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3 **Raman shifts of *c*-BN as an ideal P-T sensor for studying water-rock interactions in**
4 **a diamond anvil cell**

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12 **ABSTRACT**

13 Cubic boron nitride (*c*-BN) has the same structure as diamond. Thus, it shows very
14 inert reaction activity in different chemical environments, even under high-pressure (P)
15 and high-temperature (T) conditions. Furthermore, the P and T-dependent Raman shift of
16 *c*-BN (e.g., TO mode) can be distinguished from that of the diamond anvil (*c*-BN at ~
17 1054 cm⁻¹ v.s. diamond at ~ 1331 cm⁻¹ at ambient conditions), making *c*-BN a potential
18 P-T sensor for diamond anvil cell (DAC) experiments. However, the Raman shift of *c*-
19 BN has not been well studied at high P-T conditions, especially at temperatures above
20 700 K. In this study, we systematically calibrated the Raman shift of the TO mode (ν_{TO})
21 for synthetic *c*-BN grains at high-P and high-T conditions up to 15 GPa and 1300 K. Both

22 ruby (Mao et al. 1986) and $\text{Sm}^{2+}:\text{SrB}_4\text{O}_7$ (Datchi et al. 2007) were used as internally
23 consistent standards for calibration of *c*-BN P-T sensor. Experimental results showed that
24 Raman shift of *c*-BN is negatively correlated with temperature ($\frac{\partial \nu_{\text{TO}}}{\partial T} = -0.02206(71)$),
25 but positively correlated with pressure ($\frac{\partial \nu_{\text{TO}}}{\partial P} = 3.35(2)$). More importantly, we found
26 that the P-T cross derivative for Raman shift of *c*-BN ($\frac{\partial^2 \nu_{\text{TO}}}{\partial P \partial T} = 0.00105(7)$) cannot be
27 ignored, as it was assumed in previous studies. Finally, we calibrated a Raman shift P-T
28 sensor of *c*-BN up to 15 GPa and 1300 K as follows: $P = \frac{A(T) - \sqrt{A(T)^2 + 0.2194B(T, \Delta \nu)}}{0.1097}$, here
29 $A(T) = 3.47(6) + 0.00105(7)T$, $B(T, \Delta \nu_{\text{TO}}) = 2.81(51) - 0.0053(16)T - 1.78(11) \times 10^{-5} T^2 -$
30 $\Delta \nu_{\text{TO}}$. The *c*-BN Raman shift P-T sensor in this study fills the P-T gap ranging from
31 previously performed externally resistance-heated to laser-heated DAC experiments. The
32 effect of *c*-BN grain size and Raman system laser power on the calibration were also
33 tested for the P-T sensor. In addition, we conducted three sets of high P-T experiments to
34 test the practicability of *c*-BN P-T sensor for the water-rock interaction experiments in
35 DAC. Testing experiments showed *c*-BN has very stable chemical activity in water and
36 clear Raman signal at high P-T conditions in comparison with other P-T sensors (e.g.,
37 ruby, $\text{Sm}^{2+}:\text{SrB}_4\text{O}_7$, and quartz). In summary, we concluded that the Raman shift of *c*-BN
38 is an ideal P-T sensor for studying water-rock interaction in a DAC, especially at high-P
39 and high-T conditions relevant to subduction zones.

40 **Keywords:** Cubic boron nitride (*c*-BN); Raman shift; high P-T sensor; diamond anvil cell;
41 water-rock interactions

42 1. INTRODUCTION

43 Monitoring, recording, and reproducing the extensive thermodynamic variables such
44 as pressure and temperature is crucial for all kinds of high-P and high-T experiments,
45 particularly for in-situ diamond anvil cell (DAC) experiments (Mao et al. 1986; Datchi et
46 al. 2007). Dynamic shock compression measurements of materials could provide a
47 primary pressure scale, i.e., an equation of state (EoS) where the density is a function of
48 absolute pressure determined independently from a secondary standard. For instance, the
49 accuracy determinations of the molar volumes of metals (e.g., Au, Ag, or Cu) by X-ray
50 diffraction (XRD) can give precision to the corresponding pressure measurements (Mao
51 et al.1986; Datchi et al.1997; Fei et al., 2007; Trots et al.2013). However, to use these
52 EoS pressure scales, the shock Hugoniot data must be reduced to an isothermal EoS,
53 which can reduce the accuracy of the pressure scale (Goncharov et al.2007). On the other
54 hand, in-situ XRD patterns for EoS have to be collected from a synchrotron radiation
55 station, which dramatically limits the use of the EoS P-T sensor for DAC experiments in
56 a routine laboratory.

57 In a routine laboratory, pressure in a DAC is normally monitored by a fluorescence
58 shift P-T sensor (e.g., ruby, $\text{Sm}^{2+}:\text{SrB}_4\text{O}_7$, $\text{Sm}^{2+}:\text{YAG}$) in previous works (Barnett et al.

59 1973; Mao et al. 1986; Hess and Schiferl. 1989; Datchi et al. 1997; Trots et al. 2013).
60 Ruby, Cr³⁺-bearing α -Al₂O₃, has been widely used as the pressure sensor for DAC
61 experiments in past half a century (Barnett et al. 1973; Piermarini et al 1975; Mao et al.
62 1978; Mao et al. 1986; Datchi et al. 2007). Recently, Shen et al. (2021) re-calibrated the
63 fluorescence shift of the ruby from the well-calibrated melting line of mercury and found
64 ruby pressure sensor has a very high precision of less than ± 0.2 MPa at room
65 temperature. Sm²⁺:SrB₄O₇, strontium borate doped with divalent samarium, is proposed
66 as a better fluorescence pressure sensor to substitute ruby at high temperature (Lacam and
67 Chateau. 1989; Datchi et al. 1997; Goncharov et al. 2005; Datchi et al. 2007; Raju et al.
68 2011; Jing et al. 2013; Rashchenko et al. 2015). The sharp and narrow ⁵D₀ – ⁷F₀
69 fluorescence line (hereafter 0–0 line) of Sm²⁺:SrB₄O₇ is more intense than the ruby R1
70 line. More importantly, the 0–0 line of Sm²⁺:SrB₄O₇ presenting a significant advantage
71 over ruby is a very small temperature response (Jing et al. 2013). Previous studies
72 suggested that Sm²⁺:SrB₄O₇ is a very good pressure sensor which allowed us to consider
73 it as “the best fluorescence sensor” for in-situ high-T and high-P experiments in DAC
74 (Datchi et al. 2007; Rashchenko et al. 2015). Similarly, the fluorescence spectrum of
75 Sm²⁺:YAG was also proposed as an alternative pressure sensor. The Sm²⁺:YAG is
76 characterized by Y1 and Y2 lines in the main doublet, whose position is almost
77 insensitive to temperature (Hess and Schiferl. 1989). Besides, the Sm²⁺: YAG is the only

78 fluorescence pressure sensor that was calibrated against “absolute pressure”, which was
79 obtained by integrating the bulk modulus determined via Brillouin spectroscopy
80 concerning volumes measured simultaneously by X-ray diffraction (Trots et al. 2013).
81 Another family of compounds, $\text{Sm}^{2+}:\text{M}^{2+}\text{FCl}$ with M representing Ba^{2+} or Sr^{2+} , was also
82 considered as potential pressure sensor because their fluorescence spectra present a single
83 and sharp line with very sensitive pressure shift (1.10 nm/GPa, nearly three times as large
84 as that of the R1 line of the ruby), and relatively small temperature coefficient (Shen et al.
85 1991). Unfortunately, the quenching of the luminescence, due to the increased coupling
86 of the $5d$ and $4f$ electronic states of Sm^{2+} , limits the use of $\text{Sm}^{2+}:\text{M}^{2+}\text{FCl}$ compounds to
87 pressures below 20 GPa (Datchi et al. 1997). In most cases, the ruby and Sm^{2+} -doped P-T
88 sensor material (e.g., $\text{Sm}^{2+}:\text{SrB}_4\text{O}_7$, $\text{Sm}^{2+}:\text{YAG}$, and $\text{Sm}^{2+}:\text{M}^{2+}\text{FCl}$) have high solubility in
89 aqueous fluids at high-T and high-P conditions, which will result in contamination of
90 water-bearing experimental system in DAC (Datchi et al. 2000; Tropper and Manning
91 2007; Andreani et al. 2013). On the other hand, the easy broadening of the fluorescence
92 peaks at high temperatures significantly reduces the accuracy of pressure measurement
93 using these P-T sensors for experiments performed in the DAC (see R1 and R2 peaks for
94 ruby in Fig. S1; Ragan et al. 1992; Goncharov et al. 2005; Datchi et al. 2007; Trots et al.
95 2013) and the strong fluorescence effect of these sensors at high temperature could cover
96 the Raman signal of small samples in DAC as well. Furthermore, the applicable

97 temperature range of most fluorescence P-T sensors is less than 900 K. Until now, there is
98 still no suitable P-T sensor to bridge the P-T gap ranging from externally resistance-
99 heated DAC (< 900 K) to laser-heated DAC experiments (> 1500 K).

100 The Raman vibration of materials could effectively reflect their structure
101 information at different P-T conditions. Temperature can significantly increase the
102 molecular movement, bond length, and interaction, which can be reflected in the Raman
103 frequency shift. On the other hand, pressure also affects the Raman shift because the
104 pressure-induced reduction of atomic space will change the material structure. Therefore,
105 the Raman shift of materials such as ¹²C diamond, quartz, carbonates, or zircon could also
106 be calibrated as a P-T sensor for high P-T experiments. Raman shift of culet face of the
107 diamond anvils (¹²C diamond) has been proposed as a P-T sensor for DAC experiments
108 (Tardieu et al.1990; Datchi et al. 2007; Ono et al. 2014), however, it is only suitable for
109 high-pressure conditions (> 15 GPa) due to its limited sensitivity of determining pressure
110 in the low-pressure range (Datchi et al. 2007). Raman shifts of common minerals (e.g.,
111 carbonates, quartz, zircon) is a function of pressure and temperature, which can be also
112 calibrated as a P-T sensor (Schmidt and Ziemann. 2000; Schmidt and Ziemann. 2013;
113 Farsang et al. 2018; Cui et al. 2020; Farsang et al. 2021). For example, Schmidt and
114 Ziemann (2000) calibrated quartz Raman shift as a sensor at pressure and temperature
115 conditions up to 2.1 GPa and 800 °C. Facq et al. (2014, 2016) and Farsang et al. (2018)

116 established pressure sensors of different carbonate minerals using *P*- and *T*-induced
117 Raman shifts. Recently, Farsang et al. (2021) re-calibrated Raman shift P-T sensors for
118 most of the carbonates at pressure and temperature conditions up to 6 GPa and 500 °C.
119 Cui et al. (2020) calibrated the zircon Raman shift P-T sensor at pressure and temperature
120 up to 10 GPa and 1000 °C. There are many other minerals that can be used to determine
121 experimental pressure based on their Raman shifts, however, most common minerals
122 could easily dissolve in water and change the experimental system as well at high
123 temperature, which limits their implication as P-T sensor for water-rock interaction
124 experiments in DAC at high P-T conditions.

125 The zinc-blende structure of cubic boron nitride (*c*-BN) is a face-centered-cubic
126 lattice with space group *F*-43*m* (Ono et al. 2015), which has the same structure as
127 diamond. Thus, *c*-BN also shows many excellent physio-chemical properties, such as
128 inert chemical activity, good thermal stability, a wide bandgap, and low dielectric
129 constant (Albe et al. 1997; Datchi and Canny 2004). The *c*-BN Raman spectrum exhibits
130 two intense lines at 1054 cm⁻¹ and 1305 cm⁻¹ under ambient conditions, corresponding to
131 the Brillouin zone center transverse optical (TO) and longitudinal optical (LO) modes,
132 respectively (Datchi and Canny 2004). The Raman peak of LO mode is normally
133 overlapped with that of the diamond anvils at high P-T conditions (1305 cm⁻¹ for LO
134 mode of *c*-BN vs. 1331 cm⁻¹ for the diamond at the ambient condition). However, the P-T

135 dependent TO mode Raman shift of *c*-BN can be distinguished from that of the diamond
136 anvil at ambient conditions (e.g., 1054 cm⁻¹ for TO mode of *c*-BN vs. 1331 cm⁻¹ for
137 diamond), even at high-P and high-T conditions (Datchi and Canny 2004). Both the inert
138 chemical activity of *c*-BN and distinguishable TO mode Raman signal from the diamond
139 anvil make *c*-BN a potential P-T sensor for DAC experiments, especially for studying the
140 water-rock interaction at high-P and high-T conditions relevant to subduction zones.

141 The first *c*-BN was synthesized from hexagonal boron nitride (*h*-BN) at 6 GPa and
142 1400 °C by using magnesium metal (Mg) as a catalyst (Wentorf, R.H. 1957). Natural *c*-
143 BN has also been discovered in coesite-bearing chromium-rich ultra-mafic rocks in Tibet,
144 which was believed to have formed in the highly reduced deep mantle (Dobrzhinetskaya
145 et al. 2009). Previous studies have reported the Raman frequency shift of *c*-BN with
146 pressure up to 34 GPa at room temperature (Sanjurjo et al. 1983; Aleksandrov et al. 1989)
147 and with temperature up to 1840 K at room pressure (Alvarenga et al. 1992; Herchen and
148 Capelli. 1993), respectively. Although Datchi and Canny (2004) also studied the Raman
149 shift of *c*-BN at pressure up to 21 GPa and temperatures up to 700 K, however, studies of
150 the Raman shift of *c*-BN at simultaneous high-P and high-T conditions are still rare,
151 giving rise to larger deviation in the existing *c*-BN P-T sensor, especially at high-
152 temperature conditions (> 700 K). The lack of reliable calibration could lead to either an
153 underestimation or overestimation of pressure from *c*-BN Raman shift P-T sensor.

154 Here, we systematically studied in-situ Raman shift of *c*-BN (TO mode) at room-
155 pressure and high-temperature (1 bar, 293 - 1300 K), room-temperature and high-pressure
156 (293 K, 1 bar - 46 GPa), and high-temperature and high-pressure (293 - 1100 K, 1 bar -
157 15 GPa), respectively. We established a *c*-BN Raman shift P-T sensor, which can be used
158 to accurately determine the pressure and/or temperature in the DAC at least up to 15 GPa
159 and 1300 K. Furthermore, we conducted three sets of high P-T experiments to test the
160 practicability of *c*-BN sensor in the water-rock interaction experiments up to 5 GPa and
161 1023 K.

162 **2. EXPERIMENTAL METHODS**

163 The pure synthetic *c*-BN powders were ordered from Zhengzhou Zhongnan JETE
164 Superrabrasives Co., LTD, Henan, China, whose shock compression behavior has been
165 studied by Hu et al. (2018). Both the ruby sphere (BETSA company, France) and
166 Sm²⁺:SrB₄O₇ powders (synthesized by Zhao et al. 2016) were used as internal standards
167 to calibrate the Raman shift P-T sensor of *c*-BN.

168 **2.1 In-situ Raman measurements**

169 In-situ Raman spectra were collected from a diamond window of the DAC or a
170 sapphire window of the heating stage under the Renishaw Raman system (RM1000) at
171 HPSTAR. The 532 nm line of Nd:YAG laser was used as the excitation source. The
172 Raman signal of the Si metal line at 520 cm⁻¹ was used to calibrate the Raman system

173 before and after each measurement. By comparing the Raman shift on the same *c*-BN
174 grain under different laser power, we found that *c*-BN grains could be heated due to the
175 absorption of laser light, which may result in a significant Raman shift. The Raman shift
176 was systematically analyzed for *c*-BN grains with different sizes (8 - 90 μm) using
177 different laser power (4 - 20 mW). Finally, the laser power of the Raman system below 5
178 mW on *c*-BN grains was used for this calibration and later water-rock interaction
179 experiments. The measurements at each P-T condition were performed after equilibrating
180 the system for 5-10 minutes to obtain a steady-state Raman frequency.

181 **2.2 Room-P and high-T experiments**

182 A heating stage (HRTS1000; Shanghai Photographic Instrument Co., Ltd) with a
183 sapphire window was used to heat samples at ambient pressure. The thermocouple is
184 buried under the center of the reaction chamber of the heating stage. To obtain an
185 accurate experimental temperature, we used several metals (e.g., 63A solder, tin, zinc,
186 and silver) with well-known melting points (183 $^{\circ}\text{C}$, 232 $^{\circ}\text{C}$, 420 $^{\circ}\text{C}$, and 961 $^{\circ}\text{C}$) to
187 calibrate the real temperature of the heating stage (Fig. S2a) before calibration
188 experiments (Yang et al. 1993). The temperature of the heating stage (actual temperature)
189 on the sample position in the reaction chamber was calibrated from room T to 1000 $^{\circ}\text{C}$.
190 Particles of 0.2 - 0.5 mm^2 of each metal material were pre-indented into thin foils with a
191 diameter of 0.5 mm to 1 mm (Fig. S2b), and then loaded into the center of the heating

192 stage reaction chamber. External cooling water at 20 °C was used to protect the whole
193 heating state body from high temperature. A protective hydrogen-argon mixture gas (98 %
194 Ar + 2 % H₂) was slowly blown into the reaction chamber to avoid any oxidation of the
195 metal samples by air. The heating rate was set to be 5 °C / min during the heating process.
196 When the instantaneous shrinkage of a metal foil was observed under the microscope, a
197 real temperature was recorded. In most cases, the shrinkage time of the metal foils was
198 less than 2 s and the temperature interval was less than 0.1 °C (Fig. S2b). The calibration
199 processes were repeated several times to ensure that the calibration deviation was less
200 than 1 °C. Based on the temperature difference between the controller of the heating stage
201 and the real temperature (i.e., the known melting temperature of the given metal), we
202 corrected the controller temperature by linear fitting (Fig. S2c).

203

204 **2.3 Room-T and high-P & high-T and high-P experiments**

205 Both room-T and high-P experiments & high-P and high-T experiments were carried
206 out in a high-temperature BX-90 DAC equipped with two IIa diamond anvils. The culet
207 diameter of diamond anvils was 500 μm or 250 μm, depending on the target pressure (i.e.,
208 500 μm for experimental pressure below 16 GPa, 250 μm for the experimental pressure
209 from 16 GPa to 46 GPa). For the room-T and high-P experiment, high-pressure (0.16 GPa)
210 pure helium gas was filled into the reaction chamber as pressure media using a high-

211 pressure gas loader at HPSTAR. Several *c*-BN crystals, ruby spheres, and Sm²⁺:SrB₄O₇
212 powders were loaded together into a 100 μm hole in the rhenium gasket (Fig. S3). For
213 high-T and high-P experiments, the same pressure transmitting medium as room-T and
214 high-P experiment has been used, an external resistance heater of Ni-Cr wire was placed
215 around diamond anvils for heating samples (Figs. S3a, S3b), and the temperature was
216 measured with a K-type thermal couple mounted to one of the diamond anvils by high-
217 temperature cement (Fig. S3c). The temperature deviation in the DAC reaction chamber
218 was checked by the melting points of tin (232 °C) within 2 °C. The experimental pressure
219 was calculated by fluorescence shift of the ruby sphere (Mao et al. 1986) and/or the
220 Sm²⁺:SrB₄O₇ powders (Datchi et al. 2007) as below:

$$221 \quad P_{R1} = 248.4 \left[\left(\frac{\Delta\lambda}{\lambda_0} \right)^{7.665} - 1 \right] \text{ (Mao et al. 1986)} \quad (1)$$

$$222 \quad P_{0-0} = 3.9096 \Delta\lambda \frac{1+0.03378\Delta\lambda}{1+0.0105\Delta\lambda} \text{ (Datchi et al. 2007)} \quad (2)$$

223 Where P is in GPa and λ is in nm. λ is the fluorescence shift of the ruby (R1) or
224 Sm²⁺:SrB₄O₇ (0–0) Lines, and λ₀ is the initial wavelength at ambient conditions (293 K, 1
225 atm). Here, λ₀ = 693.973 (13) nm for ruby, λ₀₋₀ = 685.125 (9) nm for Sm²⁺:SrB₄O₇, and
226 Δλ = λ (P, T) – λ₀. Previous studies usually ignored the temperature effect for the
227 Sm²⁺:SrB₄O₇ P-T sensor because of its minimal effect on the 0–0 line shift (Lacam and
228 Chateat 1989; Leger et al. 1990; Raju et al. 2011; Fig. S4a and S4b). Here, we re-fitted
229 the temperature dependence of fluorescence shift of the Sm²⁺:SrB₄O₇ at high

230 temperatures (Fig. S5). In order to obtain a more accurate pressure value than before, we
231 also deducted the fluorescence shift caused by temperature when calculating the pressure
232 by the $\text{Sm}^{2+}:\text{SrB}_4\text{O}_7$ sensor. In Fig. S5, we compared the results with linear fitting from
233 Lacam et al. (1989), Raju et al. (2011), and Jing et al. (2013) where they give a slope of
234 -1×10^{-4} , 2×10^{-5} and -2×10^{-4} nm / K respectively. Datchi et al. (2007) used a
235 nonlinear equation of $\Delta\lambda_{0-0} = 8.7(12) \times 10^{-5} \Delta T + 4.62(60) \times 10^{-6} \Delta T^2 - 2.38(70) \times 10^{-9} \Delta T^3$.
236 Our experimental result is more consistent with that from Datchi et al. (2007).
237 Therefore, we used the $\text{Sm}^{2+}:\text{SrB}_4\text{O}_7$ pressure scale formula reported by Datchi et al.
238 (2007) to improve the pressure calibration of the TO mode of *c*-BN in this study. The big
239 difference between our study and other studies (Lacam et al. 1989; Raju et al. 2011; Jing
240 et al. (2013) may result from different laser power used for the Raman system (see below
241 for more detailed discussions). In this study, the pressure deviations given by two
242 independent standards (ruby and $\text{Sm}^{2+}:\text{SrB}_4\text{O}_7$) are consistently kept within $\pm 5\%$,
243 indicating that the calibrated *c*-BN P-T sensor in this study is internally consistent with
244 ruby and $\text{Sm}^{2+}:\text{SrB}_4\text{O}_7$ in our experimental P-T ranges.

245 In total, 108 measurements at 54 P-T conditions were collected from ambient
246 conditions to high-T and high-P conditions up to 1300 K and 46 GPa for *c*-BN P-T sensor
247 calibration in this study (Table S1). To test the practicability of *c*-BN in the water-rock
248 interaction experiment, we carried out three sets of DAC experiments to characterize the

249 changes of different pressure sensors (e.g., *c*-BN, Sm²⁺:SrB₄O₇, and ruby) in the water-
250 quartz system at high-P and high-T conditions up to 5 GPa and 1023 K. The Raman
251 peaks of *c*-BN at different P-T conditions were fitted with the symmetric Gauss + Lorentz
252 function after subtracting a linear baseline. All fitting parameters (e.g., peak positions) are
253 summarized in Table S1.

254 **3. RESULTS AND DISCUSSION**

255 **3.1 Temperature-dependent Raman shift of *c*-BN**

256 We measured *c*-BN Raman spectra from room T to 1223 K with 50 K intervals at
257 room pressure (Fig. 1a) and obtained a relationship between the Raman shift and
258 temperature (Fig. 1b). The *c*-BN LO mode gradually weakens and disappears with an
259 increase of temperature to ~ 1100 K. However, the TO mode peak of *c*-BN remains in
260 good shape in all high-temperature experiments (Fig. 1a), and is used to fit the correlation
261 between the Raman shift and temperature by the polynomial equation as below:

$$262 \nu_{\text{TO}}(P = 0, T) = 1054.58(16) - 0.02206(71) T - 1.1000(70) \times 10^{-5} T^2 \quad (R^2=0.999)$$

263 (3)

264 As shown in Fig. 1b, our fitted curve is well included in the previous study and R² is
265 0.999. The temperature dependences of Raman shift of *c*-BN at ambient pressure have
266 been investigated in previous studies (Alvarenga et al., 1992; Herchen and Capelli, 1993;
267 Datchi and Canny, 2004). A comparison for the temperature-induced Raman shift of *c*-

268 BN is shown in Table 1. In comparison with previous studies, our results are in good
269 agreement with Datchi and Canny (2004) in the low-temperature range (< 600 K) (Fig.
270 1b), however, there is a slight deviation above 600 K due to limited data (less than 723 K)
271 from Datchi and Canny (2004). The *c*-BN Raman shift from earlier works (e.g.,
272 Alvarenga et al. 1992 and Herchen and Cappelli 1993) are systematically deviating from
273 our data at the same temperature conditions, which may be due to an overall shift of the
274 early Raman spectrometer or different laser power used for a different measurement. The
275 heating equipment used by Alvarenga et al. (1992) and Herchen and Capelli (1993) were
276 not systematically calibrated before use at high temperature (see calibration in section
277 2.2), which may also result in significant uncertainty in the experimental temperature.
278 Alvarenga et al. (1992) obtained a linear Raman shift of *c*-BN with temperature with a
279 slope at $-0.038(2)$ cm^{-1}/K from 300 to 1600 K, whereas Herchen and Capelli (1993) found
280 a nonlinear behavior from 300 to 1840 K. However, we re-fitted the data of Alvarenga et
281 al. (1993) and found that the TO frequency and temperature can be better expressed as a
282 non-linear relationship in Fig. 1b. The goodness of fit (R^2) is 0.951 for the linear
283 expression and 0.969 for the polynomial expression indicating that Alvarenga et al. (1993)
284 may have used the improper function to fit the relationship between the TO frequency
285 and temperature. On the other hand, the different initial Raman shift value and irregular
286 evolution of *c*-BN Raman shift with temperature in early works (e.g., Alvarenga et al.

287 1992; Herchen et al. 1993) may result from excessive laser heating (see discussion in
288 section 3.4).

289

290

291 **3.2 Pressure-dependent Raman shift of *c*-BN**

292 The systematical Raman shift of *c*-BN with increasing pressure at room temperature
293 is shown in Fig. 2a. The correlation between the Raman shift and pressure from 1 bar to ~
294 46 GPa are fitted as a polynomial in Fig. 2b in comparison with that from previous
295 studies (Sanjurjo et al., 1983; Datchi and Canny; 2004; Ono et al., 2015). Sanjurjo et al.
296 (1983) studied the pressure dependence of *c*-BN Raman shift at room temperature up to
297 8.2 GPa using a 4:1 methanol-ethanol mixture as a pressure medium in a DAC, and a
298 linear function with a slope at $3.39(8) \text{ cm}^{-1}/\text{GPa}$ was chosen to fit experimental data.
299 Datchi and Canny (2004) studied the pressure-dependent Raman shift of *c*-BN up to 21
300 GPa in a helium gas pressure medium in a DAC and found a linear migration rate of
301 $3.27(2) \text{ cm}^{-1}/\text{GPa}$ below 10 GPa, but a nonlinear relationship at higher pressure. In
302 addition, Ono et al. (2015) extended experimental pressure up to 90 GPa using NaCl as a
303 pressure medium, with a dv/dP slope at $3.41(7)$ and $2.04(7) \text{ cm}^{-1}/\text{GPa}$ at 1 bar and 90 GPa,
304 respectively.

305 In this study, *c*-BN Raman shift can be well linearly fitted with a variation of 3.35 (3)
306 $\text{cm}^{-1}/\text{GPa}$ below 10 GPa, which is pretty consistent with that from previous studies
307 (Sanjurjo et al. 1983; Datchi and Canny 2004; Ono et al. 2015). When the pressure is
308 below 25 GPa, the difference in pressure between linear and polynomial functions is
309 relatively small. However, if the linear fitting of Sanjurjo et al. (1983) is extrapolated to
310 high pressure (> 25 GPa), the difference between Sanjurjo et al. (1983) and our study
311 increases rapidly as the pressure increases. This is the fatal uncertainty of pressure
312 calibration. Our experimental results are consistent with Ono et al. (2015) as well, the
313 slight difference above 25 GPa may attribute to non-hydrostatic NaCl pressure media
314 used by Ono et al. (2015), which may undergo a phase transformation from B1- to B2-
315 type at 25 GPa. The correlation between the Raman shift and pressure is clearly non-
316 linear, which can be well reproduced as below (Fig. 2b):

$$317 \nu_{\text{TO}}(P, T = 293\text{K}) = 1055.37(18) + 3.35(2)P - 0.011(1)P^2 \quad (R^2=0.999) \quad (4)$$

318 **3.3 Coupled effect of P-T on *c*-BN Raman shift**

319 To test the coupled effect of pressure and temperature on the Raman shift of *c*-BN in
320 this study, we carried out seven sets of high-temperature and high-pressure experiments
321 in the DAC (Fig. S6). During the heating process, both the volume expansion of the
322 reaction chamber in the DAC and hole deformation in the Re gasket affected the pressure
323 changes on the samples at high-temperature and high-pressure conditions.

324 Combining all the data we obtained from room pressure and high temperature, room
325 temperature and high pressure (here pressure less than 15 GPa was selected), and high
326 temperature and pressure experiments, we fitted the curved surface of Raman shift (ν_{TO})
327 using a best polynomial expression (Fig. 3) as the following equation:

$$\begin{aligned} 328 \quad \Delta\nu_{\text{TO}}(P, T) = & 2.81(51) - 0.0053(16)T - 1.78 \times 10^{-5}(11)T^2 + 3.47(6)P - \\ 329 \quad & 0.055(6)P^2 + 0.00105(7)P \cdot T \quad (R^2=0.998) \end{aligned} \quad (5)$$

330 $\Delta\nu_{\text{TO}}(P, T)$ is the difference of Raman shift between high-P and/or high-T conditions of
331 interest [$\nu_{\text{TO}}(P, T)$] and the reference at ambient conditions [$\nu_0(P_0, T_0)$]. Eq. 5 shows that
332 the Raman shift of *c*-BN is a coupled function of pressure and temperature, rather than a
333 simple linear superposition of pressure and temperature, indicating a significant P-T cross
334 derivative term on *c*-BN sensor.

335 The fitting surface can be used to determine the pressure from the Raman shift of *c*-
336 BN at a given temperature if Eq. 5 is converted to Eq. 6 as below:

$$337 \quad P = \frac{A(T) - \sqrt{A(T)^2 + 0.2194B(T, \Delta\nu)}}{0.1097} \quad (6)$$

$$\begin{aligned} 338 \quad A(T) = & 3.47(6) + 0.00105(7)T, B(T, \Delta\nu) = 2.81(51) - 0.0053(16)T - \\ 339 \quad & 1.78(11) \times 10^{-5} T^2 - \Delta\nu_{\text{TO}}. \end{aligned}$$

340 We compared pressure determined from the $\text{Sm}^{2+}:\text{SrB}_4\text{O}_7$ by Eq. 2 and from *c*-BN
341 by Eq. 6 at high temperature in Table S1. It shows that almost all deviations between
342 pressure determined from the $\text{Sm}^{2+}:\text{SrB}_4\text{O}_7$ by Eq. 2 and *c*-BN by Eq. 7 are within 3%.

343 Only several points at low pressure are above 10% because of the relatively small
344 denominator (the pressure determined from the $\text{Sm}^{2+}:\text{SrB}_4\text{O}_7$ by Eq. 2), indicating that an
345 accurate pressure at high-temperature can be determined by using Raman shift P-T sensor
346 of *c*-BN in this study. Datchi and Canny (2004) also calibrated the temperature- and
347 pressure-dependent Raman shift of *c*-BN up to 700 K and 21 GPa. They fitted all
348 experimental data over the whole P-T range as:

$$349 \quad P = -43.4783\{A(T) + \sqrt{A(T) - 0.046[v_{\text{TO}}(P, T) - B(T)]}\} \quad (7)$$

$$350 \quad A(T) = 3.325 (11) - 0.00022 T, B(T, \Delta\nu) = 1058.4 (2) - 9.6 (7) \times 10^{-3} T -$$
$$351 \quad 1.54 (7) \times 10^{-5} T^2.$$

352 We compared the pressure derivation $\Delta P(\%) = (P_1 - P_2)/P_1 * 100\%$ between Eq. 6
353 from this study and Eq. 7 from Datchi and Canny (2004) at different experimental
354 temperatures and pressure (Fig. 4). As Fig. 4a showed, the pressure derivation between
355 Eq. 6 in this study and Eq. 7 is less than 15% in our experimental pressure range.
356 However, Fig. 4b showed that with an increase of experimental temperature, the pressure
357 given by Eq. 7 from Datchi and Canny (2004) seems to overestimate pressure because the
358 cross-derivative term of both the pressure and temperature is underestimated in their
359 study. The coupling effect coefficient ($\frac{\partial^2 v_{\text{TO}}}{\partial P \partial T} = 0.000222$) from Eq. 7 (Datchi and Canny
360 2004) is smaller than Eq. 6 ($\frac{\partial^2 v_{\text{TO}}}{\partial P \partial T} = 0.00105(7)$) in this study. Therefore, pressure
361 deviation cannot be ignored for the Raman shift P-T sensor of *c*-BN, especially at high-

362 temperature conditions. Further, we also extended the Raman shift of *c*-BN in a higher
363 temperature range from 700 K in Datchi and Canny (2004) to 1300 K in this study. The
364 Raman shift P-T sensor of *c*-BN in this study could fill the gap of P-T sensors ranging
365 from externally resistance-heated to laser-heated DAC experiments.

366

367

368 **3.4 Effect of laser power and grain size on Raman shift of *c*-BN**

369 During calibration experiments, we found that *c*-BN grains can be heated due to the
370 absorption of laser light of the Raman system, which may result in a significant Raman
371 shift of *c*-BN. The Raman shift of the TO mode for *c*-BN may change from 1054 cm⁻¹ to
372 1045 cm⁻¹ with laser power increasing from 1 to 20 mw at room pressure (Fig. 5).
373 Therefore, it is necessary to control the Raman shift caused by laser heating. Thus, we
374 recommend using laser power as low as possible to avoid deviations caused by heat
375 absorption when using *c*-BN to be a P-T sensor. On the other hand, Werninghaus et al.
376 (1997) also reported that grain size effects, such as the presence of nanocrystalline *c*-BN
377 or amorphous BN, will cause a significant Raman shift. In order to explore the influence
378 of grain size on *c*-BN Raman shift, we also compared Raman shift of *c*-BN with grain
379 sizes of 8 - 12 μm, 38 - 45 μm, 53 - 63 μm, and 75 - 90 μm using different laser power,
380 respectively (Fig. 5). We linearly fitted the correlation between the Raman shift of *c*-BN

381 and the laser power for different grain sizes (Fig. 5). In summary, we found that there is a
382 systematical non-negligible Raman shift of *c*-BN with different sizes under the same laser
383 power (Fig. 5). The larger the grain size of *c*-BN is, the less thermal effect was observed.
384 Furthermore, the effect of laser power on the Raman shift of *c*-BN can be linearly fitted
385 with a negative slope, which is the same as the temperature effect we discussed before
386 (section 3.1).

387 In summary, the grain size of *c*-BN should be as large as possible for better heat
388 dissipation, while the laser power should be as small as possible (< 5 mW) to avoid
389 heating of *c*-BN grains, if we choose *c*-BN as the P-T sensor for DAC experiments.

390

391 **4. Application of *c*-BN P-T sensor to water-rock interaction experiments**

392 In the first testing experiment, two *c*-BN crystals and a piece of quartz crystal were
393 loaded in the sample chamber with water as the pressure medium (Fig. 6). Experimental
394 pressure is determined by the Raman shift of *c*-BN using Eq. 6. We also calculate the
395 pressure in the sample chamber by using Raman shift P-T sensor of quartz from Schmidt
396 et al. (2000) at the low-pressure range (< 2 GPa). The pressure calculated from *c*-BN
397 (1.83 GPa) is in good agreement with that from the quartz P-T sensor (1.68 GPa), which
398 further indicates the reliability of *c*-BN Raman shift P-T sensor in this study. During
399 heating, we observed the quartz crystal become smaller and the Raman intensity of the

400 quartz gradually weakened until completely disappeared at the temperature of 1023 K,
401 indicating quartz dissolves in water with the increase of temperature. However, the
402 Raman intensity of *c*-BN did not get weak with an increase of temperature or pressure in
403 our experimental temperature and pressure range. The *c*-BN crystals maintain a good
404 crystal form before and after high-temperature and high-pressure experiments, indicating
405 that no dissolution of *c*-BN in water occurs at high P-T conditions. Therefore, we have
406 reason to believe that *c*-BN has very stable properties in water and can accurately
407 determine the pressure in DAC at high-temperature and high-pressure conditions up to
408 1023 K and 5.00 GPa or higher.

409 In the second testing experiment, a piece of quartz crystal and some Sm²⁺:SrB₄O₇
410 powders were placed in the sample chamber with water (Fig. 7). The pressure is
411 determined by the fluorescence shift of Sm²⁺:SrB₄O₇ using Eq. 2. The pressure calculated
412 from Sm²⁺:SrB₄O₇ (1.32 GPa) is also in good agreement with that from quartz (1.41 GPa)
413 by Schmidt et al. 2000. With the increase of temperature, both Sm²⁺:SrB₄O₇ and quartz
414 showed an obvious dissolution phenomenon, and Raman intensities of quartz and
415 Sm²⁺:SrB₄O₇ gradually weakened. The temperature for the complete dissolution of quartz
416 is between 873 K and 1023 K, and the temperature when Sm²⁺:SrB₄O₇ is completely
417 dissolved is between 673 K and 1023 K. The reaction or dissolution of quartz or

418 $\text{Sm}^{2+}:\text{SrB}_4\text{O}_7$ in water at high temperatures makes it hard to obtain a Raman signal, thus
419 cannot be used as pressure sensors at these P-T conditions.

420 In the third testing experiment, a piece of quartz crystal and a ruby sphere were
421 loaded in the sample chamber with water (Fig. 8). The pressure calculated from ruby
422 using Eq. 1 (1.44 GPa) is also in good agreement with that from the quartz P-T sensor
423 (1.41 GPa) by Schmidt et al. (2000). With the increase of temperature, the quartz
424 completely dissolved in water at 673 K. The linewidth of the fluorescence lines of ruby
425 rapidly increases with temperature, and the doublet R1–R2 becomes a broad, unresolved
426 band above ~573 K (Fig.S1, Fig. 8), which makes it impossible to obtain the real pressure.
427 Further, the stronger fluorescence of ruby at high temperature could cover the Raman
428 signal of the fluid in the sample chamber of DACs as well. Therefore, ruby is also not a
429 suitable pressure sensor for most of the water-rock interaction experiments in DAC at
430 high temperatures.

431 In summary, we found that *c*-BN Raman shift P-T sensor has a big advantage in the
432 determination of pressure for water-rock interaction experiments in DAC. Raman shift of
433 *c*-BN is an ideal P-T sensor for water-rock reaction experiments at high-temperature and
434 high-pressure conditions relevant to subduction zones.

435 **5. CONCLUSION AND IMPLICATION**

436 We have systematically studied the Raman shift of *c*-BN up to 46 GPa and 1300 K.
437 To our knowledge, this is the first report of Raman shift of *c*-BN at simultaneous high-
438 pressure and high-temperature (>700 K) conditions. Our experimental results showed that
439 Raman shift of *c*-BN is a coupled function of pressure and temperature (i.e., $\frac{\partial^2 \nu_{\text{TO}}}{\partial P \partial T} =$
440 0.00105(7)), which is ignored in previous studies. Further, we tested and discussed the
441 influence of grain size and laser power on Raman peak shift of *c*-BN and found that the
442 grain size should be as large as possible and the laser power should be as small as
443 possible when using *c*-BN as a P-T sensor in DAC. The testing experiments of water-rock
444 reaction in this study showed *c*-BN P-T sensor can successfully determine the accurate
445 pressure in water-rock reaction experiments in DAC at high temperature and high
446 pressure up to 1023 K and 5.00 GPa or higher. Finally, combining the inert chemical
447 activity of *c*-BN and its distinguishable Raman signal from the diamond anvil at high-P
448 and high-T conditions, we propose that *c*-BN will be an ideal P-T sensor for studying
449 water-rock interactions in a DAC at high-pressure and high-temperature conditions
450 relevant to subduction zones. The *c*-BN P-T sensor will be very useful for experimenters
451 who want to investigate fluid properties in high-pressure metamorphic rocks (e.g.,
452 eclogite or granulite) in subduction zones.

453

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617

618 **FIGURES CAPTIONS**

619 **Figure 1:** (a) Evolution of the TO and LO Raman mode of *c*-BN with changing
620 temperatures at room pressure. (b) Raman shifts of the TO mode of *c*-BN as a function of
621 temperature at room pressure in this study in comparison with that from previous studies
622 (Datchi and Canny (2004); Alvarenga et al. (1992); Herchen and Capelli (1993)). The red
623 dotted line, blue dotted line and black dotted line are the data fitting results of this study,
624 Datchi and Canny (2004) and Herchen and Capelli (1993), respectively. The green
625 straight line and the green dotted dashed line are linear fitting and polynomial fitting of
626 the results of Alvarenga et al. 1992, respectively. Error bars are smaller than symbols.

627 **Figure 2:** (a) Evolution of the TO mode Raman spectra of *c*-BN at different pressures at
628 room temperature. (b) Raman shifts of the TO mode of *c*-BN as a function of pressure at
629 room temperature in this study in comparison with that from previous studies (Datchi and
630 Canny 2004; Ono et al. 2015). The red dotted line, black dotted line, blue dotted line, and
631 green dotted line is the data fitting results of this study, Datchi and Canny 2004, Ono et al.
632 2015, and Sanjurjo et al. 1983, respectively. Error bars are smaller than symbols.

633 **Figure 3:** 3D-surface fitting of Raman shifts of *c*-BN as a function of temperature (K)
634 and pressure (GPa).

635 **Figure 4:** (a) Pressure difference $\Delta P(\%)$ changes with increasing experimental pressure
636 (P). (b) Pressure difference $\Delta P(\%)$ changes with increasing experimental temperature (T).
637 $\Delta P(\%) = (P_1 - P_2) / P_1 * 100\%$, P_1 is the pressure determined by Eq. 6, P_2 is the pressure
638 determined by Eq. 7.

639 **Figure 5:** Raman shift of the TO mode of *c*-BN linearly changes with different grain
640 sizes as a function of laser power.

641 **Figure 6:** The real-time in-situ photomicrographs and the corresponding Raman spectra
642 of *c*-BN (TO mode) and quartz in DAC at 1.83 GPa and 293 K, 5.00 GPa and 573 K, 0.48
643 GPa and 873 K, 0.65 GPa and 1023 K, respectively. The pressure and temperature are
644 determined by Eq. 6 and the thermocouple, respectively.

645 **Figure 7:** The real-time in-situ photomicrographs and the corresponding fluorescent
646 spectra of $\text{Sm}^{2+}:\text{SrB}_4\text{O}_7$ and quartz in DAC at 1.32 GPa and 293 K, 4.58 GPa and 573 K,
647 773 K, 1023 K, respectively. The pressure and temperature are determined by Eq. 2 and
648 the thermocouple, respectively. The latter two sets of experiments could not determine
649 the pressure due to $\text{Sm}^{2+}:\text{SrB}_4\text{O}_7$ and quartz dissolution or reaction with water.

650 **Figure 8:** The real-time in-situ photomicrographs of ruby and quartz in DAC at 1.44 GPa
651 and 293 K, 573 K, 873 K, 1023 K, respectively, the Raman spectra of quartz at 1.44 GPa

652 and 293 K and the fluorescent spectra of ruby at 1.44 GPa and 293 K, 573 K, 873 K, 1023
653 K, respectively. The experimental pressure and temperature determined by the Eq. 1 and
654 the type K thermocouple, respectively.

655

656 **Supplementary Figure Captions**

657

658 **Figure S1:** Representative fluorescence spectra (R doublet) of a ruby sphere at 1 atm and
659 different temperatures.

660 **Figure S2:** Heating stage in room-P and high-T experiment (a) The HRTS1000 heating
661 stage under the Raman system microscope. (b) Temperature calibration process using the
662 melting points of four metals (63A solder, tin, zinc, and silver). (c) Temperature
663 correction curve from the reading temperature to the actual temperature.

664 **Figure S3:** High-temperature diamond anvil cell (BX-90) used for all high-P experiments.
665 (a) Photograph of the high-temperature BX-90 DAC. (b) Section view of BX90 DAC of
666 which (1) outer cylinder part, (2) inner piston part, (3) screws for generating loading force,
667 (4) pack of conical spring washers, (5) setscrews for diamond anvils alignment, (6) safety
668 setscrews, (7) optional miniature Ni-Cr resistance heater, (8) K-type thermal couple, (9)
669 rhenium gasket, and (10) diamond anvils (modified from Kantor et al. 2012). (c) Detailed
670 drawing of the sample chamber and diamond anvils: the ruby sphere, $\text{Sm}^{2+}:\text{SrB}_4\text{O}_7$

671 powder and *c*-BN are placed into the sample chamber of the rhenium gasket at the same
672 time and it is filled with pure helium gas as the pressure media.

673 **Figure S4:** (a) Fluorescent spectra of the 0–0 line of Sm²⁺:SrB₄O₇ under various
674 temperatures at room pressure. (b) The fitting process of the fluorescent spectra of the 0–
675 0 line of Sm²⁺:SrB₄O₇ at high pressure and high temperature.

676 **Figure S5:** Wavelength shift of the 0–0 line of Sm²⁺:SrB₄O₇ as a function of temperature
677 at room pressure. Error bars smaller than symbols are not shown for clarity.

678 **Figure S6:** Pressure-temperature paths of the diamond anvil experiments for each run.

679

680 TABLES CAPTIONS

681 **Table 1:** High P-T induced Raman shift of *c*-BN in comparison with that from previous
682 studies.

683 **Table S1:** All results of the high-P and/or high-T experiments in this study.

Table 1. High P-T induced Raman shift of *c*-BN in comparison with that from previous studies

ν_0	T	dv/dT	d^2v/d^2T	P	dv/dP	d^2v/d^2P	$d^2v/(dP \cdot dT)$	Reference
1053.6 (2)	293-1223	-0.0221(7)	$1.100(7) \times 10^{-5}$	0-46	3.35(2)	-0.011(1)	0.00105(7)	This study
1054	300-723			0-21	3.33	-0.018	0.00022	Datchi and Canny 2004
	300-1600	-0.038						Alvarenga et al. 1992
				0-8.2	3.39			Sanjurjo et al. 1983
				0-83.9	3.35	-0.013		Ono et al. 2015
	300-1840	-0.01	2.8×10^{-5}					Herchen et al. 1993

Notes: the values of ν_0 were measured at ambient pressure (1 bar) and temperature (20 °C) in cm^{-1} . T is the temperature in K; P is the pressure in GPa.

Fig. 1

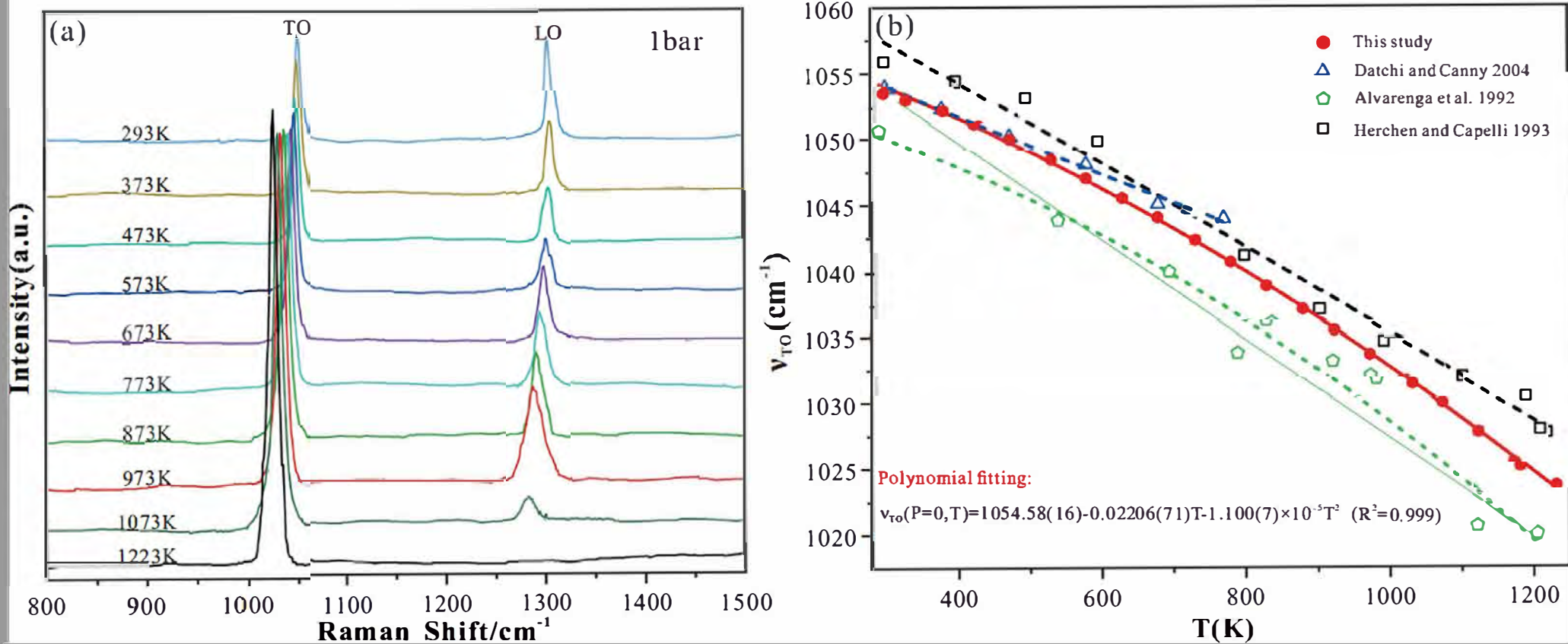


Fig. 2

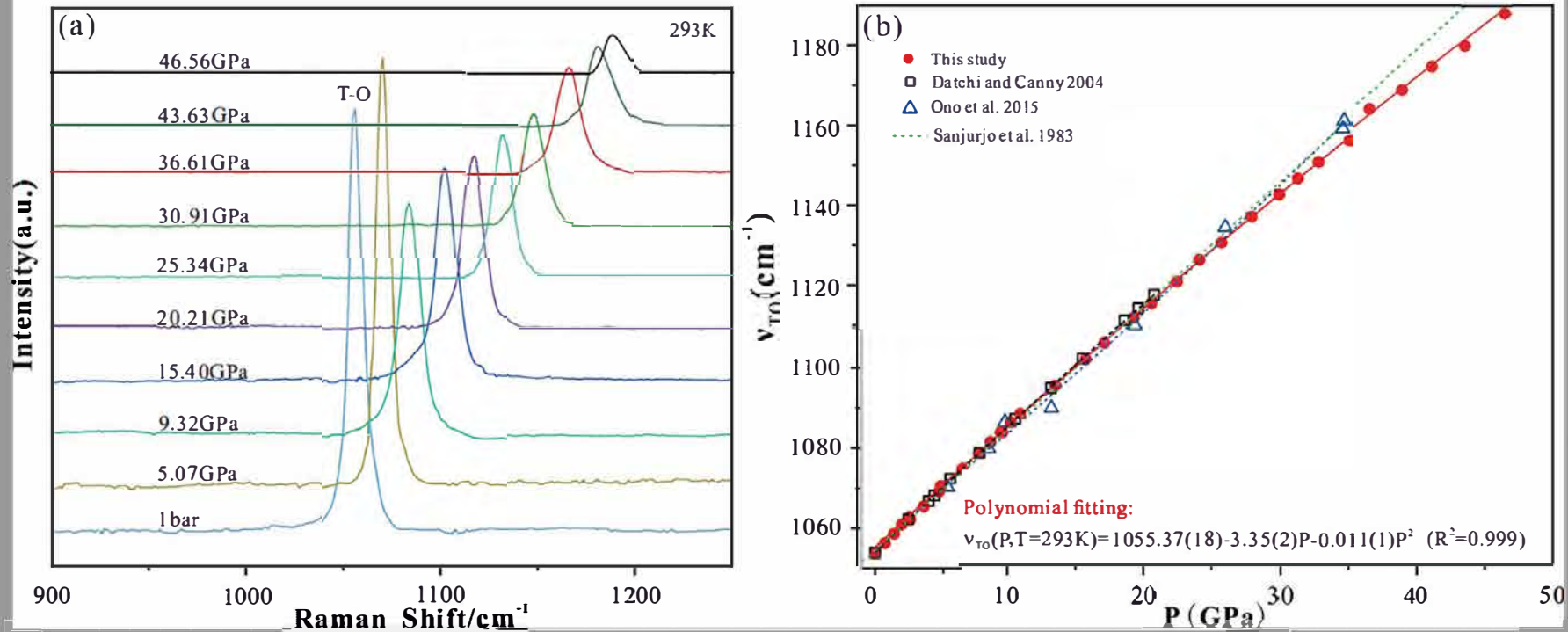
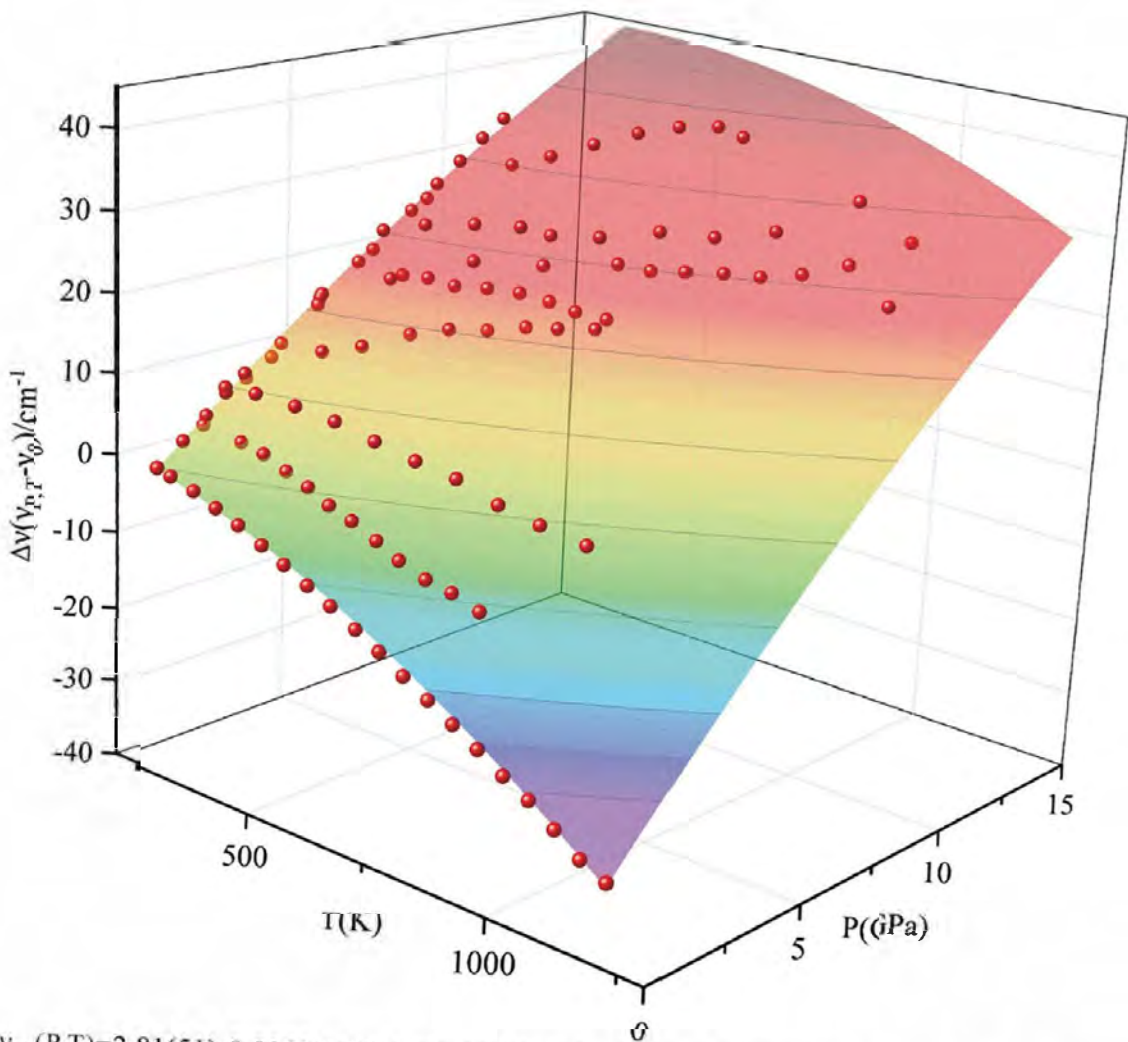


Fig. 5



$$\Delta v_{\text{TO}}(P,T) = 2.81(51) - 0.0053(16)T - 1.78 \times 10^{-5}(11)T^2 + 3.47(6)P - 0.055(6)P^2 + 0.00105(7)P \cdot T \quad (R^2 = 0.998)$$

Fig. 4

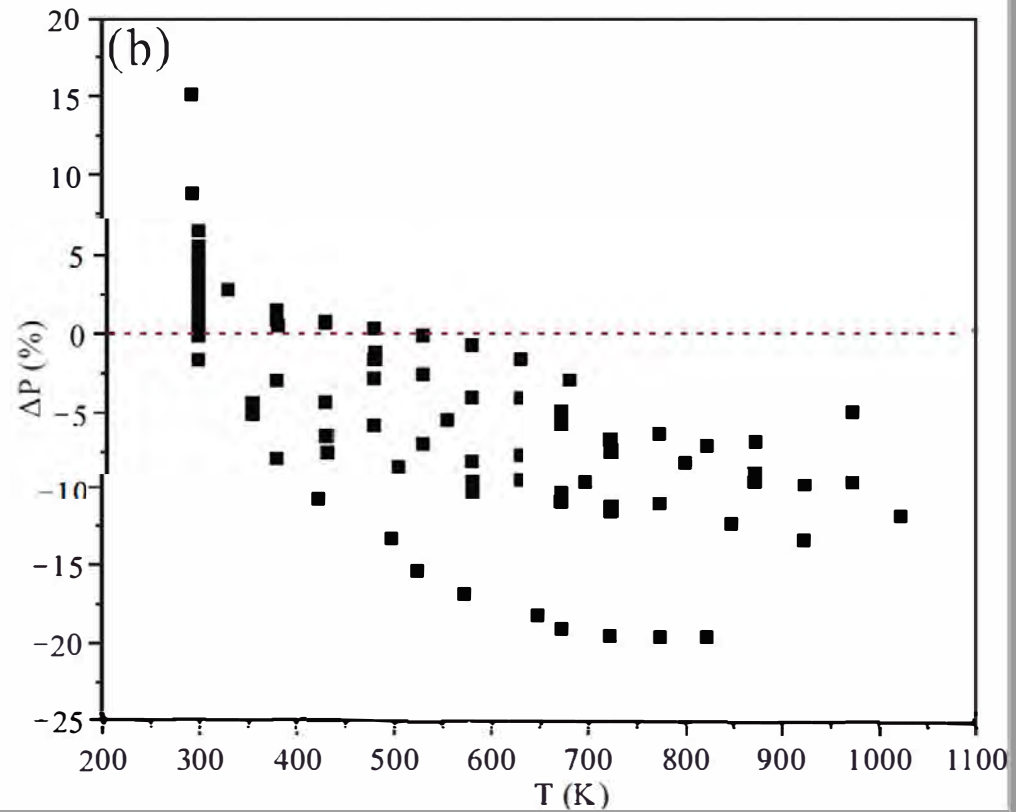
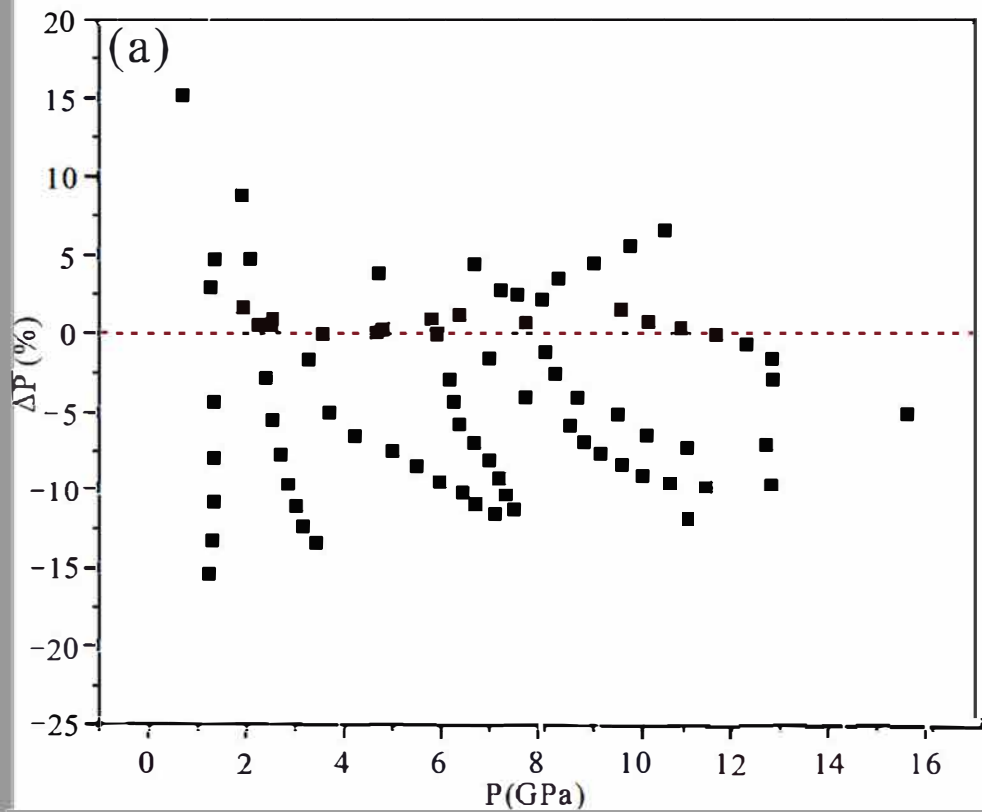


Fig. 5

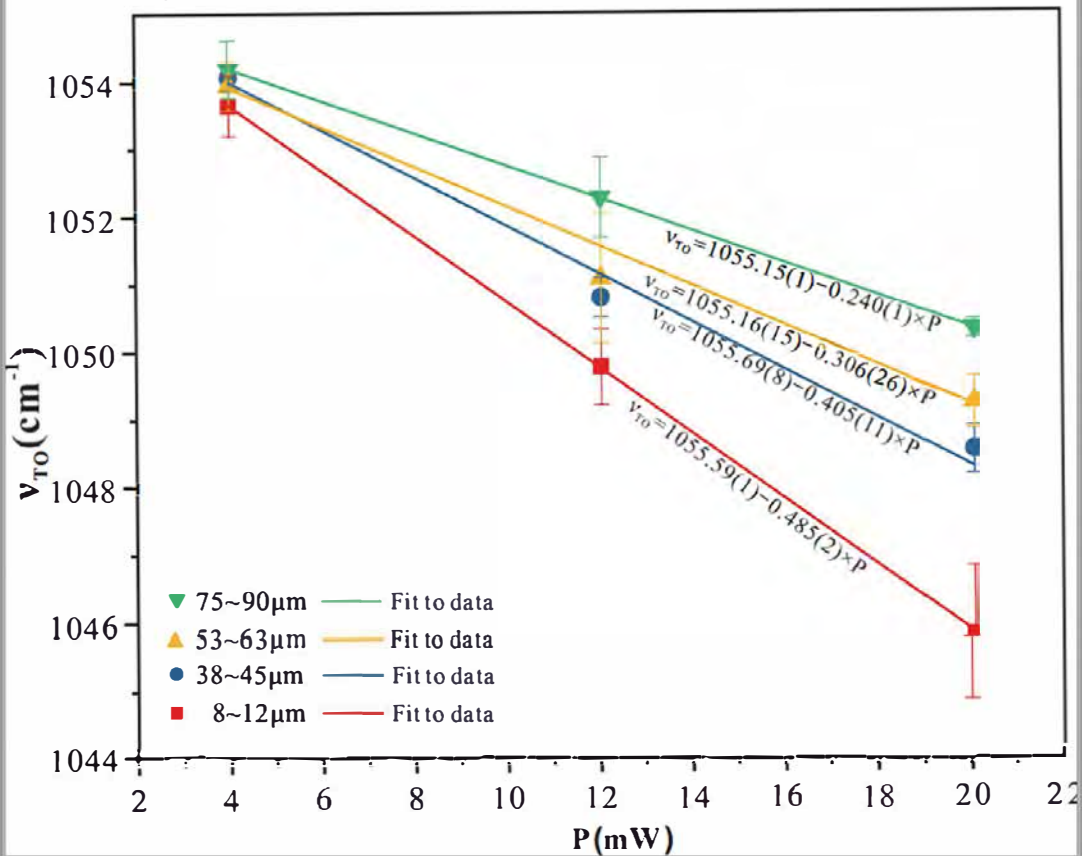


Fig. 6

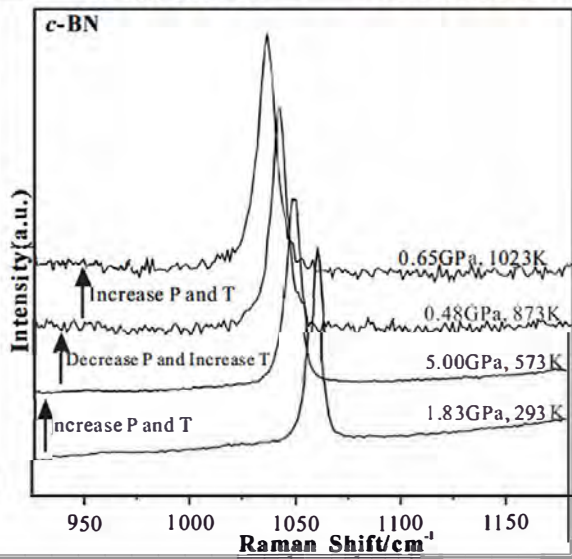
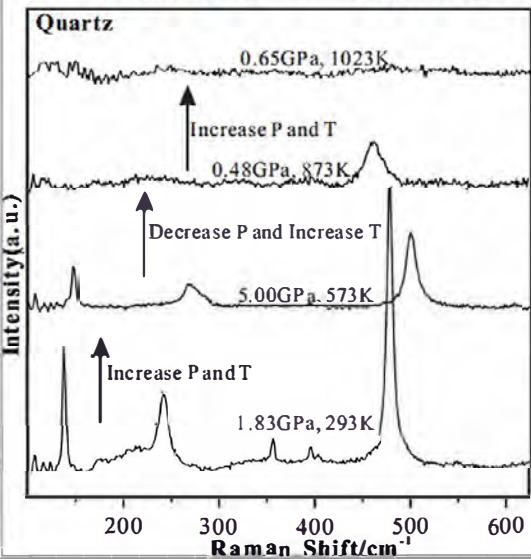
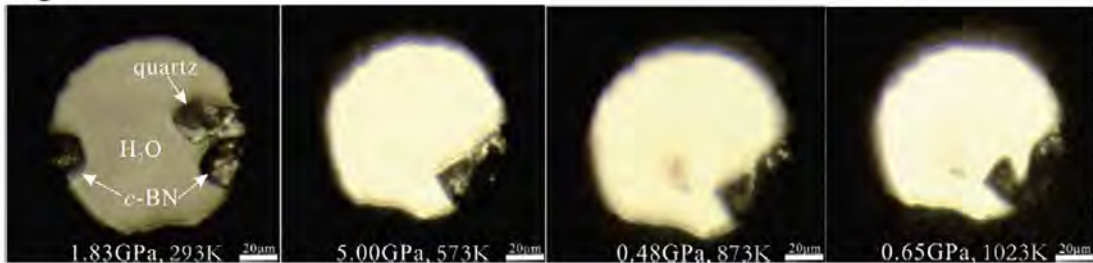


Fig. 7

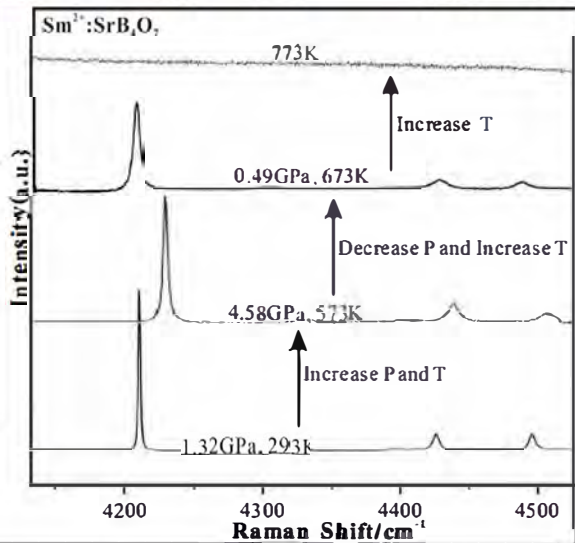
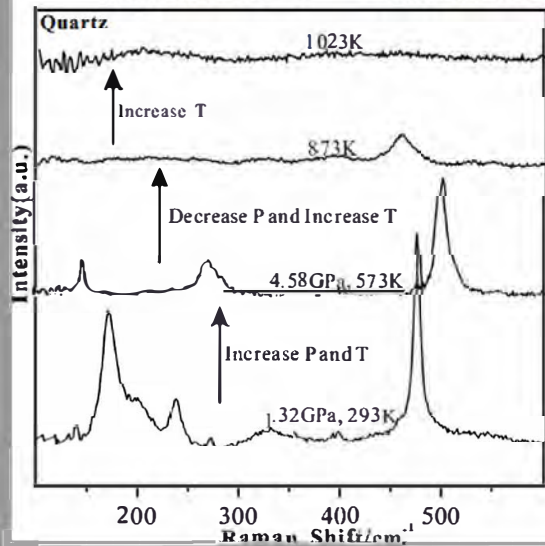
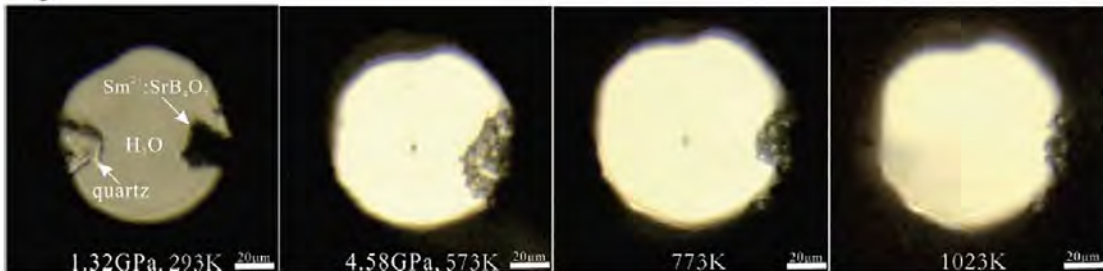


Fig. 8

