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3	Structure and equation of state of Ti-bearing davemaoite: new insights into the chemical
4	heterogeneity in the lower mantle
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Abstract

17 Davemaoite (CaSiO₃ perovskite) is considered the third most abundant phase in the pyrolytic 18 lower mantle and the second most abundant phase in the subducted mid-ocean ridge basalt 19 (MORB). During the partial melting of the pyrolytic upper mantle, incompatible titanium (Ti) 20 becomes enriched in the basaltic magma, forming Ti-rich MORB. Davemaoite is considered an 21 important Ti-bearing mineral in subducted slabs by forming a Ca(Si,Ti)O₃ solid solution. 22 However, the crystal structure and compressibility of Ca(Si,Ti)O₃ perovskite solid solution at 23 relevant pressure and temperature conditions were not systematically investigated. In this study, 24 we investigated the structure and equations of state of Ca(Si0.83Ti0.17)O3 and Ca(Si0.75Ti0.25)O3 25 perovskites at room temperature up to 82 GPa and 64 GPa, respectively, by synchrotron X-ray 26 diffraction (XRD). We found that both Ca(Si_{0.83}Ti_{0.17})O₃ and Ca(Si_{0.75}Ti_{0.25})O₃ perovskites have 27 a tetragonal structure up to the maximum pressures investigated. Based on the observed data and 28 compared to pure CaSiO₃ davemaoite, both Ca(Si_{0.83}Ti_{0.17})O₃ and Ca(Si_{0.75}Ti_{0.25})O₃ perovskites 29 are expected to be less dense up to the core-mantle boundary (CMB), and specifically ~1-2 % 30 less dense than CaSiO₃ davemaoite in the pressure range of the transition zone (15-25 GPa). Our 31 results suggest that the presence of Ti-bearing davemaoite phases may result in a reduction in the 32 average density of the subducting slabs, which in turn promotes their stagnation in the lower 33 mantle. The presence of low-density Ti-bearing davemaoite phases and subduction of MORB in 34 the lower mantle may also explain the seismic heterogeneity in the lower mantle, such as large 35 low shear velocity provinces (LLSVPs).

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Keywords: Ti-bearing davemaoite, perovskite, Ca(Si,Ti)O₃ solid solution, equation of state, slab
 stagnation, density

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Introduction

40 Seismic observations and mineral physics studies have provided a comprehensive understanding 41 of the mineralogy of the Earth's mantle (Dziewonski and Anderson 1981; Kaminsky 2012). 42 According to the pyrolytic composition model for the mantle, davemaoite, a perovskite-structured 43 CaSiO₃, is widely considered the third most abundant phase in the lower mantle (Kesson et al. 44 1998; Murakami et al. 2005; Ringwood 1962; Sun et al. 2016), making up an estimated 6-12% of 45 its volume. This is supported by the discovery of davemaoite as inclusion in superdeep diamonds 46 (Anzolini et al. 2018, 2016; Tschauner et al. 2021). Furthermore, davemaoite is the second most 47 abundant phase in subducted oceanic crust, comprising an estimated 23-30% due to the higher Ca 48 content of oceanic crust and surpassed only by bridgmanite (Hirose et al. 2005; Kesson et al. 1994; 49 Kudo et al. 2012; Steeve et al. 2019; Stixrude and Lithgow-Bertelloni 2012). Nestola et al. (2018) 50 found evidence of inclusions of davemaoite in diamond originating from the subducted oceanic crust. The seismic velocity of the subducted oceanic crust or davemaoite can be used to explain 51 52 the presence of the large low shear velocity provinces (LLSVPs), as suggested by Thomson et al. 53 (2019).

54 Numerous studies have been conducted over the past several decades to investigate the 55 physical and chemical properties of davemaoite, given its abundance and importance. Previous 56 studies have focused on the structure and equations of state of davemaoite (Mao et al. 1989; Yagi 57 et al. 1989; Shim et al. 2000a, 2000b, 2002; Kawai and Tsuchiya 2014; Chen et al. 2018; Sun et 58 al. 2022); sound velocities or elastic properties at high pressures (Tamai and Yagi 1989; Akber-59 Knutson et al. 2002; Steeve et al. 2019; Thomson et al. 2019); the synthesis of davemaoite (Liu 60 and Ringwood 1975); its phase transition at high pressure-temperature (P-T) conditions 61 (Komabayashi et al. 2007); and its presence as diamond inclusions (Nestola et al. 2018; Tschauner

et al. 2021). However, there is still little consensus on the structure and physical properties of davemaoite, particularly its equation of state (EoS) and elastic properties, primarily due to the fact that davemaoite is unquenchable and readily converts to glass after decompression (Liu and Ringwood 1975; Mao et al. 1989). This unrecoverable nature prevents the synthesized davemaoite from being further analyzed at ambient conditions.

67 Furthermore, davemaoite formed within the subducted slabs may contain considerable 68 amounts of Ti by forming Ca(Si,Ti)O₃ solid solution, as Ti is an incompatible element that tends 69 to enrich in basaltic magma during partial melting of the upper mantle (Corgne and Wood, 2005; 70 Hirose et al. 2004), which then in turn forms the Ti-rich mid-ocean ridge basalt (MORB) (Gale et 71 al. 2013; Hirose et al. 2005, 2004; Ricolleau et al. 2010). When the oceanic plates subduct to the 72 lower mantle, Ti-rich davemaoite phase may form from the MORB under high-pressure and high-73 temperature conditions of the mantle. Indeed, Ca(Si,Ti)O₃ perovskite as inclusions in superdeep 74 diamonds were found to contain high Ti, as much as ~40–60 mol% of CaTiO₃ in the solid solution 75 ((Thomson et al. 2016).

76 If mantle heterogeneity such as LLSVPs is the result of the enrichment of recycled oceanic 77 material as proposed by Thomson et al. (2019), the physical properties of Ti-bearing davemaoite 78 should also be investigated to provide better reference to real geological conditions. The basic 79 understanding of the series of Ti-rich Ca(Si,Ti)O₃ perovskites was previously experimentally 80 investigated by Leinenweber et al. (1997) and Sinelnikov et al. (1998). However, Leinenweber et 81 al. (1997) focused on investigating the structure and lattice parameters of $Ca(Ti_x,Si_{1-x})O_3$ at 82 ambient conditions, where 1 > x > 0.65, after quenching from multi-anvil press synthesis; 83 Sinelnikov et al. (1998) focused on reporting the data of Ca(Si_{0.23}Ti_{0.77})O₃ and Ca(Si_{0.49}Ti_{0.51})O₃ 84 for the composition between CaSiO₃ and CaTiO₃. The maximum pressure reached by Sinelnikov

85 et al. (1998) was also limited to 15 GPa. According to Leinenweber et al. (1997) and Sinelnikov 86 et al. (1998), the introduction of Ti into the solid solution system substantially changes the structure, 87 unit cell volume, and elastic properties of the solid solution under ambient conditions. Thus, it is 88 likely that Ti in the subducted MORB also changes elastic properties and structure of davemaoite, 89 likely contributing to the chemical and seismic heterogeneity in the deep mantle. Diamond-hosted 90 Ca(Si,Ti)O3 inclusions were found to contain a wide range of Ti, with Ti content equivalent to 0-91 10 mol% CaTiO₃ to as high as ~40–60 mol% (Thomson et al. 2016). Therefore, it is crucial to 92 investigate the effect of Ti on the structure and elastic properties of Ca(Si,Ti)O₃ under the relevant 93 pressure conditions of the lower mantle. 94 To gain a better understanding of the density and compressibility of Ti-rich davemaoite 95 under relevant mantle conditions, we investigated the structure, density and bulk modulus of 96 Ca(Si_{0.83}Ti_{0.17})O₃ and Ca(Si_{0.75}Ti_{0.25})O₃ davemaoite, referred to as 17TiPv and 25TiPv thereafter, 97 up to approximately 82 GPa and 64 GPa, respectively. Our study reveals that the incorporation of 98 Ti into $CaSiO_3$ to form $Ca(Si,Ti)O_3$ solid solution results in a significant decrease in its density. 99 This finding has implications for understanding mantle dynamics, as a subducting slab enriched in 100 Ti can become stagnant at the transition zone (Fukao et al. 2001, 2009) due to the enrichment of 101 the lower density of the Ti-bearing davemaoite. The enrichment of Ti-bearing davemaoite can 102 likely lead to the seismologically observed low-density feature in the lower mantle, such as 103 LLSVPs (Hernlund and Houser 2008; Deschamps et al. 2012, 2012; Frost and Rost 2014). 104 105 **Experimental Methods** 106 We prepared Ca1.08Si0.83Ti0.17O3.08 and Ca1.02Si0.75Ti0.25O3.02 glass as our starting materials. Both

107 starting materials were prepared by mixing and grinding appropriate stoichiometric ratios

(100:17:83 and 4:3:1 in the molar fraction, respectively) CaCO₃, TiO₂, and SiO₂ together with acetone in the agate motor. For each composition, the mixture was placed in a platinum crucible in an oven at 200 °C overnight to remove moisture. The dehydrated mixture was then heated to 900 °C overnight to decarbonate the CaCO₃ portion into CaO, followed by heating to 1600 °C until fully molten. The molten glass was then rapidly quenched with a water bath.

113 The quenched 17TiPv and 25TiPv glass was recovered from the crucible and broken into 114 fragments of approximately 1-2 mm in size and cast into epoxy. Chemical compositions were 115 determined using the electron probe microanalysis (EPMA) method using the field emission electron microprobe JEOL JXA-8500 F in the Department of Earth Sciences of the University of 116 117 Hawaii, with an acceleration voltage of 15-20 keV, a current of 20-30 A, and a spot size of up to 5 118 μm. Pure CaTiSiO₅ sphene was used as the standard to calibrate Ca, Si, and Ti. The average 119 chemical compositions of 17TiPv glass and 25TiPv glass are Ca1.08Si0.83Ti0.17O3.08 and 120 Ca1.02Si0.75Ti0.25O3.02, respectively (Table S1 and Table S2). The chemical composition was 121 homogeneous between the probe points in both samples. Based on such chemical compositions, 122 the (Si+Ti):O should be 1:3 in chemical formula of perovskite structure, and there may be a trace 123 amount of excess CaO (<4 wt%). The phases after laser heating would be Ca(Si,Ti)O₃ with minor 124 CaO, although the CaO was not detected by high pressure X-ray diffraction (XRD) experiments 125 of Ca(Si,Ti)O₃ perovskites in the later part of this study.

The 17TiPv and 25TiPv glass recovered from the crucible were further ground into finer grains within 10-100 μm. A few grains of each sample were loaded into diamond anvil cells (DAC) with neon using the gas-loading system at GSECARS, Advanced Photon Source, Argonne National Laboratory (Rivers et al. 2008). A short symmetric DAC with diamonds of 250 μm in culet size was used for 17TiPv glass and a regular symmetric DAC with diamonds of 300 μm in

131 culet size was used for 25TiPv, respectively. X-ray diffraction experiments were performed at 132 Beamline 13-IDD of the Advanced Photon Source, Argonne National Laboratory. The wavelength 133 of the X-ray was 0.2952 Å. In the experimental setup, the Pilatus 1M CdTe detector calibrated by 134 the powder LaB₆ NIST standard was used at a distance of 207 mm from the diamond anvil cell. 135 17TiPv glass and 25TiPv glass were compressed to ~20 GPa and ~15 GPa, respectively. The 136 sample was then heated by a double-sided laser heating system using the 100 W fiber laser (IPG 137 Photonics) with a wavelength of 1064 nm and a spot size of 12 µm flat top. After being heated to 138 high temperatures above 1800 K, the glass sample turned into an opaque perovskite phase. When 139 the formation of the perovskite phase was detected by X-ray diffraction (XRD), the laser heating 140 was repeated several times to completely transform the glass into crystalline phases. We collected 141 XRD data on the synthesized perovskite samples that were compressed to 82 and 64 GPa for 142 17TiPv and 25TiPv, respectively, with a typical pressure interval of 2-3 GPa by the gas membrane 143 at 300 K. At each pressure point, a step scan XRD measurement was performed for 17TiPv and 144 25TiPv, with a step interval of 0.5°, total rotation range of 34° and 25°. The collection time was 145 one second per step and two seconds per step for 17TiPv and 25TiPv, respectively.

146 XRD images with masked diamond peaks were integrated into 1-D diffraction patterns 147 using Dioptas software (Prescher and Prakapenka 2015). Unit cell volumes of gold were 148 determined by PDIndexer (Seto et al. 2010), which were subsequently used to determine the 149 sample pressures using the EoS by Fei et al. (2007). PeakPo (Shim 2020) and GSAS-II were used 150 to analyze the lattice parameters of the diffraction pattern of Ti-bearing davemaoite (Fig. S1). After 151 determining the lattice parameters, unit cell volume, as well as pressure points, we used EoSFit7-152 GUI (Gonzalez-Platas et al. 2016) and Pytheos (Shim 2017) to calculate the second-order Birch-153 Murnaghan EoS of 17TiPv and 25TiPv.

154

Results and Discussion

155 Syntheses and structures of Ca(Si,Ti)O₃ perovskites

The amorphous samples were compressed to a pressure of approximately 15-20 GPa, within the stability field of the Ca-perovskite phase (Milani et al. 2021). Afterward, we conducted a widescan XRD measurements before and after heating to identify the formation of the perovskite phase. As expected, the XRD pattern of the samples before heating showed amorphous scattering. However, the perovskite peaks appeared upon laser heating to approximately 2000 K, as shown in Figure 1.

The synthesized 17TiPv and 25TiPv perovskites both exhibited a tetragonal structure. The subtle splitting of the peaks of the tetragonal perovskite phase, (200), (002), (211), (112), is apparent in 17TiPv (Fig. 1a). In contrast, such splitting is more subtle in 25TiPv: only showing a shoulder or overlap each other, but still distinguishable (Fig. 1b). The pressure of 17TiPv became 27.2 GPa after laser heating at approximately 20 GPa. The pressure of 25TiPv did not change significantly, slightly changing from 15 GPa to 15.5 GPa after laser heating.

168 Representative XRD pattern of 17TiPv and 25TiPv are shown in Figure 2 with a focus on 169 the 2θ range of 6-15°, where most of the peaks are strong and distinguishable. The representative 170 full range XRD raw data is shown in Fig. S2. The peaks with corresponding *d*-spacing are 171 normalized to the maximum intensity of the first two overlapping diffraction peaks of the 172 perovskite structure, (112) + (200) and (110) + (101) for 17TiPv and 25TiPv, respectively. Our 173 observations agree well with the space group of 14/mcm and P4/mmm for 17TiPy and 25TiPy. 174 respectively. Several possible space groups have been reported for pure CaSiO₃, *I4/mmm*, *P4/mmm*, 175 P4/mbm, and I4/mcm (Shim et al. 2002), all of which could be observed given that there is subtle 176 difference in free energy among them and the experimental XRD pattern would be sensitive to

177 local stress and pressure gradient in the DAC (Chen et al. 2018). According to Chen et al. (2018), 178 the expected XRD pattern of *I4/mcm* has an XRD pattern similar to that of *P4/mmm*, but with an 179 additional peak (211), between the equivalent positions of (110) and (111) of the ideal cubic 180 perovskite structure. Our XRD data show that the peak (211) is present in 17TiPv but not in 25TiPv. 181 Other space groups, *I4/mmm* and *P4/mbm* reported by Shim et al. (2002) also contain additional 182 peaks between the equivalent positions of (110) and (111) of the ideal cubic perovskite structure with a 20 lower than that of (211) of I4/mcm. These peaks are not observed in both 17TiPv and 183 184 25TiPv, suggesting I4/mcm and P4/mmm structures for 17TiPv and 25TiPv, respectively. The 185 observed 20 and the corresponding d-spacing of the representative XRD pattern of 17TiPv and 186 25TiPv agree well with the calculated values for their specific structures, which can be found in 187 Table S3 and Table S4.

188 The c/a ratios of 17TiPv and 25TiPv are compared with those of CaSiO₃ reported by Ono 189 et al. (2004) and Sun et al. (2022) in Fig. 3. Our study shows that the c/a ratios of both 17TiPv and 190 25TiPv increase monotonically with pressure. Ono et al. (2004) and Sun et al. (2022) show 191 opposite pressure dependence of c/a ratio despite both studies report CaSiO₃ adopts P4/mmm 192 structure. The c/a ratio of CaSiO₃ reported by Ono et al. (2004) and Sun et al. (2022) decreases 193 and increases monotonically with rising pressure, respectively. Due to the sensitivity of structure 194 of perovskites to the strain-stress condition (Chen et al. 2018), there is little consensus regarding 195 the trend of pressure dependence of c/a ratio of davemaoite.

The structure of 17TiPv is demonstrated by analyzing the step-scan XRD pattern at 27.2 GPa (Fig. 2a). The XRD pattern shows that the (004) and (220) peaks split and can be recognized, indicating that 17TiPv adopts a tetragonal structure. We also observed the (211) peak at $2\theta \sim 8^{\circ}$, which is evidence of an *I4/mcm* structure. The representative XRD pattern of 17TiPv with 20

200 converted to *d*-spacing is shown in Fig. 2b. We found that the splitting of (004) and (220) becomes 201 more obvious as the pressure increases. The c/a ratio of 17TiPv increases monotonically with 202 pressure: ~1.42 at ~7 GPa and rises to ~1.44 at 82 GPa (Fig. 3). No pressure-induced phase 203 transition is observed for 17TiPv up to 82 GPa.

204 The XRD data of 25TiPv were collected after the perovskite sample was synthesized at 205 15.5 GPa. We observed that at 15.5 GPa, most of the peaks have distinguishable shoulders (see 206 peaks (200), (002), (211), (112), (220), and (202) in Fig. 2c). However, we did not observe the 207 peak (211) at $2\theta \sim 8^{\circ}$ for the structure *I4/mcm*, indicating that 25TiPv likely adopts a space group 208 *P4/mmm*. Interestingly, we found that the c/a ratio becomes higher than 1.0 above approximately 209 32 GPa. In Fig. 2d, at 16.6 GPa, (200) has a stronger intensity than (002), and (002) is presented 210 as a shoulder with a lower d-spacing. At ~29 GPa and ~40 GPa, the (200) and (002) almost 211 overlapped each other, forming a symmetric (200) + (002) peak. As the pressure increases, the 212 (200) + (002) peak becomes asymmetric again, and a shoulder appears at higher *d*-spacing. Chen 213 et al. (2018) summarized the calculated relative positions and intensities of (200) and (002) in the 214 P4/mmm space group. The calculated (200) has a stronger intensity than (002). If the c-axis has a 215 shorter length than the a-axis, (200) has a lower 2θ (or higher d-spacing) than (002), and vice versa. 216 Assuming that the relative intensity of diffraction peaks of 25TiPv generally follows the 217 calculation, at 16.6 GPa, (200) has a stronger intensity and higher d-spacing than (002). This 218 feature can be described by the crystal structure, P4/mmm with c < a. Whereas, at ~52 GPa and 219 ~64 GPa, (200) has a stronger intensity and a lower d-spacing than (002), better described by the 220 crustal structure, P4/mmm with c > a. Interestingly, further data processing to constrain the lengths 221 of a-axis and c-axis assuming that the relative intensities of (200) and (002) follow the theoretical 222 calculation reveals that the c/a ratio becomes nearly one at approximately 32 GPa (Fig. 3). The c/a ratio of 25TiPv is ~ 0.99 at 15 GPa and temporarily becomes pseudocubic structure at 32 GPa when the c/a ratio approaches ~1.0. However, the c/a ratio of 25TiPv continues to increase above 32 GPa without changing its structure *P4/mmm* to ~1.015 at 64 GPa. Further investigations are needed to determine whether this feature is a potential phase transition by using the diamond anvil cell with wider opening angle, which provides complete access to higher diffraction angle peaks for a comprehensive refinement of the structure.

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230 Equation of state (EoS) of Ca(Si,Ti)O₃ perovskites

231 The lattice parameters for 17TiPv and 25TiPv were determined from the XRD measurements 232 (Table S5 and Table S6). Unit-cell volumes under ambient conditions are not available, because 233 the 17TiPv sample is not quenchable and became amorphous after being decompressed to ambient 234 pressure, similar to pure CaSiO₃ davemaoite. The 25TiPv sample was compressed to 64 GPa, but 235 the diamond broke under further compression. The observed value of unit cell volumes, as well as 236 fitted second-order Birch-Murnaghan Equations of State (EoS) of 17TiPv and 25TiPv, are shown 237 in Fig. 4. The formula units in the unit cell (Z) of I4/mcm and P4/mmm are Z = 4 and Z = 1, 238 respectively. Molar volumes under ambient conditions (Vo) of 17TiPv and 25TiPv are 28.75 239 cm³/mol and 29.68 cm³/mol, respectively. The calculated initial bulk moduli (K_0) of 17TiPv and 240 25 TiPv are 224 ± 2 GPa and 205 ± 4 GPa, and their pressures derivatives of the bulk modulus (K') 241 are all fixed at 4.0. The residuals are typically within ± 0.5 % (Fig. S3). Including or excluding 242 decompression data has a minor effect on the second-order Birch-Murnaghan Equations of State 243 (EoS) of 17TiPv (Fig. S4). The stress-normalized pressure (f-F) figure in supplementary material 244 (Fig. S4) shows both 17TiPv and 25 TiPv have well-fit horizontal slopes in the *f-F* plot, suggesting 245 their derivatives of the bulk modulus are near 4.0.

246 The EoS's of 17TiPv and 25TiPv were compared with those of pure CaSiO₃ davemaoite 247 (Mao et al. 1989; Yagi et al. 1989; Shim et al. 2000a; Sun et al. 2022), and pure CaTiO₃ perovskite 248 (Guennou et al. 2010) (Fig. 4), with the unit cell volumes of all phases converted to molar volume. 249 Despite the lack of consensus on the EoS of pure CaSiO₃ davemaoite, the molar volumes of all Ti-250 bearing phases investigated exceed those of pure CaSiO₃ davemaoite, with that of pure CaTiO₃ 251 perovskite being the highest followed by 25TiPv and 17TiPv. Our results show that the molar 252 volume of Ca(Si,Ti)O₃ solid solutions increases monotonically with increasing Ti content, 253 consistent with the relatively larger radius of Ti than that of Si. The EoS's of Ca(Si,Ti)O₃ 254 perovskites were summarized and their molar V_0 and K_0 were plotted as a function of Ti 255 concentration (Table 1 and Fig. 5) (Caracas et al. 2005; Chizmeshya et al. 1996; Fischer et al. 256 1993; Guennou et al. 2010; Jung and Oganov 2005; Kawai and Tsuchiya 2014; Kung and Rigden 257 1999; Leinenweber et al. 1997; Mao et al. 1989; Ono et al. 2004; Shim et al. 2000b, 2000a; 258 Sinelnikov et al. 1998; Sun et al. 2022, 2016; Tarrida and Richet 1989; Thomson et al. 2019; Truffet 259 et al. 2023; Wang et al. 1996; Yagi et al. 1989). Although the EoS of pure CaSiO₃ davemaoite 260 remains controversial, there are two interesting features in Fig. 5. First, in Fig. 5a, the molar Vo 261 rises almost linearly with increasing Ti content. Leinenweber et al. (1997) showed that the 262 Ca(Si,Ti)O₃ solid solution has orthorhombic structure when molar fraction of CaTiO₃ is greater 263 than 50%, and its tetragonal structure when the composition is exactly $Ca(Si_{0.5}Ti_{0.5})O_3$ or less Ti 264 concentration. It is intriguing that the molar V₀ rises linearly with the Ti content, even if the whole 265 solid solution undergoes a drastic change in symmetry. The molar V_0 of pure CaSiO₃ davemaoite is around 27.5 cm³, and the molar V_0 of pure CaTiO₃ perovskite is around 34 cm³. The 266 267 compositions between CaSiO₃ and CaTiO₃ have a molar V_0 close to the trend line.

268 The K_0 is expected to decrease monotonically with Ti content, since Ti has a larger radius 269 and is more compressible than Si. The dashed lines are plotted to guide the eyeline based on what 270 we observed and current reported K_0 of the Ca(Si,Ti)O₃ solid solution. The K_0 drops most 271 dramatically from 17TiPv (~224 GPa) to 25TiPv (~209 GPa); then drops less drastically from 272 25TiPv to CaTiO₃ 50% (~185 GPa). However, above CaTiO₃ 50%, K₀ only barely drops across 273 this compositional range, from 185 GPa at Ca(Si_{0.5}Ti_{0.5})O₃ to around 180 GPa in pure CaTiO₃ with 274 a 50% increase in the molar fraction of CaTiO₃. This could be because the Ca(Si,Ti)O₃ solid 275 solution adopts a tetragonal structure or orthorhombic structure, when Ti concentration is below 276 and above CaTiO₃ 50%, respectively (Leinenweber et al. 1997; Sinelnikov et al. 1998). At ambient conditions, the radius of octahedral site Ti^{4+} is ~0.63 Å, and the radius of Si^{4+} is ~0.4 Å, with the 277 278 former more compressible. From CaSiO₃ to Ca(Si_{0.5}Ti_{0.5})O₃, the crystal lattice has a more 279 symmetric tetragonal structure dominated by Si⁴⁺. The crystal lattice may prefer its higher 280 symmetry and smaller octahedral sites as more large-radius Ti atoms are introduced. As a result, 281 K_0 may be more sensitive to the introduction of Ti atoms in this composition range. While from 282 $Ca(Si_{0.5}Ti_{0.5})O_3$ to $CaTiO_3$, the crystal lattice is an orthorhombic structure and the SiO₈ or TiO₈ 283 octahedral are already highly tilted, given that the octahedral sites are dominated by larger Ti^{4+} . Since Ti⁴⁺ already dominates the lattice, elastic properties are largely controlled by Ti⁴⁺ that already 284 exists; replacing more Si^{4+} with Ti^{4+} does not make K_0 decreases substantially with the Ti content 285 286 from Ca(Si0.5Ti0.5)O3 to CaTiO3.

Between pure CaSiO₃ davemaoite and 17TiPv, previous literature shows a confusing result, as K_0 of pure CaSiO₃ davemaoite reported by former studies is still controversial. If the K_0 is around 220-230 GPa as Chen et al. (2018), Shim et al. (2000b, 2000a), or Sun et al. (2022), had reported, the softening of K_0 is modest from the composition range of CaSiO₃ (~225 GPa) to

291 17TiPv (~209 GPa). Whereas, if the K_0 is around 250 GPa (Shim et al. 2002; Sun et al. 2016), 292 there may be another linear relationship where K_0 drops less drastically than the range of 17TiPv 293 to 25TiPv, but still a considerable drop. If the K_0 is around 280 GPa (Mao et al. 1989; Tarrida and 294 Richet, 1989; Yagi et al. 1989), the K_0 decreases dramatically and linearly from CaSiO₃ to 25TiPv. 295 A more complicated and non-linear relationship between K_0 and Ti content across the entire 296 composition range cannot be ruled out, due to the little consensus of elastic properties of pure 297 CaSiO₃ davemaoite and among the literature.

298 Densities and elastic properties of Ca(Si,Ti)O₃ solid solutions

299 We calculated the density of 17TiPv and 25TiPv as a function of pressure based on their EoS's and 300 chemical composition then compared their densities with the density of pure CaSiO₃ davemaoite 301 (Shim et al. 2000b; Sun et al. 2022) and pure CaTiO₃ perovskite (Guennou et al. 2010) calculated 302 from the EoS's in the literature (Fig. 6a). In Fig. 6a, we plot the density of davemaoite calculated 303 from Shim et al. (2000b) and Sun et al. (2022), because both studies were based on room 304 temperature experiment, and their maximum pressure range is above 100 GPa summarized in Table 305 1. They have the same temperature conditions as this study and their data intensively cover most 306 of the pressure range of the mantle. The EoS of davemaoite reported prior to 2000s with a higher 307 *K*₀ (~270-280 GPa), i.e., Mao et al. (1989); Tarrida and Richet (1989); Yagi et al. (1989), when the 308 structure was constrained to a cubic configuration and the experiments were performed under 309 relatively non-hydrastatic conditions (Table 1). Studies after the 2000s generally reported Ko 310 between 224 GPa and 236 GPa and molar V_{θ} around 27.5 cm³, e.g. Ono et al. (2004); Shim et al. 311 (2000a, 2000b); Sun et al. (2022), etc, leading to similar results as Shim et al. 2000b and Sun et al. 312 2022 in Fig. 6a. For the density of CaTiO₃ perovskite, we plot the density calculated from Guennou

et al. (2010). Although Truffet et al. (2023) has a greater pressure range, Truffet et al. (2023) is a
thermal EoS study, and the data coverage at room temperature is sparse.

315 The density of Ca(Si,Ti)O₃ solid solutions under ambient conditions decreases 316 monotonically with increasing Ti concentration (Fig. 6a). Adding Ti into the Ca(Si,Ti)O₃ solid 317 solutions increases the formula weight of Ca(Si,Ti)O₃ solid solution. However, the unit cell volume 318 increases with increasing Ti content. As a result, the effect of the Ti-driving expansion of cell 319 volume overwhelms the effect of formula weight increasing, and the net effect is the density 320 decreasing. In addition, the K_0 decreases with rising Ti-contents, and thus, the unit cell volume of 321 Ca(Si,Ti)O₃ solid solutions shrinks more rapidly with rising Ti-content. As a result, the density of 322 25TiPv becomes similar to the density of 17TiPv at approximately 100 GPa. Nevertheless, the 323 densities of 17TiPv and 25TiPv never reach the density of pure CaSiO₃ davemaoite even below 324 140 GPa. However, if the Ti content continues to increase, the crystal lattice becomes more 325 compressible and the density rises more rapidly (see the rapid rise in the density increase of CaTiO₃ 326 in Fig. 6). The CaTiO₃ is relatively more compressible, and its density exceeds the density of 327 25TiPv, 17TiPv, and CaSiO₃ at 25 GPa, 45 GPa, and 110 GPa, respectively.

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Geophysical Implications

We have determined the crystal structure, EoS, and densities of Ca(Si,Ti)O₃ pervoskites with 17% and 25% Ti. The densities of Ca(Si,Ti)O₃ perovskites were compared with those of the Preliminary Reference Earth Model (PREM) (Dziewonski and Anderson 1981), other major mantle minerals, e.g. ferropericlase, ringwoodite, and bridgmanite (Tange et al. 2009, 2012; Chang et al. 2015; Wolf et al. 2015), and pure tetragonal-structured CaSiO₃ perovskite (Shim et al. 2000b; Sun et al. 2022) in Fig. 6a. Pure MgSiO₃ bridgmanite has a similar density as PREM; however, Fe-bearing (Mg_{0.87}Fe_{0.13})SiO₃ bridgmanite is a chemically more relevant phase for the mantle. Tschauner et

al. (2021) reported that davemaoite adopts a cubic structure in the diamond inclusion, in contrary
to most recent high-pressure results of tetragonal-structured CaSiO₃ as reported in Table 1. We
therefore only compare our results with those for tetragonal-structured CaSiO₃.

339 We first compare the 17TiPv and 25TiPv with CaSiO₃ perovskite and we found that 17TiPv 340 and 25TiPv are less dense than pure tetragonal-CaSiO₃ perovskites in the investigated pressure 341 ranges at 300 K. At the transition zone pressure of ~15-25 GPa, both 17TiPv and 25TiPv are ~0.05-342 0.1 g/cm³ (or ~1-2 %) less dense than pure tetragonal CaSiO₃ davemaoite (Sun et al. 2022), and 343 (Mg_{0.87}Fe_{0.13})SiO₃ bridgmanite (Wolf et al. 2015). Although this study was conducted in 300 K, a 344 pure CaSiO₃ davemaoite turns into cubic phase transition at the temperature as low as 490-580 K 345 (Komabayashi et al. 2007), much lower than the temperature of cold slab. The 17TiPv and 25TiPv 346 may also undergo such a phase transition. The tetragonal-cubic phase transition is a second-order 347 phase transition without any volume change (Komabayashi et al. 2007). Consequently, the relative 348 density difference between CaSiO₃ and that of 17TiPv and 25TiPv would still be ~1%. By 349 replacing the CaSiO₃ in the traditional model of MORB with Ti-bearing davemaoite, it is expected 350 that the average density of MORB adapting Ti-bearing davemaoite is lower than that of 351 conventional aspect.

Seismological studies have suggested that some subducting slabs become stagnant before penetrating deeper into the lower mantle (Okino et al. 1989; Fukao et al. 1992, 2009; Fukao and Obayashi 2013). Several models have been proposed to explain the slab stagnation. (1) The negative Clapeyron slope of the ringwoodite to bridgmanite and ferro periclase transition (the postspinel transition) and/or the delay of the onset of this transition due to the lower temperature of the slabs (Christensen and Yuen 1985; King et al. 2015). (2) Based on the first hypothesis, some studies suggested that the water in slabs makes the Clapeyron slope of the post-spinel transition negatively

359 greater, and hence, water further favors slab stagnation (Litasov et al. 2005, 2006). (3) The slabs 360 could also stagnate when it encounters a sudden increase in viscosity of the lower mantle (Gurnis 361 and Hager 1988). (4) Some studies consider slab stagnation as a temporary geometry caused by 362 trench retreat (Christensen 1996) or lateral mantle flow (Peng et al. 2021a, 2021b). Our results on 363 densities of Ti-bearing CaSiO₃ perovskites may provide new insight into slab stagnation, as density 364 changes due to phase transitions of mantle minerals play a key role in slab dynamics (Christensen 365 and Yuen, 1985; King et al. 2015; Litasov et al. 2006, 2005). 17TiPv and 25TiPv are ~1-2% less dense than pure CaSiO₃ (Fig. 6a). As the second most abundant phase in the subducted MORBs, 366 367 the presence of Ti-bearing davemaoite may lower the density of the subducting slabs and thus can 368 be among the factors contributing to their stagnation in the deep mantle.

369 Pure CaSiO₃ davemaoite is traditionally considered an "invisible" phase, due to its similar 370 density with (Mg0.87Fe0.13)SiO₃ bridgmanite (Fig. 6a). However, our results indicate that Ti-bearing 371 davemaoites have lower density than that of (Mg_{0.87}Fe_{0.13})SiO₃ bridgmanite and pure CaSiO₃ 372 davemaoite, making them seismologically distinguishable within the lower mantle. The 373 accumulation of subducted oceanic crust enriched with Ti-bearing davemaoite in the lowermost 374 mantle may be responsible for the LLSVPs. Thomson et al. (2019) proposed that the LLSVPs can 375 be explained by CaSiO₃ davemaoite based on the lower shear wave velocity of Ca(Si₀₆Ti_{0.4})O₃ 376 observed with ultrasonic experiments. Due to the unavailability of shear wave velocity data at lower mantle conditions, we calculated the bulk sound velocity $(\sqrt{K/\rho})$, where K and ρ are bulk 377 378 modulus and density, respectively) of pure CaSiO₃ davemaoite (Shim et al. 2000b; Sun et al. 2022), 379 pure CaTiO₃ perovskite (Guennou et al. 2010), 17TiPv and 25TiPv from their EoS's as a reference 380 (Fig. 6b). 25TiPv presents a lower bulk sound velocity than that of pure CaSiO₃; however, the bulk 381 sound velocity of 17TiPv is similar to that of pure CaSiO₃ in the investigated pressure range. If the 382 shear moduli (G) also follow the trend as bulk sound velocity in Fig. 6b, the 17TiPv would have a 383 similar G as $CaSiO_3$, supporting Thomson's hypothesis with stoichiometrically more reasonable 384 Ti content. 25TiPv with higher Ti content would have a lower G and thus would be more favorable 385 for the formation of LLSVPs. Although the specific Ti concentration in the LLSVPs is unknown, 386 the accumulation of oceanic crust with Ti-bearing davemaoites during the history of plate tectonics 387 might increase the Ti content of davemaoite phase in the lower mantle stoichiometrically close to 388 25TiPv, in turn account for the LLSVPs in the lowermost mantle. However, the bulk sound velocity 389 is a simplified reference to generally understand the seismic wave velocity of material. The definition of bulk sound velocity, $\sqrt{K/\rho}$, is not identical to either P wave velocity defined as $V_P =$ 390 $\sqrt{(K + \frac{3}{4}G)/\rho}$, or S wave velocity defined as $V_S = \sqrt{G/\rho}$, where shear modulus is indispensable 391 392 to the calculation. Further investigations of the shear velocities and shear moduli of Ti-bearing 393 davemaoite under lowermost mantle conditions are needed to support this hypothesis.

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Figure 1. (a) Integrated XRD patterns before and after heating of 17TiPv and (b) unrolled projections of the 2-D raw diffraction areas near splitting (004) and (220) peaks of 17TiPv. (c)

- 593 Integrated XRD patterns before and after heating of 25TiPv and (d) unrolled projections of the 2-
- 594 D raw diffraction areas near splitting (200) and (002) peaks of 25TiPv.



597 Figure 2. Representative integrated XRD patterns (a) and d-spacing (b) of 17TiPv, respectively. 598 Representative integrated XRD patterns (c) and d-spacing of 25TiPv (d), respectively. 17TiPv and

- 599 25TiPv are in *I4/mcm* and *P4/mmm* space group, respectively, based on our observation: 17TiPv
- 600 (*I4/mcm*) has an additional peak (211) at around $2\theta = 8^{\circ}$, where such peak is not observed in 25TiPv
- 601 (*P4/mmm*). Question marks are unidentified peaks.



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603 Figure 3. The c/a ratio of 17TiPv (red) and 25TiPV (blue) in this study, and CaSiO₃ reported by

Ono et al. 2004 and Sun et al. 2022. Both 17TiPv and 25TiPv have *c/a* ratio increases

605 monotonically with pressure. The 25TiPv has a *c/a* ratio becoming near 1.0 at 32 GPa without

606 changing space group after 32 GPa. However, Ono et al. (2004) and Sun et al. (2022) reported

607 opposite result regarding the c/a ratio of CaSiO₃ with the same space group observed. Therefore,

608 the c/a ratio is sensitive to the stress strain condition of each study.



Figure 4. The comparison of Birch–Murnaghan equations of state among 17TiPv (red), 25TiPv (blue), pure CaSiO₃ (Chen et al. 2018; Mao et al. 1989; Shim et al. 2000b; Sun et al. 2022; Yagi et al. 1989), and pure CaTiO₃ (Guennou et al. 2010). Opening squares of 17TiPv are decompression data. The dash lines are extrapolated regions without experimental data coverage. The unit cell volumes were converted into molar volume to have the same standard.



617 *Figure 5.* Molar V_0 (a) and K_0 (b) of Ca(Si,Ti)O₃ solid solution reported by literature as function

618 of CaTiO₃ molar fraction. The dash lines are to guide the eye.

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633 TABLE 1 Equation of state parameters of Ca(Si,Ti)O₃ perovskite.

Reference	Pressure	$V_{\theta} (\mathrm{cm}^3)^{\mathrm{a}}$	K ₀ (GPa)	K'	Symm. ^b	P_{\max}					
	medium				-	(GPa) ^c					
Ca(Si _{0.83} Ti _{0.17})O ₃ Experiment											
This work	Ne	28.75(6)	224(2)	4(fixed)	Tet $(I4/mcm)$	82.4					
		Ca(Sia 75Ti	0.25)O3 Exne	eriment	(i minem)						
This work	Ne	29.68(5)	205(4)	4(fixed)	Tet	63.6					
	1.00	29:00(0)	200(1)	(Intea)	(P4/mmm)	02.0					
		Ca(Si0 51Ti	(0.49)O3 Expe	eriment	(1 ////////////////////////////////////						
Sinelnikov et al.	N/A ^e	30.37(2)	185(5)	4(1)	Tet	15					
(1998)"											
Cincluiters et al	NT/A e	$Ca(SI_{0.23})$	182(2)		Orth	15					
$(1008)^d$	IN/A°	32.48(2)	182(2)	4(1)	Orth	15					
(1990)		CaSi	O2 Experime	nt							
Mao et al. (1989)	N/A ^f	27 32(5)	281(4)	4	Cub	134					
Yagi et al. (1989)	N/A ^f	27.32(3) 27.45(4)	288(13)	40(2)	Cub	112					
Tarrida and Richet	N/A ^f	27.46(6)	275(15)	4	Cub	96					
(1989)		(!)	_/()	-							
Wang et al. $(1996)^d$	N/A ^e	27.45(2)	232(8)	4.8(3)	Cub	12					
Shim et al. $(2000a)^d$	NaCl and Ar	27.45(1)	236(4)	3.9(2)	Cub	96					
Shim et al. (2000b)	NaCl and Ar	27.45(3)	236(4)	3.9(2)	Cub	108					
Shim et al. (2002)	Ar	27.45	255(5)	4	Tet	46					
Ono et al. $(2004)^{d}$	N/A ^f	27.63(2)	235(9)	4	Tet	106					
Sun et al. (2016) ^d	NaCl	27.340(6)	249(4)	4	Tet	151					
Chen et al. (2018)	Ne	27.820(6)	228(6)	4	Tet	62					
Thomson et al. $(2019)^d$	NaCl	27.76(4)	224(4)	4	Tet	13					
Sun et al. (2022)	NaCl	27.46(2)	227(21)	4.0(3)	Tet	200					
	/ .	CaSiC	D3 Computat	ion		/ .					
Chizmeshya et al. (1996)	N/A	27.47	227	4.29	Cub	N/A					
Jung and Oganov (2005)	N/A	28.24	219	4.08	Tet	N/A					
Caracas et al. (2005)	IN/A N/A	20.82	249	4.09	Tet	N/A N/A					
(2014)	10/A	27.01	205.5	4.70	Cub	1N/A					
(2014)	(2014) CaTiO2 Experiment										
Fischer et al (1993)	n-pentane	33.87	177(3)	5 1(8)	Orth	3					
(1990)	and methyl-	22.07	1// (3)	0.1(0)	orun	5					
	butane										
Kung and Rigden	n-pentane	34.17 (2)	175	5.78	Orth	3					
(1999)	and iso-	~ /									
. /	pentane										
Guennou et al. (2010)	N/A ^f	33.62 (7)	181.0 (6)	4	Orth	60					
Truffet et al. (2023) ^d	Ne	33.62 (7)	180.6 (4)	4	Orth	170					

634 Note: The numbers in parentheses are uncertainties for the last digit. Vo, Ko, and K' without

635 parentheses are assumed to be a fixed value.

- ^a Volume of per mole of chemical formula (calculated from unit cell volume if not reported).
- ^b The column of symmetry; Cub, Tet, and Orth represent cubic, tetragonal, and orthorhombic
- 638 perovskite structures, respectively.
- ^c Maximum pressure with experimental data coverage.
- ^d Works of thermal equation of state; data coverage at 300 K might not be dense.
- ^eMulti-anvil press experiment and thus, pressure medium is not comparable to diamond anvil cell
- 642 experiments.
- ^f Did not load pressure medium or the pressure medium was not clearly mentioned by the literature.
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