1	Revision 1: Ilmenite phase transformations in suevite from the Ries impact structure
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2	(Germany) record evolution in pressure, temperature, and oxygen fugacity conditions
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10	
11	Abstract Aggregates of ilmenite with varying amounts of rutile, ferropseudobrookite, and
12	pseudorutile in suevites from the Ries impact structure have been analyzed by light microscopy,
13	analytical scanning electron microscopy, electron microscope analysis, and Raman spectroscopy
14	to constrain their formation conditions. The 10s-100s µm aggregates comprise isometric ilmenite
15	grains up to 15 $\mu$ m in diameter that form a foam structure (i.e., smoothly curved grain boundaries
16	and 120° angles at triple junctions). Grains with foam structure show no internal misorientations,
17	indicating a post-impact formation. In contrast, ilmenite grains with internal misorientation
18	occurring in the core of the aggregates are interpreted as shocked remnant ilmenite originating
19	from the target gneisses. They can contain twin lamellae that share a common $\{11\overline{2}0\}$ plane with
20	the host and the <i>c</i> -axis is oriented at an angle of $109^{\circ}$ to that of the host. Similarly, the new grains
21	with foam structure display up to three orientation domains, sharing one common $\{11\overline{2}0\}$ plane

orientation relationship likely reflects a cubic supersymmetry resulting from the transformation of 23 the initial ilmenite upon shock (>16 GPa) to a transient perovskite-type high-pressure phase 24 (liuite), subsequent retrograde transformation to the polymorph wangdaodeite, and then back-25 transformation to ilmenite. Whereas the new grains with foam structure formed from complete 26 transformation, the twin domains in the shocked ilmenite are interpreted to represent only partial 27 28 transformation. Ferropseudobrookite occurs mostly near the rim of the aggregates. An intergrowth of ferropseudobrookite, ilmenite, and rutile as well as magnetite or rarely armalcolite 29 occurs at contact with the (devitrified) matrix. The presence of ferropseudobrookite indicates high 30 temperature (>1140°C) and reducing conditions. The surrounding matrix provided  $Mg^{2+}$  to form 31 the ferropseudobrookite-armalcolite solid solution. Rutile can occur within the aggregates and/or 32 along the ilmenite boundaries; it is interpreted to have formed together with iron during the 33 decomposition of ilmenite at lower temperatures (850°-1050°C). We suggest magnetite in the 34 rims formed by electrochemical gradients driven by the presence of a reducing agent, where Fe<sup>2+</sup> 35 within ilmenite diffused towards the rim. Subsequent cooling under oxidizing conditions led to 36 the formation of magnetite from the iron-enriched rim as well as pseudorutile around ilmenite 37 grains. 38

Our study demonstrates that the specific crystallographic relationships of ilmenite grains with foam structure indicate a back-transformation from high (shock) pressures >16 GPa; moreover, the presence of associated Fe-Ti oxides helps indicate local temperature and oxygen fugacity conditions.

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44 Keywords

45 ilmenite shock effects; transformation twinning; liuite; wangdaoedeite; ferropseudobrookite;
46 armalcolite

47

# 48 Introduction

Ilmenite ( $Fe^{2+}TiO_3$ ) is a common Fe-Ti-bearing opaque mineral in magmatic and metamorphic 49 rocks. Its deformation and transformation to ferrimagnetic phases during impact cratering can 50 change the magnetic properties of impactites (e.g., El Goresy 1964; El Goresy et al. 2010; 51 52 Dellefant et al. 2022), which in turn can result in large-scale magnetic anomalies in impact structures (Pilkington and Grieve 1992; Scott et al. 1997; Plado et al. 1999; Pilkington and 53 Hildebrand 2000; Ugalde et al. 2005; Pohl et al. 2010). Heating of ilmenite under oxidizing 54 conditions leads to the formation of hematite (Fe<sub>2</sub>O<sub>3</sub>), rutile (TiO<sub>2</sub>), and nFe<sub>2</sub>O<sub>3</sub> x nTiO<sub>2</sub> from 55 500°-800°C, whereas a complete transformation of ilmenite to pseudobrookite (Fe<sub>2</sub>TiO<sub>5</sub>) occurs 56 at temperatures above 800°C (Fu et al. 2010). Pseudorutile ( $Fe^{3+}_{2}Ti_{3}O_{9}$ ) forms as a transitional 57 58 phase from ilmenite alteration towards rutile (Teufer and Temple 1966), commonly observed under high oxygen fugacity and below 700°C (Gupta et al. 1991) in combination with iron 59 removal by a leaching agent, such as water, along grain boundaries (Mücke and Bhadra 60 61 Chaudhuri 1991) and/or structural discontinuities within the grain (Temple 1966; Grey and Reid 1975). 62

Under reducing conditions, ilmenite dissociates to elemental iron and rutile at temperatures ranging from 850° to 1050°C (Taylor et al. 1972) and forms ferropseudobrookite ( $Fe^{2+}Ti_2O_5$ ) at >1140°C (Lindsley 1965; Keil et al. 1970; Tuthill and Sato 1970; Sargeant et al. 2020), which is mostly found in extraterrestrial material (Keil et al. 1970; Fujimaki et al. 1981). Armalcolite [( $Fe^{2+},Mg$ )Ti<sub>2</sub>O<sub>5</sub>] can be formed from ferropseudobrookite when part of  $Fe^{2+}$  is substituted by

Mg<sup>2+</sup> (Anderson et al. 1970). High-temperature experiments with reducing agents can produce a 68 rim of pure iron around ilmenite grains (Gupta et al. 1990; Zhao and Shadman 1990) with various 69 70 amounts of rutile, ilmenite, and elemental iron (Gupta et al. 1990; Kucukkaragoz and Eric 2006; Sargeant et al. 2020). Ilmenite is reduced with carbon in an argon atmosphere at 1000°C, 71 resulting in Ti-oxides and finely distributed iron within the core and a rim of metallic iron (Gupta 72 73 et al. 1990), as well as in veinlets of Ti-oxides within the grain as similarly reported by H<sub>2</sub>reduced ilmenites (Sargeant et al. 2020). Experiments of synthetic ilmenite under reducing 74 conditions with CO between 900 and 1100°C resulted in a core of ilmenite encompassed by an 75 inner rim of rutile and an outer rim of pure iron (Zhao and Shadman 1990). The formation of the 76 iron-enriched rim is attributed to the migration of iron to the grain boundaries (Zhao and 77 Shadman 1990; Kucukkaragoz and Eric 2006), whereas finely distributed iron within the grain is 78 interpreted by a movement of a reducing agent into and out of the grain (Li et al. 2012; Dang et 79 al. 2015; Sargeant et al. 2020). 80

81 Liu (1975) described a high-pressure phase transition of ilmenite to an orthorhombic perovskite 82 structure at 14 GPa at 1400-1800°C based on X-ray diffraction data. Furthermore, Leinenweber et al. (1991) documented an unquenchable, high-pressure polymorph of ilmenite with a 83 perovskite structure forming at 16 GPa and  $\approx 27^{\circ}$ C using a diamond-anvil cell and in-situ powder 84 diffraction. Upon decompression, the high-pressure phase transformed into a meta-stable FeTiO<sub>3</sub> 85 86 phase with a lithium niobate structure (Mehta et al. 1994). Further studies documented these high-pressure phase transitions as a function of temperature [7-10 GPa at 1327-1527°C (Mehta et 87 al. 1994); 10-20 GPa at ≈400°C (Ming et al. 2006); 20-30GPa at ≈1337°C (Nishio-Hamane et al. 88 89 2012)]. The orthorhombic perovskite-type FeTiO<sub>3</sub> phase named liuite was reported by Ma and Tschauner (2018) from the Shergotty martian meteorite. Xie et al. (2020) described a natural 90 occurrence of the lithium niobate-type FeTiO<sub>3</sub> named wangdaodeite from the Suizhou L6 91

92	chondrite. Wangdaodeite was reported for the first time in naturally occurring rocks from
93	Zipplingen in the Ries impact structure by Tschauner et al. (2020).

Further shock experiments with pressures ranging from 23 to 53 GPa reported quenchable orthorhombic Fe<sub>2</sub>TiO<sub>4</sub> (Hashishin et al. 2014), Fe<sub>2</sub>TiO<sub>4</sub> (Ca<sub>2</sub>TiO<sub>4</sub>-Type) + TiO<sub>2</sub> and wüstite (FeO) + TiO<sub>2</sub> (Nishio-Hamane et al. 2012), wüstite and orthorhombic FeTi<sub>3</sub>O<sub>7</sub> (Nishio-Hamane et al. 2010), as well as Fe<sub>1- $\delta$ </sub>Ti<sub> $\delta$ </sub>O and Fe<sub>1+ $\delta$ </sub>Ti<sub>2- $\delta$ </sub>O<sub> $\delta$ </sub> (Wu et al. 2009). Based on hybrid-density functional modelling, Wilson et al. (2005) predicted a phase transition of FeTiO<sub>3</sub> above 65 GPa to TiO<sub>2</sub> (cotunnite) + FeO (wüstite).

Studies of ilmenite shock effects in ilmenite from natural impactites, however, are relatively 100 101 scarce (Minkin and Chao 1971; Sclar et al. 1973; Syono et al. 1981; Tschauner et al. 2020; 102 Dellefant et al. 2022). This study aims to shed light on the potential of ilmenite as a recorder of temperature, pressure, and oxygen fugacity conditions in impactites based on microfabric and 103 phase relation analyses. To this aim, we investigated 35 ilmenite aggregates within suevites from 104 the Ries impact structure (Zipplingen and Aumühle) with optical light microscopy (transmitted 105 and reflected), Raman spectroscopy, and scanning electron microscopy (SEM) in combination 106 107 with electron backscatter diffraction (EBSD), as well as electron microprobe (EMP) 108 measurements.

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# 110 Geological overview

The ≈26 km diameter Ries impact structure formed about 14.8 Ma ago (Rocholl et al. 2017, 2018;
Schmieder et al. 2018a, 2018b; Schwarz et al. 2020; Di Vincenzo 2022). The Ries impactor hit a
600-700 m thick, sedimentary layer of limestones, shales, and sandstones that overlay Variscan

basement (Bolten and Müller 1969; Pohl et al. 1977) mainly composed of various paragneisses, 114 amphibolites, and metagranites (Schmidt-Kaler 1969; Pohl et al. 1977; Hüttner and Schmidt-115 116 Kaler 1999; Stöffler et al. 2013). The Ries structure can be divided into the inner crater, the inner wall [being presumably the rim of the primary inner crater, (Wünnemann et al. 2005)], the 117 megablock zone, and the outer crater rim (Stöffler et al. 2013). Suevite constitutes the most 118 119 important ejecta blanket that contains polymict impact breccia bearing both lithic clasts and cogenetic melt particles (Stöffler and Grieve 2007; Stöffler et al. 2013). On a microscopic scale, 120 (devitrified) glass represents a heterogeneous distribution of incompletely mixed melts of 121 different compositions interspersed with vesicles indicating gas phases (von Engelhardt et al. 122 1995). Based on the occurrence of high-temperature phases such as baddeleyite + SiO<sub>2</sub> formed 123 from zircon, fused SiO<sub>2</sub> formed from quartz, or pseudobrookite and rutile formed from ilmenite 124 (El Goresy and Chao 1976), von Engelhardt et al. (1995) suggested the initial melt temperature 125 exceeded 2000°C. Glass fragments in suevites were reported to contain ilmenite with 126 127 accompanying rutile, magnetite, and pseudobrookite (El Goresy 1964), as well as armalcolite and polycrystalline ilmenite (El Goresy and Chao 1976). 128

Here, we studied ilmenite aggregates in suevite from a small outcrop between the outer and inner ring in the northwestern part of the Ries impact structure, close to the village of Zipplingen (sample CT827a, Lat 48.92615°, Long 10.40889°) and in a shocked gneiss clast within suevite sampled close to the contact of the underlying bunte breccia in the Aumühle quarry at the northeastern rim of the structure (sample R20-16A, Lat 48.97151°, Long 10.62908°).

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#### 135 Methods

Uncovered and polished thin sections (25 μm) were prepared from the samples and investigated
by polarization microscopy (Leica DM2700 P) using both reflected and transmitted light.
Photomicrographs were taken with a Leica MC170 HD camera and processed with the Leica
Application Suite X 3.08.19082 software.

In-situ micro-Raman spectroscopy was carried out at the Mineralogical State Collection Munich 140 (SNSB) to identify the iron-bearing phases with a HORIBA JOBIN YVON XploRa ONE system. 141 The Raman spectrometer is equipped with a Peltier-cooled CCD detector and edge filters. An 142 143 1800g/mm grating was used in combination with a 532 nm 2ω-Nd:YAG laser on iron-bearing phases in an attenuated mode (10% laser power) corresponding to max. 0.9 mW on the sample 144 surface to avoid oxidation (Bauer et al. 2011; Kaliwoda et al. 2021), which could transform 145 146 magnetite to hematite thereby leading to a false phase determination. Hole and slit diameters were  $300 \ \mu\text{m}$  and  $100 \ \mu\text{m}$ , respectively, using an integration time of 2 x 16 s. On the sample surface, a 147 100x long working distance objective resulted in a 0.9 µm laser spot size. Wavelength calibration 148 was conducted with a pure Si-wafer chip on the predominant  $520\pm1$  cm<sup>-1</sup> peak. The precision in 149 the Raman peak position is estimated at  $\pm 1.5$  cm<sup>-1</sup>. 150

The samples were studied with a Hitachi SU5000 SEM, equipped with a NordlysNano high-151 sensitivity EBSD detector (Oxford Instruments), energy-dispersive X-ray spectroscopy (EDS) 152 detector (Oxford Instruments), field emission gun, as well as backscattered electron (BSE) and 153 154 secondary electron (SE) detectors at the Department for Earth and Environmental Sciences, LMU Munich. Thin sections were chemo-mechanically polished with a colloidal silicon suspension 155 (Syton) for EBSD to reduce the surface damage produced from preparation. SEM observations 156 157 used accelerating voltages of 20 kV and a working distance ranging from 10 to 25 mm. The sample holder was tilted at 70° with respect to the electron beam. EBSD patterns were acquired 158

and automatically indexed with the AZtec analysis software 4.2 (Oxford Instruments). Step sizes 159 for automatic mapping were between 0.5 µm and 2 µm, depending on the resolution and the 160 161 desired EBSD-pattern quality. EBSD data were processed by the Channel software 5.12.74.0 (Oxford Instruments). The inferred different crystallographic orientations were analyzed by 162 characterizing the smallest angle required to rotate one orientation into another, i.e., by the 163 164 misorientation angle and the misorientation axis [e.g., Wheeler et al. (2001)]. Grains are detected by a misorientation angle threshold of 10°. Stereograms in the proceeding figures are equal-angle 165 projections of the lower hemisphere with viewing directions identical to the EBSD map. 166

The chemical compositions of ilmenite, ferropseudobrookite, armalcolite, rutile, and pseudorutile were measured using a Cameca SX-100 electron microprobe (EMP) at the Department for Earth and Environmental Sciences, LMU Munich. As measurement conditions, 15 kV accelerating voltage, 40 nA sample current, and a focused beam were applied. Calibration was performed using silicate and oxide standards: periclase (Mg), albite (Na), bustamite (Mn), Fe<sub>2</sub>O<sub>3</sub> (Fe), wollastonite (Ca, Si), rutile (Ti), and orthoclase (K, Al).

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#### 174 **Results**

## 175 Sample description

The investigated suevite sample from Zipplingen (CT827a) consists of  $\approx 85\%$  (devitrified) glass, in which roundish to elongate aggregates of Fe-Ti-oxides with a long axis of 10s to 100s of  $\mu$ m are homogeneously distributed (Fig. 1a, b). The aggregates consist mostly of ilmenite with various amounts of rutile, ferropseudobrookite, and sphene. Magnetite occurs as  $\mu$ m-sized grains dispersed in the matrix and often occurs as a rim around the Fe-Ti-oxide aggregates (Fig. 1a).

Calcite grains (≈15%) can have (devitrified) glass as inclusions (Fig. 1b, yellow arrows) and calcite can be included in (devitrified) glass (Fig. 1b, orange arrows). Irregular phase boundaries of calcite are concave with respect to the glass matrix, where the sharp point of the cusps of the silicate glass-calcite phase boundary points towards the silicate glass (Fig. 1b).
Sample R20-16A from the Aumühle quarry can macroscopically be recognized as a gneiss clast within suevite by its lighter color and foliation. In polarized light, it is characterized by a

187 cryptocrystalline matrix with <10% vesicles (Fig. 1c, d). Elongate ilmenite aggregates with a long 188 axis of several 10s to 100s of  $\mu$ m, as well as secondary hematite, occur homogeneously 189 distributed in the matrix (Fig. 1c). Minerals in the gneiss clast were almost completely 190 amorphized due to the meteorite impact, as evidenced by the cryptocrystalline matrix (Fig. 1d), 191 indicating shock pressures of 35-45 GPa, representing shock stage F-S5 (Stöffler et al. 2018) or II 192 (IUGS system; Stöffler and Grieve 2007).

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# **194 Phase determination of Fe-Ti-oxides**

The Fe-Ti bearing phases in the aggregates were determined by optical reflected light microscopy 195 and confirmed by Raman spectroscopy, chemical EDS and EMP analyses, as well as EBSD 196 patterns. Ilmenite has the main Raman peaks at 227, 333, 371, and 449 cm<sup>-1</sup> with the prominent 197 peak at 683 cm<sup>-1</sup> (Fig. 2a) (Wang et al., 2004). The rutile spectra have main Raman peaks at 240. 198 446, and 611 cm<sup>-1</sup> (Fig. 2b) (Mazza et al., 2007). Ferropseudobrookite has prominent Raman 199 peaks at 134, 204, 317, and 643 cm<sup>-1</sup> and a relatively wide peak at 773 cm<sup>-1</sup> (Fig. 2c), in 200 201 accordance with spectra of isostructural karooite (MgTi<sub>2</sub>O<sub>5</sub>) (Liermann et al. 2006; He et al. 2017) in combination with a Fe/Ti ratio of  $\approx 1:2$  based on EDS and EMP analysis. Peak 202 broadening is associated with a disordered crystal structure (Cynn et al. 1992; Liermann et al. 203

2006). Magnetite shows typical Raman peaks at 294 and 530 cm<sup>-1</sup> with the main peak at 661 cm<sup>-1</sup> 204 (Fig. 2d) (Wang et al., 2004). Within the matrix, magnetite spectra display peaks at 538 and 666 205 cm<sup>-1</sup> (Fig. 2e) and thus differ slightly from magnetite at the Fe-Ti-oxide aggregate rim (compare 206 Fig. 2d). We observed additional peaks from the crystallized matrix at 324, 368, and 392 cm<sup>-1</sup> and 207 a strong peak at 1002 cm<sup>-1</sup> (Fig. 2e). Sphene displays the main Raman peaks at 166, 254, 335. 208 468, 548, and 608 cm<sup>-1</sup> (Fig. 2f) in agreement with Pantić et al. (2014). Within the Aumühle 209 suevite sample (R20-16A), ilmenite has major Raman peaks at 224, 329, 372, and 681 cm<sup>-1</sup> (Fig. 210 2g) (Wang et al., 2004). The Raman peaks of pseudorutile are very broad and overlapping, with 211 main peaks at about 153, 264, 312, 419, 557, 599, 693, and 823 cm<sup>-1</sup> (Fig. 2h) as reported by 212 213 Imperial et al. (2022).

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# 215 Microfabrics of ilmenite and associated Fe-Ti-phases in suevite from Zipplingen

Ilmenite within the Zipplingen suevite (CT827a) has a composition ranging from Fe0.88 216 Mn<sub>0.08</sub>Mg<sub>0.06</sub>Ti<sub>0.98</sub>O<sub>3</sub> to Fe<sub>0.88</sub>Mn<sub>0.12</sub>Mg<sub>0.02</sub>Ti<sub>0.98</sub>O<sub>3</sub> (Fig. 3a-g; Tab. 1) with small amounts of 217  $Al_2O_3$ . Chemical profiles from the core of the ilmenite aggregate towards the rim conducted with 218 the EMP show an increase in the Fe/Ti ratio (0.83 - 1.03), which correlates with an increase in 219 the Al<sub>2</sub>O<sub>3</sub> content (0.07-0.33 wt.%) and a decrease of the analytical total (98.56-96.92 wt.%) (Fig. 220 3g, h; Tab. S1). The deviation from a total of 100 wt.% of ilmenite can be explained by the 221 presence of a Fe<sub>2</sub>O<sub>3</sub> component, which decreases the total wt.% when expressed as FeO. Rutile 222 223 occurs with variable amounts of FeO (Tab. S1: #1-#4), up to 20 µm in diameter in the cores of 224 the aggregates (Fig. 3a, g) or distributed as µm-sized grains along ilmenite boundaries (Fig. 3c, 225 d). (Sub-)euhedral magnetite grains,  $<5 \mu m$  in diameter, can occur at the rim or within the matrix at a close distance to the ilmenite aggregate (Figs. 3a, c, e, f, 4a). 226

227 We distinguish two different types of ilmenite grains in the aggregates based on their internal misorientations, grain shape, and composition (Figs. 3-6): (i) Small, isometric grains of up to 15 228 µm in diameter show smoothly curved grain boundaries and 120° angles at triple junctions, 229 forming a so-called foam structure (Fig. 3a, b, d). They have generally low internal 230 misorientations with angles  $<3^{\circ}$  (Figs. 4b, c, e, 6b). (ii) In contrast, in the core of the aggregates 231 232 coarse ilmenite grains can occur that have a short wavelength lamellar internal misorientation with misorientation angles of  $>10^{\circ}$  (Figs. 4 c, d, 5a). These deformed grains can contain a few  $\mu$ m 233 wide twin lamellae that are characterized by a misorientation angle of  $\approx 75^{\circ}$  around a 234 misorientation axis parallel to  $[\overline{2}110]$ , which corresponds to *c*-axes at an angle of  $\approx 109^{\circ}$  and a 235 common  $\{11\overline{2}0\}$  plane (Fig. 5a, c, e). This crystallographic relationship does not correspond to 236 the known (0001),  $\{10\overline{1}1\}$ , and  $\{10\overline{1}2\}$  twin systems of ilmenite (Minkin and Chao 1971; Sclar 237 et al. 1973; Syono et al. 1981; Dellefant et al. 2022). In some aggregates, only isolated ilmenite 238 grains with internal misorientations are preserved (Fig. 6b, white arrows). The deformed 239 240 ilmenites have similar crystallographic orientations within single aggregates (Fig. 6e, f).

The ilmenite grains with foam structure and low internal misorientation (Fig. 4a-c, 5a, 6) can 241 242 show crystallographic orientations scattering largely around that of the deformed ilmenite with internal misorientations (Fig. 4d, e, 5a, c, e, 6). The orientations depict maxima that correspond to 243 the twin relationship, i.e., they share a common  $\{11\overline{2}0\}$  plane and *c*-axes that are at an angle of 244 ≈109° (Figs. 5, 6f). Up to three distinct orientation domains were observed with 109° or 99° 245 246 angles, respectively, between the c-axes (Fig. 5b, d, f, 6f). The grains with foam structure can 247 show varying grain sizes (Figs. 5b, 6a, b). Larger grains (>10  $\mu$ m) are commonly concave and have a high number of neighbors (>6, Fig. 6a, white arrows), and smaller new grains are mostly 248 convex and have fewer neighbors (<6), which suggests some grain boundary migration driven by 249

250	the reduction in interfacial free energy, i.e., grain growth (e.g., Nicolas and Poirier, 1976; Evans
251	et al., 2001). Strain-free grains with foam structure can contain minor amounts of alumina (0.02-
252	0.11 wt.%), whereas the deformed grains are alumina-free (Tab. 1).

Ferropseudobrookite occurs at the boundary of the aggregate with the matrix, or along cracks 253 (Fig. 7b-f), and is locally associated with sphene (CaTi[O|SiO<sub>4</sub>]) (Figs. 7b, c, 8b), with grains 254 diameters of up to 10s of µm in diameter (Figs. 7b-f, 8a, b, d). Ferropseudobrookite displays a 255 symplectitic intergrowth with ilmenite and rutile towards the core of the aggregate (Fig. 7c-e) 256 257 with random orientations (Fig. 8d, f), whereas ilmenite displays one maximum but also dispersed crystallographic orientations (Fig. 8c, e) and can occur locally as a fine-grained rim around the 258 aggregate (Fig. 8b, c). In contrast to ilmenite, ferropseudobrookite has minor Mn concentrations 259 260 (Fig. 8b, g). Locally, at the Fe-Ti-oxide boundary towards the matrix, Mg-enrichment indicates 261 compositions within the ferropseudobrookite-armalcolite solid solution  $(Fe_{1.02})$ 1). 262  $Mn_{0.04}Mg_{0.13}Ti_{1.79}Si_{0.09}Al_{0.02}Ca_{0.01}Na_{0.01}O_5;$ Fig. 8h: Tab. The occurrence of 263 ferropseudobrookite and/or rutile in the aggregates correlates with the presence of µm-sized (sub-264 )euhedral magnetite at the rim or a close distance within the matrix (Fig. 7f).

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#### 266 Microfabrics of ilmenite and associated Fe-Ti-phases in suevite from Aumühle

Within sample R20-16A from the Aumühle quarry, ilmenite aggregates have an average 267 composition of  $Fe_{0.92}Mg_{0.04}Mn_{0.04}TiO_3$ , with a very similar polycrystalline microfabric as 268 Zipplingen sample CT827a, possessing a foam structure of single grains up to 15 µm in diameter 269 270 (Figs. 3a, b, 9a) with intergranular cracks (Fig. 9b). No ferropseudobrookite was detected, instead, pseudorutile can be present as µm-sized rim surrounding single ilmenite grains, as larger 271 crystals with some ilmenite remnants in the core (Fig. 9c, d), or completely replacing ilmenite 272

with elongated pores exhibiting a preferred orientation (Fig. 9e). A few aggregates have a  $\leq 5 \mu m$ wide rim of 0.5  $\mu m$  x 2 $\mu m$ -sized intergrown ilmenite needles (Fig. 9f).

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# 276 Distribution of aggregates within the samples

Of 35 ilmenite orientation investigations within aggregates from Zipplingen and Aumühle, three display three distinct maxima with a common  $\{11\overline{2}0\}$  plane, eight display two orientation maxima with a common  $\{11\overline{2}0\}$  plane, and ten display one weak to moderate maxima, whereas 14 display only dispersed orientations. Three aggregates show coarse irregular ilmenite with short wavelength lamellar misorientations, in two of these aggregates the deformed ilmenites have twin lamellae.

In total, 23 aggregates of Fe- and/or Ti-bearing phases within the Zipplingen suevite (CT827a) were investigated with SEM/EBSD, where ilmenite is always present (Table 2). Aggregates range from 10  $\mu$ m x 10  $\mu$ m to 160  $\mu$ m x 80  $\mu$ m in size and often show a distinct rim (n=15) comprised of different Fe-Ti-oxides. We observed no size or distribution dependence of aggregates with (10  $\mu$ m x 10  $\mu$ m - 160  $\mu$ m x 80  $\mu$ m) or without rims (40  $\mu$ m x 20  $\mu$ m - 160  $\mu$ m x 50  $\mu$ m), as well as no preferred occurrence of distinct phases as a function of size. The aggregates are distributed homogeneously throughout the thin section.

From Aumühle (R20-16A), 12 aggregates of Fe- and/or Ti-bearing phases were investigated with SEM/EBSD where ilmenite was consistently present (Table 3). Aggregates range from 10  $\mu$ m x 5  $\mu$ m to 180  $\mu$ m x 120  $\mu$ m in size and can display rims (n=4) composed of ilmenite. The aggregates are distributed homogeneously throughout the thin section.

#### 295 **Discussion**

Ilmenite grains from Variscan gneisses shocked to low pressure-temperature conditions from the 296 Ries impact structure show neither a comparable foam structure (Figs. 3a, b, 4, 5, 6) nor short 297 wavelength lamellar misorientation patterns, nor twinning (Figs. 4, 5a), as described in this study. 298 The coarse ilmenite with short wavelength lamellar internal misorientation (Figs. 4, 5a) is 299 interpreted to be preserved shocked ilmenites originating from the target gneisses. The strain-free 300 grains with foam structure (Figs. 3a, b, 4, 5) document an equilibrium fabric and some grain 301 302 growth, which is indicative of isostatic conditions during their formation (e.g., Nicolas and Poirier, 1976; Evans et al. 2001). Therefore, they are interpreted to have formed after the shock 303 event. Possible processes for the formation of the new grains, which will be discussed in the 304 305 following, might be (i) strain-induced grain boundary migration, i.e., recrystallization of the originally shocked ilmenite, (ii) crystallization from a Fe-Ti-oxide glass or melt formed upon 306 shock loading and unloading or (iii) phase transformations from transient high-pressure phases 307 308 formed during shock compression.

Similar ilmenite aggregates as those studied here have been found in kimberlite pipes (Pasteris 309 1980; Haggerty 1991; Kostrovitsky et al. 2020), which were interpreted as having formed from 310 cumulates in the deep mantle that were deformed and recrystallized during ascent to the crust 311 (Frick 1973; Mitchell 1973). Recrystallization, i.e., grain boundary migration driven by the 312 313 reduction in strain energy, to replace the shocked original ilmenite grain would result in sutured grain boundaries (Drury and Urai 1990), which is rarely observed for some remnant shocked 314 grains (Fig. 6a, b). In contrast, the foam structure indicates that the reduction in interfacial free 315 energy was the driving force for the grain boundary microstructure formation (e.g., Nicolas and 316 317 Poirier, 1976; Evans et al. 2001). Furthermore, recrystallization cannot explain the specific

crystallographic relationships with up to three distinct orientation maxima with a common { $11\overline{2}0$ } plane, comparable to the twin relationship of the shocked remnant grains (Fig. 5b, d, f, 6). Nor can recrystallization explain the occurrences of the different Fe-Ti-phases. The same holds for crystallization of ilmenite grains from a glass or melt. Foam-structured ilmenite aggregates also occur in the suevite sample from Aumühle, which does not display high-temperature phases, such as ferropseudobrookite or pseudobrookite, indicating that temperatures were insufficient for melting.

We suggest that (1) the ilmenite aggregates with foam structure and (2) specific crystallographic relationship similar to the twin relationship in remnant shocked grains, both reflect phase transformations from transient high-pressure phases that were generated upon shock compression and decompression, as discussed below. We then discuss the formation conditions of the other Fe-Ti phases in the aggregates.

330

# 331 Ilmenite transformation twins and associated orientation relationship between new grains

The observed twin lamellae in shocked ilmenite with a misorientation angle of  $\approx 75^{\circ}$  around a misorientation axis parallel to [ $\overline{2}110$ ], and a common { $11\overline{2}0$ } plane (Fig. 5a, c, e) has not been reported for ilmenite so far. Shocked ilmenites in nature and laboratory experiments generate (0001), { $10\overline{11}$ }, and { $10\overline{12}$ } twins (Minkin and Chao 1971; Sclar et al. 1973; Syono et al. 1981; Dellefant et al. 2022). No similar mechanical twins are observed in our study.

The two observed twin domains (host and lamellae) in Figure 5a, c, e are characterized by an angle of  $\approx 109^{\circ}$  between the *c*-axes, i.e., the threefold axes (Fig. 5e); this angle is very close to the value of the tetrahedral angle between threefold axes in the cubic system (109.47°). The threefold

axes of the two twin domains in Figure 5e span a plane which is perpendicular to the apparent 340 interface plane of the host and the twin (Fig. 5a, c, e). The crystallographic preferred orientations 341 of the aggregates with foam structure (Fig. 5f) are characterized by three orientation domains, 342 where the c-axes enclose similar angles of 109° and 99°, respectively. Furthermore, each of the 343 three distinct orientation domains shows one common  $\{11\overline{2}0\}$  plane with the other two domains. 344 reflecting the same twin operation as in Figure 5a, c, e that maps any two of the domains onto 345 each other. These characteristics suggest a twinning process related to a cubic supersymmetry: 346 While two of the three orientation domains have a common  $\{11\overline{2}0\}$  plane, the deviation from a 347 cubic matrix does not allow a third domain to have a common pole with the other two 348 simultaneously (e.g., Bueble et al. 1998; Hahn and Klapper 2006; Janovec and Přívratská 2006), 349 explaining the 99° angle between the c-axes of two of the observed orientation domains. The 350 space group of ilmenite is  $R\overline{3}$  (No. 148) (Barth and Posnjak 1934) corresponding to point group 351  $\overline{3}$ . A twin operation must not be element of the space- or point group of the twinned phase and the 352 twin operation is, accordingly, not an element of  $\overline{3}$ . The observed twin operation can be expressed 353 as (a) a 180° rotation around the bisecting angle axis of the c-axes of the twin pair (Fig. 5e) 354 combined with (b) a 180° rotation around the c-axis of one of the twins. Further, combining the  $\overline{3}$ 355 point group symmetries of the ilmenite twin domains in their observed mutual orientation with 356 the (a) 180° rotation, leads to a compound supersymmetry equal to the cubic crystallographic 357 point group m $\overline{3}$ m (Fig. 10). The order of point group m $\overline{3}$ m is 48, whereas the order of point group 358  $\overline{3}$  is 6. Therefore, eight (48/6 = 8) equivalent twin domains are theoretically possible. The 359 rhombohedral  $\overline{3}$  axis can be chosen along any of the four <111> directions of the cubic 360 compound supersymmetry. Moreover, for each choice of <111><sub>cub</sub> direction, there are two 361 choices for the orientation of the  $\vec{a}$ ,  $\vec{b}$ -axes, since m $\overline{3}$ m also has a threefold rotation axis parallel 362

to the  $\overline{3}$  roto-inversion. In the case of the pole figure shown in Figure 5e, the projection is approximately along a fourfold axis of the supersymmetry group m $\overline{3}$ m with only two of the 8 theoretically equivalent twin states realized in the investigated ilmenite grain. In Figure 5f, however, the projection is approximately along a threefold axis of the m $\overline{3}$ m compound supersymmetry and three orientation domains out of the 8 possible orientation domains are present. The Bain lattice-correspondence matrix between the rhombohedral and the cubic lattice, expressed for one out of the 8 possibilities is:

370 
$$(\vec{a} \quad \vec{b} \quad \vec{c})_{cub} = (\vec{a} \quad \vec{b} \quad \vec{c})_{hex} \begin{pmatrix} -1/3 & -1/3 & -2/3 \\ +1/3 & -2/3 & -1/3 \\ +1/6 & +1/6 & -1/6 \end{pmatrix} (Eq. 1)$$

371 
$$(\vec{a} \quad \vec{b} \quad \vec{c})_{hex} = (\vec{a} \quad \vec{b} \quad \vec{c})_{cub} \begin{pmatrix} 1 & 0 & 2 \\ 0 & -1 & 2 \\ 1 & -1 & -2 \end{pmatrix} (Eq. 2)$$

372 (the columns of the matrix give the new lattice basis vectors in terms of the old)

Twins where the twin element is a symmetry element of a supergroup G of the space- or point group H of the twinned phase are frequent in cases where a phase transition occurs from a highsymmetry phase of symmetry G (usually at high temperature) to a low-symmetry phase with subgroup symmetry H of G. They are called transformation twins (Janovec and Přívratská 2006).

A high-pressure polymorph of ilmenite, liuite, with an orthorhombic (space group Pnma, No.62) perovskite structure (Ma and Tschauner 2018) is reported to form at pressures >16 GPa (Liu 1975; Leinenweber et al. 1991). The Pnma perovskite structure results from a simple displacive distortion of the ideal cubic Pm $\overline{3}$ m perovskite structure (Fig. 10) as similarly discussed by Wang et al. (1991). Upon decompression, liuite transforms first to wangdaodeite (Xie et al. 2020), which is a meta-stable FeTiO<sub>3</sub> with a lithium niobate structure, space group R3c (No.161) (Abrahams et al. 1966; Weis and Gaylord 1985; Mehta et al. 1994; Ma and Tschauner 2018) and

a related generation of transformation twins due to the many symmetry-equivalent ways of 384 forming wangdaodeite from liuite. Moreover, Wangdaodeite finally transforms to ilmenite (space 385 group  $R\overline{3}$ ) with resulting transformation twins reported in ilmenite (Leinenweber et al. 1994; 386 Mehta et al. 1994). Typically, transformation twins result from phase transitions of the displacive 387 or order-disorder type (e.g., Bueble et al. 1998; Hahn and Klapper 2006; Janovec and Přívratská 388 2006). However, the ilmenite, LiNbO<sub>3</sub>, and perovskite structure types are topologically different 389 but nevertheless have related unit cell geometries (Fig. 10). Therefore, the observed twins are 390 391 interpreted as transformation twins generated by phase transitions with some reconstructive character, albeit with a topotactic relationship between the lattices involved. The experimentally 392 observed prograde and retrograde phase transition sequence (Wang et al. 1991; Leinenweber et 393 al. 1994; Mehta et al. 1994) is sketched in Figure 10a. Leinenweber et al. (1994) argued that as a 394 result of the retrograde perovskite-type to LiNbO<sub>3</sub>-type phase transition, the mirror planes 395 (100)<sub>orth</sub>, (010)<sub>orth</sub>, and (001)<sub>orth</sub> in the Pnma perovskite point group disappear, with the latter 396 causing the formation of twinning as similarly discussed by Wang et al. (1991). We follow 397 essentially a similar line of argument and extend it by the new information of the 109° angle 398 between the *c*-axes of the twin domains: additionally, it must be considered that the symmetry of 399 the orthorhombic perovskite liuite Pnma (point group mmm) is also a subgroup of the cubic 400 perovskite symmetry  $Pm\bar{3}m$  (point group  $m\bar{3}m$ ). Thus, the  $m\bar{3}m$  symmetry is a common 401 supergroup of the lattices of all involved phases (Fig. 10b) and becomes the inherent compound 402 symmetry of the twinning process, without an actual cubic perovskite phase necessarily needing 403 404 to occur during the impact or the retrograde process. Therefore, the transformation from ilmenite 405 to liuite upon shock compression and transformation to wangdaodeite and finally ilmenite upon 406 unloading (Fig. 10) explains the observed twinning in shocked ilmenite and orientation 407 relationship of new grains in our EBSD maps.

Wangdaodeite has recently been detected in gneiss inclusions within suevites from Zipplingen, 408 the same locality as our samples, where the documented BSE images of Tschauner et al. (2020, 409 Fig. 1b) show similar microstructures, as those analyzed here (Figs. 3, 7, 8a). However, with 410 Raman spectroscopy, we did not detect wangdaodeite as there was no indication of the 411 characteristic peak of  $739\pm1$  cm<sup>-1</sup> compared to the Raman spectrum of ilmenite (Xie et al. 2020). 412 413 The metastability of naturally occurring wangdaodeite in the Ries structure was suggested by Tschauner et al. (2020) due to doping of a few mol.% of MnTiO<sub>3</sub>, which increases the kinetic 414 barrier for a back-transformation to ilmenite. In contrast to the ilmenite-wangdaodeite-bearing 415 gneiss inclusions investigated by the latter authors, our sample is comprised of ilmenites within a 416 (devitrified) glass matrix indicating higher temperatures, which could favor the back-417 transformation to ilmenite despite a few mol.% of MnTiO<sub>3</sub> being present (Fig. 8h; Tab. 1). 418

Similar microstructures as observed in our study have been documented within shocked zircons, 419 where the high-pressure phase reidite occurs as lamellae or, with increasing shock pressures, 420 421 replaces the entire grain. For the latter, back-transformation to µm-sized granular zircon leads to a systematic orthogonal disorientation relationship of  $\approx 90^{\circ}$  around < 110 > directions between 422 adjacent granules, topotactically controlled by the crystallographic structure of the reidite, leading 423 424 to a similar orientation relationship of the metastable reidite lamellae and the host zircon (Erickson et al. 2017; Timms et al. 2017; Cavosie et al. 2018). Similarly, the foam structure of 425 426 ilmenite with distinct orientation maxima of the crystallographic orientations of ilmenite grains can be used as a geobarometer to indicate pressure conditions >16 GPa realized for example in 427 428 impactites and kimberlites.

429

## 430 Ferropseudobrookite and armalcolite formation conditions

Ferropseudobrookite (Fe<sup>2+</sup>Ti<sub>2</sub>O<sub>5</sub>) requires >1140°C and a low oxygen fugacity to form (Lindsley 1965; Keil et al. 1970; Tuthill and Sato 1970; Sargeant et al. 2020). In contrast, ferrous-ferric pseudobrookite solid solution forms >800°C under oxidizing conditions (Fu et al. 2010). Therefore, in the Zipplingen sample (CT827; Figs. 7c-f, 8b, d), we interpret that the ferropseudobrookite at the boundary of the Fe-Ti-oxide near the matrix, representing quenched melt, formed from ilmenite as a result of shock-heating after shock-unloading at >1140°C and low oxygen fugacity according to:

438 
$$2\text{FeTiO}_3 \rightarrow \text{FeTi}_2\text{O}_5 + \text{FeO} (\text{Eq. 3})$$

Maximum temperatures could have exceeded the FeO-TiO<sub>2</sub> melting point of ≈1400°C (Eriksson
and Pelton 1993) locally at the aggregate boundary towards the silicate melt. However, the foam
structure of ilmenite grains with specific crystallographic relationship documents that solid-state
phase transformations dominated.

Armalcolite  $[(Fe^{2+},Mg)Ti_2O_5]$  is stable above  $1010\pm 20^{\circ}C$  at ambient pressure (Lindsley et al. 443 1974) with a pseudobrookite structure (Bowles 1988). Armalcolite was first documented in lunar 444 445 samples from the Apollo 11 (Anderson et al. 1970) and Apollo 17 missions (Haggerty 1973; El Goresy et al. 1974) and later on Earth in kimberlites (e.g., Haggerty 1975) and lower crustal / 446 mantle xenoliths (e.g., Grégoire et al. 2000; Hayob and Essene 1995; Spiridonov et al. 2019). El 447 Goresy and Chao (1976) first described the occurrence of armalcolite and associated ilmenite in 448 the Ries impact structure within glasses in suevite from diverse localities based on FeO/MgO of 449 21.1 to 4.5. Microscopic observation and microprobe data of a rutile core and an armalcolite rim 450 suggest a reaction of pre-existing rutile and silicate melt (El Goresy and Chao 1976). In our 451 study, chemical interaction of the silicate melt, represented by the (devitrified) matrix with the 452 Fe-Ti-oxide aggregates, presumably led to the formation of a rim of Mg-bearing 453

454 ferropseudobrookite-armalcolite solid solution formed from ilmenite (Fig. 8a, b, d, f, h), as also
455 suggested by El Goresy and Chao (1976):

456 
$$2(Fe,Mg)TiO_3 \rightarrow (Fe,Mg)Ti_2O_5 + FeO (Eq. 4)$$

457 An interaction of ferropseudobrookite with the surrounding silicate melt, where the latter 458 provides MgO in exchange for FeO could also produce armalcolite:

459 
$$FeTi_2O_5 + MgO \rightarrow (Fe,Mg)Ti_2O_5 + FeO (Eq. 5)$$

In each case, FeO would be enriched within the silicate melt close to the Fe-Ti-oxide aggregates. 460 Temperature stability experiments found that armalcolite decomposes unless rapidly quenched 461 462 (Lindsley et al. 1974). Adjacent to the Fe-Ti-oxide aggregate and within fractures, sphene can be present (Figs. 2f, 7c, 8b), which was also likely generated from a chemical reaction with the 463 silicate melt. In contrast to our observations, lunar armalcolites exhibit an inverse core-rim 464 relationship of ilmenite and armalcolite with the latter forming the core (El Goresy et al. 1974; 465 Smyth 1974; Kesson and Lindsley 1975), which therefore indicates different possible modes of 466 formation. 467

468

# 469 Formation of rutile within aggregates and magnetite outside aggregates

From the 23 Fe-Ti-oxide aggregates analyzed from Zipplingen, 10 display magnetite rims on the boundary towards the (devitrified) matrix and rutile within the aggregate core (Fig. 3, 7a-e, Tab.1). The phase assemblage resembles a Fe concentration gradient, where the rim is enriched in Fe, residing within  $\mu$ m-sized sub(-euhedral) magnetite (Fe<sub>3</sub>O<sub>4</sub>; Fig. 3a, g). Further inwards, the Fe content decreases to ilmenite (FeTiO<sub>3</sub>) and finally, Fe is absent, where rutile (TiO<sub>2</sub>) exists (Figs. 3a, g, 7a-e). Moreover, the occurrence of ferropseudobrookite and/or rutile within the Fe-Ti-oxide

aggregates (Figs. 7c-f, 8b, d) correlates with the presence of um-sized (sub-)euhedral magnetite at 476 the rim or in proximity to the (devitrified) glass. Only two ilmenite aggregates were observed 477 478 without magnetite rims — they contain neither rutile nor ferropseudobrookite (Tab. 2). We suggest that the association of µm-sized rutile along ilmenite grain boundaries included in the 479 aggregates (Figs. 3c, d, 7c-e) was generated due to the thermal decomposition of ilmenite to rutile 480 and Fe<sup>0</sup> after generation of the ilmenite aggregates, i.e., after shock unloading and under a low 481 oxygen fugacity at temperatures ranging from 850° to 1050°C (Taylor et al. 1972) along sites of 482 increased diffusivity like grain boundaries:

484 
$$\operatorname{FeTiO}_3 \rightarrow \operatorname{TiO}_2 + \operatorname{Fe} + \frac{1}{2}\operatorname{O}_2 (\operatorname{Eq.} 6)$$

483

However, a sub-solidus re-equilibration, where ferropseudobrookite first formed at grain 485 boundaries at the expense of ilmenite, with a subsequent back-transformation during cooling 486 below 1140°C (Lindsley 1965) could, furthermore, lead to the formation of ilmenite and rutile 487 (Haggerty and Lindsley 1969; Haggerty 1983; Anovitz et al. 1985): 488

489 
$$\operatorname{FeTi}_2O_5 \rightarrow \operatorname{FeTi}O_3 + \operatorname{Ti}O_2 (\operatorname{Eq.} 7)$$

Experiments under temperatures from 900° to 1350°C with ilmenite and reducing agents of C 490 (Gupta et al. 1990), CO (Zhao and Shadman 1990), or H<sub>2</sub> (Sargeant et al. 2020) generally form 491 492 elemental iron at the rim, which is interpreted as a migration of iron towards the grain boundaries (Zhao and Shadman 1990; Kucukkaragoz and Eric 2006). Furthermore, ilmenite, rutile, and iron 493 within the product aggregates can form through diffusion of a reducing agent into ilmenite along 494 vacancies and/or pathways through interstitial structures, such as grain boundaries (Merk and 495 Pickles 1988; Dang et al. 2015; Sargeant et al. 2020). 496

The presence of a reducing agent could donate electrons to ilmenite at the rim, where  $Fe^{2+}$  is 497 reduced to  $Fe^{0}$ . More available electrons would then generate a gradient attracting  $Fe^{2+}$  to diffuse 498 from within the aggregate, which agrees with fast  $Fe^{2+}$  diffusion in ilmenite modelled by 499 Kuganathan et al. (2019). As a result,  $Fe^{2+}$  diffuses towards the rim and vacancies towards the 500 core, thereby leaving TiO<sub>2</sub> inside the ilmenite aggregates, as observed in our study (Figs. 3a, g, 501 502 6a). Consistently, we observe an increasing Fe/Ti ratio for ilmenite from the core of the aggregates towards the rim (Fig. 3g, h; Tab. S1). Therefore, we interpret the process of iron 503 enrichment at the rims of the aggregates in terms of an electrochemical solid-state model, where 504 ilmenite aggregates serve as the anode and a reducing agent as the cathode, which generates an 505 electrochemical potential as a driving force for  $Fe^{2+}$  diffusion inside the ion-conducting ilmenite 506 (Kuganathan et al. 2019). 507

The high-temperature (>1140°C) phase transformation of ilmenite to ferropseudobrookite (Eq. 3) 508 only occurs at a low oxygen fugacity and results in the formation of FeO (Lindsley 1965), which 509 510 was not observed throughout the sample. We suggest that the absence of FeO together with the pores within ferropseudobrookite (Fig. 7c-e) as well as magnetite along the rim (Fig. 7b, e) are 511 also the result of  $Fe^{2+}$  migration towards the Fe-Ti-oxide aggregate – melt/matrix boundary. In 512 general, the oxygen fugacity and thus the  $Fe^{3+}/Fe^{2+}$  ratio increases with decreasing temperature in 513 514 silicate melts (Sack et al. 1981; Kilinc et al. 1983). Therefore, oxidation of elemental iron along 515 the Fe-Ti-oxide boundaries could form magnetite at intermediate oxygen fugacity conditions, as experimentally produced from 400° to 700°C (Mitchell et al. 1982; Kuroda and Mitchell 1983). 516 The experiments of Kuroda and Mitchell (1983) produced dispersed oriented magnetite as the 517 518 first iron oxide phase to form, as observed in our study:

519 
$$3Fe + 2O_2 \rightarrow Fe_3O_4$$
 (Eq. 8)

520 The formation of armalcolite leads to a local enrichment of FeO in the silicate melt (Eqs. 4, 5), 521 which likely contributed to the (sub-)euhedral magnetite due to oxidation, as observed within the 522 matrix (Figs. 1a, 2e, 3e, 7a, f):

523 
$$3\text{FeO} + \frac{1}{2}\text{O}_2 \rightarrow \text{Fe}_3\text{O}_4 (\text{Eq. 9})$$

524 Magnetite formation by oxidation of ilmenite can be excluded because the reaction would 525 stoichiometrically lead to the additional formation and local enrichment of TiO<sub>2</sub>, which is not 526 observed at the Fe-Ti-oxide boundaries:

527 
$$3FeTiO_3 + \frac{1}{2}O_2 \rightarrow Fe_3O_4 + 3TiO_2$$
 (Eq. 10)

Furthermore, oxidation of ilmenite above  $800^{\circ}$ C would only form pseudobrookite (Fe<sub>2</sub>TiO<sub>5</sub>) and hematite (Fe<sub>2</sub>O<sub>3</sub>) (Fu et al., 2010). Neither of the ferric phases were observed in the sample material from Zipplingen.

Fluorite-structured TiO<sub>2</sub> and rock salt-structured FeO were experimentally generated from 531 ilmenite at shock pressures >60 GPa (Liu 1975; King and Ahrens 1976). In our study, TiO<sub>2</sub> is 532 observed structurally only as rutile (Fig. 2b) and generally occurs within the cores of the Fe-Ti-533 oxide aggregates (Fig. 3a, g) or along grain boundaries of single ilmenite grains (Figs. 3c, d, 7c, 534 d, e), and not together with a Fe-rich phase, as could be expected for the breakdown of ilmenite to 535 536 FeO and TiO<sub>2</sub>. We cannot exclude that other breakdown reactions of ilmenite took place in addition to the described breakdown reactions in the presence of a reducing agent but we do not 537 observe evidence from the microfabric. 538

The investigated sample from Zipplingen consists of  $\approx 15\%$  of calcite, which either occurs as globules within the (devitrified) silicate matrix or as coarse xenomorphic calcite grains incorporating globules of (devitrified) silicate glass (Fig. 1b). The calcite phase boundary is

irregular and concave with respect to the silicate matrix. Philpotts and Ague (2009) used such 542 observations to interpret a silicate and an adjacent carbonatic melt based on the viscosities and, 543 therefore, the surface energies of the respective systems. Graup (1999) discussed a liquid 544 545 immiscibility of silicate and carbonatic melt in suevites from Zipplingen, which is in agreement with Osinski et al. (2008), who considered melting as the common behavior of calcite in impact 546 547 cratering. On the other hand, Hörz et al. (2015, 2020) proposed devolatilization of calcite as the dominant process based on SEM, electron microprobe analysis, thermal analyses, and shock 548 experiments. Temperatures of a melt vapor mixture formed from basement gneisses, where the 549 ilmenites from our study were likely derived, were modelled to be in excess of 2000°C (von 550 Engelhardt and Graup 1984). Carbonaceous sedimentary target rocks were first hit by the 551 impactor and thus experienced even higher temperature conditions (Bolten and Müller 1969). 552 Thermal decomposition of calcite produces  $CO_2$  at temperatures above  $\approx 650^{\circ}C$  (Galwey and 553 Brown 1999) and CO when temperatures exceed 1500°C (Itoh et al. 1993). Furthermore, the 554 555 initial impact melt was presumably at pressure conditions above the stability conditions for CO<sub>2</sub>. Above 30 GPa, oxygen and diamond can form, which during subsequent unloading may react to 556 CO (Nellis et al. 1991; Tschauner et al. 2001). Overall, we suggest CO as a potential reducing 557 558 agent, which was mixed into the silicate melt due to suevite formation.

Ilmenites in lunar basaltic breccia have  $\approx 3 \ \mu m$  spaced, parallel veins of rutile in combination with droplets of metallic iron inside glass with a well-developed flow texture containing trains of minute iron spherules (Sclar et al. 1973). Ilmenite aggregates were not reported to be enriched in Fe at the ilmenite-glass phase boundary. The presence of a reducing agent as a result of the Ries impact could, therefore, explain the iron-enriched rim of the Fe-Ti-oxide aggregates in comparison to lunar samples, where elemental iron is occurring within the grain (Sclar et al. 1973).

566

#### 567 **Pseudorutile formation**

In the Aumühle gneiss (R20-16A), pseudorutile ( $Fe_2Ti_3O_9$ ) forms a rim around individual 568 ilmenite grains within the aggregates (Fig. 9c, d) or as a nearly complete replacement of coarse 569 570 ilmenite (Fig. 9e). Broad peaks in the Raman spectra indicate a fairly amorphous structure (Fig. 2h) (Imperial et al. 2022). Ilmenite alteration to pseudorutile occurs when water acts as a 571 transport medium for iron leaching (Mücke and Bhadra Chaudhuri 1991) at a high oxygen 572 573 fugacity below 700°C (Gupta et al. 1991), which proceeds along grain boundaries and structural discontinuities within the grain (Teufer and Temple 1966; Temple 1966; Grey and Reid 1975). 574 Ilmenite aggregates in sample R20-16A have intergranular microcracks (Fig. 9b), which could 575 have allowed fluids to circulate along the grain boundaries, thereby favoring pseudorutile 576 formation surrounding single ilmenite grains during a late stage of hydrothermal activity. A 577 paleomagnetic study by Sleptsova et al. (2022) shows that the suevites from Aumühle, including 578 from sample R20-16A, were partially overprinted by hydrothermal activity, compatible with our 579 observations. Arp et al. (2013) suggested hydrothermal activity lasted  $\approx 250$  ka after the Ries 580 581 impact.

582

#### 583 **Outlook and implications**

The interpretations of our study rely on experiments conducted in thermodynamic equilibrium. The shock conditions during meteorite impacts, however, are far from thermodynamic equilibrium conditions (Melosh 1989; Stöffler et al. 2018). Future shock experiments with subsequent high-temperature quenching under controlled oxygen fugacity should yield insight

into more realistic temperature, pressure, and oxygen fugacity conditions of shocked ilmenites in
the natural environment.

Ilmenite-bearing material, such as the regolith of the moon, may constitute an in-situ oxygen 590 source (Badescu 2012 and references therein) and produces water when heated to temperatures 591 592 between 850° and 1050°C and reduced with  $H_2$  (Li et al. 2012; Dang et al. 2015; Sargeant et al. 2020). Our study demonstrates that ilmenite microfabrics can be largely modified due to 593 meteorite impacts, resulting in aggregates that consist of various Fe- and/or Ti-bearing phases, 594 595 which change the physical, chemical, and/or magnetic properties of the host rocks. Our findings might also be relevant for the consideration of breakdown reactions of Fe/Ti-phases in the scope 596 of in-situ resource utilization (ISRU) on planetary objects, whose surfaces are blanketed by 597 598 meteorite impact structures.

599

#### 600 Conclusions and summary

The homogeneously distributed ilmenite aggregates within the investigated samples contain a suite of Fe-Ti-oxide phases created by pressure-, temperature-, and oxygen-fugacity-changes during impact cratering. Figure 11 summarizes our proposed scenario of how the modification of ilmenite via solid-state transformation through time can explain the phase assemblages in the Ries impact structure.

606 (I) Upon shock loading, target ilmenites were (partly) transformed into the high-pressure phase 607 liuite (>16 GPa), then transformed upon unloading to wangdaodeite and subsequently ilmenite 608 forming a foam structure (Fig. 11-I). In  $\approx$ 5% of the observed cases, shocked ilmenite is preserved 609 in the core of the aggregates, which can contain transformation twins that share a common

610  $\{11\overline{2}0\}$  plane with the host and the *c-axis* is oriented at an angle of 109° to that of the host (Figs. 611 4, 5a, 6b). The transformation from the high-pressure phase liuite is reflected by the 612 crystallographic relationship of the new grains that exhibit up to three orientation domains that 613 share a common  $\{11\overline{2}0\}$  plane and whose *c*-axes enclose angles of 109° and 99°, respectively, 614 similar to the twin lamellae within the only partly transformed remnant shocked ilmenite (Fig. 5), 615 which are suggested to represent transformational twins (Fig. 10).

616 (II+III+IV) Heat production during shock unloading leads to a decrease in oxygen fugacity, however, high-temperature gradients produce different phases and microfabrics. The formation of 617 ferropseudobrookite from ilmenite, occurring mostly at aggregate boundaries and along fractures 618 619 (Fig. 11-II), indicates temperatures >1140°C with a low oxygen fugacity. Rutile grains along boundaries of ilmenite in the aggregates (Fig. 11-III) indicate thermal decomposition of ilmenite 620 between 850° and 1050°C favored along sites of increased diffusivity. The presence of a reducing 621 622 agent (likely CO) generated an electron enrichment, which resulted in an electrochemical potential driving  $Fe^{2+}$  to migrate from within the ilmenite aggregate cores towards the rim, 623 forming TiO<sub>2</sub> inside the ilmenite aggregates (Fig. 11-IV). Depending on whether the influence of 624 625 heating or reduction locally prevailed, either µm-sized rutile on grain boundaries (Fig. 11-III) or larger TiO<sub>2</sub> residues were formed (Fig. 11-IV), respectively. The chemical interaction of the 626 silicate melt with ferropseudobrookite formed rare armalcolite-ferropseudobrookite solid solution 627 (Fig. 8a, b, h). 628

629 (V) Oxygen fugacity increased as temperature decreased after the impact, which oxidized the 630 iron-enriched rim of the ilmenite aggregates, thereby resulting in the formation of  $\mu$ m-sized (sub-631 )euhedral magnetite (Fig. 11-V).

(VI) In the aftermath of the impact at temperatures <700°C in combination with high oxygen</li>
fugacity, pseudorutile locally formed along intergranular cracks within the ilmenite aggregate in
the presence of a fluid (Fig. 11-VI).

635

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968 Fig. 1 Polarized light micrographs of ilmenite- and other Fe-Ti-oxide aggregates within samples from the Ries impact structure. **a**, **b**) Glass fragment (Flädle; sample CT827a) from Zipplingen. 969 a) Fe-Ti-oxide (fto) aggregates are associated with magnetite (Mag); reflected light. b) 970 Homogeneously distributed Fe-Ti-oxide aggregates (white arrows) and calcite (Cal) within the 971 matrix in reflected polarized light. Note the (devitrified) glass globules within calcite (Cal, yellow 972 arrows) and calcite globules within the matrix (gl, orange arrows). The inset displays the irregular 973 974 glass-calcite boundary in transmitted light with crossed polarizers. Note that calcite is concave with respect to the glass matrix. c, d) Strongly shocked gneiss clast (sample R20-16A) in suevite 975 from the Aumühle quarry that contains ilmenite (Ilm) aggregates, secondary hematite (Hem), and 976

977 vesicles (vs) within a cryptocrystalline matrix (cr); (c) single plane-polarized reflected light, (d)
978 transmitted light and crossed polarizers.

979

Fig. 2 Raman spectra of (a) ilmenite, (b) rutile, (c) ferropseudobrookite, (d) magnetite rim surrounding ilmenite, (e) magnetite within the (devitrified) glass matrix, and (f) sphene within sample CT827a; as well as (g) ilmenite and (h) pseudorutile within sample R20-16A.

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Fig. 3 SEM analyses of Fe-Ti-oxide aggregates embedded in a (devitrified) glass matrix (gl) 984 sample CT827a, Zipplingen; (a-e, g) are BSE images and (f) is a SE image. a, b) Foam-structured 985 ilmenite (Ilm) aggregate containing rutile (Rt; black) with a magnetite (Mag; white) rim. The 986 yellow rectangle displays the area depicted in (b). c, d) Foam-structured ilmenite (IIm) aggregate 987 988 with isolated small rutile (Rt; black) grains along ilmenite boundaries. Note that the image contrast was optimized to show the orientation contrast of ilmenite grains (grey shades), rutile 989 990 appears black. Magnetite (Mag; white) occurs at the rim. The yellow rectangle displays the area 991 depicted in (d). e, f) (Sub-)euhedral magnetite (Mag) occurring at the rim of Fe-Ti-oxide aggregates. The yellow rectangle displays the area depicted in (f). g) Fe (green) and Ti (purple) 992 map of the aggregate consisting of ilmenite (Ilm) with rutile (Rt) in the core and a rim of 993 magnetite (Mag). The yellow arrow indicating the Fe/Ti gradient displayed in (h). h) Increasing 994 995 Fe/Ti gradient from the core towards the rim of the aggregate based on EMP data (Tab. S1).

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Fig. 4 EBSD data of ilmenite aggregate, sample CT827a, Zipplingen. a) Phase map of ilmenite
(red), magnetite (blue), and rutile (yellow). b) Orientation map (all Euler coloring) of (a). c)

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999 Grain Reference Orientation Distribution map (GROD) displaying the angular deviation from the 1000 average grain orientation. Note the short wavelength misorientation pattern in the core. **d**, **e**) Pole 1001 figures of the  $\{0001\}$ ,  $\{11\overline{2}0\}$ , and  $\{10\overline{1}1\}$  planes of the (d) coarse grain in the core and (e) 1002 polycrystalline ilmenite rim.

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1004 Fig. 5 Crystallographic orientations within ilmenite aggregates, sample CT827a, Zipplingen. a) Orientation map (inverse pole figure of z-direction) of rim and core, which displays twin lamellae 1005 (orange; twin boundaries in white). The blue and yellow triangles depict the positions of 1006 representative core and lamellae orientations depicted in (e). b) Orientation map (inverse pole 1007 figure of z-direction) with green, blue, and pink triangles, which depict the positions of 1008 representative crystallographic orientations displayed in (f). c, d) Pole figures and density plots of 1009 the  $\{0001\}$ ,  $\{11\overline{2}0\}$ , and  $\{10\overline{1}1\}$ , corresponding to the orientation map in (a) and (b), 1010 1011 respectively. e) Representative orientations of the host (blue triangle) and lamellae (yellow triangle) depicted in (a) of the  $\{0001\}$ ,  $\{11\overline{2}0\}$ , and  $\{10\overline{1}1\}$  planes. **f**) Representative orientations 1012 (green, blue, and pink triangles) of the three different crystallographic domains depicted in (b) of 1013 the  $\{0001\}, \{11\overline{2}0\}, \text{ and } \{10\overline{1}1\} \text{ planes.}$ 1014

1015

**Fig. 6** EBSD data of ilmenite aggregate within sample CT827a, Zipplingen. **a)** All Euler orientation map. Note the larger grains with concave grain boundaries indicated by the three arrows as well as the region strongly enriched in  $TiO_2$  displayed by the white line. **b)** GROD map displaying the angular deviation from the average grain orientation. Note the grains with internal misorientation and irregular grain boundaries indicated by the white arrows. **(c-e)** Ilmenite pole figures of the {0001}, {11 $\overline{2}0$ }, and {10 $\overline{1}1$ } planes of **c**) grains <5 µm, **d**) grains 5 – 10 µm, **e**) grains >10 µm. **f**) Orientation of all grains with GROD coloring as displayed in (b). Blue circles indicate that the enclosed orientation data is mostly derived from grains with low internal misorientation, whereas red circles indicate that the enclosed orientation data is mostly derived from grains with relatively high internal misorientation.

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Fig. 7 Ferropseudobrookite in Fe-Ti oxide aggregates, sample CT827a, Zipplingen. a, b) Fe-Ti-1027 oxide aggregate with magnetite (Mag) at the rim and within the (devitrified) matrix (gl); BSE 1028 images. Yellow rectangles in (b) indicate the close-ups in (c), (d), and (e). c, d) 1029 Ferropseudobrookite (Fpb) along a fracture within the ilmenite aggregate, where sphene (Spn) is 1030 1031 locally present. Note symplectic intergrowth of ferropseudobrookite and ilmenite (Ilm) as well as rutile grains [Rt; dark grey in (c) and white in (d)] along ilmenite grain boundaries at the top of 1032 the images; BSE and reflected single plane-polarized light, respectively. e) Close-up of rutile (Rt) 1033 along ilmenite (Ilm) boundaries, as well as a symplectic intergrowth of ilmenite and 1034 ferropseudobrookite (Fpb) at the contact to the (devitrified) matrix (gl), where (sub-)euhedral 1035 magnetite grains (Mag) also occur; BSE image. f) Traces of (sub-)euhedral magnetite grains 1036 (Mag) within the (devitrified) matrix (gl) in close distance to an intergrowth of ilmenite (Ilm) and 1037 porous ferropseudobrookite (Fpb). 1038

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Fig. 8 EBSD and EDS data of ilmenite and ferropseudobrookite aggregates, sample CT827,
Zipplingen. a) Reflected light micrograph of a Fe-Ti-oxide aggregate. The red rectangle depicts
the area displayed in (b-d, g, h). b) Phase distribution map of ilmenite (Ilm; red),

ferropseudobrookite (Fpb; green), sphene (Spn; light blue), magnetite (Mag; dark blue), and rutile
(Rt; yellow). c) Ilmenite orientation map (inverse pole figure of z-direction). d)
Ferropseudobrookite orientation map (inverse pole figure of z-direction). e) Pole figures of the
{0001}, {1120}, and {1011} planes of ilmenite displayed in (c). f) Pole figures of the {100},
{010}, and {001} planes of ferropseudobrookite displayed in (d). g) Manganese (Mn) and h)
magnesium (Mg) distribution within the Fe-Ti-oxide aggregate.

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Fig. 9 BSE images of Fe-Ti-oxide microfabrics within suevite sample R20-16A from Aumühle. 1050 a) Coarse ilmenite grain (Ilm) within a cryptocrystalline matrix (cr) containing vesicles (vs). The 1051 yellow rectangle depicts the area in (b). b) Ilmenite (Ilm) with foam structure. c) Fe-Ti-oxide (fto) 1052 reaction fabric within a vesicle-bearing (vs) cryptocrystalline matrix (cr). The yellow rectangle 1053 depicts the area in (d). d) Ilmenite with a rim of pseudorutile (Prt). Remnants of ilmenite in the 1054 bottom right are mostly covered by pseudorutile. e) Former ilmenite grain transformed almost 1055 completely to pseudorutile (Prt) with preferentially oriented pores/cracks and remnants of foam-1056 structured ilmenite (Ilm). f) Foam-structured ilmenite (Ilm) with a finer-scaled ilmenite rim (Ilm 1057 rim) embedded in a cryptocrystalline matrix (cr). 1058

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**Fig. 10** High-pressure phase transformations of  $FeTiO_3$ . **a**) Phase transformations during loading and unloading of  $FeTiO_3$  with their respective symmetries, space group numbers (No.), as well as crystallographic lattice parameters a, b, and c. **b**) Group theoretical analysis of the symmetry conversion (blue dashed lines) of ilmenite, wangdaodeite, and liuite with the respective Baincorrespondence matrix. Note that all symmetries derive from the cubic perovskite

supersymmetry. Point groups and their respective order are in orange. The red dashed line indicates reconstructive positional shifts of the phase transitions. The generation of transformation twins occurs from the transformation of liuite to wangdaodite.

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Fig. 11 Summary and interpretations of this study. Pressure (P; green), temperature (T; red), and 1069 1070 relative oxygen fugacity (fO<sub>2</sub>; blue) conditions as a function of time based on the observed characteristic Fe-Ti-oxide microfabrics. I) ilmenite 1071 Foam-structured (Ilm). II) Ferropseudobrookite (Fpb; dark grey) along a fracture within a Fe-Ti-oxide aggregate. Note that 1072 rarely observed armalcolite (Arm) likely formed from Fpb. III) Rutile (Rt) along grain 1073 boundaries of ilmenite (Ilm). IV) Ilmenite (Ilm) aggregate containing rutile (Rt) and with a rim 1074 enriched in Fe (Fe<sup>†</sup>). V) (Sub-)euhedral magnetite (Mag). VI) Pseudorutile (Prt) as rim around 1075 grains of ilmenite aggregates (Ilm). (bI) – (bV) sample CT827a and (bVI) sample R20-16A. Note 1076 that reactions as a function of temperature (red arrows) are always coupled to oxygen fugacity 1077 1078 reaction (blue arrows) conditions and vice versa, in contrast to the high-pressure reaction (green 1079 arrows). The dashed lines represent pressure, temperature, and oxygen fugacity regimes in which the corresponding reactions take place, except for reaction IV, which presumably takes place up 1080 1081 to the ferropseudobrookite stability field (1140°C). BSE images (bI)-(bIV), (bVI) and SE image (bV). Pressure conditions after Leinenweber et al. (1991); temperature and oxygen fugacity 1082 conditions after Lindsley (1965), Keil et al. (1970), Tuthill and Sato (1970), Taylor et al. (1972), 1083 Mitchell et al. (1982), Kuroda and Mitchell (1983), Gupta et al. (1991), and Sargeant et al. 1084 (2020).1085

**Tab. 1** EMP data (wt.%) of ilmenite (Ilm), ferropseudobrookite (Fpb), and armalcolite (Arm)
from sample CT827a (Figs. 4, 8) as well as ilmenite and pseudorutile (Prt) from sample R20-16A

1088	(Fig. 9c, d). Note that the detection limit in ppm is given in brackets for a concentration below
1089	0.004 wt.% and/or a not detectable (n.d.) component (*=data as $Fe_2O_3$ ). The deviation of the total
1090	wt.% can be explained for ilmenite due to the presence of a Fe <sub>2</sub> O <sub>3</sub> component, which binds more
1091	oxygen and thus would increase the calculated total. Porosity of ferropseudobrookite(-armalcolite
1092	solid solution) (Fig. 7c-f) leads to a decrease in the total wt.%, whereas in pseudorutile, the
1093	presence of water and/or hydroxyl groups decreases the total wt.%.

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Phase	Fig.	FeO /	TiO <sub>2</sub>	MnO	MgO	SiO <sub>2</sub>	Al <sub>2</sub> O <sub>3</sub>	Na <sub>2</sub> O	CaO	K <sub>2</sub> O	Total
	-	Fe <sub>2</sub> O <sub>3</sub> *			-						
Ilm	4 (core)	41.35	51.86	3.80	1.55	0.08	n.d. (253)	0.03	0.05	n.d. (239)	98.72
Ilm	4 (core)	41.52	51.91	3.91	1.55	0.10	n.d. (253)	0.01	0.05	0.02	99.07
Ilm	4 (core)	41.41	51.91	3.90	1.54	0.07	n.d. (253)	n.d. (370)	0.06	n.d. (239)	98.88
Ilm	4 (rim)	40.86	51.89	3.65	2.08	0.09	0.02	0.02	0.06	(249)	98.67
Ilm	4 (rim)	41.88	51.27	3.90	1.47	0.06	0.09	0.02	0.09	0.01	98.80
Ilm	4 (rim)	41.44	51.61	4.01	1.60	0.10	0.11	0.02	0.07	(240)	98.96
Ilm	8	41.83	50.79	5.68	0.41	0.06	n.d. (239)	0.03	0.04	0.01	98.87
Ilm	8	41.42	51.28	5.49	0.43	0.10	(239)	0.02	0.05	0.01	98.85
Ilm	8	42.14	50.68	5.74	0.40	0.07	n.d. (239)	0.03	0.04	(242)	99.11
Fpb	8	33.89	59.69	1.86	0.37	0.75	0.10	0.02	0.56	0.02	97.31
Fpb	8	34.47	59.39	2.20	0.21	0.55	0.11	0.01	0.10	0.02	97.07
Fpb	8	34.31	60.33	1.72	0.40	0.45	0.14	n.d. (379)	0.12	0.01	97.50
Arm	8	30.54	57.72	0.93	2.61	3.17	0.82	0.10	0.29	0.16	96.45
Arm	8	29.83	59.98	1.20	2.05	1.35	0.32	0.04	0.20	0.04	95.09
Arm	8	31.08	60.30	1.32	1.64	2.51	0.30	0.09	0.33	0.06	97.76
Ilm	9c, d	50.74	44.46	1.70	0.52	0.07	n.d. (242)	0.02	0.03	(249)	97.56
Ilm	9c, d	46.74	48.66	1.94	0.66	0.04	0.01	0.03	0.06	0.03	98.16
Ilm	9c, d	48.56	46.64	1.70	0.69	0.08	n.d. (242)	0.04	0.03	n.d. (241)	97.78
Prt	9c, d	41.93*	52.84	1.06	0.11	0.29	n.d. (242)	0.01	0.04	0.01	96.33
Prt	9c, d	41.96*	52.53	0.81	0.07	0.33	n.d. (242)	0.02	0.04	(233)	95.79
Prt	9c, d	41.42*	52.80	0.80	0.08	0.37	n.d. (242)	0.01	0.06	n.d. (233)	95.56

1095

1096 **Tab. 2** Investigated rim and core of aggregates of Fe- and/or Ti-bearing phases (Ilm = ilmenite;

1097 Mag = magnetite; Rt = rutile; Spn = sphene; Fpb = ferropseudobrookite; Arm = armalcolite)

1098 within sample CT827a. The lowercase numbers indicate their occurrence in %.

Size [µm]	Rim phase(s)	Core phase(s)	
160 x 80	Ilm <sub>50</sub> , Arm <sub>50</sub>	Fpb <sub>60</sub> , Ilm <sub>35</sub> , Spn <sub>&lt;5</sub> , Rt <sub>&lt;5</sub>	
120 x 100	Mag	Ilm <sub>85</sub> , Fpb <sub>10</sub> , Spn <sub>&lt;5</sub> , Rt <sub>&lt;5</sub>	
160 x 70	Mag	Ilm <sub>85</sub> , Spn <sub>&lt;15</sub> , Rt <sub>&lt;1</sub>	

Size [µm]	Rim phase(s)	Core phase(s)
100 x 40	Ilm <sub>50</sub> , Rt <sub>50</sub>	Ilm
70 x 50	Mag	Ilm <sub>&gt;95</sub> , Rt <sub>&lt;5</sub>
100 x 30	-	Ilm>95, Rt<5

170 x 50	Ilm <sub>80</sub> , Mag <sub>20</sub>	Ilm <sub>95</sub> , Rt <sub>5</sub>
160 x 50	-	Ilm
120 x 80	Fpb	Ilm <sub>85</sub> , Rt <sub>&lt;10</sub> , Spn <sub>&lt;5</sub>
100 x 90	Ilm	Ilm <sub>95</sub> , Rt <sub>5</sub>
120 x 60	Mag <sub>&gt;95</sub> , Ilm <sub>&lt;5</sub>	Ilm <sub>&gt;95</sub> , Rt <sub>&lt;5</sub>
100 x 60	Mag	Ilm>95, Rt<5
80 x 70	Mag	Ilm <sub>85</sub> , Fpb <sub>10</sub> , Mag <sub>&lt;5</sub> , Rt <sub>&lt;5</sub>
80 x 60	-	Ilm <sub>&gt;95</sub> , Rt <sub>&lt;5</sub>
70 x 60	-	Ilm <sub>&gt;95</sub> , Rt <sub>&lt;5</sub>

60 x 50	-	Ilm>95, Rt<5
140 x 20	-	Ilm>95, Rt<5
100 x 25	Mag	Ilm <sub>95</sub> , Rt <sub>5</sub>
70 x 30	Mag	Ilm <sub>95</sub> , Rt <sub>5</sub>
40 x 30	Mag	Ilm <sub>85</sub> , Rt <sub>15</sub>
40 x 30	-	Spn <sub>50</sub> , Mag <sub>20</sub> , Ilm <sub>15</sub> , Rt <sub>15</sub>
40 x 20	-	Ilm
10 x 10	Mag	Ilm>95, Rt<5

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**Tab. 3** Investigated rim and core of aggregates of Fe- and/or Ti-bearing phases (Ilm = ilmenite;

1101 Prt = pseudorutile; Rt = rutile) within sample R20-16A. The lowercase numbers indicate their

1102 occurrence in % (\* = sub-aggregates of fractured ilmenite).

Size [µm]	Rim phase(s)	Core phase(s)
180 x 120	Ilm	Ilm <sub>95</sub> , Prt <sub>5</sub>
220 x 70	Ilm	Ilm
70 x 30	-	Ilm
50 x 20	Ilm	Ilm <sub>&gt;95</sub> , Prt <sub>&lt;5</sub>
40 x 25	Ilm	Ilm
40 x 20	-	Ilm>95, Rt<5

Size [µm]	Rim phase(s)	Core phase(s)
20 x 10	-	Ilm <sub>95</sub> , Prt <sub>5</sub>
15 x 10	-	Ilm <sub>&gt;95</sub> , Prt <sub>&lt;5</sub>
*140 x 50	-	Ilm
*40 x 15	-	Ilm
*30 x 20	-	Ilm
*10 x 5	-	Ilm

1103

## 1104 Supplementary Data

1105	<b>Tab. S1</b> EMP data [wt.%] of profile (step size $\approx 1 \ \mu m$ ) from core (#1) to rim (#29) depicted in
1106	Figure 3g, h with the at.%-ratio of iron and titanium (Fe/Ti). Note that Fe/Ti increases towards
1107	the aggregate's rim and #1-#4 correspond to rutile with varying amounts of FeO. The detection
1108	limit in ppm is given in brackets for either a concentration below 0.004 wt.% or a not detectable

- 1109 (n.d.) component. The deviation of the total wt.% can be explained due to the presence of  $Fe_2O_3$ ,
- 1110 which binds more oxygen and thus would increase the calculated total.

#	FeO	TiO <sub>2</sub>	MnO	MgO	SiO <sub>2</sub>	Al <sub>2</sub> O <sub>3</sub>	Na <sub>2</sub> O	CaO	K <sub>2</sub> O	Total	Fe/Ti
1	32.16	61.35	2.17	2.72	0.24	(249)	0.01	0.06	0.01	98.72	0.58
2	7.13	86.93	0.64	0.59	0.26	(251)	0.02	0.08	0.01	95.67	0.09
3	2.57	95.05	0.20	0.17	0.34	n.d. (251)	0.01	0.07	0.02	98.44	0.03
4	12.35	86.26	0.62	1.00	1.30	0.03	0.01	0.08	0.02	101.71	0.16
5	35.43	59.39	2.14	2.93	0.93	0.09	0.02	0.12	0.01	101.09	0.66
6	36.66	57.19	2.39	3.20	0.47	0.02	0.01	0.08	n.d. (249)	100.05	0.71
7	35.76	55.36	2.43	3.27	0.80	0.05	0.02	0.09	0.02	97.81	0.72
8	29.87	67.05	1.74	2.32	0.63	0.04	n.d. (260)	0.09	(247)	101.75	0.50
9	32.45	58.88	2.31	2.87	0.94	0.06	0.05	0.11	0.02	97.73	0.61
10	18.29	74.21	1.40	1.81	0.76	0.09	0.05	0.13	0.02	96.76	0.27
11	23.87	69.43	1.62	2.07	0.46	0.07	0.02	0.09	(248)	97.65	0.38
12	37.60	55.63	2.16	2.82	0.69	0.10	0.01	0.14	0.02	99.19	0.75
13	39.98	52.47	2.43	3.41	0.27	0.09	n.d. (380)	0.10	0.02	98.78	0.85
14	39.53	52.66	2.49	3.47	0.23	0.07	n.d. (380)	0.09	0.02	98.56	0.83
15	39.52	52.82	2.52	3.47	0.22	0.11	n.d. (380)	0.11	0.01	98.80	0.83
16	39.42	52.68	2.58	3.53	0.27	0.10	0.03	0.09	0.01	98.70	0.83
17	39.56	52.42	2.49	3.53	0.26	0.10	0.01	0.11	n.d. (244)	98.48	0.84
18	39.57	52.39	2.45	3.45	0.27	0.11	0.01	0.10	0.01	98.38	0.84
19	39.59	52.40	2.43	3.51	0.31	0.13	0.03	0.08	0.01	98.54	0.84
20	39.84	52.11	2.40	3.37	0.31	0.14	0.03	0.10	0.01	98.34	0.85
21	39.94	51.97	2.45	3.35	0.25	0.14	0.01	0.10	(238)	98.22	0.85
22	39.95	51.98	2.40	3.46	0.29	0.14	0.02	0.13	0.01	98.43	0.85
23	40.21	51.90	2.36	3.36	0.33	0.18	0.02	0.10	0.02	98.51	0.86
24	39.96	51.44	2.29	3.27	0.72	0.23	0.02	0.16	0.02	98.14	0.86
25	40.39	50.87	2.33	3.38	0.40	0.24	n.d. (367)	0.16	0.02	97.81	0.88
26	40.64	50.83	2.27	3.36	0.27	0.24	0.01	0.15	0.02	97.81	0.89
27	40.85	50.50	2.18	3.31	0.36	0.25	0.01	0.18	0.03	97.68	0.90
28	41.24	49.38	2.03	3.22	0.46	0.33	n.d. (373)	0.20	0.04	96.92	0.93
29	41.44	45.28	1.99	3.01	7.75	1.75	0.22	0.29	0.28	102.03	1.02

1111

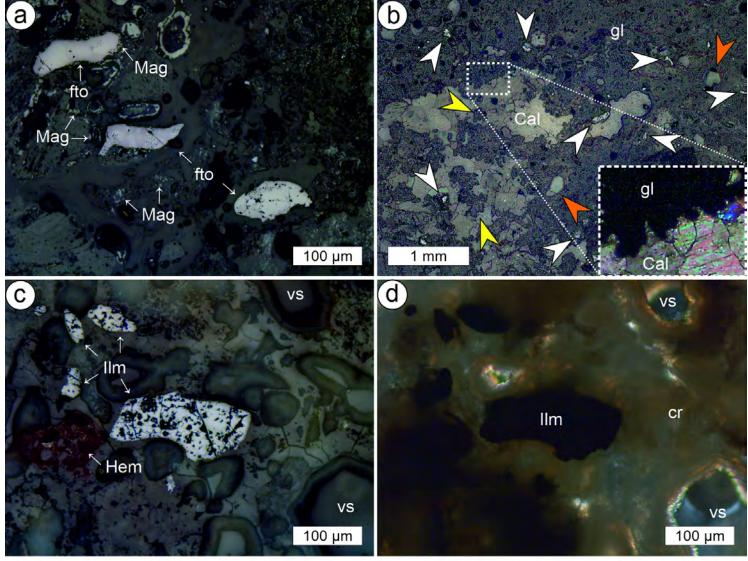
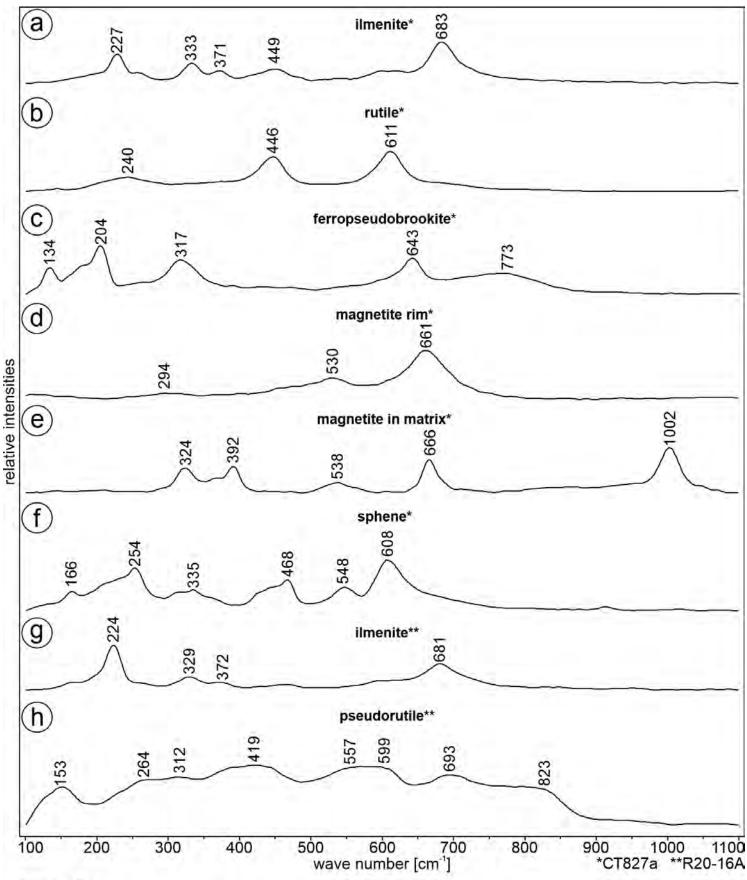
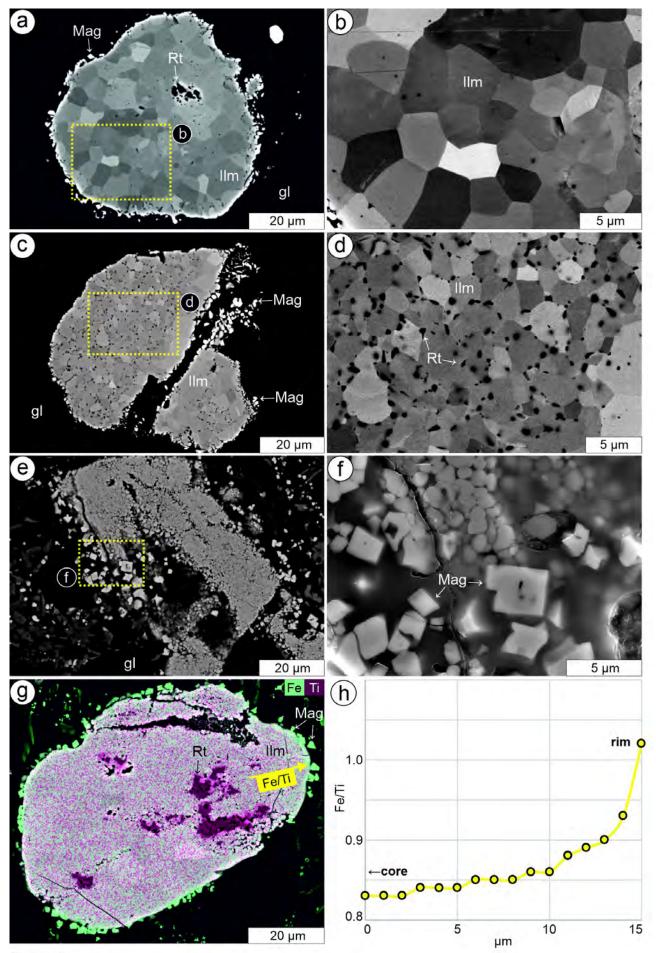


Figure 1







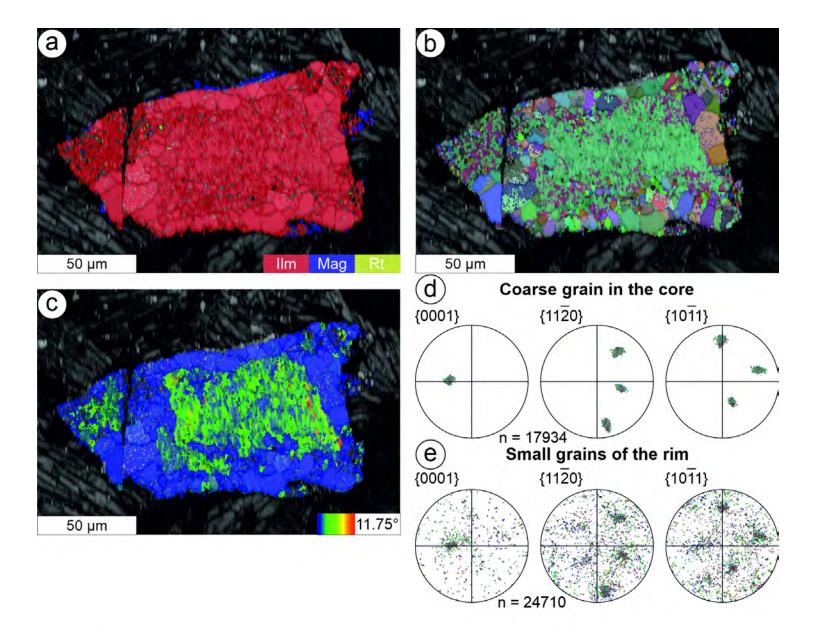
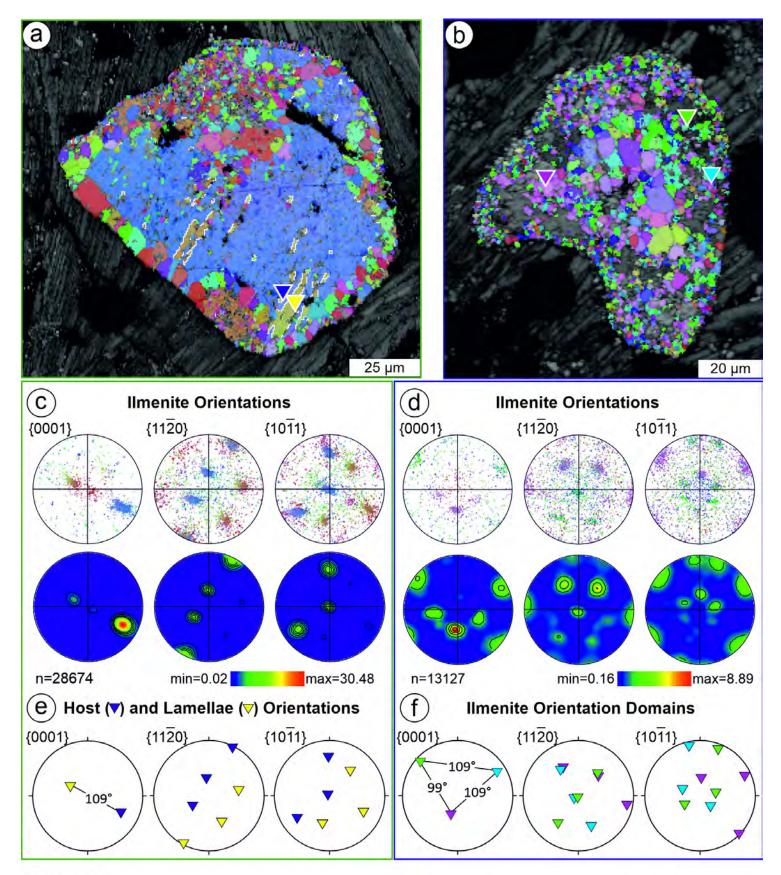
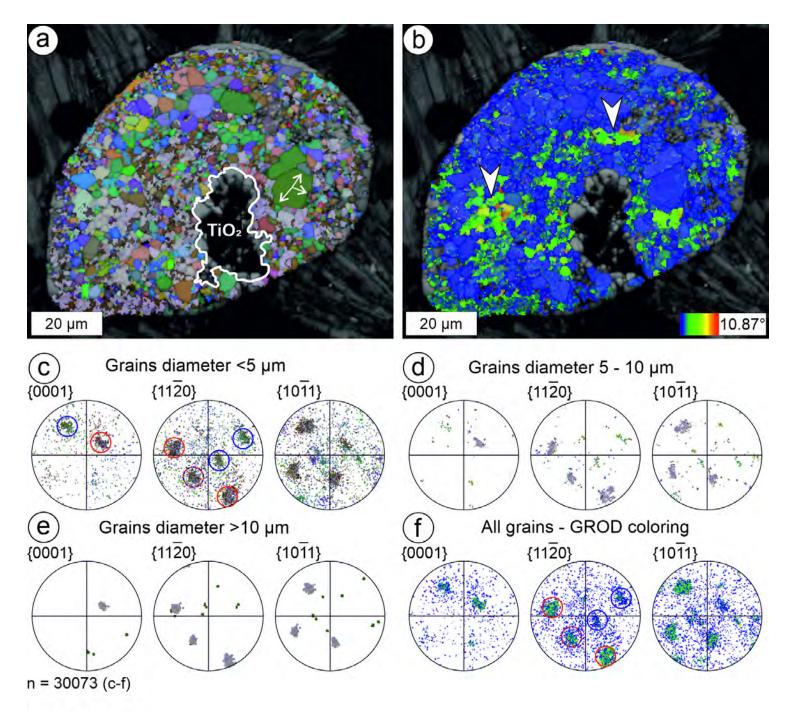


Figure 4







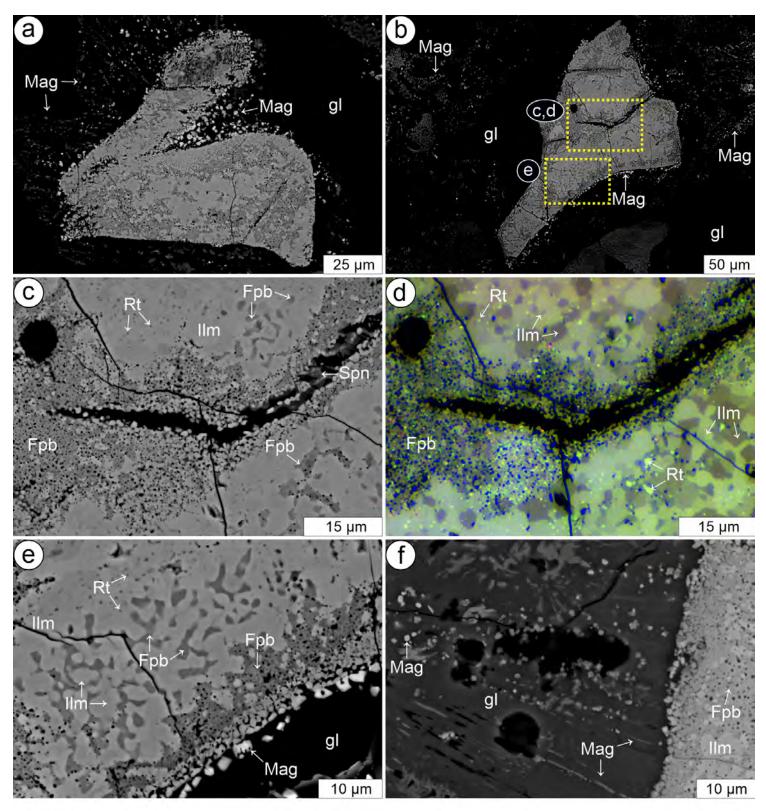
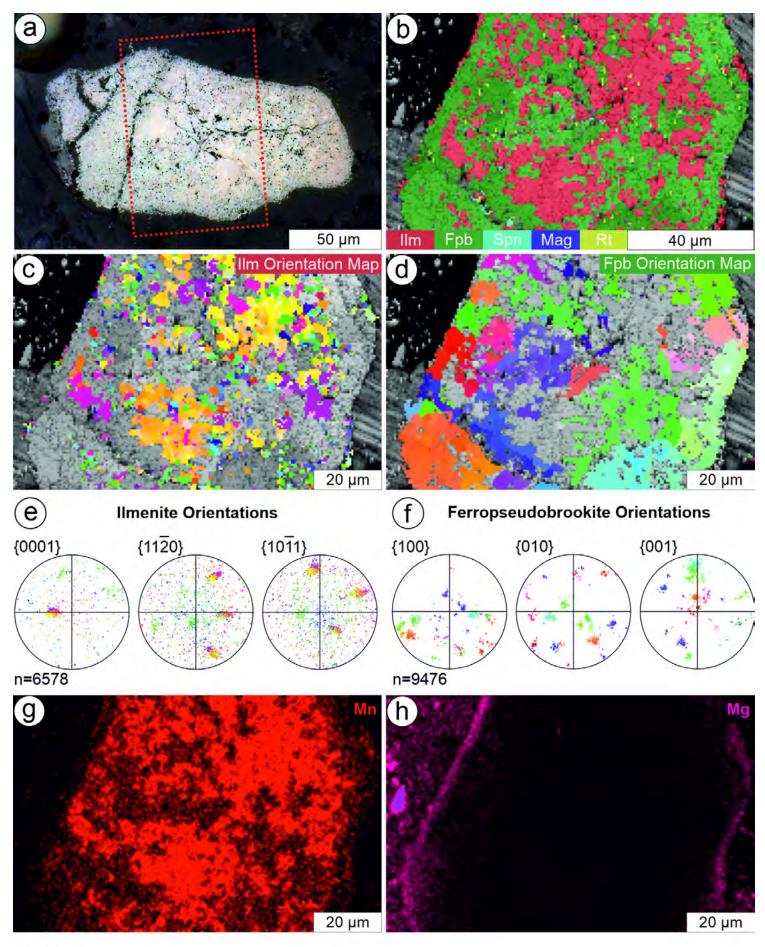
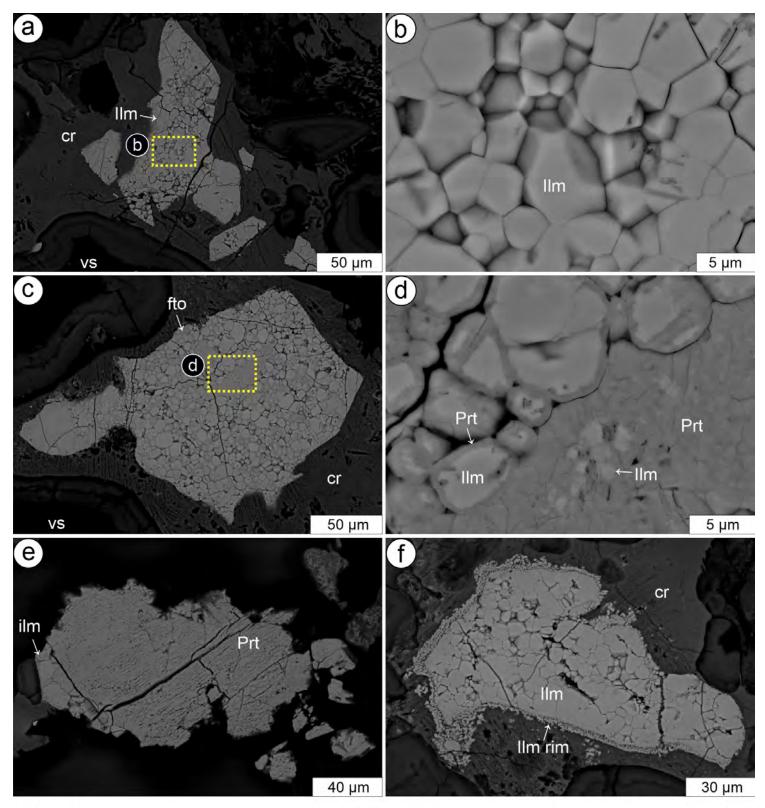
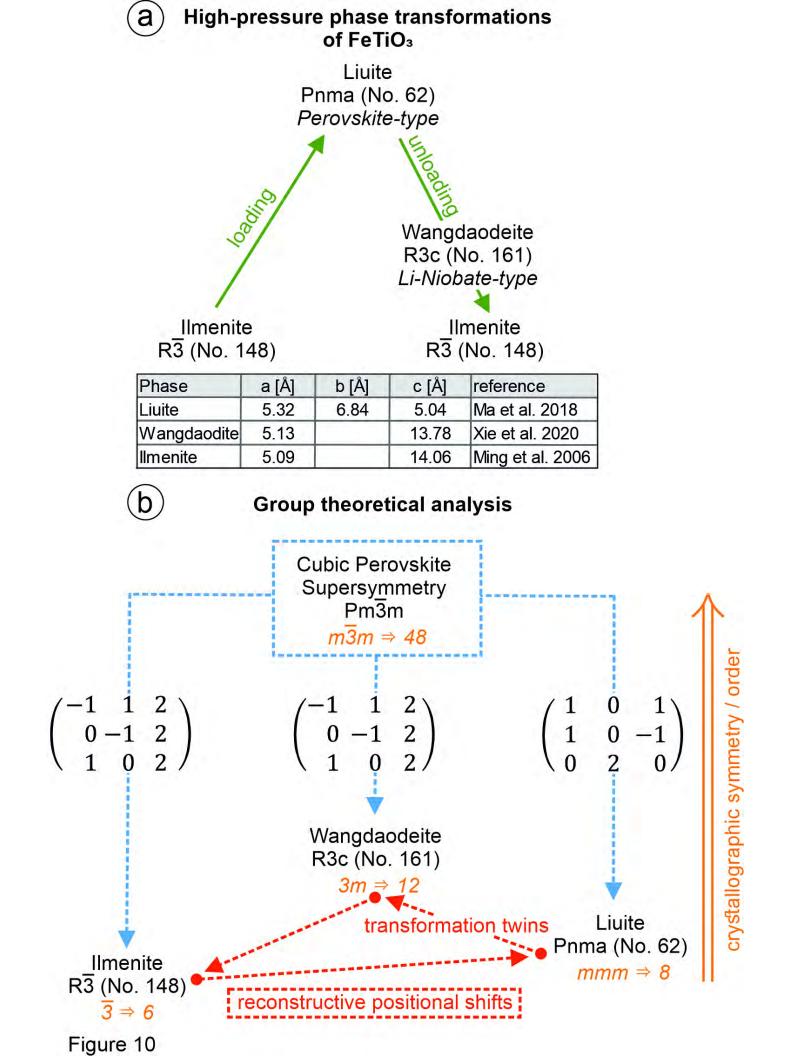


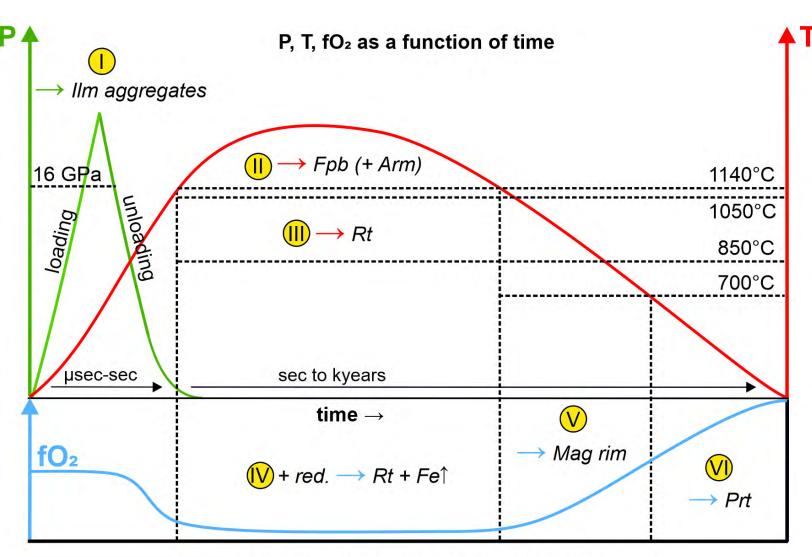
Figure 7











## **Characteristic Fe-Ti-oxide microfabrics**

