

1 **Revision 1**

2 **Crystal versus melt compositional effects on the partitioning of the**  
3 **first-row transition and high field strength elements between**  
4 **clinopyroxene and silicic, alkaline, aluminous melts**

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## Abstract

22

23       The first-row transition element (FRTE) and high field strength element (HFSE)  
24 systematics are powerful tools for tracking the source and evolution of mantle-derived  
25 magmas. Clinopyroxene is generally considered a key fractionating mineral controlling  
26 the partitioning of trace elements between melt and residual solid during mantle melting.  
27 Although partitioning of FRTE and HFSE between clinopyroxene and basaltic melts  
28 has been well studied, experimental constraints on their partitioning behaviors in the  
29 presence of siliceous, aluminous and alkali-rich melts are still lacking. Here we present  
30 clinopyroxene-silicic melt (67–69 wt% SiO<sub>2</sub>) partitioning experiments at 1 bar pressure  
31 and 1070–1100 °C for Co, Mn, Ni, Cu, Zn, Fe, Sc, Cr, V, Ti, Zr, Hf, Nb, and Ta. Run  
32 products consist of diopsidic clinopyroxene coexisting with various melt compositions  
33 with non-bridging oxygen to tetrahedral cation ratio (NBO/T) ranging from 0.10 to 0.22.  
34 Using our new partition coefficients (*D*s) and combining literature data we assess some  
35 of the effects of crystal chemistry and the melt composition on the partitioning of FRTE  
36 and HFSE in this simple system.

37       We show that partitioning of FRTE varies from mildly incompatible (e.g., *D* =  
38 ~0.1–1 for V, Cu, and Zn) to highly compatible (e.g., *D* > 10 for Cr and Ni), with highest  
39 compatibilities observed for Ni (*D*<sub>Ni</sub> = 13–34). The partitioning of HFSE varies from  
40 highly incompatible (*D* = 0.01–0.08) for Nb and Ta to mildly incompatible (*D* =  
41 0.18–0.82) for Zr, Hf, and Ti. Our measured clinopyroxene-melt *D*s are consistent with  
42 the theoretical predictions of the lattice strain model. *D*s data for most tri-, tetra-, and  
43 pentavalent elements tend to increase with increasing tetrahedrally-coordinated Al

44 content, in agreement with those anticipated from crystal-chemical considerations. In  
45 contrast to <sup>iv</sup>Al, clinopyroxene Na content has very little effect on trace element  
46 partitioning due to its insufficient incorporation into clinopyroxene at relatively low-  
47 pressure conditions. These data further support a significant control of melt  
48 composition/structure on partitioning for highly polymerized melts. In general,  
49 measured *D*s roughly increase to different extent with increasing polymerization of the  
50 melt (i.e., lower NBO/T or higher ASI). For our equilibrium melt compositions, *D*s for  
51 several FRTE such as Co and Ni correlate well with the melt molar  $Mg^{2+}/(M^+ + M^{2+})$   
52 whereas *D*s for FRTE vary as a function of the melt alkalis. These well-defined trends  
53 support the role of melt NBO species (e.g.,  $Mg^{2+}$ ) or complexing ligands (e.g.,  $Na^+$  and  
54  $K^+$ ) in controlling the partitioning of these elements.

55 Overall, our new *D*s data demonstrate that even very small changes in melt major-  
56 element compositions can greatly affect element partitioning in strongly polymerized  
57 silicic systems. These findings have important implications relevant to petrogenetic  
58 studies of the interaction between silicic melt and peridotite that occurs at shallow  
59 mantle conditions in a variety of tectonic settings.

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61 **Key words: clinopyroxene, melt composition, FRTE, HFSE, partition coefficient**

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## 1. Introduction

65

66 Siliceous, alkali-rich and aluminous glasses have been widely reported in mantle-  
67 derived spinel peridotite xenoliths from a variety of geographic and tectonic localities  
68 (e.g., Xu et al., 1996; Vannucci et al., 1998; Yaxley and Kamenetsky, 1999; Miller et  
69 al., 2012), including continental alkali basalt provinces, oceanic hot spots, and  
70 subduction-related settings. The glasses contain from 40 to >70 wt% SiO<sub>2</sub>, up to 17 wt%  
71 total alkalies, and up to 25 wt% Al<sub>2</sub>O<sub>3</sub> (Yaxley and Kamenetsky, 1999). As a result, the  
72 interaction between peridotite and such silicic melts has received considerable attention  
73 as an effective mechanism of mantle metasomatism (e.g., Coltorti et al., 2000; Qian et  
74 al., 2015), which may contribute to the heterogeneous enrichment of erupted lavas. In  
75 particular, phase-equilibria experiments further show that these kinds of melts can be  
76 readily in equilibrium with harzburgitic or lherzolithic mineral assemblages at low  
77 pressure (Draper and Green, 1997, 1999). Knowledge of element partitioning between  
78 mantle minerals and silicic, alkaline, aluminous melts is therefore pivotal to  
79 understanding upper mantle processes such as equilibrium partial melting and  
80 metasomatism.

81 Clinopyroxene (cpx) is a common mineral in upper mantle and magmatic rocks  
82 that can significantly control the partitioning of trace elements between melt and  
83 surrounding solid matrix (e.g., Lundstrom et al., 1994; Wood and Trigila, 2001; Sun  
84 and Liang, 2013; Bédard, 2014; Li, 2018; Beard et al., 2019; Ma and Shaw, 2021). Thus,  
85 cpx–melt trace element partition coefficients ( $D^{\text{cpx/melt}}$ ) are key for interpreting the  
86 geochemical characteristics of the silicic melt and ambient mantle peridotite involved

87 in the interactions. Previous studies investigating cpx–melt partitioning largely focus  
88 on mafic to ultramafic systems (e.g., Gaetani and Grove, 1995; Blundy et al., 1998;  
89 Lundstrom et al., 1998; Hill et al., 2000; Bonechi et al., 2021), with an emphasis on the  
90 importance of crystal-chemical control on trace element partitioning in basaltic melts.  
91 For example, a number of experimental studies demonstrated the dependence of rare  
92 earth element (REE) and high field strength element (HFSE) partitioning on the  
93 abundance of tetrahedral Al (<sup>iv</sup>Al) or Ca-Tschermakite content in cpx (e.g., Gaetani and  
94 Grove, 1995; Hill et al., 2000). As a consequence, most existing theoretical frameworks  
95 treat the crystal composition as the dominant role in determining cpx–melt partitioning  
96 behavior (e.g., Wood and Blundy, 1997; Sun and Liang, 2012, 2013; Mollo et al., 2018),  
97 while the potential influence of the corresponding melt is less considered.

98       The effects of melt composition/structure on trace element partitioning have been  
99 investigated by using coexisting immiscible silicate melts (Watson, 1976; Ryerson and  
100 Hess, 1978; Schmidt et al., 2006) or through mineral–melt equilibrium approaches (e.g.,  
101 Gaetani, 2004; Prowatke and Klemme, 2005; Huang et al., 2006; Miller et al., 2006;  
102 Evans et al., 2008; Qian et al., 2015; Michely et al., 2017; Schoneveld and O’Neill,  
103 2019). The pioneering studies of Watson (1976) and Ryerson and Hess (1978)  
104 experimentally demonstrated that depolymerized melts preferentially accommodate  
105 cations with a high charge density (i.e., high charge/cation radius) such as transition  
106 metals, REE, and HFSE, whereas low-charge-density cations such as Cs are better  
107 hosted in polymerized melts. Mineral-melt partitioning studies have shown that melt  
108 composition/structure effects dominate trace element partitioning when the degree of

109 melt polymerization exceeds a threshold value (Gaetani, 2004; Mysen, 2004).  
110 Consistent with this notion, partition coefficients between minerals and melts for REE  
111 could increase by an order of magnitude when the melt changes from basaltic to granitic  
112 composition (e.g., Tiepolo et al., 2007; Bédard, 2014).

113 Despite the potential role of melt compositional variations in controlling crystal–  
114 melt partitioning, only a few experimental studies have been conducted to determine  
115 the cpx–melt partition coefficients in highly silicic systems (Klein et al., 2000; Huang  
116 et al., 2006). This is primarily due to the difficulty in growing large cpx crystals and  
117 achieving equilibrium partitioning in high-SiO<sub>2</sub> melts. Although Klein et al. (2000)  
118 reported  $D_{\text{REE}}$  and  $D_{\text{HFSE}}$  between cpx and andesitic to granodioritic melts, the wide  
119 temperature range (900–1150 °C) makes it difficult to identify and quantify the  
120 compositional effects. Huang et al. (2006) investigated the effect of melt composition  
121 on trace element partitioning between cpx and silicic melts; however,  $D_s$  were only  
122 determined for a limited number of trace elements (e.g., REE, Ti, Zr, and Sr). So far  
123 there is still a lack of data for showing the crystal versus melt compositional effects on  
124 the partitioning behavior of geochemically important trace elements, such as the first-  
125 row transition element (FRTE) and HFSE, between cpx and silicic melt.

126 Accordingly, in this study, we present new partition coefficient data derived from  
127 synthetic cpx in equilibrium with anhydrous Si-rich melts whose composition is typical  
128 of silicic xenolith glasses reported in the literature (Vannucci et al., 1998). Experiments  
129 were conducted by using Pt-crucibles in a high-temperature furnace, at temperatures of  
130 1070–1100 °C, and at atmospheric pressure and oxygen fugacity conditions.

131 Specifically, we carried out experiments by changing Na<sub>2</sub>O content in the system CaO–  
132 MgO–Al<sub>2</sub>O<sub>3</sub>–SiO<sub>2</sub>–Na<sub>2</sub>O–(K<sub>2</sub>O)–Fe<sub>2</sub>O<sub>3</sub>–TiO<sub>2</sub> to cover a large compositional range  
133 such as in natural systems. Since sodium is known to play a significant role in changing  
134 cpx crystal chemistry and the structure of silicate melts (e.g., Bennett et al., 2004;  
135 Michely et al., 2017; Beard et al., 2019), these measurements enable an assessment of  
136 the effect of Na on cpx–melt partitioning. Indeed, rapid diffusive infiltration of sodium  
137 into partially molten peridotite is considered as an effective mechanism to account for  
138 the creation of silica-rich glasses in mantle xenoliths (Lundstrom 2000).

139 Our new results, particularly when combined to literature data, clearly show that  
140 trace element partitioning in silicic melts differ from those in basaltic melts, and we  
141 present a scheme for evaluating the specific influence of different crystal and melt  
142 compositional variables on the partitioning of FRTE and HFSE. The data presented in  
143 this study thus have important implications for the partitioning behavior of trace  
144 elements in evolved silicic systems, particularly for trace element systematics relevant  
145 to mantle partial melting or metasomatic processes operated by highly silicic melts.

146

## 147 **2. Experimental and analytical methods**

### 148 **2.1. Starting materials**

149 Compositions of the starting materials are presented in Table 1 and their utilization  
150 in each experiment is presented in Table S1. As a base starting material, all runs utilized  
151 a silicic melt similar in composition to sample HISIL of Huang et al. (2006), based on  
152 a glass composition in a harzburgite xenolith from the Canary Islands (Vannucci et al.

153 1998). The starting material HISILA is a synthetic powder, made to correspond to  
154 coexisting silica-rich glass (90% by mass) and chromian diopside (10%) in a  
155 harzburgite xenolith (PAT2-68; Vannucci et al. 1998). The HISILB has a major element  
156 composition similar to that of HISILA but with  $K_2O$  replaced by  $Na_2O$ . This Na-rich  
157 HISILB with  $\sim 9$  wt%  $Na_2O$  content was designed to isolate the effect of sodium on the  
158 cpx–melt partitioning of trace elements. An additional composition, HISILC, has the  
159 same major element concentration as HISILB but is doped with different levels of  
160 minor and trace elements. Benefiting from seeding a few large crystals, two  
161 experiments were run with HISILA plus a proportion of diopside powder (4–8 wt%),  
162 namely HISILA4D and HISILA8D.

163 The starting compositions were chosen on the basis of several criteria: (i)  
164 composition HISILA has been studied both in nature and experiments such that phase  
165 relations are well constrained (Vannucci et al. 1998; Huang et al., 2006), ensuring the  
166 crystallization of cpx in equilibrium with a high proportion of quenched melt; (ii) cpx  
167 crystallizes at nearly constant equilibrium conditions, preventing complications in the  
168 interpretation of results stemming from the  $P$ – $T$ – $fO_2$  effects; (iii) the liquids produced  
169 cover a range of compositions, sufficient for defining the melt compositional  
170 dependence of partitioning.

171 The starting materials were synthesized by mixing reagent-grade oxides ( $SiO_2$ ,  
172  $Al_2O_3$ ,  $MgO$ , and  $TiO_2$ ) and carbonates ( $CaCO_3$ ,  $Na_2CO_3$ , and  $K_2CO_3$ ).  $FeO$  was added  
173 as a mixture of high-purity  $Fe$  and  $Fe_2O_3$  powders. After an initial grinding of 30 min  
174 in an agate mortar in ethanol, the mixtures were dried and placed in a platinum crucible

175 to decarbonate in a 1 atm furnace at 800 °C overnight, and then fused at 1450 °C for 2  
176 h. The quenched melt was re-ground in ethanol for 3 h and dried again. Selected trace  
177 elements (Rb, Sr, Sc, V, Cr, Mn, Co, Ni, Cu, Zn, Zr, Hf, Nb, Ta, La, Ce, Nd, Sm, Eu,  
178 Gd, Tb, Dy, Ho, Er, Yb, Lu) were then added to the starting mixture as oxides or  
179 nitrate/chloride solutions in an amount of totaling ~0.88 wt%, resulting in individual  
180 trace element concentrations of ~100 to 500 ppm (Table S2). The mixture with dopants  
181 was ground and heated to 1450 °C for 2 h, and then quenched in cool water. Analyses  
182 of the silicate glass confirmed its chemical homogeneity. The produced glass was re-  
183 ground and stored for subsequent experiments.

## 184 **2.2. One-atmosphere experiments**

185 A series of partitioning experiments were conducted by using Pt-crucibles in a  
186 high-temperature furnace. Run conditions were anhydrous and atmospheric with regard  
187 to pressure (1 atm) and oxygen fugacity (i.e., around  $\log fO_2 = -0.7$ ). The experimental  
188 set-up was chosen, in part because silica- and alkali-rich liquids, preserved as glass  
189 inclusions in spinel peridotite xenoliths, are mostly restricted to the shallowest (< 45  
190 km) part of the mantle (Hirschmann et al., 1998). Thus, experiments were designed to  
191 simulate the equilibration of silicic melts with cpx in close to the upper mantle  
192 conditions at low pressure.

193 In order to maximize crystal size and ensure equilibration of the growing crystals  
194 and melt, we implemented a cycled time-temperature path over the course of the study  
195 (Table S3; Fig. S1). In detail, the experiments were initially heated to super liquidus  
196 temperatures (1300–1450 °C) for 0.5–2 h to ensure complete melting of the powders

197 and chemical homogenization prior to crystallization. Runs were then cooled and  
198 maintained at 1150 °C, above the liquidus (1125 °C), for 2–3 h. Subsequently, runs  
199 were oscillated between 1130 °C and the target equilibrium temperatures (1070–  
200 1100 °C) with a slow cooling rate of 6–7.2 °C/h and a rapid heating rate of 90–200 °C/h,  
201 and the experiments were held at the equilibrium temperature for 5–10 h between  
202 oscillations. After several rounds of temperature cycle, runs were decreased to the final  
203 equilibrium temperature at rate of around 1 °C/h and maintained for 40–100 h to  
204 enhance crystal growth. A part of run products derived from HISILA, HISILB, HISILC,  
205 HISILA4D, and HISILA8D compositions was re-used as new starting materials for  
206 subsequent runs (Table S1). This experimental approach can create equilibrium cpx–  
207 melt pairs with a wide range of melt compositions and the crystalline precursors can  
208 serve as a nucleation point for crystals. At the temperature and atmospheric conditions  
209 of our experiments, no significant loss of Fe and Na from the melt is expected, as  
210 confirmed by electron microprobe analyses of the final fused glasses. All experiments  
211 were quenched by quickly dropping the Pt-crucibles into cool water. The recovered  
212 samples were then mounted in epoxy resin and polished for chemical analysis. A  
213 summary of the experimental conditions is given in Table S1.

### 214 **2.3. EPMA analysis**

215 Cpx and quenched glasses were analyzed for the major and minor element contents  
216 (Si, Al, Fe, Mg, Ca, Na, K, Ti, Cr, Mn, and Ni) by using a JEOL JXA-8100 electron  
217 probe micro-analyzer (EPMA) at the State Key Laboratory of Isotope Geochemistry,  
218 Guangzhou Institute of Geochemistry, Chinese Academy of Sciences (CAS). The

219 analytical conditions were an acceleration voltage of 15 kV, a beam current of 20 nA,  
220 and a defocused beam of 1  $\mu\text{m}$ . Peak counting time was as follows: 10 s for Na and K;  
221 20 s for Si, Al, Mg, Ca, and Fe; and 40 s for minor elements. Calibration was performed  
222 by using the following natural and synthetic reference materials: chrome diopside (Si,  
223 Mg, and Ca), almandine (Fe and Al), albite (Na), orthoclase (K),  $\text{Cr}_2\text{O}_3$  (Cr), rutile (Ti),  
224 and MnO (Mn). Data reduction was performed by using the ZAF correction program.  
225 The reproducibility and accuracy of the measurements were verified by using a set of  
226 secondary standards, which agree well with the recommended values (Table S4).  
227 Analytical precision is typically < 1% RSD for Si and < 5% RSD for analytes with  
228 concentrations > 0.5 wt%.

#### 229 **2.4. LA-ICP-MS analysis**

230 Major and trace element analyses of cpx and glasses were performed by LA-ICP-  
231 MS using either a GeolasPro 193 nm ArF excimer laser system attached to an Agilent  
232 7700e ICP-MS at the CAS Key Laboratory of Crust–Mantle Materials and  
233 Environments, University of Science and Technology of China, or a Photon Machines  
234 Excite 193 nm ArF excimer laser attached to an Agilent 7700x ICP-MS at the Nanjing  
235 FocuMS Technology Co. Ltd. Both instruments employ custom-built, fast-washout  
236 sample cells and use He as the sample chamber carrier gas (Hou et al., 2020). In both  
237 cases, the instrumental operating parameters were optimized by ablating NIST SRM  
238 610 to obtain maximum signal intensity for  $^{238}\text{U}$ , while keeping the  $\text{ThO}^+/\text{Th}^+$  ratio  
239 below 0.5% and the  $\text{U}^+/\text{Th}^+$  ratio close to 1.

240 The sequence of analysis began with four analyses of reference glasses (NIST

241 SRM 610, BHVO-2G, BCR-2G and BIR-1G), followed by analyses of samples and  
242 then four analyses of reference glasses again. NIST SRM 610 was repetitively analyzed  
243 every eight sample analyses for time-drift correction. Each individual analysis included  
244 a background acquisition of the first 30 s (gas blank) followed by 40 s for ablation and  
245 then 35 s for gas flow washing. We used a beam size of 30  $\mu\text{m}$  under single-spot mode  
246 for glasses while a beam size ranging between 10 and 20  $\mu\text{m}$  under single-spot or line  
247 scanning mode for cpx. Though the use of small beam size for cpx may reduce the  
248 analyte sensitivity, this problem was largely counteracted due to the high doping levels  
249 of the starting mixes (Table S2). For cpx analysis, only laser signals free of melt  
250 inclusions and spikes were integrated to yield the elemental concentration.

251 We applied the calibration strategy of summed metal oxide normalization for data  
252 reduction (Halicz and Günther 2004, Guillong et al. 2005, Liu et al. 2008), in which the  
253 sum of all metal oxides is normalized to 100% m/m. Three USGS glass reference  
254 materials (BHVO-2G, BCR-2G, and BIR-1G) were used for external standardization  
255 (He et al., 2016), which would provide more precise and accurate elemental  
256 concentrations relative to calibration by NIST SRM 610 (Liu et al. 2008). Off-line  
257 selection and integration of background and ablation signals, and time-drift correction  
258 and quantitative calibration were performed by ICPMSDataCal software (Liu et al.  
259 2008). Compared with previous calibration applying internal standardization, this  
260 method is able to obtain major and trace element contents simultaneously without an  
261 internal standard element pre-determined by EPMA.

262 The anticipated measurement precision (measured as relative standard deviation)

263 and accuracy (expressed as relative deviation from the reference value) for silicate  
264 glasses of this calibration method are better than 5% for major elements and 5–10% for  
265 trace elements, based on analyses of USGS and MPI-DING reference glasses (Liu et al.  
266 2008; He et al., 2016). To assess the accuracy of cpx analyses, we also used USGS,  
267 MPI-DING, and CGSG glasses as secondary standards. Under these operating  
268 conditions, at beam sizes equal or smaller than 20  $\mu\text{m}$ , precision and accuracy typically  
269 remain better than 10–20% for most elements (Table S5).

270

## 271 **3. Results**

### 272 **3.1. Run products**

273 All experiments produced cpx crystals with the coexisting melt quenched into  
274 glass and no other phases present (Table S1). Figure 1 shows the back-scattered electron  
275 (BSE) images of two representative cpx crystals from runs 5H05-1 and 9H07-2. Cpx  
276 crystals are generally euhedral/subhedral and blade shaped without zoning (Fig. 1),  
277 ranging from 10 to 50  $\mu\text{m}$  in the narrowest dimension, with the largest cpx found in the  
278 melts of the HISILB/ HISILC composition. Based on optical inspection the modal  
279 proportions of cpx crystals are generally < 20%, whereas the quenched melts are clear  
280 glasses without any dendritic quench texture (Fig. 1).

### 281 **3.2. Major element compositions of the silicate glass and cpx**

282 *Glass.* The major element composition of the run product glasses from LA-ICP-  
283 MS analyses is summarized in Table 2. The glasses in all runs are homogenous in

284 chemistry as illustrated by the small standard deviations in major element analyses  
285 (Table 2). On a total-alkalis vs. silica diagram the glasses are predominantly  
286 trachydacitic in composition, ranging from 8.8 to 9.7 wt% in total alkali ( $\text{Na}_2\text{O} + \text{K}_2\text{O}$ )  
287 content and from 67.6 to 69.1 wt% in the  $\text{SiO}_2$  content, resembling the silicic glass  
288 observed in a harzburgite xenolith (PAT2-68) from Vannucci et al. (1998). The glasses  
289 are also characterized by high  $\text{Al}_2\text{O}_3$  (14.10–15.51 wt%), and low MgO (1.35–2.56  
290 wt%),  $\text{FeO}_t$  (1.54–2.11 wt%),  $\text{TiO}_2$  (0.11–0.2 wt%) and CaO (1.81–3.18 wt%). Their  
291 alumina saturation index (ASI: molar ratio  $\text{Al}_2\text{O}_3/(\text{Na}_2\text{O} + \text{K}_2\text{O} + \text{CaO})$ ) is 0.66 to 0.89,  
292 resulting in a range of compositions from metaluminous to peralkaline, and the Mg#s  
293 ( $= 100 \times \text{Mg}/(\text{Mg} + \text{Fe})$  in molar fraction) of these glasses range from 60.1 to 74. The  
294 degree of polymerization can be described by the NBO/T parameter (the number of  
295 non-bridging oxygens per tetrahedral cation; Mysen and Virgo, 1980), which ranges  
296 from 0.10 to 0.22 in our experiments (Table 2).

297 **Cpx.** The major element composition of the run product cpx from EPMA analyses  
298 is summarized in Table 3. Cpx is also homogeneous in a single run but its composition  
299 slightly varies with changes of experimental starting composition and conditions (Table  
300 3). Cpx is Ca- and Mg-rich (21.16–23.3 wt% CaO; 17.34–18.5 wt% MgO) but Al-poor  
301 (1.7–3.1 wt%) and contains 0.49–0.95 wt%  $\text{Na}_2\text{O}$ . In general, all Fe (1.89–2.61 wt%  
302  $\text{FeO}_t$ ) is assumed to be present as ferric iron in cpx based on the high oxygen fugacity  
303 during the runs. For reference, cpx compositions are reported in cations and end-  
304 member mole fractions on a six-oxygen basis in Table S6. All cpx crystals produced in  
305 these experiments are diopside (Di74–88) in composition with relatively low aegirine

306 (Ae0–5) and enstatite (En1–12) components, whereas their Mg#s range from 92.5 to  
307 94.2. The produced cpx in this study therefore resembles the typical diopside coexisting  
308 with silicic melts in mantle xenoliths (Vannucci et al., 1998), in terms of high SiO<sub>2</sub>,  
309 CaO, MgO, and low Al<sub>2</sub>O<sub>3</sub> contents.

### 310 **3.3. Cpx–melt partition coefficients**

311 Partition coefficients in each experimental run, calculated from LA-ICP-MS or  
312 EPMA analyses of cpx–glass pairs (Tables 2–3, S7–S8), are summarized in Table 4.  
313 All partition coefficients are reported with 1σ errors calculated by propagating the  
314 uncertainties (relative standard deviation) on the concentrations in each phase:  $\sigma_D =$   
315  $D^{cpx/melt} \times \sqrt{RSD_{Cpx}^2 + RSD_{Cmelt}^2}$ . Both concentrations and partition coefficients for  
316 Ni, Cr, Fe, and Ti by the two analytical methods agree within errors and show no  
317 systematic bias (Fig. 2), except for Mn. Reasons for the slight offset in Mn data from  
318 the 1:1 line in Fig. 2 most likely arise from the analytical limit for low-concentration  
319 elements by EPMA, and in this case,  $D_{Mn}$  were calculated based on LA-ICP-MS  
320 analyses of cpx. The partition coefficients for LILE and REE from nine representative  
321 runs, together with literature data, are depicted in Fig. S2. As expected, both Na and Sr  
322 are incompatible in cpx. Notably,  $D_{Sr}$  (0.16–0.33) compare well with the range reported  
323 by Huang et al. (2006) and those determined from silica-rich glass–diopside pairs in  
324 mantle xenoliths (Vannucci et al. 1998), but are elevated relative to typical cpx–basalt  
325 values (e.g., 0.089–0.115 in Lundstrom et al., 1994). The REE show a gradual increase  
326 in compatibility from light REE (e.g.,  $D_{La} = 0.26–0.4$ ) to middle–heavy REE (e.g.,  $D_{Yb}$   
327  $= 1.27–2.27$ ), with a maximum occurring in the middle REE, typically Dy. This REE

328 partitioning pattern is in excellent agreement with previous results experimentally  
329 measured in silicic system by Huang et al. (2006) while is much higher than those  
330 between cpx and basaltic or andesitic melts (e.g., Hart and Dunn, 1993; Skulski et al.,  
331 1994; Hill et al., 2000). Because the cpx-silicic melt partitioning for LILE and REE has  
332 been fully addressed (Huang et al., 2006), we focus here on FRTE and HFSE  
333 partitioning behavior.

334 A comparison of our measured FRTE and HFSE partition coefficients from nine  
335 representative runs with those from the literature is shown in Fig. 3. To the first order,  
336 the trends in measured  $D_{\text{FRTE}}$  and  $D_{\text{HFSE}}$  are similar among experiments. Most FRTE  
337 (i.e., Mn, Co, Ni, Sc, and Cr) are moderately to highly compatible in cpx, with Ni being  
338 the most compatible element. Cu and Zn are in general incompatible, with  $D$  values  
339 between 0.1 and 1. As regards the multivalent transition metals, Fe is slightly  
340 compatible in cpx whereas V behaves incompatibly and exhibits the lowest  $D$  values  
341 (0.06–0.14). Our  $D_{\text{V}}$  values are considerably lower than most other experimental  
342 partitioning studies (e.g., Toplis and Corgne, 2002; Mallmann and O'Neill, 2009; Li,  
343 2018; Holycross and Cottrell, 2022). The difference is likely due to the higher oxygen  
344 fugacity ( $f_{\text{O}_2}$ ) in the system of this study, resulting in elevated V oxidation states  
345 (predominantly  $\text{V}^{5+}$ ). Overall, the compatibility of the FRTE increases in the order  $\text{V} <$   
346  $\text{Cu} < \text{Zn} < \text{Fe} < \text{Mn} < \text{Co} < \text{Sc} < \text{Cr} < \text{Ni}$  (Fig. 3). With respect to HFSE, the  $D$ s vary  
347 from highly incompatible for Nb and Ta and to mildly incompatible for Zr, Hf, and Ti  
348 (Fig. 3). For comparison, the FRTE and HFSE partition coefficients obtained here are  
349 generally above or at the high end of the literature data associated with basalts (Fig. 3).

350 Finally, at the run conditions, the limited equilibrium temperature (1070 to 1100 °C)  
351 range exerts no obvious influence on  $D$  values for FRTE and HFSE (Fig. S3).

### 352 **3.4. Equilibrium partitioning and Henry's law behavior**

353 Several lines of evidence support the attainment of chemical equilibrium during  
354 the experiments. First, run durations were much longer than those of previous studies  
355 that reached equilibrium (e.g., Hart and Dunn, 1993; Hill et al., 2000; Bennett et al.,  
356 2004). Second, major and trace element compositions of cpx and melt phase in each  
357 run were found to be widely homogeneous with standard deviations usually in the range  
358 of 10% RSD or below (Tables 2–3; Tables S7–S8). To be more specific, EPMA analyses  
359 of core–rim transect across the cpx grain revealed no major or minor element zonation  
360 (Fig. 4a; Table S9), especially for slow diffusing cations,  $Al^{3+}$  and  $Ti^{4+}$ , indicating steady  
361 state cpx growth without the development of a diffusive boundary layer (Holycross and  
362 Watson 2016; 2018). Moreover, the random analyses of major element in glass also  
363 revealed no detectable variations outside of analytical errors (Fig. 4b; Table S9). Third,  
364 the partition coefficients for K in our experiments (0.01–0.015) are close to those  
365 previously reported (e.g., 0.015–0.045 in Johnson, 1998), precluding the enrichment in  
366 highly incompatible elements due to fast non-equilibrium growth (Huang et al., 2006).  
367 Finally, the  $D$ s for most investigated trace elements can be well described by the crystal  
368 lattice strain model, indicating that our measured partition coefficients represent  
369 equilibrium values.

370 Henry's law behavior of element partitioning between cpx and melt in our  
371 experiments is assumed for the following reasons. First, the total dopant levels ( $<0.9$

372 wt%) in the starting material utilized in this study as well as the element concentrations  
373 measured in the run products (Tables S2, S8) are well below the anticipated upper limits  
374 of Henrian behavior (Watson, 1985). Second, the experiments with HISILB/HISILC as  
375 starting materials show comparable trace element  $D$ s under the same experimental  
376 conditions (e.g., runs 6H05-2 and 7H05-3; see Table 4), despite the fact that the dopants  
377 for trace elements are different. Therefore, cpx–melt partitioning in our experiments  
378 should obey Henry’s law.

379

## 380 **4. Discussion**

### 381 **4.1. Lattice strain model**

382 The lattice strain model (LSM) has been widely applied to trace element  
383 partitioning between minerals and melt over the last few decades (e.g., Brice, 1975;  
384 Blundy and Wood, 1994; Wood and Blundy, 1997; Mollo et al., 2020). In general, the  
385 partitioning of isovalent elements into the same crystallographic site of a mineral  
386 always shows a near-parabolic dependence on the ionic radii (e.g., an Onuma diagram;  
387 Onuma et al., 1968), which can be quantitatively described by the lattice strain equation  
388 (Blundy and Wood 1994) as follows:

389

$$390 \quad D_i = D_0 * \exp \left( \frac{-4\pi EN_A}{RT} \left( \frac{r_0}{2} (r_i - r_0)^2 + \frac{1}{3} (r_i - r_0)^3 \right) \right) \quad (1)$$

391

392 where  $D_i$  is the partition coefficient of element  $i$  with ionic radius  $r_i$ ,  $D_0$  is the partition  
393 coefficient for the strain-free substitution,  $r_0$  is the optimum ionic radius of the strain-

394 free lattice site,  $E$  is the apparent Young's modulus (GPa) of the lattice site,  $R$  is the  
395 universal gas constant ( $=8.31$  J/mol/K),  $N_A$  is Avogadro's number, and  $T$  is temperature  
396 in Kelvin. Of these lattice terms,  $D_0$  and  $r_0$  define the apex location of a parabola on a  
397 plot of  $D_i$  vs. ionic radius. The apparent Young's modulus  $E$  determines the width of the  
398 parabola, which reflects the elastic response of the crystallographic site to lattice strain  
399 due to the incorporation of elements of non-ideal size (Blundy and Wood, 2003).

400 In cpx, the sixfold coordinated M1 site incorporates most transition elements and  
401 HFSE, including divalent (Co, Ni), trivalent (Cr, Sc), tetravalent (Ti, Zr, Hf), and  
402 pentavalent (V, Nb, Ta) cations, whereas the eightfold coordinated M2 site  
403 preferentially incorporates the larger cations like monovalent Na, K, and Rb, divalent  
404 Ca, Mn, Sr, and Ba, trivalent REE, and tetravalent U and Th (e.g., Blundy and Wood  
405 1994, 2003; Hill et al. 2000). The LSM was applied to our  $D$ s by a non-linear weighted  
406 least-squares regression using Eq. (1), which calculates the best-fit values of the lattice  
407 strain parameters ( $r_0$ ,  $D_0$ , and  $E$ ) for both M-sites in cpx. In presence of a low number  
408 of measured  $D$  for isovalent elements entering the same site, it failed to constrain the  
409 lattice strain parameters in regression analysis. Therefore, we choose to fit only the  $D_0$   
410 and  $E$  parameters, whereas  $r_0$  for each experiment is fixed in a reasonable range  
411 depending on the valence of the trace elements (Dalou et al., 2018). The resultant  
412 parabolas of  $D$  vs. ionic radius derived from two individual experiments are shown in  
413 Fig. 5 (remaining experiments are shown in Fig. S4), and the fitting parameters are  
414 listed in Table S10. These lattice strain parabolas thus enable us to assess the quality of  
415 the fit of the experimentally determined  $D$  values to the LSM. Parabola fits for di-, tri-,

416 tetra-, and pentavalent elements are described below.

417 For cations entering the M1 site, the fits of the experimental partitioning data for  
418 divalent ( $\text{Co}^{2+}$ ,  $\text{Ni}^{2+}$ ,  $\text{Mg}^{2+}$ , and  $\text{Zn}^{2+}$ ), trivalent ( $\text{Al}^{3+}$ ,  $\text{Cr}^{3+}$ , and  $\text{Sc}^{3+}$ ), tetravalent ( $\text{Ti}^{4+}$ ,  
419  $\text{Zr}^{4+}$ , and  $\text{Hf}^{4+}$ ), and pentavalent ( $\text{V}^{5+}$ ,  $\text{Nb}^{5+}$ , and  $\text{Ta}^{5+}$ ) cations form distinct parabolas  
420 (Fig. 5). Though parabolas for HFSE and 3+ cations on the M1 site are defined by just  
421 three elements they are in excellent agreement with the experimentally determined  
422 values for all experiments. Note the very few cations that fall off the parabola in some  
423 cases are  $\text{Cu}^{2+}$  and  $\text{Fe}^{3+}$ , which may be due to analytical uncertainties or the crystal-field  
424 effects associated with the mineral-melt partitioning of transition metals (e.g., Mn, Co,  
425 Ni, Cu, Fe, Cr, and V; Blundy and Wood, 1994; Cartier et al., 2014).

426 Consistent with previous studies (e.g., Hill et al., 2000; Dygert et al., 2014;  
427 Bonechi et al., 2021), we observe a higher Young's modulus and hence tighter parabola  
428 on the M1 site with respect to M2 site (Fig. 5; Table S10). Notably, the high effective  
429 Young's moduli for 3+ cations (1554–2100 GPa) obtained on the M1 site is overall  
430 close to the measured value for diopsidic cpx (1635 GPa; Hill et al., 2000). The inferred  
431 values for  $E^{4+}$  (~1147–2472 GPa) from Ti-Hf-Zr regression are also consistent with the  
432 large values calculated in previous studies (~800–2400 GPa; Lundstrom et al., 1998;  
433 Hill et al., 2011; Baudouin et al., 2020). These observations are accounted for by the  
434 elastic properties of the lattice site in which the M1 site is stiffer than the M2 site in cpx  
435 (Hill et al., 2011).

436 The larger divalent cation  $\text{Mn}^{2+}$  prefers the M2 site along with  $\text{Ca}^{2+}$  and  $\text{Sr}^{2+}$  and  
437 they define parabolas of wider shape, which yield  $D_0$  of 8–15,  $r_0$  of 1.068–1.086 Å, and

438  $E$  of 268–304, respectively. Among these divalent cations,  $\text{Ca}^{2+}$  is the cation closer to  
439 the apex of parabolas and Mn is more compatible than Sr. It is also clear in Fig. 5 that  
440 the parabolas for trivalent elements on the M2 site match well with our experimentally  
441 determined  $D_{\text{REE}}$  values. Lattice strain parameters for REE partitioning are  $D_0 = 1.58$ –  
442 2.41,  $r_0 = 1.021$ – $1.038 \text{ \AA}$ ,  $E = 235$ – $365$ , consistent with the range ( $D_0 = 1.09$ – $3.62$ ,  $r_0$   
443  $= 1.006$ – $1.041$ ,  $E = 255$ – $366$ ) experimentally obtained in silicic systems (Huang et al.,  
444 2006).

#### 445 **4.2. Effect of crystal chemistry on $D_{\text{FRTE}}$ and $D_{\text{HFSE}}$**

446 In general, element partitioning between minerals and melts depends on both  
447 crystal chemistry and the melt composition. Although our original intention was to  
448 isolate the effect of sodium, the cpx composition of these experiments shows more  
449 differences than simply Na content. The melt composition is also considerably different  
450 among experiments. Therefore, it presents a challenge in determining the effect of any  
451 single compositional parameter independently. However, an advantage of our  
452 experimental design is that all experiments were conducted in a simple low-variance  
453 system by using a common crystal composition ( $\text{CaMgSi}_2\text{O}_6$ ). This approach allows us  
454 to evaluate the relative importance of crystal versus melt compositional effects on cpx–  
455 melt partitioning under specific conditions.

456 The systematic relationship between  $D$ s and ionic radii along the Onuma parabola  
457 (Fig. 5) is a clear indication that the incorporation of trace elements into the crystal site  
458 is controlled by cpx structural constituent. Previous studies have shown that  $^{\text{iv}}\text{Al}$  and

459 Na content in cpx are the primary compositional variables in determining the cpx–melt  
460 partition coefficient (e.g., Lundstrom et al., 1994; Skulski et al., 1994; Hill et al., 2000;  
461 Bennett et al., 2004; Michely et al., 2017). The cpx crystals in this study are diopsidic  
462 with variable Al<sub>2</sub>O<sub>3</sub> and Na<sub>2</sub>O contents, similar to those in equilibrium with basaltic  
463 melts in many previous partitioning studies (e.g., Forsythe et al., 1994; Johnson, 1998;  
464 Adam and Green, 2006). To evaluate and quantify the control of these variables on  
465 FRTE and HFSE partitioning, our data were compared with these previous studies.

466 Tetrahedrally coordinated Al in cpx has been argued to increase *D*s for tri-, tetra-,  
467 and pentavalent cations by providing a charge balancing mechanism for their  
468 substitution into M site (e.g., Lundstrom et al., 1994; Gaetani and Grove, 1995; Hill et  
469 al., 2000). For example, Wood and Trigila (2001) demonstrated that partition  
470 coefficients increase by one order of magnitude for Ti and two orders of magnitude for  
471 Zr, Nb, and Ta when the <sup>iv</sup>Al of cpx increases from 0.02 to 0.5. In this study, the Al<sub>2</sub>O<sub>3</sub>  
472 contents in cpx range from 1.7 to 3.1 wt%, resulting in <sup>iv</sup>Al varying from 0.05 to 0.10  
473 (p.f.u.). In Fig. 6, *D*s for Sc, Zr, Hf, and Nb are plotted against <sup>iv</sup>Al in cpx (see also Fig.  
474 S5 for Cr, Ti, and Ta). To first order, *D*<sub>Cr</sub> and all *D*<sub>HFSE</sub> are systematically correlated with  
475 <sup>iv</sup>Al within our new experimental dataset, supporting the <sup>iv</sup>Al as a control on highly  
476 charged element partitioning. For Sc, this dependency is more subtle given the small  
477 *D*<sub>Sc</sub> range (3.62–6.35). Interestingly augite–basalt partitioning of Sc has been also  
478 shown to be relatively constant, but rather dependent on experimental *T* (Dygert et al.,  
479 2014). It is not surprising then that once the literature data is considered, the global  
480 dataset might not be fitted with one universal trend, which may be expected given the

481 big gaps in melt compositions and experimental  $P$ - $T$  conditions.

482 Sodium in cpx may also exert an important control on high-valence trace element  
483 partitioning by increasing the charge-balanced configurations (Wood and Blundy 1997;  
484 Blundy et al., 1998; Bennett et al., 2004). Bennett et al. (2004) performed experiments  
485 in the NCMAS system and observed that the introduction of Na into cpx could lead to  
486 an increase by at least three orders of magnitude for  $D_{\text{HFSE}}$  relative to those in the Na-  
487 free system. Our study utilized both high- and low-Na compositions in the starting  
488 materials, which produced a range of Na content in cpx (expressed in moles p.f.u., Na  
489 = 0.035–0.067). However, as shown in Figs. 7 and S6, there is no indication that the Na  
490 in cpx has any significant relationship with Sc, Cr, and HFSE partitioning. This can be  
491 explained by the fact that our experiments were done with much less Na than achieved  
492 by experiments in Bennett et al. (2004), in which a remarkable effect of Na in cpx has  
493 only been observed for more Na-rich (2.2–13.5 wt% Na<sub>2</sub>O) cpx at high-pressure  
494 conditions.

495 In sum, the data examined in this study confirm that an <sup>iv</sup>Al-controlled substitution  
496 mechanism for highly charged transition metal and HFSE partitioning extends to silicic  
497 systems. When the results from these and other partitioning studies are compared (Figs.  
498 6–7), it becomes apparent that the elevated partition coefficients in our experiments  
499 cannot be reconciled by considering only the cpx composition. We would further argue  
500 for a role of melt composition on FRTE and HFSE partitioning which is discussed in  
501 the following section.

### 502 **4.3. Effect of melt composition on $D_{\text{FRTE}}$ and $D_{\text{HFSE}}$**

503 It has long been recognized that melt composition influences the melt structure,  
504 and thus trace element  $D$ s, that generally increase with increasing melt polymerization  
505 (e.g., Ryerson and Hess, 1978; Mysen, 2004). Our new results, combined with those  
506 from previous studies, are useful for elucidating melt composition/structure controls on  
507 FRTE and HFSE partitioning. Basically, NBO/T has been widely used as an expression  
508 of the melt structure, and more specifically the degree of polymerization (with lower  
509 values indicating more polymerized melts; Mysen and Virgo, 1980; Mysen et al., 1985).  
510 Several studies have emphasized that the influence of melt structure becomes more  
511 effective when NBO/T < 0.49 (e.g., Gaetani, 2004; Huang et al., 2006; Mollo et al.,  
512 2016). Other studies showed that trace element partitioning depends on the major  
513 element components in melt (e.g., O'Neill and Eggins, 2002; Evans et al. 2008; Dygert  
514 et al., 2013). In this regard, Leitzke et al. (2016) observed that in basaltic systems,  $D$   
515  $\frac{\text{cpx/melt}}{\text{HFSE}}$  decreases with increasing  $\text{TiO}_2$  in the melt. Some studies have also highlighted  
516 that the molar  $\text{Ca}^{2+}/(\text{M}^+ + \text{M}^{2+})$  of the melt, where  $\text{M}^+$  and  $\text{M}^{2+}$  are  $\text{Na}^+ + \text{K}^+$  and  $\text{Fe}^{2+}$   
517  $+ \text{Ca}^{2+} + \text{Mg}^{2+}$  cations, respectively, exerts a significant control on  $D_{\text{REE}}^{\text{cpx/melt}}$  (Huang et al.,  
518 2006; Mollo et al., 2016). Furthermore, the global fit of cpx–melt partitioning data  
519 revealed that most  $D$ s increase as melt MgO, Mg#, CaO, and FeO contents drop, and  
520 as melt  $\text{SiO}_2$  and  $\text{Na}_2\text{O} + \text{K}_2\text{O}$  increase (Bédard, 2014).

521 Our dataset offers the opportunity to shed further light on these melt compositional  
522 effects. Taking into account of the structural environment of FRTE and HFSE in silicic  
523 melt (e.g., Schmidt et al., 2006), we investigate several melt variables that potentially

524 influence cpx–melt partitioning. In Figs. 8–10 we illustrate the variations in  $D_s$  with  
525 different melt parameters (NBO/T,  $Mg^{2+}/(M^+ + M^{2+})$ , ASI, and  $Na_2O + K_2O$ ,  
526 respectively). It is noteworthy to mention that these variables are interrelated to some  
527 extent, so they cannot be considered exclusive of one another.

528 **4.3.1. Partitioning of FRTE and HFSE as a function of NBO/T.** In Fig. 8,  
529 partition coefficients for Co, Ni, Mn, Cu, Zn, Cr, Sc, Ti, Zr, Hf, Nb, and Ta from this  
530 study and the literature are plotted against NBO/T. Among the global dataset, in which  
531 cpx composition remains nearly constant (i.e., diopsidic with  $^{iv}Al < 0.2$ ) over a range  
532 of NBO/T (0–1.2; Fig. 8), we are able to assess the isolated effect of melt composition.  
533 The systematic correlations of most  $D_s$  and especially  $D_{Co}$ ,  $D_{Ni}$ , and  $D_{Mn}$  with NBO/T  
534 are broadly consistent with previous work which established that the polymerization of  
535 melt influences remarkably the partitioning of high  $Z^+/r_i$  cations such as FRTE and  
536 HFSE (e.g., Ryerson and Hess, 1978; Schmidt et al., 2006). In detail, our new  
537 experimental data fall within or extend the trends established by the global dataset at  
538 the low NBO/T end of the distribution (Fig. 8). Moreover, in agreement with Gaetani  
539 (2004), we observe stronger dependence of the  $D_s$  on NBO/T in more polymerized  
540 melts with  $NBO/T < 0.49$  when compared to the cpx–basalt partitioning.

541 Although the partitioning data discussed above are consistent with a melt  
542 structural control, it has been pointed out in many previous studies that melt  
543 compositional effects are not fully captured by NBO/T (e.g., Bennett et al., 2004; Huang  
544 et al., 2006; Michely et al., 2017). Indeed, NBO/T treats structural role of individual  
545  $M^+$  ( $Na^+$  and  $K^+$ ) and  $M^{2+}$  ( $Mg^{2+}$ ,  $Ca^{2+}$ , and  $Fe^{2+}$ ) cations the same, and therefore does

546 not account for their fundamentally different behaviors in affecting melt structure (e.g.,  
547 Mysen et al., 1985). The same is also likely true for some aspects of the NBO/T– $D_s$   
548 relationships in Fig. 8. For example, in light of our new data,  $D_{\text{HFSE}}$  does not increase  
549 regularly with decreasing NBO/T from 0.22 to 0.1. As demonstrated, the highest  $D_{\text{HFSE}}$   
550 commonly shares a characteristic moderate NBO/T (e.g., maximum  $D_{\text{HF}}$  at NBO/T =  
551 0.15). Thus, the above considerations suggest that NBO/T is insufficient as a single melt  
552 compositional term to describe the partitioning behavior of FRTE and HFSE in our  
553 experiments.

554 **4.3.2.  $D_{\text{FRTE}}$  and melt  $\text{Mg}^{2+}/(\text{M}^+ + \text{M}^{2+})$ .** Partition coefficients obtained in this  
555 study confirm that, for a highly polymerized silicate melt with NBO/T < 0.2, small  
556 changes in melt structure have a dramatic effect. While NBO/T cannot adequately  
557 account for all subtleties in the strongly polymerized melt, a more subtle investigation  
558 of the influence of melt structure on trace element partitioning is necessary. The  
559 compositional changes in the melt among our experiments mostly reflect variation in  
560 Mg, Ca, and total alkali content, depending on starting materials and the proportion of  
561 crystals. Informed by the negative correlation of  $D_{\text{REE}}^{\text{cpx/melt}}$  with melt molar  $\text{Ca}^{2+}/(\text{M}^+ +$   
562  $\text{M}^{2+})$  in silicic systems (Huang et al., 2006), we test  $\text{Mg}^{2+}/(\text{M}^+ + \text{M}^{2+})$  as a potential melt  
563 descriptor to account for the influence of melt structure on FRTE partitioning.  
564 Rationally, transition metals with similar charge and size to  $\text{Mg}^{2+}$  are expected to show  
565 more favorable incorporation into the melts with higher ratio of molar  $\text{Mg}^{2+}/(\text{M}^+ + \text{M}^{2+})$ ,  
566 thereby reducing their partitioning into the crystalline phase. Besides, as our melts vary  
567 in  $\text{Al}_2\text{O}_3/(\text{Na}_2\text{O} + \text{K}_2\text{O} + \text{CaO})$  with almost constant  $\text{SiO}_2$  contents (Table 2), the ASI

568 parameter is used as an alternative indication of melt polymerization (Prowatke and  
569 Klemme, 2005).

570 Fig. 9 displays the relationships between  $D_{\text{FRTE}}$  with ASI (Fig. 9a-b) and  $\text{Mg}^{2+}/(\text{M}^+$   
571  $+ \text{M}^{2+})$  (Fig. 9c-d) for the experimental data reported here. It shows that, with the  
572 exception of Cu, Zn, and Sc showing small  $D_s$  variations, partition coefficients of Co,  
573 Mn, Ni, and Cr are positively correlated with melt ASI. These well-defined trends agree  
574 well with the notion that minerals crystallized in the more strongly polymerized melts  
575 with higher ASI generally show increased  $D_s$ , which is also well established in the  
576 literature (e.g., Prowatke and Klemme, 2005). On the other hand,  $D_s$  of Co, Ni, and to  
577 a lesser extent Mn, show negative dependencies on melt  $\text{Mg}^{2+}/(\text{M}^+ + \text{M}^{2+})$ , although the  
578 trends for the other transition metals are not clear enough to be definitive. These results  
579 strongly recall those of Huang et al. (2006) and further indicate a direct link between  
580 melt Mg and FRTE partitioning behavior. Such a causal relationship could be expected  
581 from the structural role of Mg in the melt.

582 Previous studies have suggested that the constitution of the melt network is  
583 governed by an equilibrium reaction (e.g., Stebbins, 1987):  $2\text{Q}^n \leftrightarrow \text{Q}^{n-1} + \text{Q}^{n+1}$  (where  
584  $\text{Q}^n$  denotes species in silicate melt with different numbers of bridging oxygen;  $0 \leq n \leq$   
585  $4$ ). The reaction shifts to the right as the charge density (i.e.,  $Z^+/r_i$ ) of the network-  
586 modifying cations increases (Maekawa et al., 1991; Mysen, 2004). Thus, as the  $Z^+/r_i$   
587 increases with increasing  $\text{Mg}^{2+}/(\text{M}^+ + \text{M}^{2+})$  in the melt, the network tends to be less  
588 regular with both larger and smaller sites produced in the structure. This increase in the  
589 number of available sites would be critical for accommodating some transition metals

590 in the melt, resulting in lower  $D_s$ . Particularly, the charge and ionic radius of both  $\text{Co}^{2+}$   
591 and  $\text{Ni}^{2+}$  (in 6-fold coordination, 0.65 and 0.69 Å, respectively; Shannon, 1976) make  
592 it much easier to substitute for  $\text{Mg}^{2+}$  (0.72 Å in 6-fold coordination) than other mono-  
593 or divalent cations (e.g.,  $\text{Na}^+$ ,  $\text{Ca}^{2+}$ ) in the melt, thus giving rise to the strong negative  
594 correlations between their  $D_s$  and the melt  $\text{Mg}^{2+}/(\text{M}^+ + \text{M}^{2+})$ .

595 **4.3.3.  $D_{\text{HFSE}}$  and melt alkalis.** Several previous studies have proposed that the  
596 effect of melt composition on mineral–melt  $D_s$  is related to trace element solubility in  
597 silicate melts (e.g., Watson, 1979; O'Neill and Eggins, 2002; Bennett et al., 2004;  
598 Dygert et al., 2013; Michely et al., 2017). In this sense, the formation of melt complexes  
599 involving the trace element of interest and a major element in the melt, such as CaO  
600 (e.g.,  $\text{CaMoO}_3$  and  $\text{CaMoO}_4$  complexes; O'Neill and Eggins, 2002), could change the  
601 activity coefficient of the trace element, thereby affecting its partitioning behavior. For  
602 HFSE, melt complexation with alkalis is thought to significantly affect its solubility in  
603 high-silica melts (e.g., Watson, 1979; Horng et al., 1999; Linnen and Keppler, 2002).  
604 For example, Watson (1979) demonstrated that Zr solubility in silicic melts increases  
605 with increasing alkali in excess of  $\text{Al}_2\text{O}_3$  because of the formation of melt  $\text{Na}_4\text{Zr}(\text{SiO}_4)_2$   
606 and  $\text{K}_4\text{Zr}(\text{SiO}_4)_2$  complexes.

607 The remarkable feature of the results presented here is that all HFSE partition  
608 coefficients depend on melt composition in a similar way, which are illustrated by the  
609 negative correlations between  $D_{\text{HFSE}}$  and melt total alkali ( $\text{Na}_2\text{O} + \text{K}_2\text{O}$ ) contents (Fig.  
610 10). These correlations indicate that some fundamental chemical properties of the melts  
611 governed by alkalis control the activity coefficients of HFSE. Indeed, other than  $\text{Zr}^{4+}$ ,

612 previous studies have also suggested that  $K^+$  has a strong influence on enhancing the  
613 solubilities (and lowering activity coefficients) of both Nb and Ta, which forms  $Nb^{+5}$ -  
614 O-K and  $Ta^{+5}$ -O-K species in peralkaline  $SiO_2$ - $K_2O$ - $Al_2O_3$  melts (Hornig et al., 1999).  
615 It follows that the strong negative dependencies of  $D_{HFSE}$  on melt total alkalis could be  
616 interpreted as evidence for the formation of alkali-HFSE complexes in melts, which  
617 enhances the solubility of HFSE and, consequently, leads to a decrease of their  $D_s$ .

618 Alternatively, HFSE may act as network-formers and copolymerize with  $SiO_2$  (e.g.,  
619  $Ti^{4+}$ ,  $Zr^{4+}$ , and  $P^{5+}$  replace  $Si^{4+}$  in tetrahedral coordination; Ryerson, 1985) when the  
620 silicic melts evolve towards more alkaline, explaining the negative dependencies of  
621  $D_{HFSE}$  on melt alkalis. A more definitive constraint on the HFSE partitioning must await  
622 future detailed research on their structural environment in the melts of our compositions.

623

## 624 **5. Implications**

625 Mantle partial melting and metasomatism are two fundamental processes that have  
626 contributed to the chemical heterogeneities of the Earth and play a key role in the  
627 evolution of the lithosphere (e.g., Albarède, 1998; Lundstrom, 2000; Gao et al., 2004;  
628 Yu et al., 2006; Kiseeva et al., 2013; Koppers et al., 2021). The most common approach  
629 to quantifying melt extraction and metasomatic processes that occur in the lithospheric  
630 mantle has been to use the partition coefficients between minerals and melts (silicate  
631 and/or carbonate) and mineral compositions in mantle-derived xenoliths (e.g., Ionov,  
632 2010; Kiseeva et al., 2017, 2018; Li et al., 2022).

633 This study extends previous experimental work constraining the partitioning and

634 extraction of trace elements during melting of the shallow mantle, which has been on  
635 silicic, alkaline, aluminous systems (Huang et al., 2006). To address this aspect, our  
636 experiments were performed at relatively low pressure and the cpx compositions  
637 resemble those from natural depleted peridotite. We build upon our new partition  
638 coefficient dataset, as well as comparison with previous work, to explore the extents to  
639 which crystal versus melt compositions control FRTE and HFSE partitioning between  
640 cpx and silicic melts. This discussion is particularly relevant to petrogenetic studies of  
641 the interaction between silicic melt and peridotite that occurs in a variety of tectonic  
642 settings, as has been documented by interstitial xenolith glasses (e.g., Xu et al., 1996;  
643 Coltorti et al., 2000; Miller et al., 2012).

644 Our newly determined  $D_s$  in silicic systems differ significantly from those in mafic  
645 systems in two aspects. First, most trace elements exhibit considerably higher  
646 compatibilities in cpx from silicic melts relative to those from more mafic melts (Figs.  
647 3 and 8). Second, our results show marked variability in  $D_s$  for a number of trace  
648 elements (e.g., two-fold or greater variations for  $D_{Ni}$ ,  $D_{Cr}$ ,  $D_{Nb}$ , and  $D_{Ta}$ ) within melt  
649 compositions that are restricted to two-percent silica interval (67.6–69.1 wt% SiO<sub>2</sub>).  
650 We demonstrate that coupled substitution involving Al in the tetrahedral site is the  
651 major charge balance mechanism by which the highly charged trace elements are  
652 incorporated into the cpx. As for the Na-coupled substitution, the cpx compositions  
653 seem not high enough in Na content for this mechanism to be significant.

654 Our findings in conjunction with literature data also highlight the strong influence  
655 of melt composition/structure on FRTE and HFSE partition coefficients. With

656 increasing polymerization of the melt (i.e., lower NBO/T or higher ASI),  $D_{\text{FRTE}}$  and  
657  $D_{\text{HFSE}}$  roughly increase to different extent. Moreover, we identified simple correlations  
658 between  $D_s$  and a range of melt parameters (i.e., melt  $\text{Mg}^{2+}/(\text{M}^+ + \text{M}^{2+})$  and total alkali  
659 contents) for individual FRTE and HFSE, supporting the role of melt NBO species or  
660 complexing ligands in controlling the partitioning of these elements. The implication is  
661 that even very small changes in melt major-element compositions can greatly affect  
662 element partitioning in strongly polymerized silicic systems.

663 It should be further stressed that the effect of melt composition will not only affect  
664 the partitioning of trace elements between melt and cpx, but also additional minerals  
665 such as amphibole (Tiepolo et al., 2007), rutile (Schmidt et al., 2004), titanite (Prowatke  
666 and Klemme, 2005), and apatite (Prowatke and Klemme, 2006). Thus, in the context of  
667 melt-lithosphere interactions, elemental partition coefficients for mantle crystalline  
668 phases are expected to shift towards higher values in the presence of silica-rich liquids  
669 that react with peridotite. Of particular, our experimentally determined  $D_s$  for Ti, Zr,  
670 REE, and Sr agree well with the partitioning between diopside and silicic glass in  
671 harzburgite xenoliths (PAT2-68 in Vannucci et al. 1998), indicating these results may  
672 be applicable to lithospheric melting. As such, this new experimental dataset is of  
673 interest for the further modeling of multi-step polybaric mantle melting and  
674 metasomatism, e.g., as has been proposed for the genesis of hump-shaped REE patterns  
675 in mantle cpx via the melting of a depleted spinel lherzolite source and subsequent  
676 equilibration with the melt enriched in  $\text{SiO}_2$ ,  $\text{Al}_2\text{O}_3$ , and alkalis at shallower depth  
677 (Lundstrom, 2000; Huang et al., 2006).

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981 **Table 1.** Major element (wt% oxides) compositions of starting materials, as measured by LA-ICP-MS under single-spot (30  
982  $\mu\text{m}$ ) analytical mode.

Stating material	<i>n.</i>	SiO <sub>2</sub>	Al <sub>2</sub> O <sub>3</sub>	FeO	MgO	CaO	Na <sub>2</sub> O	K <sub>2</sub> O	TiO <sub>2</sub>	Cr <sub>2</sub> O <sub>3</sub>	Total
HISILA	8	68.2(1)	14.26(7)	1.65(3)	2.57(1)	3.49(7)	4.74(1)	3.84(2)	0.155(4)	0.053(3)	98.9
HISILB	8	67.5(1)	14.49(8)	1.65(1)	2.69(2)	3.38(7)	8.96(6)	–	0.115(3)	0.048(1)	98.9
HISILC	8	67.6(1)	14.28(10)	1.69(2)	2.63(3)	3.47(5)	8.9(1)	–	0.204(3)	0.077(1)	98.9
HISILA4D	8	67.9(3)	14.26(13)	1.76(5)	2.64(12)	3.56(17)	4.73(5)	3.85(4)	0.153(3)	0.062(3)	98.9
HISILA8D	7	67.7(3)	14.23(15)	1.73(3)	2.75(18)	3.69(29)	4.7(1)	3.91(9)	0.150(2)	0.065(2)	98.9

983 *Notes:* numbers in parentheses for this and subsequent tables indicate one standard deviation ( $1\sigma$ ) of replicate analyses in terms of last significant  
984 numbers: 68.2(1) should be read as  $68.2 \pm 0.1$ .

985

986 **Table 2.** Major element (wt% oxides) compositions of glasses, as measured by LA-ICP-MS under single-spot (30 μm)  
 987 analytical mode.

Run #	1H04-1	2H04-2	3H04-3	4H04-4	5H05-1	6H05-2	7H05-3	8H07-1	9H07-2	10H08-1	11H08-2	12H09-1	13H09-2	14H09-3
Starting material	HISILA	HISILB	HISILA4D	HISILA8D	*HISILA	*HISILB	*HISILC	HISILB	HISILC	†HISILA	†HISILB	‡HISILA	‡HISILB	‡HISILC
n	12	10	10	13	10	10	11	11	15	13	13	10	12	10
SiO <sub>2</sub> (wt%)	68.07(26)	68.21(42)	68.77(47)	69.11(43)	68.28(24)	67.59(34)	67.78(32)	68.87(21)	68.11(33)	68.55(45)	68.97(54)	68.42(48)	68.84(43)	69.01(53)
Al <sub>2</sub> O <sub>3</sub>	14.51(17)	14.89(27)	14.78(30)	15.27(28)	14.80(13)	14.65(26)	14.10(17)	14.88(22)	14.49(22)	14.93(29)	15.51(35)	14.84(28)	15.10(45)	14.90(32)
FeO	1.66(7)	2.11(13)	1.65(8)	1.54(3)	1.86(8)	1.75(5)	1.60(16)	1.62(4)	1.92(4)	1.94(7)	1.60(7)	1.66(5)	1.65(11)	1.67(11)
MgO	2.45(14)	2.26(34)	2.06(30)	1.38(8)	2.26(21)	2.46(17)	2.56(5)	1.83(10)	2.36(14)	1.91(16)	1.35(18)	2.34(8)	1.99(35)	1.81(22)
CaO	3.18(29)	2.30(25)	2.66(28)	2.40(9)	2.84(16)	3.03(25)	3.12(17)	2.43(19)	2.85(25)	2.55(16)	1.81(12)	2.84(26)	2.06(21)	2.31(31)
Na <sub>2</sub> O	4.79(6)	9.14(17)	4.88(6)	4.98(7)	4.86(4)	9.45(20)	9.41(11)	9.16(10)	8.95(9)	4.91(8)	9.44(17)	4.95(10)	9.22(16)	9.14(12)
K <sub>2</sub> O	4.02(10)		4.08(15)	4.21(8)	3.98(8)					3.95(11)		4.01(13)		
TiO <sub>2</sub>	0.159(4)	0.115(3)	0.162(5)	0.159(4)	0.164(3)	0.112(3)	0.198(5)	0.114(4)	0.198(4)	0.154(3)	0.116(4)	0.162(2)	0.115(2)	0.197(6)
Cr <sub>2</sub> O <sub>3</sub>	0.04(1)				0.024(2)	0.03(1)		0.03(1)	0.05(1)	0.02(1)		0.02(1)	0.02(1)	0.02(1)
MnO	0.047(2)	0.030(3)	0.039(6)	0.031(2)	0.048(3)	0.034(3)	0.059(3)	0.029(2)	0.055(5)	0.043(4)	0.029(3)	0.045(5)	0.031(4)	0.046(2)
SUM	98.93	99.04	99.08	99.08	99.12	99.10	98.82	98.97	98.99	98.95	98.83	99.30	99.04	99.10
ASI	0.81	0.77	0.86	0.89	0.85	0.69	0.66	0.76	0.72	0.88	0.82	0.84	0.80	0.78
Mg# (mole%)	72.4	65.7	69.1	61.5	68.4	71.5	74.0	66.8	68.7	63.7	60.1	71.5	68.2	65.9
Mg <sup>2+</sup> /(M <sup>+</sup> +M <sup>2+</sup> )	15.94	13.29	13.99	9.81	14.99	13.67	14.18	11.16	13.70	13.05	8.54	15.38	12.13	11.09
NBO/T	0.17	0.18	0.14	0.10	0.15	0.21	0.22	0.16	0.20	0.13	0.12	0.15	0.15	0.15

988

989 **Table 3.** Major element (wt% oxides) compositions of cpx, as measured by EPMA.

Run #	1H04-1	2H04-2	3H04-3	4H04-4	5H05-1	6H05-2	7H05-3	8H07-1	9H07-2	10H08-1	11H08-2	12H09-1	13H09-2	14H09-3
Starting material	HISILA	HISILB	HISILA4D	HISILA8D	*HISILA	*HISILB	*HISILC	HISILB	HISILC	†HISILA	†HISILB	‡HISILA	‡HISILB	‡HISILC
n	7	9	4	9	8	8	8	5	6	5	5	6	7	7
SiO <sub>2</sub> (wt%)	52.58(35)	53.86(25)	53.15(14)	53.19(55)	52.84(28)	53.66(25)	53.61(23)	53.78(19)	53.44(24)	52.65(8)	53.67(40)	52.75(22)	53.68(21)	53.22(18)
TiO <sub>2</sub>	0.10(1)	0.07(1)	0.096(4)	0.10(2)	0.12(2)	0.06(1)	0.13(1)	0.07(2)	0.12(2)	0.11(1)	0.07(1)	0.13(2)	0.07(2)	0.13(2)
Al <sub>2</sub> O <sub>3</sub>	2.59(21)	1.91(14)	2.66(24)	2.97(22)	3.09(16)	1.89(12)	1.83(15)	1.97(13)	1.88(10)	3.10(19)	1.70(9)	2.95(12)	2.09(11)	1.93(8)
FeO	2.13(8)	1.89(7)	2.24(15)	2.02(7)	2.52(14)	2.17(11)	2.12(9)	2.22(7)	1.98(3)	2.61(23)	2.21(9)	2.39(6)	2.27(10)	2.10(5)
MnO	0.08(1)	0.04(1)	0.10(01)	0.06(2)	0.09(1)	0.04(2)	0.08(1)	0.05(1)	0.08(2)	0.096(4)	0.05(2)	0.09(2)	0.05(2)	0.07(1)
MgO	18.50(22)	17.34(14)	18.33(37)	17.89(9)	18.35(15)	17.82(16)	17.40(14)	17.66(8)	17.48(11)	18.18(24)	17.60(7)	18.31(22)	17.59(18)	17.36(21)
CaO	21.32(20)	23.30(26)	21.72(61)	22.48(67)	21.16(56)	23.06(27)	23.02(27)	23.07(27)	22.56(26)	21.65(19)	22.83(24)	21.94(15)	22.70(41)	23.05(25)
K <sub>2</sub> O	0.05(2)		0.06(2)	0.04(1)	0.06(2)					0.05(1)		0.05(1)		
Na <sub>2</sub> O	0.49(2)	0.81(5)	0.51(3)	0.50(6)	0.53(6)	0.87(3)	0.88(6)	0.91(9)	0.93(4)	0.52(1)	0.90(5)	0.53(2)	0.95(4)	0.90(5)
NiO	0.44(2)	0.35(2)	0.36(2)	0.21(2)	0.44(2)	0.34(2)	0.53(3)	0.34(2)	0.57(2)	0.44(2)	0.29(2)	0.48(1)	0.30(4)	0.56(3)
Cr <sub>2</sub> O <sub>3</sub>	0.31(2)	0.22(1)	0.53(4)	0.21(4)	0.26(3)	0.12(2)	0.14(3)	0.17(4)	0.40(6)	0.23(3)	0.29(8)	0.28(6)	0.18(5)	0.30(6)
Total	98.60(33)	99.80(31)	99.76(48)	99.47(66)	99.46(49)	100.04(26)	99.71(32)	100.25(27)	99.45(34)	99.64(29)	99.61(34)	99.91(36)	99.89(30)	99.61(30)

990

991 **Table 4.** Cpx–melt partition coefficients.

Run #	1H04-1	2H04-2	3H04-3	4H04-4	5H05-1	6H05-2	7H05-3	8H07-1	9H07-2	10H08-1	11H08-2	12H09-1	13H09-2	14H09-3
Starting material	HISILA	HISILB	HISILA4D	HISILA8D	*HISILA	*HISILB	*HISILC	HISILB	HISILC	†HISILA	†HISILB	‡HISILA	‡HISILB	‡HISILC
Si*	0.77(1)	0.79(1)	0.77(1)	0.77(1)	0.774(5)	0.79(1)	0.79(1)	0.781(4)	0.78(1)	0.77(2)	0.78(1)	0.77(2)	0.78(1)	0.77(1)
Al <sup>IV</sup> *	0.13(1)	0.08(1)	0.13(1)	0.12(1)	0.14(1)	0.11(1)	0.10(1)	0.11(1)	0.10(1)	0.15(1)	0.09(1)	0.15(1)	0.10(1)	0.11(1)
Al <sup>VI</sup> *	0.049(4)	0.045(4)	0.054(5)	0.076(6)	0.072(4)	0.019(1)	0.030(2)	0.027(2)	0.030(2)	0.058(4)	0.022(1)	0.044(2)	0.039(2)	0.016(1)
Fe <sub>total</sub> *	1.28(7)	0.90(6)	1.36(11)	1.31(5)	1.35(10)	1.24(7)	1.32(15)	1.36(5)	1.03(3)	1.35(13)	1.38(8)	1.44(6)	1.37(11)	1.26(9)
Fe <sub>total</sub>	1.22(6)	1.05(15)	1.47(18)	1.47(13)	1.66(13)	1.16(19)	1.16(13)	1.36(14)	0.88(16)	1.19(20)	1.29(23)	1.45(14)	1.30(13)	1.30(27)
Fe <sup>3+</sup> ,MI*	1.28(7)	0.90(6)	1.27(10)	1.13(4)	1.19(9)	1.24(7)	1.32(15)	1.36(5)	1.03(3)	1.35(13)	1.38(8)	1.44(6)	1.37(11)	1.26(9)
Mg <sub>total</sub> *	7.55(45)	7.68(1.16)	8.88(1.29)	12.96(71)	8.10(76)	7.24(50)	6.80(14)	9.64(54)	7.41(45)	9.53(83)	13.01(1.74)	7.82(28)	8.84(1.54)	9.60(1.20)
Mg <sup>MI</sup> *	6.60(39)	7.35(1.11)	7.82(1.13)	11.75(65)	6.99(65)	6.87(48)	6.48(14)	9.16(51)	6.97(42)	8.37(73)	12.32(1.65)	6.95(25)	8.32(1.45)	9.19(1.14)
Ca*	6.70(61)	10.14(1.11)	8.18(89)	9.36(44)	7.44(45)	7.61(62)	7.38(42)	9.48(74)	7.90(71)	8.50(52)	12.64(85)	7.71(70)	11.01(1.13)	10.00(1.34)
Na*	0.102(4)	0.088(6)	0.104(6)	0.101(13)	0.110(12)	0.092(4)	0.093(6)	0.100(10)	0.104(4)	0.106(3)	0.095(5)	0.108(4)	0.103(4)	0.098(5)
K*	0.013(4)		0.015(5)	0.010(4)	0.015(6)					0.0135(4)		0.013(3)		
Ti*	0.64(6)	0.63(11)	0.59(3)	0.64(10)	0.73(11)	0.56(9)	0.64(7)	0.59(16)	0.62(12)	0.71(7)	0.60(11)	0.82(10)	0.63(15)	0.67(9)
Ti	0.78(19)	0.77(13)	0.58(13)	0.72(12)	0.83(15)	0.57(12)	0.69(10)	0.63(34)	0.74(12)	0.81(15)	0.64(09)	0.73(17)	0.62(09)	0.71(14)
Cr*	8.46(1.58)				11.07(1.71)	4.44(1.16)		6.22(2.13)	8.44(1.63)	14.71(5.19)		13.62(4.88)	8.83(3.20)	11.97(4.01)
Cr	10.80(3.30)				13.19(2.24)	5.07(2.09)		10.22(3.66)	9.12(1.97)	19.01(7.17)		14.24(4.58)	11.77(4.65)	11.86(3.34)
Mn*	1.62(24)	1.33(35)	2.57(43)	1.94(71)	1.88(26)	1.25(51)	1.41(22)	1.74(47)	1.52(30)	2.24(21)	1.68(61)	2.05(44)	1.74(79)	1.58(24)
Mn	2.61(42)	2.34(35)	2.73(45)	2.96(61)	3.03(31)	1.99(31)	2.08(19)	2.54(49)	2.05(22)	3.72(47)	3.19(67)	2.84(44)	2.55(42)	2.85(41)
Ni*	13.88(2.40)	23.76(5.59)	28.00(6.82)	31.82(4.93)	20.30(1.63)	15.06(2.12)	12.79(1.58)	20.66(2.47)	15.77(2.10)	27.57(5.09)	34.02(3.80)	21.39(3.90)	24.68(5.97)	23.82(4.16)
Ni	14.41(3.10)	20.44(5.16)	18.23(5.58)	30.66(4.98)	21.15(3.00)	14.45(3.34)	12.56(2.90)	18.46(2.14)	15.35(2.30)	26.35(6.45)	40.21(6.80)	19.78(4.05)	26.49(6.01)	25.78(5.51)
Sc	3.99(45)	4.11(49)		5.17(85)	4.38(43)	3.62(47)	3.66(42)	5.56(79)	4.07(43)	5.29(1.20)	6.35(1.18)	4.90(1.00)	4.80(76)	5.34(45)
V	0.14(3)		0.108(4)			0.09(2)	0.12(4)	0.08(2)		0.10(3)	0.09(3)		0.07(1)	0.06(1)
Co	3.25(47)	3.31(45)	3.02(52)	4.46(84)	3.49(31)	2.76(44)	3.09(22)	3.96(57)	3.07(27)	4.14(70)	4.54(88)	3.61(49)	3.76(75)	3.91(30)
Cu			0.14(8)		0.33(11)	0.21(8)	0.16(1)		0.16(9)	0.37(6)		0.33(2)	0.21(8)	0.10(3)
Zn	1.04(11)		0.70(9)		0.94(9)	0.64(8)	0.60(5)	0.77(9)	0.60(5)	0.83(10)	0.79(12)	0.98(18)	0.85(14)	0.71(22)
Nb			0.03(1)		0.05(1)	0.03(1)	0.03(2)	0.02(1)	0.03(1)	0.04(2)	0.02(1)	0.03(1)	0.016(2)	0.011(3)
Ta			0.06(3)		0.08(2)	0.05(2)	0.04(2)	0.03(1)	0.07(1)	0.08(3)	0.05(2)	0.08(2)	0.04(1)	0.02(1)

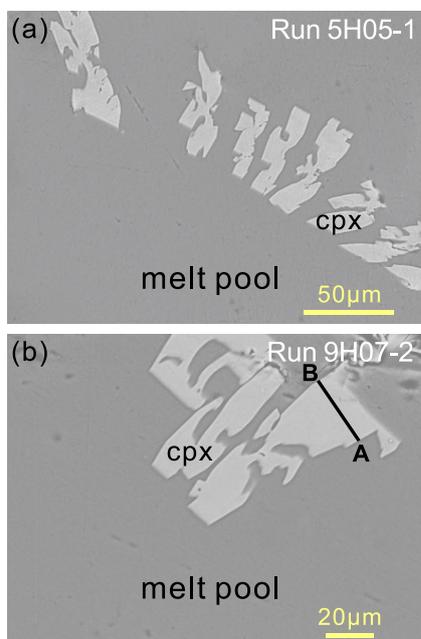
Zr	0.31(4)		0.18(8)		0.30(7)	0.21(4)	0.18(4)	0.21(4)	0.25(2)	0.31(5)	0.20(2)	0.31(5)	0.25(4)	0.28(5)
Hf	0.44(3)		0.33(3)		0.54(7)	0.36(6)	0.34(5)	0.36(11)	0.49(7)	0.53(12)	0.37(4)	0.63(6)	0.51(7)	0.53(10)
Rb	0.031(16)	0.099(26)	0.023(14)	0.122(38)	0.032(11)	0.027(15)	0.033(31)	0.009(11)	0.017(15)	0.034(22)	0.016(5)	0.019(3)	0.007(7)	0.008(6)
Sr	0.16(1)		0.19(3)		0.20(5)	0.27(4)	0.26(3)	0.25(7)	0.27(3)	0.18(2)	0.33(3)	0.18(4)	0.29(4)	0.30(5)
La	0.26(5)		0.36(6)		0.29(3)	0.35(6)	0.40(8)	0.33(6)	0.40(5)	0.30(4)	0.39(5)	0.34(6)	0.39(8)	0.39(4)
Ce	0.38(2)		0.56(13)		0.49(6)	0.50(5)	0.51(5)	0.43(3)	0.50(3)	0.46(4)	0.57(11)	0.57(8)	0.53(11)	0.50(9)
Nd	0.88(7)	1.17(13)	0.94(12)	1.02(10)	1.15(8)	1.07(15)	0.94(10)	1.14(15)	1.12(11)	0.95(16)	1.23(11)	1.29(19)	1.23(18)	1.20(11)
Sm	1.69(13)		1.54(25)	1.65(21)	2.04(16)	1.31(15)	1.36(15)	1.93(27)	1.87(11)	1.94(20)	2.07(35)	2.00(19)	1.95(24)	2.01(21)
Eu	1.77(10)		1.72(31)	1.82(28)	2.15(18)	1.58(16)	1.54(12)	1.91(28)	1.73(17)	1.94(31)	2.22(48)	2.18(20)	2.13(33)	2.25(25)
Gd	1.71(16)	1.63(22)		1.70(14)	1.87(22)	1.49(21)	1.42(19)	2.08(16)	1.60(21)		2.05(16)	2.01(34)	2.04(30)	2.44(20)
Tb	1.69(43)	1.55(18)		1.69(19)	2.08(16)	1.54(19)	1.63(18)	2.04(25)	1.78(12)	1.95(32)	2.13(21)	2.19(14)	2.12(27)	2.56(33)
Dy			1.75(29)	1.75(26)	2.47(22)	1.48(20)	1.43(16)	2.09(28)	1.94(12)	1.96(27)	2.22(21)	2.23(20)	2.19(31)	2.15(23)
Ho	1.90(8)	1.68(19)	1.60(26)	1.76(26)	2.26(17)	1.49(13)	1.52(17)	2.04(15)	1.76(11)	1.90(27)	2.06(17)	2.23(20)	1.90(24)	2.23(27)
Er	1.92(36)	1.68(17)	1.41(14)	1.82(25)	2.40(17)	1.56(16)	1.66(22)	2.12(29)	1.64(7)	2.02(28)	2.01(14)	2.41(9)	1.97(24)	1.99(18)
Yb	1.67(8)	1.51(12)	1.31(17)	1.51(24)	2.15(27)	1.30(20)	1.27(26)	1.65(26)	1.57(17)	1.93(37)	1.94(13)	2.27(15)	1.82(28)	1.68(32)
Lu	1.65(16)	1.47(17)	1.44(16)	1.53(23)	2.08(13)	1.19(21)	1.08(14)	1.41(22)	1.54(16)	1.55(19)	1.54(10)	1.85(16)	1.55(16)	1.65(24)

992 Si\*: Partition coefficients calculated from EPMA analyses of cpx. All other partition coefficients were calculated based on LA-ICP-MS data.

993

## Figures and captions

994 **Fig. 1**

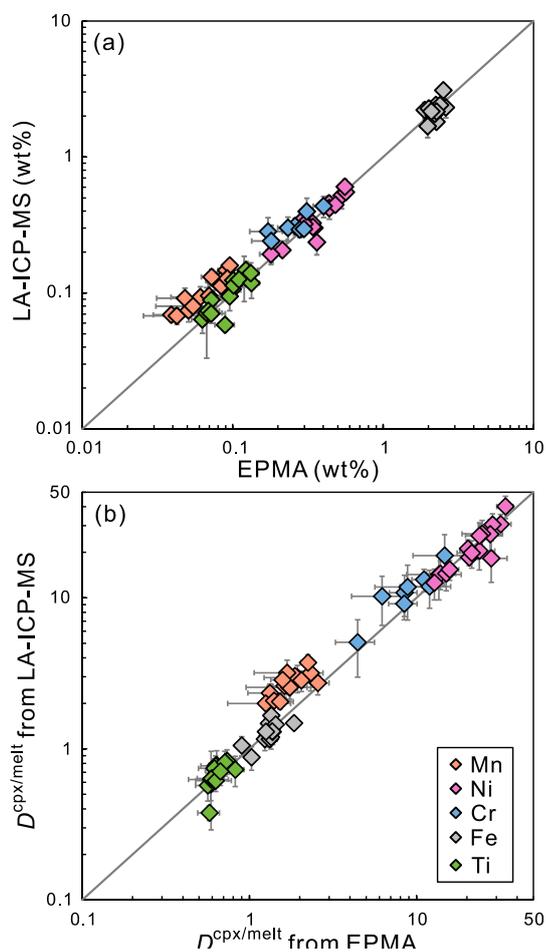


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997 **Fig. 1.** Back-scattered electron images of experimental products in representative runs  
998 (a) 5H05-1 and (b) 9H07-2 showing euhedral-subhedral cpx coexisting with melt. The  
999 experimental charge has large melt pools free of microcrystals allowing micro-beam  
1000 analyses. The line A–B in panel b represents the analytical traverse shown in Figure 4.  
1001 Note that the dark area near the crystal is due to a polishing problem.

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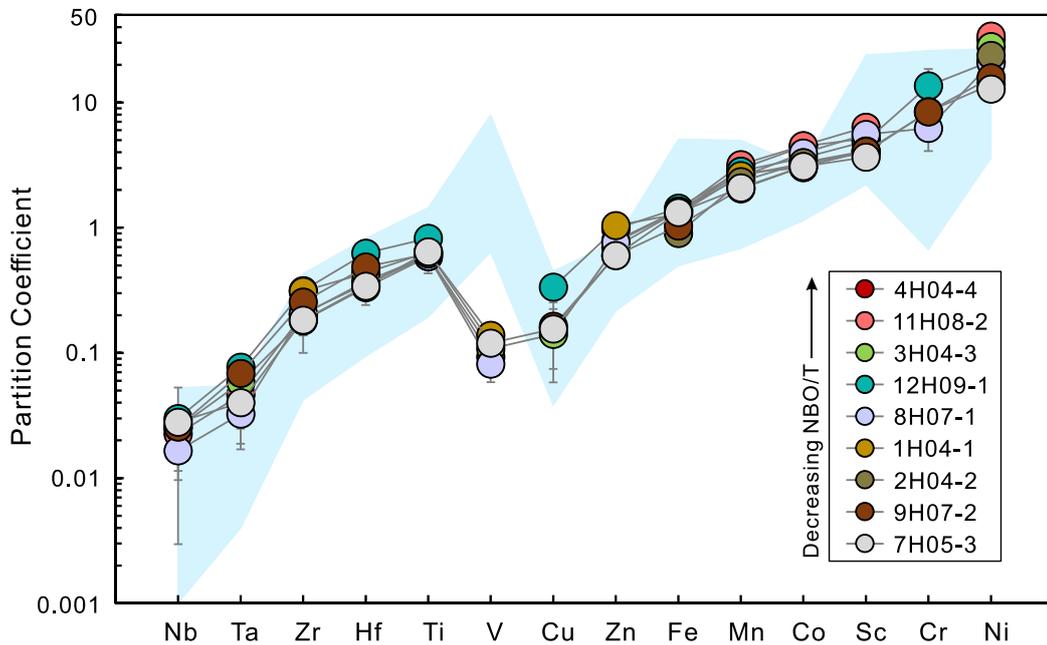
1005 **Fig. 2**



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1008 **Fig. 2.** Comparison between EPMA and LA-ICP-MS analyses of MnO, NiO, Cr<sub>2</sub>O<sub>3</sub>,  
1009 FeO<sub>t</sub>, and TiO<sub>2</sub> concentrations for (a) cpx and (b) corresponding partition coefficients.  
1010 Error bars on this and subsequent diagrams represent one standard deviation ( $1\sigma$ ). The  
1011 plots demonstrate very good agreement between the two analytical techniques. The  
1012 solid grey line is the 1:1 ratio line.  
1013

1014 **Fig. 3**

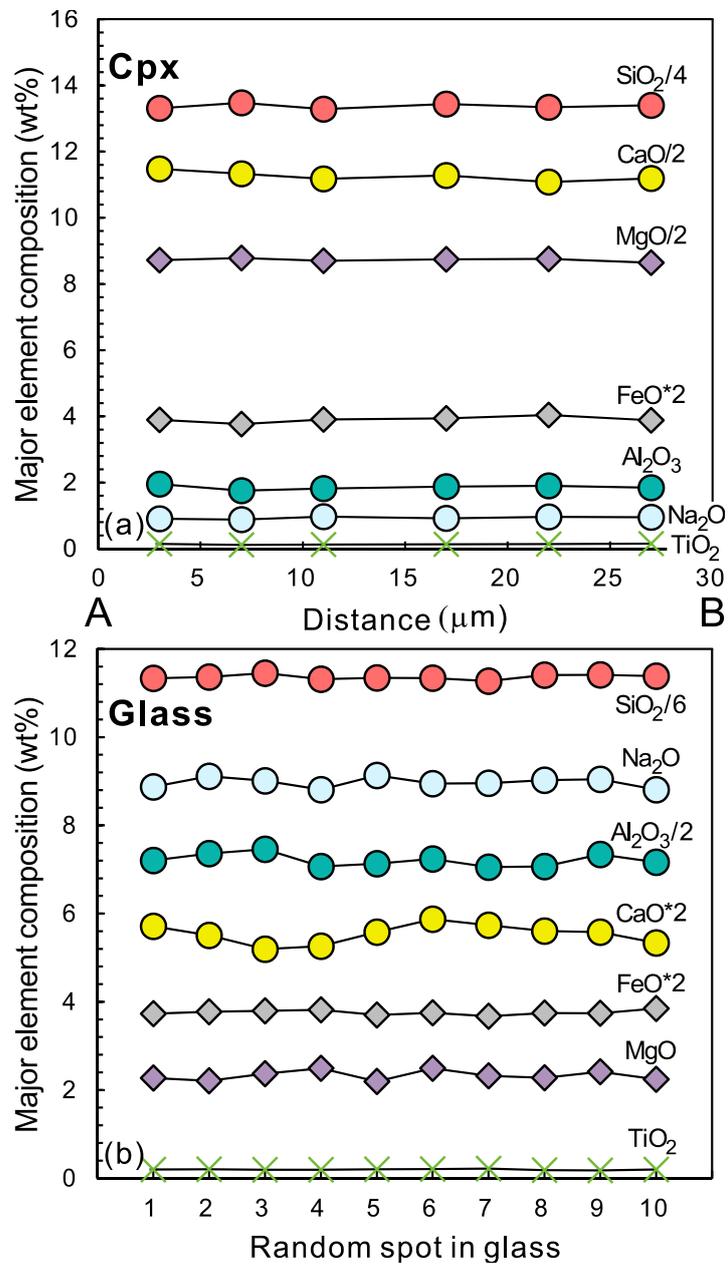


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1017 **Fig. 3.** Cpx–melt FRTE and HFSE partition coefficient patterns obtained from nine  
1018 representative runs with a range of melt NBO/T (0.10–0.22) in this study. The  
1019 partitioning data from the literature (Luhr and Carmichael, 1980; Dunn, 1987; Yurimoto  
1020 and Sueno, 1987; Forsythe et al., 1994; Skulski et al., 1994; Johnson, 1998; Toplis and  
1021 Corgne, 2002; Norman et al., 2005; Adam and Green, 2006; Huang et al., 2006; Fedele  
1022 et al., 2009; Liu et al., 2015), filtered based on diopsidic cpx with  $^{iv}Al < 0.20$ , are shown  
1023 as blue shaded field for comparison.

1024

1025 **Fig. 4**

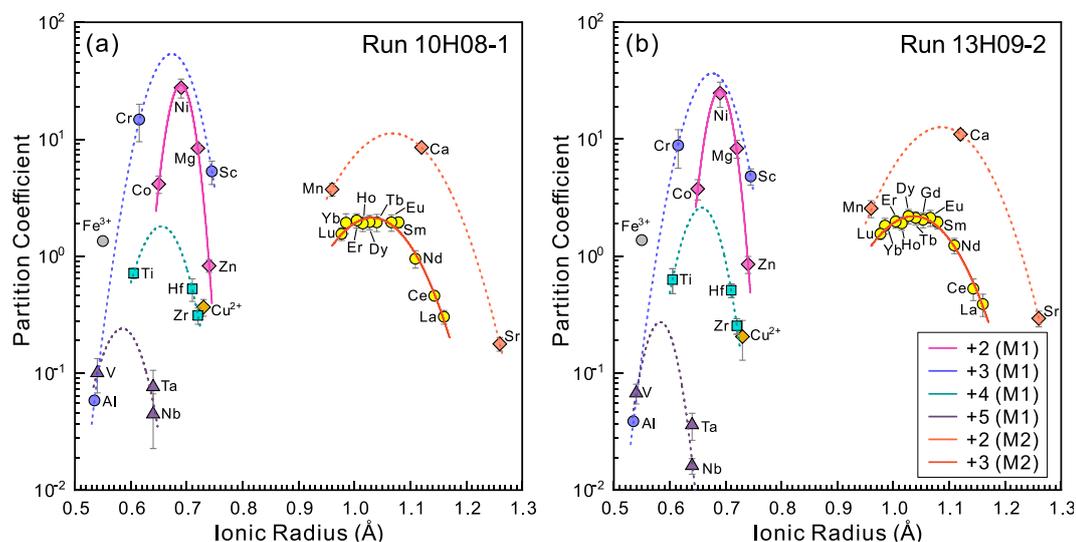


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1028 **Fig. 4.** Profiles of major-element compositional variations in clinopyroxene  
 1029 and glass. (a) Variation in composition of clinopyroxene across the  
 1030 analytical traverse A–B in Figure 1, showing minor zoning in major  
 1031 elements. (b) Variation in glass composition among 10 random spots. No  
 1032 variations outside of analytical errors are detectable.

1033 **Fig. 5**

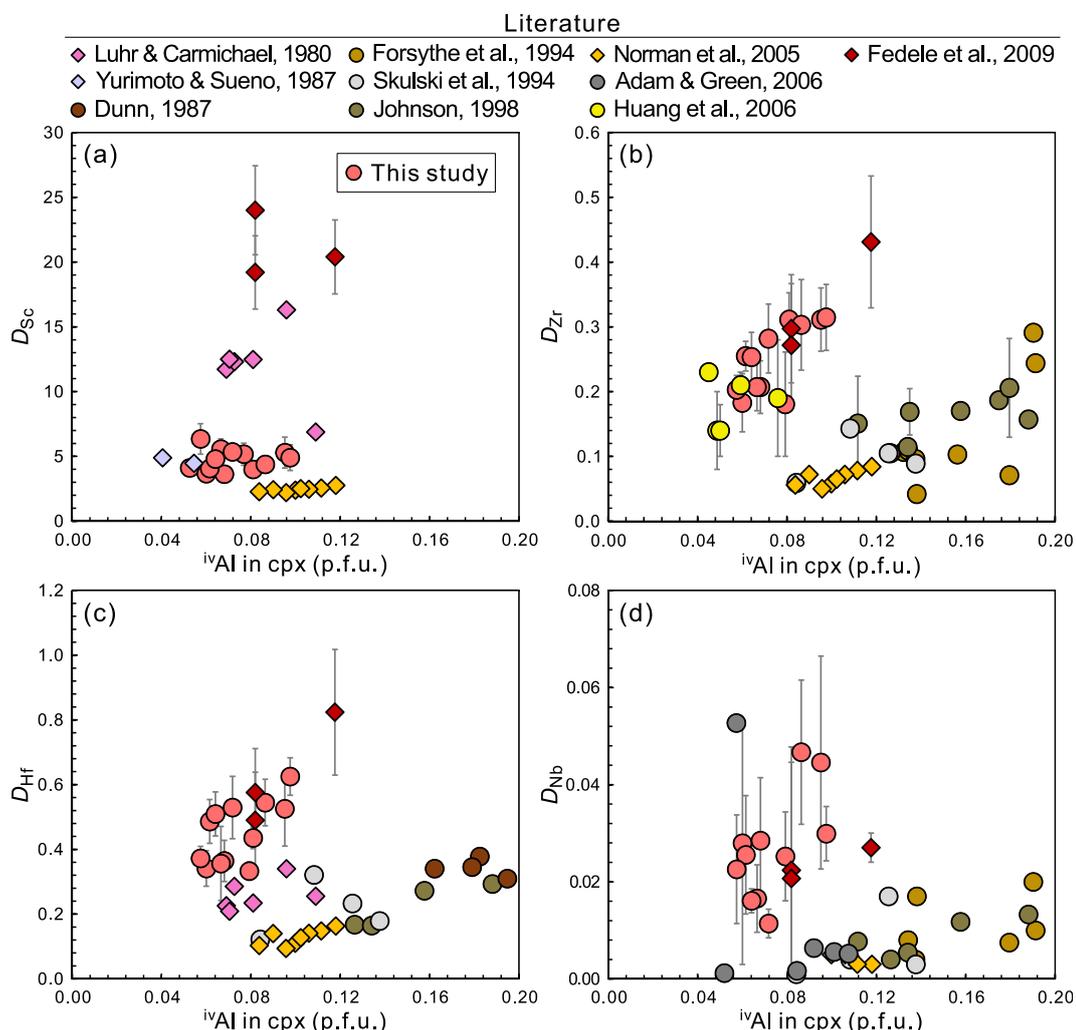


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1036 **Fig. 5.** Onuma diagrams showing partition coefficients for di-, tri-, tetra-, and  
 1037 pentavalent cations between cpx and melt as a function of ionic radii (in Å; Shannon,  
 1038 1976). Ionic radii in VI- and VIII-fold coordination are taken for cations on the M1 and  
 1039 M2 site, respectively. The partitioning data are from two runs (a)10H08-1 and (b)  
 1040 13H09-2 reported in this study. Color-coded curves represent least-squares fits of the  
 1041 data to the lattice strain model of Blundy and Wood (1994) as expressed in Eq. (1).  
 1042 Lattice strain parameter for tri-, tetra-, and pentavalent cations on the M1 site and  
 1043 divalent cations on the M2 site are unconstrained, thus parabolas are shown as dashed  
 1044 curves. For the fitting of trivalent cations on M1 site,  $D_{Al}$  on the M1 site was calculated  
 1045 as the difference between total Al and the  $^{iv}Al$  partitioning into the tetrahedral site. Note  
 1046 that  $Cu^{2+}$  and  $Fe^{3+}$  were excluded from the fitting since they fall outside the parabolas  
 1047 defined by divalent and trivalent cations, respectively.

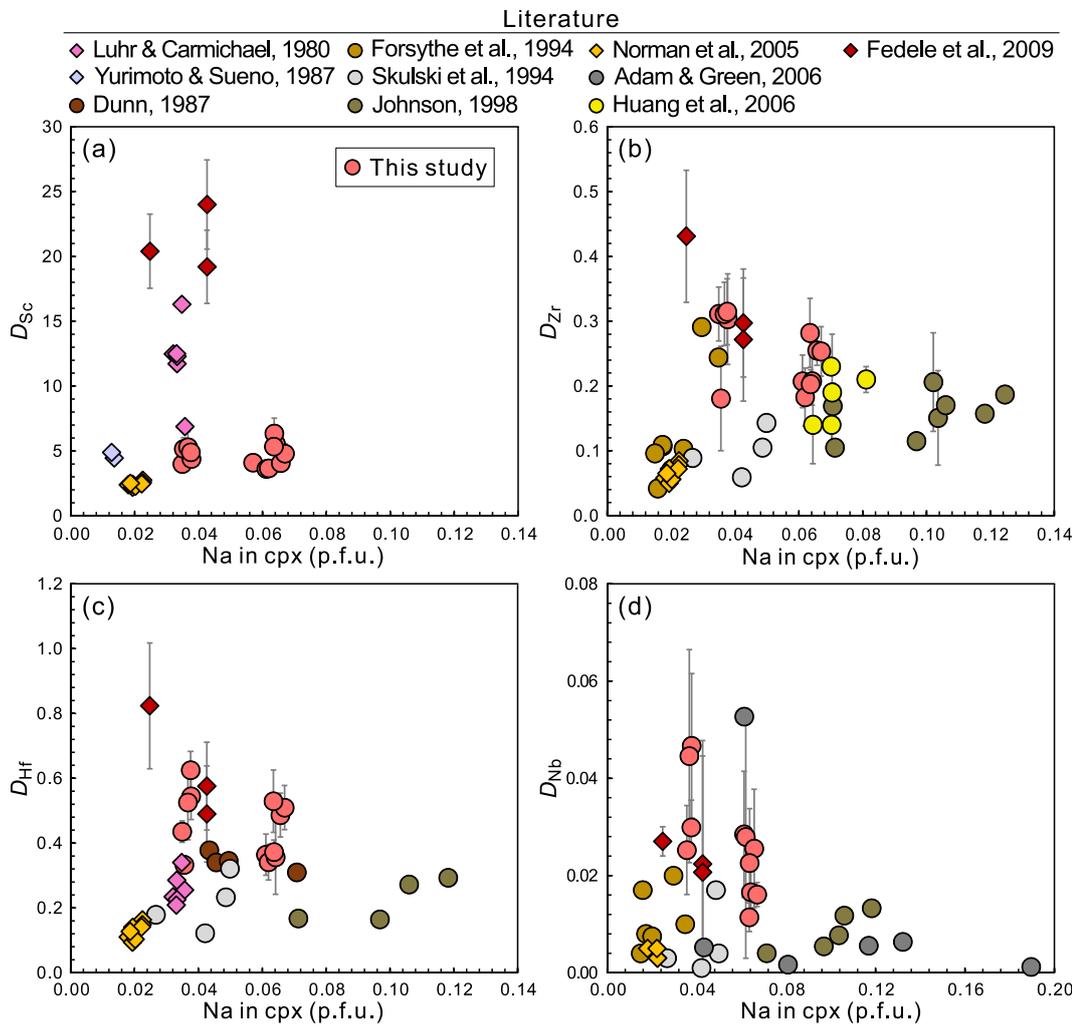
1048 **Fig. 6**



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**Fig. 6.** Partition coefficients of (a) Sc, (b) Zr, (c) Hf, and (d) Nb between cpx and melt as a function of  $ivAl$  content (p.f.u. = per formula unit, calculated for six oxygens) of the cpx. Other partition coefficients for diopsidic cpx with  $ivAl < 0.2$  from the literature are shown for comparison.

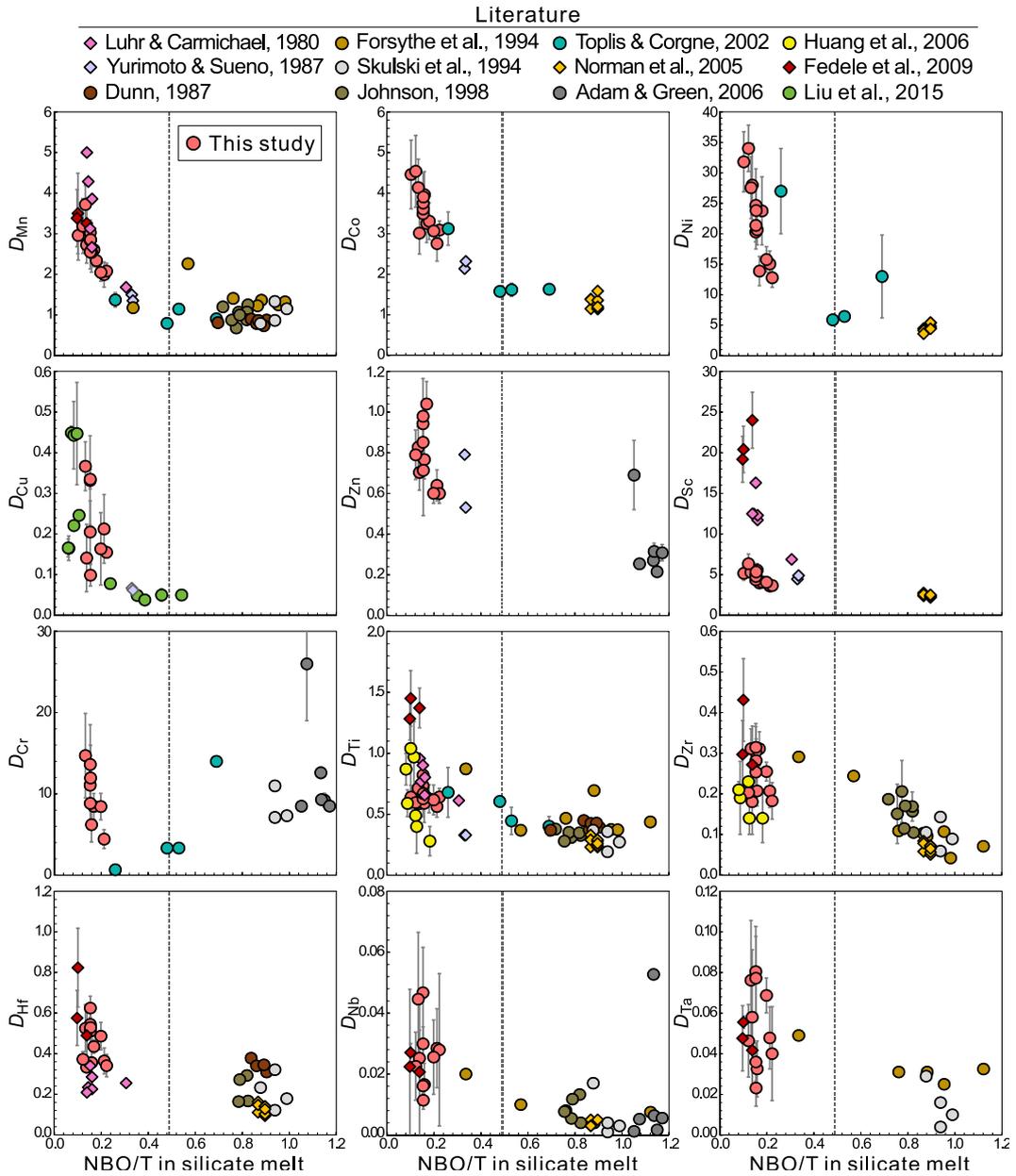
1055 **Fig. 7**



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1057 **Fig. 7.** Partition coefficients of (a) Sc, (b) Zr, (c) Hf, and (d) Nb between cpx and melt  
 1058 as a function of Na content (p.f.u.) of the cpx. Other partition coefficients for diopsidic  
 1059 cpx with  $^{iv}Al < 0.2$  from the literature are shown for comparison.

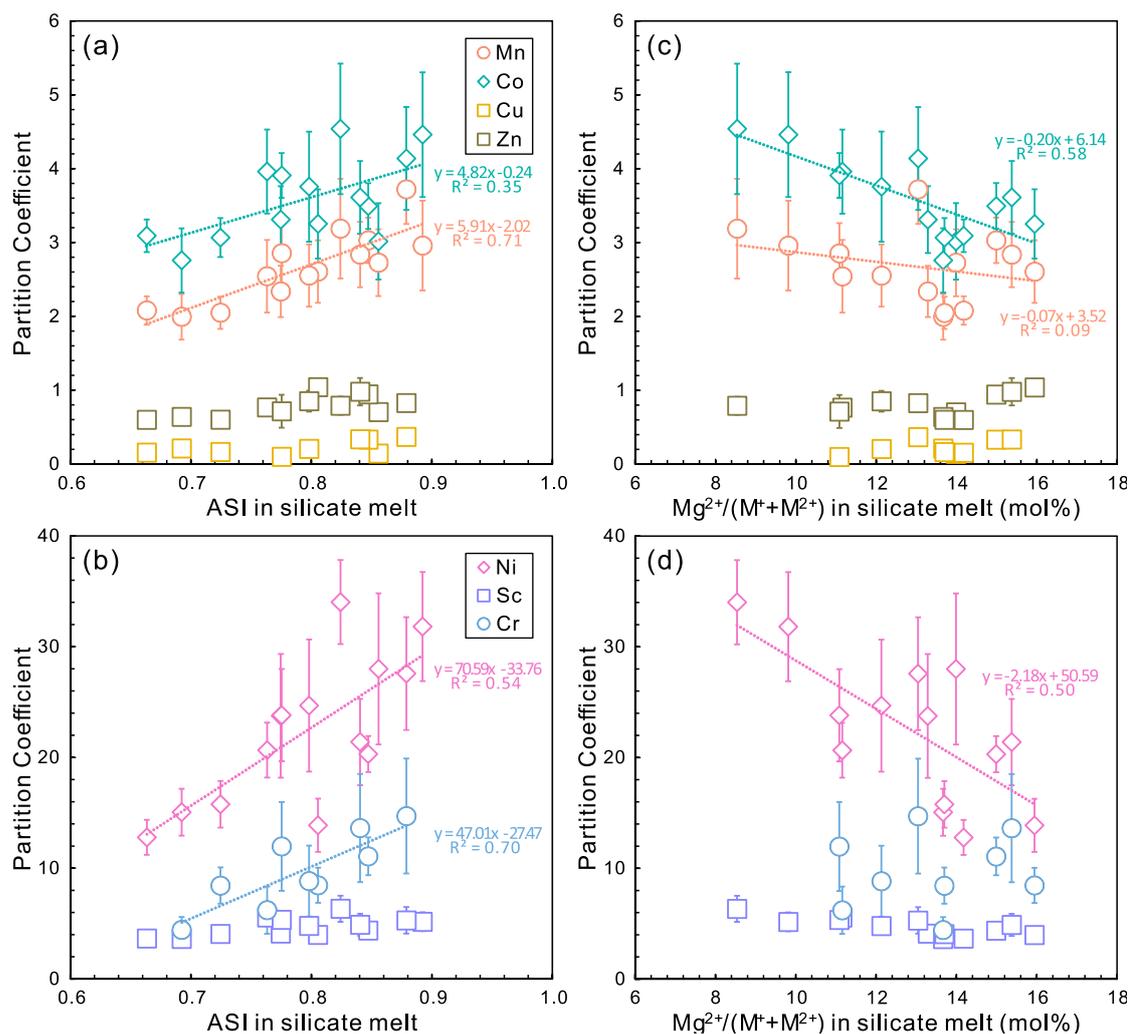
1060 **Fig. 8**



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1063 **Fig. 8.** Partition coefficients of Mn, Co, Ni, Cu, Zn, Sc, Cr, Ti, Zr, Hf, Nb, and Ta  
1064 between cpx and melt as a function of NBO/T of the melt. In a completely polymerized  
1065 melt NBO/T = 0, and in a completely depolymerized melt NBO/T = 4. The dashed gray  
1066 vertical line corresponds to NBO/T = 0.49; at lower NBO/T values, melt structure  
1067 significantly influences trace element partitioning (Gaetani 2004). Other partition  
1068 coefficients for diopsidic cpx with <sup>iv</sup>Al < 0.2 from the literature are shown for  
1069 comparison.

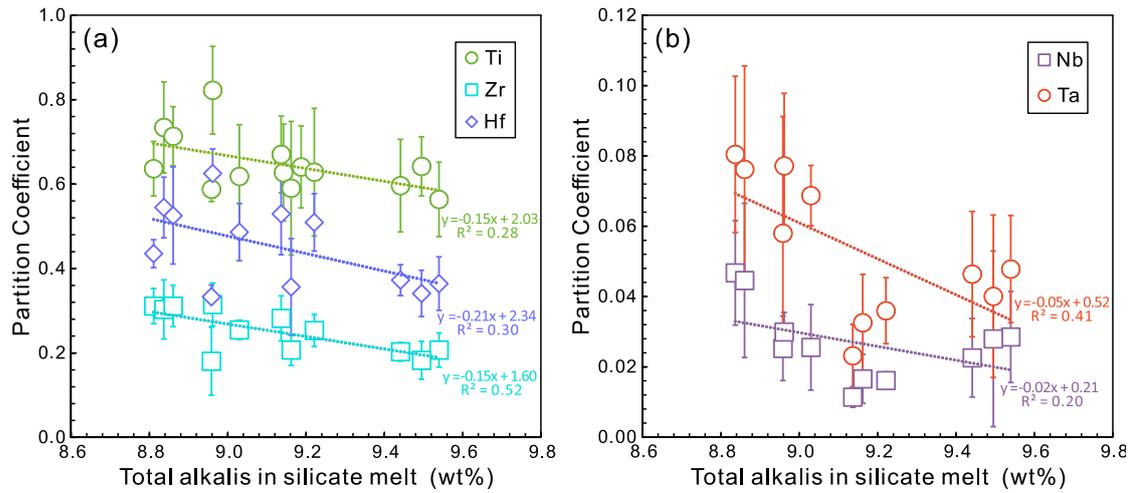
1070 **Fig. 9**



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**Fig. 9.** Partition coefficients of Mn, Co, Cu, Zn, Ni, Sc, and Cr between cpx and melt as a function of (a-b) ASI and (c-d)  $Mg^{2+}/(M^+ + M^{2+})$  (mol%) of the melt.

1075 **Fig. 10**



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1077 **Fig. 10.** Partition coefficients of (a) Ti, Zr, Hf, and (b) Nb and Ta between cpx and melt  
1078 as a function of total alkalis ( $\text{Na}_2\text{O} + \text{K}_2\text{O}$ , wt%) of the melt.