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2 **Cation ordering, twinning and pseudo-symmetry in silicate garnet: the study**  
3 **of a birefringent garnet with orthorhombic structure**

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## ABSTRACT

18 The crystal structure of a birefringent garnet ( $\sim\text{Adr}_{53}\text{Grs}_{47}$ ) that occurs as a late-stage rim on  
19 andradite from Stanley Butte, Graham County, Arizona is analyzed and refined using single-  
20 crystal XRD. The structure has an orthorhombic  $I\frac{2}{a}1\frac{2}{a}$  (unconventional setting for  $Fddd$ ) space  
21 group symmetry, with unit cell parameters of  $a = b = 11.966(3) \text{ \AA}$ ,  $c = 11.964(3) \text{ \AA}$ ,  $\alpha = \beta = 90^\circ$ ,  
22  $\gamma = 90.29(2)^\circ$ ,  $V = 1713.0(7) \text{ \AA}^3$ ,  $Z = 8$ . The orthorhombic garnet displays very high  
23 birefringence ( $\delta \sim 0.021$ ) produced by the strong Fe-Al ordering in the octahedral sites, with Fe  
24 occupancies of 0.804 and 0.221 in  $Y_1$  and  $Y_2$  sites, respectively. Diffraction peaks (such as 101  
25 and 103) violating the  $Ia\bar{3}d$  symmetry of cubic garnet are obvious even in powder XRD pattern.  
26 The homogenization temperatures of the fluid inclusions suggest that the low crystallization  
27 temperature is responsible for the ordered orthorhombic structure. The strong ordering state of  
28 the structure and the sharp boundaries in the chemical zoning in the crystal (between  $\sim\text{Adr}_{53}\text{Grs}_{47}$   
29 and  $\sim\text{Adr}_{100}$ ) indicate the orthorhombic intermediate grandite garnet is a thermodynamically  
30 stable phase at low temperature, separated by wide miscibility gaps from the pure end members  
31 (grossular and andradite) with cubic structures. Most of the previously reported triclinic garnet  
32 structures are likely artifacts produced by pseudo-merohedral twinning of less ordered  
33 orthorhombic structure, as indicated by the characteristic pairing pattern of different Y-sites with  
34 the same occupancies.

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36 Keywords: orthorhombic garnet, Fe-Al ordering, non-cubic garnet, birefringent garnet, pseudo-  
37 merohedral twinning, fluid inclusion

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## INTRODUCTION

40 Silicate garnets are common rock-forming orthosilicate minerals that occur in earth's crust and  
41 mantle, which have a general formula of  $X_3Y_2[SiO_4]_3$ , with X representing divalent cations in  
42 triangular dodecahedron (distorted cube) (i.e.,  $Ca^{2+}$ ,  $Fe^{2+}$ , and  $Mg^{2+}$ ), and Y representing trivalent  
43 cations in regular octahedral site (i.e.,  $Cr^{3+}$ ,  $Al^{3+}$ , and  $Fe^{3+}$ ) (Geiger 2008, 2016; Grew et al.  
44 2013). The tetrahedrally coordinated Si site is known as Z site, which can be occupied by  
45 elements such as Al, Fe or V in non-silicate garnets (Grew et al. 2013). A significant amount of  
46 structural  $OH^-$  (water) may be incorporated in these nominally anhydrous minerals, mostly  
47 through the hydrogarnet substitution (Lager et al. 1989; Grew et al. 2013; Geiger and Rossman  
48 2018, 2020). Although silicate garnets generally have cubic ( $Ia\bar{3}d$ ) symmetry, birefringence is  
49 often observed in the grossular (Grs:  $Ca_3Al_2Si_3O_{12}$ ) - andradite (Adr:  $Ca_3Fe^{3+}_2Si_3O_{12}$ ) solid  
50 solution (grandite garnet). [The IMA-CNMNC approved mineral symbols (abbreviations) are  
51 used to describe the garnet compositions in this paper (Warr 2021).] Similar effects have been  
52 recently reported in the less studied uvarovite (Uv:  $Ca_3Cr_2Si_3O_{12}$ ) – grossular solid solution  
53 (Andrut and Wildner 2001, 2002; Wildner and Andrut 2001; Andrut et al. 2002). Fine scale  
54 oscillatory zoning, supposedly related to the anisotropy, is also commonly observed in grandite  
55 garnets (Lessing and Standish 1973; Murad 1976; Jamtveit 1991; Jamtveit et al. 1993, 1995;  
56 Pollok et al. 2001; Antao 2013; Antao et al. 2015). Some of the sub-micron periodic zoning near  
57 the surface can create spectacular iridescent colors similar to those observed in labradorite,  
58 making them gemstones (Ingerson and Barksdale 1943; Akizuki et al. 1984; Hainschwang and  
59 Notari 2006; Nakamura et al. 2017).

60 Various reasons have been proposed to explain the optical anomaly of the calcium silicate  
61 garnets, including plastic deformation (Allen and Buseck 1988), magneto-optical effects from  
62 rare-earth elements substituting for Ca (Blanc and Maisonneuve 1973), non-cubic orientation of  
63 the  $OH^-$  groups (Rossman and Aines 1986), or residual strain from chemical zoning or twin  
64 boundaries (Chase and Lefever 1960; Lessing and Standish 1973; Foord and Mills 1978; Antao  
65 2013a,b, 2021a,b; Antao and Klincker 2013; Antao et al. 2015). Among all the hypothesized  
66 reasons, symmetry reduction due to cation ordering remains the most widely proposed (Takéuchi  
67 and Haga 1976; Takéuchi et al. 1982; Gali 1983, 1984; Akizuki 1984, 1989, 1989; Allen and  
68 Buseck 1988; Hatch and Griffen 1989; Kingma and Downs 1989; Griffen et al. 1992;  
69 Shtukenberg et al. 2001, 2002, 2005; Frank-Kamenetskaya et al. 2007; Kobayashi et al. 2013;

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70 Badar et al. 2016; Nakamura et al. 2016). However, the real symmetry of the birefringent  
71 calcium silicate garnets remains ambiguous with many different structures refined. The graph of  
72 all centrosymmetric subgroups of the  $Ia\bar{3}d$  space group that preserves the  $I$ -centered translational  
73 symmetry, along with their relations to each other is shown in Figure 1. Most birefringent garnet  
74 structures were refined with triclinic  $I\bar{1}$  symmetry (Takéuchi et al. 1982; Allen and Buseck 1988;  
75 Kingma and Downs 1989; Wildner and Andrut 2001; Kobayashi et al. 2013; Nakamura et al.  
76 2016). Tetragonal garnet based on X-sites ordering in the almandine-grossular solid solution  
77 system has also been reported (Griffen et al. 1992; Cesare et al. 2019). Recently, a trigonal  
78 hydrous garnet species with ordered Fe-Al distribution in Y-sites and  $Si/(OH)_4$  ordering in  
79 tetrahedral (Z) sites was discovered (Krivovichev et al. 2021). Orthorhombic garnet was first  
80 reported in a grandite garnet ( $Adr_{33}Grs_{67}$ ) by Takéuchi and Haga (1976), however, many later  
81 studies of metrically orthorhombic garnet structures concluded they are pseudo-orthorhombic  
82 with triclinic ( $I\bar{1}$ ) symmetry (Takéuchi et al. 1982; Shtukenberg et al. 2005; Frank-  
83 Kamenetskaya et al. 2007).

84 The problem is further complicated as some of the symmetry violating reflections are suspected  
85 to be produced by multiple diffraction (Rossmanith and Armbruster 1995). Twinning in  
86 anisotropic garnet has been discussed (Takéuchi et al. 1982; Hatch and Griffen 1989), yet never  
87 taken into account during structure refinements. Most of the published structure models did not  
88 include any raw diffraction data, making it difficult to judge the validity of the models. Structural  
89 analyses from powder diffraction may bypass the potential complexity introduced by twinning,  
90 but significant peak overlaps from minute lattice distortion and anisotropic strains are very  
91 difficult to resolve, which often lead to ambiguous symmetry determination (Antao 2013a;  
92 Tančić et al. 2020). The poorly understood structural model makes it hard to determine whether  
93 the cation ordering is kinetically induced during crystal growth (Shtukenberg et al. 2001, 2005;  
94 Frank-Kamenetskaya et al. 2007), or thermodynamically driven during low-temperature phase  
95 transition (Engi and Wersin, 1987; Hatch and Griffen 1989; Jamtveit 1991; Becker and Pollok,  
96 2002), with the existing data being used to support arguments in both directions.

97 In this paper, we report the structure of an orthorhombic grandite garnet single crystal  
98 ( $I\frac{2}{a}1\frac{2}{d}$  space group, an unconventional setting for  $Fddd$ ) showing very high birefringence ( $\delta \sim$   
99 0.021). The grandite crystal has much stronger Fe/Al ordering in the octahedral sites than any of

100 the previously published structures, with significant lattice distortion from cubic symmetry. The  
101 fluid inclusions in the samples indicate low crystallization temperatures, providing strong  
102 evidence for a thermodynamically stable phase for grandite with composition close to  $\text{Adr}_{50}\text{Grs}_{50}$ .  
103 The effect of twinning on the intensity distribution in the diffraction data is also discussed in  
104 detail, providing a potential explanation for previously reported structures with different space  
105 group symmetry.

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## 107 **SAMPLES AND EXPERIMENTAL METHODS**

108 The specimen (UW Mineral Collection 7579) was collected from Stanley Butte, San Carlos  
109 Indian Reservation, Graham County, Arizona. The site and similar minerals were described  
110 before (Ross 1925; Anthony et al. 1995). It occurs as a gangue mineral in a skarn deposit. The  
111 euhedral crystals occur on the wall of a hydrothermal vein (Figures S1, S2, S3). The  
112 orthorhombic garnet rim that grew around early-stage andradite, formed at a very late stage  
113 (Figure 2). Associated minerals include epidote and quartz. The olive-green euhedral  
114 dodecahedral crystals range from 2 mm to 10 mm in diameter (Figure S1, 2). The orthorhombic  
115 garnet with high interference colors occurs as an outer rim (~2 mm to 4 mm thick) that coats the  
116 core andradite (Figure 2). It is colourless in thin section. Refractive indices are calculated based  
117 on average refractive index (~1.810) for the garnet composition (Deer et al. 1982, p. 486). The  
118 measured birefringence is ~ 0.021, and its  $2V(+)$  angle is  $75(5)^\circ$  (see Figure S5 for its  
119 interference figure).

120 The chemical composition of the garnet sample was analyzed with a CAMECA SXFive field  
121 emission electron probe at 15 kV and 10 nA beam current with a 5  $\mu\text{m}$  beam size, using  
122 microcline, jadeite, augite, forsterite, tephroite, fluorite, and quartz as standards.  $\text{H}_2\text{O}$  is  
123 calculated by analysing the O by EPMA and assigning the excess O to  $\text{H}_2\text{O}$ . The strongly  
124 birefringent rim shows an average composition of  $(\text{Ca}_{2.91}\text{Fe}^{2+}_{0.06}\text{Mn}^{2+}_{0.02})(\text{Fe}^{3+}_{1.06}\text{Al}_{0.94})$   
125  $(\text{Si}_{2.93}\square_{0.07})(\text{O}_{11.71}\text{F}_{0.14}\text{OH}_{0.15})$ , or  $\sim\text{Adr}_{53}\text{Grs}_{47}$ . The isotropic core and zones, on the other hand,  
126 show compositions close to the andradite end member. The results of the chemical analyses are  
127 listed in Table 1.

128 Double polished sections were used for the fluid inclusion analyses. Measurement of  
129 homogenization and freezing temperature were conducted at the University of Wisconsin-  
130 Madison using a Linkam LMS600 heating-freezing stage mounted on an Olympus BX50  
131 microscope with a 100x objective lens. The fluid-inclusion populations in our garnet sample are  
132 predominantly monophasic liquid inclusions with small numbers of two-phase inclusions, as  
133 shown in Figure 3. Our measured homogenization temperatures ( $T_h$ ) of primary liquid-rich  
134 inclusions range from 157 °C to 170 °C (Figure 3). Freezing point depressions ( $T_m$ ) are about -  
135 11 °C, corresponding to salinities of ~14 equiv. wt.% NaCl (Vanko et al. 1988; Bodnar 1994;  
136 Hurai et al. 2015). Previous research on skarn systems similar to the source of these  
137 garnets estimates formation pressures ranging from ~300 to ~1000 bars (Theodore et al. 1986;  
138 Boni et al. 1990; Myers 1994). Trapping temperatures for the orthorhombic garnet inclusions are  
139 estimated to be ~ 180 °C to ~220 °C by using the range of trapping pressures with the isochores  
140 appropriate for the measured  $T_h$  and salinity (Bodnar 1994).

141 The areas with the highest uniform interference color (see 2 circled areas inside circles in Figure  
142 2c) were selected for single-crystal XRD work and were cut from petrographic thin sections.  
143 Twinned crystals were eliminated through careful screening using single-crystal XRD method. X-  
144 ray diffraction data were collected on a Bruker Quazar APEXII single-crystal diffractometer  
145 with a MoK $\alpha$  I $\mu$ S source and APEX2 detector. The instrument was operated at a voltage of  
146 50 kV and current of 0.6 mA. Unit-cell parameters were calculated and refined using *APEX3*  
147 software. The refinement of the structures was performed with *JANA2006* software (Petříček et  
148 al. 2014) on F<sup>2</sup>. The 3D crystal structure was visualized by *VESTA* (Momma and Izumi 2011).  
149 The experimental and crystallographic information are listed in Table 2. Powder XRD data were  
150 collected with a 2-D image-plate detector and a 0.1 mm collimator using a Rigaku Rapid II  
151 instrument (Mo-K $\alpha$  radiation) in the Department of Geoscience, University of Wisconsin-  
152 Madison. Two-dimensional diffraction patterns were converted to conventional  $2\theta$  vs. intensity  
153 plots using the Rigaku 2DP software. Transmission electron microscope (TEM) samples were  
154 prepared by depositing a suspension of crushed grains (picked from the thin section next to the  
155 crystal used for single-crystal XRD) on a lacy carbon-coated Cu grid. The bright-field and high-  
156 resolution TEM images with the selected-area electron diffraction (SAED) patterns were  
157 obtained using a Philips CM200-UT transmission electron microscope operated at 200 kV in the

158 Materials Science Center, University of Wisconsin-Madison. The chemical composition was  
159 obtained using Energy-dispersive X-ray spectra (EDS) with a Li-drifted Si detector.

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## RESULTS

### 162 Single-crystal XRD

163 Based on the body-centered lattice with no symmetry constraints, the unit-cell parameters are  
164 refined to be  $a = 11.967(3) \text{ \AA}$ ,  $b = 11.965(3) \text{ \AA}$ ,  $c = 11.964(3) \text{ \AA}$ ,  $\alpha = 90.00(1)^\circ$ ,  $\beta = 90.01(1)^\circ$ ,  $\gamma$   
165  $= 90.29(2)^\circ$ . The  $\gamma$  angle deviates significantly from  $90.0^\circ$  (much more than the uncertainty in the  
166 measurement), whereas  $\alpha$  and  $\beta$  do not, which means the tetragonal and trigonal space groups  
167 can be eliminated for the symmetry of this structure. Space group  $Ibca$  can also be ruled out  
168 since it also requires right angles between axes in the body-centered setting. This means the only  
169 space groups possible based on the unit-cell parameters are  $I \frac{2}{a} 1 \frac{2}{d}$  (unconventional setting for  
170  $Fddd$ ) and its subgroups (Figure 1). The reconstructed sections of the diffraction pattern along  
171  $[001]$ ,  $[010]$  and  $[1\bar{1}0]$  axes are shown in Figure 4. No reflections violating the body-centered  
172 lattice was detected. The systematic extinction of the  $a$ -glide plane is preserved in the  $hk0$  section  
173 (Figure 4a), but not in the  $h0l$  section (Figure 4c) as the symmetry operation in this direction is  
174 not allowed by the  $\gamma$  angle. The  $0kl$  section is identical to the  $h0l$  section (Figure S6a). Only  
175 reflections with  $h+k+l = 4n$  are present in the  $hhl$  section (Figure 4d), indicating the  $d$ -glide plane  
176 along the diagonal  $(x, x, z)$  is also preserved. The  $h\bar{h}l$  section shows the same systematic  
177 extinction following the  $d$ -glide symmetry, whereas the  $hkh$  section shows no such extinction  
178 (Figure S6b, c). These systematic extinctions in the diffraction pattern clearly show that the  
179 structure has a space group symmetry of  $I \frac{2}{a} 1 \frac{2}{d}$  ( $Fddd$ ). The diffraction pattern is overall very  
180 clean, and no splitting of reflections was observed even at high Bragg angles. No merohedral  
181 twinning is possible for this centrosymmetric orthorhombic symmetry, and the deviation from  
182 cubic symmetry ( $\gamma$  angle) is large enough for any pseudo-merohedral twinning to be easily  
183 detected during the crystal screening. Therefore, it is clear that the diffraction pattern is collected  
184 from a well-ordered single crystal of orthorhombic garnet.

185 The reflection intensity statistics further support an orthorhombic  $I \frac{2}{a} 1 \frac{2}{d}$  symmetry. The  $R_{\text{int}}$   
186 values for different space groups on the  $I \frac{2}{a} 1 \frac{2}{d}$  branch in Figure 1 are listed in Table 3. The  $R_{\text{int}}$   
187 value dramatically drop tetragonal  $I \frac{4_1}{a} 1 \frac{2}{d}$  to orthorhombic  $I \frac{2}{a} 1 \frac{2}{d}$  symmetry. However, the  
188 improvement of the  $R_{\text{int}}$  value for space groups lower than  $I \frac{2}{a} 1 \frac{2}{d}$  is marginal. The 12 “observed”  
189 reflections violating the systematic extinction condition of the  $I \frac{2}{a} 1 \frac{2}{d}$  space group (Table 3) are  
190 all at low diffraction angles with very weak intensities ( $I < 5\sigma$ ), and none of them are visible in  
191 the raw diffraction data (Figure 4). Note that this is the first time a garnet structure shows  
192 unambiguous orthorhombic symmetry with clear systematic extinctions. Most previous reports  
193 of orthorhombic garnet structures used a monoclinic setting with **b**-axis as the unique axis,  
194 presumably to save the trouble from lattice transformation when going to the monoclinic  
195 subgroup. However, there are 2 possible monoclinic subgroups of the orthorhombic  $I \frac{2}{a} 1 \frac{2}{d}$  space  
196 group (Figure 1), with no particular reason for **b**-axis being unique. Therefore, **c**-axis should be  
197 the unique axis by default, considering two of the axes (**a** and **b**) are symmetrically equivalent in  
198 the body-centered cell setting. Making **c**-axis the unique axis with  $a = b$  and  $\gamma > 90^\circ$  is also  
199 consistent with the subgroup relation ( $I \frac{4_1}{a} \bar{3} \frac{2}{d} \rightarrow I \frac{4_1}{a} 1 \frac{2}{d} \rightarrow I \frac{2}{a} 1 \frac{2}{d}$ ) as shown in Figure 1,  
200 following the unconventional subgroup symbols listed in International Table of Crystallography  
201 Volume A (Hahn 2002).

202 The structure solution and refinement of the orthorhombic garnet is straightforward. A structure  
203 solution could be reached easily using the charge flipping method, with Ca and Si assigned to the  
204 cube-like (8-coordinated) sites and tetrahedral sites respectively. Fe and Al are assigned to the  
205 larger and smaller Y site ( $Y_1$  and  $Y_2$  with Wyckoff letter  $8c$  and  $8d$ ) respectively, and all the  
206 anion sites are assigned with O. After the initial convergence of the refinement, partial  
207 occupancies of Al and Fe are assigned to the  $Y_1$  and  $Y_2$  sites to account for the partial  
208 disordering in the structure, with the total occupancy of each site constrained to 1. Occupancies  
209 of the two Si sites are also relaxed to account for  $(\text{OH}, \text{F})_4$  substitution. The isotropic atomic  
210 displacement parameters (ADPs) are relaxed to harmonic ADPs for every atom in the structure in  
211 the last step, with the ADPs of Fe and Al in the same Y sites constrained to be identical. The  
212 final structure is checked to make sure there is no significant residual electron density, and all the

213 atoms have positive definite ADPs. All the atoms are refined as neutral atoms using the form  
214 factors from International Table Vol C tables 4.2.6.8 and 6.1.1.1. The chemical formula of the  
215 final refined structure is  $\text{Ca}_3(\text{Fe}_{0.804}\text{Al}_{0.196})(\text{Al}_{0.779}\text{Fe}_{0.221})\text{Si}_{2.917}\square_{0.083}\text{O}_{12}$ , which is very close to  
216 the composition from the EPMA analyses. Using the form factors of charged ions would result  
217 in a composition further from the EPMA analysis. Fe or Mn in the X sites and F in the anion  
218 sites are not considered in the refinement for simplicity, as disordering in every site of the  
219 structure would cause instability in the refinement. The slightly lower Fe and Si concentration  
220 compared to the EPMA analyses result can be explained by the anion sites only occupied by O in  
221 the refinement. If F is introduced to the anion sites, increasing the average electron density in the  
222 structure (larger scale parameter), the electron density of the cation sites would also increase  
223 accordingly, resulting in higher Fe and Si occupancies with a composition closer to the EPMA  
224 results.

225 The asymmetric unit of the refined structure is illustrated in Figure 5. The atom coordinates of  
226 the structure in the body-centered pseudo-cubic cell with  $I\frac{2}{a}1\frac{2}{a}$  symmetry are provided in Table  
227 4, along with the anisotropic temperature parameters (ADPs) of each atom in Table 5. The  
228 complete crystallographic information is provided in the Supplementary Material [data block I of  
229 Crystallographic Information File (CIF)], along with the same structure based on the  
230 conventional *Fddd* setting using the transformation  $\mathbf{a}' = \mathbf{a} + \mathbf{b}$ ,  $\mathbf{b}' = \mathbf{b} - \mathbf{a}$ ,  $\mathbf{c}' = \mathbf{c}$  (transformation  
231 matrix 1,1,0; -1,1,0; 0,0,1) (data block II). The crystal structure of the same crystal at 100K is  
232 also provided in the CIF file (data block III). Note that the reported structure in this paper (Table  
233 2, 4 and 5) is refined against symmetry averaged intensities without absorption correction (data  
234 blocks I – III) to avoid artifacts from data processing, the result of which is even slightly better  
235 than the symmetry-averaged absorption-corrected data. This means that simply averaging the  
236 symmetry equivalent peaks with highly redundant data is quite effective at correcting the errors  
237 from the experiment (absorption corrected data is still better for refinement against unaveraged  
238 intensities). The statistics listed in Table 3, on the other hand, are from absorption-corrected  
239 intensities (with *SADABS* in *APEX3*) using multi-scan method in Laue group  $\bar{1}$  (point group 1, so  
240 no symmetry bias is introduced from absorption correction). The structure refined against the  
241 unaveraged absorption-corrected intensities (15867 reflections total) is also provided in the CIF  
242 file (data block IV), in case the reader wants to perform symmetry or Hamilton tests on the

243 structure (Hamilton 1965). It is obvious that the symmetry loss is mainly due to the strong Fe-Al  
244 ordering in the Y site (an ordering parameter of  $\sim 0.8$ ), which is responsible for the significant  
245 lattice distortion with  $\gamma = 90.29(2)^\circ$  (largest in all reported non-cubic garnets). Small but obvious  
246 difference in the tetrahedral site occupancies is also detected in the structure, indicating the (OH,  
247 F) substitution prefers the more symmetrical  $Si_2$  site over  $Si_1$ . All the bond distances in the  
248 structure are listed in Table 6. The  $Y_1$ -O bonds are about 0.05 Å longer than the  $Y_2$ -O bonds. The  
249 relations between the Y-O bond lengths and the Fe occupancies are consistent with previously  
250 published results from Cr-rich ugrandite garnet (Shtukenberg et al. 2005). The Si-O bonds of the  
251 (OH, F)<sub>4</sub> preferred  $Si_2$  site are about 0.01 Å longer than those of  $Si_1$ -O bonds.

### 252 **Twinning and average structure of twinned garnet**

253 Pseudo-merohedral twinning cannot be avoided in non-cubic garnets, no matter what mechanism  
254 is causing the cation ordering in the structure. However, there are no reported considerations  
255 about possible twins in “single-crystal” structural refinements. The strongly birefringent zone in  
256 the garnet sample is obviously twinned, as indicated by the fluctuations in the interference colors  
257 (Figure 2). A diffraction frame of a screened crystal showing obvious peak splitting from  
258 twinning is provided in Figure S7. It is hard to estimate the average twin domain size of the  
259 birefringent zone, but based on the crystal picked for the single-crystal XRD, the larger domains  
260 are at least a few hundred microns in diameter. The high-resolution TEM image in Figure 6  
261 shows two twin domains smoothly transform into each other without an obvious boundary. The  
262 fast Fourier transform (FFT) patterns of the two regions show diffraction patterns of [100] and  
263 [001] zone-axes respectively, corresponding to the  $0kl$  and  $hk0$  sections of the X-ray diffraction  
264 pattern (Figure 4a,c). The lattice distortion of the sample in this study is large enough for any  
265 twinning to be easily detected, but a less ordered structure may have pseudo-merohedral  
266 twinning that is much harder to identify due to smaller deviation from cubic lattice. Powder X-  
267 ray diffraction is one way to bypass the problem introduced by twinning, which is used by  
268 several previous structural studies of anisotropic garnet (Kingma and Downs 1989; Antao 2013a,  
269 b, 2021a, b; Antao and Klincker 2013; Antao et al. 2015; Tančić et al. 2020). The powder  
270 diffraction pattern of the birefringent zone of the garnet is shown in Figure 7. The diffraction  
271 pattern seems overall very similar to a cubic garnet, as the peak splitting due to lattice distortion  
272 is impossible to resolve in a lab based powder X-ray diffractometer. However, some reflections

273 violating the  $Ia\bar{3}d$  symmetry (i.e. 101, 103, 303) can be clearly seen in the diffraction pattern,  
274 which has not been observed in any previous studies. Diffraction peaks from powder XRD of the  
275 orthorhombic garnet are listed in Table S1.

276 Lowering in symmetry (or dissymmetrization) from cubic  $Ia\bar{3}d$  to orthorhombic  $I\frac{2}{a}1\frac{2}{d}$  would  
277 give rise to 6 possible different orientations, each of which corresponds to one possible pseudo-  
278 merohedral twin domain. Two distinct twin operations are required to generate all 6 orientations,  
279 namely  $\{010\}$   $a$ -glide and  $\{101\}$   $d$ -glide, which are the glide symmetries lost from the  $Ia\bar{3}d \rightarrow$   
280  $I\frac{2}{a}1\frac{2}{d}$  transformation. Note that twinning by rotational operations (i.e., 2-fold around  $\langle 101 \rangle$ , 3-  
281 fold around  $\langle 111 \rangle / \langle 1\bar{1}1 \rangle$ , and 4-fold around  $\langle 100 \rangle / \langle 001 \rangle$ ), in which the composition planes are  
282 parallel instead of perpendicular to the operation axes, would be technically different from  
283 twinning by glide-planes. Although in theory the exact twin operation may be deduced from the  
284 optical indicatrices across sharp twin boundaries [i.e., the crystals reported by Hariya and  
285 Kimura (1978), Akizuki (1984) and Jamtveit (1991)], they are not distinguishable from the  
286 diffraction pattern, especially when the distortion from cubic lattice is very small. The 6 different  
287 orientations (twin domains) can be labeled TD<sub>oo</sub>, TD<sub>od</sub>, TD<sub>od'</sub>, TD<sub>ao</sub>, TD<sub>ad</sub>, TD<sub>ad'</sub> following  
288 the symmetry relations among one another, as illustrated in Figure 8 [i.e. TD<sub>oo</sub> is related to  
289 TD<sub>od</sub> by a (101)-glide, and to TD<sub>ad'</sub> by a (100)-glide plus a (011)-glide].

290 A twinned orthorhombic garnet crystal may contain domains of all 6 different orientations with  
291 different proportions. Any two twin domains related by  $\{101\}$   $d$ -glide operation would create  
292 reflections violating the extinction conditions of  $I\frac{2}{a}1\frac{2}{d}$  space group, resulting in a triclinic  
293 symmetry for the diffraction pattern. Reducing the symmetry from  $I\frac{2}{a}1\frac{2}{d}$  to  $I\bar{1}$  would  
294 differentiate each Y site into four symmetrically different sites (Shtukenberg et al. 2005), which  
295 are labeled for TD<sub>oo</sub> in Figure 8. All the possible twin domains have the same topology and only  
296 the position of Al and Fe are swapped in a certain way from one another (Figure 8). Therefore, if  
297 an  $I\bar{1}$  structure is refined against a twinned diffraction pattern of an orthorhombic garnet, the  
298 result would be a weighted average of the 6 possible orientations. Assuming the relative  
299 proportion of each twin domain in a twinned crystal is  $t_{oo}$ ,  $t_{od}$ ,  $t_{od'}$ ,  $t_{ao}$ ,  $t_{ad}$  and  $t_{ad'}$  respectively, the  
300 corresponding Fe occupancy in each Y site of the corresponding  $I\bar{1}$  structure would be:

$$Fe_{Y1,1} = Fe_{Y1,3} = (t_{oo} + t_{od} + t_{od'})Fe_{Y1} + (t_{ao} + t_{ad} + t_{ad'})Fe_{Y2}$$

$$Fe_{Y1,2} = Fe_{Y1,4} = (t_{oo} + t_{ad} + t_{ad'})Fe_{Y1} + (t_{ao} + t_{od} + t_{od'})Fe_{Y2}$$

$$Fe_{Y2,1} = Fe_{Y2,4} = (t_{ao} + t_{ad} + t_{ad'})Fe_{Y1} + (t_{oo} + t_{od} + t_{od'})Fe_{Y2}$$

$$Fe_{Y2,2} = Fe_{Y2,3} = (t_{ao} + t_{od} + t_{ad'})Fe_{Y1} + (t_{oo} + t_{ad} + t_{od'})Fe_{Y2}$$

301 Note that the refined structure is not an exact average because the intensity of a certain reflection  
302 from a single crystal is proportional to the square of the structure factor. Nonetheless, the  
303 unusual pattern shown in the equations above is expected for the  $I\bar{1}$  structure refined from a  
304 twinned orthorhombic data set, where pairs of Y sites that are not symmetrically equivalent have  
305 the same occupancies. This is observed in almost all of the previously published  $I\bar{1}$  structures  
306 (Takéuchi et al. 1982; Kingma and Downs 1989; Wildner and Andrut 2001; Shtukenberg et al.  
307 2005; Frank-Kamenetskaya et al. 2007). Hatch and Griffen (1989) first noticed this phenomenon  
308 in the  $I\bar{1}$  garnet structure reported by Takéuchi et al. (1982), but misinterpreted this as a single  
309 order parameter in different orientations, instead of twinning. The Landau theory (based on  
310 group theory of the  $Ia\bar{3}d$  space group) used by Hatch and Griffen (1989) only applies to the  
311 temperatures close to the phase transition, whereas the studied  $I\bar{1}$  structures are supposedly  
312 ordered further at much lower temperatures, therefore has no constraints to preserve this special  
313 pattern. We have simulated a series of twinned diffraction data of the orthorhombic garnet  
314 structure and used them to refine structures in the  $I\bar{1}$  space group. The results are provided in the  
315 Appendix, which demonstrate different ordering patterns can indeed result from twinning.

316 It should be noted that twinning by (100) glide plane flips all the  $Y_1$  sites to  $Y_2$  sites (and vice  
317 versa), with the orientations of all the symmetry elements unchanged (Figure 8). This means a  
318 crystal with only (100) glide twins would still show an  $I\frac{2}{a}1\frac{2}{d}$  space group in the symmetry test,  
319 but the Fe/Al occupancies would be averaged out between  $Y_1$  and  $Y_2$  sites, resulting in a smaller  
320 order parameter. The (OH, F) ordering in the Si sites, however, would be preserved, as they  
321 remain in the same positions after this twin operation. This also means the only pseudo-  
322 monoclinic symmetry that can be created by averaging different twin domains is  $I11\frac{2}{d}$ , and  
323 monoclinic structures of  $I\frac{2}{a}$  space group can only be created by further ordering (Figure 1 and  
324 Appendix).

325

326

## DISCUSSIONS

327 The main reason causing the ambiguity of the cation ordering mechanism for the anisotropic  
328 grandite garnet is the large variety of different structures reported. The heterogeneity often  
329 observed in these birefringent garnet crystals also make the case more complicated. There are  
330 obviously many different factors controlling the final appearance and ordering states of the  
331 grandite garnet, including growth kinetics and thermodynamics of the solid solution. The  
332 structure reported in this paper shows an unambiguous orthorhombic symmetry, with much  
333 stronger Fe/Al ordering between two Y sites than any previously reported garnet structures. The  
334 composition very close to the ideal stoichiometry of a perfectly ordered structure  $\text{Adr}_{50}\text{Gr}_{50}$  also  
335 strongly suggest a thermodynamically stable phase exist for intermediate compositions in the  
336 solid solution at low temperature. A schematic phase diagram of the grossular-andradite solid  
337 solution is presented in Figure 9 based on the studied garnets and previously reported garnets  
338 with known temperature range by Jamtveit (1991).

339 Growth kinetics of garnet plays an important role in the formation of birefringent garnet, as  
340 evident by the zoning and sector twinning that are often observed (Hariya and Kimura 1978;  
341 Akizuki 1984; Jamtveit 1991), and may affect the cation ordering in the structure as many  
342 studies suggested (Callen 1971; Gali 1983; Akizuki 1984; Shtukenberg et al. 2001, 2005; Frank-  
343 Kamenetskaya et al. 2007). However, the “growth dissymmetrization” phenomenon is mostly  
344 theoretical with no direct evidences, and kinetics is mostly known to favor the growth of  
345 disordered instead of ordered structures under metastable conditions (Carpenter and Putnis 1985;  
346 Pollok et al. 2001). It is not possible for a strongly ordered orthorhombic structure to be created  
347 only from the “growth dissymmetrization”. Most models for growth-induced ordering predicts an  
348  $I\bar{1}$  structure (Shtukenberg et al. 2005; Frank-Kamenetskaya et al. 2007), and growth normal to  
349  $\{110\}$  planes that is required to create orthorhombic symmetry (Gali 1983) should create  
350 perfectly oriented domains relative to the growth surface, instead of the randomly twinned  
351 domains as observed in the sample studied in this paper (Figures 2 and 6). The large order  
352 parameter ( $\sim 0.8$ ) of the structure studied in this paper also cannot be explained by growth  
353 kinetics. As demonstrated in the previous section, pseudo-merohedral twinning can create all  
354 different kinds of pseudo-symmetry, which none of the previous structure studies have  
355 considered in their refinements. It is most likely that the large varieties of different non-cubic

356 structures are results of twinned crystals of different domain proportions. It needs to be noted  
357 that the orthorhombic symmetry is only expected for Y site ordering, and ordering in the X or Z  
358 sites could result in different subgroup symmetries [i.e. tetragonal for X site ordering (Griffen et  
359 al. 1992; Cesare et al. 2019) and trigonal for Z site ordering (Krivovichev et al. 2021)], which are  
360 easily distinguishable based on chemical analysis. Moreover, submicron chemical zoning and  
361 intergrowths are very common in anisotropic garnets (Ingerson and Barksdale 1943; Akizuki et  
362 al. 1984; Hainschwang and Notari 2006; Antao 2013b; Nakamura et al. 2017), which means  
363 some of the previously reported structures may not even be from a single homogeneous phase.  
364 Note that the symmetry can be conclusively determined in this paper in a twin-free crystal only  
365 because the ordering is strong enough to significantly distort the lattice from cubic symmetry. In  
366 a less ordered and more densely twinned crystal, the lattice with strained twin boundaries can  
367 easily show metrically cubic symmetry, similar to the effect observed in micro-tweed structure of  
368 orthoclase with metrically monoclinic lattice (Eggleton and Buseck 1980; Xu et al., 2019).

369 The strongest evidence for the thermodynamic stability of orthorhombic garnet structure for  
370 intermediate compositions at low temperature (Figure 9) comes from the oscillatory growth  
371 zoning between  $\sim\text{Adr}_{100}$  (pure andradite) and  $\sim\text{Adr}_{53}\text{Grs}_{47}$  with particularly consistent  
372 compositions and sharp boundaries (Figure 2). The same phenomena have been reported by in  
373 garnet from Colorado, Thera Island (Aegean Sea), and southern Norway (Lessing and Standish  
374 1973; Murad 1976; Jamtveit et al. 1993, 1995; Pollok et al. 2001). Similar phenomena between  
375  $\sim\text{Adr}_{10}\text{Grs}_{90}$  and  $\sim\text{Adr}_{50}\text{Grs}_{50}$ , as well as coexisting unzoned isotropic and anisotropic grandite  
376 garnets in skarn zones around hydrothermal veins, have also been reported in other birefringent  
377 garnets (see Jamtveit 1991). This indicates the presence of a miscibility gap between  $\sim\text{Adr}_{100}$  and  
378  $\sim\text{Adr}_{50}\text{Grs}_{50}$  (also between  $\sim\text{Adr}_0$  and  $\sim\text{Adr}_{50}$ ) at low temperature in the solid solution series  
379 (Figure 9). The lamellar intergrowths observed by Hirai and Nakazawa (1986), which are  
380 interpreted as exsolution textures also indicate immiscibility for certain compositions. Note that  
381 thermodynamic stability of the orthorhombic garnet phase means birefringent garnet can be  
382 formed by both direct growth at low temperature and through phase transformation during  
383 cooling, which is consistent with the various zoning and textures observed in natural and  
384 synthetic garnets. Jamtveit (1991) successfully models the oscillatory zoning pattern by  
385 considering the miscibility gaps in the solid solution using the theory of nonlinear dynamics. The  
386 ordered orthorhombic structure perfectly explains the presence of the miscibility gaps in the solid

387 solution (Figure 9), where the Gibbs free energy is dramatically decreased for intermediate  
388 grandite through Fe/Al ordering, driving the phase separation between intermediate grandite and  
389 pure end members. It is possible that the periodic lamellar intergrowth/zoning found in iridescent  
390 garnet (Ingerson and Barksdale 1943; Akizuki et al. 1984; Hainschwang and Notari 2006;  
391 Nakamura et al. 2017) could also be formed during oscillatory growth suggested by Jamtveit  
392 (1991).

393 Thermal information of previously reported anisotropic garnet structures are very limited,  
394 making it difficult to constrain the shapes of the miscibility gaps in the phase diagram (Figure 9).  
395 The fluid inclusions trapped in the garnet, along with the chemical composition of different  
396 zones in the garnet studied in this paper, shows that the miscibility gap is almost at its widest at  
397  $\sim 200^\circ\text{C}$ . Grandite with compositions of  $\sim\text{Adr}_{95}\text{Grs}_5$  and  $\sim\text{Adr}_{65}\text{Grs}_{35}$  were found coexisting in  
398 the skarn (sample R8923) at  $\sim 400^\circ\text{C}$  by Jamtveit (1991), which indicates the miscibility gap is  
399 asymmetrical tilting towards the andradite end member, instead of the shape from molecular  
400 simulation by Becker and Pollok (2002). Similar compositional gap (between  $\sim\text{Grs}_{65}\text{Adr}_{35}$  and  
401  $\sim\text{Grs}_{90}\text{Adr}_{10}$ ) was reported in the grossular-rich side (Engi and Wersin, 1987). Garnet crystals  
402 from skarn deposits have documented homogenization temperatures of fluid inclusions that  
403 range from  $\sim 400^\circ\text{C}$  to  $\sim 150^\circ\text{C}$  (Boni et al. 1990), which means all the garnets found in skarn  
404 with intermediate compositions are at least partially ordered. Optical anisotropy can be recreated  
405 from an isotropic natural grandite (heated) with composition of  $\sim\text{Adr}_{90}\text{Grs}_{10}$  at  $600^\circ\text{C}$  (Hariya  
406 and Kimura 1978), but synthetic grandite shows an isotropic/anisotropic transition at much lower  
407 temperature. It should be noted that the ordering kinetics of cations in the garnet structure is  
408 poorly understood. The Al/Si ordering rate in feldspar structures are known to change  
409 dramatically with the amount of  $\text{H}_2\text{O}$  in the composition (Yund and Tullis 1980). Therefore, a  
410 large variation in the cation ordering rate in garnet may be expected considering the wide range  
411 of (OH) content that can be incorporated into the structure.

412 The residual strain caused by lattice discrepancy in chemically zoned garnet has been proposed  
413 to produce birefringence (Akizuki 1984; Hirai and Nakazawa 1986; Antao and Klincker 2013).  
414 However, the pure andradite zone in the sample studied in this paper shows no observable optical  
415 anomaly, despite the sharp boundaries with the orthorhombic grandite with drastically different  
416 composition ( $\sim\text{Adr}_{53}\text{Grs}_{47}$ ). Stress-induced birefringence is expected to be very low in garnets

417 (Lynch et al. 1973), and dissipates quickly away from the strained interface, as shown by the  
418 birefringent halos around zircon and coesite inclusions in garnets (Campomenosi et al. 2018).  
419 None of the previously reported anisotropic garnet crystals are pure end member, which means it  
420 is likely that cation ordering is at least partially involved in all observed optical anomalies in  
421 grandite garnet. Reported garnets with high birefringence ( $\sim 0.010$ ) all have compositions close  
422 to  $\text{Adr}_{50}\text{Grs}_{50}$  (Mariko and Nagai 1980; Hirai and Nakazawa 1986; Frank-Kamenetskaya et al.  
423 2007), and the birefringence vs composition plots (Mariko and Nagai 1980; Shtukenberg et al.  
424 2001) show that the higher the birefringence, the closer the composition is to the middle of the  
425 solid solution. Theoretical calculation using a quantum mechanical approach indicates that a  
426 fully ordered structure with composition of  $\text{Adr}_{50}\text{Grs}_{50}$  has the highest birefringence ( $\delta = 0.023$ )  
427 (Lacivita et al. 2013), which is consistent with the observed birefringence ( $\delta \sim 0.021$ ) of the  
428 orthorhombic garnet in this paper.

429

430

## IMPLICATIONS

431 The crystal structures and ordering states of natural birefringent garnet crystals are  
432 thermodynamically controlled, which can record information regarding the cooling history of the  
433 host rock. The textures observed in anisotropic garnet can also be twin domains or even  
434 exsolution lamellae from solid state phase transitions, instead of just growth features recording  
435 the crystallization condition as suggested by the “growth dissymmetrization” theory. The phase  
436 diagram of the grossular-andradite solid solution should be studied in detail to better constrain  
437 the shapes and positions of the miscibility gaps between the orthorhombic and the cubic (pure  
438 end members) phases, which may potentially be used as geothermometer and/or geobarometer.  
439 Based on the reported structure of uvarovite, it is very likely that the uvarovite-grossular solid  
440 solution has a similar phase relation with orthorhombic Cr/Al ordering for intermediate  
441 compositions at low temperature. The solvi between orthorhombic Al-Cr garnet and its end-  
442 members (grossular & uvarovite) will be narrower than the solvi in the grossular-andradite  
443 system (Figure S8), because the difference between  $\text{Al}^{3+}$  and  $\text{Cr}^{3+}$  is smaller than the difference  
444 between  $\text{Al}^{3+}$  and  $\text{Fe}^{3+}$  in their ionic radii and chemical hardness (Xu et al. 2017).  
445 Most of the previously reported non-cubic garnet structures, especially those with  $I\bar{1}$  space group  
446 symmetry, are most likely from twinned orthorhombic crystals, potentially also containing

447 intergrowth of more than one phase with different compositions. These structures should be  
448 revisited to better evaluate their structures and ordering states. The conclusions made based on  
449 triclinic (pseudo-)symmetry should be cited with immense caution. Future studies of anisotropic  
450 garnet structures should carefully take potential twinning into consideration during the  
451 refinement. Structure studies of garnet with powder XRD, although bypassing the twinning  
452 problem, miss important information to determine the correct symmetry due to significant peak  
453 overlapping, therefore powder XRD is not reliable when used by itself. Combination of single-  
454 crystal diffraction and high-resolution TEM, plus selected-area electron diffraction (SAED), as  
455 demonstrated in previous study (Jin et al. 2018), is the most reliable method for determining  
456 crystal structures with pseudo-symmetry. Careful examination of raw single-crystal diffraction  
457 data is critical, rather than simply following the automated data reduction procedure in the  
458 programs, especially when studying crystals with potential twinning and pseudo-symmetry.

459

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- 666

## 667 Captions of figures

668 Figure 1. The centrosymmetric subgroup graph of the space group  $Ia\bar{3}d$ , along with the lattice  
669 parameter constraints and Wyckoff symbols of the X, Y, and Z sites in each space group.  
670 All the Hermann–Mauguin notations in the graph are based on the body-centered cubic  
671 setting (1<sup>st</sup> position-[001], 2<sup>nd</sup> position-[111], 3<sup>rd</sup> position-[110]), for easily comparing the  
672 symmetry elements among different space groups. Some unconventional space group  
673 symbols are created this way, with the corresponding conventional symbols in  
674 parenthesis. The unconventional settings are necessary in this case to differentiate the two  
675 distinct settings of the monoclinic subgroup  $C2/c$ . The multiplicity of each Wyckoff  
676 position may be different from the conventional space group setting due to changes in  
677 unit cell volumes (i.e., the  $I\frac{2}{a}1\frac{2}{a}$  unit cell has half the volume of the conventional  $Fddd$   
678 unit cell). The critical symmetry elements ( $4_1$ ,  $2/a$ ,  $2/c$ ,  $2/d$ , and 3) that are lost for each  
679 supergroup-subgroup relation is marked on the arrows ( $2/c$  is a symmetry in the [100]  
680 direction which is not explicit in the Hermann-Mauguin notation in cubic setting). Note  
681 that these unconventional space group symbols are used in the International Tables of  
682 Crystallography Volume A. for the “maximal non-isomorphic subgroups” (Hahn 2002).

683 Figure 2. Optical image (crossed polarizer) of a thin section cut perpendicular to [111] of cubic  
684 unit cell setting (a) and back-scattering electron microscopic image (high resolution  
685 image provided in Figure S3) (b) from lower part area of image (a) showing oscillatory  
686 zoning between orthorhombic garnet and andradite. Optical image (crossed polarizer) of  
687 a thin section (50  $\mu\text{m}$  thick) cut perpendicular to  $\sim b$ -axis (c) and back-scattering electron  
688 microscopic image (d) showing oscillatory zoning between orthorhombic garnet and  
689 andradite. The crystal also displays sector twinning. Heterogeneous interference colours  
690 in the orthorhombic garnet layers result from domains with different orientations in twin  
691 relationships. The areas with uniform highest interference colour (second order red here,  
692 see areas inside white circles) were selected for single-crystal XRD work. See Figure S4  
693 in the supplementary material for the cutting planes / positions in the ideal  $\{110\}$  garnet  
694 shape for the two thin sections.

695 Figure 3. Photomicrographs of fluid inclusions inside the oscillatory andradite and orthorhombic  
696 garnet zone during microthermometry. V = vapor, L = liquid,  $T_h$  = homogenization  
697 temperature,  $T_f$  = freezing temperature of liquid phase, and  $T_m$  = melting temperature of  
698 ice.

699 Figure 4. Reconstructed sections of the X-ray diffraction data showing symmetry elements  
700 through systematic extinctions. The remaining mirror planes (*d*-glide in real space) and  
701 associated extinctions are marked with dash lines and dash circles.

702 Figure 5. The asymmetric unit of the orthorhombic garnet structure projected along *a*-axis. The  
703 2-fold rotation symmetry relating the Y sites in the cubic structure is lost, creating two  
704 distinct  $Y_1$  (red) and  $Y_2$  (cyan) sites.

705 Figure 6. High-resolution image showing neighboring domains with (101)-glide twin  
706 relationship. The fast Fourier transform (FFT) patterns of the two areas show [010] and  
707 [001] zone-axis diffraction patterns of the orthorhombic structure respectively,  
708 corresponding to the *h0l* and *hk0* sections of the XRD data in Figure 4.

709 Figure 7. Powder XRD data along with the calculated pattern from the refined orthorhombic  
710 structure.

711 Figure 8. All 6 possible twin orientations of orthorhombic garnet. Only the Y octahedra are  
712 shown. Red octahedra are  $Y_1$  (Fe) sites and cyan ones are  $Y_2$  (Al) sites. The twin domains  
713 are labeled based on their symmetry relation (twin operation) with the reference domain  
714  $TD_{00}$ . The Y sites of  $TD_{00}$  are labeled based on the  $I\bar{1}$  symmetry in the figure. Note that  
715 the same labels apply to all 6 orientations in the figure, because there is no standard or  
716 preferred orientation in the triclinic symmetry.

717 Figure 9. A phase diagram modified from Becker and Pollok (2002) showing stability field for  
718 the ordered orthorhombic garnet with  $I\frac{2}{a}1\frac{2}{a}$  symmetry at low temperature. Only a  
719 temperature of 200 °C for the coexisting andradite and the orthorhombic garnet and data  
720 from Jamtveit (1991) with known temperature range of 300 °C – 400 °C are labeled

721 based on the studied sample. The upper temperature limit for  $I \frac{2}{a} 1 \frac{2}{d}$  structure is not  
722 determined.  
723

## 724 **Appendix**

725 A series of diffraction data of twinned orthorhombic crystals of various proportions are  
726 simulated to study the effect of pseudo-merohedral twinning on the structure refinement. The  
727 single-crystal data of the orthorhombic structure is simulated using JANA2006 program with a  
728 noise level of 1. The data is then multiplied with a twin matrix and added to the original data at a  
729 certain ratio. Each resulting simulated data is represented by a 6-digit number which indicates  
730 the relative proportion of each individual twin domain ( $t_{oo}$ ,  $t_{od}$ ,  $t_{od'}$ ,  $t_{ao}$ ,  $t_{ad}$ ,  $t_{ad'}$ ). For example,  
731 111111 is an intergrowth of TD<sub>oo</sub>, TD<sub>od</sub>, TD<sub>od'</sub>, TD<sub>ao</sub>, TD<sub>ad</sub>, TD<sub>ad'</sub> twin domains with same  
732 volume; 200100 is an intergrowth of TD<sub>oo</sub> and TD<sub>ao</sub> twin domains with 2:1 volume ratio, but  
733 without TD<sub>od</sub>, TD<sub>od'</sub>, TD<sub>ad</sub>, TD<sub>ad'</sub> domains. Although higher (pseudo-)symmetry may be  
734 present, we refined all the simulated diffraction data as a single crystal  $I\bar{1}$  structure. The resulting  
735 Fe occupancies of each Y site are listed in Table A1, along with the R factors of each structure  
736 refinement. If higher pseudo-symmetry is present, equivalent Y sites are coded with the same  
737 color.

738 The structure 100100 shows no Fe-Al ordering between Y<sub>1</sub> and Y<sub>2</sub> sites, but the symmetry is still  
739 orthorhombic. As discussed in the main text, the (100) glide plane twin law would only switch  
740 the positions of Y<sub>1</sub> and Y<sub>2</sub>, but keep the Si sites unchanged, which is why even though the Y<sub>1</sub>  
741 and Y<sub>2</sub> shows same occupancy, they are still symmetrically different sites. Structure 200100  
742 obviously shows an orthorhombic structure with less ordered Y sites. Structure 111000 shows a  
743 [111] 3-fold rotoinversion symmetry. Structure 110110 shows pseudo-tetragonal  $I4_1/acd$  space  
744 group symmetry with all Y sites equivalent to each other. And with all six twin domains of equal  
745 volume to each other, the structure 111111 restores all the symmetry elements in the  $Ia\bar{3}d$  space  
746 group. It needs to be emphasized that all this pseudosymmetry produced by a certain twin  
747 domain component would fail any symmetry test regarding the extinction conditions. All the  
748 twinned crystals show no systematic extinctions required for screw axes and glide planes.

749 The paired-occupancy pattern as discussed in the main text are not exactly followed in the  
750 refined structure. This should be expected since the “average” structure of all the twin domains is  
751 not really an average of each site in the structure, and there is no symmetry constraint to make  
752 the occupancies equal to each other. The intensity of a certain reflection from a single crystal is

753 proportional to the square of the structure factor, whereas the reflection from a twinned crystal is  
 754 the arithmetic mean of the intensities from different domains. The difference between Fe  
 755 occupancies in  $Y_{1\_1}$  and  $Y_{1\_3}$  can be up to about 0.1 in structure 210000 and 510000 (refined  
 756 from simulated twinned diffraction data) (Table A1). It is also interesting that the fake  $I\bar{1}$   
 757 structure can show an even wider range of Fe occupancies than the real orthorhombic structure,  
 758 even though the average order between  $Y_1$  and  $Y_2$  is always lower in the  $I\bar{1}$  structures. The  
 759 highest Fe ordering in the limited variations we tested is 0.888 of  $Y_{1\_3}$  site in the structure  
 760 210000 (refined from simulated twinned diffraction data) (Table A1), and the lowest is 0.146 of  
 761  $Y_{2\_4}$  in the same structure.

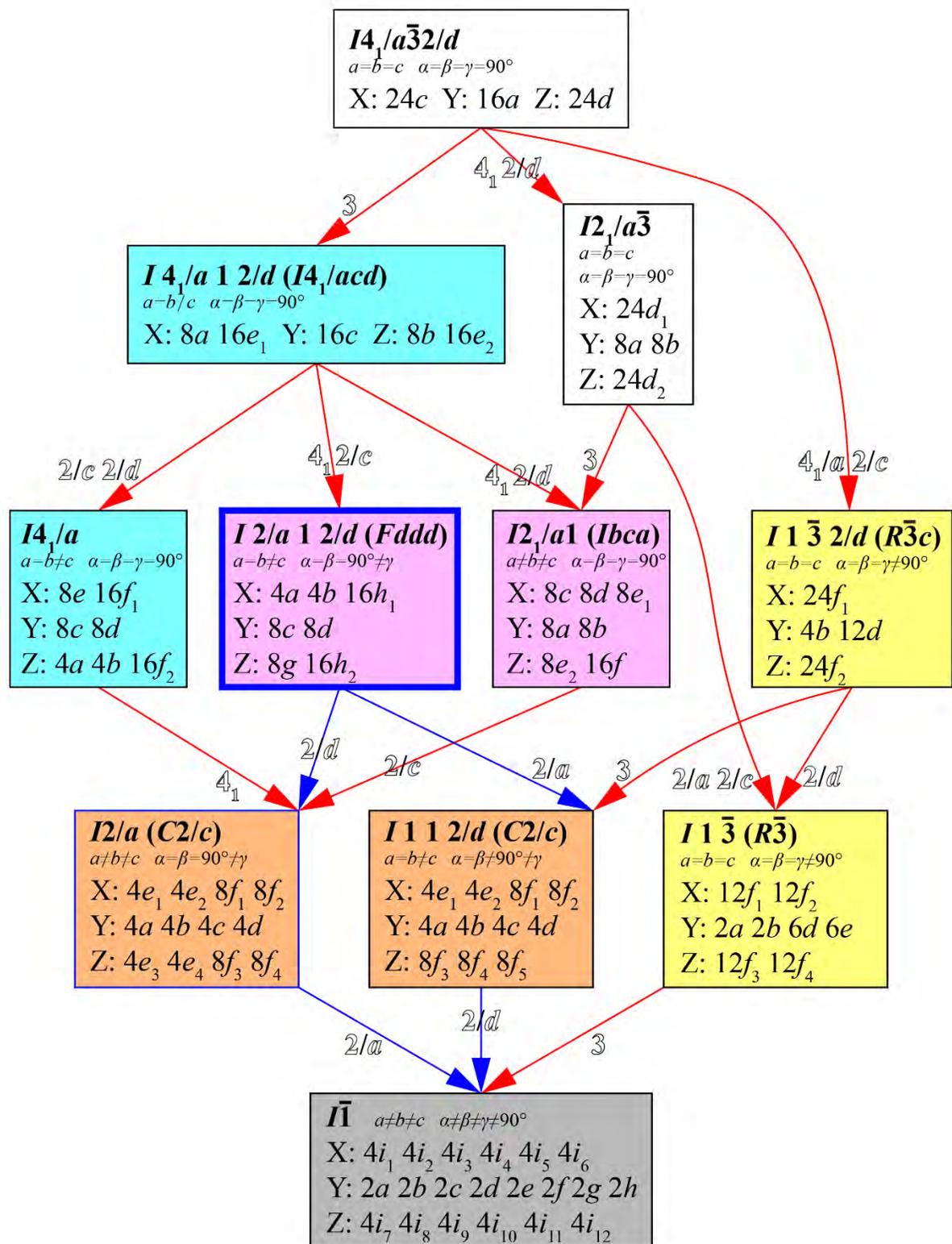
762 Table A1: Fe occupancies of the structures refined from simulated diffraction pattern of twinned  
 763 crystals. The 6-digit number gives the relative proportion of each individual twin domain (TD<sub>00</sub>,  
 764 TD<sub>01</sub>, TD<sub>02</sub>, TD<sub>03</sub>, TD<sub>04</sub>, TD<sub>05</sub>). The difference between the mean occupancies over each  
 765 quartet of octahedra ( $x_{Oct1}-x_{Oct2}$  as defined by Shtukenberg et al, 2005) is also listed in the table  
 766 for each structure.

Site	Coordinate	Twin Domains and Their Proportions					
		100000	100100	200100	110000	210000	510000
$Y_{1\_1}$	(0,0,0)	0.8118(8)	0.501(11)	0.666(10)	0.841(7)	0.799(6)	0.800(5)
$Y_{1\_2}$	(1/2,0,0)	0.8121(8)	0.497(11)	0.667(10)	0.551(7)	0.676(6)	0.753(5)
$Y_{1\_3}$	(1/4,1/4,1/4)	0.8137(8)	0.505(11)	0.673(10)	0.843(7)	0.888(6)	0.887(5)
$Y_{1\_4}$	(1/4,1/4,3/4)	0.8140(8)	0.505(11)	0.675(10)	0.484(7)	0.607(6)	0.690(5)
$Y_{2\_1}$	(0,1/2,0)	0.2288(8)	0.497(11)	0.347(10)	0.190(6)	0.236(6)	0.240(5)
$Y_{2\_2}$	(1/2,1/2,0)	0.2284(8)	0.500(11)	0.347(10)	0.482(7)	0.362(6)	0.288(4)
$Y_{2\_3}$	(1/4,3/4,1/4)	0.2297(8)	0.504(11)	0.348(10)	0.549(7)	0.427(6)	0.349(5)
$Y_{2\_4}$	(3/4,1/4,1/4)	0.2287(8)	0.505(11)	0.349(10)	0.188(6)	0.146(6)	0.151(4)
R factor %		0.38	5.48	4.58	2.95	2.63	1.92
Symmetry		Orthorhombic	Orthorhombic	Orthorhombic	Monoclinic	Triclinic	Triclinic
$x_{Oct1}-x_{Oct2}$		0.584	0.0005	0.3225	0.3275	0.44975	0.5255
Site	Coordinate	Twin Domains and Their Proportions					
		111000	302010	302100	321000	110110	111111
$Y_{1\_1}$	(0,0,0)	0.823(9)	0.815(8)	0.767(9)	0.811(8)	0.490(13)	0.498(13)
$Y_{1\_2}$	(1/2,0,0)	0.409(9)	0.576(8)	0.488(9)	0.604(8)	0.487(13)	0.489(13)
$Y_{1\_3}$	(1/4,1/4,1/4)	0.857(9)	0.750(8)	0.802(9)	0.879(8)	0.501(13)	0.504(13)
$Y_{1\_4}$	(1/4,1/4,3/4)	0.395(9)	0.663(8)	0.553(9)	0.519(8)	0.515(13)	0.513(13)

Y <sub>2_1</sub>	(0,1/2,0)	0.407(9)	0.511(8)	0.530(9)	0.283(8)	0.501(13)	0.499(13)
Y <sub>2_2</sub>	(1/2,1/2,0)	0.408(9)	0.158(8)	0.250(9)	0.367(8)	0.498(13)	0.495(13)
Y <sub>2_3</sub>	(1/4,3/4,1/4)	0.396(9)	0.211(8)	0.231(9)	0.452(8)	0.514(13)	0.510(13)
Y <sub>2_4</sub>	(3/4,1/4,1/4)	0.394(9)	0.435(8)	0.477(9)	0.210(8)	0.498(13)	0.495(13)
R factor %		3.87	3.54	4.28	3.54	5.90	6.11
Symmetry		p-Trigonal	Triclinic	Triclinic	Triclinic	p-Tetragonal	p-Cubic
X <sub>Oct1</sub> -X <sub>Oct2</sub>		0.21975	0.37225	0.2805	0.37525	-0.0045	0.00125

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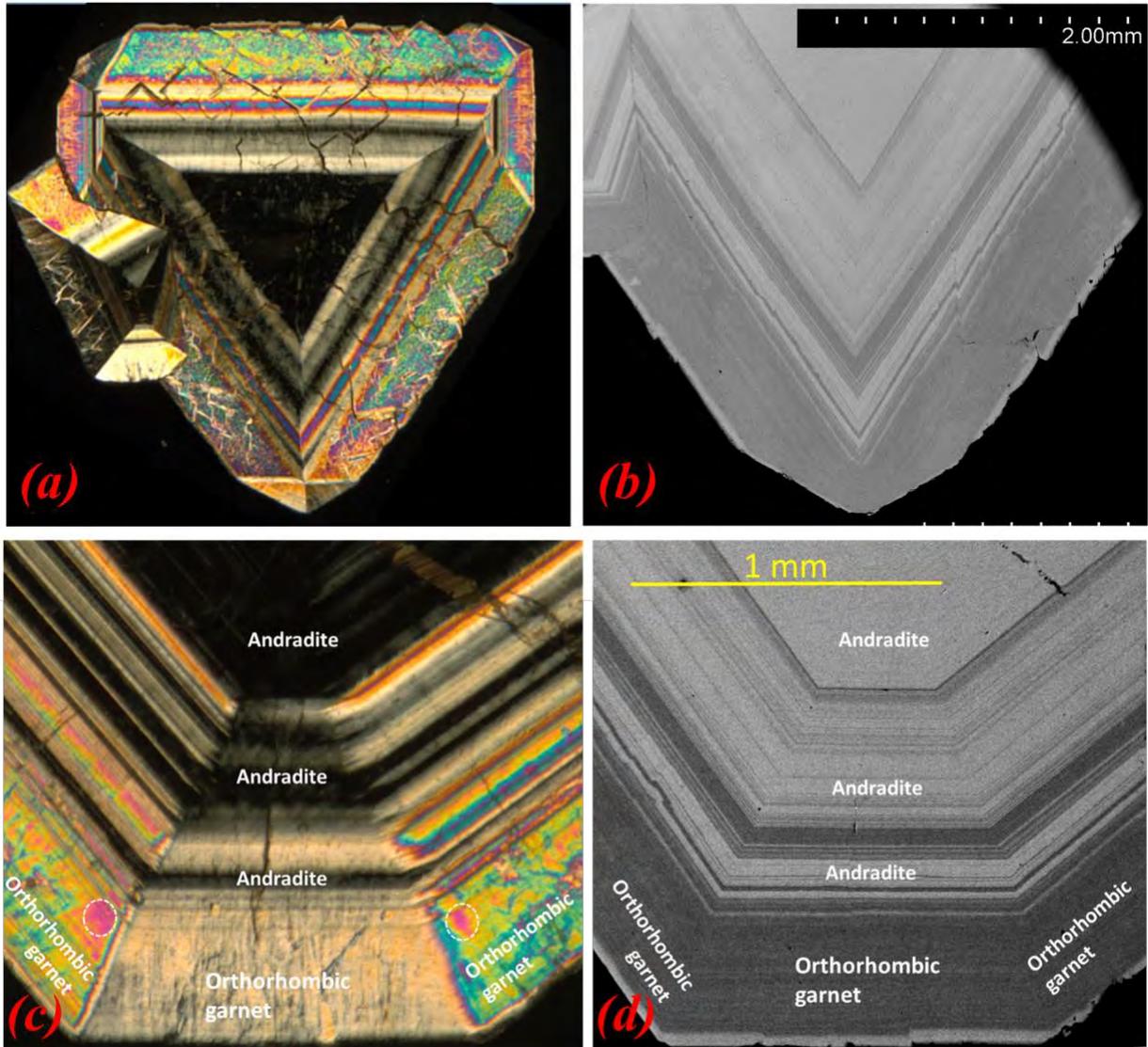
Figures



770 Figure 1.

771

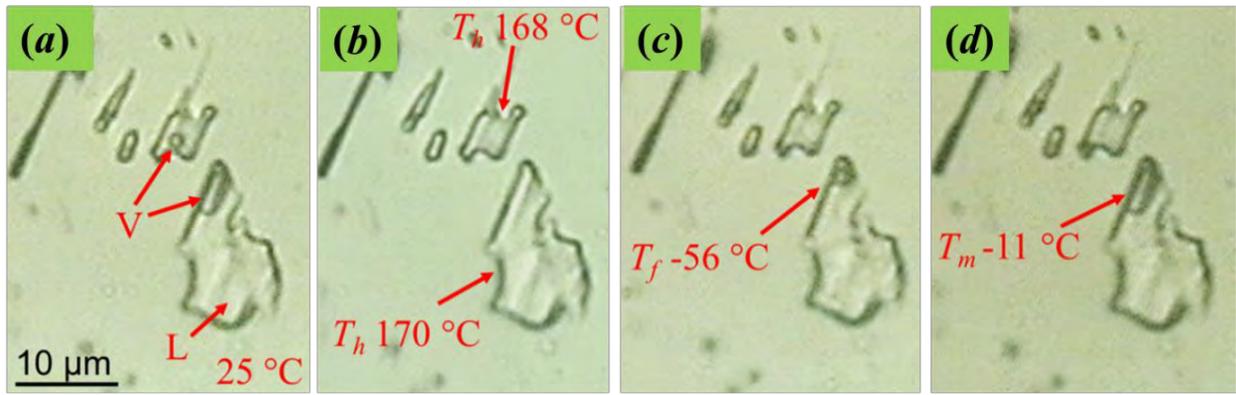
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774 Figure 2.

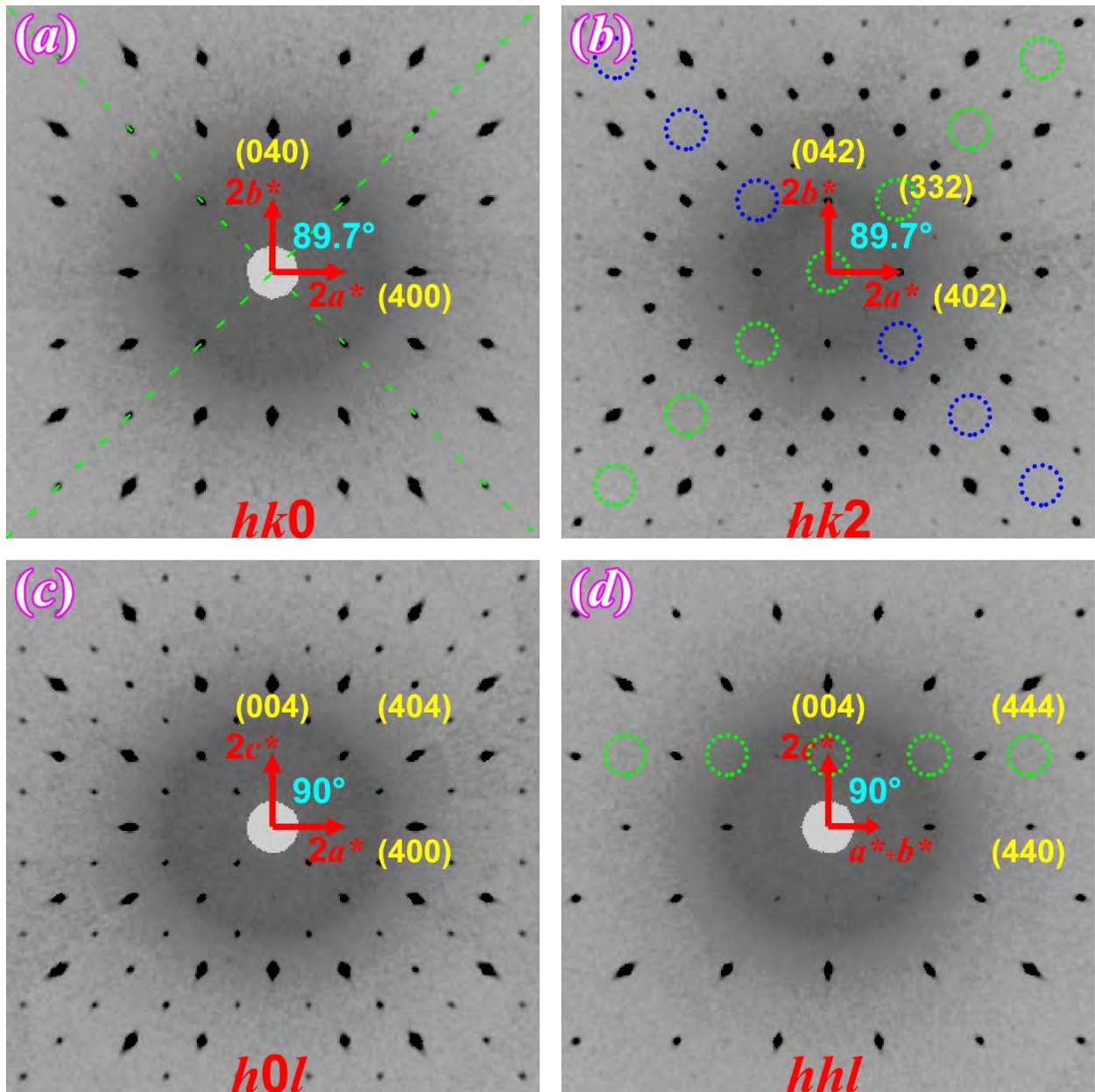
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777 Figure 3.

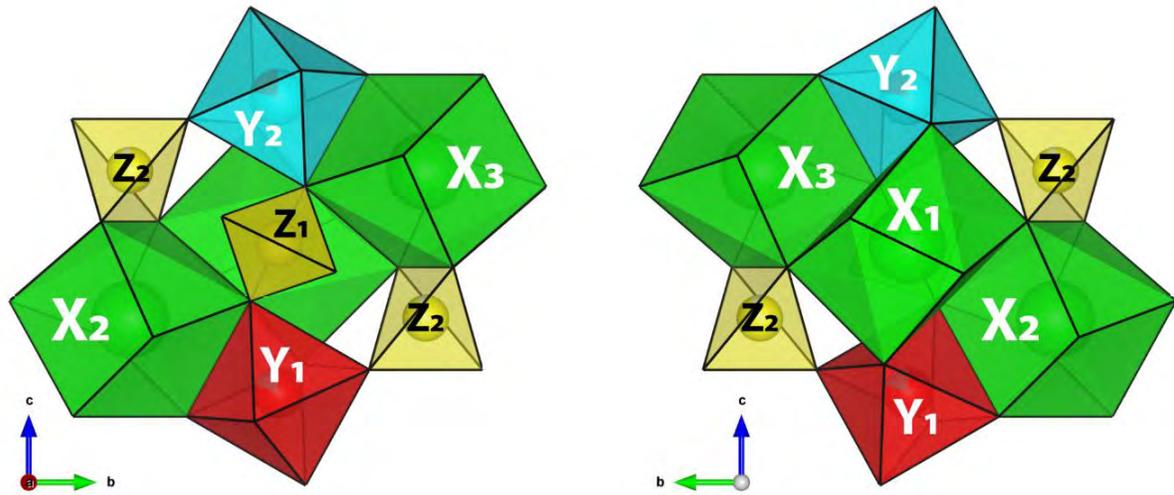
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780 Figure 4.

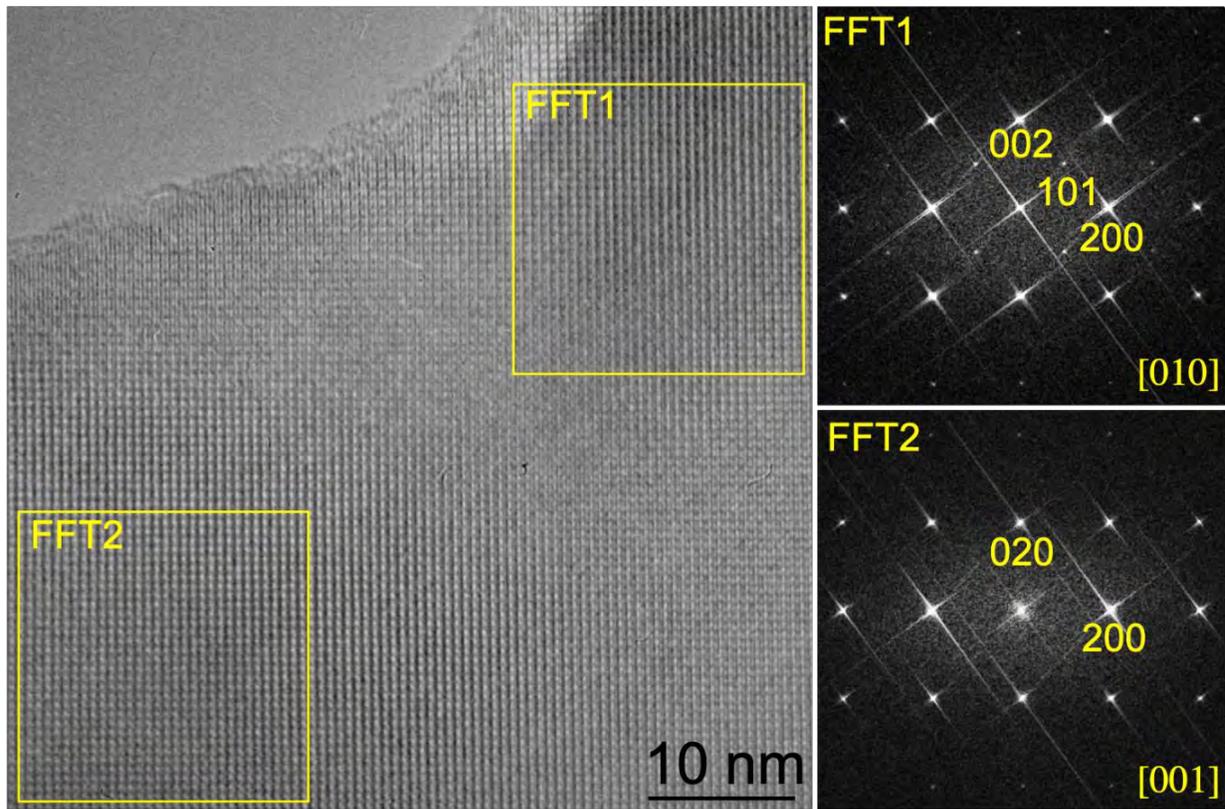
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783 Figure 5.

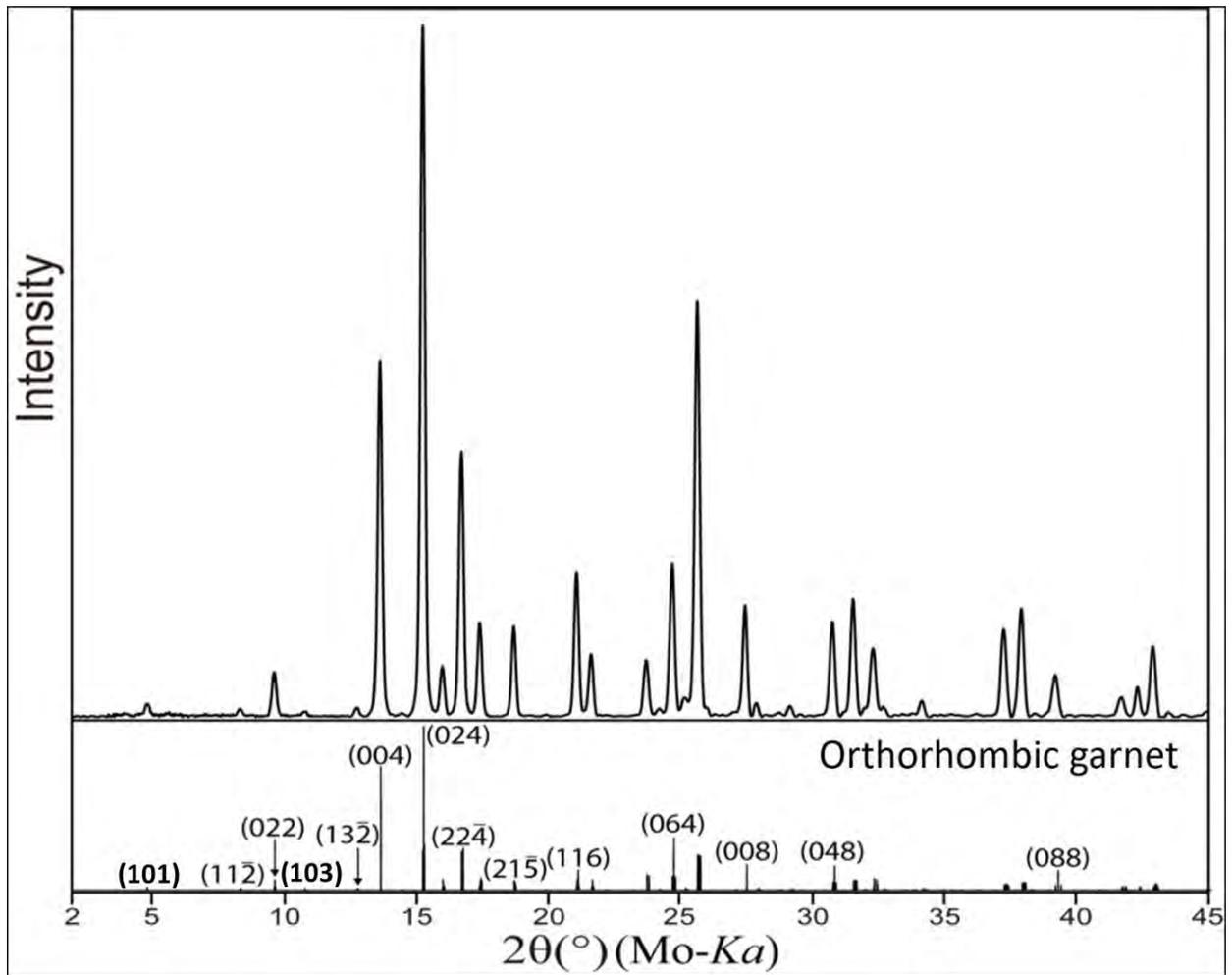
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786 Figure 6.

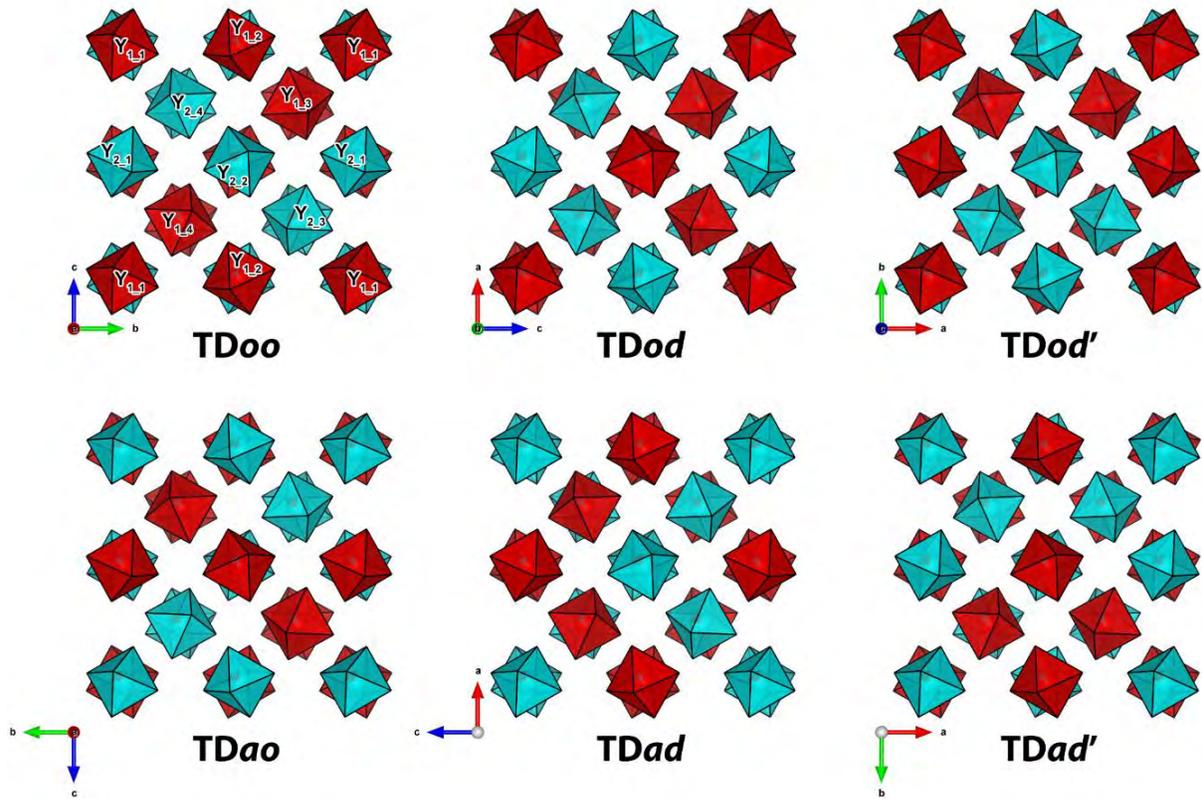
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789 Figure 7.

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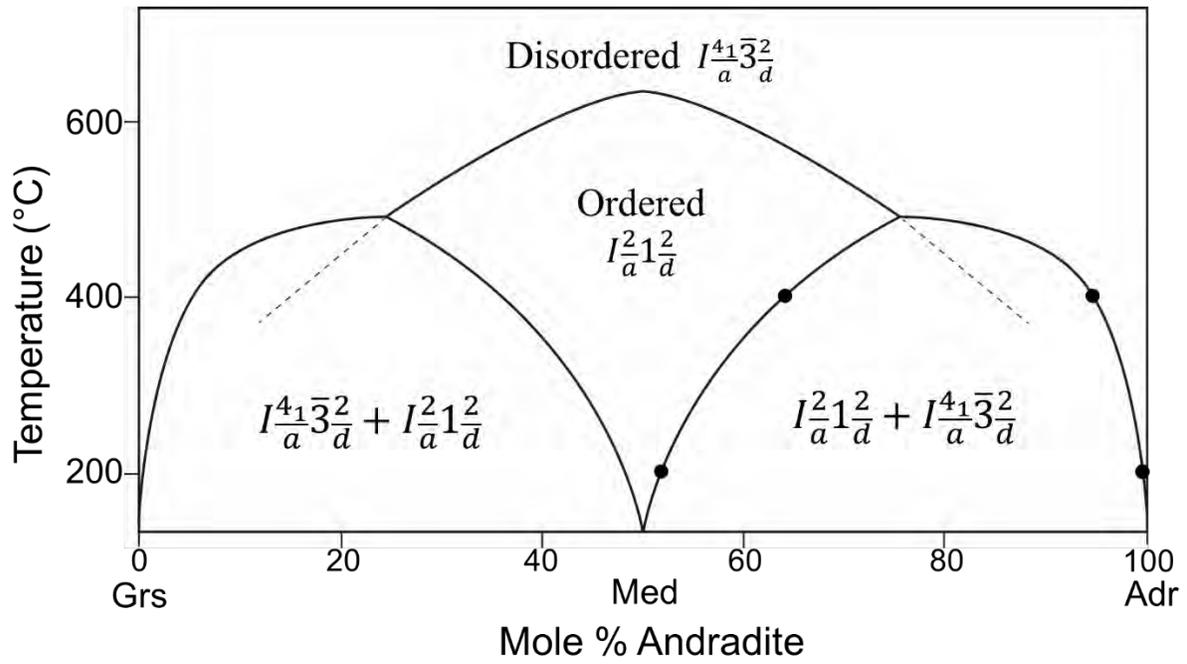
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792 Figure 8.

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797 Figure 9.

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## Tables

Table 1. Chemical data for orthorhombic garnet and andradite collected from areas in Figure. 2

Element Wt%	Orthorhombic Garnet							Andradite				Intermediate	
	1	2	3	4	5	6	Avg	1	2	3	Avg	1	2
Si	17.05	16.94	17.33	17.20	17.24	16.99	17.12	16.38	16.29	16.45	16.37	16.70	16.35
Al	5.33	5.37	5.34	5.27	5.16	5.32	5.30	0.01	0.00	0.02	0.01	2.62	2.35
Ca	24.43	24.20	24.31	24.33	24.30	24.35	24.32	23.15	23.19	23.09	23.14	24.00	24.12
Fe	13.08	13.05	12.88	12.84	13.26	12.98	13.01	22.09	22.15	22.33	22.19	17.64	17.89
Mn	0.23	0.26	0.33	0.28	0.29	0.29	0.28	0.13	0.15	0.11	0.13	0.20	0.29
O	39.39	39.54	39.56	39.61	39.41	39.55	39.51	37.48	37.66	37.54	37.56	38.65	38.71
F	0.46	0.57	0.71	0.53	0.43	0.67	0.56	0.24	0.24	0.38	0.29	0.30	0.38
Total	99.96	99.93	100.46	100.07	100.08	100.13	100.11	99.47	99.68	99.92	99.69	100.11	100.08
Site occupancy													
Ca <sup>2+</sup>	2.94	2.90	2.90	2.91	2.93	2.91	2.91	2.94	2.94	2.92	2.93	2.96	2.96
Fe <sup>2+</sup>	0.08	0.08	0.05	0.04	0.07	0.06	0.06	0.02	0.01	0.03	0.02	0.04	0.00
Mn <sup>2+</sup>	0.02	0.02	0.03	0.02	0.03	0.03	0.02	0.01	0.01	0.01	0.01	0.02	0.03
X <sub>Σ</sub>	3.05	3.00	2.98	2.97	3.02	2.99	3.00	2.97	2.96	2.96	2.97	3.01	2.99
Fe <sup>3+</sup>	1.05	1.05	1.05	1.06	1.08	1.06	1.06	2.00	2.00	2.00	2.00	1.52	1.57
Al <sup>3+</sup>	0.95	0.95	0.95	0.94	0.92	0.94	0.94	0.00	0.00	0.01	0.00	0.48	0.43
Y <sub>Σ</sub>	2.00	2.00	2.00	2.00	2.00	2.00	2.00	2.00	2.00	2.00	2.00	2.00	2.00
Si <sup>4+</sup>	2.93	2.89	2.95	2.93	2.96	2.90	2.93	2.97	2.94	2.97	2.96	2.94	2.86
□	0.07	0.11	0.05	0.07	0.04	0.11	0.07	0.03	0.06	0.03	0.04	0.07	0.14
Z <sub>Σ</sub>	3.00	3.00	3.00	3.00	3.00	3.00	3.00	3.00	3.00	3.00	3.00	3.00	3.00
O <sup>2-</sup>	11.81	11.57	11.75	11.68	11.89	11.55	11.71	11.83	11.69	11.81	11.78	11.76	11.44
F <sup>-</sup>	0.12	0.14	0.18	0.14	0.11	0.17	0.14	0.07	0.06	0.10	0.08	0.08	0.10
OH <sup>-</sup>	0.07	0.29	0.07	0.18	0.00	0.28	0.15	0.11	0.25	0.09	0.15	0.16	0.47
φ <sub>Σ</sub>	12.00	12.00	12.00	12.00	12.00	12.00	12.00	12.00	12.00	12.00	12.00	12.00	12.00

\*All the cations are normalized to 12 (O+F) and number for H (or OH) is calculated based on charge balance.

Table 2. Experimental details of X-ray diffraction and structure refinement.

Crystal data	
Chemical formula	$\text{Ca}_3(\text{Fe}_{0.804}\text{Al}_{0.196})(\text{Al}_{0.779}\text{Fe}_{0.221})(\text{Si}_{2.917}\square_{0.083})\text{O}_{12}$
Color	yellow-brownish green
Crystal size (mm)	0.09×0.09×0.06
Temperature (K)	298
Crystal System	Orthorhombic
Space Group	$I\frac{2}{a}1\frac{2}{a}(Fddd)$
<i>a, b, c</i> (Å)	11.966(3)
	11.966(3)
	11.964(3)
$\alpha, \beta, \gamma$ (°)	90
	90
	90.29(2)
<i>V</i> (Å <sup>3</sup> )	1713.0(7)
Data collection	
Exposure time (s/frame)	40
Scanning width (°)	0.5
Runs	3 $\omega$ +1 $\phi$
Total reflections	15579
Independent reflections	1318
observed [ <i>I</i> > 3 $\sigma$ ( <i>I</i> )] reflections	963
<i>R</i> <sub>int</sub>	0.081
$\theta$ values (°)	$\theta_{\text{max}} = 30.5,$
	$\theta_{\text{min}} = 2.4$
( <i>sin</i> $\theta$ / $\lambda$ ) max (Å <sup>-1</sup> )	0.715
Range of <i>h, k, l</i>	<i>h</i> = -17→16
	<i>k</i> = -17→17
	<i>l</i> = -17→15
Refinement	
<i>R</i> [ <i>F</i> <sup>2</sup> >3 $\sigma$ ( <i>F</i> <sup>2</sup> )]	0.0194
<i>wR</i> <sup>2</sup> [ <i>F</i> <sup>2</sup> >3 $\sigma$ ( <i>F</i> <sup>2</sup> )]	0.0638
<i>R</i> (all)	0.0268
<i>wR</i> <sup>2</sup> (all)	0.0668
GOF(obs)	1.05
GOF(all)	1.19
No. of parameters	100
No. of constraints	12
$\Delta\rho_{\text{max}}, \Delta\rho_{\text{min}}$ (eÅ <sup>-3</sup> )	0.31, -0.27

Table 3. Intensity statistics of the integrated peaks for different space groups using the absorption-corrected data (multi-scan method in Laue group  $\bar{1}$ ).

Space Group	Point Group	R <sub>int</sub> (obs/all)	# averaged	Redundancy	Systematic Extinction (obs/all)
$I\bar{1}$	$\bar{1}$	1.68/1.75	3421/5208	3.225	0/0
$I\frac{2}{a}$	$\frac{2}{m}$	1.82/1.89	1907/2735	6.141	3/451
$I11\frac{2}{d}(a+b)$	$\frac{2}{m}$	1.87/1.93	1894/2703	6.213	10/304
$I11\frac{2}{d}(a-b)$	$\frac{2}{m}$	1.90/1.97	1903/2704	6.211	2/243
$I\frac{2}{a}1\frac{2}{d}$	$\frac{2}{m}\frac{2}{m}\frac{2}{m}$	1.96/2.02	1069/1455	11.543	12/928
$I\frac{4_1}{a}1\frac{2}{d}$	$\frac{4}{m}\frac{2}{m}\frac{2}{m}$	6.71/6.74	653/792	21.206	435/1707
$I\frac{4_1}{a}\bar{3}\frac{2}{d}$	$\frac{4}{m}\bar{3}\frac{2}{m}$	7.38/7.40	277/297	56.549	745/2659

Table 4. Atom coordination and occupancies of the refined structure in the body-centered pseudo-cubic cell setting.

Site	Atom	Occ.	x	y	z	$U_{\text{equiv}}$	Multiplicity
X <sub>1</sub>	Ca	1	0.24962(3)	0.37526(3)	0.50068(2)	0.00760(10)	16
X <sub>2</sub>	Ca	1	0	0.25	0.125	0.00787(13)	4
X <sub>3</sub>	Ca	1	0	0.25	0.625	0.00772(13)	4
Y <sub>1</sub>	Fe	0.804(3)	0	0	0	0.00590(12)	8
	Al	0.196(3)					
Y <sub>2</sub>	Al	0.779(3)	0	0	0.5	0.00458(17)	8
	Fe	0.221(3)					
Si <sub>1</sub>	Si	0.987(2)	0.24871(3)	0.12523(3)	0.50188(3)	0.00558(14)	16
Si <sub>2</sub>	Si	0.936(2)	0	0.25	0.37632(4)	0.00469(18)	8
O <sub>1</sub>	O	1	0.65182(8)	0.03911(8)	0.04540(8)	0.0095(3)	16
O <sub>2</sub>	O	1	0.04825(8)	0.65426(8)	0.03938(8)	0.0101(3)	16
O <sub>3</sub>	O	1	0.03821(8)	0.04671(7)	0.65148(8)	0.0091(3)	16
O <sub>4</sub>	O	1	0.34486(7)	0.53948(8)	0.45162(8)	0.0092(3)	16
O <sub>5</sub>	O	1	0.04656(8)	0.34807(8)	0.46275(8)	0.0098(3)	16
O <sub>6</sub>	O	1	0.46078(8)	0.04677(8)	0.34455(8)	0.0093(3)	16

Table 5. Anisotropic temperature parameters of atoms in the refined structure.

		U <sup>11</sup>	U <sup>22</sup>	U <sup>33</sup>	U <sup>12</sup>	U <sup>13</sup>	U <sup>23</sup>
X <sub>1</sub>	Ca	0.00786(17)	0.00584(17)	0.00911(19)	0.00117(14)	0.00097(9)	-0.00002(9)
X <sub>2</sub>	Ca	0.00832(18)	0.00832(18)	0.0070(3)	0.0020(2)	0	0
X <sub>3</sub>	Ca	0.00825(18)	0.00825(18)	0.0066(3)	0.0002(2)	0	0
Y <sub>1</sub>	Fe	0.0056(2)	0.0055(2)	0.0066(2)	0.00108(13)	0.00002(10)	-0.00009(10)
Y <sub>2</sub>	Al	0.0042(3)	0.0045(3)	0.0050(3)	0.0012(2)	-0.00002(15)	-0.00048(15)
Si <sub>1</sub>	Si	0.0052(2)	0.0050(2)	0.0066(3)	0.00114(19)	0.00012(13)	0.00015(13)
Si <sub>2</sub>	Si	0.0048(3)	0.0049(3)	0.0044(4)	0.0009(2)	0	0
O <sub>1</sub>	O	0.0091(5)	0.0094(5)	0.0101(5)	0.0017(4)	0.0004(4)	-0.0008(4)
O <sub>2</sub>	O	0.0095(5)	0.0088(5)	0.0119(5)	0.0010(4)	-0.0005(4)	0.0011(4)
O <sub>3</sub>	O	0.0081(4)	0.0095(5)	0.0096(5)	0.0007(4)	-0.0005(4)	0.0004(4)
O <sub>4</sub>	O	0.0090(5)	0.0087(5)	0.0099(5)	0.0012(4)	0.0000(4)	-0.0009(4)
O <sub>5</sub>	O	0.0086(5)	0.0096(5)	0.0113(5)	0.0017(4)	0.0006(4)	0.0009(4)
O <sub>6</sub>	O	0.0092(5)	0.0089(4)	0.0099(5)	0.0004(3)	0.0000(4)	-0.0003(4)

Table 6. All the bond distances (Å) in the structure.

Site	oxygen site	bond distance	Site	oxygen site	bond distance
X <sub>1</sub> (Ca) (16)	O <sub>1</sub>	2.3500(14)	Y <sub>1</sub> (Fe) (8)	O <sub>2</sub>	1.9878(13)
	O <sub>1</sub>	2.5050(15)		O <sub>6</sub>	2.0011(13)
	O <sub>2</sub>	2.4894(15)		O <sub>4</sub>	2.0033(13)
	O <sub>3</sub>	2.3437(13)		average	1.9974(13)
	O <sub>4</sub>	2.3413(14)	Y <sub>2</sub> (Al) (8)	O <sub>1</sub>	1.9508(13)
	O <sub>4</sub>	2.4914(15)		O <sub>3</sub>	1.9503(13)
	O <sub>5</sub>	2.4915(13)		O <sub>5</sub>	1.9556(13)
	O <sub>6</sub>	2.3444(13)		average	1.9522(13)
	average	2.4196(14)	Si <sub>1</sub> (16)	O <sub>1</sub>	1.6474(12)
X <sub>2</sub> (Ca) (4)	O <sub>2</sub>	2.3496(13)		O <sub>3</sub>	1.6463(11)
	O <sub>6</sub>	2.5013(15)		O <sub>4</sub>	1.6425(12)
	average	2.4255(14)		O <sub>6</sub>	1.6428(11)
X <sub>3</sub> (Ca) (4)	O <sub>3</sub>	2.4975(15)	average	1.6448(11)	
	O <sub>5</sub>	2.3344(13)	Si <sub>2</sub> (8)	O <sub>2</sub>	1.6537(11)
	average	2.4160(14)		O <sub>5</sub>	1.6584(11)
Ca average	Ca average	2.4203(14)	average	1.6561(11)	