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#### Abstract

We show, by single crystal diffraction studies in laser-heated diamond anvil cells, that Ca<sub>2</sub>CO<sub>4</sub> 24 orthocarbonate, which contains  $CO_4^{4-}$  tetrahedra, can be formed already at ~20 GPa at ~1830 K. 25 i.e. at much lower pressures than other carbonates with  $sp^3$ -hybridized carbon. Ca<sub>2</sub>CO<sub>4</sub> can also 26 be formed at ~89 GPa and ~2500 K. This very broad p, T-range suggests the possible existence of 27 Ca<sub>2</sub>CO<sub>4</sub> in the Earth's transition zone and in most of the lower mantle. Raman spectroscopy 28 shows the typical bands associated with tetrahedral  $CO_4^{4-}$ -groups. DFT-theory based calculations 29 reproduce the experimental Raman spectra and indicate that at least in the athermal limit the 30 phase assemblage of  $Ca_2CO_4 + 2SiO_2$  is more stable than  $2CaSiO_3 + CO_2$  at high pressures. 31

Keywords: carbonate, Ca<sub>2</sub>CO<sub>4</sub>, structure, x-ray diffraction, Raman spectroscopy, density
 functional theory

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#### Introduction

Carbonates play a crucial role in the long-term global carbon cycle as they contain  $\approx 60 - 70\%$  of 36 the carbon present on Earth's surface and in its crust (Hirschmann, 2010; McKenzie et al., 2016; 37 Ridgwell, 2005; Wallmann, 2001). As carbonates are partly incorporated into oceanic 38 lithosphere, they may be transported into the deep mantle via subduction (Clift, 2017; Kelemen 39 and Manning 2015; McKenzie et al., 2016). Models propose a carbon influx on the order of 40 several tenth-megatons per year, which is mainly due to the contribution of carbonate sediments 41 and altered carbonaceous oceanic crust (Hirschmann, 2010; Kelemen and Manning, 2015). 42 43 Within the last two decades, several studies showed that all major carbonates, such as  $CaCO_3$ (Bayarjargal et al., 2018), MgCO<sub>3</sub> (Binck et al., 2020b), FeCO<sub>3</sub> (Cerantola et al., 2017) and 44

 $CaMg(CO_3)_2$  (Binck et al., 2020a; Merlini et al., 2017), may be stable at the p, T conditions of the 45 Earth's mantle, while displaying a variety of different high p, T structures (e.g. Binck et al., 46 2020a; Cerantola et al., 2017; Chariton et al., 2020; Merlini et al., 2017; Gavryushkin et al., 2017; 47 Ono et al., 2007). It has been proposed that carbonates in contact with mantle silicates and metals 48 in the deep Earth are reduced and, depending on the stage of reduction, form metal carbides, or 49 diamonds (Palyanov et al., 2013; Rohrbach et al., 2011; Stagno et al., 2012, 2013). However, 50 51 carbonate inclusions in diamonds originating from the deep mantle offer direct evidence for the presence of oxidized carbon in the form of carbonates in at least some regions of the mantle 52 (Brenker et al., 2007; Kaminsky et al., 2016). Cold oxidized subducting slabs with low reaction 53 kinetics are believed to provide suitable conditions for hosting carbonates even at depths of the 54 lower mantle (Maeda et al., 2017; Martirosyan et al., 2016; Walter et al., 2011). 55

A recent remarkable discovery are carbonates, in which  $sp^3$ -hybridization of carbon leads to the 56 formation of  $CO_4^{4-}$  tetrahedra instead of triangular  $sp^2$ -hybridized  $CO_3^{3-}$ -groups (Binck et al., 57 2020b; Boulard et al., 2011, 2012, 2015; Cerantola et al., 2017; Merlini et al., 2015, 2017; 58 Lobanov et al., 2017). It is now of great interest to determine whether carbonates with  $sp^3$ -59 hybridized carbon form solid solutions with their silicate analogs and thus provide an alternative 60 major host of carbon in the deeper mantle regions. Up to now, carbonates with CO<sub>4</sub>-groups were 61 synthesized at pressures >70 GPa, which led to the conclusion that only the deep lower mantle 62 may provide the required thermodynamic conditions for their occurrence (Binck et al., 2020b; 63 64 Boulard et al., 2011, 2012, 2015; Cerantola et al., 2017; Merlini et al., 2015, 2017; Lobanov et al., 2017). 65

Recent theoretical studies, however, have indicated that carbonate polymorphs in the system
CaO-CO<sub>2</sub> which contain CO<sub>4</sub>-groups may be stable at pressures and temperatures corresponding

68	to the conditions of Earth's transition zone and uppermost lower mantle (Sagatova et al., 2020;
69	Yao et al., 2018). Specifically, Yao et al., (2018) predicted that Ca <sub>3</sub> CO <sub>5</sub> -Cmcm and CaC <sub>2</sub> O <sub>5</sub> -Pc
70	may be stable phases at pressures >11 and >33 GPa respectively, while Sagatova et al., (2020)
71	found $Ca_2CO_4$ - <i>Pnma</i> to be a stable phase at pressures >13 GPa.
72	While the phase diagram of CaCO <sub>3</sub> is fairly well constrained for the pressure and temperature
73	conditions of Earth's upper and lower mantle (Bayarjargal et al., 2018; Gavryushkin et al., 2017;
74	Ishizawa et al., 2013; Lobanov et al., 2017; Ono et al., 2007), experimental studies focusing on
75	different compositions in the system CaO-CO <sub>2</sub> have not been reported up to now. In this study,
76	we have used a multidisciplinary experimental and theoretical approach, which allowed us to
77	verify the formation of calcium orthocarbonate ( $Ca_2CO_4$ - <i>Pnma</i> ) at <i>p</i> , <i>T</i> -conditions of the Earth's
78	transition zone and lower mantle. We report the experimentally determined crystal structure of
79	calcium orthocarbonate (Ca <sub>2</sub> CO <sub>4</sub> -Pnma) and present the first experimental and theoretical Raman
80	spectra of the new compound.

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#### **Experimental Methods**

## 83 Preparation of high-pressure, high-temperature experiments

For low pressure and moderate temperature Raman spectroscopy and x-ray diffraction experiments (up to ~23 GPa and ~2300 K), Boehler-Almax type (Boehler, 2006) diamond anvil cells (DAC) were employed. Raman-compatible diamonds with low birefringence, ultra-low fluorescence and culets of 300-350  $\mu$ m in diameter were inserted in WC seats. The opening angles of the cells were 48° and 70°, respectively. A BX-90 type DAC (Kantor et al., 2012) was used for high-pressure, high-temperature x-ray diffraction experiments (up to ~93 GPa and ~2500 K), with a 120  $\mu$ m culet diamond inserted in a WC seat. Depending on the culet size, sample

chambers of 60 - 175  $\mu$ m in diameter were laser drilled in Re gaskets pre-indented to ~15 - 45 µm. The diamond anvil cells for the experiments at moderate pressures were loaded with natural CaCO<sub>3</sub> single crystals, surrounded by compacted CaO powder (analytical grade chemPUR 94 99.95%) and ruby pressure markers. The CaCO<sub>3</sub> single crystals had edge lengths ranging from ~15 - 60 µm. Before loading, CaO and CaCO<sub>3</sub> were dried for 48 h at 460°C, in order to remove 96 surface moisture. The DAC employed for the high-pressure experiments was loaded with calcium 97 azide (CaN<sub>6</sub>) containing CaO and a nitrogen pressure transmitting medium.

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#### 99 Raman spectroscopy in the LH-DAC

Raman spectra were measured at the Institute of Geosciences at the Goethe Universität Frankfurt 100 101 in 0.5 - 4 GPa steps upon compression and decompression covering a range between ambient pressure and ~23 GPa. A frequency doubled 532.14 nm Nd:YAG Oxxius laser (LCX 532S) was 102 focused on the sample with a spot size of 6 µm. Spectra were collected in backscattering 103 geometry, using a grating spectrometer (Acton, SP-2356) equipped with a CCD detector (Pixis 104 256E) and a microscope objective (Mitutoyo). The spectral resolution of the spectrometer is 3 105 cm<sup>-1</sup> (Bayarjargal et al., 2018). The laser power was set to 430 mW and spectra were collected for 106 50 s in a frequency window of 100 - 1500 cm<sup>-1</sup>, using a grating of 1800 grooves/mm. The 107 estimated laser power on the sample was around ~250 mW. The pressure was determined before 108 109 and after the Raman measurement, using the ruby reference scales for non-hydrostatic (Mao et al., 1978) conditions. The accuracy of our pressure determination by the ruby scale was  $\sim 2$  GPa, 110 while pressure gradients may have caused uncertainties up to 3 GPa during and after laser 111 heating. 112

The sample was heated from both sides with a pulsed CO<sub>2</sub>-laser (Diamond K-250 from Coherent, 113  $\lambda = 10.6 \,\mu\text{m}$ ), reaching temperatures up to ~2300 K. For the determination we used the same set-114 up as for the Raman measurements, while the grating was set to 150 grooves/mm. In order to 115 achieve coupling of the CO<sub>2</sub>-laser from both sides of the sample, the laser power was set in a 116 range between 1 - 6 W, depending on the pressure and the loading of the diamond anvil cell. The 117 heating laser was focused on the sample so that the diameter of the heated area was around 25 118 um, which nearly covered all of the single crystals. We moved the heating laser across the 119 sample, while heating for about 5 min per position. The position of the Raman laser with respect 120 to the heated areas on the sample was controlled using an optical camera. The thermal emission 121 of the sample, as well as the Raman signal was measured with a spatial resolution of around 5 - 6 122  $\mu$ m, i.e. the areas for the measurements were significantly smaller than the heating spots. The 123 temperatures during laser heating were determined by the two-color pyrometer method, 124 employing Planck and Wien fits (Benedetti et al., 2004). We assume a typical uncertainty 125 associated with radiometric temperature measurements in LH-DACs of ~10 %. 126

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#### 128 High-pressure single-crystal x-ray diffraction

High pressure single-crystal x-ray diffraction (SC-XRD) data were acquired at the P02.2 and 13-IDD beamlines of PETRA III (DESY, Hamburg, Germany) and the Advanced Photon Source (APS, Chicago, USA), respectively. Two different phase assemblages were compressed up to ~20 and ~89 GPa at ambient temperature. The samples were laser-heated in a temperature range between ~1830 and ~2500 K once the cells had reached the desired pressures. The sample material of the low-pressure cell was heated using the CO<sub>2</sub>-laser heating system described above. The high-pressure cell was heated using the double-sided YAG laser-heating system at the 13-

136 IDD beamline of the APS, with  $CaN_6$  acting as the laser-absorber. Temperatures were measured 137 using the thermal emission (Mezouar et al., 2017). The pressures inside the sample chambers 138 were either determined using the ruby fluorescence method (Mao et al., 1978), or from the 139 vibrational mode of the stressed diamond anvil (Akahama and Kawamura, 2006).

The synthesized polycrystalline samples were characterized upon compression and 140 decompression by SC-XRD measurements. At the P02.2 beamline, a Perkin Elmer XRD 1621 141 detector was employed with an x-ray beam ( $\lambda = 0.2901$  Å, or  $\lambda = 0.2887$  Å, depending on the 142 experiment) focused down to about 2 x 2  $\mu$ m<sup>2</sup>. At the GSECARS beamline, a Pilatus CdTe 1M 143 detector was used along with an x-ray beam ( $\lambda = 0.2952$  Å) focused down to 3 x 3  $\mu$ m<sup>2</sup>. On the 144 polycrystalline samples, a full x-ray diffraction mapping of the sample chamber was performed 145 after laser-heating in order to identify the most promising sample positions for a single-crystal 146 data collection. On the locations where the most intense single-crystal reflections were detected, 147 single-crystal data were acquired in step-scans of  $0.5^{\circ}$  from  $36^{\circ}$  to  $+36^{\circ}$   $\omega$  and 5 s exposure time. 148 The CrysAlisPro software (Rigaku Oxford Diffraction, 2018) was utilized for the single-crystal 149 data analysis. To calibrate the instrumental model in the CrysAlisPro software, i.e., the sample 150 to-detector distance, detector's origin, offsets of goniometer angles, and rotation of both x-ray 151 beam and the detector around the instrument axis, a single crystal of orthoenstatite 152  $((Mg_{1.93}Fe_{0.06})(Si_{1.93}, Al_{0.06})O_6, Pbca \text{ space group, } a = 8.8117(2) \text{ Å}, b = 5.18320(10) \text{ Å}, and c =$ 153 18.2391(3) Å) was used. The same calibration crystal was used at all beamlines. The analysis 154 155 procedure in the CrysAlisPro software includes the peak search, the removal of the diamond anvils' parasitic reflections and saturated pixels of the detector, finding reflections belonging to a 156 unique single crystal, the unit cell determination and the data integration and absorption 157 158 corrections. The crystal structures were then solved with SHELXT structure solution program

(Sheldrick, 2015) using intrinsic phasing and refined within the JANA2006 software (Petříček et 159 al., 2014). CSD 2026976 contains the crystallographic data for Ca<sub>2</sub>CO<sub>4</sub> at 89 GPa (Laniel, 2020). 160 FIZ These data be obtained free of charge from Karlsruhe 161 can via www.ccdc.cam.ac.uk/structures. Further structural data of Ca<sub>2</sub>CO<sub>4</sub> at 20.1(2) GPa and 89.0(8) 162 GPa are given in the associated CIF. 163

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# 165 **Density functional theory**

In order to obtain theoretical Raman spectra, density functional perturbation theory (DFPT) 166 167 calculations were performed employing the CASTEP code (Clark et al., 2005). The code is an implementation of Kohn-Sham DFT based on a plane wave basis set in conjunction with 168 pseudopotentials. The plane wave basis set allows to achieve numerically converged results in a 169 straightforward manner, as the convergence is controlled by a single adjustable parameter, the 170 plane wave cut-off, which we set to 1020 eV. The norm-conserving pseudopotentials were 171 generated 'on the fly' from the information provided in the CASTEP data base. These 172 pseudopotentials have been tested extensively for accuracy and transferability (Lejaeghere et al., 173 2016). All calculations employed the GGA-PBE exchange-correlation functional (Perdew et al., 174 1996). The Brillouin zone integrals were performed using Monkhorst-Pack grids (Monkhorst and 175 Pack, 1976) with spacings between grid points of less than 0.037 Å<sup>-1</sup>. Geometry optimizations 176 were defined as being converged when the energy change between iterations was  $<0.5 \times 10^{-6}$ 177 eV/atom, the maximal residual force was < 0.01 eV/Å, and the maximal residual stress was <178 0.02 GPa. Phonon frequencies were obtained from density functional perturbation theory (DFPT) 179 calculations. Raman intensities were computed using DFPT in the 2n + 1 theorem approach 180 (Miwa, 2011) for a powder sample. We have shown in numerous studies that the relative 181

intensities of Raman bands for carbonates are very well reproduced and predicted with this
approach (Nguyen-Thanh et al., 2016; Biedermann et al., 2017; Bayarjargal et al., 2018; Binck et
al., 2020a; Binck et al. 2020b). Reaction enthalpies were computed in the athermal limit by
comparing the sum of the enthalpies of the reactants to those of the products at prescribed
pressures.

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## Results

# **189** Crystal structure of Ca<sub>2</sub>CO<sub>4</sub>-*Pnma*

Our single-crystal diffraction measurements unequivocally show the appearance of a new set of diffraction spots upon laser-heating, which was indexed with an orthorhombic unit cell (space group *Pnma*). The structure was readily solved and refined for data sets obtained at 20.1(2) and 89.0(8) GPa (Supplementary<sup>1</sup> data). All other data sets between 10 and 93 GPa were successfully indexed with the same unit cell (Supplementary<sup>1</sup> data).

195  $Ca_2CO_4$ -*Pnma* has 28 atoms (Z = 4 formula units) in the unit cell and is characterized by carbon 196 that is coordinated by oxygen fourfold, forming isolated CO<sub>4</sub> tetrahedra. Ca atoms occupy two 197 non-equivalent crystallographic positions and form CaO<sub>9</sub> and CaO<sub>11</sub> polyhedra (Fig. 1). 198 Geometric parameters of the crystal structure at 20.1(2) and 89.0(8) GPa are listed in Tab. 1.

The CO<sub>4</sub> tetrahedra share corners, edges and faces with the CaO<sub>11</sub> polyhedra, but only share corners and edges with the CaO<sub>9</sub> polyhedra. The CaO<sub>9</sub> polyhedra are irregularly shaped, while the CaO<sub>11</sub> polyhedra form fivefold-capped trigonal prisms (Sagatova et al., 2020). CaO<sub>9</sub> and CaO<sub>11</sub> polyhedra are connected via their faces. Upon compression from 20.1(2) to 89.0(8) GPa, the overall unit cell volume decreases by ~20 %. The C-O bond lengths range from 1.378(14) to 1.386(7) Å at 20.1(2) GPa. They shorten by ~3 % when a pressure of 89.0(8) GPa is reached (C-

O bond range 1.323(16) - 1.381(11) Å). The volume of the CO<sub>4</sub> tetrahedra decreases by ~9 % 205 from 20.1(2) GPa ( $V = 1.35 \text{ Å}^3$ ) to 89.0(8) GPa ( $V = 1.23 \text{ Å}^3$ ). The distortion of a coordination 206 polyhedron can be defined by an index D, with  $D = 1/n \sum_{i=1}^{n} \frac{|li - lav|}{|lav|}$ , where  $l_i$  is the distance from 207 the central atom to the *i*th coordinating atom and  $l_{av}$  is the average bond length (Baur, 1974). The 208 distortion of the CO<sub>4</sub> polyhedra at 20.1(2) GPa is small with D = 0.002, and increases only 209 slightly to 0.015 upon pressure increase to 89.0(8) GPa. The average bond lengths of the CaO<sub>9</sub> 210 and CaO<sub>11</sub> coordination polyhedra decrease from 2.360 to 2.174 Å and 2.532 to 2.338 Å over the 211 same pressure range, corresponding to a pressure-induced volume decrease of 20.8 and 20.5 %, 212 respectively. 213

In summary, our SC-XRD experiments demonstrated that  $CaO + CaCO_3$  reacts at pressures as low as ~20 GPa and ~1830 K to form  $Ca_2CO_4$ . Further, the CaO in CaN<sub>6</sub> reacted with the diamond culet at ~89 GPa and ~2500 K to form  $Ca_2CO_4$ . The new phase formed in the experiments is a calcium orthocarbonate with a structure that has recently been predicted (Sagatova et al., 2020).

We calculated structural parameters for Ca<sub>2</sub>CO<sub>4</sub>-Pnma for pressures ranging from 0 to 100 GPa 219 using density functional theory calculations. Our Mulliken population analysis of the C-O bonds 220 221 clearly shows that all four C-O bonds in a CO<sub>4</sub> tetrahedron are similar, i.e. all four bonds have high populations of ~0.65  $e^{-1}$ Å<sup>3</sup>, indicative of covalent bonding. The calculated volume data were 222 fitted using a third-order Birch-Murnaghan equation of state and are shown alongside 223 experimental data (Fig. 2a). The extrapolated unit cell volume at ambient pressure is 302.0(3) Å<sup>3</sup> 224 and the bulk modulus and its first pressure derivative are  $K_0 = 108(1)$  GPa and  $K'_0 = 4.43(3)$ , 225 respectively. The experimentally determined unit cell dimensions closely resemble those obtained 226 by DFT calculations derived in this work and by Sagatova et al., (2020) (Fig. 2a and b). The 227

DFT-calculated volumes of the CO<sub>4</sub>, CaO<sub>9</sub> and CaO<sub>11</sub> polyhedra were further fitted using thirdorder Birch-Murnaghan equations of state and are shown together with the experimental volumes of the solved crystal structures (Fig. 3 and Supplementary<sup>1</sup> data). The bulk modulus of the CO<sub>4</sub> groups is 360(38) GPa with  $K'_0 = 6.70(12)$  and  $V_0 = 1.433(5)$  Å<sup>3</sup>. For the CaO<sub>9</sub> and CaO<sub>11</sub> polyhedra, the bulk moduli are 99(3) GPa and 106(3) GPa with  $K'_0 = 4.30(8)$  and  $V_0 = 29.32(8)$ , as well as  $K'_0 = 4.31(9)$  and 40.09(11) Å<sup>3</sup>, respectively.

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## 235 Experimental and theoretical Raman spectra of Ca<sub>2</sub>CO<sub>4</sub>-*Pnma*

Raman spectra of CaO + CaCO<sub>3</sub> were measured upon cold compression up to 22.8(2) GPa. 236 237 Raman signals of a sequence of meta-stable high-pressure polymorphs are expected when calcite is used as CaCO<sub>3</sub> precursor instead of aragonite (Bayarjargal et al., 2018; Koch-Müller et al., 238 2016). CaO- $Fm\bar{3}m$ , on the other hand, is known to be Raman inactive and should not contribute 239 to the measured signal. We obtained CaCO<sub>3</sub>-III at around ~6 GPa, which eventually transformed 240 to CaCO<sub>3</sub>-VI at above ~15 GPa (Supplementary<sup>1</sup> data). We increased the pressure to ~23 GPa 241 242 before laser heating. When heating the CaCO<sub>3</sub>-VI single crystals to ~1830 and to ~2255 K in two distinct experiments, they reacted immediately with the surrounding CaO- $Fm\bar{3}m$ . In both runs, 243 the pressure after heating dropped by approximately 1.6 - 2.7 GPa. Raman spectra that have been 244 obtained after quenching the sample to ambient temperatures show well-resolved new features 245 (Fig. 4), which cannot be explained by the Raman spectra of CaCO<sub>3</sub>-VI or of aragonite 246 (Supplementary<sup>1</sup> data). At 19.9(4) GPa, the most prominent feature of the newly synthesized 247 phase is a strong Raman band at 996.4 cm<sup>-1</sup> (Fig. 4). Furthermore, at least eight new Raman 248 bands are observable in a range of 557-774 cm<sup>-1</sup>, and more than twelve new modes appear in the 249 low frequency range between 121-399 cm<sup>-1</sup>. We observe CaCO<sub>3</sub>-VI in the non-heated areas, 250

which demonstrates the localized increase of temperature upon laser heating. In both runs, we were not able to find any evidence for the presence of aragonite, suggesting a strong reactivity in the CaO-CaCO<sub>3</sub> system if sufficient energy is provided.

254 According to group theory, forty-two Raman active modes are expected for Ca<sub>2</sub>CO<sub>4</sub>-Pnma:  $\Gamma_{\text{Raman}} = 13A_{\text{g}} + 8B_{1\text{g}} + 13B_{2\text{g}} + 8B_{3\text{g}}$ . The most prominent characteristic of Ca<sub>2</sub>CO<sub>4</sub>-*Pnma* is an intra-255 molecular Raman band that is appearing around 1000 cm<sup>-1</sup>. Raman bands in the same frequency 256 interval have been shown to be distinctive of the intra-molecular vibrations of the CO<sub>4</sub> tetrahedra 257 in CaCO<sub>3</sub>-P2<sub>1</sub>/c (Lobanov et al., 2017) and MgCO<sub>3</sub>-C2/m (Binck et al., 2020b). Fig. 5 258 demonstrates a good match between the theoretical and experimental Raman modes indicating 259 that Ca<sub>2</sub>CO<sub>4</sub>-*Pnma* is the only carbonate with  $sp^3$ -hybridized carbon present in the sample. We 260 observe at least twenty-three experimental Raman modes which can be assigned to the theoretical 261 spectrum at 20 GPa (Fig. 5). Missing Raman bands are due to the resolution of the spectrometer 262 or high background of the diamonds. A calculated Raman spectrum of Ca<sub>2</sub>CO<sub>4</sub>-Pnma at 83 GPa 263 implies positive Grüneisen-parameters for all modes (Supplementary<sup>1</sup> data). According to 264 previous calculations, Ca<sub>2</sub>CO<sub>4</sub>-Pnma is thought to decompose below 10 GPa (Sagatova et al., 265 2020). We were able to acquire Raman spectra of Ca<sub>2</sub>CO<sub>4</sub>.Pnma down to 4.1(2) GPa. Below that 266 pressure the Raman spectra do not exhibit any of the characterizing features of Ca<sub>2</sub>CO<sub>4</sub> anymore 267 (Supplementary<sup>1</sup> data). Instead, at ambient conditions, a diffuse spectrum clearly showing the 268 lattice vibrations of the CO<sub>3</sub>-stretching mode is present (Supplementary<sup>1</sup> data). Since the 269 270 distinctive features of crystalline carbonate are absent, we infer that its amorphization has taken place due to the cold pressure relaxation. As the decomposition may require an activation energy, 271 the actual stability field of Ca<sub>2</sub>CO<sub>4</sub>-Pnma currently remains unknown. 272

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#### **Discussion and Implications**

Our results demonstrate that calcium orthocarbonate (Ca<sub>2</sub>CO<sub>4</sub>-Pnma) can be formed at the p, T-275 conditions of Earth's upper transition zone at ~20 GPa and ~1830 K and can persist to at least the 276 mid-lower mantle conditions (up to  $\sim$ 93 GPa and  $\sim$ 2500 K). During the investigation of the CaO-277 CaCO<sub>3</sub> system, we neither found indications for the previously predicted Ca<sub>3</sub>CO<sub>5</sub>-Cmcm phase 278 and its high-pressure polymorphs (Sagatova et al., 2020; Yao et al., 2018), nor for the predicted 279 meta-stable Ca<sub>2</sub>CO<sub>4</sub>- $P2_1/m$  phase (Yao et al., 2018). 280 Our constraints on the formation conditions of Ca<sub>2</sub>CO<sub>4</sub>-Pnma imply that some compounds in the 281 CaO-CO<sub>2</sub> system may have comparatively low  $sp^2$ - $sp^3$  transition pressures, corresponding to 282 lower upper mantle conditions. This stands in strong contrast to the  $sp^2$ - $sp^3$  crossover of pure 283 CaCO<sub>3</sub>, which was observed at pressures >100 GPa (Lobanov et al., 2017; Oganov et al., 2006; 284 Ono et al., 2007; Pickard and Needs, 2015). Although the solubility of carbon into major mantle 285 minerals has been shown to be low (Shcheka et al., 2006), the  $sp^2-sp^3$  transition in carbonates 286 287 may have a significant impact on the silicate-carbonate equilibria and thus the potential for

comparatively low pressures with large-volume presses.

A topological analogy of Ca<sub>2</sub>CO<sub>4</sub>-*Pnma* with the silicates larnite  $\beta$ -Ca<sub>2</sub>SiO<sub>4</sub> (Barbier et al., 1985) and flamite  $\alpha'_H$ -Ca<sub>2</sub>SiO<sub>4</sub> (Rashchenko et al., 2019) has been reported (Sagatova et al., 2020). Carbon substitution of SiO<sub>4</sub>-groups by CO<sub>4</sub>-groups is, however, unlikely as the former are more voluminous by a factor of ~2 than the latter (e.g. Hugh-Jones et al., 1997; Kudo and Takéuchi, 1985; Milman et al., 2001; Smyth and Bish, 1988 and references therein). Also, the compressibility of isolated CO<sub>4</sub> tetrahedra ( $K_0$  ~360 GPa) is smaller than that of SiO<sub>4</sub> tetrahedra ( $K_0 \sim 300$  GPa) which, based on crystal chemical considerations is the expected behavior due to

carbonates to exist in the mantle. Such effects can now be studied in the CaO-CO<sub>2</sub> system at

the smaller cation radius of C. However, it is now worthwhile to explore the phase stabilities of compounds in which  $CO_4$  groups and SiO<sub>6</sub>-octahedra may coexist.

We investigated whether  $sp^3$ -hybridized calcium orthocarbonate can coexist with silicates in the 299 mantle. While at the p, T-conditions of the transition zone and upper lower mantle,  $CaCO_3 + SiO_2$ 300 were shown to form  $CaSiO_3 + CO_2$  (Li et al., 2018), our DFT-calculations at 20, 30 and 40 GPa 301 show, that  $Ca_2CO_4 + 2SiO_2$  are more stable than  $2CaSiO_3 + CO_2$  (Fig. 6). This implies that 302 Ca<sub>2</sub>CO<sub>4</sub>-Pnma could exist together with silica at the p, T-conditions of the transition zone and 303 upper lower mantle. In fact, the presence of CaCO<sub>3</sub> inclusions in super deep diamonds derived 304 305 from the transition zone or the lower mantle (Brenker et al., 2007; Kaminsky et al., 2016) is inconsistent with the assumption that CaCO<sub>3</sub> will always react with the surrounding mantle. 306

Isotopic signatures (Harte, 2010; Tappert et al., 2005) and carbonate inclusions in diamonds 307 (Brenker et al., 2007; Kaminsky et al., 2016) suggest that diamonds may form during 'redox 308 freezing', i.e. when the oxidized carbonate-bearing subducting slab has reached the more reduced 309 310 transition zone or the lower mantle, and releases carbonate and/or carbonatitic melts which reduce oxidized carbon to diamond (Rohrbach et al., 2011; Stagno et al., 2013). The inverse 311 scenario called 'redox melting' occurs when reduced carbon-bearing rocks are lifted up above the 312 313 660 km discontinuity of the transition zone due to the upwelling mantle (Rohrbach et al., 2011; 314 Stagno et al., 2013). In this case, carbonate and/or carbonatitic melts form due to re-oxidation of diamond. Our results show that Ca<sub>2</sub>CO<sub>4</sub>-Pnma may form during such redox processes. Diamonds 315 that have formed during redox melting could initially conserve calcium orthocarbonate. These 316 orthocarbonates would transform or decompose once the diamond is elevated above the base of 317 thick continental crust and could then only be identified by the decomposition products. 318 Indications of the presence of calcium-bearing inclusions such as CaO, CaF<sub>2</sub> and Ca(OH)<sub>2</sub> in 319

diamond have been found in the presence of  $CO_3$ -groups (Dobrzhinetskaya et al., 2007). Such a phase assemblage would be consistent with the expected remnants of decomposed  $Ca_2CO_4$ -*Pnma*.

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- 523

## 524 Figure captions

**Figure 1.** Structure of calcium orthocarbonate ( $Ca_2CO_4$ -*Pnma*) at ~20 GPa. Carbon atoms appear to no longer be coordinated by oxygen threefold, but form isolated CO<sub>4</sub> tetrahedra. The Ca atoms occupy two different sites, and they form CaO<sub>9</sub> (orange) and CaO<sub>11</sub> (green) polyhedra.

528

Figure 2. a) Pressure-dependence of the unit-cell volume of Ca<sub>2</sub>CO<sub>4</sub>-*Pnma*. The DFT data were 529 fitted using a third-order Birch-Murnaghan EOS (Gonzalez-Platas et al., 2016; Birch, 1947) using 530 the EoS-FIT7-GUI program (Gonzalez-Platas et al., 2016). b) Linear fits (dashed lines) to the 531 532 pressure dependence of DFT lattice parameters. Colors correspond to the a (red), b (green) and c (blue) lattice parameters. Filled and open squares correspond to experimental data obtained upon 533 compression and decompression, respectively. Filled triangles correspond to DFT-calculated data 534 points. Stars correspond to the predicted lattice parameters of Ca<sub>2</sub>CO<sub>4</sub>-Pnma at 25 GPa as 535 obtained by Sagatova et al., (2020). 536

537

541

**Figure 4.** Experimental Raman spectra obtained in the CaO-CaCO<sub>3</sub> system. At ambient conditions, only CaCO<sub>3</sub> with calcite structure ( $R\overline{3}c$ ) is Raman-active, which upon cold

<sup>Figure 3. Pressure-dependence of the CO<sub>4</sub> tetrahedron volume of Ca<sub>2</sub>CO<sub>4</sub>-</sup>*Pnma*. The DFT data
were fitted using a third-order Birch-Murnaghan EOS (Gonzalez-Platas et al., 2016; Birch, 1947)
using the *EoS-FIT7-GUI* program (Gonzalez-Platas et al., 2016).

compression to 21.5(2) GPa has the CaCO<sub>3</sub>-VI structure ( $P\overline{1}$ ). After heating to ~2255 K and quenching to ambient temperature, CaCO<sub>3</sub>-VI has reacted with CaO, forming calcium orthocarbonate Ca<sub>2</sub>CO<sub>4</sub> with the *Pnma* space group.

547

548 Figure 5. Comparison of experimental (temperature quenched from ~2255 K) versus theoretical

Raman spectra of  $Ca_2CO_4$ -*Pnma* at ~20 GPa. The calculated frequencies were multiplied by a scaling factor of 1.04.

551

**Figure 6.** Pressure-dependence of the reaction enthalpy  $\Delta H$  for the reaction  $Ca_2CO_4 + 2SiO_2 \rightleftharpoons$ 552  $2CaSiO_3 + CO_2$ . Negative values indicate that the phase assemblage on the left-hand side of the 553 554 equation is stable. Cross symbols represent calculated values, the solid red line is a linear fit to these values, while the dashed red parts indicate an extrapolation. The values plotted are for CO<sub>2</sub>-555 V and orthorhombic CaSiO<sub>3</sub>-perovskite. Using either CO<sub>2</sub>-III or CO<sub>2</sub>-IV (Han et al., 2019; Lu et 556 al., 2013) would stabilize Ca<sub>2</sub>CO<sub>4</sub> further by ~0.2 eV. The enthalpy difference between cubic and 557 orthorhombic CaSiO<sub>3</sub> (Caracas et al., 2005, 2006; Komabayashi et al., 2007) is negligible in the 558 present context. 559

Pressure (GPa) Group Min Max Average *d*(C-O) 1.378(14) 1.386(7)1.383 20.1(2)∠(O-C-O) 107.5(5) 112.9(6) 110.3 d(Ca1-O)2.227(8) 2.601(2)2.360 2.258(9) d(Ca2-O)2.893(6) 2.532 89.0(8) *d*(C-O) 1.323(16) 1.381(11) 1.340 ∠(O-C-O) 116.0(9) 107.4(6) 109.5

**Table 1.** Structural parameters of Ca<sub>2</sub>CO<sub>4</sub>-*Pnma* at 20 and 89 GPa. Distances are in d(Å), angles in  $\angle(\circ)$ .

<i>d</i> (Ca1-O)	2.088(8)	2.329(4)	2.174	
<i>d</i> (Ca2-O)	2.129(6)	2.559(10)	2.338	













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