

1 Type of Revision: revision 2
2 Word count: 5451

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4 **Nitrogen diffusion in silicate melts under reducing conditions**

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11

12 **Abstract**

13

14 The behavior of nitrogen during magmatic degassing and the potential kinetic

15 fractionation between N and other volatile species (H, C, O, noble gases) are poorly known

16 due to the paucity of N diffusion data in silicate melts. To better constrain N mobility during

17 magmatic processes, we investigated N diffusion in silicate melts under reducing conditions.

18 We developed uniaxial diffusion experiments at 1 atm, 1425 °C, and under nominally

19 anhydrous reducing conditions ($fO_2 \leq IW - 5.1$, where IW is oxygen fugacity, fO_2 , reported in

20 log units relative to the iron-wüstite buffer), in which N was chemically dissolved in silicate

21 melts as nitride (N^{3-}). Although several experimental designs were tested (platinum,

22 amorphous graphite, and compacted graphite crucibles), only N diffusion experiments at IW –

23 8 in compacted graphite crucibles for simplified basaltic andesite melts were successful.

24 Measured N diffusivity (D_N) is on the order of $5.3 \pm 1.5 \times 10^{-8} \text{ cm}^2 \text{ s}^{-1}$, two orders of

25 magnitude lower than N chemical diffusion in soda-lime silicate melts (Frischat et al., 1978).

26 This difference suggests that nitride diffusivity increases with increasing degree of melt

27 depolymerization. The dependence of N^{3-} diffusion on melt composition is greater than that

28 of Ar. Furthermore, N^{3-} diffusion in basaltic-andesitic melts is significantly slower than that

29 of Ar in similarly polymerized andesitic-tholeiitic melts at magmatic temperatures (1400–

30 1450 °C; Nowak et al., 2004). This implies that N/Ar ratios can be fractionated during

31 reducing magmatic processes, such as during early-Earth's magma ocean stages.

32

33 **Keywords:** Nitrogen, diffusion, nitride, silicate melts, basaltic andesite.

34

35 **1. Introduction**

36 Understanding the behavior of volatile elements (H, C, N, noble gases) in silicate
37 melts is fundamental to better constrain their fate during the formation and evolution of Earth
38 and other planetary bodies in the inner solar system (Marty et al., 2016; Piani et al., 2020).
39 Because N, as molecular N₂, is expected to behave like noble gases, particularly Ar, during
40 magmatic processes (Marty, 1995; Miyazaki et al., 1995, 2004), the N₂/³⁶Ar ratios of Earth's
41 atmosphere and mantle are expected to be comparable. However, the N₂/³⁶Ar value of Earth's
42 mantle (>10⁶, estimated from mid-ocean-ridge basalt glasses) is two orders of magnitude
43 higher than that of the atmosphere (~10⁴) (Marty, 1995; Marty et al., 1995; Marty and
44 Humbert, 1997), perhaps as a result of kinetic disequilibrium during the formation of Earth's
45 atmosphere due to mantle degassing (Marty et al., 1995) or the distinct volatile histories of
46 these two reservoirs (Marty and Humbert, 1997). Diffusion strongly controls this ratio
47 because it governs volatile transport and degassing during magmatic activity at all stages of
48 planetary evolution (from magma ocean differentiation to volcanic eruptions). However,
49 whereas the diffusion of H₂O, CO₂, and noble gases in silicate melts has been widely studied
50 (e.g., Zhang and Stolper, 1991; Baker et al., 2005; Zhang and Ni, 2010; Lux, 1987; Roselieb
51 et al., 1995; Amalberti et al., 2018), N diffusion in silicate melts remains largely under-
52 constrained.

53 Elemental speciation fundamentally controls the diffusion of redox-sensitive species
54 such as nitrogen in silicate melts (Frischat et al., 1978; Zhang and Ni, 2010). Depending on
55 *f*O₂ conditions, N is incorporated as free molecular N₂ and NH₃ or bonded N³⁻, CN⁻, NH²⁻,
56 and/or NH₂⁻ (Libourel et al., 2003; Li et al., 2015; Dalou et al., 2017, 2019; Mosenfelder et

57 [al., 2019](#); [Boulliung et al., 2020](#); [Grewal et al., 2020](#)). At atmospheric pressure and under
58 anhydrous conditions, N is either incorporated as molecular N₂ under oxidizing conditions
59 ($fO_2 > IW - 1.5$) or chemically dissolved as nitride (N³⁻) under reducing conditions ($fO_2 \leq IW$
60 $- 1.5$; [Libourel et al., 2003](#); [Boulliung et al., 2020](#)). To date, N diffusion in silicate melts has
61 only been studied in soda-lime silicate melts at atmospheric pressure and high temperature
62 (1000–1400 °C) for molecular N₂ and nitride species ([Frischat et al., 1978](#)). In these melts,
63 physical diffusion of N₂ is significantly faster than chemical diffusion of nitride ([Behrens,](#)
64 [2010](#)). However, N solubility not only depends on fO_2 , but also on melt composition and
65 structure; when dissolved as N³⁻, N solubility increases with increasing degree of melt
66 depolymerization ([Boulliung et al., 2020](#)). These observations indicate that melt composition
67 may affect N diffusion, and, therefore, that N diffusivities determined for soda-lime silicate
68 melts cannot be representative of those in natural silicate melts.

69 In this study, various experimental approaches were tested to determine uniaxial N
70 diffusion in silicate melts of simplified basaltic andesite and highly depolymerized, magma
71 ocean-like compositions to address the influence of melt composition on N diffusion under
72 conditions in which N is chemically incorporated as nitride. Uniaxial diffusion experiments
73 were conducted at 1425 °C, 1 atm, and $IW - 8$ and $IW - 5.1$. Nitrogen concentration profiles in
74 the quenched run products were determined by *in-situ* secondary ion mass spectrometry
75 (SIMS). The results provide new insights into the N diffusion mechanism(s) in naturally
76 occurring silicate melts, thereby improving our understanding of the behavior of N during
77 reducing magmatic processes.

78

79 **2. Experimental and analytical methods**

80 Two starting materials of different simplified iron-free compositions were synthesized
81 and their bulk compositions determined by electron microprobe analysis at Université de

82 Lorraine (Service Commun de Microscopie Electronique et de Microanalyses X, Nancy,
83 France): (i) a basaltic andesite composition (AND1; 53.6 wt% SiO₂, 20.7 wt% Al₂O₃, 9.9
84 wt% MgO, 15.0 wt% CaO, 0.45 wt% Na₂O, 0.16 wt% K₂O) and (ii) a mafic composition
85 (MO2; 47.4 wt% SiO₂, 4.6 wt% Al₂O₃, 19.8 wt% MgO, 26.1 wt% CaO). These starting glass
86 materials are characterized by extreme NBO/T values (the number of non-bridging oxygen
87 atoms per tetrahedrally coordinated cations) of 0.5 (AND1) and 2.1 (MO2); the latter is
88 comparable to that estimated for the terrestrial magma ocean ([Ringwood, 1966](#); [Dasgupta and](#)
89 [Grewal, 2019](#)). To prepare these starting materials, oxide and carbonate powders were dried
90 and decarbonated at 1000 °C in a muffle furnace for 12 h. After mixing in appropriate
91 proportions, they were fused in a platinum crucible at 1400 °C in a muffle furnace for 4 h and
92 quenched in air. The fused starting glasses AND1 and MO2 were ground for 1 h for use in the
93 different experiments.

94 Different experimental supports were tested to study uniaxial N diffusion. Compacted
95 graphite (15 mm long, 6 mm outer diameter, 1.5 mm wall thickness) and vitreous graphite (12
96 mm long, 7 mm outer diameter, 1 mm wall thickness) cylinders were used for experiments at
97 IW -8, whereas Pt cylinders (13 mm long, 3 mm outer diameter, 0.2 mm wall thickness) were
98 used for experiments at IW -5.1 ([Table 1](#)). To obtain bubble-free glass prior to the diffusion
99 experiments, the starting powder was incrementally filled into the cylinders and melted for 30
100 min under a CO₂ flux (300 cm³ min⁻¹) at 1600 °C for Pt crucibles (as described by [Amalberti](#)
101 [et al., 2018](#)) and under Ar flux (300 cm³ min⁻¹) at 1450 °C for graphite cylinders to minimize
102 potential atmospheric nitrogen contamination. The filled cylinders were subjected to a final
103 melting step at 1550 °C for Pt crucibles and 1450 °C for graphite crucibles during ~15 h
104 under the same Ar and CO₂ fluxes to ensure the complete removal of bubbles.

105 Uniaxial N diffusion experiments were performed in a Gero vertical furnace at
106 atmospheric pressure, 1425 ± 1 °C, and under a controlled N₂-CO(-CO₂) atmosphere at log

107 $fO_2 = -17.4$ and -14.5 (IW -8 and IW -5.1 , respectively; [Table 1](#)). Diffusion at such low
108 oxygen fugacities is expected to result in discernable and measurable N concentration
109 gradients, whereas low N solubilities in silicate melts under more oxidizing conditions (log
110 $fO_2 > IW -1.5$) limit N contents to a few ppm ([Libourel et al., 2003](#); [Boulliung et al., 2020](#)).
111 The gas flux was maintained at $300 \text{ cm}^3 \text{ min}^{-1}$ by TYLAN mass flow controllers. The
112 graphite cylinders were placed into an alumina crucible, whereas the platinum cylinders were
113 held in place using Pt wire (0.3 mm). Diffusion experiments using compacted graphite and
114 vitreous graphite cylinders lasted 3 h, but were restricted to 1 h with Pt crucibles to minimize
115 Pt loss by evaporation at high temperature under reducing conditions ([Darling et al., 1970](#))
116 ([Table 1](#)). Regardless of melt composition, these experimental durations are too short to reach
117 gas-melt equilibrium; based on experiments using 2–5 mm diameter silicate melt spherules in
118 graphite crucibles, equilibrium is expected to be achieved in 24 hours ([Humbert, 1998](#);
119 [Boulliung et al., 2020](#)). After quenching in air, run products were fixed onto a flat ceramic
120 support using CrystalbondTM adhesive and cut in half along the direction of diffusion. The
121 obtained glass half-cylinders were then mounted in high-purity indium, polished, and gold
122 coated.

123 N concentrations were quantified by *in-situ* measurements of $^{14}\text{N}^{16}\text{O}^-$ secondary ions
124 using the CAMECA 1280 HR2 SIMS at the Centre de Recherches Pétrographiques et
125 Géochimiques (Nancy, France) operating with a 10 kV Cs^+ primary ion beam, a current of
126 ~ 10 nA, and a nominal mass resolution of $\sim 13,000$ (for details, see [Füri et al., 2018](#)). N
127 contents were measured along profiles from each exterior edge of the glass cylinders towards
128 their center ([Fig. 1](#)). Uncertainties on N contents (2σ) are $\leq 4\%$ for N contents > 100 ppm,
129 $\leq 30\%$ between 10 and 100 ppm, and 50–100% below 10 ppm. Carbon (CO_2) and water (H_2O)
130 contents were measured in one sample (along profile A of AND1; [Fig. 1](#)) using the CAMECA

131 IMS-1270 E7 ion microprobe (see supplementary material for details) to assess the potential
132 presence and abundance of these volatiles.

133

134 **Results**

135 Previous gas-melt equilibrium experiments at IW -5.1 yielded N contents of $68.1 \pm$
136 6.0 ppm and 35.6 ± 10.7 ppm for AND1 and MO2, respectively, and those at IW -8 yielded
137 2672 ± 40 ppm and 5987 ± 63 ppm, respectively, after 24 hours (Boulliung et al., 2020).
138 These experiments demonstrated that N solubility is highest in depolymerized melts under
139 very reducing conditions.

140 The run products of diffusion experiments performed with platinum crucibles at IW $-$
141 5.1 contain <5 ppm N, and concentrations are indistinguishable within analytical
142 uncertainties. The glasses produced using vitreous graphite crucibles at IW -8 have
143 homogeneous N concentrations (208 ± 10 ppm in AND1 and 400 ± 47 ppm in MO2) and lack
144 any apparent diffusion profiles. Therefore, N diffusion coefficients could not be determined in
145 experiments using platinum and vitreous graphite crucibles.

146 A significant loss of melt ($\sim 50\%$) and partial crystallization was observed in the
147 experiment on the MO2 melt using a compacted graphite crucible at IW -8 . Given its high
148 degree of depolymerization ($NBO/T = 2.1$) and correspondingly low viscosity, this melt loss
149 was likely caused by melt migration. We assume that minor evaporation of major oxides
150 (mainly SiO_2 and MgO) during the bubble elimination steps (at $1450^\circ C$ under an Ar flux),
151 affected the liquidus temperature (initially estimated just below $\sim 1400^\circ C$) and promoted
152 crystallization upon quenching. Due to these limitations, no nitrogen diffusion profiles were
153 observed for the MO2 melt. In contrast, using the same experimental design, the AND1 melt
154 was successfully quenched to a crystal-free glass, and N concentration gradients were
155 determined along four profiles (A–D; Fig. 1). The N concentrations range from 1068 ± 41 to 4

156 ± 3 ppm, 142 ± 5 to 7 ± 4 ppm, 220 ± 9 to 8 ± 4 ppm, and 2001 ± 80 to 541 ± 22 ppm along
157 profiles A, B, C, and D, respectively (Fig. 1, Table 2). Although the surface of the melt adopts
158 the shape of a concave meniscus (Fig. 1E), N concentrations along the gas-melt interface are
159 nearly homogeneous (906 ± 37 to 1068 ± 41 ppm; supplementary Fig. A1).

160

161 Discussion

162 The N contents of glasses produced in experiments using Pt crucibles are too low (<5
163 ppm) to detect concentration gradients by SIMS. Although the 1-h experimental duration was
164 probably too short to dissolve a significant amount of N, even at low fO_2 conditions (IW -5.1)
165 under which N contents are expected to be relatively important at gas-melt equilibrium (i.e.,
166 tens of ppm; Boulliung et al., 2020), longer experiments were not possible because Pt
167 crucibles are prone to evaporative losses at such strongly reducing conditions. The run
168 products of experiments using vitreous graphite are N-rich (208 ± 10 ppm in AND1, 400 ± 47
169 ppm in MO2), but were likely homogenized by convection induced by the bottom-up
170 migration of silicate melts along the vitreous graphite crucible. We therefore conclude that
171 platinum and vitreous graphite are not appropriate for experiments on N diffusion in silicate
172 melt under reducing conditions.

173 On the other hand, the uniaxial diffusion experiment using compacted graphite yielded
174 variable N concentration gradients in glass AND1 (Fig. 1). The occurrence of N gradients
175 along profiles B, C, and D (Fig. 1) is surprising since no direct gas-melt interaction was
176 expected at the edges or bottom of the glass cylinder. These gradients may be the result of N
177 migration along or diffusion through the graphite crucible. Nevertheless, the central part of
178 the glass cylinder is N-free (Table 2) which suggests that no or very limited convection
179 movements occurred in the melt during the experiment. The N diffusion coefficients can be

180 determined from each profile using the one-dimensional diffusion equation in a semi-infinite
181 medium with a constant interface concentration (Crank, 1975):

$$182 \quad \frac{(C(x,t) - C_1)}{(C_0 - C_1)} = \operatorname{erf} \left(\frac{x}{2\sqrt{Dt}} \right) \quad (1)$$

183
184 where $C(x,t)$ is the concentration at distance x from the gas-melt interface after time t (the
185 experimental duration), C_0 is the initial concentration in the glass (here, $C_0 = 0$), C_1 is the
186 concentration at the gas-melt interface ($x = 0$), and D is the diffusion coefficient. For profile
187 A, we chose the starting point ($x = 0$) to be at the center of the meniscus, whereas the starting
188 point for the other profiles was chosen arbitrarily. In each case, the concentration at the gas-
189 melt interface was obtained by extrapolating the diffusion profile to $x = 0$. Finally, we
190 linearized the left-hand side of equation (1) by plotting the inverse error function against
191 distance to extract the diffusion coefficient as (Van Orman et al., 1998):

$$192 \quad D = (4m^2t)^{-1} \quad (2)$$

193 The slope of the best-fit line, m , was derived using IsoplotR (Vermeesch 2018) with an
194 uncertainty that takes into account the uncertainties on measured N concentrations and on x
195 positions (5 μm). The diffusion coefficients are reported in Figure 1, and the inversion error
196 function of diffusion profile A is presented as a representative example in Figure 2. Because
197 of the meniscus shape of the gas-melt interface (Fig. 1E), N diffusion may not be strictly one-
198 dimensional along profile A. Thus, the main source of uncertainty for profile A is related to
199 the position of the origin of the diffusion profile ($x = 0$). The overall uncertainty was
200 evaluated by comparing the diffusion coefficient for a starting point at the center ($4.2 \pm 0.4 \times$
201 $10^{-8} \text{ cm}^2 \text{ s}^{-1}$) and at mid-height of the meniscus ($6.2 \pm 0.6 \times 10^{-8} \text{ cm}^2 \text{ s}^{-1}$). By coupling the
202 standard deviation related to the geometrical and the analytical uncertainties, the average
203 diffusion coefficient is equal to $5.3 \pm 1.5 \times 10^{-8} \text{ cm}^2 \text{ s}^{-1}$ along profile A. The diffusion
204 coefficients obtained for the four profiles differ significantly, with lower values for profiles B,
205 C and D (Fig. 1), likely due to distinct boundary conditions; given that N diffusion along

206 profiles B, C, and D may have been controlled by delayed transport of nitrogen through the
207 graphite container wall, these three values are excluded from further discussions.

208 To date, the diffusion of molecularly (N_2) and chemically incorporated nitrogen (N^{3-})
209 in silicate melts has only been studied in soda-lime silicate compositions at atmospheric
210 pressure and 1000–1400 °C (Frischat et al., 1978). The authors studied N_2 diffusion by
211 heating a pre-saturated glass slab under an oxygen atmosphere and chemical diffusion by
212 heating a glass containing silicon nitride under an Ar- H_2 atmosphere. Under these conditions,
213 N diffuses faster as N_2 than N^{3-} in soda-lime silicate melts (Frischat et al., 1978; Behrens,
214 2010). However, nitride diffusivity increases more rapidly than N_2 diffusivity with increasing
215 temperature because of its higher activation energy (244 kJ mol⁻¹ for N^{3-} vs. 161 kJ mol⁻¹ for
216 N_2 ; Frischat et al., 1978). Indeed, N_2 displays an intrinsic diffusivity resembling that of noble
217 gases (Dingwell and Webb, 1990; Dingwell, 2006), whereas N^{3-} diffusivity seems to be
218 extrinsic, as it increases significantly with increasing temperature.

219 Under nominally anhydrous and very reducing conditions (i.e., IW -8), N is expected
220 to dissolve predominantly as nitride species (Libourel et al., 2003), likely bonded as Si-N
221 (Boulliung et al., 2020), whereas C-N and N-H species are inferred to represent a minor
222 fraction of the total N budget (see supplementary material for details). After extrapolation of
223 the data from Frischat et al. (1978) to 1425 °C, N^{3-} diffusion in soda-lime silicate melts is two
224 orders of magnitude higher ($\sim 6.0 \times 10^{-6} \text{ cm}^2 \text{ s}^{-1}$) than in our melt AND1 ($5.3 \pm 1.5 \times 10^{-8} \text{ cm}^2$
225 s^{-1}). As soda-lime silicate melts are more depolymerized (NBO/T = 0.7) than basaltic andesite
226 melts (NBO/T = 0.5), it appears that N^{3-} diffusion increases with melt depolymerization.
227 However, since the experimental conditions (e.g., fO_2) of the study of Frischat et al. (1978)
228 are not clearly indicated, it is difficult to ascertain that the NBO/T value is the only parameter
229 responsible for this difference. Although additional experiments, including distinct melt
230 compositions, are needed to better understand the effect of melt polymerization on N^{3-}

231 diffusion, we note here that similar behavior is observed for O^{2-} , for which the activation
232 energy decreases with increasing depolymerization (e.g., [Leshner, 2010](#)). As in the case of
233 oxygen, nitride diffusion requires the breaking of Si-O bonds, and the increased occurrence of
234 Si-O-M (where M is a network-modifying cation and O a non-bridging oxygen) compared to
235 Si-O-Si bounds (with O a bridging oxygen) in highly depolymerized melts enhances nitride
236 diffusion by oxygen substitution ([Boulliung et al., 2020](#)). To compare the diffusivity of
237 network formers (such as Si and Al) and O^{2-} (e.g., [Dingwell and Webb, 1990](#)) with that of
238 nitride obtained for profile A, we used the Eyring equation:

$$239 \quad D = \frac{kT}{\eta\lambda} \quad (3)$$

240 where D is the diffusion coefficient in $m^2 s^{-1}$; k is the Boltzmann constant, T is the
241 temperature in K, η is the melt viscosity in Pa s, and λ is the jump distance, which is fixed at 3
242 Å for the diffusion of network formers (e.g., [Nowak et al., 2004](#)). The melt viscosity (η) of
243 AND1 at 1425°C is 2.19 P as (calculated using the model of [Giordano et al., 2008](#)) (see
244 supplementary material for details). The Eyring diffusivity at 1425°C, $\sim 3.6 \times 10^{-7} cm^2 s^{-1}$, is
245 less than one order of magnitude higher than the diffusion coefficient obtained for N in this
246 study ($5.3 \pm 1.5 \times 10^{-8} cm^2 s^{-1}$; supplementary [Fig. A2](#)). This similarity suggests that
247 movements of network formers are involved in nitride diffusion, reinforcing the conclusion of
248 [Boulliung et al. \(2020\)](#) that N diffuses as a Si-N species under highly reducing conditions in
249 basaltic andesite melts.

250 As with N^{3-} and O^{2-} , noble gas diffusivity also increases with increasing degree of melt
251 depolymerization. For example, at 1400 °C, [Amalberti et al. \(2018\)](#) reported that the argon
252 diffusion coefficient (D_{Ar}) in a synthetic basaltic melt with NBO/T = 1 ($9.3 \times 10^{-7} cm^2 s^{-1}$) is
253 ~ 3 times higher than that in a synthetic Hawaiitic melt with NBO/T = 0.7 ($2.9 \times 10^{-7} cm^2 s^{-1}$;
254 [Nowak et al. 2004](#)). The relative difference in NBO/T between the soda-lime silicate melt
255 (NBO/T = 0.7) used by [Frischat et al. \(1978\)](#) and our melt AND1 (NBO/T = 0.5) is

256 comparable to that between basaltic and Hawaiitic melts. However, D_N is ~ 110 times higher
257 in soda-lime silicate melts than in AND1, indicating that nitride diffusion is more dependent
258 on the degree of melt polymerization than Ar diffusion in this temperature range.
259 Furthermore, at the same degree of melt polymerization ($NBO/T = 0.5$), $D_{Ar} = 1.5 \times 10^{-7} \text{ cm}^2$
260 s^{-1} in andesitic-tholeiitic melts at 1400–1450 °C and 500 MPa (Nowak et al., 2004), is ~ 3
261 times higher than the D_N value observed for our AND1 melt ($5.3 \times 10^{-8} \text{ cm}^2 \text{ s}^{-1}$). Since
262 increased pressure slows down noble gas diffusion in silicate melts (Roselieb et al., 1996;
263 Behrens and Zhang, 2001; Zhang et al., 2007), D_{Ar} is expected to be even larger at 1 atm. This
264 difference between D_{Ar} and D_N suggests that under reducing conditions (i.e., N as N^{3-}),
265 diffusion during magmatic processes significantly fractionates the N/Ar ratio.

266

267 **Implications**

268 Our results imply the efficient transport of N as nitride under magma ocean conditions
269 during early stages of Earth's formation (i.e., fO_2 between IW -3 and IW -5 for 70% of
270 Earth's mass accreted, Rubie et al., 2011; Wade and Wood, 2005; ~ 2000 K, Rubie, 2007;
271 Solomatov, 2007; and $NBO/T \sim 2$, Ringwood, 1966; Javoy et al., 2010; Dasgupta and Grewal,
272 2019). Under such conditions favorable to nitrogen diffusion as nitride, the N/Ar ratio may be
273 kinetically fractionated during degassing and/or ingassing and melt migration. Therefore, the
274 different N/Ar ratios of Earth's mantle and atmosphere (e.g., Marty, 1995) may have resulted
275 from an early stage of atmospheric formation when Earth's magma ocean was highly
276 reducing. During later stages of magma ocean differentiation, for redox conditions between
277 IW -3 and IW -1.5 (i.e., when Earth had accreted more than 70% of its final mass, Rubie et
278 al., 2011), N dissolves not only as N^{3-} , but as other species such as N-H or C-N complexes,
279 and, to a lesser extent, N_2 (e.g., Dalou et al., 2019; Grewal et al., 2020; Boulliung et al. 2020)
280 depending on fH_2 and fO_2 . Since the speciation of volatile elements affects their diffusivity in

281 silicate melts (e.g., [Frischat et al., 1978](#); [Zhang and Ni, 2010](#)), the diffusion and transport
282 mechanism(s) of such species needs to be investigated through multi-component experiments
283 over a range of P - T - fO_2 conditions. With this respect, the experimental design (gas-melt
284 interaction in compacted graphite crucibles), analytical protocol (*in situ* SIMS analysis), and
285 first results presented here represent an important step towards improving our understanding
286 of N behavior within planetary magma oceans.

287

288 **Figure captions**

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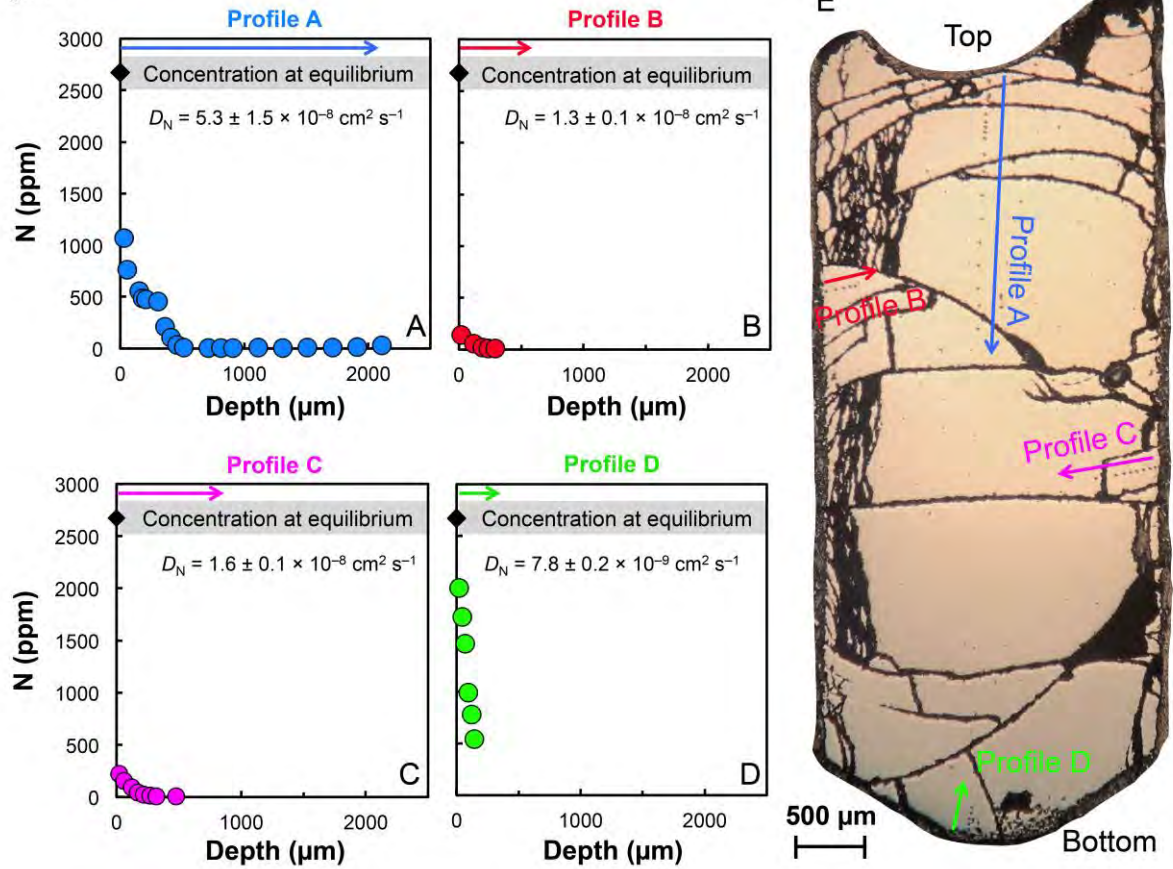
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291 **Fig. 1.** Nitrogen concentration profiles A–D measured in glass cylinder AND1 (1 atm, 1425
292 °C, 3 h, IW –8); the respective SIMS spot analyses are the small dark spots in the photograph
293 in panel E. The indicated equilibrium nitrogen concentration (2672 ± 40 ppm) is that
294 determined from the 24-h gas-melt equilibrium experiment on AND1 under similar
295 experimental conditions ([Boulliung et al., 2020](#)). The diffusion coefficient (and the
296 uncertainty thereof) along profile A takes into account the geometrical and analytical
297 uncertainties (see text for details).

298

299 **Fig. 2.** Error function inversion of profile A ([Fig. 1](#)) as a function of depth in the sample (for x
300 = 0, i.e., a starting point at the center of the meniscus). The nitrogen diffusion coefficient D_N
301 was calculated using the inverse error function according to equations (1) and (2). The linear
302 regression between the inverse function of concentration and depth was calculated using
303 IsoplotR ([Vermeesch, 2018](#)).

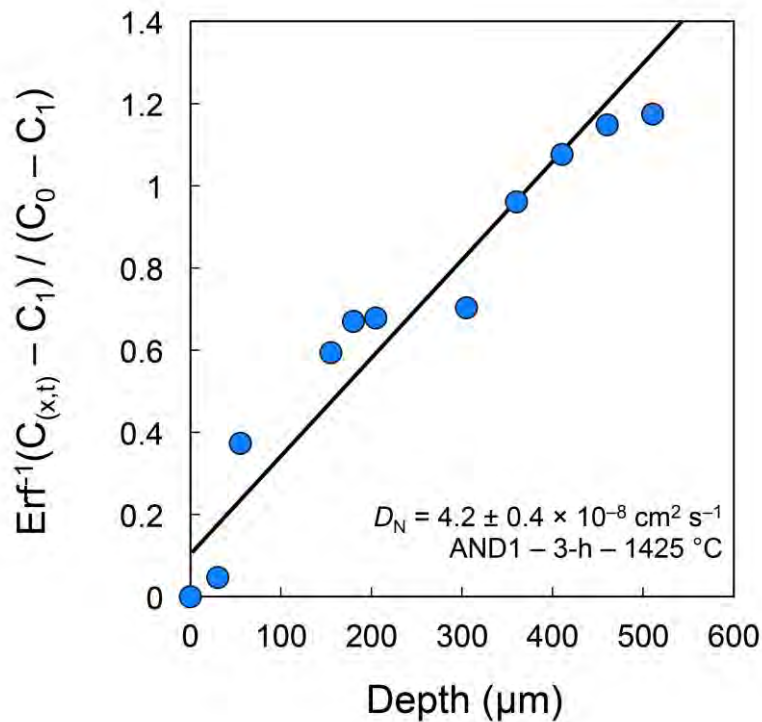
Figure 1



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Figure 2



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Table 1. Run conditions and fractional gas mixtures of nitrogen nominally uniaxial diffusion (1–3 h) at $P = 1$ atm, $T = 1425$ °C for both AND1 and MO2 melt compositions. These nominal fO_2 values were calculated using the JANAF and Thermodata database (see [Boulliung et al., 2020](#), for details).

fO_2 (IW)	$\log fO_2$	Sample holder	Duration (hours)	CO	CO ₂	N ₂
IW –8	–17.4	Vitreous graphite	3	0.2	–	0.8
IW –8	–17.4	Compacted graphite	3	0.2	–	0.8
IW –5.1	–14.5	Platinum	1	0.4994	0.0006	0.5

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360 Table 2: N contents measured along the profiles presented in Figure 1A–D. SIMS spot
 361 references refer to the profile (A–D) and number of each analysis along the profile as
 362 indicated by the arrows in Figure 1E.
 363

	SIMS spot reference	Depth (μm)	N (ppm)
364			
365			
366	A-1	30	1068 ± 41
367	A-2	55	762 ± 15
368	A-3	155	556 ± 11
369	A-4	180	485 ± 10
370	A-5	205	477 ± 10
371	A-6	305	454 ± 9
372	A-7	360	212 ± 5
	A-8	410	104 ± 4
373	A-9	460	36 ± 5
	A-10	510	11 ± 3
374	A-11	710	4 ± 3
	A-12	810	5 ± 3
375	A-13	910	6 ± 4
	A-14	1110	7 ± 4
376	A-15	1310	6 ± 4
	A-16	1510	7 ± 4
377	A-17	1710	9 ± 4
	A-18	1910	15 ± 3
378	A-19	2110	33 ± 3
379	B-1	20	142 ± 5
	B-2	120	56 ± 4
380	B-3	190	22 ± 3
	B-4	240	10 ± 5
381	B-5	290	7 ± 4
382	C-1	20	220 ± 9
	C-2	60	156 ± 6
383	C-3	120	89 ± 9
	C-4	170	41 ± 6
384	C-5	220	20 ± 6
	C-6	270	13 ± 4
385	C-7	320	10 ± 5
386	C-8	480	8 ± 4
387	D-1	20	2001 ± 80
	D-2	45	1720 ± 69
	D-3	70	1460 ± 58
388	D-4	95	993 ± 40
	D-5	120	777 ± 31
389	D-6	145	541 ± 22

390

391 **Acknowledgments**

392 Delphine Lequin is thanked for her help during the diffusion experiments. Technical support
393 by Cécile Deligny, Etienne Deloule, Nordine Bouden, and Johan Villeneuve (SIMS), and
394 Olivier Rouer (electron microprobe), is gratefully acknowledged, and we thank François
395 Faure for constructive discussions. Robert Dennen is thanked for English editing.
396 Constructive comments by associate editor Charles E. Lesher, Harald Behrens, and an
397 anonymous reviewer helped to improve the manuscript. This work was supported by the
398 European Research Council (ERC) under the European Union's Horizon 2020 research and
399 innovation program (grant agreement no. 715028). This is CRPG contribution 2733.

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