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2 **Phase Diagram and Thermal Expansion of**  
3 **orthopyroxene-, clinopyroxene- and**  
4 **ilmenite-structured MgGeO<sub>3</sub>**

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7 Simon A. Hunt<sup>1, 2, \*</sup>, James R. Santangeli<sup>2</sup>, David P. Dobson<sup>2</sup>, Ian G. Wood<sup>2</sup>.

8  
9 1 – Department of Materials, University of Manchester, Sackville Street Building, Manchester. M1 3BB. UK

10 2 – Department of Earth Sciences, University College London, Gower Street, London. WC1E 6BT. UK

11 \* - [simon.hunt@manchester.ac.uk](mailto:simon.hunt@manchester.ac.uk)

12  
13 **Abstract**

14 *The MgGeO<sub>3</sub> system is a low-pressure analogue for the Earth forming (Mg,Fe)SiO<sub>3</sub> system and*  
15 *exhibits recoverable orthopyroxene, clinopyroxene and ilmenite structures below 6 GPa. The*  
16 *pressure-temperature conditions of the clinopyroxene to ilmenite phase transition are reasonably*  
17 *consistent between studies, having a positive Clapeyron slope and occurring between 4 and 7*  
18 *GPa in the temperature range 900-1600 K. There are, though, significant discrepancies in the*  
19 *Clapeyron slope of the orthopyroxene to clinopyroxene phase transition in existing works which*  
20 *also disagree on the stable phase at ambient conditions. The most significant factor in these*  
21 *differences is the method used; high-pressure experiments and thermophysical property*

22 *measurements yield apparently contradicting results. Here, we perform both high pressure and*  
23 *temperature experiments as well as thermal expansion measurements to reconcile the*  
24 *measurements. High pressure and temperature experiments, yield a Clapeyron slope of  $-1.0_{-0.7}^{+1.0}$*   
25 *MPa/K for the MgGeO<sub>3</sub> orthopyroxene-clinopyroxene phase transition, consistent with previous*  
26 *high pressure and temperature experiments. The MgGeO<sub>3</sub> orthopyroxene-clinopyroxene-ilmenite*  
27 *triple point is determined to be at 0.98 GPa and 752 K, with the ilmenite phase stable at ambient*  
28 *conditions. The high temperature (> 600 K) thermal expansion of the clinopyroxene phase is*  
29 *greater than that of the other phases. Debye-Grüneisen relationships fitted to the volume-*  
30 *temperature data give Debye temperatures for the orthopyroxene, clinopyroxene and ilmenite*  
31 *phases of 602(7) K, 693(10) K and 758(13) K and  $V_0$  of 897.299(16) Å<sup>3</sup>, 433.192(10) Å<sup>3</sup> and*  
32 *289.156(6) Å<sup>3</sup> respectively. The Clapeyron slopes calculated directly from the Debye-Grüneisen*  
33 *relationships are consistent with previous thermophysical property measurements. The presence*  
34 *of significant anharmonicity and/or formation of defects in the clinopyroxene phase at high-*  
35 *temperatures, which is not apparent in the other phases, accounts for the previous contradictions*  
36 *between studies. The inferred increased heat capacity of the clinopyroxene corresponds to an*  
37 *increase in entropy and an expanded phase field at high temperatures.*

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### 39 **Keywords**

40 Equation of State, Heat Capacity, Thermal Expansion, high-pressure, phase transition,

41 anharmonicity

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## Introduction

The MgO-GeO<sub>2</sub> system has long been studied because of its similarity to the MgO-SiO<sub>2</sub> system but with structure types stable at lower pressures. Like MgSiO<sub>3</sub>, MgGeO<sub>3</sub> transforms from the ilmenite structure to both the perovskite structure (at around 12 GPa, Leinenweber et al. 1994) and the post-perovskite structure (around 65 GPa, Akaogi et al. 2005; Hirose et al. 2005). Below 10 GPa, MgGeO<sub>3</sub>, like MgSiO<sub>3</sub>, exhibits orthopyroxene, clinopyroxene and ilmenite structures (Ringwood and Seabrook 1962; Kirfel and Neuhaus 1974; Ozima and Akimoto 1983; Ross and Navrotsky 1988). Both the MgO-GeO<sub>2</sub> and MgO-SiO<sub>2</sub> systems have ambient pressure olivine structures that transform to spinel structures at elevated pressure and temperature (Ross and Navrotsky 1987; Akaogi et al. 1989). The germanate system is not, though, a perfect low-pressure reflection of the silicate system. Pure GeO<sub>2</sub> is stable in the rutile structure at ambient conditions and above 1310 K transforms to that of quartz (Laubengayer and Morton 1932). This contrasts with SiO<sub>2</sub>, in which quartz is the ambient pressure phase and the rutile structure is stable above 9 GPa as stishovite. These two silica phases are separated by coesite, which has not been found in GeO<sub>2</sub>. Neither has the wadsleyite ( $\beta$ -Mg<sub>2</sub>SiO<sub>4</sub>) structure been observed in Mg<sub>2</sub>GeO<sub>4</sub>. Above about 4 GPa and 1000 K in MgGeO<sub>3</sub>, the ilmenite structure is stable and there is no appearance of the Mg<sub>2</sub>SiO<sub>4</sub> + SiO<sub>2</sub> field that separates the clinopyroxene and ilmenite phases in the MgSiO<sub>3</sub> system. An additional intermediate phase occurs in the MgO-GeO<sub>2</sub> binary that is not present in the MgO-SiO<sub>2</sub> binary. Initially reported as Mg<sub>4</sub>GeO<sub>6</sub> (Robbins and Levin 1959; McCormick 1964) and subsequently refined to be Mg<sub>3.5</sub>Ge<sub>1.25</sub>O<sub>6</sub> with space group *Pbam* (von Dreele et al. 1970; Kostiner and Bless 1971), at ambient pressure it is stable below 1768 K, whereupon it breaks down to Mg<sub>2</sub>GeO<sub>4</sub> + MgO (Robbins and Levin 1959).

67           There is more reported complexity in the silicate pyroxenes (e.g. Ohi and Miyake 2016)  
68 than in the germanate equivalents. Silicate clinopyroxene, with the space group  $C2/c$ , is stable  
69 above about 7 GPa (Ulmer and Stalder 2001). At low pressures, this transforms to either low-  
70 clinoenstatite (space group  $P2_1/c$ ) or, above 1073 K, orthoenstatite (space group  $Pbca$ ). Above  
71 ~1270 K, orthoenstatite transforms to an unrecoverable protoenstatite (space group  $Pbcn$ )  
72 (Murakami et al. 1982).

73           Only the orthoenstatite ( $Pbca$ ) and high-clinopyroxene ( $C2/c$ ) structures have been  
74 recovered in  $MgGeO_3$ , although the low-clinopyroxene form has been observed in other pyroxene-  
75 structured germanates (e.g.  $LiFeGe_2O_6$ , Redhammer et al. 2010). At ambient pressure, Robbins  
76 and Levin (1959) report that orthopyroxene- $MgGeO_3$  transforms to a high-clinopyroxene structure  
77 ( $C2/c$ ) above  $1828 \pm 5$  K. They comment that it is possible that the high-clinopyroxene phase is the  
78 back-transformation product of a ‘proto-pyroxene’ phase. This would be consistent with the  
79  $MgSiO_3$ - $MgGeO_3$  binary that has the enstatite—protoenstatite transition temperature increasing  
80 from ~1310 K to 1620 K between 0 and 12.5 mol.%  $MgGeO_3$  (Sarver and Hummel 1963). In  
81 addition, Lyon (1968) compared the observed diffraction peaks of the recovered high-temperature  
82 clinopyroxene phase unfavourably to the  $C2/c$  space group but did not make or analyse the  
83 clinopyroxene himself. Ozima and Akimoto (1983) argued that protoenstatite is the stable  
84  $MgGeO_3$  phase above ~1200 K but this is in disagreement with most other studies.

85           At pressures greater than ~4 GPa  $MgGeO_3$  adopts the ilmenite structure (Kirfel and  
86 Neuhaus 1974; Ozima and Akimoto 1983; Ross and Navrotsky 1988) (Figure 1). The  
87 clinopyroxene-ilmenite phase boundary is at lower pressure in the study of Ross and Navrotsky  
88 (1988) but is conceivably within the systematic errors of the experiments.

Insert Figure 1 here.

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90           There however remains disagreement between studies as to the phase diagram of MgGeO<sub>3</sub>  
91 below the ilmenite stability field. The extensive experimental study of Kirfel and Neuhaus (1974)  
92 proposed a negative Clapeyron slope for the orthopyroxene-clinopyroxene phase boundary with  
93 orthopyroxene as the low-temperature–low-pressure phase (Figure 1a). This is topologically  
94 consistent with Ringwood and Seabrook (1962) who, though, synthesised the clinopyroxene phase  
95 at 0.5 GPa. They therefore questioned whether the shallowness of the phase boundary was caused  
96 by shear stresses rather than pressure, as has been observed in natural pyroxenes (Coe and Kirby  
97 1975). Yamanaka et al. (1985) transformed pre-synthesised clinopyroxene phase to orthopyroxene  
98 phase at ~1173 K and ambient pressure and attempted, but did not observe, the reverse  
99 orthopyroxene to clinopyroxene phase reaction at 973 K and ambient pressure. The phase  
100 boundaries of Kirfel and Neuhaus (1974) are consistent with the high-pressure and temperature  
101 experimental constraints of Ross and Navrotsky (1988) (Figure 1b) but not with the positive  
102 Clapeyron slope derived by calorimetric measurements. The calorimetric Clapeyron slope makes  
103 the clinopyroxene phase the low-temperature–low-pressure phase and the stable phase at ambient  
104 conditions. Interestingly, the reported experimental results of both Kirfel and Neuhaus (1974) and  
105 Ross and Navrotsky (1988) are consistent with the reported sample synthesis conditions of all  
106 other studies (Roth 1955; Ringwood and Seabrook 1962; Liebermann 1974; Sato et al. 1977; Kirfel  
107 et al. 1978; Ito and Matsui 1979; Ashida et al. 1985; Okada et al. 2008) (Figure 1b) except that of  
108 Ozima and Akimoto (1983) (Figure 1c).

109           Ozima and Akimoto (1983) grew orthopyroxene phase and two types of clinopyroxene  
110 MgGeO<sub>3</sub> crystals from flux at ambient pressure. They argued that clinopyroxene is the low  
111 temperature phase, which transforms to ortho- and the proto-pyroxene structures with increasing

112 temperature. They also reported the stability of orthopyroxene phase between 3 and 5 GPa and  
113 ~1300 K where all other authors report the clinopyroxene phase.

114 Only a few measurements of the thermophysical properties of  $\text{MgGeO}_3$  phases have been  
115 made. Ross and Navrotsky (1988) used high temperature solution calorimetry and room  
116 temperature vibrational spectroscopy measurements of the three phases which they amalgamated  
117 to produce a thermodynamically consistent phase diagram (Figure 1b). Yamanaka et al. (1985)  
118 measured the thermal expansion and thermal evolution of the crystal structures of the ortho- and  
119 clinopyroxene phases between 293 K and 1373 K, but only with very large temperature steps.  
120 Ashida et al. (1985) measured the room-temperature compressibility, thermal expansion between  
121 300 and 1000 K, and heat capacity of ilmenite-structured  $\text{MgGeO}_3$  between 150 and 700 K.  
122 Tsuchiya and Tsuchiya (2007) used density functional theory to compute an thermal equation of  
123 state for ilmenite-structured  $\text{MgGeO}_3$  but did not investigate the pyroxene phases.

124 In this study, we attempt to unify the differing phase diagram topographies presented by  
125 experiments and thermodynamics. We performed high pressure and temperature quenching  
126 experiments to directly constrain the phase diagram of the  $\text{MgGeO}_3$  system and we also measured  
127 the thermal expansion of the three phases at ambient pressure. Fitting the thermal expansion  
128 volumes with a Debye-Grüneisen zero-pressure equation of state enables us to estimate the heat  
129 capacity and Clapeyron slope of the phase relations to compare with the direct experimental  
130 constraints.

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## Methods

### Sample Synthesis and high-pressure Phase Relations

Samples of orthopyroxene-structured  $\text{MgGeO}_3$  were synthesised from stoichiometric mixtures of  $\text{MgO}$  and  $\text{GeO}_2$ . The oxides were ground in an agate pestle and mortar under isopropanol and pressed into pellets. The pellets were sintered at 1373 K for 72 h in air. Pellets were then crushed and ground under isopropanol for 20 minutes to generate a fine powder and analysed by X-Ray powder diffraction to determine their purity. This sintering and analysis process was repeated, with minor additions of  $\text{MgO}$  or  $\text{GeO}_2$  based on the phase fractions determined from the diffraction patterns, until pure orthopyroxene-structured  $\text{MgGeO}_3$  was synthesised.

Phase relations of  $\text{MgGeO}_3$  at pressures less than  $\sim 1.5$  GPa were examined using a piston-cylinder cell (Boyd and England 1960) and above  $\sim 1.5$  GPa with a Walker type multi-anvil apparatus (Walker et al. 1990) at University College London. Multiple samples ( $\text{MgGeO}_3$  and pressure calibrants) were contained in capsules of double wrapped 25  $\mu\text{m}$  Inconel625 ( $\text{Ni}_{61}\text{Cr}_{22}\text{Mo}_9\text{Fe}_5$ ) foil, 3 mm long and 2.5 mm diameter; the samples were separated by disks of Inconel625.

Piston-cylinder experiments used  $\text{BaCO}_3$  pressure media with graphite furnaces (OD 7.7 mm, ID 6.2 mm) and the inner parts of the cell, surrounding the capsule, were made from  $\text{MgO}$ . The capsule was loaded centrally with respect to the length of the furnace. Temperature was monitored using a K-Type thermocouple which was 1 mm above the end of the sample capsule (2.5 mm above the centre of the furnace). A 1 mm thick corundum plate was placed between the sample capsule and the end of the thermocouple to prevent sample deformation.

The multi-anvil assembly used an 18 mm octahedral cell compressed by anvils with 11 mm corner truncations. The furnace comprised a 14 mm long double wrap of 25  $\mu\text{m}$  thick Inconel 625

158 foil, with a diameter of 4 mm, inside a zirconia sleeve with an outer diameter of 6.25 mm. The  
159 inner parts were made of MgO as a confining medium and were large enough to hold the 3 mm  
160 long by 2.5 mm in diameter capsule. Temperature was monitored at the end of the sample capsule  
161 using a D-type (W<sub>97</sub>Re<sub>3</sub>—W<sub>75</sub>Re<sub>25</sub>) thermocouple. Thermocouples were shielded from the furnace  
162 using thin walled corundum tubes with 0.8 mm outer diameter. The effects of pressure on  
163 thermocouple-emf were assumed to be negligible. The multi-anvil cell was calibrated using SiO<sub>2</sub>,  
164 Ni<sub>2</sub>SiO<sub>4</sub>, Fe<sub>2</sub>SiO<sub>4</sub> and Mg<sub>2</sub>GeO<sub>4</sub> phase relations, in the same experiments as the MgGeO<sub>3</sub> phase  
165 relations were determined. The pressure calibrations indicated a large decrease in pressure in the  
166 temperature range 870-1370 K. The average pressure depression was 1.9 GPa, relative to a Bi  
167 calibration curve measured at room temperature.

168         The conditions of the synthesis and phase relations experiments, as well as the run products  
169 are listed in Table 1. The phases in all recovered samples were identified by X-ray diffraction. The  
170 phase boundary positions were determined by the first occurrence of the high pressure phase and  
171 pass through the middle of the constraints. The experiments that constrain the position of the phase  
172 boundaries are noted in Table 1.

173         The samples of the high-pressure clinopyroxene-structured and ilmenite-structured  
174 MgGeO<sub>3</sub> for thermal expansion measurements were synthesised from the orthopyroxene phase  
175 using the same 18/11 multi-anvil cell. The clinopyroxene-structured MgGeO<sub>3</sub> was synthesised at  
176 2.5 GPa and 1173 K for 3 hours and MgGeO<sub>3</sub> ilmenite phase was synthesised at 5 GPa and 1373  
177 K for 3 hours. Each sample synthesis was recovered, ground in an agate pestle and mortar under  
178 isopropanol before analysis by X-ray powder diffraction to confirm phase purity. Several syntheses  
179 were required to produce the amount of material required for the thermal expansion measurements.

## 180 **Thermal Expansion Measurements by X-ray Powder Diffraction**



181 Ideally, all three samples would be measured by X-ray diffraction simultaneously, removing the  
182 possibility of any inconsistencies between measurements. However, there is a significant degree  
183 of peak overlap between the orthopyroxene phase and the other phases due to the large number of  
184 diffraction peaks in the orthopyroxene phase (see Figure 2). The orthopyroxene phase was  
185 therefore run on its own while the clinopyroxene and ilmenite phases were mixed together and  
186 measured at the same time. The samples used for the diffraction experiments were mixed with  
187 high purity MgO (Aldrich 99.99%). The MgO was fired overnight at 1073 K in air before mixing.  
188 The purpose of the MgO was to provide good constraints to the specimen transparency and  
189 displacement in the subsequent refinements – constraints that are not well provided by low  
190 symmetry materials.

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Insert Figure 2 here.

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193 Unit-cell volumes as a function of temperature were investigated by X-ray diffraction using  
194 the same method as previous studies (e.g. Hunt et al. 2017; Pamato et al. 2018). We used a  
195 PANalytical X'Pert Pro diffractometer, with Bragg–Brentano parafocusing reflection geometry.  
196 The cobalt X-ray source is monochromated, by a Ge (1 1 1) Johansson geometry focusing  
197 monochromator, to produce a Co  $K\alpha_1$  incident beam, the wavelength of which is assumed to be  
198 1.788996 Å (Hölzer et al. 1997). The X-ray tube was operated at 40 kV and 30 mA. Variable-  
199 width divergence and anti-scatter slits were used, together with a 10 mm wide incident-beam mask,  
200 so as to illuminate a 10 × 8.5 mm area of the sample. Both the incident and diffracted beams had  
201 0.04 radian Soller slits to reduce the axial divergences. The X-ray detector was an 'X'celerator'  
202 position-sensitive detector with an angular range of  $2\theta = \pm 1.061^\circ$  and an effective fixed step size

203 of 0.0167°. Data were collected over the  $2\theta$  range from 20° to 154.9°. Prior to the experiments  
204 reported here, the zero  $2\theta$  angle of the diffractometer was determined using an Si standard.

205         Diffraction data between 40 and 300 K were collected using an Oxford Cryosystems  
206 PheniX-FL low-temperature stage (Wood et al. 2018) and between 298 and 1173 K using an Anton  
207 Paar HTK1200N heated stage. The data were collected starting with the lowest temperature, in 20  
208 K steps for the low-temperature stage and in 25 K steps for the high-temperature stage. In the  
209 PheniX-FL cold stage, the sample was held in an atmosphere of helium exchange gas. The  
210 temperature was cooled as quickly as possible ( $\sim 3 \text{ K min}^{-1}$ ) to 80 K and then at  $1 \text{ K min}^{-1}$  to 40  
211 K. After equilibration for 10 min at 40 K the data were collected. Subsequent increases in  
212 temperature were at  $1 \text{ K min}^{-1}$  and the data were measured after approximately 15 min  
213 equilibration. The data collections were  $\sim 90$  min long. In the high-temperature stage, the sample  
214 was heated at  $5 \text{ K min}^{-1}$ , after which it was equilibrated for 10 min and the data collected in a time  
215 of  $\sim 90$  min.

216         Measurements in the high-temperature stage were made with the sample chamber open to  
217 the atmosphere. Data were collected in the orthopyroxene phase up to 1173 K. No additional  
218 diffraction peaks, produced by either back-transformation of the high pressure phases to  
219 orthopyroxene- $\text{MgGeO}_3$  or by reaction to form olivine- $\text{Mg}_2\text{GeO}_4$ , were observed in any of the high  
220 temperature measurements or in diffraction patterns taken from the samples after recovery to room  
221 temperature. The measurements of the clinopyroxene and ilmenite phases were stopped at 923 K  
222 when an additional diffraction peak appeared in the data.

223  
224 The data were analysed using the GSAS suite of programs (Larson and Von Dreele 2000; Toby  
225 2001), after conversion from variable to fixed divergence slit geometry by the diffractometer

226 manufacturer's X'pert HighScorePlus software package. The data was refined sequentially starting  
227 at with the data collected closest to room temperature. The initial model for each refinement was  
228 the converged fit of the preceding temperature, except for the data closest to room temperature  
229 which used the volumes and atomic coordinates from Yamanaka et al. (1985, 2005) as their starting  
230 point. Example diffraction patterns, from orthopyroxene and clinopyroxene + ilmenite, collected  
231 at 298 K are presented in Figure 2.

232 As has been previously found, the diffraction patterns of the clinopyroxene phase showed  
233 all the peaks consistent with the space group  $C2/c$  and none of the additional peaks indicative of  
234 the  $P2_1/c$  space group. The clinopyroxene (space group  $C2/c$ ) and ilmenite (space group  $R\bar{3}$ )  
235 phases were analysed using the Le Bail method (Le Bail et al. 1988). The Le Bail method produces  
236 less biased cell parameters when fitting data from samples that might not be a true random powder  
237 pattern but at the cost of freely adjusting the intensity of each sample peak. In each refinement, the  
238 lattice parameters for the two  $MgGeO_3$  phases (the triple-hexagonal cell was used for the ilmenite)  
239 and  $MgO$  (space group  $Fm\bar{3}m$ ) were refined. Also refined were the peak profiles, specimen  
240 displacement and transparency, which were constrained to be the same for each sample in the  
241 diffraction pattern. At each temperature the fit achieved convergence with the data.

242 The larger unit-cell and many overlapping low-intensity peaks in the orthopyroxene phase  
243 (space group  $Pbca$ ) prevent the Le Bail method from producing well constrained fits to the  
244 diffraction data. Therefore the Rietveld method of refinement was used (Rietveld 1969). All the  
245 atoms in each phase (orthopyroxene-structured  $MgGeO_3$  or  $MgO$ ) were constrained to have the  
246 same isotropic thermal parameters ( $U_{iso}$ ) and the two phases were constrained to have the same  
247 specimen transparency and displacement. Each pattern in the temperature series was initially fitted  
248 for the background coefficients and the unit cells of the  $MgO$  and orthopyroxene-structured

249 MgGeO<sub>3</sub> phases, whilst the peak profile parameters, atomic positions and thermal parameters of  
250 the orthopyroxene phase were fixed. When this refinement had converged these parameters were  
251 sequentially allowed to refine until all parameters were determined.

252 In the initial fits of the orthopyroxene phase data there was a correlation between the unit-  
253 cell parameters and the poorly constrained sample transparency. Therefore, the data were re-fitted  
254 with a fixed sample transparency. For the low-temperature stage data, the mean of the consistent  
255 transparencies was used. For the high-temperature stage data, the free-fit transparencies decreased  
256 with temperature, slowly below 773 K and more quickly at higher temperatures. These values were  
257 smoothed using two 1<sup>st</sup> degree polynomials which intersected at the change in slope in the  
258 transparency values. The data were subsequently re-refined using the transparency value of the  
259 polynomial at the relevant temperature.

260 In both 40 K diffraction patterns, reflections from  $\beta$ -nitrogen were observed. It was  
261 included in the refinement with space group *P6<sub>3</sub>/mmc*. Following the unphysical model of Vegard  
262 (1932), the N<sub>2</sub> molecules were assumed to be spherical and the size compensated for by unusually  
263 large thermal vibrations.

264 There is a small offset ( $<2$  in  $10^4$ ) in the lattice parameters and volumes between the high  
265 and low temperature diffraction stages. To compensate for this in the subsequent analysis the high  
266 temperature data were multiplied by a scale factor to align them with the low temperature data.  
267 The scaling was done by minimising the residuals of a third-degree polynomial passing through  
268 the 10 data between 220–398 K. The scaling factors are listed in Table 3. The scaling has minimal  
269 effect on the fitted thermophysical parameters; with and without scaling the data, they were all  
270 within two standard errors of each other and most parameters were within one standard error.

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## Results

### Phase Relations at high pressure and temperature

The pressure-temperature conditions at which we have synthesised the different phases of MgGeO<sub>3</sub> are presented in Figure 3a and Table 1. The phase boundaries constrained here are compared to those of other studies in

Figure 3b. We find that below ~ 900 K the reaction kinetics are slow and the reaction does not run to completion even after 20 hrs. This is consistent with the inferred experimental constraints of Kirfel and Neuhaus (1974) who report no experiments below 923 K. Above ~ 900 K reaction kinetics are fast, running to completion within 1 hour at 1123 K and 24 hours at 973 K.

Insert Figure 3 here.

Insert Table 1 here.

There is disagreement between studies as to which phase is the stable phase of MgGeO<sub>3</sub> between ~ 900 K and 1273 K at ambient pressure (Figure 1). Kirfel and Neuhaus (1974) find the orthopyroxene phase is stable in this temperature range (Figure 1a), whilst Ross and Navrotsky (1988) argue the clinopyroxene phase is the stable ambient pressure below 973 K (Figure 1b). Ozima and Akimoto (1985) argue the orthopyroxene phase is stable between 1083 and 1173 K, transforming to the clinopyroxene or protopyroxene phase at lower and higher temperatures respectively.

291           During our high-temperature stage X-ray diffraction measurements (up to 1173 K) we did  
292 not observe any sign of the orthopyroxene phase transforming to the clinopyroxene phase despite  
293 the sample being hotter than 900 K for over 34 hours and hotter than 1023 K for 22 hours by the  
294 end of the experiment. However, during the clinopyroxene and ilmenite phase X-ray diffraction  
295 measurements, a new diffraction peak appeared around 900 K; identification of which was  
296 prevented by termination of the measurement. No phase transformation is to be expected at  
297 temperatures less than 900 K based on the reaction kinetics inferred from the high -pressure  
298 experiments in this study. Therefore, the onset of reaction around 900 K in the clinopyroxene or  
299 ilmenite phase but not in the orthopyroxene phase indicates that the orthopyroxene phase is the  
300 stable phase at ~ 900 K and ambient pressure. Between 900 K and 1173 K, the orthopyroxene  
301 phase remains the stable phase on the basis that there is no sign of transformation during the X-  
302 ray diffraction measurements. This is consistent with the observations of Yamanaka et al (1985),  
303 who transformed the clinopyroxene phase to orthopyroxene at 1173 K in 1 hour but the  
304 orthopyroxene phase showed no signs of transformation after 24 hours at 973 K. The  
305 orthopyroxene phase is therefore shown to be the stable phase at ambient pressure at temperatures  
306 between 900 K and 1828±5 K. We note that it is not possible to determine if the orthopyroxene  
307 phase ceases to be the stable phase below 900 K because of retarded reaction kinetics.

308           Our experimental phase diagram is consistent with most of the previous experimental data  
309 (Figure 1) and like previous studies we find that ilmenite-structured MgGeO<sub>3</sub> is the high pressure  
310 phase. The clinopyroxene—ilmenite phase boundary is constrained to have a Clapeyron slope of  
311  $6.4 + 0.1/-0.6$  MPa/K and to intercept 0 GPa at 600 K. This is in better agreement with the  
312 experimentally constrained 5.45 MPa/K of Kirfel and Neuhaus (1974) (Table 2) than the value of  
313 1.9(4) MPa/K calculated by Ross and Navrotsky (1988) from calorimetry data. Our

314 clinopyroxene—ilmenite phase boundary is ~ 1GPa lower in pressure than that of Kirfel and  
315 Neuhaus (1974) (Figures 1a, 3b) which could be due to differences in pressure calibration between  
316 the studies.

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Insert Table 2 here.
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319 The orthopyroxene-clinopyroxene phase boundary occurs at <1 GPa, consistent with the  
320 experiments of Ringwood and Seabrook (1962) and Kirfel and Neuhaus (1974). Our experiments  
321 give a Clapeyron slope of  $-1.0 \pm 1.0 / -0.7$  MPa/K which intercepts 0 GPa at 1736 K. Given that the  
322 phase boundary is sub-parallel to the temperature axis, the value of 1736 K is very close to the 1  
323 atmosphere value of 1828 K determined by Robbins and Levin (1959). Our phase boundary is in  
324 excellent agreement with the study of Kirfel and Neuhaus (1974); who also have a negative but  
325 slightly shallower Clapeyron slope of -0.82 MPa/K but who fixed their 0 GPa intercept to be 1828  
326 K (Table 2, Figures 1 and 3). Our phase boundary would be in even better agreement with Kirfel  
327 and Neuhaus (1974) if we also fixed the 0 GPa intercept to 1828 K. These results contradict the  
328 studies of Ozima and Akimoto (1983) and Ross and Navrotsky (1988) who both argued for a  
329 positive Clapeyron slope.

330 Ozima and Akimoto (1983) report simultaneously flux growing orthopyroxene,  
331 clinopyroxene and ‘clinopyroxene from protopyroxene’ phases in the temperature range 928-1228  
332 K. Yamanaka et al. (1985) quenched the orthopyroxene phase from a flux at 1823 K and recovered  
333 the clinopyroxene phase by slow cooling from a flux at 1633 K. Both studies argue for  $C2/c$   
334 clinopyroxene as the low temperature phase and neither report observing a  $P2_1/c$  low-  
335 clinoenstatite phase. Ozima and Akimoto (1983) argue for the protoenstatite phase being stable

336 above 1173 K based on the recovery of two different styles of clinopyroxene phase from their flux.  
337 The incompatibility of these studies with our and other studies is possibly due to flux  
338 contamination compared to direct growth from oxides.

339 The phase boundaries in our experiments converge towards lower pressure and temperature  
340 and in the absence of other phases the triple point is at 0.98 GPa and 752 K. This is in good  
341 agreement with the triple point position of Kirfel and Neuhaus (1974) which is at 1.07 GPa and  
342 517 K.

343

#### 344 **Thermal Expansion**

345 The unit-cell volumes of the MgGeO<sub>3</sub> orthopyroxene, clinopyroxene and ilmenite phases are listed  
346 in Table 3 and plotted, as normalised values, in Figures 4, 5 and 6 respectively. The unit-cell  
347 parameters measured here at 298 and 300 K show a good correspondence to previously reported  
348 values at ambient conditions (Table 4) and have significantly lower standard errors. Our values for  
349 clino- and orthopyroxene phases are within the range of previous values. For the ilmenite phase  
350 our more precise values correspond reasonably to those of previous studies; our measured unit-  
351 cell volume and *c*-axis agree within 2 standard errors and the *a*-axis agrees to 5 standard errors of  
352 the values of Ashida et al (1985).

353

Insert Table 3 here.

354

Insert Figure 4 here.

355



Insert Figure 5 here.

356

Insert Figure 6 here.

357

Insert Table 4 here.

358

359           Only two other studies have reported high-temperature unit-cell parameters for the  
360 MgGeO<sub>3</sub> phases studied here: Yamanaka et al. (1985) and Ashida et al. (1985). Yamanaka et al.  
361 (1985) report the unit-cell parameters of clinopyroxene-structured MgGeO<sub>3</sub> at five temperatures  
362 between 298 and 1023 K and at 298, 1223 and 1373 K for the orthopyroxene-structured MgGeO<sub>3</sub>.  
363 Their unit cell parameters are always larger than those reported here (Figure 4 and Figure 5) and  
364 the difference between their data and ours increases with temperature. Conversely, the unit-cell  
365 volumes from Ashida et al. (1985) are always smaller than those measured here; again the  
366 difference increases with temperature. The increasing differences in unit cell parameter with  
367 temperature might imply that the differences come from inaccuracies in thermometry. We have  
368 calibrated our experimental apparatus using the melting point of gold (Pamato et al. 2018) and the  
369 thermal expansivity of silicon (Lindsay-Scott et al. 2007) and find excellent agreement with  
370 literature data for these two calibration methods. We therefore believe the present thermometry  
371 and unit-cell parameter measurements to be the most accurate of the three studies. If the differences  
372 between the present data and the previous studies are due to inaccuracies in thermometry it would  
373 require temperature errors in excess of 10% in the previous studies.

374 The Le Bail method does not give the atomic positions of the atoms within the unit-cell but  
375 the Rietveld refinement used for the orthopyroxene phase gives atomic positions which, while  
376 poorly constrained, are overall consistent with those of Yamanaka et al. (1985).

377  
378 **Volume-temperature models.** For many materials, at elevated temperatures (>298 K), the change  
379 in unit-cell volume and axes with temperature is well approximated by the integral over a low  
380 degree polynomial. This empirical form of thermal expansion has the formulation (Fei 1995):

$$l(T) = l_{T_{\text{ref}}} \exp \left[ \int_{T_{\text{ref}}}^T \alpha(T) dT \right] \quad 1$$

381 where  $l_{T_{\text{ref}}}$  is a unit-cell parameter (i.e. V, a, b, c) at a reference temperature,  $T_{\text{ref}}$ , and  $\alpha(T)$  is the  
382 thermal expansivity, the 2<sup>nd</sup> degree form of which is:

$$\alpha(T) = a_0 + a_1 T + a_2 T^{-2} \quad 2$$

383 Here, the reference temperature is taken as 298 K and values of  $l_{T_{\text{ref}}}$ ,  $a_0$ ,  $a_1$  and  $a_2$  for the unscaled  
384 high-temperature stage data in this study are listed in Table 5. Also listed are the equivalent  
385 parameters for the experimental data of Yamanaka et al. (1985) and Ashida et al. (1985) as well  
386 as calculations of Tsuchiya and Tsuchiya (2007). The data density of this study enabled utilisation  
387 of the 2<sup>nd</sup> degree form of the thermal expansivity but the more sparse data of the other studies mean  
388 it is only reasonable to report the 0<sup>th</sup> or 1<sup>st</sup> degree forms.

389

Insert Table 5 here.
----------------------

390

391 For the clinopyroxene phase, except for the  $\beta$  angle, the values of  $a_0$  derived from the study  
392 of Yamanaka et al. (1985) are more than one standard error greater than those in this study, which

393 is not unexpected given their larger unit-cell parameters. Only the volumetric thermal expansivity  
394 of the ilmenite-structured phase is calculable from the data of Ashida et al. (1985) and this is within  
395 error of our value at the reference temperature (298 K) but does not increase at the same rate as  
396 that of this study. Again this might be expected because of the smaller volumes measured in that  
397 study.

398 In the orthopyroxene phase, the axial thermal expansions occur in the same order as those  
399 of Yamanaka et al. (1985),  $\alpha_b > \alpha_c > \alpha_a$ , but with slightly different values. In the clinopyroxene  
400 phase both studies agree that  $\alpha_b > \alpha_a > \alpha_c$ . The order of these axial thermal expansions is the same  
401 as those in the equivalent polymorphs in MgSiO<sub>3</sub> (Hugh-Jones 1997).

402 For the ilmenite phase, the calculated thermal expansion of Tsuchiya and Tsuchiya (2007)  
403 is less than that measured here.

404  
405 At low-temperatures the thermal expansion does not follow a simple low-order polynomial form.  
406 Instead, the Debye model can be used to provide a theoretical description of the internal energy  
407 ( $U$ ) that, when combined with further theory, leads to a representation of the thermal expansion  
408 that has the observed low-temperature functional form. The internal energy predicted by the Debye  
409 model is (e.g. Poirier 2000):

$$U(T) = 9Nk_B T \left(\frac{T}{\theta_D}\right)^3 \int_0^{\theta_D/T} \frac{x^3}{\exp(x) - 1} dx \quad 3$$

410 where  $N$  is the number of atoms in the unit cell,  $k_B$  is Boltzmann's constant and  $\theta_D$  is the Debye  
411 temperature. The purely harmonic oscillators in this model do not predict thermal expansion and,  
412 therefore, this model has been combined with the Grüneisen approximations to form a zero  
413 pressure, quasi-harmonic, equation of state (Wallace 1998), in which the effects of thermal

414 expansion are considered equivalent to elastic strain, induced by thermal pressure. Further  
415 expansions to the internal energy model that account for intrinsic anharmonicity (Oganov and  
416 Dorogokupets 2004) have also been developed. Previous studies (e.g. Vočadlo et al. 2002;  
417 Lindsay-Scott et al. 2007) have shown that a second order Grüneisen approximation, combined  
418 with quasi-harmonic Debye internal energy model, provides a satisfactory description of data  
419 covering a wide range of temperatures, albeit sometimes additional modifications are required (e.g.  
420 Hunt et al. 2017; Pamato et al. 2018). It has the form (Wallace 1998):

$$V(T) = V_0 + \frac{V_0 U}{Q - bU} \quad 4$$

421 where

$$Q = V_0 K_0 / \gamma \quad 5$$

422 and

$$b = (K' - 1) / 2 \quad 6$$

423  $V(T)$  is the unit-cell volume at finite temperature,  $V_0$  is its 0 K value,  $K_0$  and  $K'$  are the bulk modulus  
424 and its first pressure derivative, and  $\gamma$  is a Grüneisen parameter, which in the derivation is assumed  
425 to be constant. In assuming that  $\gamma$  is independent of temperature it becomes equivalent to the  
426 thermodynamic Grüneisen parameter,  $\gamma_{\text{th}}$  (Wallace 1998); for a full explanation the reader is  
427 referred to Poirier (2000) and Wallace (1998). Expressions of the same functional form as Equation  
428 4 may also be used to describe the behaviour of the axes of the crystal but with different  
429 interpretation of the values of the parameters  $Q$  and  $b$  (Lindsay-Scott et al. 2007).

430 Equation 4 was fitted to the unit-cell parameter—temperature data by minimising the sum  
431 of the weighted least-squared residuals; the axial values were fitted as measured, rather than  
432 pseudo-cubic values. The standard errors of the measurements were used to calculate the weights.  
433 Unlike in our previous studies (Hunt et al. 2017; Pamato et al. 2018) the 40 K data point was not

434 additionally weighted because of greater scatter in the low-temperature unit-cell parameters of the  
435 orthopyroxene phase. The additional scatter is possibly due to the increased number peaks in the  
436 orthopyroxene unit-cell and the larger numbers of free parameters in the GSAS fitting. We expect  
437 this scatter to have only a minor effect on the subsequent analysis.

438         There is a suggestion of negative thermal expansion in orthopyroxene phase at the lowest  
439 temperatures, both in the axial and volumetric values. The 40 K data, where the negative thermal  
440 expansion may be observed, is also where we observe  $\beta$ -N<sub>2</sub> in the diffraction pattern. The data  
441 density and 40 K temperature limit of the PheniX-FL stage preclude fitting the data with multiple  
442 Debye or Einstein internal energy modes. We note that negative thermal expansion is observed at  
443 low temperatures in the germanate pyroxenes CoGeO<sub>3</sub> (Redhammer et al. 2010a) and  $P2_1/c$   
444 LiFeGe<sub>2</sub>O<sub>6</sub> (Redhammer et al. 2010b) but not in the silicate jadeite (Knight and Price 2008).

445         Table 6 lists the thermophysical parameters returned by the fit and the values derived from  
446 these. The fits are plotted in Figures 4, 5 and 6 as lines. The model provides a good fit to the data,  
447 except at perhaps the very lowest temperatures in the orthopyroxene phase. The residuals are  
448 slightly systematic with temperature but higher and lower order Grüneisen approximations result  
449 in worse fits. There is not sufficient temperature range in the data to adequately constrain  
450 anharmonic internal energy models (Oganov and Dorogokupets 2004).

451         For each phase, the volumetric Debye temperature is bracketed by the axial Debye  
452 temperatures and, for each phase, the  $a_0$  values (Equation 2, Table 5) are proportional to  $1/Q$  (Table  
453 6). The Debye temperatures reported from heat capacity by Ashida et al. (1985) for ilmenite phase  
454 (Table 6) are larger than we calculate here, whereas the Debye temperatures calculated for  
455 clinopyroxene and ilmenite phases from the sound velocity data of Liebermann (1974) are slightly  
456 smaller. The acoustic Debye temperature has been found to be up to ~80 K lower than the thermal

457 Debye temperature in pyroxene structured phases (Yang and Ghose 1994) and it would appear that  
458 same is true for ilmenite structured phases.

459

Insert Table 6 here.

460  
461 The values of  $b$  imply volumetric  $K'$  values (Table 6) significantly greater than the 3.6  
462 measured in ilmenite-structured  $\text{MgGeO}_3$  (Ashida et al. 1985) and the 4.59 calculated for ilmenite-  
463 structured  $\text{MgGeO}_3$  (Tsuchiya and Tsuchiya 2007). They are also greater than the values of  $K'$   
464 typically reported for oxides and silicate minerals ( $3 < K' < 8$ , Knittle 1995). The values of  $K'$  in  
465  $\text{MgSiO}_3$  pyroxenes have been measured to be greater than 5.6(29) (Angel and Hugh-Jones 1994;  
466 Angel and Jackson 2002) but no values exist for  $\text{MgGeO}_3$  pyroxene-structured phases.

467 The Grüneisen parameter ( $\gamma$ , Equation 5) is derived using the bulk modulus. The bulk  
468 modulus of ilmenite-structured  $\text{MgGeO}_3$  has been measured to be 187(2) GPa (Sato et al. 1977),  
469 195 GPa (Ashida et al. 1985) and 180(2) GPa (Yamanaka et al. 1985). The bulk modulus of  
470 orthopyroxene-structured  $\text{MgGeO}_3$  is 115 GPa (Kandelin et al. 1983, Ross and Navrotsky 1988)  
471 and Liebermann (1974) measured  $K_s = 131$  GPa at 0.75 GPa pressure in the clinopyroxene phase.  
472 The adiabatic bulk modulus ( $K_s$ ) is related to the isothermal Bulk modulus ( $K_T$ ) by:

$$K_S = K_T(1 + \gamma_{th}\alpha T) \quad 7$$

473 Assuming that  $K_T$  is isothermal and equal to  $K_0$  and that  $\gamma_{th}$ , the thermodynamic Grüneisen  
474 parameter is equal to  $\gamma$  (Equation 5, Wallace 1998), it is possible to solve for  $\gamma$  using a value of  $K_s$ .

475 Using the values above and for the clinopyroxene phase Equation 7, we calculate  $\gamma$  to be  
476 0.86, 1.01 (at 300 K) and 1.23 for the orthopyroxene, clinopyroxene and ilmenite phases  
477 respectively (Table 6). The effect of converting  $K_s$  to  $K_T$  for the clinopyroxene phase is minimal;

478 calculating  $\gamma$  directly using  $K_s$  yields a value of 1.02(1). For ilmenite, Tsuchiya and Tsuchiya  
479 (2007) used density functional theory to calculate a value for  $\gamma$  of 1.24, which is very close to our  
480 value.

481  
482 The thermal expansivity ( $\alpha$ ) is the derivative of the unit-cell parameter with respect to temperature:

$$\alpha(T) = \frac{1}{l(T)} \left( \frac{dl}{dT} \right) \quad 8$$

483 where  $l$  is the unit cell length (for axial expansivities) or volume (for volumetric expansivities).  
484 The point-to-point volumetric thermal expansivity of the  $\text{MgGeO}_3$  phases is plotted in Figure 7,  
485 together with the thermal expansion calculated from the polynomial and Debye-Grüneisen models.  
486 At low temperatures the orthopyroxene phase has a greater thermal expansion than the  
487 clinopyroxene or the ilmenite phases, which is reflected in the lower Debye temperature in this  
488 phase (Table 6). Consistent with the polynomial thermal expansion coefficients and the  $b$ -values  
489 of the Debye model, at elevated temperatures, the thermal expansivity of the orthopyroxene phase  
490 increases with temperature significantly slower than that of the other two phases.

491

Insert Figure 7 here.

492

### 493 **Heat Capacity**

494 The Debye model of internal energy also can be used to calculate a heat capacity. The isochoric or  
495 volumetric heat capacity is the change in internal energy ( $U$ ), at constant volume, with temperature:

$$C_V = \left( \frac{\partial U}{\partial T} \right)_V \quad 9$$

496 The molar heat capacity derived from the Debye model for internal energy (Equation 3) is  
497 therefore:

$$C_V = 9nN_A k_B \left(\frac{T}{\theta_D}\right)^3 \int_0^{\theta_D/T} \frac{x^4}{(\exp(x) - 1)^2} dx \quad 10$$

498 where  $n$  is the number of atoms per formula unit and  $N_A$  is Avogadro's number. Equation 10 uses  
499 Debye theory and so at high-temperatures asymptotes to the classical limit  $3Nk_B$ , known as the  
500 Dulong-Petit law. The volumetric heat capacity (Equation 10) can be obtained using the Debye  
501 temperatures calculated in the fit to the thermal expansion from Equations 4 and 7.

502 Whilst this can sometimes be preferred, this limit does not take into account factors such  
503 as anharmonicity, or disordering (including defect formation), which are important because they  
504 can raise the heat capacity of a material beyond the classical limit. Therefore, an alternative  
505 formulation of the heat capacity is to use the coefficient of thermal expansion, the thermodynamic  
506 Grüneisen parameter ( $\gamma_{th}$ ) and the incompressibility ( $K$ ) to calculate  $C_V$  as (Poirier 2000):

$$C_V = \frac{\alpha KV}{\gamma_{th}} \quad 11$$

507 This formulation contains no asymptotic limit and so may represent a more realistic model under  
508 high-temperature conditions. If it is again assumed that  $\gamma_{th} = \gamma$  and that  $K$  is athermal (i.e.  $K = K_0$ ),  
509 Equation 5 may be used to derive an expression for  $C_V$  without requiring the bulk modulus or a  
510 Grüneisen parameter:

$$C_V = \frac{\alpha V Q}{V_0} \quad 12$$

511 Neither value for  $C_V$  is directly comparable to experimental measurements of heat capacity which  
512 are made at constant pressure. The constant pressure or isobaric heat capacity ( $C_P$ ) is related to the  
513 volumetric heat capacity as follows:



$$\left(\frac{C_p}{C_v}\right) = 1 + \gamma_{th}\alpha T \quad 13$$

514 Again, by assuming the Grüneisen parameters in Equations 5, and 11, 13 are the same  $C_p$  is  
515 calculable.

516 Figure 8 plots the calculated isobaric heat capacities for the three phases studied here along  
517 with previous data. At temperatures greater than  $\sim 300$  K the heat capacities derived using equation  
518 12 become significantly greater than those from the simple Debye approximation (Equation 10).  
519 Our calculated values for the heat capacity derived directly from the Debye model (i.e. via  
520 Equation 10) are greater than the measured values of Ashida et al. (1985). At higher temperatures,  
521 though, the calculated heat capacities of Ross and Navrotsky (1988) are slightly smaller than our  
522 Debye-derived values at 298 K and are very close above 600 K. Below  $\sim 850$  K, Ross and  
523 Navrotsky (1988) and our Debye heat capacities agree that  $C_p(\text{clinopyroxene-structured MgGeO}_3)$   
524  $> C_p(\text{orthopyroxene-structured MgGeO}_3) > C_p(\text{ilmenite-structured MgGeO}_3)$ . The heat capacities  
525 derived from Equation 12 also have this condition below  $\sim 485$  K. Above 635 K, these heat  
526 capacities have  $C_p(\text{clinopyroxene-structured MgGeO}_3) > C_p(\text{ilmenite-structured MgGeO}_3) >$   
527  $C_p(\text{orthopyroxene-structured MgGeO}_3)$ , with the lower orthopyroxene phase heat capacity related  
528 to the smaller thermal expansion of the orthopyroxene phase relative to the other phases. This  
529 concords with the observations showing that, at less than 25 K in  $\text{MgSiO}_3$ , the heat capacity of  
530 clinopyroxene is greater than orthopyroxene (Drebushchak et al. 2008). It is clear from Figure 8  
531 that the available experimental heat capacity data seem better matched by values we have obtained  
532 via Equation 10 and that the approximations inherent in Equation 12 may, therefore, be leading to  
533 an overestimate of  $C_p$ . However, Fiquet et al. (1992), have reported experimental values of  $C_p$  for  
534 other germanates at high temperature that exceed the value from the Dulong-Petit law by similar

535 amounts to that shown in Figure 8, e.g. by up to 13% for olivine-structured  $Mg_2GeO_4$  and 30% for  
536 olivine-structured  $CaMgGeO_4$ .

537

Insert Figure 8 here.

538

### 539 **Thermodynamic Clapeyron slope**

540 The Clapeyron slope of a phase boundary is given by:

$$\frac{dP}{dT} = \frac{\Delta S}{\Delta V} \quad 14$$

541 where  $\Delta S$  and  $\Delta V$  are the entropy and volume change across the phase transition. The entropy of a  
542 mineral is related to its heat capacity by:

$$\left(\frac{\partial S}{\partial T}\right)_P = \frac{C_P}{T} \quad 15$$

543 and, therefore, the thermophysical parameters calculated above are sufficient to estimate the  
544 Clapeyron slope of the boundary - provided that an adequate representation of  $C_P$  can be obtained  
545 from the thermal expansion measurements. For any set of isobaric measurements, the result will  
546 be the value of the Clapeyron slope as a function of temperature along the isobar. In the present  
547 work, the thermal expansion measurements have all been made at ambient pressure and so the  
548 calculated slope will be that of the phase boundary at the point where it crosses the temperature  
549 axis in Figure 3. Thermodynamics requires that the Clapeyron slope must fall to zero as  $T$   
550 approaches 0 K; for finite temperatures, in cases where the Clapeyron slope shows a strong  
551 temperature dependence, it then will be necessary to know the temperature of the phase boundary  
552 to calculate the correct value of the Clapeyron slope. As a result, the method described here  
553 provides only a check on the correctness of the slopes of phase boundaries, rather than a method

554 for determining their absolute position in P-T space. The Clapeyron slopes, calculated using both  
555 representations of the heat capacities (Equations 10 and 12) for each phase transition, are plotted  
556 in Figure 9. The Clapeyron slopes at the 0 GPa intersection temperature are listed in Table 2.

557

Insert Figure 9 here.

558

559 Below ~300 K, there is little difference between Clapeyron slopes however they are  
560 calculated but above this temperature the possible Clapeyron slopes show a wider range depending  
561 on the assumed heat capacity (Figure 9). At high temperatures, Clapeyron slopes calculated with  
562 heat capacities directly from the Debye model (Equation 10) show asymptotic behaviour which is  
563 not present in Clapeyron slopes calculated using Equation 12. This is because the heat capacity  
564 from the Debye model asymptotes to the classical limit. With Debye-model heat capacities, the  
565 Clapeyron slope of the orthopyroxene-clinopyroxene phase boundary is always positive but when  
566 Equation 12 is used the Clapeyron slope is negative at temperatures greater than 1665 K (Figure  
567 9a). At 1736 K, where our orthopyroxene-clinopyroxene phase boundary intersects 0 GPa, the  
568 calculated Clapeyron slope is -1.8 MPa/K. For the clinopyroxene-ilmenite phase transition, the use  
569 of heat capacities obtained from Equation 12 increases the calculated Clapeyron slope but it  
570 remains somewhat less than the experimental values of either Kirfel and Neuhaus (1974) or that  
571 found in the present work (Figure 9b). However, when it is considered that these Clapeyron slopes  
572 are calculated from 0 GPa data and that there are a number of simplifying assumptions in the  
573 calculations, we believe that they strongly support conclusion that our phase boundaries as  
574 determined by direct high-P/T experiments are correct.

575

576

## Discussion

577 In this study, the experimental phase diagram is in excellent agreement with that of Kirfel and  
578 Neuhaus (1974). Both studies find a negative Clapeyron slope for the orthopyroxene-  
579 clinopyroxene phase boundary and the orthopyroxene-clinopyroxene-ilmenite phase triple point  
580 close to 0.98 GPa and 752 K. This topology is different from the MgSiO<sub>3</sub> system, in which the  
581 Clapeyron slope of the orthopyroxene-clinopyroxene transition is positive and there is no  
582 equivalent triple point. There is no evidence for a low-temperature *P2<sub>1</sub>/c* clinopyroxene-structured  
583 MgGeO<sub>3</sub> phase and it is not required by the phase diagram. However, it may be kinetically  
584 inhibited from forming below ~900 K but this seems unlikely given that in MgSiO<sub>3</sub> the *P2<sub>1</sub>/c*  
585 clinopyroxene forms spontaneously during decompression of the *C2/c* clinopyroxene. The phase  
586 diagram is also consistent with orthopyroxene-structured MgGeO<sub>3</sub> transforming directly to the  
587 *C2/c* clinopyroxene phase above 1828 K at 0 GPa. There is no significant break in slope of the  
588 orthopyroxene-clinopyroxene phase boundary (Figure 1a) that may indicate the presence of the  
589 protopyroxene (*Pbcn*) structure argued for by Ozima and Akimoto (1983). Although its presence  
590 cannot be excluded, if it does exist, it is likely to have only a small narrow, high temperature,  
591 stability field.

592 Our experimental phase diagram, and that of Kirfel and Neuhaus (1974), are consistent  
593 with the thermophysical property measurements if the heat capacity exceeds the Dulong-Petit limit  
594 (Equation 12). In this case the heat capacity estimated for the clinopyroxene phase increases faster  
595 than that of the orthopyroxene phase (Figure 8). The associated relative increase in the entropy of  
596 the clinopyroxene phase at high temperatures changes the Clapeyron slope (Equation 14),  
597 expanding the phase field at the expense of the orthopyroxene phase. Although the heat capacity  
598 of ilmenite-structured MgGeO<sub>3</sub> (Figure 8) is consistent with the Dulong-Petit limit, and would thus

599 have negligible anharmonicity, assuming heat capacities calculated using Equation 12 leads to  
600 Clapeyron slopes that correspond better to the phase boundaries determined by direct experiment.  
601 There are no measurements of heat capacity in germanate pyroxenes with which we can compare  
602 our estimates, but heat capacities well above the Dulong-Petit limit have been observed in  
603 germanate olivines (Fiquet et al. 1992) and significant anharmonicity has been observed in  
604 germanate perovskites (Andrault et al. 2015). The prevalence of anharmonicity more generally in  
605 germanate oxides points to it being present in the phases here and the cause of their apparently  
606 increased heat capacity. We note, though, that other processes can cause increases in heat capacity  
607 at high temperature, e.g. defect formation (Pamato et al., 2018).

608         Whatever is assumed about the heat capacity, for the clinopyroxene-ilmenite phase  
609 transition, our results are in broad agreement with those of Ross and Navrotsky (1988). Our phase  
610 diagram and interpretation of the thermophysical properties, though, disagrees with their results  
611 for the orthopyroxene-clinopyroxene phase boundary.

612         Although anharmonicity and/or significant formation of defects in clinopyroxene-  
613 structured  $\text{MgGeO}_3$  can reconcile the experimental observations, there is another factor that may  
614 explain the discrepancies, between studies, namely an extremely small transformation shear-stress  
615 between clinopyroxene and orthopyroxene-structured phases. The possibility that the low-pressure  
616 extent of the clinopyroxene phase field is controlled by shear-stress controlled back-transformation  
617 has been presented before (Ringwood and Seabrook 1962) and has been observed during  
618 deformation of natural pyroxenes (Coe and Kirby 1975). It is unlikely, though, that this effect has  
619 any effect on the phase diagram here because (a) our phase boundary is in almost perfect agreement  
620 with that of Kirfel and Neuhaus (1974) and (b) the clinopyroxene-structured  $\text{MgGeO}_3$  samples

621 were ground prior to the thermal expansion measurements and no back transformation was  
622 observed in the sample.

623

### 624 **Implications**

625 Based on the evidence in this study, we suggest that  $\text{MgGeO}_3$  polymorphs and the clinopyroxene  
626 in particular may be significantly more anharmonic than the equivalent  $\text{MgSiO}_3$  polymorphs. This  
627 observation unifies the experimental observations of the phase boundaries; however, there are  
628 currently no heat capacity measurements against which to test this assertion. Therefore, to fully  
629 understand the system more thermophysical property measurements and/or *in-situ* experiments  
630 must be performed.

631 The presence of potentially significant anharmonicity in clinopyroxene-structured  
632  $\text{MgGeO}_3$ , magnesium germanates (e.g. Fiquet et al. 1992) and germanates more generally (e.g.  
633 Andrault et al. 2015), is a major difference between the germanate and silicate systems. These  
634 effects must therefore be considered when using germanates as low-pressure analogues for Earth  
635 forming minerals, especially at elevated temperatures.

636

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799 *Figure 1. Phase diagrams and synthesis conditions in the MgGeO<sub>3</sub> system from previous studies. Large symbols are*  
800 *data from (a) Kirfel and Neuhaus (1974) (b) Ross and Navrotsky (1988) and (c) Ozima and Akimoto (1983). Small*  
801 *symbols in (b) are synthesis conditions reported by Roth (1955); Ringwood and Seabrook (1962); Liebermann*  
802 *(1974); Sato et al. (1977); Kirfel et al. (1978); Ito and Matsui (1979); Ashida et al. (1985); Okada et al. (2008).*  
803 *Symbols: □ – orthopyroxene, ○ – clinopyroxene, Δ - ilmenite, ▽ – protopyroxene; open – forward and filled -*  
804 *reversals, except data of Ross and Navrotsky (1988) (b) where all experiments started with both clinopyroxene and*  
805 *ilmenite so are both forward and reverse. Solid lines are phase boundaries in reference, dotted lines in (b), (c) are*  
806 *the clinopyroxene–ilmenite phase boundary of Kirfel and Neuhaus (1974) for comparison.*

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809 *Figure 2. Sample X-ray powder diffraction patterns showing the observations (points), calculated fit (upper*  
810 *line) and difference (lower trace) for (a) clinopyroxene and ilmenite MgGeO<sub>3</sub> and (b) orthopyroxene MgGeO<sub>3</sub> at*  
811 *298 K. The tick marks show the positions of the Bragg reflections from (a) clinopyroxene, MgO and ilmenite (top to*  
812 *bottom) and (b) MgO and orthopyroxene.*

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815 *Figure 3. (a) MgGeO<sub>3</sub> phase diagram from the experiments in this study and (b) comparison of our phase*  
816 *boundaries (bold solid line) with those of Kirfel and Neuhaus (1974; solid line), Ozima and Akimoto (1983; dashed*  
817 *line) and Ross and Navrotsky (1988; dotted line). Symbols: □ – orthopyroxene, ○ – clinopyroxene, Δ - ilmenite;*  
818 *open – forward and filled – reversals. Small symbols denote experiments that were kinetically inhibited. Coloured*  
819 *regions in (b) are the extent of the phase fields constrained by our experiments*

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822 *Figure 4: (top) Lattice parameters, unit-cell volumes and (bottom) residuals to fit of orthopyroxene MgGeO<sub>3</sub> (Table*  
823 *3), normalised to the 0 K values reported in Table 5. The error bars are omitted for clarity and the high-temperature*  
824 *measurements are multiplied by the scale factor. Lines are normalised fits of the Debye-Grüneisen model to the data*  
825 *of this study. Filled symbols are the data from Yamanaka et al. (1985).*

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828 *Figure 5. (top) Lattice parameters, unit-cell volumes and (bottom) residuals to fit of clinopyroxene MgGeO<sub>3</sub> (Table*  
829 *3), normalised to the 0 K values reported in Table 5. The error bars are omitted for clarity and the high-temperature*  
830 *measurements are multiplied by the scale factor. Lines are normalised fits of the Debye-Grüneisen model to the data*  
831 *of this study. Filled symbols are the data from Yamanaka et al. (1985).*

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834 *Figure 6. (top) Lattice parameters, unit-cell volumes and (bottom) residuals to fit of ilmenite MgGeO<sub>3</sub> (Table 3),*  
835 *normalised to the 0 K values reported in Table 5. The error bars are omitted for clarity and the high-temperature*  
836 *measurements are multiplied by the scale factor. Lines are normalised fits of the Debye-Grüneisen model to the data*  
837 *of this study. Filled symbols are the data from Ashida et al. (1985).*

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840 *Figure 7. Volumetric thermal expansivity of MgGeO<sub>3</sub> phases. Lines and symbols as in the legend.*

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842 *Figure 8. Calculated isobaric heat capacities of MgGeO<sub>3</sub> phases, converted from isovolumetric heat capacities*  
843 *(equation 13). Lines and symbols as in the legend. The Horizontal dashed grey line is the Dulong-Petit limit for the*  
844 *heat capacity at constant volume.*

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847 *Figure 9. Clapeyron slopes of the (a) orthopyroxene-clinopyroxene and (b) clinopyroxene-ilmenite phase*  
848 *transitions. Lines and symbols as in the legends.*

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Expt.	Type	Pressure	Temperature	Duration	Sample	
		(GPa)	(K)	(hours)	starting	recovered
synthesis	furnace	1 bar	1373	72	oxides	opx
synthesis	MA	2.5	1173	3	opx	cpx
synthesis	MA	5.0	1273	3	opx	ilm
GER001	MA	2.1	1273	2	opx	cpx
GER003	MA	3.6	1173	3	opx	cpx
GER004	MA	2.6	1273	1.33	opx	cpx
GER005	MA	3.3	1273	1.5	opx	cpx
GER006	MA	4.6	1273	1.5	opx	ilm
GER007	MA	1.3	1073	1.5	opx	cpx+opx
GER008\$	MA	2.0	823	3	opx	cpx
GER011\$	MA	2.6	823	20	opx	cpx
GER012	MA	3.9	1273	3	opx	cpx
GER013#	MA	4.2	1273	3	opx	cpx
GER014	MA	3.1	1123	3	opx	cpx
GER015\$	MA	3.0	823	3	opx	cpx
GER017	MA	3.6	1123	3	opx	1:9 cpx:ilm
GER018#	MA	3.1	1073	1	opx	9:1 cpx:ilm
GER019	MA	2.9	1073	3	opx	cpx
GER020#	MA	3.3	1123	1	opx	cpx
GER022	MA	6.1	1523	2	opx	ilm
GER024#	MA	5.6	1473	2.66	opx	ilm
GER025	MA	5.4	1493	3	opx	cpx
GER028	MA	2.5	973	3	opx	cpx+ilm
GER029#	PC	0.6	973	24	opx	opx
GER031#	PC	0.9	973	24	opx	cpx
GER032	PC	0.9	1323	4.5	opx	cpx
GER033	PC	1.2	1323	3	opx	cpx
GER035#	PC	0.3	1323	17.5	opx	opx
GER037#	PC	0.6	1273	3	opx	cpx+opx
GER042	MA	5.0	1273	3	opx	ilm
GER043	MA	5.0	1273	3	opx	ilm
GER044	MA-R	3.0	1123	3.5	ilm	ilm+cpx
GER046	MA-R	3.9	1273	3.5	ilm	ilm+cpx

850 *Table 1. Experimental conditions and outcomes used here to synthesise samples and*  
 851 *constrain the phase diagram. PC – piston cylinder experiment, MA – multi-anvil, MA-R –*  
 852 *reversal. Samples: opx – orthopyroxene, cpx – clinopyroxene and ilm – ilmenite. The*  
 853 *experiments that are used to calculate the phase boundary positions are denoted by # and those*  
 854 *inferred to be kinetically inhibited are denoted by \$.*

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Author	Type	Clapeyron Slope (MPa/K)	$\Delta S$ (J/K/mol)
<i>Orthopyroxene – clinopyroxene</i>			
This study	Experimental	$-1.0^{+1.0}_{-0.7}$	1.17 <sup>a</sup>
Kirfel and Neuhaus (1974)	Experimental	-0.82	0.96 <sup>a</sup>
Ozima and Akimoto (1983)	Experimental	28	-32.69 <sup>a</sup>
Ross and Navrotsky (1988)	Calorimetry	2.1	-3
This study	Calculated (Debye, 1736 K)	16.0(0) <sup>b</sup>	-15.67(1)
This study	Calculated ( $\alpha T$ , 1736 K)	-1.76(0) <sup>b</sup>	1.72(6)
<i>Clinopyroxene – ilmenite</i>			
This study	Experimental	$6.4^{+1.0}_{-0.6}$	-23.55 <sup>a</sup>
Kirfel and Neuhaus (1974)	Experimental	5.45	-19.98 <sup>a</sup>
Ross and Navrotsky (1988)	Calorimetry	2.0(4)	-7.6(15)
This study	Calculated (Debye, 600 K)	2.8(0) <sup>b</sup>	-10.34(7)
This study	Calculated ( $\alpha T$ , 600 K)	3.1(0) <sup>b</sup>	-11.53(9)

857 Table 2. Clapeyron slopes of the phase boundaries reported here and in previous studies. The Calculated Clapeyron  
858 slopes are reported at the temperature of the experimental phase boundary at 0 GPa; for the temperature evolution  
859 of the calculated Clapeyron slopes see Figure 9. The heat capacities used to calculate the Clapeyron slope were  
860 either the Debye heat capacity (Equation 10) or the anharmonic heat capacity ( $\alpha T$ , Equations 11, 12); see text for  
861 details. a.  $\Delta S$  values are calculated using the  $\Delta V$  from the thermal expansion measurements in this study at 900 K. b.  
862 Errors are  $<5 \times 10^{-4}$  but greater than zero.



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*and Rietveld fits using GSAS; for further discussion of the importance of systematic errors in profile refinement of X-ray powder data see e.g. Thompson and Wood (1983).  
Multiplying the high-temperature data by the scale factors aligns it with the low-temperature data; see text for details. a. Errors are  $<5 \times 10^{-4}$  degrees but greater than zero.*

	Reported temperature	V (Å <sup>3</sup> )	a (Å)	b (Å)	c (Å)	β (°)
<i>Orthopyroxene</i>						
This study	298K	900.48(2)	18.8149(2)	8.9570(1)	5.34329(7)	
	300K	900.62(2)	18.8153(3)	8.9571(2)	5.34392(9)	
Roth (1955, 1957)	-	-	18.661	8.954	5.346	
Lyon (1968)	-	-	18.745(5)	8.974(17)	5.350(1)	
Lindemann (1961)	-	-	18.649(7)	8.902(5)	5.332(5)	
Kirfel and Neuhaus (1974)	-	901.17(35)	18.859(6)	8.958(2)	5.334(1)	
Ozima (1983) & Ozima and Akimoto (1983)	-	899.69	18.8099(12)	8.9484(8)	5.3451(4)	
Yamanaka et al. (1985)	20C	901.3(3)	18.829(3)	8.952(2)	5.347(1)	
<i>Clinopyroxene</i>						
This study	298K	434.755(6)	9.60388(6)	8.93706(6)	5.15955(4)	100.969(0) <sup>c</sup>
	300K	434.695(4)	9.60374(4)	8.93675(4)	5.15925(3)	100.978(0) <sup>c</sup>
Kirfel and Neuhaus (1974)	-	440.0(5)	9.623(9) <sup>a</sup>	8.960(5)	5.196(3)	100.144(8)
Ozima (1983) & Ozima and Akimoto (1983)	-	434.27(6)	9.6010(8)	8.9323(6)	5.1592(5)	101.034(9)
Yamanaka et al. (1985)	20C	435.1(1)	9.605(2)	8.940(2)	5.160(1)	100.95(1)
<i>Ilmenite</i>						
This study	298K	290.057(4)	4.93703(3)		13.74110(12)	
	300K	290.013(3)	4.93671(2)		13.74081(9)	
Ringwood and Seabrook (1962)	-	-	4.936(5)		13.76(5)	
Kirfel and Neuhaus (1974)	-	289.8	4.936(1)		13.735(5)	
Kirfel et al. (1978)	-	289.80(14)	4.933(1)		13.734(5)	
Ito and Matsui (1979)	“ambient conditions”	-	4.9363(2)		13.7401(8)	
Ashida et al. (1985)	273K	289.93	-		-	
Tsuchiya and Tsuchiya (2007)	300K	291.48 <sup>b</sup>	4.949 <sup>b</sup>		13.743 <sup>b</sup>	

867 *Table 4. Unit cell parameters from previous studies at ambient conditions. Values not reported are marked ‘-’,*  
 868 *numbers in parenthesis are the standard error in the last digit and numbers without parenthesis did not have*  
 869 *reported errors. All unit cell parameters were measured by powder or single-crystal X-ray diffraction unless*  
 870 *otherwise noted. See note in caption of Table 3 regarding significance of values determined in this study. <sup>a</sup> value*  
 871 *was reported as twice this value. <sup>b</sup> calculated using density functional theory. <sup>c</sup> Error is <math>5 \times 10^{-4}</math> degrees but greater*  
 872 *than 0.*

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	$l_{298K}$	$a_0 (\times 10^{-6} K^{-1})$	$a_1 (\times 10^{-9} K^{-2})$	$a_2 (K)$
This study				
Orthopyroxene volume	900.47402 Å <sup>3</sup>	23.46(4)	9.65(7)	-0.169
a	18.81493(19) Å	7.01(9)	2.22(12)	-0.107
b	8.95700(7) Å	8.45(7)	4.52(9)	0.034
c	5.34326(6) Å	7.99(10)	2.92(14)	-0.096
Clinopyroxene volume	434.75207 Å <sup>3</sup>	25.79(15)	11.73(29)	-0.456
a	9.60388(5) Å	9.89(9)	4.15(16)	-0.157
b	8.93700(5) Å	11.53(9)	4.19(16)	-0.270
c	5.15954(3) Å	6.01(11)	4.28(18)	-0.068
β	100.96923 °	4.95(4)	2.22(7)	-0.120
Ilmenite volume	290.05569 Å <sup>3</sup>	23.32(20)	11.66(39)	-0.429
a	4.93707(4) Å	6.84(13)	3.89(21)	-0.122
c	13.74079(14) Å	9.58(16)	3.93(27)	-0.179
Yamanaka et al. (1985)				
Orthopyroxene volume	901.476(64) Å <sup>3</sup>	39.8(1)		
a	18.830(3) Å	10.1(3)		
b	8.953(2) Å	15.9(3)		
c	5.347 Å	13.7(2)		
Clinopyroxene volume	435.433 Å <sup>3</sup>	43.8(132)	-13.4(233)	
a	9.608(5) Å	16.7(67)	-5.1(104)	
b	8.944(7) Å	18.4(100)	-5.9(155)	
c	5.161(2) Å	7.7(54)	4.4(80)	
β	100.968(36) °	7.9(52)	-1.6(84)	
Ashida et al. (1985)				
Ilmenite volume	289.816 Å <sup>3</sup>	25.7(44)	-2.3(74)	
Tsuchiya and Tsuchiya (2007)				
Ilmenite volume	291.48 Å <sup>3</sup> a	20.9 a		

875 *Table 5. Polynomial thermal expansion parameters derived from the high-temperature stage data in Table 3,*  
876 *together with selected other data. Numbers in parenthesis are the standard error in the least significant digit;*  
877 *numbers without errors have a standard error of < 0.5 least significant digit. a – 0 GPa, 300 K values calculated*  
878 *using density functional theory.*

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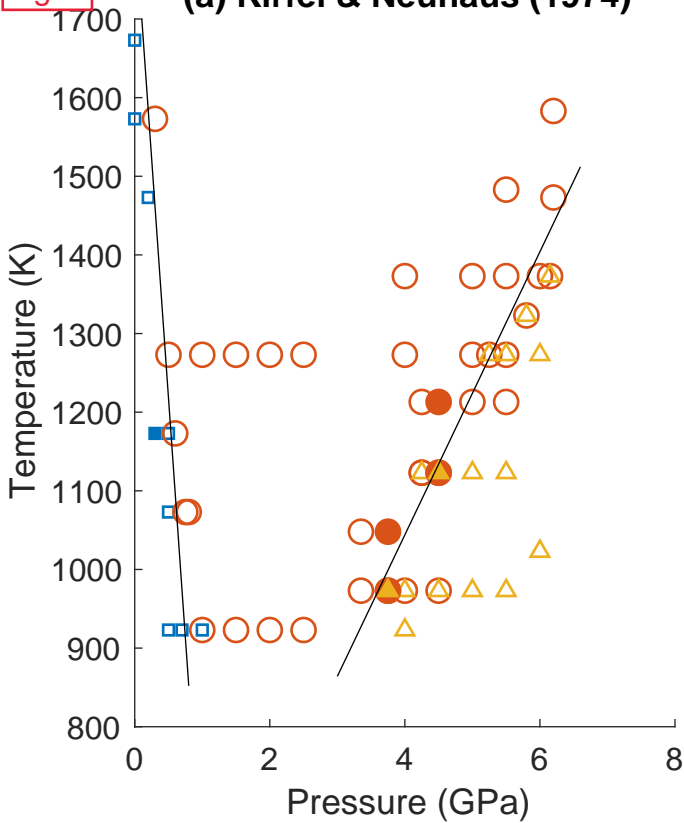
		$l_0$	$\theta_D$ (K)	$Q$ ( $\times 10^{-16}$ J)	$b$	$K'$	$\gamma$
This study, thermal expansion							
Orthopyroxene	volume	897.299(16) Å <sup>3</sup>	602(7)	1.199(6)	4.6(1)	10.3(3)	0.861(5)a
	a	18.79709(21) Å	650(16)	4.22(5)	14(1)		
	b	8.94334(11) Å	545(11)	3.11(3)	12(1)		
	c	5.33762(7) Å	635(16)	3.61(5)	13(1)		
Clinopyroxene	volume	433.192(7) Å <sup>3</sup>	693(10)	0.56(1)	5.9(4)	12.7(9)	1.01(1)b
	a	9.59029(8) Å	633(12)	1.50(2)	15(1)		
	b	8.92336(6) Å	663(9)	1.36(1)	15(1)		
	c	5.15493(3) Å	772(15)	2.03(3)	20(2)		
	$\beta$	100.8987(7) <sup>o</sup>	473(15)	3.47(5)	53(3)		
Ilmenite	volume	289.154(5) Å <sup>3</sup>	758(13)	0.44(1)	5.9(5)	12.7(11)	1.23(2)c
	a	4.93251(4) Å	801(18)	1.43(3)	18(2)		
	c	13.72349(10) Å	707(12)	1.13(1)	13(1)		
This study, from experimental Clapeyron slopes							
Clinopyroxene	Orthopyroxene - Clinopyroxene transition						1.07(4)
	Clinopyroxene - Ilmenite transition						1.08(3)
Ashida et al. (1985)							
Ilmenite	volume		819 (all data) 777 (<300K)			3.6	
Liebermann (1974)							
Clinopyroxene	volume		618				
Ilmenite	volume		657d				
Tsuchiya and Tsuchiya (2007)							
Ilmenite	volume	291.48 Å <sup>3</sup> e				4.59 e	1.24 e

881 *Table 6. Debye-Grüneisen model thermophysical parameters and derived values. a* Calculated using  $K_T=115$  GP; *b* calculated  
882 *assuming  $K_S = 131$  GPa; c* – calculated using  $K_T = 187$  GPa; *d* – reported in Okada et al. (2008); *e* – 0 GPa, 300 K values  
883 *calculated using density functional theory.*

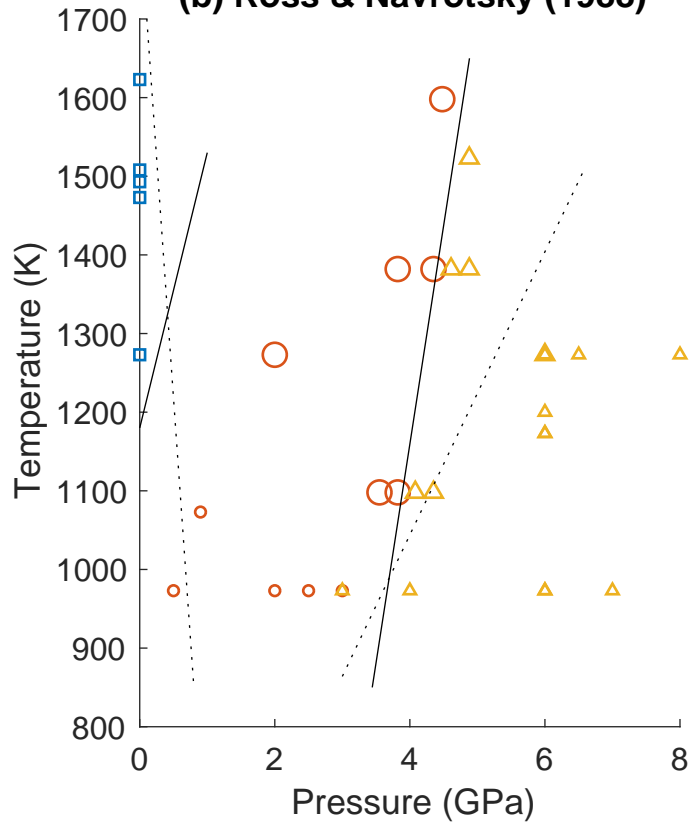
884

Fig. 1

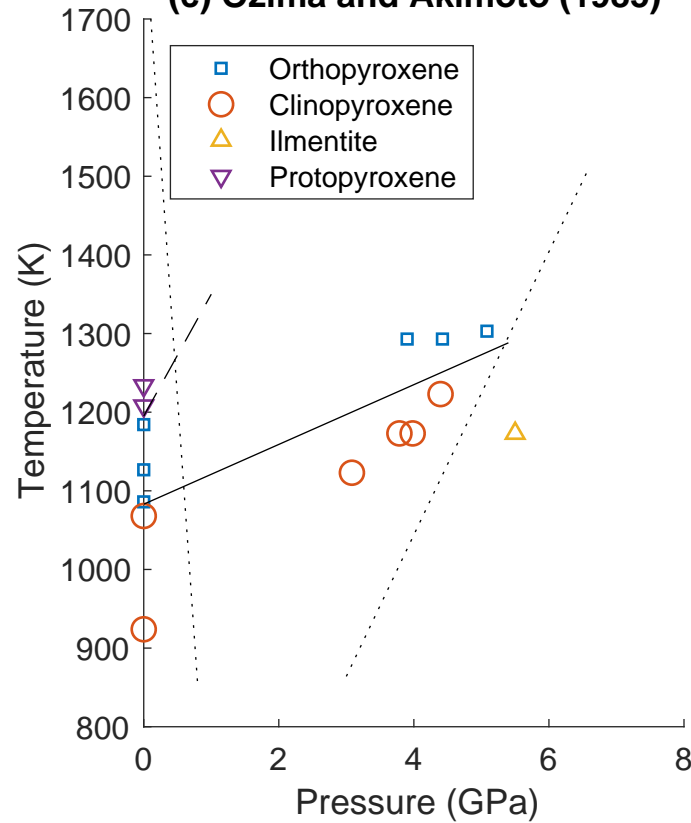
(a) Kirfel &amp; Neuhaus (1974)

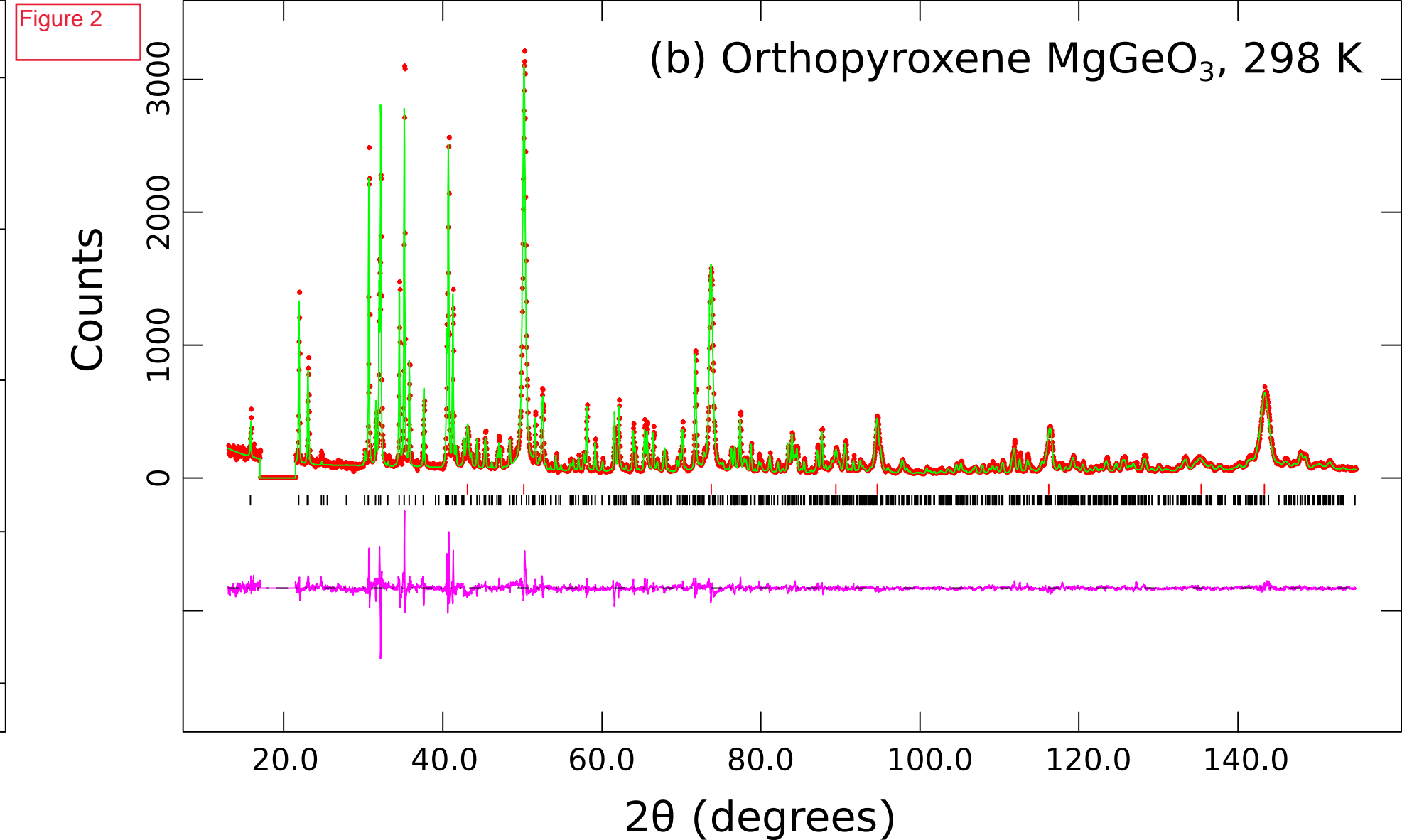
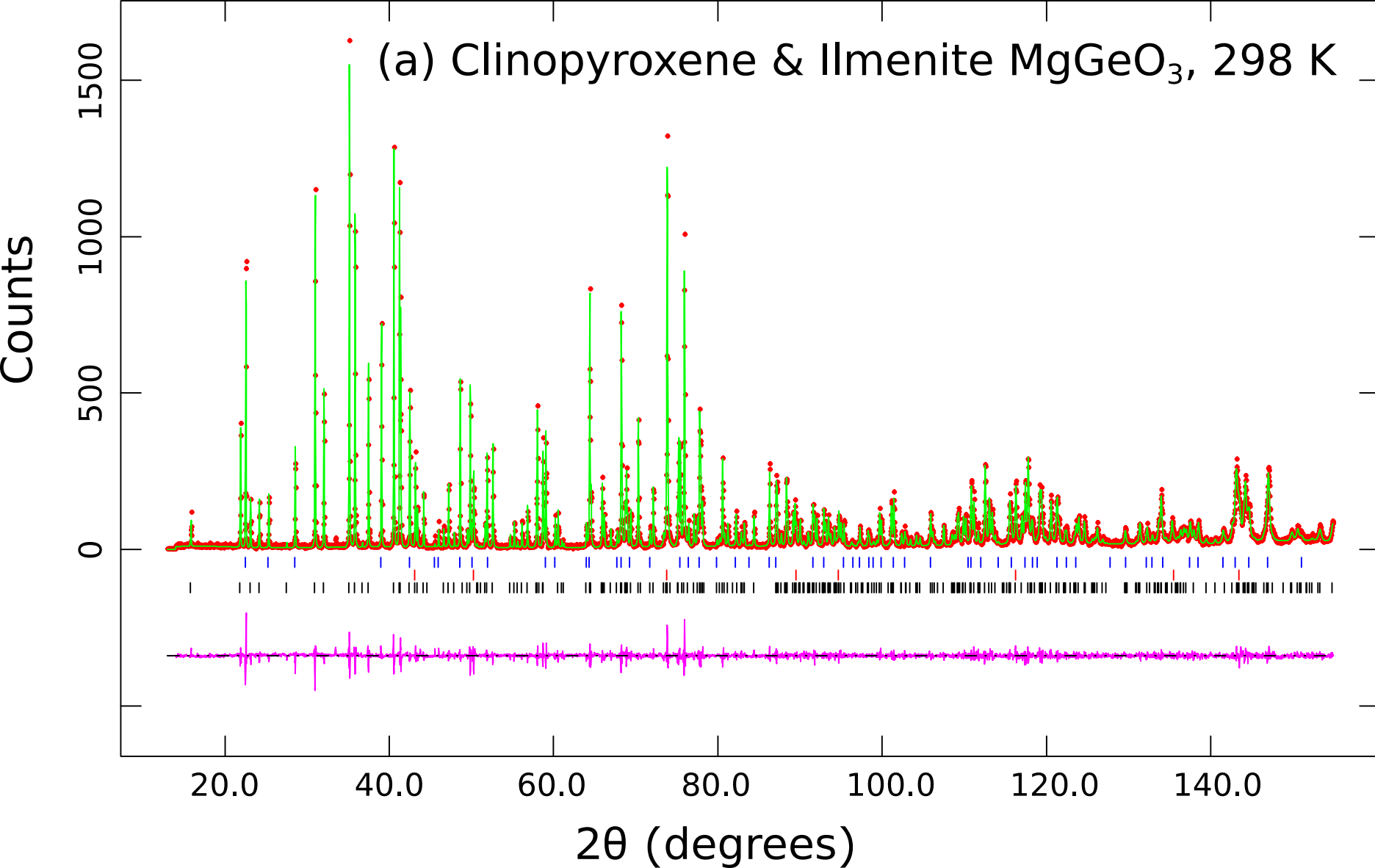


(b) Ross &amp; Navrotsky (1988)



(c) Ozima and Akimoto (1985)







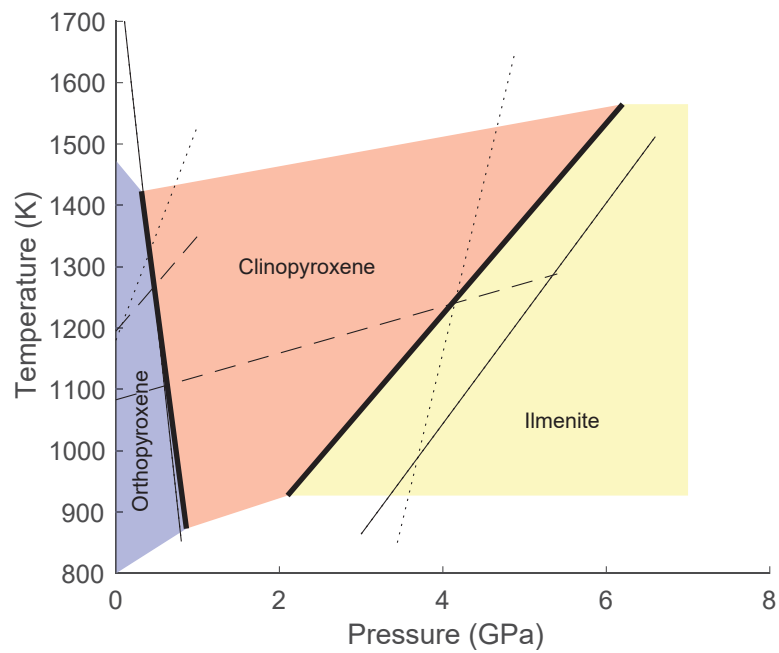
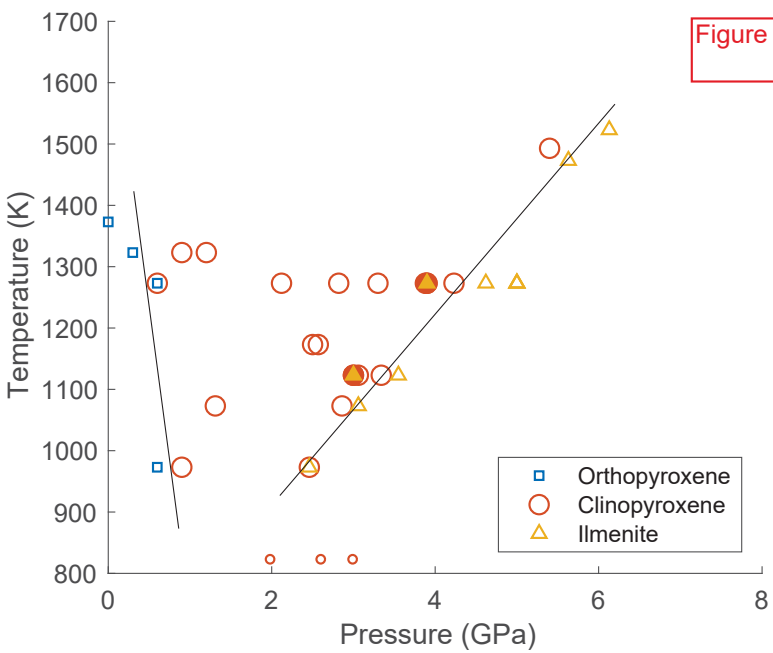
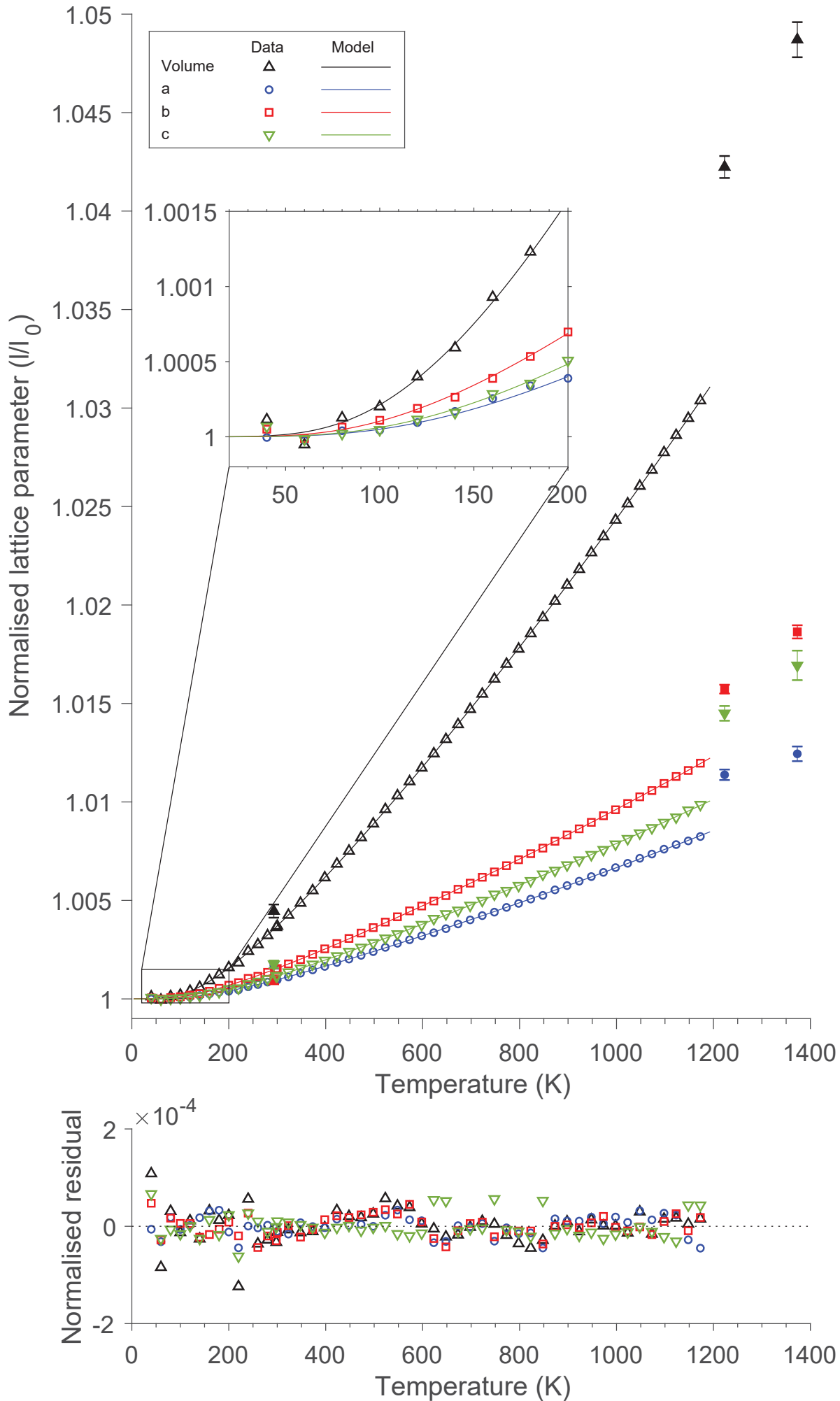


Figure 4



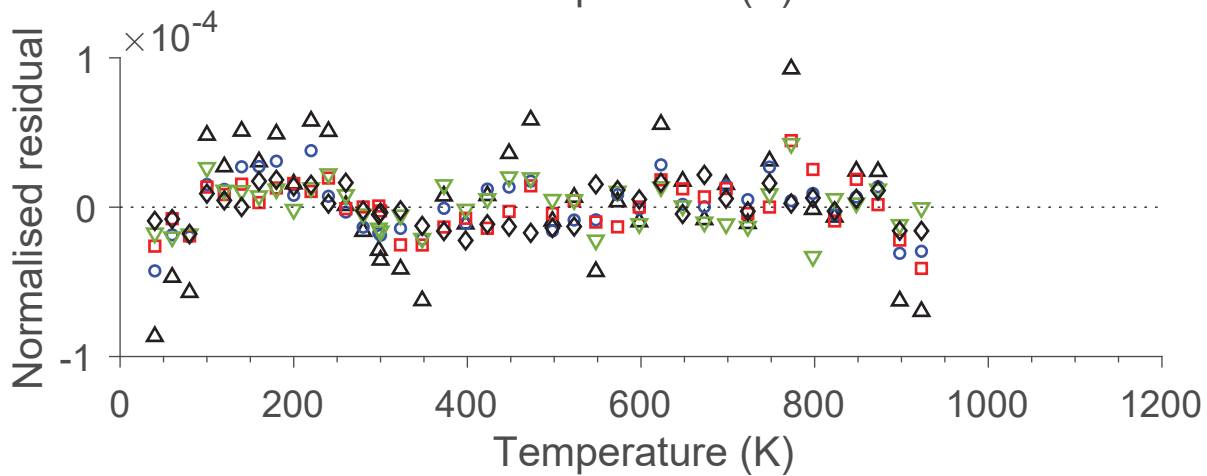
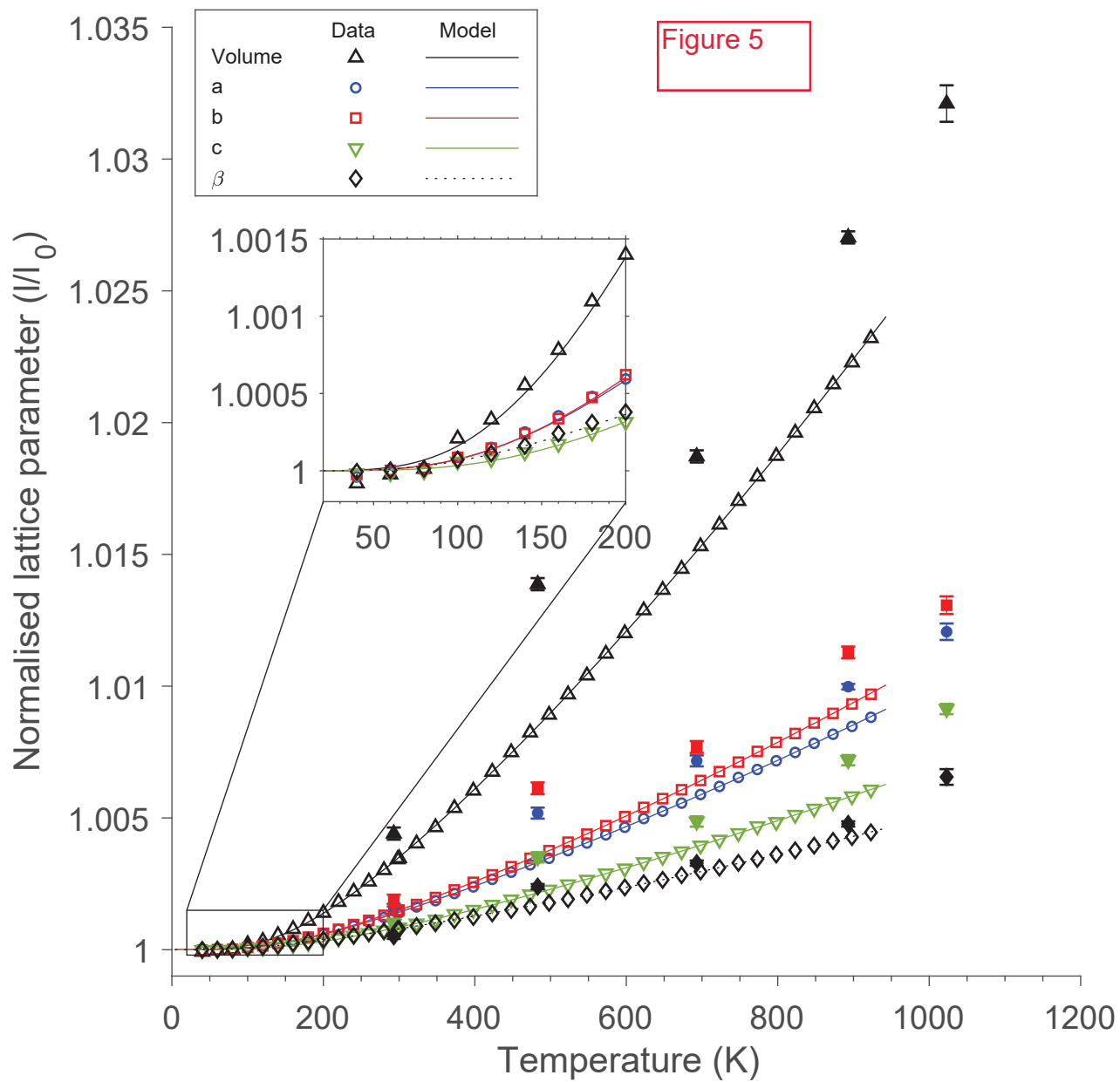
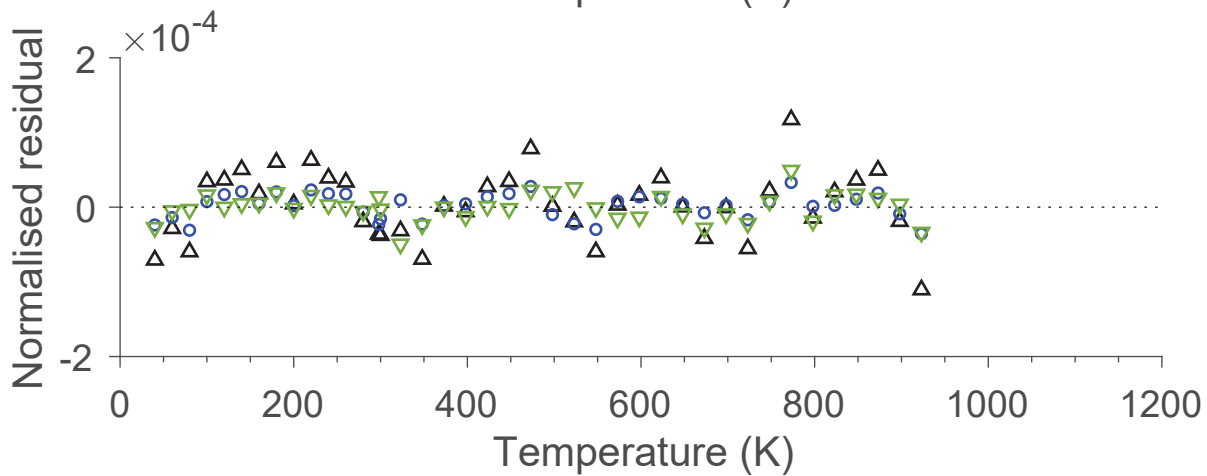
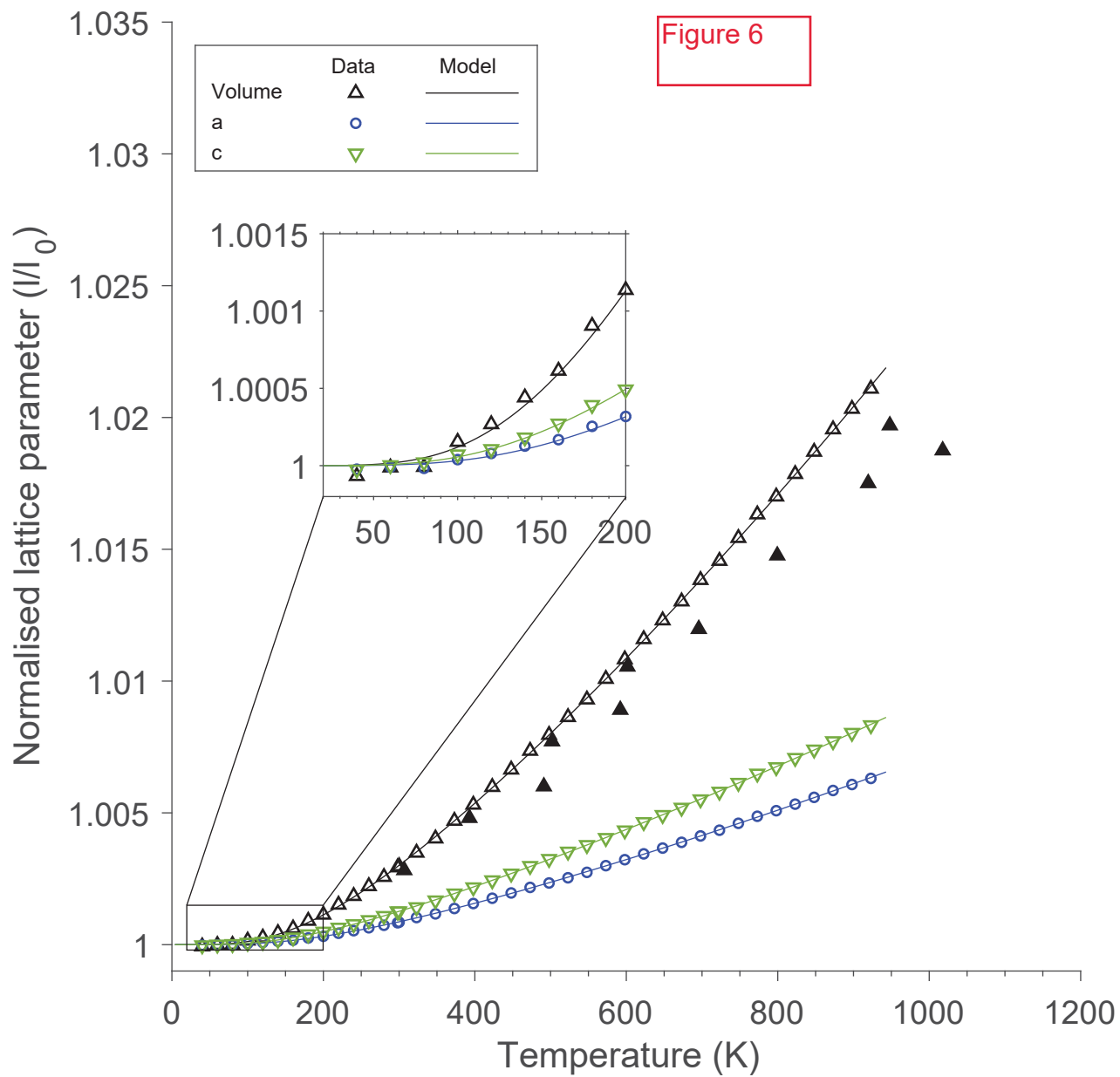
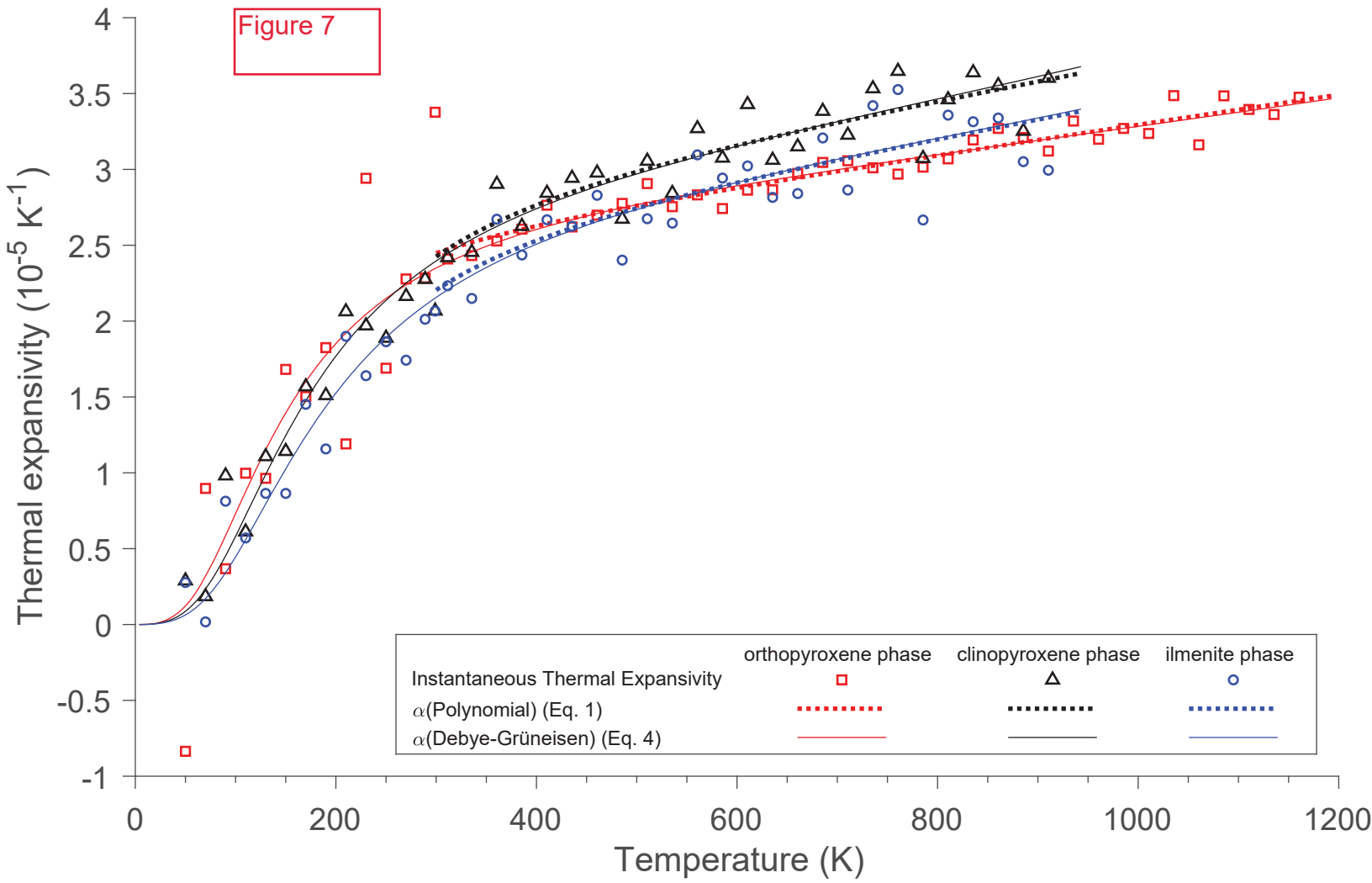
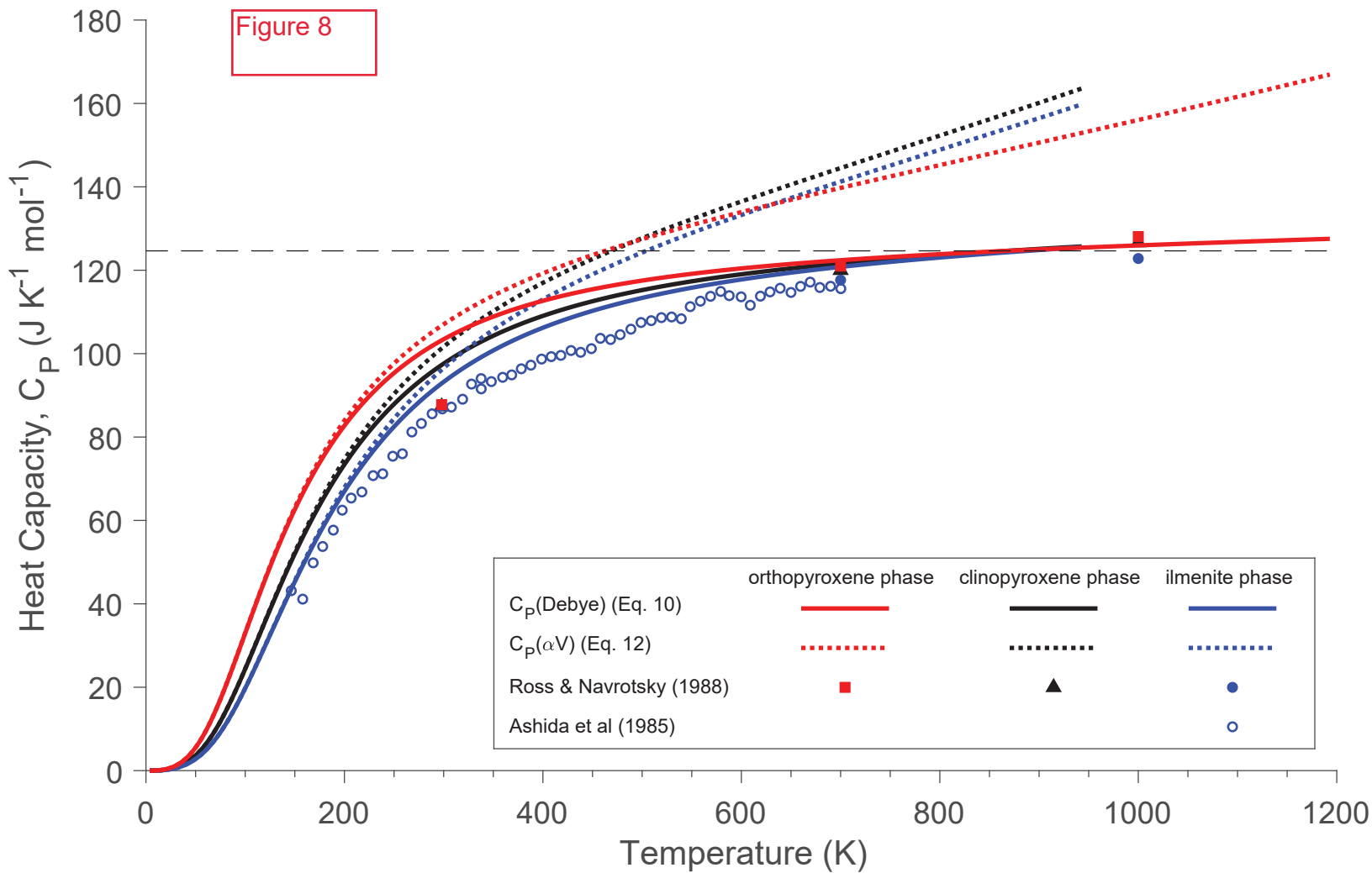


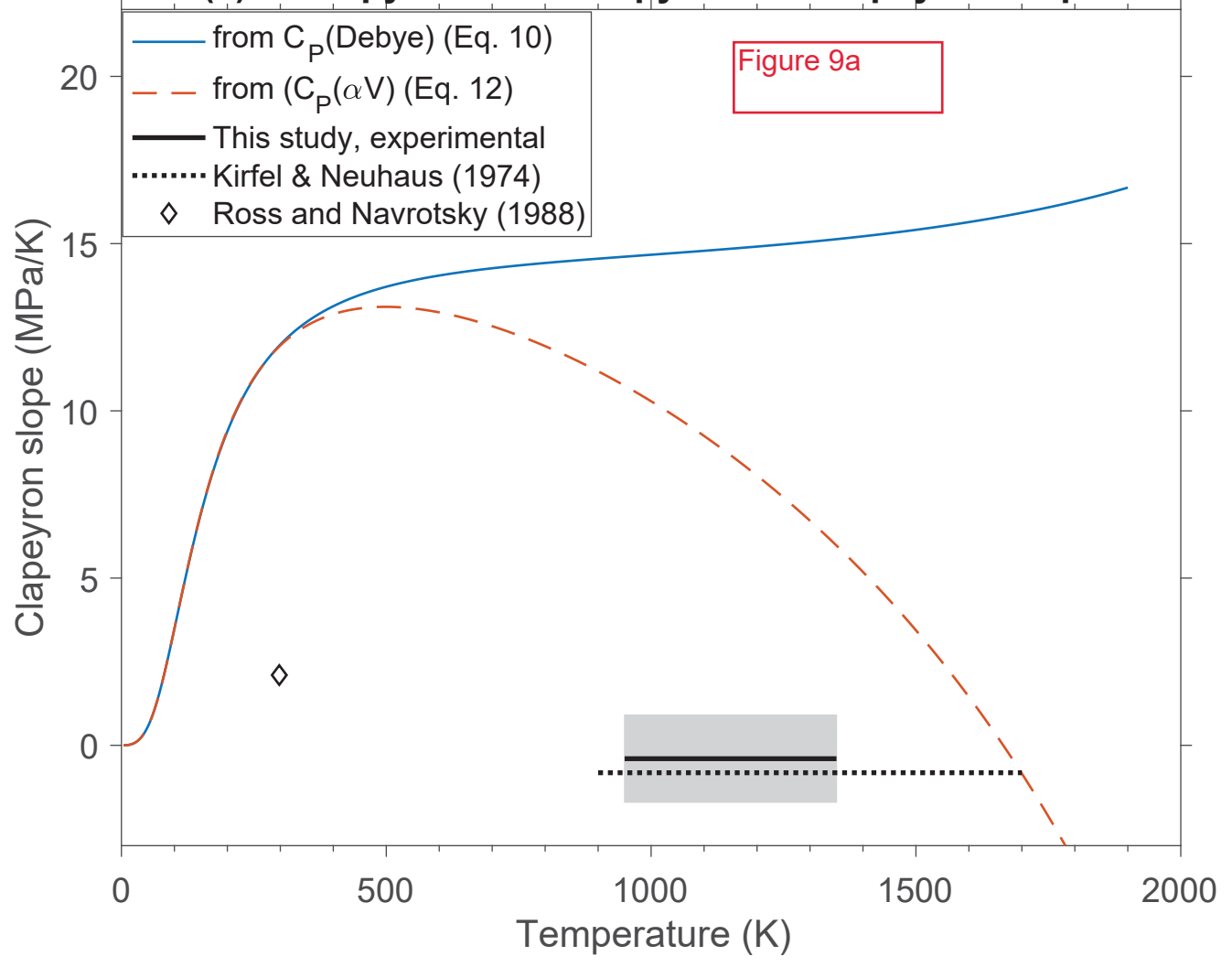
Figure 6







**(a) Orthopyroxene-Clinopyroxene Clapeyron slope**



**(b) Clinopyroxene-Ilmenite Clapeyron slope**

