

1 **Density and sound velocity of liquid Fe–S alloys at Earth’s outer core conditions**

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9 **Key Points:**

- 10 • First-principles molecular dynamics simulations of liquid Fe–S alloys are performed
11 at the Earth’s outer core conditions.
- 12 • P – T – V equations of state for liquid Fe–S alloys are established.
- 13 • Density and sound velocity of liquid Fe–S alloys along the geotherm are estimated.
- 14 • S concentration in the outer core is constrained.

15 **Abstract**

16 Pressure–temperature–volume (P – T – V) data on liquid iron–sulfur (Fe–S) alloys at
17 the Earth’s outer core conditions (~ 136 to 330 GPa, ~ 4000 to 7000 K) has been obtained by
18 first-principles molecular dynamics simulations. We developed thermal equation of state
19 (EOS) composed of Murnaghan and Mie-Grüneisen-Debye expressions for liquid Fe–S
20 alloys. The density and sound velocity are calculated and compared with Preliminary
21 Reference Earth Model (PREM) to constrain the S concentration in the outer core. Due to the
22 temperature at the inner core boundary (T_{ICB}) has not been measured precisely ($4850 \sim 7100$
23 K), we clarify that the S concentration range is from $10 \sim 14$ wt.% assuming S is the only
24 light element. Our results also show that Fe–S alloy couldn’t satisfy the seismological density
25 and sound velocity simultaneously. S element couldn’t be the only light element. Considering
26 the geophysical and geochemical constrains, we propose the outer core contains no more than
27 3.5 wt.% S, 2.5 wt.% O, or 3.8 wt.% Si. In addition, the developed thermal EOS can be
28 utilized to calculate plenty of thermal properties of liquid Fe–S alloys serving as fundamental
29 data to model the Earth’s outer core.

30

31 **1 Introduction**

32 The composition of the Earth's outer core with pressure from 136 to 330 GPa and
33 temperature from 4000 to 7000 K (Dziewonski and Anderson 1981; Koči et al. 2007; Alfè
34 2009) remains controversial. The direct geophysical data is from measuring the speed of
35 seismic wave propagation. Understanding the geodynamics in the outer core highly requires
36 the thermal properties of its composition. According to Birch's reports (Birch 1952), the
37 outer core is mainly composed of iron (Fe). While Preliminary Reference Earth Model
38 (PREM) (Dziewonski and Anderson 1981) raises that pure Fe is much denser than the
39 substance in outer core, and has a lower sound velocity. It's argued that there must exist some
40 light elements (e.g. sulfur, silicon, oxygen, carbon, and hydrogen) to decrease the density and
41 increase sound velocity (Poirier 1994; Hirose et al. 2013; Badro et al. 2015; Litasov and
42 Shatskiy 2016; Umemoto and Hirose 2020). The presence of light elements dose reduce the
43 density of ferroalloy, but not the sound velocity of each component matches the PREM data.
44 The type of light elements is still pending.

45 Owing to its siderophile nature, sulfur (S) is considered as a major light element in the
46 Earth's outer core (Huang et al. 2011; Kawaguchi et al. 2017). The concentration of S is
47 typically predicted by comparing the density and sound velocity of liquid Fe–S alloy with
48 PREM data. So far, many efforts have been done in experiment. Huang and co-authors
49 (Huang et al. 2011, 2018) measured the density and sound velocity of liquid $\text{Fe}_{92.5}\text{O}_{2.2}\text{S}_{5.3}$ and
50 $\text{Fe}_{90}\text{O}_8\text{S}_2$ up to 208 GPa and Fe–11.8 wt.% S up to 211.4 GPa through the shock-wave
51 experiment and predicted the maximum S content was around 10 wt.%. A similar experiment
52 estimated the S concentration of 2 wt.% (Zhang et al. 2016). Using laser heated diamond-
53 anvil cells device, Kawaguchi et al. measured the sound velocity of $\text{Fe}_{47}\text{Ni}_{28}\text{S}_{25}$ and
54 $\text{Fe}_{63}\text{Ni}_{12}\text{S}_{25}$ up to 58 GPa and 2480 K, and proposed the amount of S was 5.8 ~ 7.5 wt.%
55 (Kawaguchi et al. 2017). Since most experimental P – T – V data (Sanloup et al. 2002; Badro et

56 al. 2007; Jing et al. 2014; Kuskov and Belashchenko 2016) is measured at pressure and/or
57 temperature below the outer core conditions, the values at high P - T are usually extrapolated
58 from Birch's law (Badro et al., 2007). The validity of birch's law at extremely high
59 temperatures is questionable (Lin et al. 2005). In brief, due to technical challenges in
60 dynamic and static high-pressure measurement, the exact S concentration in the Earth's outer
61 core is far from reaching a consensus.

62 As a remedy, theoretical calculations has been utilized to identify the S concentration
63 (Alfè and Gillan 1998; Alfè et al. 2002a; Badro et al. 2014; Umemoto et al. 2014; Bazhanova
64 et al. 2017). Since some theoretical calculations are performed at zero K, temperature effects
65 on thermal pressure are often neglected or approximated (Bazhanova et al. 2017). Ignoring
66 the temperature impact at extreme conditions leads to erroneous pressure-volume (P - V)
67 relationship. Therefore, first-principles molecular dynamics (FP-MD) simulations are suitable
68 to study the P - T - V relations, namely, equation of state (EOS). Applied with the EOS, density
69 and sound velocity can be obtained. Actually, the thermal EOS for liquid Fe has been
70 reported (Vočadlo et al. 2003; Bouchet et al. 2013; Ichikawa et al. 2014; Wagle and Steinle-
71 Neumann 2019), whereas it is rare for liquid Fe-S alloys. The latter plays a central role on
72 predicting the S concentration in the outer core.

73 In this work, the P - T - V data of liquid Fe-S alloys with different S concentrations (0,
74 5.5, 11.5, and 18.1 wt.%) are obtained by FP-MD simulations with pressure and temperature
75 up to 330 GPa and 7000 K. We establish thermal EOS for liquid Fe-S alloys, and then
76 calculate the density and sound velocity along the geotherm of the outer core. In order to
77 constrain S content, the current data is compared to PREM data. Besides, the density and
78 sound velocity of liquid Fe-S alloys provided by FP-MD calculations are fundamental to
79 build geophysical and geochemical model of the Earth's outer core.

80 2 Computational Methods

81 FP-MD are based on the density functional theory (DFT) as implemented in the
82 Vienna Ab initio Simulation Package (VASP) (Kresse and Hafner 1993; Kresse and
83 Furthmüller 1996). The exchange-correlation functional is the Perdew-Burke-Ernzerhof
84 (PBE) (Perdew et al. 1996) variant of the generalized gradient approximation (GGA).
85 Projector augmented wave (PAW) pseudopotentials (Blöchl 1994) use fourteen valence
86 electrons ($3p^63d^74s^1$) with the outmost cutoff radius (r_c) 2.2 a.u. for iron, and six valence
87 electrons ($3s^23p^4$) with $r_c = 1.5$ a.u. for sulfur to avoid state overlapping under high
88 temperature and pressure conditions. Valence electrons states are expanded into plane waves
89 with a cutoff energy of 500 eV. The finite temperatures for electronic structure and force
90 calculations are implemented within the Fermi-Dirac smearing (Mermin 1965). The
91 reciprocal Brillouin zone is sampled by gamma point. These setting values are sufficient to
92 produce converging results. Molecular dynamics runs have been performed in the canonical
93 ensemble (NVT) using Nosè-Hoover thermostat (Nosé 1984; Hoover 1985) to control the
94 temperature. Every simulation runs for 7 ps with timestep 1.0 fs. The first 2 ps are to reach
95 the equilibrium. P and T values are averaged statistically from the last 5 ps. By the block
96 method (see the supplementary material part), the estimated statistic errors of pressure are
97 within 0.35 GPa. The running steps are long enough to collect accurate averages.

98 The initial configuration of liquid Fe including 108 atoms is built by melting the
99 perfect face-centered cubic (fcc) phase Fe with density of 11.893 g/cm^3 at 10000 K for 10 ps.
100 The liquid state is confirmed by the Fe–Fe radial distributional function (RDF) and mean
101 square displacement (MSD) of Fe atoms, which increases with time in liquid and is a constant
102 in solid fcc Fe (see Fig. S3). The initial configurations of liquid Fe–S alloys are obtained via
103 randomly replacing Fe atoms by S to reach the target S concentration, melted at 10000 K for

104 10 ps, and then quenched to target temperature. Different pressures are obtained by varying
105 the box size.

106 **3 Results and discussions**

107 In this work, we study four concentrations of S (0, 5.5, 11.5, and 18.1 wt.%) of liquid
108 Fe–S alloys at pressure ranges from 136 to 330 GPa and temperature from 4000 to 7000 K.
109 Figure 1 shows the comparison of isothermal density of liquid Fe at 6000 K with previous
110 available data (Brown and McQueen 1986; Anderson and Ahrens 1994; Dewaele et al. 2006;
111 Umemoto et al. 2014). It is clearly seen that the density of liquid Fe is slightly higher than
112 values by EOS fitted to experimental thermal expansion, enthalpy, and sound velocity etc.
113 (Anderson and Ahrens 1994), and lower than DFT values (Umemoto et al. 2014). The
114 derivations are within 1.2% and 1.4%, respectively. We state that the difference with
115 previous DFT calculation is possibly caused by technical details (e.g. cell size, exchange-
116 correlation functional, pseudopotential parameters, and so on). Compared with shock-wave
117 compression density 12.44 g/cm³ at 260 GPa and 6069 K (Brown and McQueen 1986), the
118 calculated value is 12.30 g/cm³ with derivation around 1.13 %. It's comparable to the
119 experimental uncertainty. The agreement confirms the FP-MD data is reliable.

120 To calculate thermal properties at given P and T , the EOS of liquid Fe–S alloys are
121 indispensable. Here, the P – T – V relations are analyzed combining Murnaghan (Murnaghan
122 1944) and Mie-Grüneisen-Debye (MGD) EOS (Ichikawa et al. 2014)

$$123 \quad P(V, T) = P_{T_0}(V) + \Delta P_{th}(V, T), \quad (1)$$

124 $P(V, T)$ is the total pressure, P_{T_0} is the pressure at a reference temperature T_0 . P_{T_0} is expressed as
125 Murnaghan EOS (Murnaghan 1944)

$$126 \quad P_{T_0} = \frac{K_{T_0}}{K'_{T_0}} \left[\left(\frac{V_0}{V} \right)^{K'_{T_0}} - 1 \right], \quad (2)$$

127 K_{T_0} is thermal bulk modulus at zero pressure and K'_{T_0} is the derivative of K_{T_0} over pressure at
128 T_0 . V_0 is the volume at zero pressure. The thermal pressure ΔP_{th} is expressed as

$$129 \quad \Delta P_{th}(V, T) = \frac{\gamma(V)}{V} (E(V, T) - E(V, T_0)), \quad (3)$$

130 $\gamma(V)$ is Grüneisen parameter. Here we assume $\gamma(V) = \gamma_0$ that is independent with pressure and
131 temperature in the target P - T range. The internal thermal energy $E(V, T)$ is represented by a
132 second-order polynomial of temperature with a volume dependent second-order coefficient as

$$133 \quad E = 3nR \left[T + e_0 \left(\frac{V}{V_0} \right)^g T^2 \right], \quad (4)$$

134 where R is the gas constant ($R = 8.314 \text{ J}\cdot\text{mol}^{-1}\cdot\text{K}^{-1}$) and n is the number of atoms in the
135 formula unit (Fe: $n = 1$; Fe-S: $n = 2$). The first term corresponds to atomic contribution,
136 while the second term represents the electronic contribution.

137 The present EOS is comprised of four equations, which requires six parameters (V_0 ,
138 K_{T_0} , K'_{T_0} , γ_0 , e_0 , and g) to calculate the pressure. These parameters are fitted to the FP-MD
139 P - T - V data by least squares algorithm (see Table 1.). We set the temperature 6000 K as the
140 reference temperature T_0 in order to constrain the reference isotherm as tightly as possible in
141 the broad P - T range. We discuss the accuracy of Murnaghan EOS to describe the isothermal
142 pressure in the supplementary material part.

143 How does the introduction of S change the thermal properties of liquid Fe alloy? The
144 isothermal density of liquid Fe-S alloys is presented in Figure 2. The general trend is the
145 density decreases with temperature and S concentration. It is consistent with the conclusions
146 about liquid Fe-S and Fe-H alloys (Umamoto et al. 2014; Umamoto and Hirose 2015). The
147 results indicate the density of Fe-11.5 wt.% S composition is close to the PREM data.

148 Based on the EOS, we calculate thermal properties including isothermal bulk modulus
149 (K_T), adiabatic bulk modulus (K_S), and sound velocities (V_p) of liquid Fe–S alloys. The
150 isothermal bulk modulus (K_T) is defined by $K_T = -V \left(\frac{\partial P}{\partial V} \right)_T$. The adiabatic bulk modulus (
151 K_S) is determined by $K_S = (1 + \alpha \gamma T) K_T$, where α is volumetric thermal expansion coefficient.
152 αK_T is both temperature and volume dependent and can be calculated by $\alpha K_T = \left(\frac{\partial P}{\partial T} \right)_V$. The
153 calculated αK_T values of Fe–S alloys at different P and T are listed in Table S1. For liquid Fe,
154 $\alpha K_T = 0.011$ GPa K⁻¹ at 80 GPa, 2500 K (Seagle et al. 2006), and $\alpha K_T = 0.95 \times 10^{-2}$ at 150
155 GPa and 4000 K (Ichikawa et al. 2014). Here we get $\alpha K_T = 0.01162$ GPa K⁻¹ for Fe at 4000 K
156 that is in good agreement with previous reports. We assume Grüneisen parameter $\gamma_0 = 1.5$
157 based on previous first-principles calculations and experimental measurements (Brown and
158 McQueen 1986; Anderson and Ahrens 1994; Wasserman et al. 1996; Alfè et al. 2002b;
159 Belonoshko 2010). The value is widely used in reported experimental or theoretical analysis
160 on the similar issue (Huang et al. 2011; Umemoto et al. 2014). The sound velocity V_p of
161 liquid Fe–S alloys is calculated by $V_p = \sqrt{\frac{K_S}{\rho}}$, where ρ is the density of the substance. The
162 uncertainty of sound velocity mainly caused by Grüneisen parameter is within 1.79 %.

163 In the following, we compare the isothermal sound velocity of liquid Fe–S alloys with
164 PREM (see Figure 3). Obviously, it increases with pressure. Introducing S atoms into liquid
165 Fe alloy does increase the sound velocity. The curves of Fe–5.5 wt.% S are close to the
166 PREM data. Combined to the density- and sound velocity-pressure relations, the likely S
167 concentration should be between 5.5 and 11.5 wt.%. In addition, the results (see solid lines)
168 also indicate that sound velocity is insensitive to temperature. The same conclusion was

169 reported in the study of liquid Fe–S and Fe–H alloy (Umemoto et al. 2014; Umemoto and
170 Hirose 2015).

171 **4 Geophysical implications**

172 Based on the Murnaghan-MGD EOS, we estimate the S concentration in the outer
173 core. The adiabatic temperature profile along the geotherm is given by

$$174 \quad T = T_{ICB} \left(\frac{\rho}{\rho_{ICB}} \right)^\gamma, \quad (5)$$

175 where T_{ICB} and ρ_{ICB} are temperature and density at the inner core boundary (ICB), ρ is the
176 density. γ is Grüneisen parameter.

177 It is known that T_{ICB} relies closely on melting temperature (T_m) of Fe at ICB pressure.
178 According to different approaches, T_m varies from 4850 K by extrapolation of static
179 compression measurement (Boehler 1993), 6230 ± 500 K by fast x-ray diffraction (Anzellini et
180 al. 2013) to 5400 ~ 7100 K via first-principles or semi-classical calculations (Wasserman et
181 al. 1996; Stixrude et al. 1997; Belonoshko et al. 2000; Laio et al. 2000; Alfè 2009;
182 Belonoshko et al. 2017, 2019). Since T_m at ICB pressure has not been measured precisely,
183 four anchoring temperatures of 4850 K (Boehler 1993), 5400 (Laio et al. 2000), 6350 (Alfè
184 2009; Anzellini et al. 2013), and T_{ICB} 7100 (Belonoshko et al. 2000) are tested to estimate
185 the temperature at the core mantle boundary (CMB). The adiabatic temperature profiles are
186 presented in Figure 4, and the corresponding T_{CMB} are 3562, 3966, 4664, and 5214 K,
187 respectively.

188 We elaborate on the density and sound velocity profiles for S concentration 5.5 ~ 11.5
189 wt.% ferroalloy along the geotherm with $T_{ICB} = 4850, 5400, 6350,$ and 7100 K, as displayed
190 in Figure 5., The most possible Fe–S alloy composition in the outer core is then predicted. If
191 T_{ICB} is around 4850 K (Boehler 1993), we predict Fe–14.0 wt.% S reproduce the density of

192 PREM data by interpolation. With T_{ICB} around 5400 K (Laio et al. 2000), it is Fe–13.0 wt.%
193 S that reproduce the PREM density profile. Applied with the same T_{ICB} , previous first-
194 principles approach estimated the maximum S concentration was 14 wt.% (Umemoto et al.
195 2014) while shock-wave compression experiment (Huang et al. 2013, 2018) gave out 10.0
196 wt.%. Our result is comparable with previous first-principles one within the calculation
197 deviation. Another well agreed T_{ICB} by both theory (Alfè 2009) and experiment is around
198 6350 K (Anzellini et al. 2013). Estimated with this value, it is Fe–11.5 wt.% S that could
199 reproduce the PREM density profile. Since recent first-principles calculations argued a
200 “high” T_m (~ 7100 K) of Fe (Belonoshko et al. 2000, 2017, 2019), we apply $T_{ICB} = 7100$ K to
201 estimate the S concentration in the outer core and find that Fe–10 wt.% S could explain the
202 PREM density quite well.

203 As the Figure 5 shown, the sound velocity along the geotherm increases with the
204 sulfur concentration and is insensitive with the T_{ICB} temperature. Our predicted slopes of
205 sound velocity versus pressures are in good agreement with the PREM one and better than
206 the ones from previous first-principles approach (Umemoto et al. 2014) and shock-wave
207 compression experiment (Huang et al. 2013, 2018). It should be noted that the predicted
208 sound velocity of liquid Fe–S alloys in this work is still higher than PREM data by about
209 1.74 % at the bottom and 3.04% at the top of the outer core.

210 Since Fe–S alloy can't match the deficits in density and sound velocity
211 simultaneously, there must be other light elements in the outer core, for example, oxygen or
212 silicon. Earlier first-principles calculation (Badro et al. 2014) showed that the outer core must
213 contain oxygen to synchronously satisfy the seismological density and sound velocity.
214 However, the shock-wave compression experiment (Huang et al. 2011) ruled out oxygen as a
215 major light element and stated oxygen concentration should be less than 2.5 wt.%. Recent
216 experiment with the same method reported that 9.0 wt.% Si in the outer core and 3.8 wt.% Si

217 in the inner core could match the PREM data (Huang et al. 2019). Besides, melting
218 experiment demonstrated eutectic liquid in Fe–Fe₃S binary system contains less than 15 wt.%
219 S at 15 GPa (Fei et al. 2000), 9.5 wt.% S at 136 GPa (CMB pressure), or 6.0 wt.% S at 254
220 GPa (Mori et al. 2017). One pace further, we deduce the S partition at 330 GPa (ICB pressure)
221 is approximately 3.5 wt.%. Combination of the geophysical and geochemical constrains, we
222 propose the outer includes no more than 3.5 wt.% S, 2.5 wt.% O, or 3.8 wt.% Si. Detailed
223 calculations about the density and sound velocity of Fe–S–O–Si alloys should be performed
224 in future work.

225

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Figure captions

371 **Figure 1** The density of liquid Fe at 6000 K is compared with reported data. (a) is
372 experimental solid hcp Fe (Dewaele et al. 2006). (b) is first-principles calculation at 6000 K
373 (Umemoto et al. 2014). (d) is experimental liquid Fe at 6000 K derived from parameterized
374 EOS (Anderson and Ahrens 1994). The filled blue square is density at 6000 K, 260 GPa by
375 shock-wave experiment (Brown and McQueen 1986).

376 **Figure 2** Isothermal density as a function of pressure for liquid Fe–S alloys. PREM is
377 included for comparison (Dziewonski and Anderson 1981). The cubic, circle, and up-/down-
378 triangle symbols are FP-MD P – T – V data. The red, green, blue, and magenta solid lines are
379 the data from Murnaghan (M) EOS.

380 **Figure 3** Isothermal sound velocity profiles of liquid Fe–S alloys. PREM is included for
381 comparison (Dziewonski and Anderson 1981).

382 **Figure 4** Adiabatic temperatures profiles in the outer core with four different anchoring
383 temperatures (Boehler 1993; Stixrude et al. 1997; Belonoshko et al. 2000; Laio et al. 2000;
384 Alfè 2009; Anzellini et al. 2013).

385 **Figure 5** Density and sound velocity as a function of pressure along the geotherm. Open
386 squares are shock-wave experimental data by Huang (Huang et al. 2018). Open triangles
387 represent data from first-principles calculation by Umemoto (Umemoto et al. 2014).

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389 Table 1 Murnaghan-MGD EOS parameters for liquid Fe–S alloys

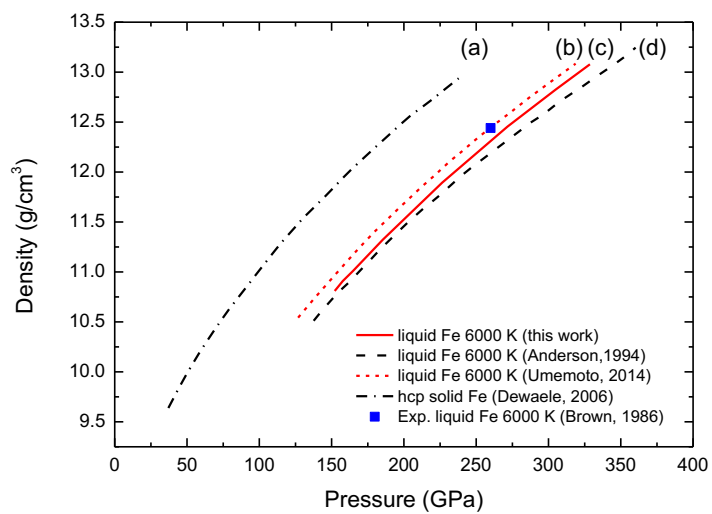
parameters	V_0 ($\text{\AA}^3/\text{atom}$)	K_{T0} (GPa)	K_{T0}'	γ_0	e_0 (K^{-1})	g
Fe	13.55	137.96	3.42	1.50	1.16491083e-04	1.12396676e+00
Fe–5.5wt.%S (Fe ₉₈ S ₁₀)	13.99	117.79	3.46	1.50	-4.00572800e-05	1.19346252e+00
Fe–11.5wt.%S (Fe ₈₈ S ₂₀)	14.37	107.38	3.40	1.50	-2.88105736e-05	5.06933449e-01
Fe–18.1wt.%S (Fe ₇₈ S ₃₀)	15.68	75.15	3.46	1.50	-5.10315845e-05	1.61538473e+00

390 Note: the parameters are fitted based on the reference temperature 6000 K.

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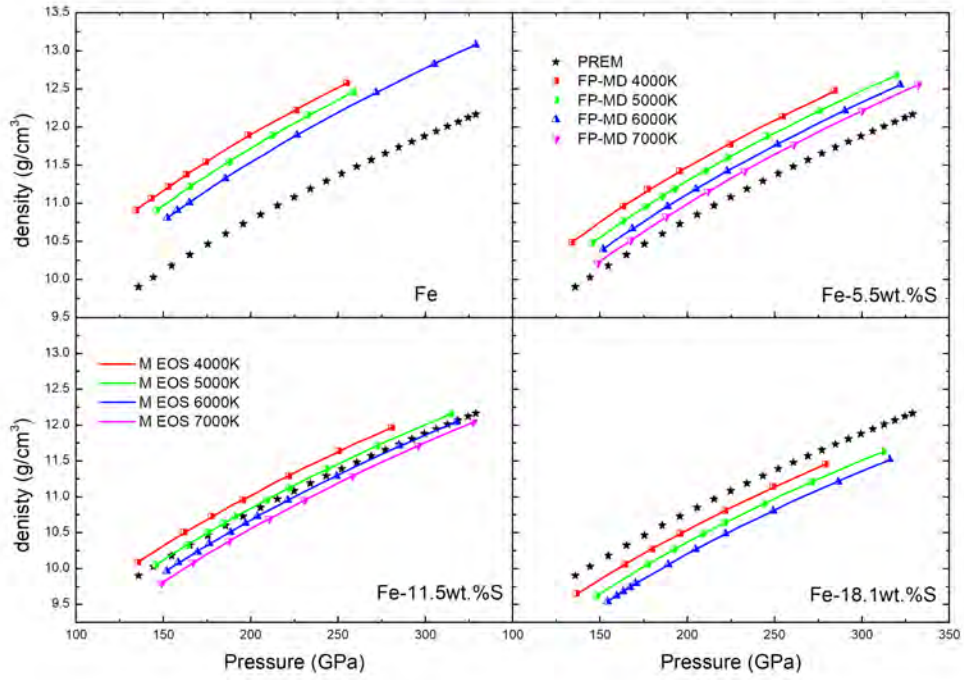


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Figure 1

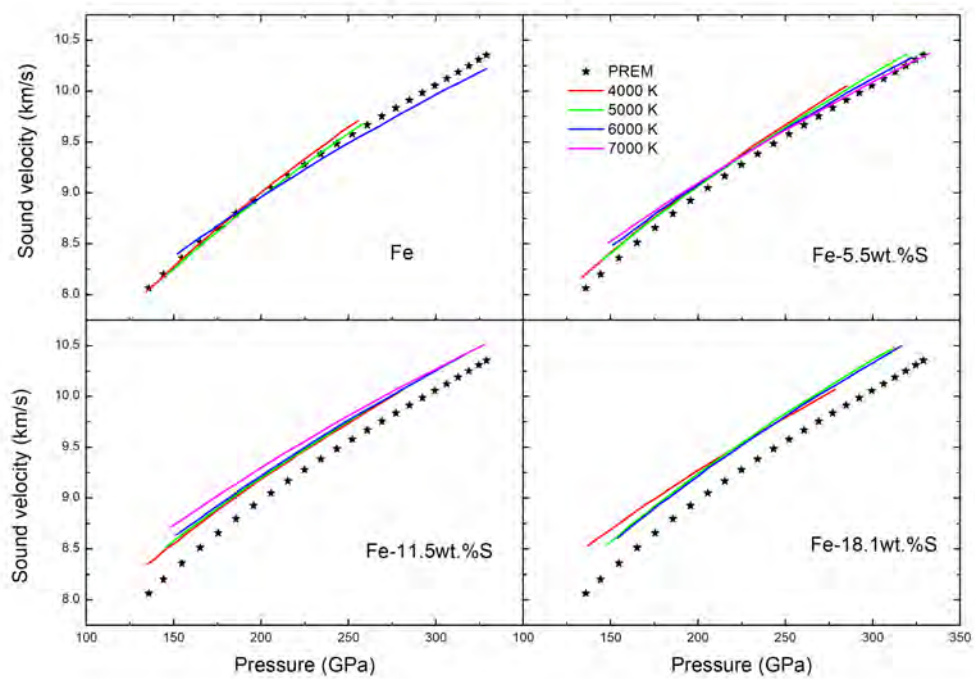


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Figure 2

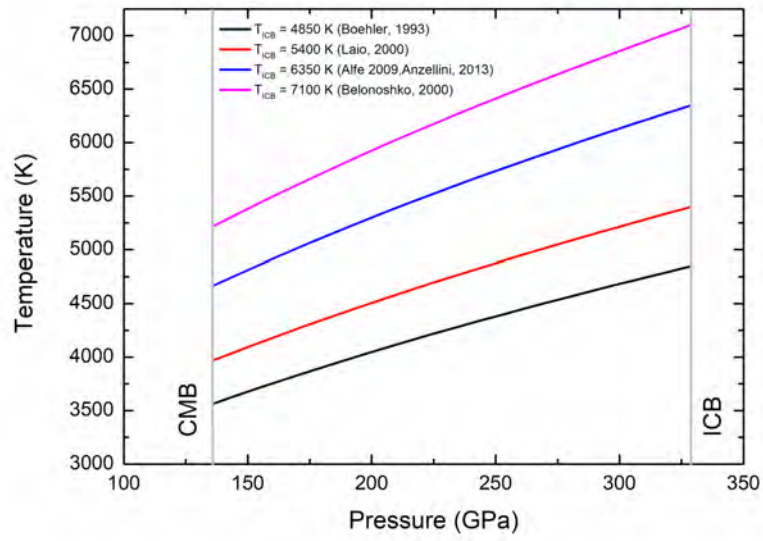


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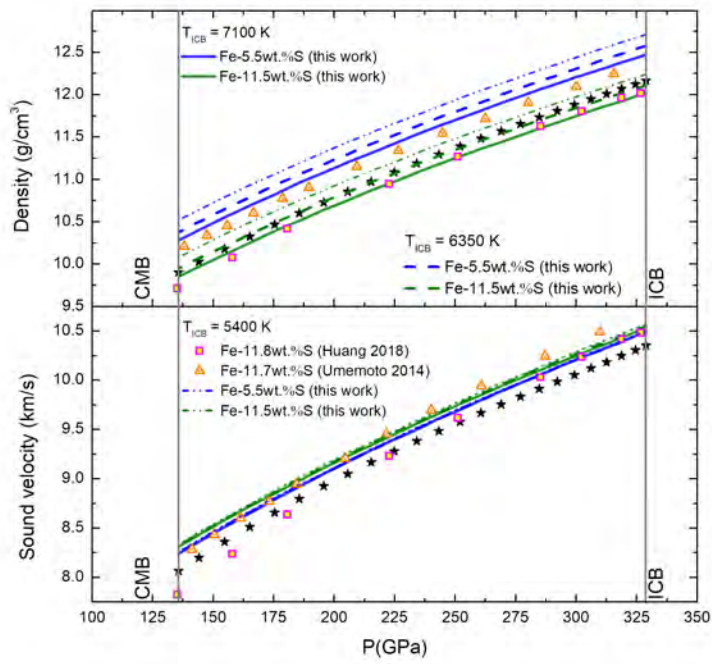
Figure 3



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Figure 4.

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Figure 5