

2 Diffusion of F and Cl in dry rhyodacitic melt

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24 **Abstract**

25

26 Chemical diffusion of F and Cl has been experimentally determined in a rhyodacitic melt obtained
27 from remelting a sample of Hekla pumice (Iceland). Diffusion couple experiments were conducted
28 in a vertical tube furnace over a temperature range of 750 – 950°C and in air for durations of 1 to
29 35 days. Concentration profiles of F and Cl were obtained for the quenched sample using an
30 electron microprobe.

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32 Fluorine and chlorine exhibit Arrhenian behaviour over the range of temperature investigated here.
33 The pre-exponential factors of F and Cl are $D_0(\text{F}) = 4.3 \times 10^{-4}$ and $D_0(\text{Cl}) = 1.6 \times 10^{-5} \text{ m}^2/\text{s}$. Fluorine
34 diffusion coefficients vary in the order of 1×10^{-15} to $1 \times 10^{-13} \text{ m}^2/\text{s}$ whereas Cl diffusivity is up to
35 2 orders of magnitude slower. The activation energies for F and Cl diffusivities are equal within
36 error at 223 +/- 31 and 229 +/- 52 kJ/mol, respectively.

37

38 The difference in diffusivity between F and Cl is particularly pronounced in the melt of our study,
39 compared to results obtained for other magmatic melt compositions. This means that the potential
40 for diffusive fractionation exists and may occur especially under conditions of magma ascent and
41 bubble growth, as this would favor partitioning of the relatively fast-diffusing halogens into
42 growing bubbles, due to H₂O exsolution. A dependence of diffusivity on atomic radius observed
43 here is enhanced over that observed in more basic, less viscous melts, indicating that diffusive
44 fractionation is more likely to be pronounced in more silicic, more viscous systems. A proper
45 parameterisation and modelling of diffusive fractionation of halogens in actively degassing
46 volcanic systems thus holds the potential of serving as a tool for quantifying the processes
47 responsible for volcanic unrest.

48

49 **Keywords: halogens (F, Cl); diffusion; silicate melt; diffusion couple; experimental**

50 **volcanology**

51 **1. Introduction**

52

53 Volatiles are an important constituent of melts and play a significant role in igneous processes. The
54 chemical composition of the volatile phase crucial for the style of volcanic eruptions is dependent
55 on parameters such as solubility, partitioning and diffusion of the different species, which in turn
56 are controlled by compositional parameters (e.g. initial volatile content, magma composition) and
57 ambient conditions (e.g. pressure, temperature). While the dominant components of volcanic
58 volatiles are water (H₂O), carbon dioxide (CO₂), and sulfur-species, halogens (F, Cl, Br, I) can be
59 highly concentrated in silicic melts (e.g. tin and topaz rhyolite, Carroll and Webster, 1994; Webster
60 and Duffield, 1994) and are a significant component of volcanic gases (Symonds et al. 1994;
61 Aiuppa et al. 2009; Webster et al. 2018) venting from active silicic volcanic centers. Halogens
62 affect magma viscosity and diffusivities (e.g. Dingwell et al. 1985; Dingwell and Hess 1998;
63 Baasner et al. 2013) as well as phase equilibria (Manning 1981) and directly influence the stability
64 of hydrous minerals and halogen-bearing phases in igneous rocks (e.g. Micas, Fluorite, Topaz;
65 Webster et al. 2018). Schipper et al. 2017 showed that the degassing of F and Cl into isolated pores
66 of slowly cooling magma can cause silica redistribution and the formation of vapor-phase
67 cristobalite. Halogen degassing is also essential for the formation of ore deposits in active volcanic
68 systems and can have a major impact on Earth's climate and environment including potential
69 destruction of ozone in the stratosphere (Aiuppa et al. 2001, 2009; Bobrowski et al. 2003, 2007;
70 von Glasow et al. 2009; Boichu et al. 2011; Surl et al. 2015; Roberts 2018). The escape of F and
71 Cl from ascending magma has been argued to occur at relatively shallow depths (Spilliaert et al.
72 2006). Thus, their study in volcanic gas emissions can help identify and characterize pre- or syn-
73 eruptive volcanic degassing and might augment volcano monitoring.

74 The potential for widely varying diffusivities between the halogens may mean that diffusive
75 fractionation between the halogens may provide further insights into the timing of magma ascent
76 and volatile exsolution (Alletti et al. 2007). Halogens are also increasingly routinely measured in
77 volcanic plumes.

78 Despite investigations to date, our knowledge of halogen solubility and diffusivity in silicate melts
79 is far from complete. Rhyolitic volcanoes are known to commonly be the most explosive and
80 hazardous and typically comprise melts with the highest F concentrations (e.g. Aiuppa et al., 2009).
81 Yet there, halogen diffusivity data is scant for such systems. Instead, most studies describing
82 volatile behavior in silicate melts have concentrated on H₂O, CO₂ and S. Previous studies on F and
83 Cl diffusion are available for simplified model systems such as albite, jadeite or Na-aluminosilicate
84 (Dingwell and Scarfe, 1984; Dingwell and Scarfe, 1985, only for F), basaltic (Alletti et al. 2007)
85 and phonolitic melts (Balcone-Boissard et al. 2009; Böhm and Schmidt 2013). Only few studies
86 exist on more evolved dacitic to rhyolitic melt compositions (Bai and Koster van Groos 1994;
87 Fortin et al. 2017; Yoshimura 2018) and those have concentrated solely on Cl diffusion. Further,
88 most previous studies have been conducted at higher experimental temperatures than those relevant
89 for natural systems.

90 While some authors report a pressure dependence of the diffusion coefficients (e.g. 1.0 - 1.5 GPa;
91 Dingwell and Scarfe, 1984; Bai and Koster van Groos, 1994) most recent studies have found only
92 minor pressure effects (e.g. 0.5 - 1.0 GPa, Alletti et al., 2007; Balcone-Boissard et al., 2009). The
93 addition of H₂O is reported to enhance diffusivities up to an order of magnitude (Baker and
94 Balcone-Boissard 2009; Böhm and Schmidt 2013). For a complete compilation of all studies on
95 halogen-diffusion in silicate melts until 2010, see reviews by Baker and Balcone-Boissard (2009)
96 and Zhang et al. (2010) and for a recent comprehensive review about halogens in volcanic systems
97 of mafic to intermediate compositions see Webster et al. (2018).

98 In this study we apply the diffusion couple technique on remelted, mechanically homogenized
99 glasses synthesized from natural pumice of the Hekla H3 eruption, one of Hekla's most recent
100 Plinian silicic eruptions (2879 BP; 2.2 km³ DRE; e.g. Sverrisdottir, 2007; Thordarson and Larsen,
101 2007; Weber and Castro, 2017). We determine the chemical diffusion of F and Cl in melts of
102 rhyodacitic composition over a temperature range relevant for these magmatic systems in order to
103 improve and expand our understanding of volatile transport in magmatic systems. Even though
104 diffusion is not the only mechanism responsible for volatile transport in melts it is one of the
105 fundamental processes necessary to understand and model volcanic degassing.

106

107 **2. Experimental and analytical technique**

108

109 The initial sample synthesis was carried out in the laboratories of the Earth Science department of
110 the LMU Munich. Experimental starting glasses were synthesized from natural rhyodacitic pumice
111 of the Hekla H3 eruption (Table 1). The pumice was synthesized in a 1 atm high-T furnace at 1400–
112 1550°C for 2–7 days to produce volatile-depleted glass (Table 1). During the synthesis, the melt
113 was slowly stirred using a platinum-rod attached to a viscometer to improve devolatilization and
114 ensure homogeneity of the whole batch. In addition, the viscosity of the glass was monitored and
115 the synthesis was continued until the viscosity reached a steady state, i.e. the viscosity did not
116 change any more over a time scale of several hours, which we interpreted to reflect the bulk loss
117 of volatiles. After the first synthesis, the batch was split and crushed. One half was doped with
118 halogen-bearing Na-salts (NaF, NaCl, NaBr, NaI) so that each halogen would account for 1 weight
119 percent of the whole batch after doping. Bromine and iodine, even though present in the melt, are
120 the subject of another study in progress. However, it has to be taken into account that all halogen

121 species are diffusing in our experiments and that this might have an effect on the results. The other,
122 halogen-depleted half of the batch was in turn doped with a certain amount of NaCO₃ to account
123 for the sodium enrichment in the halogen-bearing glass. After thorough mixing, each half was then
124 re-homogenized by high-T stirring at the same conditions as in the first synthesizing step for 4–7
125 days. It is noted that, even though the samples were synthesized at high temperatures for several
126 days, the halogen-depleted sample still contained few 100's of ppms of halogens (Table 1).
127 However, the diffusion of halogens nevertheless occurred under a significant chemical potential
128 gradient between the two diffusion couple halves. The small amount of halogens in the depleted
129 half is therefore not expected to have an effect on the resulting diffusivities. After synthesis,
130 cylinders with a diameter of 4.6 mm were drilled out of the glass within the Pt crucible and cut into
131 small discs with a thickness of ~2 mm using a diamond wire saw. The discs were polished on one
132 side to ensure proper contact between the two diffusion couple halves during the experiment. The
133 final major-element compositions of the two different anhydrous starting glasses are presented in
134 table 1. Sodium enrichment in the experimental glasses compared to the original H3 pumice is
135 attributed to the addition of halogens by Na-halogenides in the enriched and the addition of
136 sodiumcarbonate in the depleted glass.

137 The capsules for each diffusion couple were constructed from 5 mm Pt tube with a wall-thickness
138 of 0.2 mm. The bottom of the capsule was closed with a Pt lid which was pressed onto and welded
139 to the tube using an arc-welder. Afterwards the bottom of the capsule was tamped down to ensure
140 cylindrical geometry. The halogen-doped glass discs were loaded into the bottom part of the
141 capsules and the halogen-depleted discs on top of them with both discs touching at the polished
142 surfaces (Fig. 1a). Each capsule was closed by welding a second lid to the top of the capsule to
143 ensure closed system conditions (Fig. 1b).

144 The diffusion experiments were conducted in the laboratories of the University of Mainz. For each
145 experiment one diffusion couple was first weighed and then loaded into an alumina tube which is
146 closed on one side. The tube was inserted vertically into a small opening on the top of the vertical
147 tube furnace which was preheated to the target temperature. All experiments were carried out at
148 atmospheric pressure. The capsule was placed upright on the closed bottom of the tube and located
149 in the hot spot of the furnace chamber. The temperature was monitored during the entire run using
150 the internal furnace thermocouple and an additional K-type thermocouple which was in direct
151 contact with the diffusion couple capsule. Temperatures are estimated to be accurate to $\pm 2^\circ\text{C}$ based
152 on the accuracy of the thermocouple device. The experimental durations ranged from one to 35
153 days depending on the temperature investigated (Table 2). The time needed for the capsule to heat
154 to the target temperature ranged between 3 and 5 minutes. The samples were quenched by sliding
155 the alumina tube out of the furnace and letting it cool in air to room temperature, leaving the
156 assembly in an upright position. Typical cooling times to a temperature of 200°C were in the range
157 of 3 to 4 minutes. After each experiment the samples were weighed again to confirm closed system
158 conditions during the experiment. To investigate a possible pressure effect on halogen diffusion
159 one experiment (HX2b) was conducted at 900°C and 100 MPa in a cold-seal pressure vessel within
160 a horizontal tube furnace. For this experiment Ag-Pd was used as capsule material. Even though
161 heating and quenching times were longer due to the more complex setup, the related error is
162 considered small compared to the long relative experimental durations (see 3.4 for further
163 discussion).

164 After quenching each capsule was embedded in epoxy, cut open parallel to the diffusion direction,
165 and polished for electron microprobe analysis (EMPA). All analyses were performed with an
166 electron microprobe of the type JEOL JXA 8200 in the Department of Geoscience of the University
167 of Mainz. F and Cl concentrations were measured simultaneously along with most major elements.

168 An acceleration voltage of 15 kV, a beam current of 12 nA and a probe diameter of 5 μm were used
169 for all quantitative analyses. For F and Cl dwell times of 120 s and 30 s were used, respectively.
170 The detection limits for F and Cl were both 60 ppm (1σ). Standards used for calibration are SrF₂
171 (F), tugtupite (Cl), VG-2 (Ca, Mg), VG-A99 (Fe, Si), MnTi (Mn, Ti) and orthoclase (Al, K). VG-
172 2, VG-A99 and a natural obsidian standard were analyzed repeatedly during each analytical
173 session. The data were corrected with the ZAF method. The profiles were acquired with a step-size
174 of 20 μm on the ends of the profile and smaller step-sizes of 10 and 5 μm for analyses closer to the
175 diffusion interface. This way it was possible to achieve a high spatial resolution of data necessary
176 to resolve slow diffusion but also cover a large range of diffusion widths while keeping analysis
177 time low. Additionally, one qualitative 2D compositional map of F, Cl and some major elements
178 (Si, Al, Na, Ca, Fe, Ti) was acquired for sample HX8 using a pixel-size of 3x3 μm , an acceleration
179 voltage of 15 kV and a beam current of 100 nA. Dwell times were 200 ms for each element.

180

181 **3. Results**

182 3.1 F and Cl diffusion coefficients

183

184 Diffusion coefficients for each experimental temperature were derived from the diffusion profiles
185 acquired by EPMA. In each sample both F and Cl were analyzed. Typical examples of diffusion
186 profiles are presented in Fig. 2. For each sample two profiles with distances ranging from 100 –
187 700 μm (except only one profile for HX9) were analyzed and processed to confirm homogeneity.
188 Additional backscattered electron images of all samples and a compositional map of sample HX8
189 (950°C) further confirmed the homogeneity of the major elements and the concentration gradient
190 of F and Cl along the diffusion interface produced by this experimental setup (Fig. 3). The semi-

191 quantitative compositional map was acquired with a size of 500 x 350 pixels and a spot size of 3
192 μm . Even though no quantification has been done to the raw data, the map still shows the sharp
193 difference in initial halogen contents of the two diffusion couple halves, in addition to the relative
194 homogeneity of major element composition across the charges. Concentrations of Si, Al, Fe and Ti
195 are very homogeneous in the investigated area. Ca concentration is mostly homogeneous but is
196 enhanced at some points close to the diffusion interface, likely due to the growth of apatite
197 microlites. Na concentration is homogeneous in most parts but is increased in small irregular
198 patches close to the diffusion interface. These Na-rich areas also correspond to low-concentration
199 zones of F and Cl. Profile 2 of HX8 was acquired crossing one of those low-concentration patches.
200 Despite these micro-scale compositional heterogeneities, the compositional map allowed us to
201 select appropriate traverse positions and discard the data points acquired in one of these patches to
202 calculate the diffusion coefficients just from the unaffected data. The compositional map along
203 with backscattered electron images additionally confirmed that no convection occurred in our high-
204 viscosity rhyodacitic diffusion couples. This agrees well with previous studies which demonstrated
205 that even in low-viscosity basaltic melts no convection influenced the experiments (Alletti et al.
206 2007; Balcone-Boissard et al. 2009). Even though a slight downward bend of the interface was
207 observed in sample HX8 (950°C, Fig. 3a) there is no effect on the resulting diffusion coefficients
208 calculated from linear profiles. A compilation of all analyzed diffusion profiles and the complete
209 compositional map can be found in the supplementary material.

210

211 Most concentration profiles show a relatively smooth transition between the enriched and the
212 depleted halves of the sample, which is especially true for the higher temperature runs (e.g. Fig.
213 2b). Diffusion lengths, i.e. the length over which significant departure from the original
214 concentration occurred, are up to 850 μm for F and 150 μm for Cl in the high-T experiments.

215 Experiments run at lower temperatures developed only short diffusion profiles (several tens of μm),
216 especially those of Cl. Profiles of Cl concentration in sample HX6 (750°C) were particularly short
217 with only few points falling in the transition zone between enriched and depleted endmembers.
218 Samples HX3 (850°C) and HX4 (800°C) had lower maximum concentrations than all other
219 experiments, supposedly due to our use of a piece of glass that was positioned close to the surface
220 of the stirred sample. Therefore, three additional experiments were performed at the lowest
221 temperatures (750, 800 & 850°C) running for longer experimental duration to improve the diffusion
222 width and thus data quality. However, as diffusivity has previously been shown to be concentration-
223 independent (e.g. Dingwell and Scarfe, 1984; Bai and Koster van Groos, 1994; Baker and Balcone-
224 Boissard, 2009; Zhang et al., 2010), we included the data of HX3 and HX4 in the results.

225

226 The diffusion coefficients for each profile and each species were calculated by fitting the
227 concentration data to the concentration-independent equation for constant one-dimensional
228 diffusivity between two semi-infinite media in a cartesian coordinate system (Crank 1975):

229

$$230 \quad C(x, t) = \frac{C_{low} + C_{high}}{2} + \frac{C_{low} - C_{high}}{2} * \operatorname{erf}\left(\frac{x - x_0}{2 * \sqrt{D * t}}\right) \quad (1)$$

231

232 where $C(x, t)$ is the concentration [ppm] at distance x [m] after the experimental time t [s]. C_{low}
233 [ppm] and C_{high} [ppm] are the concentrations of the respective halogen-poor and halogen-enriched
234 starting glasses. Parameter x_0 describes the position of the diffusion interface between the two
235 individual media and D [m^2/s] is the diffusion coefficient. The term *erf* defines the error function.
236 Parameters D , x_0 , C_{low} and C_{high} were determined by fitting equation (1) to the respective data.
237 Analyses with totals below 97 wt.% or F and Cl values below 2 times the detection limit (120 ppm,
238 2σ) were discarded. Fitting was carried out using the non-linear least squares method implemented

239 in a *MatLab*-based computer program. The program is easy to use, offers control over all important
240 parameters and provides the user with information about the goodness of fit (e.g. R^2 , plot of
241 residuals etc.) and confidence intervals calculated from the fit. Additionally, the user can fix C_{low}
242 and C_{high} when necessary, e.g. when the profile obtained by EPMA is too short to cover the end-
243 member concentrations of the starting glasses but information about the highest and lowest
244 concentrations is available from another profile analyzed in the same sample. The results obtained
245 by using this program were validated by cross-checking some results with commercially available
246 curve-fitting software (*CurveExpertPro*). Before fitting the data were filtered for outliers. All
247 calculated diffusion coefficients are compiled in table 2 together with the experimental conditions.

248

249 3.2 Temperature dependence of F and Cl diffusion

250

251 Diffusion coefficients of both F and Cl increase with increasing temperature (Fig. 4, Table 2). The
252 calculated diffusion coefficients for F range from $1.6 \times 10^{-15} \text{ m}^2/\text{s}$ (750°C) to $1.3 \times 10^{-13} \text{ m}^2/\text{s}$
253 (950°C). Those for Cl are up to 2 orders of magnitude smaller and range from $1.7 \times 10^{-17} \text{ m}^2/\text{s}$
254 (750°C) to $4.5 \times 10^{-15} \text{ m}^2/\text{s}$ (950°C). Diffusion profiles acquired in a single sample may yield
255 slightly different diffusion coefficients (F: <0.2 log units; Cl: <0.5 log units, both except HX7)
256 which partly accounts for the variability in diffusivities, cast in terms of an uncertainty value.
257 However, the results for one sample are in general within the same order of magnitude. Experiment
258 HX2b which was run at elevated pressure of 100 MPa and 900°C yielded diffusion coefficients of
259 F and Cl comparable to those obtained at atmospheric pressure and the same temperature (Table
260 2). We therefore conclude that pressure does not play a significant role in modifying F and Cl
261 diffusion in rhyodacitic melt over a P – T range relevant for volcanic processes. This agrees well

262 with the findings of other recent studies (Baker and Balcone-Boissard 2009; Zhang et al. 2010;
263 Böhm and Schmidt 2013).

264

265 As shown in previous studies F and Cl diffusion in silicate melts follow Arrhenian behavior
266 described by the following equation:

267

$$268 \quad D = D_0 * e^{-\frac{E_A}{R*T}} \quad (2)$$

269

270 where D is the diffusion coefficient [m²/s], D₀ is the pre-exponential factor [m²/s], E_A is the
271 activation energy [J/mol], R is the universal gas constant [8.3145 J/(mol K)] and T is the
272 temperature [K]. In order to determine the constants D₀ and E_A necessary to describe diffusion
273 behavior of the individual species, the diffusion coefficients calculated with equation (1) are plotted
274 in an Arrhenius diagram of log(D) vs. 10000/T [K] (Fig. 4) and the data is fitted to the linear
275 equation:

276

$$277 \quad \log(D) = -\frac{E_A}{\ln(10)*R*T} + \log(D_0) \quad (3)$$

278

279 E_A correlates with the slope of the line in the Arrhenius diagram and D₀ represents the intercept at
280 10000/T = 0. From our experiments we calculated pre-exponential factors of D₀ = 4.3 x 10⁻⁴ for F
281 and D₀ = 1.6 x 10⁻⁵ for Cl. The activation energies of F and Cl are similar (E_A (F) = 223.7 ± 31
282 kJ/mol; E_A (Cl) = 229.1 ± 52 kJ/mol) as can be seen from the slopes of the trendlines. The
283 uncertainty values of the activation energies represent the 95 % confidence interval calculated
284 during fitting.

285

286 3.3 Experimental and analytical uncertainties

287

288 The errors given for the diffusion coefficients (Table 2) were calculated during the fitting process
289 and represent the 95 % confidence intervals based on the fitted datapoints only. However, other
290 sources of uncertainty that may have affected the final Arrhenius parameters include: heating and
291 quenching times, spot distance, accuracy of temperature measurements and homogeneity of the
292 glass. After the experiments, few microlites (<1 vol.%) were present in some samples. In particular,
293 some apatite crystals up to 30 μm length crystallized close to the interface mainly in the halogen-
294 depleted sample. However, because we selected analysis traverses so as to avoid these microlites,
295 we did not observe and anomalous F and Cl concentrations. In some samples, small patches of F-
296 depletion and Na-enrichment developed close to the interface (Fig. 3). However, those patches are
297 recognizable from the backscattered images and were either avoided for diffusion profile
298 acquisition or corrected for during data reduction.

299 Due to the long experimental durations used (minimum 1 day) the short heating and quenching
300 times with a maximum of 5 minutes are considered to have minor effects only. For example, for
301 sample HX8 which had the shortest experimental duration of 87025s (~1 day) the diffusion
302 coefficients for F and Cl would change by only $\pm 0.01 \times 10^{-13}$ or 0.02×10^{-15} respectively (both <1
303 %), for experimental durations varying ± 5 minutes. These values are well within the calculated 95
304 % confidence intervals. For longer experimental durations the error induced by 5 minutes of
305 heating and quenching would be even smaller. For example: an error of 2 hours on the experimental
306 duration of sample HX9 (750°C; ~35 days) would change the resulting diffusion coefficients for F
307 and Cl by less than 1 %. As is shown in the results section, the diffusivities of F and Cl are
308 exceedingly slow, which precludes the development of a significant concentration profile at the
309 relatively short time scales of heating and quenching (max. 10 min total). Even at the highest

310 temperature investigated, well defined profiles of 500 μm needed one day to develop. Therefore,
311 zero-time experiments have not been carried out. To assess the error introduced by a non-perfect
312 angle between the analyzed profile and the diffusion interface we used the same samples (HX8 and
313 HX9) and calculated the diffusion coefficients for the case that the profile was not perfectly
314 perpendicular to the diffusion interface but tilted by up to 10° . In this case the actual distance
315 between each spot with regard to the diffusion interface would only be $\sim 98\%$ of the target distance
316 which results in a diffusion coefficient which is smaller by less than 4% for F and Cl in both
317 samples. The combination of both effects discussed above (5 minutes longer experimental duration
318 and diffusion profile tilted by 10° to the diffusion direction) would result in F and Cl diffusion
319 coefficients both being less than 5% smaller compared to the apparent value. This is still well
320 within the limits of the 95% confidence interval calculated from the fit of the concentration profile
321 and is therefore covered by the given error bars. The accuracy of the measured temperature during
322 the experimental run may have an effect on the calculation of the Arrhenian diffusion law.
323 However, the accuracy of the thermocouple device is at least $\pm 5^\circ\text{C}$ and the thermocouple was in
324 direct contact with the capsule during the whole run. Therefore, we consider the effect of possible
325 temperature deviations to be negligible compared to the relatively large 95% confidence intervals
326 of the activation energies (Fig. 4).

327

328 **4. Discussion**

329 **4.1 Influence of temperature**

330

331 Our experiments confirm that diffusion of both F and Cl are strongly positively correlated with
332 temperature and follow Arrhenian behavior (Fig. 4). The temperature dependences of both

333 elements, represented by their activation energies, are similar and suggest the same diffusion
334 mechanism. However, in the temperature range investigated F diffusion is always faster than Cl
335 diffusion, which is also true for the theoretical case of very low or very high temperatures due to
336 the similar activation energies. The experimental temperatures were chosen according to typical
337 natural magma storage conditions of rhyodacitic systems (e.g. Weber and Castro, 2017). Using the
338 model of Giordano et al. (2008) a glass transition temperature (T_G) of 687°C for the depleted half
339 and 692°C for the enriched half of the diffusion couple was calculated. We consider no significant
340 diffusion to happen below T_G which is supported by the fact that diffusion at 750°C was already
341 so slow it took weeks to develop proper diffusion profiles. Furthermore, at these low temperatures
342 the halves of the diffusion couple did not weld together perfectly over the whole interface, so at
343 lower temperatures the conditions necessary for a diffusion couple would not have been given.
344 However, diffusion at lower temperatures could still take place and could be significant in scenarios
345 of very slow magma ascent rates (e.g. Boudon et al. 2015), or during post-eruptive lava-flow
346 degassing (e.g. Schipper et al. 2015, 2019) rather than explosive eruptions.

347

348 4.2 F and Cl diffusion in different melt compositions

349

350 Fig. 5 illustrates the results of all published studies on F and Cl diffusion in silicate melts (Watson
351 and Bender 1980; Dingwell and Scarfe 1984, 1985; Bai and Koster van Groos 1994; Alletti et al.
352 2007; Balcone-Boissard et al. 2009; Böhm and Schmidt 2013; Fortin et al. 2017; Yoshimura 2018).
353 Only the results of anhydrous experiments are shown for better comparability to our study. In
354 general, diffusion coefficients in different silicate melt compositions vary by 3 orders of magnitude
355 for both F and Cl. Melts of basaltic composition yield the highest diffusivities compared to studies
356 on other compositions such as phonolite, dacite or rhyolite. Dingwell and Scarfe (1984, 1985) were

357 the first to study the diffusion of F in different synthetic silicate melt compositions. Their results
358 for jadeite melt are comparable to those of Alletti et al. (2007) who investigated melts of natural
359 basaltic composition and also studied Cl diffusion. Phonolitic compositions in general yield slower
360 diffusion coefficients for F and Cl with the exception of potassic phonolite from the Laacher See
361 eruption which yields F diffusivities similar to those of basalt (Balcone-Boissard et al. 2009). Böhm
362 and Schmidt (2013) also studied a natural phonolitic composition but at lower temperatures which
363 are potentially more significant to natural systems. They found intermediate diffusivities for F but
364 lower diffusivities for Cl compared to other phonolitic compositions. Watson and Bender (1980)
365 studied tracer diffusion of Cl in a synthetic melt composition and found results similar to basaltic
366 compositions of Alletti et al. (2007).

367 When our data is extrapolated to the temperature range that has been used by many previous studies
368 (~1200–1450°C), F diffusivity plots close to, yet below that in basalt, jadeite or Na-phonolite,
369 having a very similar activation energy as this determined for dry basalt (Fig. 5a). For Cl, even
370 when the data is extrapolated to high temperatures, diffusivity in our samples is still about one
371 order of magnitude slower than in Na-phonolite and dacite, and even slower compared to other
372 melts (Fig. 5b).

373 Studies investigating a similar, silicic melt composition are those of Bai and Koster van Groos
374 1994, Fortin et al. 2017 and Yoshimura 2018. Bai and Koster van Groos (1994) studied Cl diffusion
375 in granitic and haplogranitic melts using different approaches of which the “high-temperature
376 series” is the most similar approach to our study. This set of experiments was run with haplogranitic
377 melt at atmospheric pressure and under dry conditions in a temperature range from 850 – 1400°C.
378 They found a pre-exponential factor of $D_0 = 3.16 \times 10^{-9}$ and a relatively low activation energy of
379 $E_A = 86.2$ kJ/mol. By contrast, our study presents an almost three times larger activation energy
380 and 5 orders of magnitude larger pre-exponential factor resulting in significantly slower

381 diffusivities than those suggested by Bai and Koster van Groos (1994). This discrepancy is most
382 likely a result of the simplified melt compositions and experimental approaches. Bai and Koster
383 van Groos (1994) equilibrated haplogranitic melt with molten NaCl instead of using a diffusion
384 couple. This way not only diffusion, but also partitioning from the molten NaCl into the melt may
385 influence the results. Yoshimura (2018) used three different experimental setups based on the
386 absorption of Cl to a piece of natural obsidian. The source of Cl was either Cl₂, Cl₂ + H₂O or molten
387 NaCl. At 750 and 950°C and under mostly dry conditions he found $D_{Cl} = 2.0 \times 10^{-17} \text{ m}^2/\text{s}$ and D_{Cl}
388 $= 3.1 \times 10^{-16}$ to $3.3 \times 10^{-15} \text{ m}^2/\text{s}$ respectively. The lower temperature results (~750–800°C) are
389 based on molten Cl₂ as Cl source and agree well with our findings at the same temperatures.
390 However, at higher temperatures (>800°C) this experimental series yields slower diffusion than
391 our experiments. Instead, the higher value of his 950°C results, based on molten NaCl as Cl source,
392 are similar to our results. It is likely that the results of both of these studies are affected by other
393 mechanisms than pure diffusion, such as solubility and depolymerization of the melt due to Na-
394 infiltration (Yoshimura 2018), of which the latter would result in increased diffusivity. In addition
395 to Na-infiltration, the simplified haplogranitic melt composition used by Bai and Koster van Groos
396 (1994) has likely caused increased diffusivity as a result of the availability of more vacancies within
397 the melt structure compared to a natural melt (see 4.3 for further discussion). The application of
398 their results to natural systems and a direct comparison to our study is therefore difficult. The
399 general results of Fortin et al. 2017 who studied diffusive fractionation of Cl-isotopes in a dacitic
400 melt are comparable to those obtained in our experiments, even though their data were collected at
401 higher temperatures. The enhanced diffusivity in the dacite melt may be explained by the lower
402 content of SiO₂ in addition to the higher amount of network modifying cations (e.g. Fe, Mg)
403 compared to the rhyodacitic composition investigated in this study.

404

405 4.3 Effects of melt structure on diffusion

406

407 It has been shown in the past that diffusion is highly dependent on the composition of the medium
408 and on its structure (e.g. Jambon and Semet, 1978; Margaritz and Hofmann, 1978; Lowry et al.,
409 1982; Henderson et al., 1985). Typical natural melt compositions comprise a wide range of
410 chemical compositions and vary greatly in their structure which is determined by various
411 parameters such as the amount of SiO₂, network modifying cations, volatiles and the resulting
412 degree of polymerization, defined by parameters such as NBO/T, i.e. the ratio of non-bridging
413 oxygen atoms over tetrahedrally coordinated ions (Mysen 1988). The classic interpretation that
414 diffusion in silicate melts depends strongly on the radius of the diffusing element has been shown
415 to be true mainly for noble gases and monovalent cations (Henderson et al. 1985; Lux 1987; Alletti
416 et al. 2007). This strong correlation of diffusivity and atomic radius was interpreted to indicate the
417 movement of these elements through vacancies in the melt structure (Henderson et al. 1985). For
418 divalent and trivalent cations this correlation does not apply, instead the diffusivities of those
419 elements seem to be mostly independent of their respective ionic radius. Rather, Henderson et al.
420 (1985) propose diffusion of divalent and trivalent elements is dominantly controlled by exchange
421 mechanisms that preserve local charge balance and is therefore mainly dependent on the proportion
422 of network modifying cations. Figure 6a illustrates the correlation of atomic radius and diffusivity
423 of the most recent studies on halogen diffusion. The data shown represents diffusivities at 1300°C.
424 As our study and this of Böhm and Schmidt (2013) only provide data up to lower temperatures,
425 diffusion coefficients have been scaled to 1300°C using equation (2). Data of Alletti et al. (2007)
426 follow a behavior similar to di- and trivalent elements for F, Cl and Br in natural basaltic melt,
427 observing almost no correlation between the diffusion coefficient and the atomic radius of the
428 diffusant. Therefore, they interpret the diffusion of halogens in basaltic melt to be dominated by

429 exchange mechanisms. Our data instead shows consistently slower diffusion of Cl which has a
430 larger radius than F. Other recent experimental data which was mostly obtained from phonolitic
431 compositions also shows a correlation between the diffusivity and the atomic radius, even though
432 very variable and not as strong as observed in our experiments. The correlation of the variable
433 diffusivity between F and Cl and their atomic radius in more evolved melts suggests that the degree
434 of polymerization does have an influence on diffusion and the diffusion mechanism. For a better
435 estimate of the degree of polymerization of the different melts, we calculated NBO/T and the
436 network modifier parameter (SM) for the melts discussed. For NBO/T we followed the method of
437 Mysen (1988) and split FeO_{tot} equally (wt.%) between Fe_2O_3 and FeO (Giordano et al. 2008). SM
438 was calculated after Giordano and Dingwell (2003) as the molar oxide sum of network modifiers:
439 $\text{SM} = \sum(\text{Na}_2\text{O} + \text{K}_2\text{O} + \text{CaO} + \text{MgO} + \text{MnO} + \text{FeO}_{\text{tot}}/2)$ [mol.%]. The highest SM value (~32) and
440 the highest NBO/T ratio (0.5) corresponds to the basaltic melt of the study of Alletti et al., (2007)
441 which comprises the fastest diffusion and almost no correlation between the atomic radius and the
442 diffusion coefficient. In contrast, the rhyodacitic melt used in our study has the lowest NBO/T ratio
443 (0.04) and the lowest SM (~12) amongst previous studies and yielded generally slow diffusion
444 with diffusion coefficients of F and Cl differing by up to 2 orders of magnitude. The phonolitic
445 melts (Balcone-Boissard et al. 2009; Böhm and Schmidt 2013) which yield intermediate
446 diffusivities range between NBO/T = 0.08–0.09 with SM ~18–20. However, the K-phonolite series
447 shows only weak correlation between diffusivity and atomic radius similar to the basaltic melt.
448 Figure 6b shows the correlation between parameter SM and the diffusion coefficient of this and
449 other recent studies. From the diagram, it can be seen that diffusivities of F and Cl are expected to
450 be similar at higher SM but deviate significantly with decreasing SM.
451 In basaltic melts where the proportion of network modifying cations (i.e. SM) is sufficiently high
452 and polymerization is relatively low, diffusion is likely dominated by the charge balancing

453 exchange of ions. As this mechanism mainly depends on the charge of the diffusant, the atomic
454 radius has no significant influence on the diffusion behavior. In more highly polymerized melts
455 with lower SM, ion-exchange is less efficient and diffusion along vacancies in the melt structure
456 becomes more important. However, among elements of the same charge, this mechanism is likely
457 most efficient for those with a small atomic radius, which potentially explains the observed
458 differences in diffusivity between F and Cl in our and other studies. Even though the correlation
459 between D, the proportion of network modifying cations and atomic radius becomes clear with
460 these results there is still the need to investigate the diffusion behavior of other halogens (Br & I)
461 and more intermediate silicate melts to validate and refine this hypothesis.

462 In addition to the obtained diffusion data, the viscosity of the sample material was measured by
463 micropenetration at three different temperatures (690, 720 and 750°C). The viscosities were used
464 to calculate the Eyring diffusivity of oxygen in our melt using the following equation (Glasstone
465 et al. 1941):

466

$$467 \quad D = \frac{k_B * T}{\lambda * \eta} \quad (4)$$

468

469 where k_B is the Boltzmann constant [J/K], T is temperature [K], η is the viscosity [Pas] and λ is the
470 effective jump distance of the diffusant [m], in this case oxygen (2.8 Å, Shimizu and Kushiro,
471 1984). The obtained diffusion coefficients were in turn fitted to the Arrhenius equation in order to
472 extrapolate the Eyring diffusivity to higher temperatures, relevant for our experiments. The
473 calculated diffusivity of oxygen plotted in an Arrhenius diagram has a similar slope (i.e. activation
474 energy) as F and Cl but is located more than 4 orders of magnitude below Cl diffusivity (Fig. 4
475 inset). With this high contrast in diffusivity, the Si-O bonds in the melt appear to behave as if in
476 fixed structural sites relative to the diffusing F- and Cl-ions (Dingwell 1990; Dingwell and Webb

477 1990), so their diffusion is not affected by self-diffusion of the silicate structure over the relevant
478 time-scales. At higher Eyring mobilities, i.e. lower viscosities, the jump frequency of Si-O bonds
479 would approach the jump frequency of the investigated diffusant and by this the observed
480 diffusivity of F and Cl would likely be increased, as each Si-O jump would allow the diffusing ion
481 to jump to another site as well. Even though the Eyring equation does not predict oxygen diffusivity
482 perfectly (e.g. Liang et al., 1996), it still gives a good approximation of the structurally defined
483 lower limit of transport rates in the melt.

484

485 4.4 Potential for diffusive fractionation

486

487 The diffusivities presented in section 3 are based upon experiments utilizing dry melt. The main
488 driving force for the growth of bubbles in a natural magma, is however, the exsolution of H₂O and
489 CO₂, neither of which is present in our experimental melt. The following discussion of diffusive
490 fractionation is therefore hypothetical, and based on the assumption that the ratio of diffusivities
491 between F and Cl is equal in dry and wet rhyodacitic melt, which has been shown to be largely the
492 case for other silicic melt compositions by previous studies (e.g. Baker and Balcone-Boissard
493 2009).

494 As shown in section 3.1 diffusivities of F and Cl differ by up to 2 orders of magnitude. This
495 variation is likely to produce fractionated volatile compositions during bubble growth, with the
496 faster diffusing F being enriched in the volatile phase compared to the slower diffusing Cl. Recent
497 studies showed that even with only small differences in diffusivity, fractionation can be significant
498 (e.g. Fortin et al. 2017) but is highly dependent on the ratio of the bubble growth rate (R) and the
499 diffusivity (D) ((R/D) Watson 2017), and of course assuming the same vapour-melt partition
500 coefficient (K) for both diffusants. Watson (2017) discussed this relationship on the basis of

501 different isotopes of the same volatile element, which have the same vapour-melt partition
502 coefficient but small differences in diffusivity and showed that purely diffusive fractionation in
503 bubbles up to 10 mm is most pronounced at R/D values $<10^4 \text{ m}^{-1}$. However, while the same
504 principles generally apply for other diffusing volatile elements such as F and Cl, the vapour-melt
505 partition coefficient cannot assumed to be identical for both of these elements and thus could
506 greatly influence the exact relative behavior of F and Cl during bubble growth and degassing. In
507 particular, F is known to behave paradoxically in silicate melts, in that it exhibits high solubility in
508 silicic melts (Carroll and Webster 1994) and usually prefers partitioning into melt rather than
509 aqueous fluid (e.g. Lowenstern et al. 2012). However, F degassing during volcanic eruptions is
510 commonly measured, e.g. in volcanic plumes and fumaroles, and preferred partitioning into the
511 vapour phase has been determined experimentally in basaltic compositions (Baker and Alletti
512 2012). F partitioning may also be influenced by late-stage crystallization of magma, the so-called
513 “second boiling” process. For example, Schipper et al. (2019) showed based on glass inclusion
514 analysis that only 2 wt.% of F and 7 wt.% of Cl were degassed during the 2011-2012 Cerdón Caille
515 eruption’s explosive phase, but the bulk of these halogens were later degassed during lava flow
516 emplacement and coincident syn-eruptive laccolith intrusion (Castro et al. 2016). Schipper et al.
517 2019 postulate that late-stage crystallization of the lava and laccolith played a key role in driving
518 otherwise soluble F out of the melt and into degassing pathways. In summary, the lack of accurate,
519 composition-specific data on the partitioning behavior of F hinders precise modelling of diffusive
520 fractionation of F and Cl during bubble growth. Future experimental efforts could profitably focus
521 on constraining F partitioning in a variety of melt compositions and under a range of different
522 degassing regimes (e.g. open- vs closed-system).

523 We speculate that diffusive fractionation will be most important when magma ascent rates are slow
524 enough to favor significant transport of F and Cl to growing bubbles, which in turn depends on the

525 characteristic spacing of bubbles within the melt, as determined by bubble number density and size
526 distribution (Gonnermann and Manga 2007). We therefore propose that diffusive halogen
527 fractionation may be most significant during scenarios of slow mass effusion such as lava dome
528 forming eruptions or the slow cooling of lava flows or shallow intrusions (Castro et al. 2016;
529 Schipper et al. 2019).

530

531 **5. Implications**

532

533 Diffusion of F and Cl in dry rhyodacitic melt was characterized experimentally using the diffusion
534 couple technique. Even though the temperature dependence of the two is very similar, there is a
535 large difference in diffusive speeds over the temperature range investigated which suggests that
536 diffusive fractionation with an increasing F/Cl ratio is a possible outcome of magma degassing
537 under conditions of relatively slow ascent and bubble growth, as would be expected during lava
538 dome formation. Diffusion of F and Cl was shown to be generally slower in our rhyodacitic melt
539 compared to basaltic or other simplified silicate melts. We interpret this to be related to the higher
540 degree of polymerization in the rhyodacite melt. This is supported by the fact that the correlation
541 between the atomic radius of the diffusing component and its diffusivity is more pronounced in
542 more silicic, more viscous melts. We propose that concentrations of F and Cl in volcanic gases,
543 when monitored, could help identify volcanic unrest. However, for a comprehensive understanding
544 of diffusive fractionation during degassing, knowledge of halogen fluid/melt partition coefficients
545 in a range of melt compositions are still needed, in addition to other parameters such as bubble
546 nucleation and growth rates under diverse magma ascent scenarios. With these important factors

547 in hand, we anticipate future modeling-based studies on diffusive fractionation processes at
548 volcanoes (e.g. Watson 2017).

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550

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694 **Tables and table captions**

695

696

697 **Table 1:** *Chemical composition of Hekla H3 pumice and the resulting synthesized glasses.*
 698 *Synthesized compositions include corrections for Na-loss. Major elements are normalized to 100*
 699 *wt.%.*

wt.%	H3 pumice ^a whole rock	enriched glass <i>n</i> = 15	1 SD	depleted glass <i>n</i> = 14	1 SD
SiO ₂	69.93	68.47	0.35	68.26	0.31
TiO ₂	0.34	0.46	0.02	0.46	0.02
Al ₂ O ₃	14.72	14.70	0.12	14.57	0.08
FeO	4.83	4.60	0.29	4.76	0.06
MgO	0.13	0.46	0.02	0.44	0.02
MnO	0.14	0.13	0.02	0.15	0.02
CaO	2.66	2.85	0.07	2.82	0.05
Na ₂ O	4.85	6.24	0.12	6.35	0.07
K ₂ O	2.33	2.10	0.03	2.20	0.03
P ₂ O ₅	0.05	-	-	-	-
F	a	0.53	0.02	0.04	0.03
Cl	a	0.26	0.01	0.02	0.02
Total	99.1	101.35	0.33	100.07	0.43

700 ^adata from Weber and Castro (2017) obtained by XRF. Significant F (1800 ppm) and Cl (500
 701 ppm) concentrations are reported for matrix glass analyses (EPMA) of the H3 pumice.

702 **Table 2:** Conditions and results of all experiments.

sample	T [°C]	time [s]	profile #	D_F	\pm^a	C_{max}	C_{min}	D_{Cl}	\pm^a	C_{max}	C_{min}
HX6	750	851880	1	1.6E-15	3.8E-16	5428	944	5.2E-17	7.8E-17	2647	292
			2	2.1E-15	5.7E-15	5473	988	7.5E-17	4.0E-17	2710	320
HX9	750	3023535	1	1.6E-15	4.3E-16	5657	996	1.7E-17	8.4E-18	2881	353
HX4	800	359334	1	2.6E-15	9.1E-16	3355	548	4.6E-17	3.5E-17	1083	189
			2	3.2E-15	1.3E-15	3409	498	8.6E-17	1.3E-16	1057	188
HX7	800	587235	1	6.6E-15	1.6E-16	5549	758	4.9E-17	1.4E-17	2732	257
			2	1.1E-14	1.9E-15	5580	671	1.6E-16	4.2E-17	2714	291
HX3	850	171502	1	1.6E-14	5.0E-15	3829	1045	3.9E-16	2.0E-16	1534	326
			2	1.9E-14	5.9E-15	3928	1014	4.5E-16	2.4E-16	1471	319
HX10	850	266855	1	1.9E-14	4.4E-15	5624	947	3.9E-16	1.2E-16	2740	318
			2	2.0E-14	5.1E-15	5482	970	7.4E-16	4.0E-16	2733	315
HX5	900	154560	1 ^b	4.7E-14	6.7E-15	5186	501	1.0E-15	2.0E-16	2762	222
			2	4.6E-14	8.9E-15	5186	501	8.7E-16	1.9E-16	2758	196
HX8	950	87025	1	1.3E-13	2.5E-14	5491	917	4.5E-15	1.1E-15	2721	302
			2	1.1E-13	1.4E-14	5304	993	1.5E-15	6.7E-16	2715	294
HX2b ^c	900	83400	1	4.7E-14	9.3E-15	5507	832	1.4E-15	4.8E-16	2726	310
			2	3.4E-14	9.2E-15	5320	859	1.3E-15	4.0E-16	2747	311

703 ^a 95% confidence interval calculated from the fit

704 ^b C_{max} and C_{min} of F fixed from profile 2

705 ^c experiment run at 100 MPa

706 **Figure captions**

707

708

709 **Fig. 1:** Capsule geometry used for diffusion couple experiments. a) Schematic section through an
710 experimental capsule. b) Finished capsule before the experiment.

711

712 **Fig. 2:** Representative concentration – distance diagrams of F and Cl obtained by electron
713 microprobe analysis. Also shown are the corresponding fits of equation 3.1 and the respective
714 diffusion coefficients as determined by fitting. a) sample HX9 (750°C / ~35 days) b) sample HX10
715 profile # 1 (850°C / ~3 days)

716 **Fig. 3:** Results of point- and map-analysis of sample HX8 (950°C). a) Backscattered electron image
717 showing the slightly bent diffusion interface and irregular lower density patches (darker). Yellow
718 lines show the position of two profiles acquired with point-analysis. b) Diffusion concentrations
719 determined for F and Cl along the left profile in a), from A to B. c) Compositional maps of F, Cl,
720 Na and Si. Maps of F and Cl show a clear concentration gradient parallel to the diffusion interface.
721 Note the high homogeneity of Si and Na apart from irregular patches of Na enrichment. Please see
722 text for further discussion.

723 **Fig. 4:** Diffusion coefficients for F and Cl determined in this study plotted on an Arrhenius
724 diagram. Solid symbols represent experiments at 1 atm while open gray symbols represent
725 pressurized conditions (100 MPa). Results from the 100 MPa series have not been used for fitting.
726 The inset shows the position of the Arrhenius laws of F and Cl relative to the calculated self-
727 diffusion of oxygen. See 4.3 for further explanation. Error bars correspond to the 95 % confidence-

728 interval calculated from fitting of the diffusion coefficients. Errors in x-direction are within the
729 symbol-size.

730

731 **Fig. 5:** Overview of Arrhenian diffusion for different melt compositions determined in this and
732 other studies. a) Fluorine b) Chlorine. Results of experiments using dry compositions are shown
733 only. Bold lines represent data obtained in this study and dashed line shows the same data
734 extrapolated to higher temperatures. Note the different y-axis scaling between the diagrams.

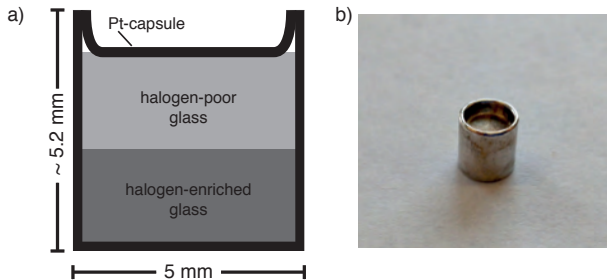
735

736 **Fig. 6:** Diagrams showing the correlation of diffusivity with the atomic radius and the sum of the
737 network modifying cations of this and other recent studies. a) Atomic radius vs. D . D of basalt and
738 Na-phonolite show only weak correlation with the atomic radius, while diffusion in rhyodacite is
739 strongly correlated. Lines marked with “*” are calculated from equation (2). b) Diffusivity of F
740 and Cl as a function of the proportion of network modifying cations in the host melt, represented
741 by the parameter SM ($= \sum \text{Na}_2\text{O} + \text{K}_2\text{O} + \text{CaO} + \text{MgO} + \text{MnO} + \text{FeO}_{\text{tot}}/2$ [mol.%]). Dashed lines
742 indicate a simplified exponential fit of the form $y = ae^{bx}$ for each set of datapoints. As they both
743 represent the same SM value, Na- and K-phonolite together have been accounted for as one
744 datapoint with averaged D . All diffusivities presented are for anhydrous conditions at 1250°C. See
745 text for further discussion.

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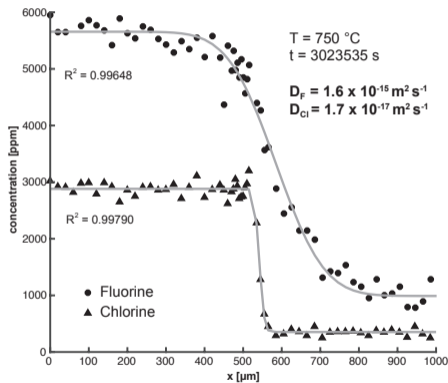
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Fig. 1



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Fig. 2 a) HX9



b) HX10

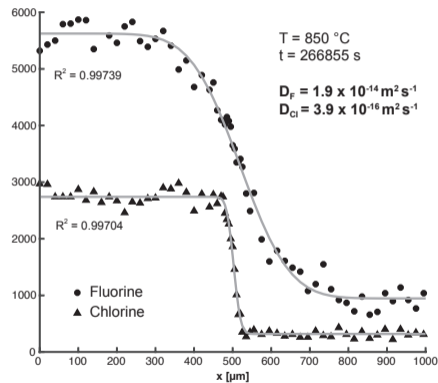


Fig. 3

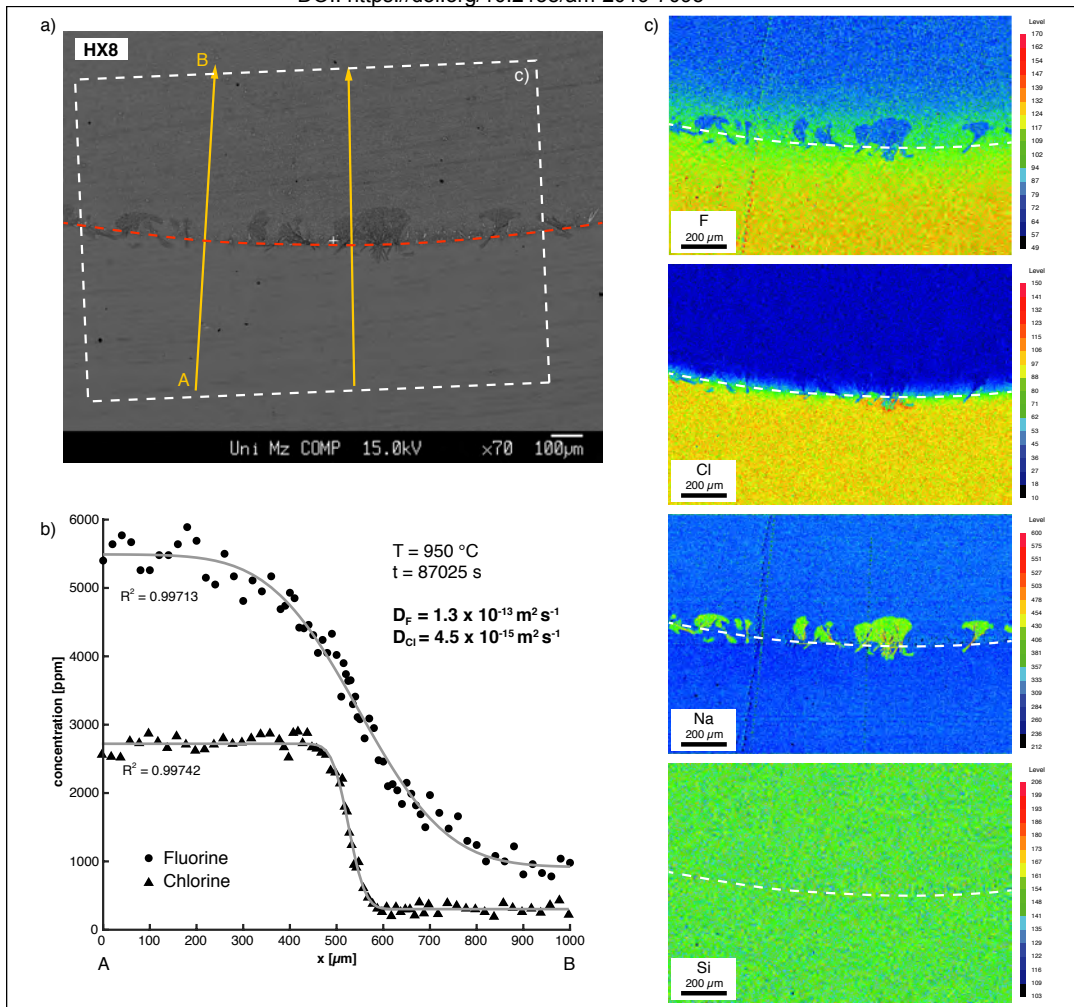


Fig. 4

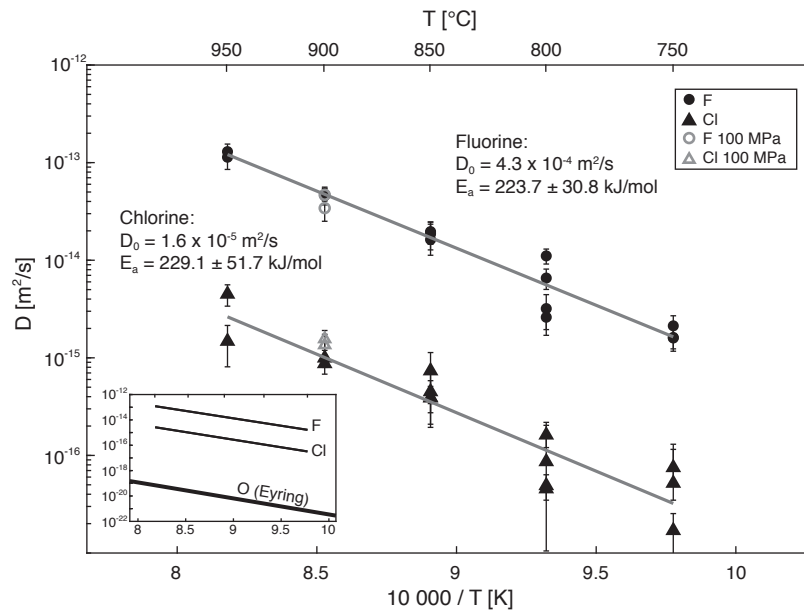


Fig. 5

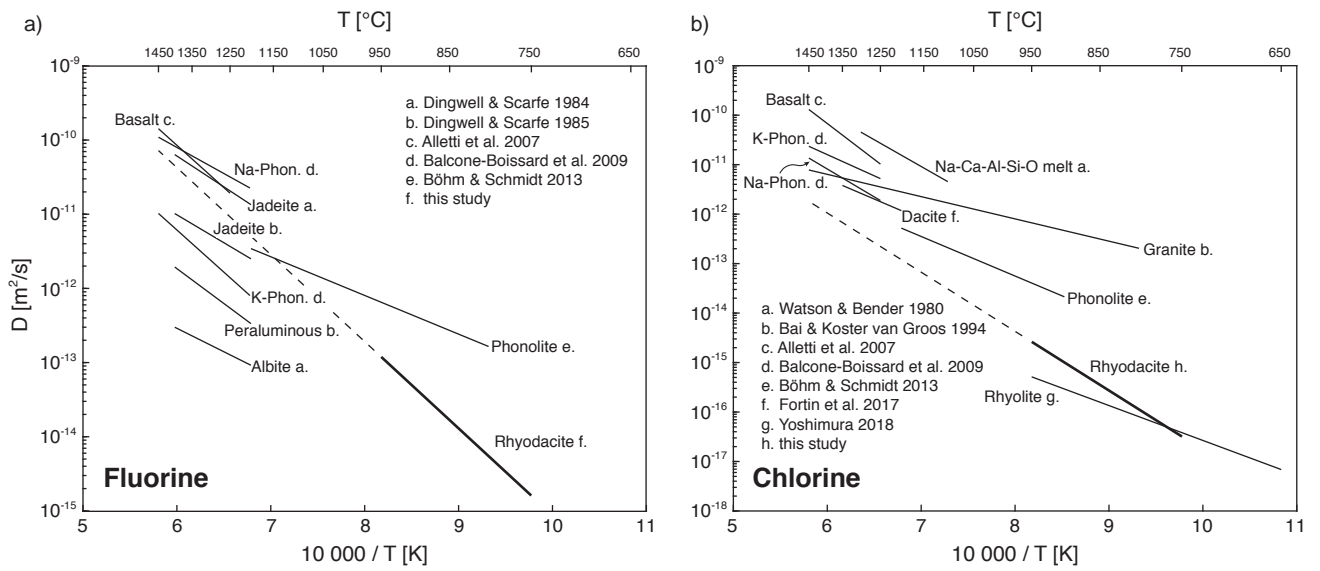


Fig. 6

