

Revision 3

1 **The spin state of Fe³⁺ in lower mantle bridgmanite**

2 Ryosuke Sinmyo^{*}, Catherine McCammon, Leonid Dubrovinsky

3 Bayerisches Geoinstitut, Universitaet Bayreuth, D-95440 Bayreuth, Germany

4 ^{*}Corresponding author. (E-mail: ryosuke.sinmyo@elsi.jp) Now at Earth-Life Science

5 Institute, Tokyo Institute of Technology

6

7 **Abstract**

8 Iron- and aluminum-bearing MgSiO₃ bridgmanite is the most abundant mineral in the
9 Earth's interior; hence its crystal chemistry is fundamental to expanding our knowledge
10 of the deep Earth and its evolution. In this study, the valence and spin state of iron in
11 well characterized Al-free Fe³⁺-rich bridgmanite were investigated by means of
12 Mössbauer spectroscopy to understand the effect of ferric iron on the spin state. We
13 found that a minor amount of Fe³⁺ is in the low spin state above 36 GPa and that its
14 proportion does not increase substantially with pressure up to 83 GPa. This observation
15 is consistent with recent experimental studies that used Mössbauer and X-ray emission
16 spectroscopy. In the Earth's deep lower mantle, Fe³⁺ spin crossover may take place at
17 depths below 900 and 1200 km in pyrolite and MORB, respectively. However, the
18 effect of spin crossover on physical properties may be small due to the limited amount

19 of Fe³⁺ in the low spin state.

20

21 **Introduction**

22 The crystal chemistry of terrestrial minerals is fundamental to understanding
23 the Earth's interior and its evolution. Iron- and aluminum-bearing MgSiO₃ bridgmanite
24 (Bdg) is the most abundant mineral in the Earth's interior and contains a substantial (up
25 to about 20 at. %) amount of iron, which is the fourth most abundant element in the
26 mantle. Bdg can incorporate a substantial amount of Fe³⁺ (e.g., McCammon 1997; Frost
27 et al. 2004), while olivine, the dominant mineral in the upper mantle, accommodates
28 essentially no Fe³⁺ (e.g., McCammon 2005). The substitution of Fe³⁺ for Fe²⁺ has a
29 large influence on physical and chemical properties of Bdg such as elastic constants,
30 iron partition coefficients and electrical/thermal conductivity (Ohta et al. 2010; 2014;
31 Boffa Ballaran et al. 2012; Sinmyo and Hirose 2013; Sinmyo et al. 2014a; Wolf et al.
32 2015; Yoshino et al. 2016). Moreover, both experiments and theoretical calculations
33 suggest that iron in Bdg undergoes spin crossover under deep mantle conditions (Badro
34 et al. 2004; Lin et al. 2012; Potapkin et al. 2013; Mashino et al. 2014). Although iron
35 spin crossover can strongly influence the physical properties of lower mantle minerals,
36 e.g., elastic constants and thermal/electrical conductivity (Lundin et al. 2008; Keppler et

37 al. 2008; Ohta et al. 2010; Catalli et al. 2011), the pressure of spin crossover in Bdg is
38 not well constrained (see reviews by Lin et al. 2013; McCammon et al. 2013 and
39 references therein). Challenges arise due mainly to the presence of iron in two different
40 valence states (Fe^{2+} and Fe^{3+}), and two different sites in the perovskite structure (8-12
41 coordinated A-site and octahedral B-site). Fe^{2+} , which is observed to occupy the A-site,
42 undergoes spin crossover from high spin to intermediate spin (McCammon et al. 2008;
43 Lin et al. 2008; Potapkin et al. 2013), although theoretical calculations do not support
44 the existence of the intermediate spin state of Fe^{2+} (Hsu et al. 2011; Metsue and
45 Tsuchiya 2012). There is general consensus that Fe^{3+} remains in the high spin state to at
46 least 100 GPa when it occupies the A-site (Catalli et al. 2010, 2011; Fujino et al. 2012;
47 Lin et al. 2012; Potapkin et al. 2013; Glazyrin et al. 2014; Kuppenko et al. 2014), but
48 undergoes spin crossover from high spin to low spin at 20~60 GPa when it occupies the
49 B-site (Catalli et al. 2010; 2011; Fujino et al. 2012; Lin et al. 2012; Kuppenko et al.
50 2015). These observations strongly suggest that the spin crossover pressure is
51 dominantly controlled by Fe^{3+} in the B-site. It has been reported that Fe^{3+} occupies the
52 A-site of Bdg regardless of the oxidation state concentration, Al content and pressure
53 (Kudoh et al. 1990; Vanpeteghem et al. 2006; Glazyrin et al. 2014), while some recent
54 studies have reported that Al-free and Fe^{3+} -bearing Bdg accommodate a detectable

55 amount of Fe^{3+} in the B-site (Catalli et al. 2011; Lin et al. 2012; Hummer and Fei 2012;
56 Kuppenko et al. 2015). Since previous studies have investigated Bdg with a limited
57 amount of Fe^{3+} , the electronic state of Fe^{3+} is not well understood, even though the
58 electronic state of iron in Al-free Fe^{3+} -rich Bdg may have a significant effect on the
59 physical/chemical properties of subducted depleted lithospheric mantle with
60 harzburgitic bulk composition (Ringwood 1991). Furthermore, it is recognized that
61 sample synthesis in the diamond anvil cell by laser heating may cause heterogeneity in
62 chemical composition and/or redox state of the sample (e.g., Fialin et al. 2008; Sinmyo
63 and Hirose 2010). Therefore, to minimize the experimental uncertainty, the $\text{Fe}^{3+}/\Sigma\text{Fe}$
64 ratio of samples pre-synthesized using a large volume press should be characterized
65 before the sample is compressed with a diamond anvil cell. Here, our study aims to
66 explore the spin state of iron under lower mantle conditions by investigating the rarely
67 studied, but well defined Al-free Fe^{3+} -rich Bdg, which provides insight into the complex
68 effects of Al^{3+} and Fe^{3+} on the spin state.

69

70 **Experimental procedure**

71 The sample was synthesized using a multi anvil apparatus. A mixture of fine
72 grained SiO_2 [55.23 wt %] + MgO [37.05 wt %] + $^{57}\text{Fe}_2\text{O}_3$ (90% enriched) [7.72 wt %]

73 was ground well for about one hour, and then dried at 1273 K in a furnace for one day.
74 We used Fe₂O₃ as a starting material to maximize Fe³⁺ content in the sample. The final
75 mix was loaded into a gold capsule and then packed into a MgO container. The
76 multi-anvil synthesis used LaCrO₃ for the heater, and was run for 45 minutes at pressure
77 and temperature conditions of 26 GPa and 1973 K, respectively. The synthesized
78 sample was examined using a field-emission-type scanning electron microscope (SEM)
79 (Leo Gemini 1530) and the chemical composition was determined using an electron
80 microprobe (JEOL JXA-8200) operated at 15 kV and 15 nA. Phase identification was
81 performed by powder X-ray diffraction (XRD) using a FR-D high-brilliance Rigaku
82 X-ray diffractometer with Mo-K α radiation operated at 55 kV and 60 mA.

83 The high-pressure Mössbauer spectroscopic study was conducted using a
84 diamond anvil cell. Diamonds with 250 μ m diameter culets were used as anvils. The
85 polycrystalline sample from the multianvil synthesis was loaded into a hole drilled in a
86 rhenium gasket. Mössbauer spectroscopy was conducted using a nominal 370 MBq
87 ⁵⁷Co high specific activity source in a rhodium matrix. The velocity scale was calibrated
88 relative to ⁵⁷Fe foil. Mössbauer spectra were analyzed using the MossA software
89 package (Prescher et al. 2012). Mössbauer spectra were collected with increasing
90 pressure up to 83 GPa, and then the pressure was gradually decreased toward ambient

91 conditions.

92

93 **Results**

94 The sample synthesized using the multianvil press was characterized by XRD,
95 Mössbauer spectroscopy and electron microprobe analysis. The electron microprobe
96 analysis yielded a chemical composition of $\text{Mg}_{0.971(11)}\text{Fe}_{0.064(4)}\text{Si}_{0.979(7)}\text{O}_3$. The grain size
97 was around 50 μm in diameter. Although a trace amount of quenched liquid material
98 was observed at the very edge of the sample, it was easily removed from the capsule
99 before selecting the Bdg sample for diamond anvil cell experiments. Most likely due to
100 the partitioning of iron between liquid and Bdg, the iron content in the Bdg was
101 depleted in comparison to the starting material. XRD measurements showed that the
102 obtained sample was a single phase of Bdg. The $\text{Fe}^{3+}/\Sigma\text{Fe}$ ratio of the perovskite was
103 determined to be 0.53(6) according to Mössbauer measurements at ambient conditions.
104 Based on this result, the chemical composition can be described as
105 $\text{Mg}_{0.971}\text{Fe}^{2+}_{0.030}\text{Fe}^{3+}_{0.034}\text{Si}_{0.979}\text{O}_3$. The total charge of the cations slightly exceeds six
106 (6.02), which may be due to vacancies in the crystal lattice (Hummer and Fei 2012). A
107 quantitative discussion is not possible, however, since the excess amount (+ 0.3 %) is
108 smaller than the uncertainty in the measurements ($\sim 1\%$).

109 Mössbauer spectra were collected up to 83 GPa at 15 pressure points in total
110 (Fig. 1). The pressure variation of the obtained hyperfine parameters center shift (CS)
111 and quadrupole splitting (QS) are summarized in Figure 2 and Table 1. The spectra at
112 ambient conditions are composed of two components corresponding to Fe^{2+} with higher
113 CS and QS values (hereafter Fe^{2+} #1) and Fe^{3+} with lower CS and QS values (Fe^{3+} #1).
114 The QS values of Fe^{2+} #1 and Fe^{3+} #1 slightly increase with increasing pressure, while
115 their CS values are almost constant (Fig. 2). Above 12 GPa, we observed a component
116 with a high QS value (Fe^{2+} #2), which corresponds to Fe^{2+} in the distorted A-site of Bdg
117 (McCammon et al. 2008; Kuppenko et al. 2014). The abundance of Fe^{2+} #2 does not
118 change substantially from 12 to 83 GPa (Fig. 3). At higher pressure, an additional
119 component was observed above 36 GPa (Fe^{3+} #2). This new component exhibits almost
120 zero CS and QS similar to the value for Fe^{3+} #1. These hyperfine parameters are
121 comparable to those of the Fe^{3+} component found in Al, Fe-bearing Bdg at around 40
122 GPa after laser heating (Kuppenko et al. 2015). As in the case of the Fe^{2+} #1 and Fe^{3+} #1
123 components, the QS values of Fe^{2+} #2 and Fe^{3+} #2 slightly increase with pressure, in
124 contrast to the constant CS value (Fig. 2). Parallel to the slight increase in abundance of
125 Fe^{3+} #2 with increasing pressure from 40 to 53 GPa, the abundance of Fe^{3+} #1 decreases
126 to compensate, maintaining an essentially constant amount of Fe^{3+} (Fig. 3). The

127 hyperfine parameters vary with pressure in a highly consistent way during compression
128 and decompression (Fig. 2), and the $\text{Fe}^{3+}/\Sigma\text{Fe}$ ratio refined independently at each data
129 point remains nearly constant throughout the experiment (Fig. 3).

130

131 **Discussion**

132 **Fe^{3+} in Al-free Bdg**

133 It is well known that the $\text{Fe}^{3+}/\Sigma\text{Fe}$ ratio in Al-bearing Bdg is remarkably high
134 (about 0.5) even under low oxygen fugacity conditions (e.g., McCammon 2005). The
135 high $\text{Fe}^{3+}/\Sigma\text{Fe}$ ratio can be explained by a coupled substitution mechanism involving
136 Fe^{3+} -Al to Mg-Si in Bdg (Frost et al. 2004; McCammon et al. 2004). In this study, the
137 $\text{Fe}^{3+}/\Sigma\text{Fe}$ ratio was about 0.5 in the Al-free Bdg synthesized from Fe^{3+} -rich starting
138 material, which is considerably higher than that for Al-free Bdg synthesized from
139 Fe^{2+} -dominant material (Frost et al. 2004; McCammon et al. 2004; Sinmyo et al. 2008).
140 This observation suggests that $\text{Fe}^{3+}/\Sigma\text{Fe}$ of Al-free Bdg depends on oxygen fugacity as
141 suggested by McCammon et al. (2004). Indeed, Hummer and Fei (2012) and Mashino et
142 al. (2014) reported a $\text{Fe}^{3+}/\Sigma\text{Fe}$ ratio close to one in Bdg synthesized from Fe^{3+} -rich
143 material. Fe^{3+} #1 most likely occupies the A-site of Bdg because ferric iron mostly
144 occupies the A-site of Bdg at ambient conditions when the ferric iron concentration is <

145 0.04 per formula unit (Hummer and Fei 2012; Sinmyo et al. 2014b). The Fe³⁺ #2
146 component shows moderate QS ~ 1 and low CS ~ 0 (Table 1). These values are
147 consistent with low spin Fe³⁺ in the octahedral sites of the high-pressure phase of
148 FeOOH (Xu et al. 2013), CaFe₂O₄ (Greenberg et al. 2013), rare earth orthoferrites with
149 the perovskite structure (Rozenberg et al. 2005), and Al,Fe-bearing Bdg (Kupenko et al.
150 2015). This observation strongly supports the theoretical work of Hsu et al. (2011), who
151 suggested that Fe³⁺ is in the low-spin state in the B-site of Bdg.

152 Pioneering work by Fujino et al. (2012) reported that an equilibrium site
153 distribution of iron may be difficult to achieve, even with annealing by laser heating.
154 However, more recent work showed that the site distribution does not change
155 substantially before and after annealing (Glazyrin et al. 2014; Kupenko et al. 2015; Lin
156 et al. 2016). Also, *in situ* single crystal XRD measurements at high pressure
157 demonstrated that the site occupancy of iron did not change remarkably before, during
158 and after laser heating (Glazyrin et al. 2014). Lin et al. (2016) also showed that the site
159 distribution does not change after annealing, although X-ray emission spectra became
160 sharp after heating. Generally, annealing is important to achieving equilibrium; however
161 at the same time laser heating may cause undesired chemical heterogeneity in the
162 sample (e.g., Sinmyo and Hirose 2010), which can lead to a large uncertainty in

163 property measurements. For more detailed knowledge in the future, it is necessary to
164 develop a heating technique with smaller temperature gradient in the sample, such as
165 externally/internally resistive heating.

166 It is notable that independent studies agree that Fe^{3+} mainly occupies the A-site
167 of Al-free Bdg with moderate Fe^{3+} content (< 0.04 per formula unit) (Lin et al. 2012;
168 Sinmyo et al., 2014b; this study) for samples synthesized at ~ 25 GPa in a large volume
169 press. Our current results strongly support a small amount of Fe^{3+} distributed onto the
170 B-site of Bdg under pressure. This suggests that A-site of Bdg is more compressible
171 than the B-site, although it should be examined by a future study.

172 **Fe^{3+} in the low-spin state**

173 The new component observed above 36 GPa (Fe^{3+} #2) can be assigned to low
174 spin Fe^{3+} in the B site based on hyperfine parameters reported by Kuppenko et al. (2015).
175 Fe^{3+} may have been in the low-spin state when it was distributed onto the B-site of
176 bridgmanite as reported by Kuppenko et al. (2015). This inference is consistent with
177 theoretical predictions (e.g., Hsu et al. 2011). Our results show that Fe^{3+} in the low-spin
178 state never exceeds $\sim 18\%$ of total iron (Fig. 3). This value corresponds to 0.01 cations
179 per formula unit of Fe in the low-spin state in the chemical formula
180 $\text{Mg}_{0.971}\text{Fe}_{0.064}\text{Si}_{0.979}\text{O}_3$. Such a minor amount of low-spin iron is consistent with recent

181 experimental studies (Fujino et al. 2012; Lin et al. 2012; Kuppenko et al. 2015). It is
182 known that spin crossover can induce pronounced anomalies in the compressibility of
183 minerals (e.g., Fei et al. 2007; Bykova et al. 2016). However, such an anomaly in Bdg
184 compression has hardly been observed, even with single crystal XRD measurements
185 using a soft pressure medium (Boffa Ballaran et al. 2012; Mao et al. 2015). This
186 behavior is possibly due to the limited amount of ferric iron undergoing spin crossover
187 at high pressure. Caracas et al. (2014) showed that spin crossover can be detected by
188 fine analysis of XRD patterns collected for iron-rich $\text{Mg}_{0.5}\text{Fe}_{0.5}\text{SiO}_3$ Bdg. Although
189 Mössbauer spectroscopy is sensitive to the electronic state of iron, a minor amount of
190 Fe^{3+} may not be recognized during analysis of a Fe^{2+} -rich sample. Previous
191 disagreement among studies using XRD, Mössbauer, and X-ray emission spectroscopy
192 (XES) can be explained by the low amount of low-spin Fe^{3+} . Notably, the obtained
193 proportion of low-spin Fe^{3+} is consistent between XES measurements (Fujino et al.
194 2012) and Mössbauer spectroscopy results (Lin et al. 2012; Kuppenko et al. 2015; this
195 study) for Bdg synthesized in the multi-anvil press.

196 Figure 4 summarizes the CS and QS values obtained in this study. Fe^{2+} #1 is
197 high spin Fe^{2+} in the A-site (e.g., McCammon et al. 2013). Based on the hyperfine
198 parameters, Fe^{2+} #2 is intermediate spin Fe^{2+} in the A-site (McCammon et al. 2008), or

199 high-spin Fe^{2+} in the A-site with a high QS value (Hsu et al. 2010). CS and QS values of
200 Fe^{3+} #1 are generally similar to Fe^{3+} in the A-site of Al-free Bdg reported at ambient
201 conditions (“M2” and “M3” components in Hummer and Fei 2012). Hyperfine
202 parameters of Fe^{3+} #2 are similar to those for low-spin Fe^{3+} reported in a previous study
203 of Bdg (Kupenko et al. 2015). We note that all of the data by Hummer and Fei (2012)
204 were measured at ambient conditions. It is thus likely that all iron in their study is in the
205 high-spin state, while in our high-pressure study we suggest that the Fe^{3+} #2 component
206 is in the low-spin state. Moreover, the parameters are also quite similar to recently
207 observed values for low-spin Fe^{3+} in Fe_2O_3 at high pressure (Bykova et al. 2016).
208 Theoretical calculations predicted a high QS value (about 2-3 mm/s) for low-spin Fe^{3+}
209 in Bdg (e.g., Hsu et al. 2011). However, our results in relation to Fe^{3+} in Bdg do not
210 show a QS value higher than 1.5 mm/s regardless of the pressure, consistent with results
211 of Lin et al. (2012) and Kupenko et al. (2015). Although the source of this inconsistency
212 is unclear, the results of theoretical calculations may be affected by iron concentration
213 and structural/magnetic configuration (Umemoto et al. 2008).

214 **Implications**

215 Figure 5 summarizes the relation between the concentration of Fe^{3+} and the
216 pressure at which the fraction of low-spin iron is saturated in Bdg. We have fitted the

217 saturation pressure empirically as a function of Fe^{3+} content using an exponential curve.
218 We used data for Bdg synthesized using a large volume press and with known $\text{Fe}^{3+}/\Sigma\text{Fe}$
219 ratio (Jackson et al. 2005; Li et al. 2006; Fujino et al. 2012; Lin et al. 2012; Kuppenko et
220 al. 2015). As the Fe^{3+} content increases, the pressure of saturation increases (Fig. 5).
221 This behavior is similar to the previously reported relation between Fe^{2+} content and
222 spin-crossover pressure in $(\text{Mg,Fe}^{2+})\text{O}$ ferroperriclinite (Fei et al. 2007; Yoshino et al.
223 2011). Such a similarity may be due to the general relation between the spin state of
224 iron and the lattice volume altered by the impurity. The amount of Fe^{3+} in the deep
225 Earth depends on the bulk composition of the rock assemblage in the lower mantle,
226 since it is sensitive to Al content and bulk iron content (Nakajima et al. 2012). The
227 concentration of Fe^{3+} in Bdg is reported to be 0.01 - 0.04, 0.04 - 0.06 and 0.15 - 0.30
228 cations pfu in the harzburgite, pyrolite and MORB bulk compositions, respectively
229 (Sinmyo et al. 2011; Nakajima et al. 2012). Spin crossover of Fe^{3+} may take place at a
230 depth of about 900 km (35 GPa) in pyrolite and 1200 km (50 GPa) in MORB
231 composition according to this study (Fig. 5). It should be deeper in the hotter mantle,
232 since the spin crossover occurs at a higher and wider pressure range at high temperature
233 (Wentzcovitch et al. 2009; Wang et al. 2015). Boffa Ballaran et al. (2012) and Mao et al.
234 (2015) suggested that the elastic properties of Bdg are not substantially altered by iron

235 spin crossover. Moreover, as shown in this study, low-spin Fe^{3+} is a minor component
236 throughout the lower mantle; thus the effect on elasticity is not likely to be visible in
237 seismic observations. However, since electrical conductivity is sensitive to the spin state
238 (Ohta et al. 2010; Yoshino et al. 2011), spin crossover may be detected in geomagnetic
239 observations. Additionally, even though the amount of low-spin Fe^{3+} may be small, spin
240 crossover can have a non-negligible effect on the partition coefficient of iron between
241 Bdg and ferropericlase (e.g., Lin et al., 2013), which can strongly influence the structure
242 and dynamics of the lower mantle. The plot shown in Fig. 5 suggests that spin crossover
243 in Fe^{3+} may take place at a depth of about 900 and 1200 km in pyrolite an MORB
244 composition, respectively. Electrical conductivity and the iron partition coefficient
245 could change at such depths due to spin crossover, although available data are currently
246 limited to Fe^{3+} -bearing bulk compositions such as pyrolite (Ohta et al. 2010; Irifune et
247 al. 2010; Sinmyo and Hirose 2013; Prescher et al. 2014). Recent studies have reported
248 that the redox state of the Earth's mantle may be more heterogeneous than previously
249 thought (Frost and McCammon 2008; Stagno et al., 2013; Kaminsky et al., 2015).
250 Nakajima et al. (2012) showed that the Fe^{3+} content is 0.02 per formula unit higher in
251 Bdg synthesized in a rhenium capsule compared to a diamond capsule. This implies that
252 the spin crossover pressure may differ by ~ 10 GPa between the rhenium-rhenium oxide

253 buffer and the CCO buffer, since the crossover pressure significantly increases with
254 increasing Fe³⁺ content in Bdg (Fig. 5). Redox heterogeneities may be detectable by
255 precise magnetotelluric measurements at lower mantle depths.

256

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470

471 **Figure captions**

472 Figure 1. Selected Mössbauer spectra of Al-free Fe³⁺-rich Bdg at different pressures and
473 room temperature. Experimental data are indicated by solid circles while the fitted curve
474 is shown by the thick solid line. Components are shaded as indicated in part (c) and the
475 fitting residual is shown beneath each spectrum.

476

477 Figure 2. Pressure dependence of center shift (CS) and quadrupole splitting (QS) for (a)
478 Fe²⁺ and (b) Fe³⁺ in Al-free Bdg. Right and left pointing triangles indicate data taken
479 during compression and decompression, respectively. Our results are largely consistent
480 with previously reported values for Al-free Bdg (Potapkin et al. 2013).

481

482 Figure 3. Relative abundance of components fitted to Mössbauer spectra of Al-free Bdg:
483 (a) Fe²⁺; (b) Fe³⁺. The dashed line shows the initial abundance of Fe²⁺ and Fe³⁺.

484

485 Figure 4. Variation of CS and QS values for Bdg. Circles, this study; squares, Al-free
486 Bdg at ambient conditions (Hummer and Fei 2012); diamonds, Al-bearing Bdg under 12

487 - 77 GPa (Kupenko et al. 2015). The diameters of the circles are proportional to the
488 pressure at which the values were obtained (i.e., larger diameters correspond to higher
489 pressures).

490

491 Figure 5. Relation between Fe^{3+} concentration (in cations per formula unit) and the
492 saturation pressure of low-spin Fe^{3+} in Bdg (see text for details). The grey line is an
493 empirical fit to the data for Bdg samples synthesized using a large volume press where
494 $\text{Fe}^{3+}/\Sigma\text{Fe}$ ratios were experimentally determined. Red symbols, Al-free Bdg; blue
495 symbols, Al-bearing Bdg. Filled symbols indicate the data used for fitting (Color
496 online). “Harzburgite”, “Pyrolite” and “MORB” denote possible concentrations of Fe^{3+}
497 in Bdg in harzburgitic, pyrolitic and basaltic rock, respectively (see text for details).

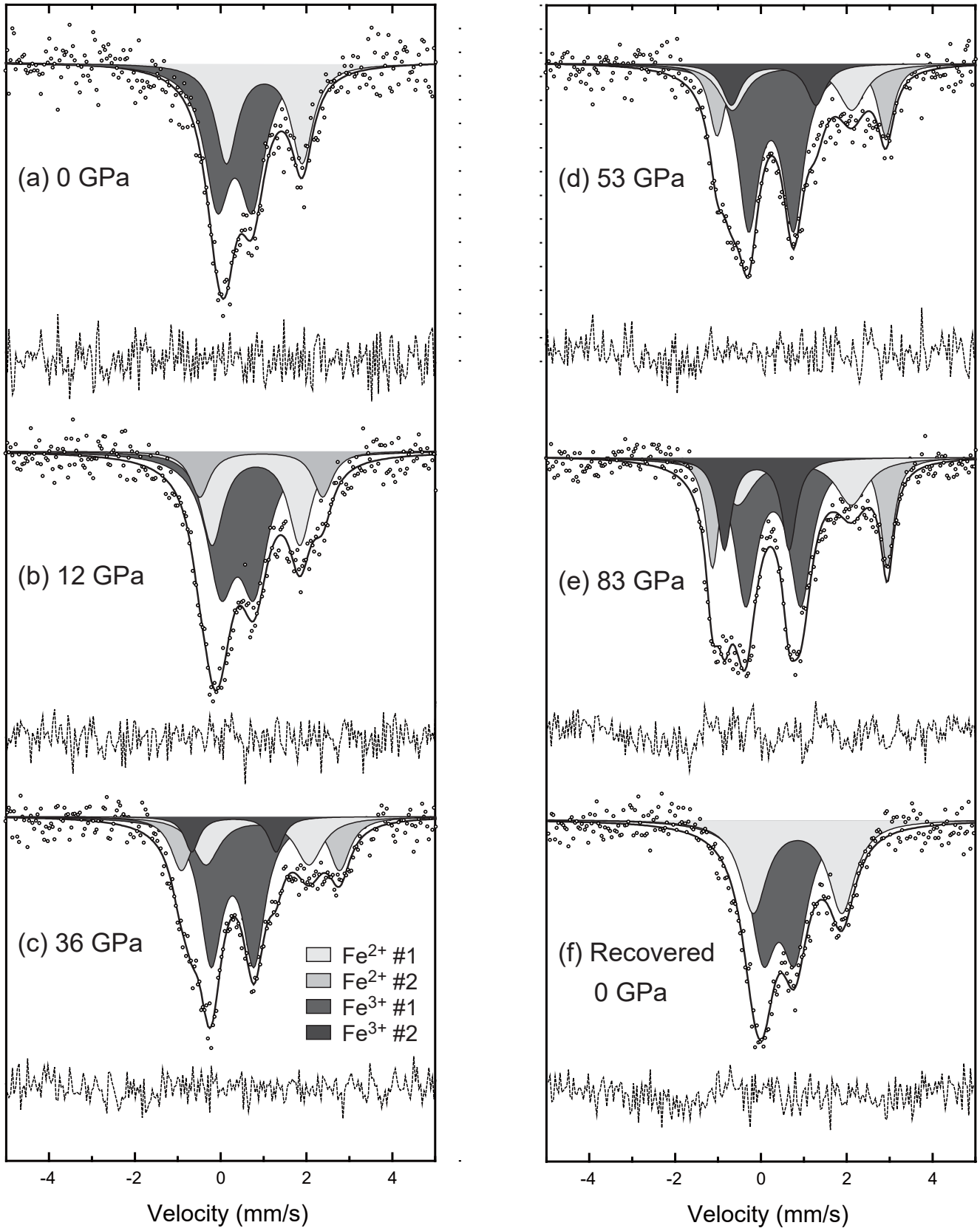


Figure 1

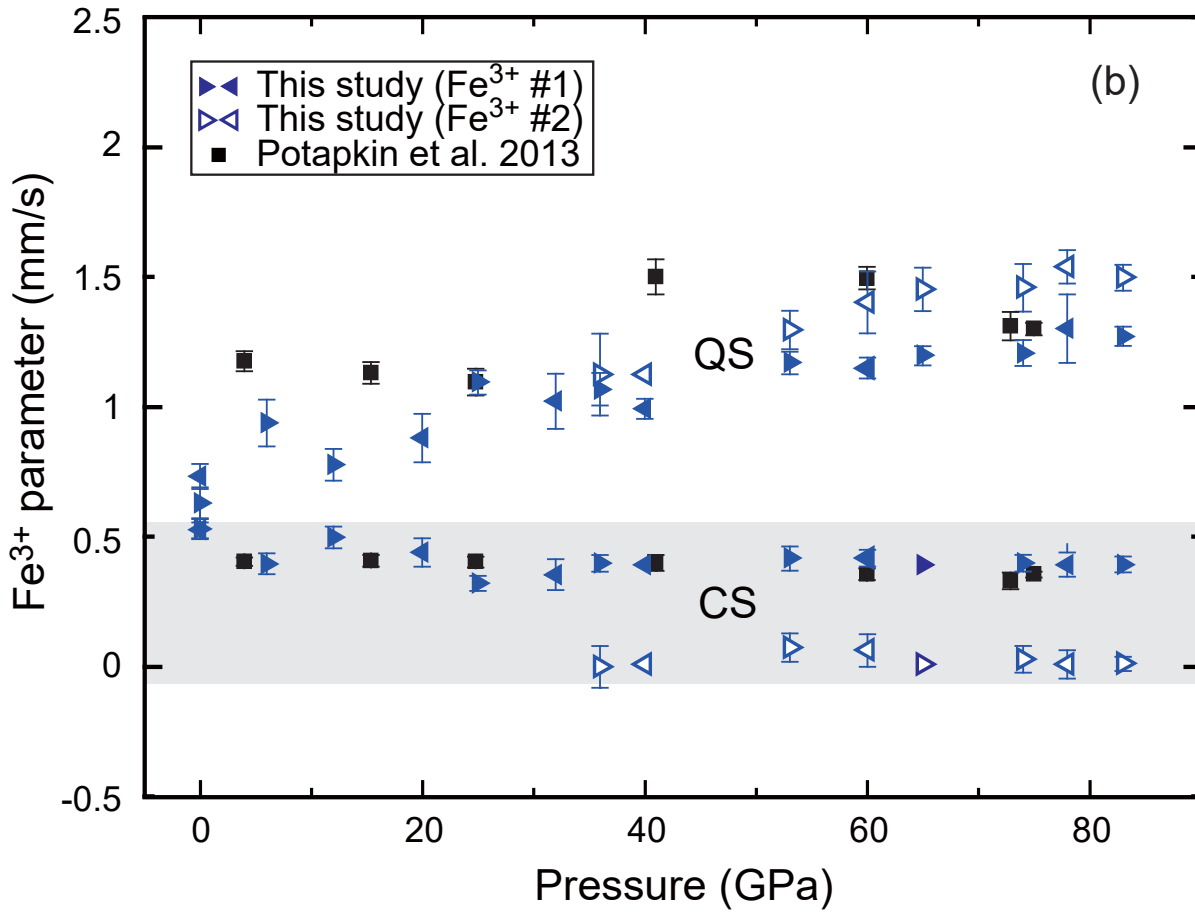
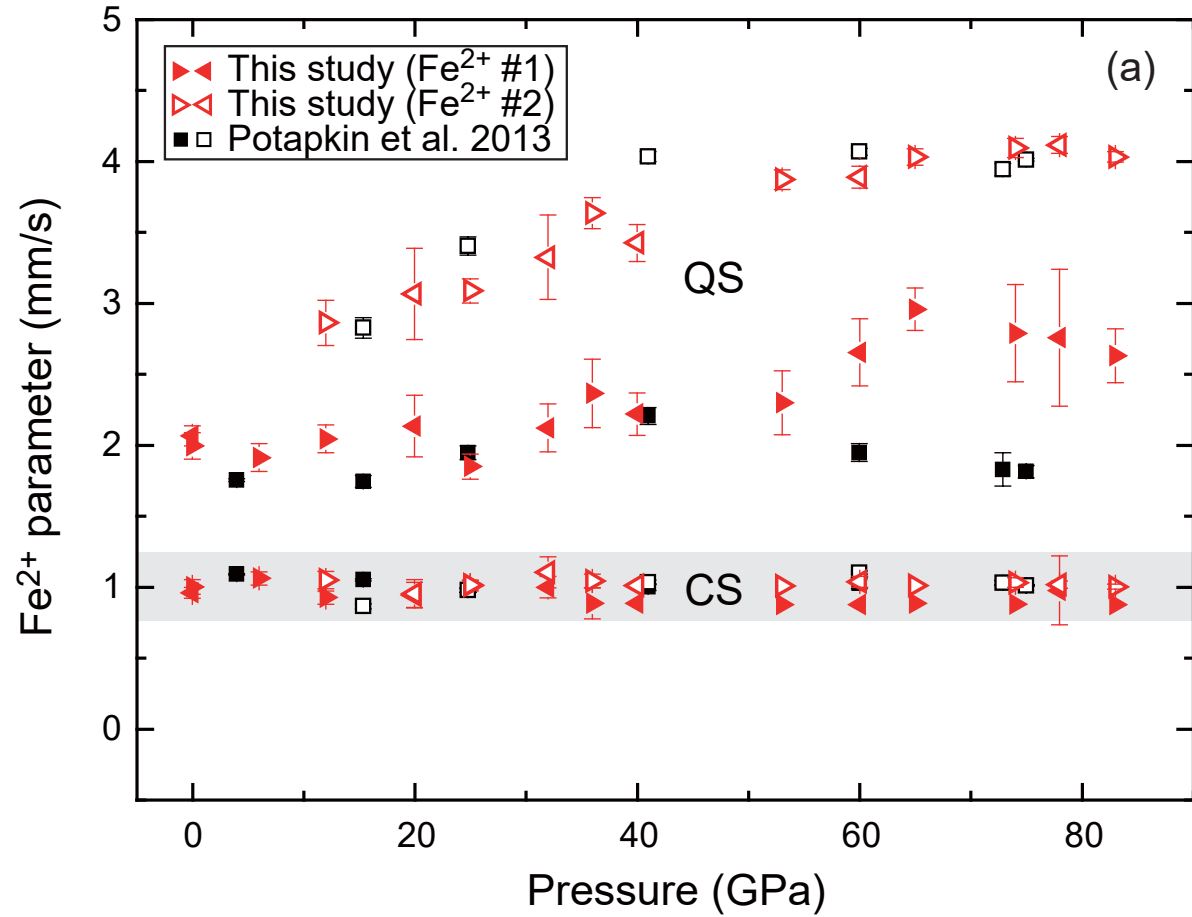


Figure 2

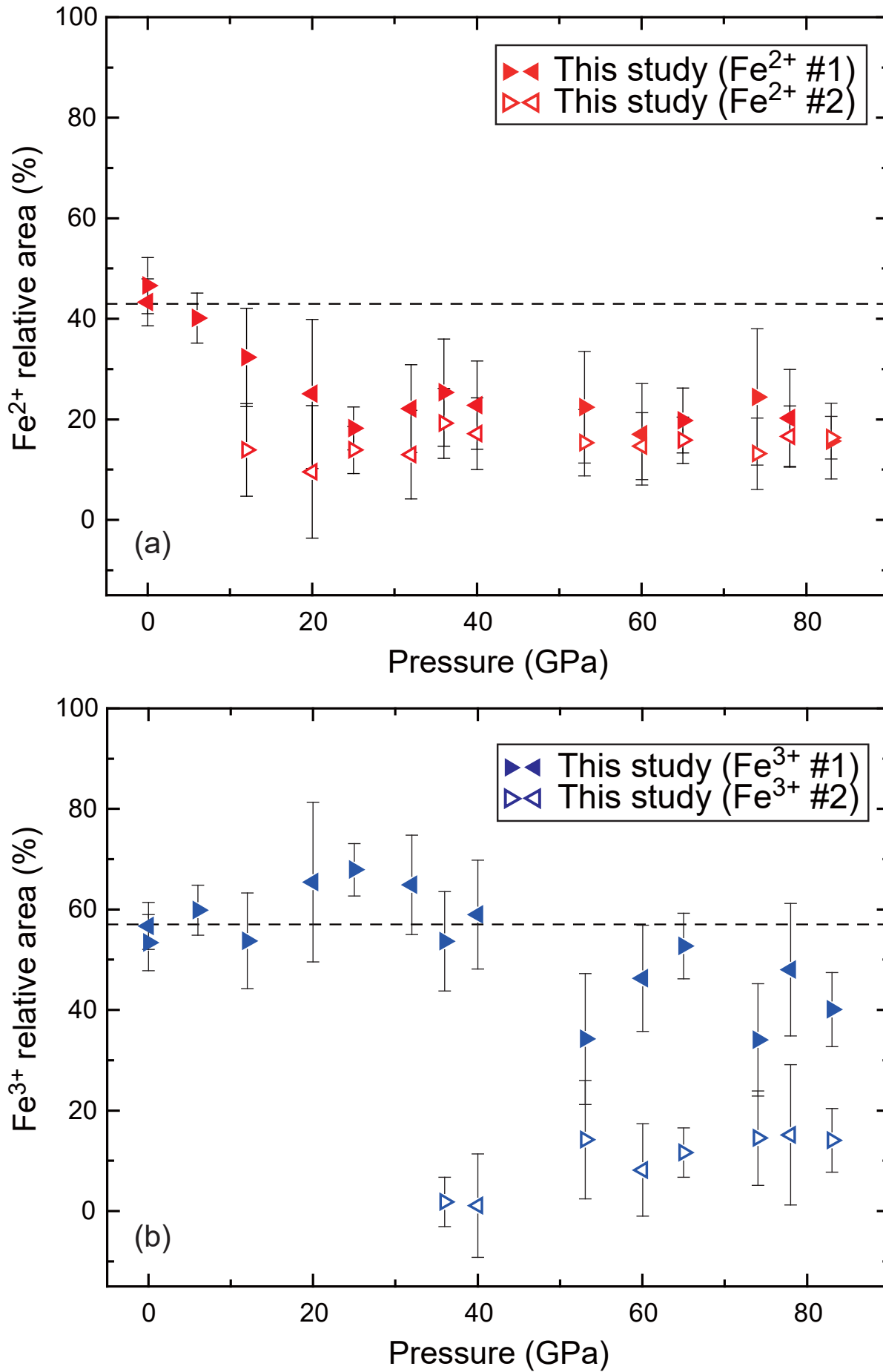


Figure 3

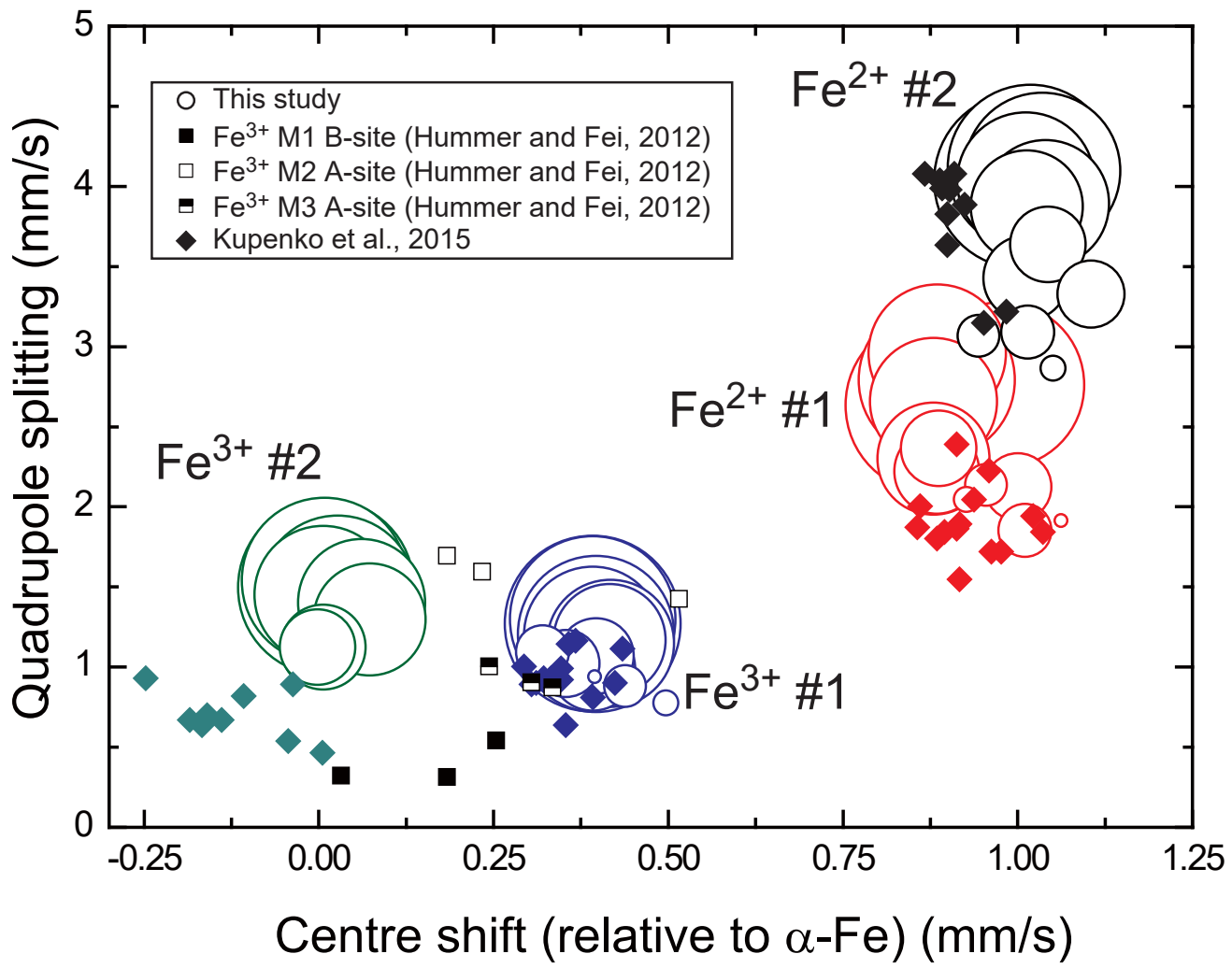


Figure 4

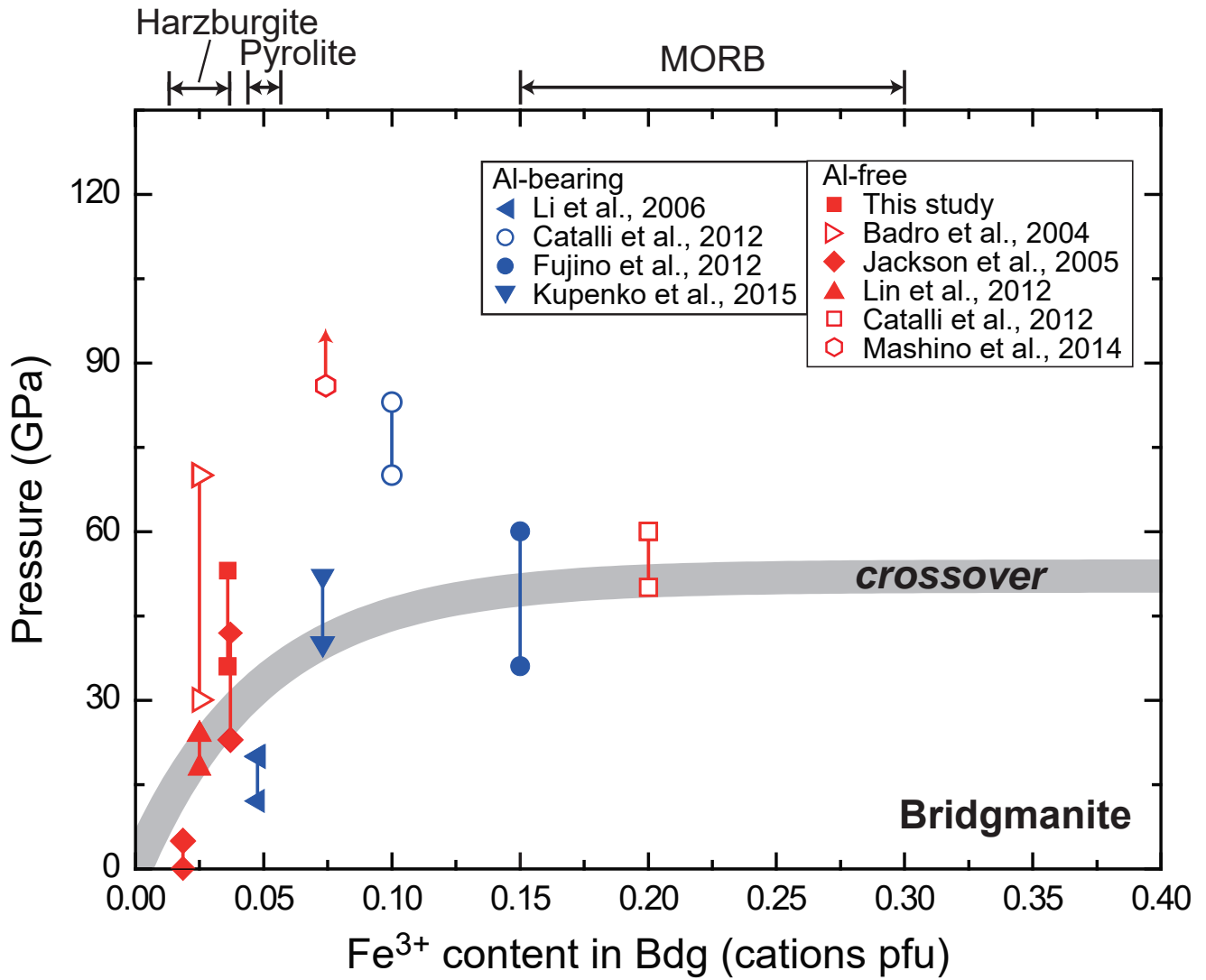


Figure 5

Table 1. Hyperfine parameters of Fe³⁺-rich bridgmanite determined from room temperature Mössbauer spectra.

Pressure (GPa)	Component	CS (mm/s)	FWHM (mm/s)	Area (%)	QS (mm/s)	Fe ³⁺ /total Fe (%)
<i>Compression</i>						
0	Fe ²⁺ #1	1.00(5)	0.75(15)	47(6)	2.00(9)	53
	Fe ³⁺ #1	0.53(4)	0.66(13)	53(6)	0.63(6)	
6	Fe ²⁺ #1	1.06(5)	0.77(13)	40(5)	1.91(10)	60
	Fe ³⁺ #1	0.40(4)	0.92(13)	60(5)	0.94(9)	
12	Fe ²⁺ #1	0.93(5)	0.64(21)	32(10)	2.05(10)	54
	Fe ²⁺ #2	1.05(6)	0.55(26)	14(9)	2.86(16)	
	Fe ³⁺ #1	0.50(4)	0.79(13)	54(10)	0.78(6)	
25	Fe ²⁺ #1	1.01(4)	0.46(15)	14(5)	1.85(9)	68
	Fe ²⁺ #2	1.01(4)	0.54(13)	18(4)	3.09(9)	
	Fe ³⁺ #1	0.32(3)	0.95(11)	68(5)	1.09(5)	
36	Fe ²⁺ #1	0.89(11)	0.92(42)	25(11)	2.36(24)	55
	Fe ²⁺ #2	1.04(5)	0.61(19)	19(7)	3.64(11)	
	Fe ³⁺ #1	0.40(3)	0.67(11)	54(10)	1.07(6)	
	Fe ³⁺ #2	0.00(8)	0.19(41)	2(5)	1.12(16)	
53	Fe ²⁺ #1	0.88	0.93(40)	26(11)	2.30(22)	54
	Fe ²⁺ #2	1.01	0.47(12)	19(6)	3.87(7)	
	Fe ³⁺ #1	0.42(5)	0.56(16)	37(12)	1.17(4)	
	Fe ³⁺ #2	0.07(6)	0.45(18)	18(11)	1.29(7)	
65	Fe ²⁺ #1	0.88	0.67(27)	20(6)	2.96(15)	64
	Fe ²⁺ #2	1.01	0.37(10)	16(5)	4.03(6)	
	Fe ³⁺ #1	0.39	0.63(7)	53(7)	1.20(4)	
	Fe ³⁺ #2	0.01	0.37(15)	12(5)	1.45(8)	
74	Fe ²⁺ #1	0.88	1.06(60)	28(13)	2.79(34)	55
	Fe ²⁺ #2	1.04(3)	0.37(13)	17(7)	4.10(7)	
	Fe ³⁺ #1	0.40(3)	0.50(12)	37(11)	1.21(5)	
	Fe ³⁺ #2	0.03(5)	0.43(17)	18(9)	1.46(9)	
83	Fe ²⁺ #1	0.88	0.85(38)	19(7)	2.63(19)	61
	Fe ²⁺ #2	1.01(2)	0.37(6)	20(4)	4.03(4)	
	Fe ³⁺ #1	0.39(3)	0.62(9)	43(7)	1.27(4)	
	Fe ³⁺ #2	0.01(3)	0.40(10)	18(6)	1.50(5)	
<i>Decompression</i>						
78	Fe ²⁺ #1	0.98(24)	0.77(41)	20(10)	2.76(48)	63
	Fe ²⁺ #2	1.02(3)	0.38(10)	17(6)	4.12(6)	
	Fe ³⁺ #1	0.39(5)	0.70(12)	48(13)	1.30(13)	
	Fe ³⁺ #2	0.01(6)	0.37(21)	15(14)	1.54(6)	
60	Fe ²⁺ #1	0.88	0.82(38)	20(12)	2.56(78)	62
	Fe ²⁺ #2	1.04(4)	0.37(6)	18(9)	3.94(8)	
	Fe ³⁺ #1	0.42(3)	0.62(9)	44(21)	1.19(15)	
	Fe ³⁺ #2	0.06(6)	0.40(10)	18(24)	1.43(9)	
40	Fe ²⁺ #1	0.88	0.68(27)	23(9)	2.22(15)	60
	Fe ²⁺ #2	1.01	0.56(21)	17(7)	3.43(13)	
	Fe ³⁺ #1	0.39	0.65(10)	59(11)	0.99(4)	
	Fe ³⁺ #2	0.01	0.64(456)	1(10)	1.12	
32	Fe ²⁺ #1	1.00(8)	0.53(23)	22(9)	2.12(17)	65
	Fe ²⁺ #2	1.10(11)	0.65(45)	13(9)	3.33(30)	
	Fe ³⁺ #1	0.35(6)	0.85(16)	65(10)	1.02(11)	
20	Fe ²⁺ #1	0.95(10)	0.61(37)	25(15)	2.14(22)	65
	Fe ²⁺ #2	0.94(9)	0.49(48)	10(13)	3.07(32)	
	Fe ³⁺ #1	0.44(6)	0.74(12)	65(16)	0.88(9)	
0	Fe ²⁺ #1	0.96(4)	0.75(12)	43(5)	2.07(7)	57
	Fe ³⁺ #1	0.53(3)	0.74(11)	57(5)	0.73(5)	

Values in italics were held fixed during fitting