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Flux growth of single crystals of MgGeO₃ polymorphs (orthopyroxene, clinopyroxene, and ilmenite) and their phase relations and crystal structures

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Abstract

Three kinds of MgGeO₃ pyroxene single crystals of 1–10 mm size were grown by the flux method: an orthopyroxene, a clinopyroxene and a "clino-from proto" phase. The lattice parameters and the space groups are (1) orthopyroxene: a = 18.8099(12), b = 8.9484(8), c = 5.3451(4)Å, Pbca, (2) clinopyroxene and (3) "clino-from proto" phase: a = 9.6010(8), b = 8.9323(6), c = 5.1592(5)Å, $\beta = 101.034(9)^\circ$, C2/c. The phase relations of MgGeO₃ were studied in the temperature range $655-1030^\circ$ C and in the pressure range 0-55 kbar. The boundary between orthopyroxene and clinopyroxene was determined as $T = 810^\circ$ C + 3.8 P(kbar), where the high-pressure and low-temperature phase is monoclinic. Single crystals of ilmenite-type MgGeO₃ (up to 100 μ m in size) were also grown in association with the pyroxene \rightarrow ilmenite transformation at 55 kbar, using the flux-grown mm-size orthopyroxene or clinopyroxene MgGeO₃ single crystals as starting materials.

Introduction

Pyroxene is one of the important constituent minerals of the earth's crust and upper mantle. Ito (1975) has established a crystal-growth technique particularly suitable for growing enstatite (MgSiO₃). Orthoenstatite single crystals provided by Ito have been successfully used for detailed structure analyses (Sasaki et al., 1982) and for Brillouin scattering measurements (Weidner et al., 1978). Germanates are generally used as geochemical models of silicates. They often display the same phases as silicates but at lower pressures more easily achieved in the laboratory. However, the phase relationships of germanate pyroxenes such as MgGeO₃ have not been studied in detail nor has the growth of single crystals of germanate pyroxenes been reported. Large single crystals of germanate pyroxene are useful as starting materials to obtain single crystals of high-pressure phases as well as for physical property measurements.

Previous studies have demonstrated that MgGeO₃ exhibits phases isomorphic to those of MgSiO₃. Roth (1955) reported the synthesis of an orthorhombic phase isomorphic to orthoenstatite. Robbins and Levin (1959) reported the synthesis of a monoclinic polymorph of MgGeO₃ in addition to the orthorhombic form. In contrast to the MgSiO₃ system, they concluded that the monoclinic phase is the high-temperature phase (above 1555°C). Furthermore, they reported that the germanate melts congruently at 1700°C which also differs from the behavior of the silicate pyroxene.

High-pressure transformation of MgGeO₃ pyroxene

 (1974) who determined phase boundaries among orthorhombic, monoclinic and hexagonal (ilmenite structure) phases.
This paper reports growth of large single crystals (1–10 mm size) of MgGeO₃ pyroxene in air by the flux method.
Based on morphological observation and X-ray diffrac-

Based on morphological observation and X-ray diffraction measurements on the synthesized single crystals, phase relations at atmospheric pressure between orthorhombic (two orthorhombic phases) and monoclinic polymorphs in MgGeO₃ pyroxene are discussed in detail. Preliminary results of high-pressure and high-temperature experiments using single-crystal MgGeO₃ pyroxene prepared in the present study as starting materials are also reported.

into the ilmenite structure was first reported by Ringwood and Seabrook (1963), and later by Kirfel and Neuhaus

Crystal growth and phase relations

We adopted a flux method to grow single crystals of orthorhombic phase of MgGeO₃. Single crystals of MgSiO₃ orthoenstatite have already been grown by Ito (1975) using such a method. We followed his procedure, including the chemical composition of the flux ($Li_2O = 51.7$ mol%, MoO₃ = 42.3 mol%, V₂O₅ = 6.0 mol%) and the temperature range for growth.

Reagent grade MoO₃, Li₂CO₃, V₂O₅, MgO, and GeO₂ of 99.999% were used as starting materials. MoO₃, Li₂CO₃, and V₂O₅ were for flux and MgO and GeO₂ were for nutrient. Mixtures of these were heated in air at 955°C in platinum crucibles for several days, slowly cooled to

Charge composition	flux	Li2CO3	7.35 g(Li ₂ 0:2.97g)
		MoOz	11.70
		V205	2.08
	nutrient	MgO	0.232
		GeO2	0.602
Mg/Ge atomic ratio		2	1.0
Nutrient/Flux			5 wt%
Soaking temperature			955°C
Soaking time			7 days
Cooling rate			1.6°C/hr
Crucible size			10 ml
Cut-off temperature			650°C
Crystal size(length)			1 - 10 mm
Products		Orthoenstatite-type	
		Clinoenstatite-type "Clino-from proto"	

Table 1. Crystal growth data for MgGeO₃ pyroxene

655°C, and quenched to room temperature. After the flux was dissolved in water, the residual materials were collected as products. In Table 1, an example of the growth conditions is shown. According to Robbins and Levin (1959) this temperature range should yield only one orthorhombic phase (orthoenstatite-type). However, we have obtained three kinds of MgGeO₃ crystals in the temperature range 955–655°C, *i.e.*, single crystals of orthorhombic MgGeO₃, single crystals of monoclinic MgGeO₃ with or without twinning, and "clino-from proto" MgGeO₃ with very fine twinning. Morphological features of these three kinds of crystals are described below.

(1) Orthorhombic MgGeO₃ (orthoenstatite-type, Pbca)

A number of single crystals of orthorhombic MgGeO₃ were grown. These are transparent and colorless with either an octagonal needle-like shape elongated along the *c*-axis (0.5 mm \times 0.5 mm \times 10 mm) or a platelet shape (3 mm \times 5 mm \times 0.5 mm). The crystal habit for this orthorhombic MgGeO₃ is completely the same as that of flux-grown orthoenstatite, *i.e.*, the axis of elongation is *c*, {100}, {010}, and {210} faces form an octagonal column, and {111} and {211} build up a peculiar feature at the top of the needle (Takei *et al.*, 1982). The (010) faces develop to make a flat surface in the platelet crystals.

(2) Type A monoclinic MgGeO₃ (clinoenstatitetype, C2/c)

According to Robbins and Levin (1959), monoclinic MgGeO₃ is reported to be stable above 1555°C at atmospheric pressure. In this study, however, a number of single crystals (with or without twinning) of the monoclinic phase of MgGeO₃ were obtained simultaneously with the orthorhombic phase in a single run. This indicates that single crystals of monoclinic MgGeO₃ were grown in the temperature range 955 to 655°C. These crystals are transparent and colorless and plate-like or needle-like with dimensions about 3 mm \times 5 mm \times 0.5 mm or 0.5 mm \times 0.5 mm \times 10 mm. When the solution was cooled very

slowly around 800°C, twinning-free single crystals of plate-like monoclinic MgGeO3 were grown (Fig. 1). As the shapes of the needle-like and plate-like monoclinic MgGeO₃ are very similar to those of orthorhombic MgGeO₃, *i.e.*, octagonal column composed of {110}, $\{100\}$, and $\{010\}$ and elongated in the *c*-axis and (010)plate, it is difficult to distinguish the monoclinic MgGeO₃ crystals from the orthorhombic MgGeO₃ using only the crystal habit. When we observe these crystals under a petrographic microscope with crossed nicols, the orthorhombic phase shows a straight extinction, while the monoclinic phase extinguishes obliquely (about 25-28°). Twins in the monoclinic phase are easy to recognize. The (100) twinning (occasionally only a single twin in a grain) seems not to be essential for crystal growth, and furthermore, this twinning is completely different from those for the type B monoclinic phase discussed below. This latter group is interpreted as resulting from phase transitions upon quenching. Therefore, these type A monoclinic MgGeO₃ crystals can be regarded as having grown as a primary phase in this temperature range.

(3) Type B monoclinic MgGeO₃ (clino-from proto-type, C2/c)

In the temperature range 955 to 655°C, a number of opaque, white and thick octagonal columnar crystals were also produced. Precession photographs and microscopic observation revealed that these are mostly monoclinic MgGeO₃ single crystals with very fine (100) twinning (plus a small amount of the orthorhombic phase). Because of their similarity to "proto-inverted-clinoenstatite" (Ito, 1975), these are considered to be an inversion product from unquenchable proto-type MgGeO₃ single crystals, i.e., "clino-from proto"-MgGeO3. As will be discussed later, this type B monoclinic MgGeO₃ (C2/c) is crystallographically identical to the type A monoclinic MgGeO₃ (C2/c). However, the appearance of these two geneses of monoclinic MgGeO₃ crystals is completely different. That is, the former is opaque due to the extensively developed twinning and cracks, and the latter is transparent with at most a small amount of twinning. A small amount of orthorhombic phase was also found to coexist within or on the surface of these type B monoclinic crystals. This orthorhombic phase is regarded as the inversion product of the protophase and/or newly grown crystals which formed on the surface during the process of cooling.

In order to elucidate further these complicated relations of the synthesis of MgGeO₃ crystals, several different kinds of heat treatments were carried out. Slowcooling of the solution from 908°C did not yield any type B monoclinic MgGeO₃ crystals, while cooling from 935°C produced a number of type B monoclinic MgGeO₃ crystals. Therefore, the stability field of proto-type MgGeO₃ at atmospheric pressure is indicated to be above about 920°C and this temperature is very close to that for enstatite (950°C). On slow-cooling from 908°C to 810°C,



Fig. 1.(i),(ii) An (010) plate of type A monoclinic MgGeO₃ single crystal (twinning-free). (i) uncrossed nicols, (ii) crossed nicols. A small single crystal of orthoenstatite-type MgGeO₃ is attached at the lower part which shows straight extinction under crossed nicols (ii). (iii) thin section of type B monoclinic MgGeO₃ ("clino-from proto" MgGeO₃) under crossed nicols. (iv) An (010) plate of "clino-from ortho" MgGeO₃ (31 kbar, 850°C) under crossed nicols.

no monoclinic MgGeO₃ phase grew at all; rather only orthorhombic MgGeO₃ was obtained. Cooling to 795°C produced both type A monoclinic MgGeO₃ and orthorhombic MgGeO₃. These results imply that single crystals of orthorhombic MgGeO₃ (orthoenstatite-type) are stable only between 920°C and about 810°C, and monoclinic MgGeO₃ grows below 810°C.

The present phase relations are completely discordant with those given by Robbins and Levin (1959). They concluded monoclinic MgGeO₃ is the high-temperature phase (above 1555°C), while we conclude that it is the low-temperature phase (below 810°C). The present phase relations at atmospheric pressure are very similar to those for MgSiO₃ enstatite (for example, Boyd and England (1965)), except that the boundary between the monoclinic and orthorhombic phase in MgGeO₂ is higher by about $200-250^{\circ}$ C than that for MgSiO₃ (566°C by Grover (1972); 630°C by Boyd and England (1965)). In this context, note that it should be possible to grow single crystals of clinoenstatite of MgSiO₃ at atmospheric pressure if we can find a solvent which dissolves considerable amounts of MgO and SiO₂ below 566°C. In Figure 1, both type A and type B monoclinic MgGeO₃ crystals are shown. If our phase relations are correct, the structural identities of all those MgGeO₃ samples studied by Roth (1955), Robbins and Levin (1959), and Kirfel and Neuhaus (1974) should be suspected. Further discussion of this point is given in the following sections.

High-pressure and high-temperature experiments

High-pressure and high-temperature experiments were carried out in the pressure range 31-55 kbar and in the temperature range 850-1030°C, with the aim of elucidating the phase relations between orthoenstatite- and clinoenstatite-type MgGeO₃ at high pressures. Both a tetrahedral anvil apparatus and a cubic anvil apparatus were used. In experiments using the tetrahedral anvil apparatus, tungstencarbide anvils with 15 mm edges and baked pyrophyllite tetrahedra with 20 mm edges were used. For the cubic anvil apparatus, a combination of 16 mm-edge anvils with 21 mm-edge cubes was adopted. Single crystals of orthoenstatite-type and clinoenstatite-type MgGeO₃ were used as the starting materials. A tubular graphite heater was used for heating these crystal samples. NaCl was adopted as the pressure-transmitting medium surrounding these crystals inside the graphite heater. Since the temperaturepressure conditions of the present study are close to the melting curve of NaCl, the NaCl insert was softened markedly although not melted. This produced a uniform stress condition for the crystals.

Pressure values were calibrated against press load at room temperature based on the well-established pressurefixed points of low-Bi (25.5 kbar), low-Ba (55 kbar) and high-Bi (77 kbar). A conventional quenching technique was used for investigating the phase relations. After the pressure was applied to the sample, temperature was brought to the desired value and held there for 30 to 90 OZIMA AND AKIMOTO: ORTHOENSTATITE-TYPE AND CLINOENSTATITE-TYPE MgGeO3

minutes. The sample was then quenched by turning off the heating power under the working pressure. After releasing the pressure, the single crystals were collected by dissolving the NaCl insert with water. Before and after the high-pressure and high-temperature treatment, the phases of the samples were examined with a petrographic microscope and a precession camera.

At 40 kbar and 900°C, no change was observed in the type A monoclinic single crystals of MgGeO₃. On the other hand, orthorhombic single crystals of MgGeO₃ changed to a monoclinic structure after the same treatment. These run-products, "clino-from ortho"-MgGeO3 crystals were confirmed to be crystallographically identical to the type A monoclinic MgGeO₃, except for the presence of a large number of twins developed through the phase transformation. This twin texture was the same as observed in the "clino-from proto"-MgGeO₃, type B monoclinic MgGeO₃ (Fig. 1). At 39 and 44 kbar and 1020°C, the type A monoclinic phase changed to an orthorhombic phase, while at 31 kbar, 850°C; 38 kbar, 900°C; and 44 kbar, 940°C; the orthorhombic phase changed to a monoclinic phase with very fine twinning. At 51 kbar and 1030°C, the orthorhombic phase did not change. Summarizing all our results including the phase relations at atmospheric pressure, the ortho-clino phase boundary is shown in Figure 2. The boundary curve is approximately determined as $T = 810^{\circ}\text{C} + 3.8 P$ (kbar). The high-temperature and low-pressure phase is the orthorhombic orthoenstatite-type and the low-temperature and high-pressure phase is the monoclinic clinoenstatitetype.

The present result is entirely different from the determination of Kirfel and Neuhaus (1974). They reported a narrow stability field for the orthoenstatite-type below 10 kbar and gave a steep negative slope for the phase boundary. The unit cell volumes of the orthoenstatitetype and the clinoenstatite-type are 899.69(12)Å³ (Z = 16) and 434.27(6)Å³ (Z = 8), respectively. The smaller cell volume of the clinoenstatite-type supports our conclusion that the clinoenstatite-type is the high-pressure phase. It is noted that Kirfel and Neuhaus (1974) did not consider the possibility of the appearance of a proto-phase and consequently did not distinguish "clino-from proto" MgGeO₃ (type B) from the primary monoclinic MgGeO₃ (type A) and from the "clino-from ortho" MgGeO₃. Furthermore, their identification of phases based on their polycrystalline powder sample could have resulted in an erroneous conclusion. In the next section we critically discuss their powder data on the basis of our own measurements on powdered samples. The present phase relations of MgGeO₃ at high-pressures and high-temperatures are similar to those of MgSiO₃ reported by Boyd and England (1965); whose ortho-clino boundary is T = $630^{\circ}\text{C} + 2.6 P$ (kbar), although their experiments were claimed to be carried out under the existence of some shearing stress (Grover, 1972).

At 55 kbar and 900°C, small single crystals of hexagonal



Fig. 2. Phase diagram of MgGeO₃. P: protoenstatite-type region; O: orthoenstatite-type region; C: clinoenstatite-type region; I: ilmenite-type region; KN: pyroxene-ilmenite phase boundary by Kirfel and Neuhaus (1974).

ilmenite-type MgGeO₃ (up to 100 μ m in size) partly replaced single crystals of both the ortho- and clino-phase (Fukizawa et al., 1982). This condition is located within the ilmenite-type region given by Kirfel and Neuhaus (1974) and not far from the phase boundary between pyroxene and ilmenite. The crystal orientation of this hexagonal phase with respect to the parent phase was examined with a precession camera. It was found that the (100) plane of the ortho- or clino-phase coincides with the (001) plane of the ilmenite phase, *i.e.*, the oxygen-closepacked plane is conserved through the transition. Although we did not determine the boundary between the pyroxene and ilmenite phases, Kirfel and Neuhaus's straight line should have a kink at the triple point (ortho, clino, ilmenite) and the estimated boundary lines are shown in Figure 2.

Conservation of the oxygen-close-packed plane through the phase transformation has recently been reported for the olivine-spinel transformation (Lacam *et al.*, 1980; Hamaya and Akimoto, 1982). Hamaya and Akimoto (1982) also demonstrated the successful growth of high-quality single crystals of Ni₂SiO₄ spinel (up to 300 μ m in size) using mm-size single crystals of olivine as starting materials. This suggests a possibility of growing sub-mm size high-quality single crystals of MgGeO₃ ilmenite using the mm-size single crystals of MgGeO₃ pyroxene synthesized in the present study as starting materials.

Crystal structure

Crystal structure of single crystals of both orthorhombic and monoclinic MgGeO₃ were studied with a preces-

h	k	1'	d _{obs} .	d _{calc} .	Intensity
2	1	0	6.4840	6.4829	m
4	0	0	4.7024	4,7025	m
0	2	0	4.4756	4.4742	Th
4	1	0	4.1630	4 1627	1767
1	2	1	3.3750	3.3752	w
4	1	1	3.2846	3.2842	vw
4	2		3.2412	3.2414	vs
3	2	1	3.0089	3.0097	W
6	1	4	2.9601	2.9587	S
5	T	1	2.9111	2.9093	W
4	2	1	2.7713	2,7716	W
1	3	1	2.5747	2.5801	w
2	0	2	2.5747	2.5708	w
5	2	1	2.5348	2.5350	m
3	1	2	2.3704	2.3707	vw
6	2	1	2,3161	2.3143	1767
0	4	0	2,2371	2,2371	w
5	0	2	2,1781	2.1787	17.4
8	1	ĩ	2,0928	2,0925	1762
ĩ	4	ĩ	2,0509	2.0513	V W 1.7
-		10 -	210309	2.0313	
4	4	0	2.0204	2.0202	m
2	4	1	2.0170	2.0157	m
3	4	1	1.9604	1.9602	w
10	1	0	1.8405	1.8408	π
5	4	1	1.8091	1.8093	w
8	3	1	1.7451	1.7453	w
7	2	2	1.7451	1.7449	w
8	1	2	1.7314	1,7319	w
0	2	3	1.6551	1.6553	vw
9	0	2	1.6472	1.6463	VW
9	э	1	1 6297	1 6201	
R	4	0	1 6210	1.6301	w
6	5	0	1 5542	1.5542	W
10	2	ĩ	1 5244	1 5240	Itt
1	3	3	1.5244	1.5245	m
6	5	1	1.4923	1.4924	m
9	4	1	1.4685	1.4684	VW
3	5	2	1.4461	1.4469	VW
7	4	2	1.4461	1.4459	VW
11	0	2	1.4395	1.4404	vw
11	з	1	1.4297	1.4295	w
11	1	2	1.4221	1.4221	VW
4	5	2	1.4179	1.4178	vw
12	3	1	1.3431	1.3430	W
12	1	2	1.3371	1.3369	W
1	0	А	1 3331	1 2320	
14	1	-	1 3289	1.3329	vw
10	5	0	1 2969	1.3207	W
	1.00		1.6707	1./900	m

Table 2. X-ray powder diffraction data of orthoenstatite-type MaGeO

x = 18.8099(12) Å, b = 8.9484(8) Å, c = 5.3451(4) Å, V = 899.69(12) Å, Z = 16, $\rho_{calc.} = 4.278$ g/cm³

sion camera and a four-circle diffractometer. Powdered samples were examined by a diffractometer.

The orthorhombic phase was confirmed to have the orthoenstatite structure (space group Pbca) with cell

Table 3. Crystal data of orthoenstatite-type MgGeO₃

	Roth(1975)	Kirfel & Neuhaus(1974)	This study	
a /Å b /Å c /Å V /Å	18.661 8.954 5.346 893.27	18.859(6) 8.958(2) 5.334(1) 901.17(35)	18.8099(12) 8.9484(8) 5.3451(4) 899.69(12)	
Group	Pbca	Pbca	Pbca	

parameters a = 18.8099(12), b = 8.9484(8), c =5.3451(4)Å. These cell parameters were determined for a single crystal using only strong reflections at high 2θ angle selectively peak-refined with a four-circle diffractometer. In Table 2 are shown observed d-spacings for a powdered sample together with the calculated ones based on the refined unit cell parameters. Our data are generally close to that by Robbins and Levin (1959). In Table 3, the crystal data are compared to those of other authors.

Table 4. X-ray powder diffraction data of clinoenstatite-type MgGeO₃

1 1 0 6.4839 6.4828 2 0 0 4.7148 4.7118 0 2 0 4.4644 4.4661 -1 1 4.2720 4.2729 0 2 1 3.3488 3.3495 2 2 0 3.2412 3.2414	m s w vw
2 0 0 4.7148 4.7118 0 2 0 4.4644 4.4661 -1 1 1 4.2720 4.2729 0 2 1 3.3488 3.3495 2 2 0 3.2412 3.2414	s w vw
0 2 0 4.4644 4.4661 -1 1 1 4.2720 4.2729 0 2 1 3.3488 3.3495 2 2 0 3.2412 3.2414	w vw
-1 1 1 4.2720 4.2729 0 2 1 3.3488 3.3495 2 2 0 3.2412 3.2414	vw
1 1 1.2725 0 2 1 3.3488 3.3495 2 2 3.2412 3.2414	V 41
2 2 0 3.2412 3.2414	
2 2 0 3.2412 3.2414	IC.
	vs
3 1 0 2.9630 2.9633	VS
-2 2 1 2.9093 2.9095	w
-3 1 1 2.7864 2.7862	W
2 2 1 2.5804 2.5801	5
-1 3 1 2.5387 2.5397	m
0 0 2 2,5310 2,5319	W
-1 1 2 2 4710 2 4713	1767
-2 0 2 2 4323 2 4329	1714
1 2 1 2 4100 2 4177	
1 3 1 2.4190 2.4177	VW
3 1 1 2.3770 2.3773	vw
0 4 0 2.2331 2.2331	w
-3 1 2 2.1226 2.1236	vw
-4 2 1 2.0533 2.0532	vw
0 4 1 2.0434 2.0432	w
2 4 0 2.0181 2.0179	m
-2 4 1 1 9280 1 9297	ww
3 3 1 1 8991 1 8991	ww
	~ ~
D I U I.0430 I.0441	iu i
2 4 1 1.0224 1.0235	w
4 2 1 1.8224 1.8215	w
3 1 2 1.7730 1.7733	VW
-5 1 2 1.6446 1.6444	VW
1 5 1 1.6400 1.6404	vw
5 1 1 1.6373 1.6370	vw
4 4 0 1.6199 1.6207	17W
-3 1 3 1 5944 1.5958	m
-5 2 1 1 5944 1 5947	
	ALL NY ALL
1.5514 1.5525	vw
4 0 2 1.5807 1.5805	vw
6 0 0 1.5702 1.5706	vw
3 5 0 1.5529 1.5529	W
-1 3 3 1.4887 1.4891	m
0 6 0 1.4887 1.4887	m
-6 0 2 1.4664 1.4663	vw
5 3 1 1.4532 1.4534	m
-4 2 1 1.4352 1.4359	WW
-3 3 3 1.4201 1.4243	w
2 2 3 1 4201 1 42245	147
2 6 0 1.4201 1.4195	w
1 5 2 1 4201 1 4190	147
_3 5 2 1 3834 1 3834	1762
L.JO24 L.JO24	V W
D I 4 I.3/22 I.3/33	VW
/ 1 0 1.3306 1.3312	W
5 5 U 1.2965 1.2966	W
0 6 2 1.2831 1.2833	ww

monoclinic C2/c $\alpha = 9.6010(8)$ Å, b = 8.9323(6) Å, c = 5.1592(5) Å, $\beta = 101.034(9)$, V = 434.27(6) Å, Z = 8, $\rho_{calc.} = 4.431 \text{ g/cm}^3$

The monoclinic MgGeO₃ crystals grown below 810°C (type A) were found to have unit cell dimensions of a =9.6010(8), b = 8.9323(6), c = 5.1592(5)Å, $\beta = 101.034(9)^{\circ}$. X-ray diffraction data taken for powdered type A monoclinic MgGeO3 are shown in Table 4 with the calculated dspacings based on the refined unit cell dimensions determined for a single crystal using high 2θ angle reflections and a four-circle diffractometer. These d-spacings are close to those by Robbins and Levin (1959) except for their three unindexed lines mentioned below. In Table 5, the present crystal data for monoclinic MgGeO3 are compared to those of Robbins and Levin (1959) and Kirfel and Neuhaus (1974). The space group of this phase is C2/cwhich is the same as that for "high-clinoenstatite" (Smith, 1969) and diopside. The present result is different from that of Kirfel and Neuhaus (1974) who deduced $P2_1/m$ or $P2_1$ on the basis of the diffraction data for polycrystalline powder by Robbins and Levin (1959). Furthermore, our value of the unit cell dimension, a, is about half of that given by them. As has been noted in the JCPDS-card No. 11-504 by L. G. Berry, three strong reflections reported by Robbins and Levin (1959) could not be indexed on the basis of the unit cell dimensions a =9.60, b = 8.92, c = 5.16Å, $\beta = 100.49^{\circ}$. In order to index these three lines, Kirfel and Neuhaus (1974) had to double the *a*-value. Consequently, they were forced to conclude the space group was P not C.

As shown in the crystal growth experiments, both monoclinic and orthorhombic phases have grown, and furthermore the "clino-from proto" phase was also obtained below 955°C. We also found that single crystals of proto-type MgGeO₃ inverted both to clinoenstatite- and orthoenstatite-type MgGeO₃ during the cooling process to room temperature. These experimental results strongly imply that the sample (polycrystalline powder) prepared by Roth (1955) ("Ge-enstatite"), Robbins and Levin (1959) ("enstatite form" and "clinoenstatite form"), and by Kirfel and Neuhaus (1974) ("ortho-MgGeO3" and "klino-MgGeO₃") might not have been a single phase. In the case of MgGeO₃, as well as MgSiO₃, phase transitions at atmospheric pressure among orthorhombic, monoclinic and proto-type orthorhombic phases do not induce a drastic change in the crystal structure. Therefore, for a powder sample of MgGeO₃, it would be very difficult to determine whether the sample is really single phase or composed of both the orthoenstatite-type and the clinoenstatite-type. The three unindexed lines in Robbins and

Table 5. Crystal data of clinoenstatite-type MgGeO₃

Robb	ins & Levin(1959) Kirfel & Neuhaus(197	4) This study
a /Å	9.60	19.246(17)	9,6010(8)
b /A	8.92	8.960(5)	8,9323(6)
c /Å	5.16	5.196(3)	5.1592(5)
β	100°49'	100°52'(3')	101.034(9) °
V /Å Space	434.01	880.0(9)	434.27(6)
Group	Ce or C2/e	$P2_1 \text{ or } P2_1/m$	C2/c

Levin (1959) imply that their "clinoenstatite form" sample contained considerable amounts of an extra phase. Actually, these three lines can be successfully indexed for an orthoenstatite-type. In particular, these three lines ((1) $d = 4.74, I/I_0 = 35, (2) d = 3.35, I/I_0 = 70, (3) d = 3.25,$ $I/I_0 = 14$) could be regarded as the (400), (121), and (420) reflections, respectively. Another possibility is as follows: As will be seen in Table 2 and 4, since d_{200} of the clinoenstatite-type is slightly larger than d_{400} of the orthoenstatite-type, the d = 4.74 line would be d_{200} of clinoenstatite-type, and d = 4.70 line previously indexed as d_{200} of clinoenstatite-type might correspond to d_{400} of orthoenstatite-type. These interpretations seem to be reasonable, since Robbins and Levin (1959) prepared their sample "clinoenstatite form" MgGeO3 above 1555°C, and on cooling, the orthoensatite-type can be expected to appear. Their "clinoenstatite form" MgGeO₃ sample should be assigned to our type B monoclinic phase, "clino-from proto" MgGeO₃, and should be distinguished from the type A (primary) monoclinic MgGeO₃.

Both "clino-from proto" and "clino-from ortho" MgGeO₃ crystals with very fine twinning structure (Fig. 1(iii), (iv)) were also confirmed to be crystallographically identical to the type A monoclinic MgGeO₃ grown at temperatures below 810°C (Fig. 1 (i), (ii)). No clinoenstatite-type MgGeO₃ crystals with P lattice type were found in this study. Detailed structure analyses of both orthoenstatite-type and clinoenstatite-type MgGeO₃ will be reported elsewhere.

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