

Refinement of the Margarite Structure in Subgroup Symmetry

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Abstract

The crystal structure of margarite- $2M_1$ from Chester County, Pennsylvania, has been refined as an ordered derivative structure in subgroup Cc of the ideal space group $C2/c$. Because of the high pseudosymmetry involved, successful refinement by least-squares required initial movement of atomic parameters away from those of the disordered phase towards those of the possible ordered models predicted by a distance least-squares program. Ordering of tetrahedral Si and Al is nearly complete in a pattern that violates centrosymmetry between the two tetrahedral sheets within a 2:1 layer. Compositionally similar tetrahedra in adjacent sheets are related instead by a pseudo two-fold axis that extends laterally through the octahedral Al atoms and is normal to the direction of intralayer shift. The two tetrahedral sheets differ significantly in Si,Al contents. A similar ordering pattern is theoretically possible in muscovite- $2M_1$ also, but was not found to be adopted by muscovite from the Diamond mine, South Dakota.

Introduction

Margarite, the dioctahedral brittle mica, has been recognized only as the $2M_1$ form. Takéuchi (1965) refined the structure for a specimen from Chester, Massachusetts, using three dimensional film data. He found the tetrahedral Si,Al cations to be disordered over the two non-equivalent sites $T(1)$ and $T(2)$ of the ideal space group $C2/c$. Since then, Gatineau and Méring (1966) because of the absence of diffuse X-ray scattering and Farmer and Velde (1973) because of the sharpness of the infrared spectra and the absence of Al-O-Al vibrations have suggested that margarite should be ordered.

One way to resolve the conflicting evidence above, as pointed out by Farmer and Velde, is to assume that the tetrahedral cations are ordered in a lower subgroup symmetry. Because of systematic absences, the only possible subgroup is Cc . This paper reports a successful refinement of margarite as an ordered derivative structure in space group Cc . Because of the similarities of the margarite and muscovite structures, the $2M_1$ form of muscovite also was reexamined in subgroup symmetry.

Experimental

A margarite crystal $0.32 \times 0.30 \times 0.02$ mm in size from Corundum Hill near Unionville, Chester County, Pennsylvania, was chosen for study because of its sharp reflections and lack of streaking due to

stacking faults. Table 1 presents the results of electron microprobe analysis, as averaged for two adjacent crystals. Scanning electron microscopic (SEM) analysis indicates that the crystals are homogeneous and that all elements have been accounted for that are present within the detection limits of the instrument.

Intensities of 1,071 independent, non-zero reflections were measured with a Syntex $P\bar{1}$ autodiffractometer in the variable-scan speed mode (Table 2). Graphite-monochromatized $MoK\alpha$ radiation was used, and only reflections for which $I > 2\sigma(I)$ were considered observed. Two reflections were monitored after every 50 measurements in order to check electronic stability. The integrated intensity I was calculated from $I = [S - (B_1 + B_2)/B_r]T_r$, where S is the scan count, B_1 and B_2 the background counts, B_r the ratio of background time to scan time, and T_r the 2θ scan rate in degrees per minute. $\sigma(I)$ is defined as equal to $T_r[S + (B_1 + B_2)/B_r^2 + q(I)^2]^{1/2}$, where q is equal to 0.003, an estimate of the standard error squared. Intensities then were corrected for Lorentz-polarization effects and for absorption. Table 3 lists the cell parameters based on least squares refinement of 15 independent high 2θ reflections.

Refinement

As a first step, the atomic coordinates of Takéuchi (1965) were refined by the least-squares program

TABLE 1. Electron Microprobe Analysis of Margarite from Chester County, Pennsylvania

Weight percent		Cations per 22 positive charges		
SiO ₂	31.58	Si	2.110	} 4.000 ^{IV}
Al ₂ O ₃	49.29	Al	3.882	
FeO	0.21	Fe ²⁺	0.012	} 2.036 ^{VI}
MgO	0.32	Mg	0.032	
CaO	11.35	Ca	0.812	} 1.011 ^{XII}
Na ₂ O	1.46	Na	0.190	
K ₂ O	0.10	K	0.009	
Sum of oxides	94.31			

ORFLS using the new data in the ideal space group $C2/c$. The residual R_1 converged to 8.4 percent after the scattering factors were adjusted for the composition indicated by microprobe analysis. Bond lengths calculated at this stage confirmed Takéuchi's earlier finding of tetrahedral disorder in the ideal space group.

Because of the high pseudosymmetry when subgroup Cc is assumed, one cannot merely expand the parameter set over the inversion centers of $C2/c$ and obtain reasonable results by least-squares refinement from the coordinates of the disordered model. Even if the parameters of pseudosymmetry-related atoms are refined independently, it was found that the R factor increased and convergence was not obtained. It proved necessary to move the atoms away from their pseudosymmetric positions by postulating their approximate positions for all feasible ordered models and then to refine each model separately to determine the correct one.

In Cc symmetry the two tetrahedral sheets within the 2:1 layer no longer are equivalent, and two different ordering models may be postulated for margarite that are consistent with disorder in the parent space group but full order in the subgroup. Approximate atomic coordinates for each ordered model were obtained by using the distance least-squares program OPTDIS of W. A. Dollase (University of California, Los Angeles). The coordinates for the disordered structure were used as input to this program along with the bond lengths and bond strengths that would be expected in the ordered models between first and second nearest neighbors.

Subsequent refinement by ORFLS showed that the procedure of producing artificially ordered models removed the previous problems of high correlations and lack of convergence. It was, however, still neces-

sary to vary the parameters of pseudosymmetry-related atoms independently. After initial increase in R_1 to about 10 percent, both ordered models converged to better residuals than for the disordered model. It was evident, however, that one ordered model could be rejected because it was moving towards the disordered model. Bonds around the postulated Al^{IV} became smaller and those around the postulated Si^{IV} became larger. This model was implausible for crystallochemical reasons in addition, as it segregated Si and Al in separate tetrahedral sheets. The second model retained its postulated ordering pattern during refinement and converged to a residual of 7.5 percent with isotropic B values. Application of Hamilton's (1965) residual ratio test indicates that this ordered model is a significant improvement over the disordered model at better than the 1 percent significance level (Table 4).

An electron density difference map did not show the position of the hydrogen atom, but did show the presence of a density of approximately one electron in the vacant octahedral position $M(1)$ in accord with the microprobe analysis of slightly over 2.0 octahedral atoms. Two final cycles of isotropic refinement incorporating this one electron in $M(1)$ plus the hydrogen atom position calculated by Giese (in preparation) and using corrections for anomalous scattering did not affect the residual significantly. Table 5 lists the final atomic coordinates and isotropic temperature factors, as compared with Takéuchi's disordered structure.

Discussion

The basic structure determined in this study is very similar to that of Takéuchi, except for the distortion to lower symmetry resulting from the ordering scheme. Pertinent data are summarized in Table 6. Table 7 presents bond length and angle calculations. No standard deviations of bond lengths are presented because only half of the atomic coordinates were varied in any one refinement cycle in order to reduce correlation effects. A correlation matrix involving all coordinates cannot be obtained, therefore, although the results from the two sets of refinements must be interrelated because the same data set was used for each. The standard deviations of the bond lengths are believed to be similar (approximately 0.003 to 0.005 Å) to those obtained in other isotropic refinements using diffractometer data with a similar ratio of number of reflections to variable parameters. Differences in isotropic B values for pseudosymmetry-related atoms (Table 5) are believed to be

TABLE 2. Observed and Calculated Structure Amplitudes

k		l		10Fo		10Fc		k		l		10Fo		10Fc		k		l		10Fo		10Fc		
h		n		10Fo		10Fc		h		n		10Fo		10Fc		h		n		10Fo		10Fc		
0	2	304	185	6	11	413	412	1	-14	628	636	7	-15	386	372	2	-23	218	221	10	-5	534	487	
0	4	271	345	6	-11	433	414	1	15	106	149	7	16	209	209	2	-24	225	162	10	6	554	526	
0	6	2029	1907	6	12	588	551	1	-15	746	762	7	-16	109	58	2	-25	390	390	10	-6	586	539	
0	8	230	182	6	-14	743	684	1	16	181	157	7	17	413	440	2	-26	331	303	10	7	162	129	
0	10	2013	1971	6	15	165	161	1	-17	276	300	7	-17	269	246	4	0	516	537	10	-8	300	277	
0	12	995	991	6	-15	129	160	1	19	117	97	7	20	162	124	4	1	200	205	10	-9	155	141	
0	14	694	719	6	16	691	662	1	-20	112	89	7	-21	260	276	4	-2	209	209	10	-10	174	165	
0	16	991	1030	6	-16	695	664	1	21	437	458	7	-21	220	207	4	-3	467	472	10	-11	210	184	
0	18	153	186	6	17	111	83	1	-21	226	256	7	-22	339	247	4	-4	562	536	10	-12	345	318	
0	20	1139	1091	6	-17	116	153	1	22	333	295	7	0	499	466	4	-4	642	607	10	-13	137	138	
0	22	761	758	6	19	147	150	1	-22	256	256	7	1	961	894	4	5	532	541	10	-14	362	356	
0	26	384	363	6	-20	903	845	1	23	205	204	9	-1	534	504	4	-5	1041	1048	10	-15	210	184	
2	0	388	346	6	20	132	173	1	-23	447	467	9	-1	534	504	4	-5	1041	1048	10	-15	210	184	
2	1	321	241	6	-21	172	173	1	24	398	404	9	2	149	114	4	-6	828	831	1	0	100	73	
2	1	282	247	6	21	134	175	1	-24	427	434	9	3	685	627	4	7	160	170	1	1	327	314	
2	2	550	433	6	-22	540	538	1	25	285	261	9	4	214	179	4	-7	457	455	1	-1	130	216	
2	2	562	442	6	22	540	538	1	-26	154	132	9	4	214	179	4	-7	457	455	1	-1	130	216	
2	3	904	770	8	0	237	264	3	0	366	366	9	-4	192	170	4	-8	214	213	1	-2	796	620	
2	3	882	766	8	-1	175	168	3	1	2261	2121	9	-5	755	691	4	-9	248	264	1	-3	736	767	
2	4	1176	1072	8	-1	161	177	3	-1	1115	1128	9	-6	118	100	4	-10	384	394	1	-3	609	616	
2	4	1183	1077	8	2	426	413	3	2	93	88	9	7	150	140	4	-10	384	394	1	-3	609	616	
2	5	1509	1384	8	-2	402	404	3	2	894	848	9	-7	418	371	4	-11	161	152	1	-4	211	203	
2	5	1508	1386	8	3	425	400	3	-3	1508	1344	9	8	271	267	4	-11	161	152	1	-4	211	203	
2	6	286	262	8	-3	403	396	3	-4	172	180	9	-8	308	276	4	-12	492	511	1	-5	904	931	
2	6	292	254	8	4	358	329	3	5	2071	1884	9	9	1299	1183	4	-13	525	517	1	-6	511	527	
2	7	629	606	8	-4	266	332	3	-5	1078	1095	9	-9	418	371	4	-13	525	517	1	-6	511	527	
2	7	628	611	8	5	832	767	3	7	685	630	9	-10	359	342	4	-13	290	285	1	7	87	105	
2	8	178	201	8	-5	798	768	3	-7	960	913	9	-10	441	410	4	-14	430	410	1	8	195	231	
2	8	190	193	8	7	371	344	3	-8	220	235	9	-11	121	99	4	-14	430	410	1	8	195	231	
2	9	82	83	8	-8	136	127	3	-9	2025	2054	9	-11	121	99	4	-14	430	410	1	8	195	231	
2	10	433	431	8	8	116	127	3	-9	1333	1295	9	13	277	230	4	-15	691	681	1	-9	495	553	
2	10	442	428	8	-8	151	121	3	10	215	201	9	-13	680	591	4	-16	230	270	1	10	199	193	
2	12	227	227	8	9	106	77	3	-10	303	286	9	-14	321	308	4	-17	230	270	1	10	199	193	
2	12	235	244	8	-10	471	187	3	11	475	417	9	-15	295	255	4	-17	341	320	1	11	541	607	
2	13	587	606	8	-10	158	182	3	-11	2752	2742	9	16	151	166	4	-18	301	309	1	-11	234	273	
2	13	589	606	8	13	300	308	3	13	746	747	9	17	574	544	4	-18	301	309	1	-11	234	273	
2	14	195	201	8	-13	344	307	3	-13	1506	1448	9	-17	159	133	4	-19	311	311	1	-12	424	444	
2	14	201	180	8	14	243	165	3	14	415	417	9	-18	322	311	4	-21	117	152	1	-12	992	1079	
2	15	631	622	8	-14	144	166	3	15	10	157	126	9	-18	322	311	4	-21	117	152	1	-12	992	1079
2	15	644	624	8	15	463	434	3	-15	235	236	11	1	204	163	4	-22	178	111	1	14	483	569	
2	16	537	541	8	-15	435	434	3	16	108	128	11	3	281	257	4	-23	178	111	1	14	483	569	
2	16	531	511	8	16	136	166	3	-17	1050	1107	11	3	281	257	4	-23	178	111	1	14	483	569	
2	17	98	82	8	-16	151	158	3	-17	443	429	11	4	390	361	4	-24	165	154	1	15	504	592	
2	18	172	168	8	17	154	150	3	-18	139	187	11	-4	439	398	4	-24	165	154	1	15	504	592	
2	18	163	174	8	-18	245	215	3	19	744	801	11	-5	115	67	6	-1	457	487	1	-17	319	363	
2	20	210	209	8	-18	246	222	3	19	1122	1079	11	-5	115	67	6	-1	457	487	1	-17	319	363	
2	20	245	249	8	20	174	216	3	20	112	25	11	6	349	322	6	-2	268	288	1	-18	114	178	
2	21	158	145	8	-20	205	222	3	-20	206	204	11	-6	423	400	6	-2	268	288	1	-18	114	178	
2	22	178	203	10	1	198	173	3	21	539	521	11	-6	423	400	6	-2	268	288	1	-18	114	178	
2	22	418	393	10	-1	177	171	3	-21	615	596	11	-10	120	55	6	-4	960	920	1	-19	134	156	
2	23	428	395	10	2	350	330	3	23	1038	1030	11	-10	120	55	6	-4	960	920	1	-19	134	156	
2	24	252	259	10	-2	166	160	3	-24	130	110	11	-11	153	133	6	-6	641	588	1	-22	244	244	
2	24	294	294	10	3	164	154	3	25	285	314	11	-11	153	133	6	-6	641	588	1	-22	244	244	
2	25	299	299	10	-3	164	154	3	-25	285	314	11	-11	153	133	6	-6	641	588	1	-22	244	244	
2	25	328	303	10	4	604	524	3	0	188	186	11	0	971	1003	6	-6	641	588	1	-22	244	244	
2	26	149	135	10	-5	185	161	3	-5	980	351	11	0	971	1003	6	-6	641	588	1	-22	244	244	
2	26	190	142	10	5	185	161	3	5	980	351	11	0	971	1003	6	-6	641	588	1	-22	244	244	
4	0	323	108	10	-5	191	160	5	-1	262	249	11	-4	755	779	6	-9	130	143	3	1	336	319	
4	1	337	285	10	6	153	160	5	-2	347	329	11	-4	755	779	6	-9	130	143	3	1	336	319	
4	1	337	285	10	-6	146	161	5	2	1504	1472	6	8	719	749	6	-9	130	143	3	1	336	319	
4	1	337	279	10	7	193	185	5	-3	489	481	11	-4	755	779	6	-9	130	143	3	1	336	319	
4	2	527	434	10	-7	175	177	5	4	653	642	11	-4	755	779	6	-9	130	143	3	1	336	319	
4	2	527	434	10	7	193	185	5	-3	489	481	11	-4	755	779	6	-9	130	143	3	1	336	319	
4	2	500	443	10	8	234	227	5	-4	1127	1076	11	-4	755	779	6	-9	130	143	3	1	336	319	
4	3	682	506	10	-8	253	227	5	5	645	636	11	-4	755	779	6	-9	130	143	3	1	336	319	
4	3	680	598	10	9	181	168	5	-5	773	762	11	-4	755	779	6	-9	130	143	3	1	336	319	
4</																								

TABLE 2, Continued

k	l	10Pc	10Pc	k	l	10Pc	10Pc	k	l	10Pc	10Pc	k	l	10Pc	10Pc
3	-2	162	178	5	-10	369	382	0	-2	1130	1311	4	1	235	230
3	3	603	651	5	-10	227	220	0	-3	115	0	4	2	411	423
3	-3	1035	1096	5	-11	391	392	0	4	295	314	4	-2	109	80
3	-4	125	92	5	12	540	582	0	6	707	783	4	3	453	448
3	5	505	524	5	13	136	163	0	-6	153	166	4	4	134	128
3	-5	509	518	5	-13	310	327	0	8	499	524	4	-4	284	287
3	7	406	493	5	14	301	353	0	10	215	219	4	-5	195	186
3	-7	200	206	5	-14	412	417	0	-10	409	530	4	-6	448	444
3	9	856	1001	5	-15	275	270	0	-12	450	500	4	-7	292	273
3	-9	231	251	5	-16	592	601	0	-14	258	382	4	8	158	184
3	11	660	751	5	-17	220	193	0	-16	410	423	4	-8	178	142
3	-11	1117	1199	7	1	474	456	1	0	106	128	4	-10	285	296
3	12	108	90	7	-1	235	217	2	1	408	468	6	0	211	160
3	-15	131	152	7	2	243	258	2	2	267	281	6	-1	195	164
3	16	112	137	7	-2	125	109	2	-2	157	219	6	-2	1026	914
3	-17	314	343	7	4	303	277	2	3	639	695	6	-3	223	184
3	-19	622	591	7	-4	130	133	2	-3	242	255	6	4	315	248
5	0	276	284	7	5	334	318	2	4	207	227	6	5	128	129
5	1	161	149	7	-5	167	181	2	-4	528	594	6	-5	130	81
5	-1	334	317	7	-6	429	401	2	5	152	184	6	-7	134	110
5	2	362	365	7	7	168	175	2	6	185	179	6	-8	458	422
5	-2	236	257	7	-7	464	456	2	-6	381	427	6	-8	458	422
5	-3	154	124	7	-8	198	193	2	7	118	155	6	-9	128	89
5	-4	290	304	7	9	181	178	2	9	138	173	1	0	128	89
5	5	176	174	7	-9	177	137	2	-9	265	305	1	-1	256	303
5	6	209	203	7	-12	171	143	2	-10	285	331	1	2	212	258
5	-6	398	405	7	-13	222	190	2	-10	301	306	1	-2	187	169
5	7	478	503	7	11	157	185	2	11	157	185	1	3	277	373
5	-7	278	302	7	-13	150	180	2	-13	150	180	1	-3	186	215
5	8	110	97	7	14	164	182	2	14	164	182	1	-5	149	147
5	9	489	490	0	0	219	253	2	15	241	274	1	-6	346	373
5	-9	395	387	0	-1	144	0	2	-15	241	274	1	-7	270	300
5				0	2	168	203	2				1	-8	434	405

artifacts resulting from separating the atoms into two sets during refinement.

The resultant mean $T-O$ bond lengths of 1.640 and 1.747 Å in one tetrahedral sheet and 1.624 and 1.730 Å in the other sheet indicate (1) nearly complete ordering for the composition $Si_{2.11}Al_{1.89}$ given by microprobe analysis, and (2) a slightly asymmetric distribution of Si,Al between the two sheets. The ordering pattern is of special interest, and is illustrated in Figure 1. Tetrahedra in the lower tetrahedral sheet (dashed) that would be related by inversion centers (small open circles) to tetrahedra in the upper sheet (full line) of the same 2:1 layer in the ideal space group $C2/c$ prove to be compositionally different when refined in subgroup Cc . A pseudo two-fold axis normal to the direction of intralayer shift and passing laterally through the two octahedral Al atoms can be seen to relate compositionally similar tetrahedra in the two sheets instead. Güven (1971a) has pointed out for muscovite- $2M_1$ that ordering may be inhibited in the ideal space group $C2/c$ because ordering would cause two apical oxygens (of Al-rich tetrahedra) along the same octahedral shared edge to be electrostatically unbalanced. This unstable situation is avoided in muscovite- $3T$, where the two tetrahedral sheets are related by a true two-fold axis instead of by inversion centers and where ordering of

TABLE 3. Unit Cell Parameters

	Takéuchi (1965)	This study
a	5.123 Å	5.1038(4) Å
b	8.886	8.8287(7)
c	19.221	19.148(1)
β	95.5°	95.46(3)

TABLE 4. Results of Refinement

	In $C2/c$	In Cc
R (%)	8.4	7.5
wR	10.7	9.5
variable parameters	40	76
data set	1,071	1,071

tetrahedral cations has been confirmed within the ideal space group $P3_112$. The Cc ordered pattern in margarite- $2M_1$ also avoids two unbalanced oxygens on the same octahedral shared edge, but the two-fold axis relating compositionally similar tetrahedra in the two tetrahedral sheets does not hold for the structure as a whole.

Tetrahedral cation occupancies as calculated from the mean $T-O$ values with the regression equation given by Hazen and Burnham (1973) for micas are listed in Table 7. The total tetrahedral Al calculated in this manner (1.895 Al) is in excellent agreement with that indicated by microprobe analysis (1.890 Al). There is a significant difference in calculated tetrahedral compositions for the two tetrahedral sheets within the 2:1 layer (1.049 and 0.846) due to the fact that ordering is more complete in the $T(1)$ and $T(11)$ sites than in the $T(2)$ and $T(22)$ sites. The sheets are internally consistent structurally in that the more Al-rich sheet also has a greater thickness and a slightly larger rotation angle α_{tet} (Table 6). The Al-rich tetrahedra are slightly flattened (larger τ) relative to their Si-rich counterparts, due to apical $T-O$ bonds shorter than mean basal $T-O$ bonds (Tables 6, 7). The bonds from the octahedral Al to the under-saturated apical oxygens attached to Al-rich tetrahedra are shorter than those to apical oxygens of Si-rich tetrahedra. The oxygen configurations about the octahedral Al are distorted in nearly the same way as in muscovite- $2M_1$, as described by Takeda and Ross (1975). Octahedral edges O(2)-O(22) and O(1)-OH(11) are nearly parallel to the x -axis, but OH(1)-O(11) is slightly more oblique than reported for muscovite.

Possible Ordering in Muscovite

The most detailed refinements of the $2M_1$ form of muscovite in its ideal space group $C2/c$ all have shown similar mean $T-O$ bond lengths as a consequence of disorder of the tetrahedral cations (e.g., Burnham and Radoslovich, 1964; Güven, 1971b). Because of the similarity of muscovite- $2M_1$ to margarite- $2M_1$, the question immediately arises whether muscovite also is ordered in subgroup symmetry Cc . Muscovite differs from margarite in its

TABLE 5. Atomic Coordinates for Margarite

	Takéuchi (1965) <i>C2/c</i>				This study <i>Cc</i>				
	x	y	z	B	x	y	z	B	
Ca(1)	0.00	0.0942	0.25	1.14	0.00	0.0932(2)	0.25	1.02(3)	
^a M(1)					0.2479	0.2507	0.4996	---	
M(2)					0.7469(7)	0.9187(4)	0.9997(2)	0.73(6)	
M(3)	0.2518	0.0815	0.00	0.29	0.2510(7)	0.0863(4)	0.9998(2)	0.55(6)	
^b T(1)	0.4628	0.9283	0.1432	0.53	0.4647(7)	0.9285(4)	0.1416(2)	0.99(7)	
T(11)					0.5364(6)	0.0752(3)	0.8549(2)	0.58(5)	
T(2)	0.4543	0.2575	0.1438	0.71	0.4567(6)	0.2572(3)	0.1444(2)	0.61(5)	
T(22)					0.5500(7)	0.7437(4)	0.8573(2)	0.97(6)	
O(1)	0.9547	0.4430	0.0553	1.25	0.960(2)	0.4436(9)	0.0512(4)	1.0(1)	
O(11)					0.046(2)	0.5585(8)	0.9390(4)	0.5(1)	
O(2)	0.3874	0.2524	0.0569	1.12	0.395(2)	0.2540(9)	0.0568(4)	0.9(1)	
O(22)					0.619(2)	0.7489(9)	0.9447(4)	0.7(1)	
O(3)	0.3597	0.0884	0.1788	0.40	0.367(2)	0.0974(9)	0.1775(5)	1.1(1)	
O(33)					0.641(2)	0.9167(9)	0.8226(4)	0.6(1)	
O(4)	0.2786	0.7839	0.1695	1.46	0.266(2)	0.7784(9)	0.1665(4)	0.5(1)	
O(44)					0.711(2)	0.214(1)	0.8304(4)	1.1(1)	
O(5)	0.2700	0.3903	0.1797	0.45	0.287(2)	0.3926(8)	0.1785(4)	0.4(1)	
O(55)					0.737(2)	0.608(1)	0.8211(5)	1.4(2)	
OH(1)	0.4492	0.5624	0.0505	0.87	0.448(2)	0.5693(8)	0.0488(4)	0.4(1)	
^c OH(11)					0.543(2)	0.4417(9)	0.9468(4)	1.0(1)	
H(1)					0.397	0.637	0.083	0.78	
H(11)					0.603	0.363	0.917	0.78	

^aCoordinates for the "vacant site" M(1) were obtained from an electron density difference map. These are unrefined coordinates included as fixed values in the final stages of refinement.

^bPseudosymmetry-related atoms in one tetrahedral sheet are indicated by doubling the last digit of the symbol for the corresponding atom in the other sheet.

^cCoordinates for the hydrogen atom are from Giese (1975), and were treated in a fashion similar to those of M(1).

tetrahedral composition of Si₃Al instead of Si₂Al₂. Because of this difference, only one ordered model is possible in subgroup *Cc* that would be disordered when averaged over the sites of the parent space group *C2/c*, and this is the same model that is adopted in margarite. The maximum degree of order then possible is to have alternation of Si and Si_{0.5}Al_{0.5} in adjacent tetrahedra.

This ordered model for muscovite-2*M*₁ has been tested using the neutron diffraction data of Rothbauer (1971) for a pegmatitic muscovite from the Diamond mine, South Dakota. The 774 reflections from Rothbauer were used to refine the ordered model in subgroup *Cc* by exactly the same procedure as for margarite. The coordinates refined smoothly away from those of the ordered model back to those of the disordered model, *i.e.*, back to coordinates consistent with space group *C2/c*. It can be concluded that the ordered model of margarite-2*M*₁ does not extend to muscovite-2*M*₁ and that avoidance of electrostatically unbalanced oxygens along the same octahedral shared edge is not sufficient driving force

TABLE 6. Pertinent Structural Details for Margarite

	Takéuchi (1965)	This study
α_{tet} (°)	20.4	^b 20.76, 20.64 (20.70)
τ_{tet} (°)	111.5	Si: 110.78, 110.41 (110.73) Al: 110.83, 110.88
ψ_{oct} (°)	58.9	58.8
Sheet thickness (Å)		^b 2.291, 2.234 (2.263)
tetrahedral	2.330	2.074
octahedral	2.074	2.074
Interlayer separation (Å)	2.832	2.868
Basal oxygen Δz_{ave} (Å)	0.19	0.20
Mean bond lengths (Å)		
T-O	1.692	T(1)=1.747, T(11)=1.624 (1.686)
	1.702	T(2)=1.640, T(22)=1.730 (1.685)
M(2,3)-O,OH	1.912	1.902, 1.910 (1.906)
Ca-O (inner)	2.458	2.455
(outer)	3.427	3.428

^aThe amount of tetrahedral rotation α_{tet} may be calculated from $\alpha = 1/2 |120^\circ - \text{mean } O_b-O_b-O_b \text{ angle}|$

^bThe value referring to the more Al-rich tetrahedral sheet is given first. The average values are in parentheses.

^cThe tetrahedral angle τ is defined as $\tau = \angle_{apical} \text{---} T \text{---} O_{basal}$, ideally $109^\circ 28'$.

^dThe mean octahedral angle ψ , ideally $54^\circ 44'$, is calculated from $\cos \psi = \text{oct. thickness} / 2(M-O,OH)$, where $M-O,OH$ is the mean of all octahedral cation to anion distances, including the vacant site.

TABLE 7. Interatomic Distances and Angles

Bond lengths (Å)			Bond angles (°)		
around T(1)					
T(1)--O(1) ^a	1.733	0(1)--O(3)	2.853	0(1)--O(3)	110.69
0(3)	1.735	0(4)	2.898	0(4)	112.02
0(4)	1.762	0(5)	2.856	0(5)	109.79
0(5)	1.758	0(3)--O(4)	2.868	0(3)--O(4)	110.18
Mean	1.747	0(5)	2.803	0(5)	106.73
(= 0.853 AI)		0(4)--O(5)	2.833	0(4)--O(5)	107.23
		Mean	2.852	Mean	109.44
around T(11)					
T(11)--O(11) ^a	1.614	0(11)--O(33)	2.643	0(11)--O(33)	108.64
0(33)	1.640	0(44)	2.698	0(44)	113.40
0(44)	1.614	0(55)	2.661	0(55)	110.30
0(55)	1.629	0(33)--O(44)	2.654	0(33)--O(44)	109.35
Mean	1.624	0(55)	2.663	0(55)	109.16
(= 0.098 AI)		0(44)--O(55)	2.588	0(44)--O(55)	105.90
		Mean	2.651	Mean	109.46
around T(2)					
T(2)--O(2) ^a	1.677	0(2)--O(3)	2.708	0(2)--O(3)	109.98
0(3)	1.629	0(4)	2.699	0(4)	110.53
0(4)	1.606	0(5)	2.734	0(5)	110.71
0(5)	1.647	0(3)--O(4)	2.612	0(3)--O(4)	107.65
Mean	1.640	0(5)	2.639	0(5)	107.30
(= 0.196 AI)		0(4)--O(5)	2.674	0(4)--O(5)	110.57
		Mean	2.678	Mean	109.46
around T(22)					
T(22)--O(22) ^a	1.676	0(22)--O(33)	2.779	0(22)--O(33)	108.56
0(33)	1.747	0(44)	2.889	0(44)	113.56
0(44)	1.777	0(55)	2.790	0(55)	110.52
0(55)	1.719	0(33)--O(44)	2.845	0(33)--O(44)	107.68
Mean	1.730	0(55)	2.772	0(55)	106.17
(= 0.748 AI)		0(44)--O(55)	2.865	0(44)--O(55)	110.04
		Mean	2.823	Mean	109.42
Ca(1)					
Ca(1)--O(3)	2.436		3.391		
0(4)	2.509		3.541		
0(5)	2.430		3.375		
0(33)	2.404		3.432		
0(44)	2.470		3.514		
0(55)	2.479		3.313		
Mean (inner)	2.455	(outer)	3.428		
M(2)					
M(2)--O(1)	1.853	0(1)--O(2)	2.774	T(1) to T(2)	
0(2)	1.930	0(22)	2.844	around 0(3)	119.22
OH(1)	1.877	OH(1)	2.730	around 0(4)	126.38
0(11)	1.927	0(2)--OH(1)	2.803	around 0(5)	118.56
0(22)	1.912	OH(11)	2.839	Mean	121.39
OH(11)	1.910	0(11)--OH(1)	2.801		
Mean	1.902	0(22)	2.760	T(11) to T(22)	
		OH(11)	2.737	around 0(33)	119.63
		0(22)--OH(11)	2.753	around 0(44)	125.17
		Mean	2.782	around 0(55)	119.89
		(unshared)		Mean	121.56
		0(1)--O(11)	2.451		
		0(2)--O(22)	2.455		
		OH(1)--OH(11)	2.346		
		Mean (shared)	2.417		
M(3)					
M(3)--O(1)	1.870	0(1)--O(2)	2.764		
0(2)	1.942	OH(1)	2.837		
OH(1)	1.890	OH(11)	2.778		
0(11)	1.999	0(2)--O(11)	2.997		
0(22)	1.870	OH(1)	2.797		
OH(11)	1.893	0(11)--O(22)	2.758		
Mean	1.911	OH(11)	2.776		
		0(22)--OH(1)	2.753		
		OH(11)	2.740		
		Mean	2.800		
		(unshared)			

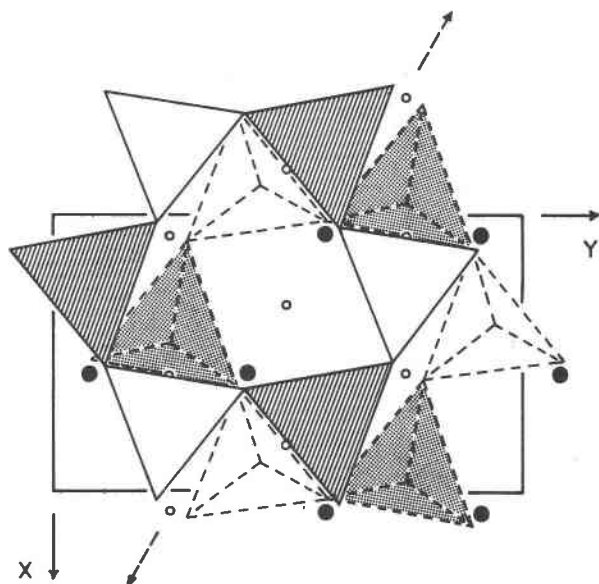
^aApical oxygen

FIG. 1. Tetrahedral ordering pattern within upper 2:1 layer of margarite-2M₁ in subgroup Cc. Al-rich tetrahedra = ruled lines and dots; octahedral Al = solid circles; inversion centers of ideal space group = small open circles; pseudo 2-fold axis = dashed arrows. From Bailey (1975).

It is encouraging to note here, as in the margarite refinement, that use of the ordered-model approach to avoid pseudosymmetry effects does not bias the least-squares refinement towards the ordered model so greatly that an incorrect model cannot be identified.

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for the ordering in this case. Possible adoption by muscovite of symmetry still lower than Cc, which would require violation of systematic absences and would permit complete order, has not been investigated.

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