

1 Revision 1

2 **Electronic properties and compressional behavior of Fe–Si alloys at high pressure**

3

4 Seiji Kamada<sup>1,2,\*</sup>, Nanami Suzuki<sup>2</sup>, Fumiya Maeda<sup>2</sup>, Naohisa Hirao<sup>3</sup>, Maki Hamada<sup>2,4</sup>,

5 Eiji Ohtani<sup>2</sup>, Ryo Masuda<sup>5</sup>, Takaya Mitsui<sup>6</sup>, Yasuo Ohishi<sup>3</sup>, Satoshi Nakano<sup>7</sup>

6

7 1: Frontier Research Institute for Interdisciplinary Sciences, Tohoku University, Sendai,  
8 980-8578, Japan.

9 2: Department of Earth Science, Tohoku University, Sendai, 980-8578, Japan.

10 3: Japan Synchrotron Radiation Research Institute, Sayo, Hyogo, 679-5198, Japan.

11 4: School of Natural System, College of Science and Engineering, Kanazawa University,  
12 Kanazawa, 920-1192, Japan.

13 5: Research Reactor Institute, Kyoto University, Osaka, 590-0494, Japan.

14 6: National Institute for Quantum and Radiological Science and Technology, Sayo,  
15 Hyogo, 679-5148, Japan.

16 7: National Institute for Materials Science, Tsukuba, 305-0044, Japan.

17 \*: Corresponding author.

18

19

### Abstract

20 **Planetary cores are composed mainly of Fe with minor elements such as Ni, Si, O,**  
21 **and S. The physical properties of Fe alloys depend on their composition. Changes**  
22 **in *c/a* ratio, center shifts, and elastic properties of Fe and Fe–Ni alloys were**  
23 **reported previously. However, such properties of Fe–light–element alloys have not**  
24 **yet been extensively studied. Si is a plausible candidate as a light element in**

25 planetary cores. Therefore, we studied the electronic properties and compressional  
26 behavior of Fe-Si alloys with a hexagonal-close-packed (hcp) structure under high  
27 pressure using synchrotron Mössbauer spectroscopy (SMS) and X-ray diffraction  
28 (XRD). Center shifts (CS) were observed at pressures of 21.4–45.3 GPa for  
29 Fe–2.8wt.%Si and of 30.9–62.2 GPa for Fe–6.1wt%.Si. Some of SMS and XRD  
30 measurements were performed under the same conditions using a newly developed  
31 system at the BL10XU beamline of SPring-8, which allowed simultaneous  
32 characterization of the electron information and crystal structure. Changes in the  
33 CS values were observed at 36.9 GPa in Fe–2.8wt%.Si and 54.3 GPa in  
34 Fe–6.1wt%.Si. The ratios of  $c/a$  in the hcp structure were measured at pressures of  
35 21.2–49.6 GPa in Fe–2.8wt.%Si and 32.9–61.4 GPa in Fe–6.1wt%.Si. The  $c/a$  ratio  
36 changed at pressures of 30–45 GPa in Fe–2.8wt.%Si and at 50 GPa in  
37 Fe–6.1wt%.Si. Changes in the CS and  $c/a$  ratio were explained according to the  
38 electronic isostructural transition in Fe–Si alloys. In addition, the transition  
39 pressure increased with increasing Si content in metallic iron. This finding is  
40 significant as changes in seismic wave velocities due to the change in  $c/a$  ratio of  
41 Fe–Si alloys with an hcp structure might be observed if Venus has a solid inner  
42 core.

43

44 **Keywords:** Synchrotron Mössbauer spectroscopy; diamond anvil cell; electronic  
45 topological transition; compressional behavior; Fe-Si alloy

46

47

## Introduction

48 Iron is one of the main constituents of Earth and other terrestrial planets, and

49 hence, is an element of interest in planetary studies. The Earth's core is divided into an  
50 outer liquid core and an inner solid core based on seismological observations. These  
51 cores are less dense than pure metallic iron under the high pressure and temperature  
52 conditions corresponding to the core conditions. This density deficit of the cores  
53 suggests a presence of one or more light elements in addition to Fe. Silicon has been  
54 proposed as a viable candidate for one of the light elements in the core as it is one of the  
55 most abundant elements on Earth (Birch 1952; Ringwood 1959).

56 Iron is stable in a hexagonal-close-packed (hcp) structure up to the conditions  
57 of the Earth's core. The physical properties of hcp Fe have been studied at conditions of  
58 high pressure and temperature (e.g., Anzellini et al. 2013; Dewaele et al. 2006; Fiquet et  
59 al. 2001; Mao et al. 1990; Ohtani et al. 2013; Sakai et al. 2014; Sakamaki et al. 2016;  
60 Tateno et al. 2010). According to previous studies, the densities and seismic wave  
61 velocities of the Earth's inner core could not be explained if the core was pure Fe.  
62 Therefore, it is important to investigate the effects of light elements on the physical  
63 properties of Fe. Phase relationships in Fe–Si systems have been investigated at high  
64 pressures and temperatures, showing that hcp structures of Fe–Si alloys were stable  
65 under the core conditions (e.g., Fischer et al., 2012; 2013; Lin et al., 2002; 2009; Tateno  
66 et al., 2015). The compressional behavior (e.g., Fischer et al. 2012; 2013; Hirao et al.  
67 2004; Tateno et al. 2015) and sound velocities of hcp Fe–Si alloys (e.g., Badro et al.  
68 2007; Lin et al. 2003b; Mao et al. 2012) have been measured at high pressure and  
69 temperature in order to evaluate the contribution of Si to the density deficit and seismic  
70 velocity in the inner core. Lin et al. (2003a) and Hirao et al. (2004) estimated the  
71 amount of Si in the Earth's inner core to be in the range of 3–5 wt.% based on studies  
72 on the compressional behavior. According to compressional behavior and sound

73 velocity studies, Badro et al. (2007) concluded that the Earth's inner core contained 2.3  
74 wt.% of Si. Antonangeli et al. (2010) investigated sound velocities of Fe–Ni–Si alloys  
75 and proposed that the amount of Si in the core is 1–2 wt.% with 4–5 wt.% Ni.

76         Recently, Glazyrin et al. (2013) reported that hcp Fe and hcp Fe–Ni alloys  
77 experienced an electronic topological transition (ETT; Lifshitz 1960), which was not  
78 accompanied by a structural transition, based on X-ray diffraction (XRD), Mössbauer  
79 spectroscopy, sound velocity measurements, and theoretical calculations. They observed  
80 changes in the evolution of the  $c/a$  ratio with increasing pressure and a drastic increase  
81 in the Mössbauer center shift (CS) at around 40 GPa. They measured Debye sound  
82 velocities of Fe and Fe–Ni alloys and observed a change in the pressure effect at around  
83 40 GPa. They reported a change in  $dV_{\text{Debye}}/dP$  before and after the ETT of hcp Fe and  
84 Fe–Ni alloys. The slope of the curve ( $dV_{\text{Debye}}/dP$ ) below 40 GPa was larger than that  
85 above 40 GPa, suggesting that the sound velocities after the ETT occurred were lower  
86 than estimated sound velocities based on sound velocities before the ETT occurred.  
87 These results clearly show that the contribution of the EET in Fe alloys should be  
88 considered when discussing the sound velocities and/or structures in solid planetary  
89 cores.

90         The ETT should also be studied at high temperature for various Fe alloys (e.g.,  
91 Fe–Si, Fe–Ni–Si, Fe–S) in order to consider any possible effects on the core. Since a  
92 change in the pressure dependency of the  $c/a$  ratio is related to the ETT, as shown by  
93 Glazyrin et al. (2013), it is also important to consider this relationship at high  
94 temperature. Ono et al. (2010) reported a change in the pressure dependency of the  $c/a$   
95 ratio of hcp Fe at room temperature at around 50 GPa, while Ono (2015) reported that a  
96 similar change in hcp Fe occurred at 2000 K at around 150 GPa. Thus, there is

97 possibility that the ETT and change in  $c/a$  may occur in the Earth and planetary cores.  
98 Although the ETT in Fe and Fe–Ni alloys was investigated at high pressure and  
99 temperature, the electronic properties and  $c/a$  ratios have not yet been investigated in  
100 Fe–light-element alloys under hydrostatic conditions. Since Si is a prime candidate for a  
101 light element in the core and substitutes Fe in the same way as Ni, we examined the  
102 structural, electronic, and compression properties of Fe–Si alloys with hcp structure up  
103 to 60 GPa using a combination of XRD and synchrotron-based Mössbauer spectroscopy  
104 (SMS). We used the data to estimate the effect of Si on the ETT and  $c/a$  evolution as a  
105 function of pressure the interior of terrestrial cores.

106

107

### Experimental procedure

108 The starting materials for the Fe–Si alloys were synthesized from pellets of  
109 powder mixtures of Fe and Si, which was 33% enriched in  $^{57}\text{Fe}$ . The pellets were melted  
110 using a double-sided laser heating system with a reducing atmosphere of 99 at.% Ar and  
111 1 at.%  $\text{H}_2$  to avoid oxidation of the samples. The molten samples were quenched to  
112 room temperature by shutting off the laser power after heating for 5 seconds with a laser  
113 power between 20 and 30 W. These samples were polished and their chemical  
114 compositions were analyzed by a field-emission scanning electron microscope  
115 (FE-SEM, JEOL JSM-7001F) with energy dispersive X-ray spectroscopy (EDS), which  
116 confirmed compositions of Fe–2.8wt.%Si (5 at.% Si) and Fe–6.1wt.%Si (12 at.% Si). A  
117 homogeneous texture was also observed under FE-SEM. Chips from the polished  
118 samples with a thickness of 15  $\mu\text{m}$  were used for energy-domain SMS and in situ XRD  
119 experiments at high pressure conditions.

120 A symmetric diamond anvil cell (DAC) with a culet size of 300 or 450  $\mu\text{m}$

121 (depending on the desired pressure) was used to generate high pressure. A pre-indented  
122 rhenium gasket with a thickness of 40  $\mu\text{m}$  was drilled to produce a sample chamber with  
123 a diameter of half of the culet size used. After placing the starting material and some  
124 ruby chips into the sample chamber, helium gas was loaded into the chamber as a  
125 pressure medium using a gas-loading system described previously (Takemura et al.  
126 2001).

127 Room-temperature XRD patterns were collected at the BL10XU beamline of  
128 SPring-8 (Ohishi et al. 2008). Typical X-ray wavelengths were 0.4134(1) or 0.4425(1)  
129  $\text{\AA}$  and pinhole collimators with a diameter of 20 or 50  $\mu\text{m}$  were used. An imaging plate  
130 (IP; Rigaku R-Axis IV<sup>++</sup>, 300  $\times$  300 mm<sup>2</sup>, 0.10 mm pixel size) or a flat panel detector  
131 (XRD0822, Perkin Elmer, Inc.) was used as an X-ray detector. X-ray diffraction images  
132 were converted into conventional one-dimensional X-ray profiles by IPAnalyzer  
133 software package, where the 1D profiles were analyzed using the PDIndexer software  
134 package (Seto et al. 2010).

135 The energy domain SMS measurements using a nuclear Bragg monochromator  
136 were performed at the BL10XU and BL11XU beamlines at SPring-8 (Mitsui et al.  
137 2009). The X-ray energy used in the SMS experiments was set to 14.4125 keV using  
138 double monochromatic crystals, while the bandwidth of the incident X-rays was set to  
139  $\sim 3$  meV using a high-resolution monochromator (HRM). The HRM was placed in front  
140 of the sample when using the BL11XU beamline or behind the sample when using the  
141 BL10XU beamline in order to record the XRD patterns. The 14.4 keV single-line <sup>57</sup>Fe  
142 Mössbauer radiation from the broadband synchrotron radiation was filtered using an  
143 electronically forbidden pure nuclear Bragg reflection (333) of a <sup>57</sup>FeBO<sub>3</sub> single crystal  
144 near the Néel temperature in an external magnetic field. The final energy bandwidth of

145 the X-ray beam was around 15 neV. The Mössbauer resonance energy of the  $^{57}\text{FeBO}_3$   
146 crystal was controlled using the Doppler effect of light by oscillating the crystal, which  
147 was mounted on a velocity transducer and operated in a sinusoidal-acceleration mode,  
148 parallel to the reflection plane. The  $^{57}\text{Fe}$  Mössbauer radiation was counted using a NaI  
149 scintillation detector. A typical data collection time for one spectrum was 2 hours.  
150 Mössbauer spectra were referenced to a standard metallic iron foil at 1 atm and room  
151 temperature in air. The MossA software package (Prescher et al. 2012) was used for  
152 computational analysis and the spectra were fitted using a Lorentzian model.

153 The experimental pressure was measured using a ruby fluorescence method  
154 (Dewaele et al. 2008). The experimental pressure was determined as the average of  
155 pressures before and after the SMS or XRD measurements. The uncertainty in the  
156 pressure was considered to be the standard deviation of the measured pressures. Since  
157 helium was used as a pressure medium, the pressure distribution or differences between  
158 the values before and after the measurements were  $\leq 0.4$  GPa. Therefore, the stress  
159 effect on the center shift and  $c/a$  ratio was suppressed.

160

## 161 **Results and discussion**

162 Mössbauer spectra and XRD patterns from the hcp phases of Fe–2.8wt.%Si and  
163 Fe–6.1wt.%Si were obtained up to 60 GPa at room temperature. The hyperfine structure  
164 and unit cell parameters are summarized in Tables 1 and 2, respectively. A typical SMS  
165 spectrum is shown in Fig. 1, where a single peak from Fe–2.8wt%Si or Fe–6.1wt%Si at  
166 all pressures was observed, suggesting they were nonmagnetic within the present  
167 experimental resolution. The relationships between the CS and pressures for the present  
168 Fe–Si alloys are shown in Fig. 2, together with those of pure iron and the  $\text{Fe}_{0.9}\text{Ni}_{0.1}$  alloy

169 by Glazyrin et al. (2013). The CS of Fe–2.8wt.%Si was decreased with increasing  
170 pressure up to 35 GPa, were constant around -0.49 mm/sec in the range of 36 to 42 GPa,  
171 then decreased again above 43 GPa. The CS of Fe–6.1wt.%Si decreased up to 54 GPa.  
172 When the sample was pressurized at 57 GPa, the CS increased drastically from -0.286  
173 mm/sec to -0.02 mm/sec. Similar changes in CS were observed by Glazyrin et al. (2013)  
174 for pure iron and the Fe<sub>0.9</sub>Ni<sub>0.1</sub> alloy; they observed an increase in CS at ~40 GPa and  
175 concluded that the ETT occurred at this pressure based on a DFT calculation. The  
176 difference in CS before and after the ETT in Fe was much smaller than that in Fe<sub>0.9</sub>Ni<sub>0.1</sub>.  
177 In this study, we also observed similar behaviors in Fe–Si alloys. The large change in  
178 CS values was not observed for Fe–2.8wt.%Si, whereas the CS for Fe–6.1wt.%Si was  
179 0.284 mm/sec and similar behavior was observed in Glazyrin et al. (2013), which was a  
180 large CS change in Fe<sub>0.9</sub>Ni<sub>0.1</sub> compared to that of Fe. According to Glazyrin et al (2013),  
181 if this transition is the ETT, this shift is due to the change of second order Doppler shift  
182 as a result of the change of the topology of Fe electronic states, and *c/a* ratio should also  
183 show anomaly. Here, we observed that adding Si to Fe increased the pressure of the CS  
184 change from 40 to 54 GPa in this study. Assuming Si randomly substitutes Fe, as the Si  
185 content increases, the average number of Si atoms that occur as a nearest neighbor atom  
186 in the hcp crystal structure increases. Since Si has less electrons than Fe, the electron  
187 density at a Fe nucleus in the hcp structure of Fe–Si alloys is smaller than that of Fe in  
188 the hcp structure of Fe. Thus, a higher pressure is required to achieve the ETT with  
189 increasing Si content.

190 XRD patterns of Fe–2.8wt.%Si and Fe–6.1wt.%Si were recorded under a  
191 pressure of 11 to 61 GPa. Only hcp structured phases in Fe–2.8wt.%Si and  
192 Fe–6.1wt.%Si were observed above 21 GPa and 29 GPa, respectively. The 100, 002,



193 101, 102, 110, and 103 reflections were observed up to the highest pressure. The  $c/a$   
194 ratios were obtained for hcp structures and their pressure dependencies are shown in Fig.  
195 3. The  $c/a$  ratios of hcp Fe-Si alloys coexisting with bcc Fe-Si alloys are not shown in  
196 Fig. 3 due to the scattered data (probably due to the influence of the bcc phase).  
197 Although the error bars for the  $c/a$  ratios are large, the  $c/a$  ratio of Fe-2.8wt.%Si had a  
198 consistent value of 1.610 up to 30 GPa, decreased slightly to 1.606 at 45 GPa, and  
199 remained constant above this pressure. On the other hand, the  $c/a$  ratio for  
200 Fe-6.1wt.%Si slightly decreased up to 50 GPa, at which point it suddenly increased up  
201 to 1.613, decreased to 1.611 at around 51 GPa, and then stayed constant under further  
202 compression. Ono et al. (2010) observed that the  $c/a$  ratio of hcp-Fe decreased up to 50  
203 GPa and then stayed nearly constant above 50 GPa. A similar anomalous behavior as  
204 observed in Fe was also observed in Fe-Si alloys. The pressure where anomalous  
205 behavior occurred increased with increasing Si contents in Fe. Since we used helium  
206 gas as a pressure medium in the SMS and XRD measurements, changes in CS and  $c/a$   
207 (as shown in Figs. 2 and 3, respectively) are not due to the pressure distribution.  
208 Therefore, the changes may have been due to the ETT.

209 The equation of states (EOS) for Fe-Si alloys were also obtained in this study.  
210 P-V data set were fitted to the 3rd order Birch-Murnaghan EOS (3BM; eq. 1) and Vinet  
211 EOS (VEOS; eq. 2).

$$212 \quad P = \frac{3}{2} K_0 \left[ \left( \frac{V_0}{V} \right)^{\frac{7}{3}} - \left( \frac{V_0}{V} \right)^{\frac{5}{3}} \right] \left[ 1 - \frac{3}{4} (4 - K') \left( \left( \frac{V_0}{V} \right)^{\frac{2}{3}} - 1 \right) \right], \quad (1)$$

$$213 \quad P = 3K_0 \left( \frac{V}{V_0} \right)^{\frac{-2}{3}} \left[ 1 - \left( \frac{V}{V_0} \right)^{\frac{1}{3}} \right] \exp \left\{ \frac{3}{2} (K' - 1) \times \left[ 1 - \left( \frac{V}{V_0} \right)^{\frac{1}{3}} \right] \right\}, \quad (2)$$

214 where  $P$  and  $V$  are the experimental pressure in GPa and the experimental volume in  $\text{\AA}^3$ ,  
215  $V_0$  is the volume at zero pressure in  $\text{\AA}^3$ ,  $K_0$  is the bulk modulus at zero pressure in GPa,  
216 and  $K'$  is the pressure derivative of the bulk modulus.  $V_0$  values for Fe–2.8wt.%Si and  
217 Fe–6.1wt.%Si were obtained based on a  $g$ – $G$  plot (Jeanloz 1981), which is used for a  
218 material that is stable only under high pressure in order to estimate its  $V_0$  value.  $G$  and  $g$   
219 represent a normalized pressure and an effective strain and they are expressed as  
220  $G=P/\{3(1+2g)^{5/2}\}$  and  $g=1/2\times\{(V_2/V_{01})^{-2/3}-1\}$ , respectively, where  $V_2$  represents the  
221 volume of a high pressure phase and  $V_{01}$  is a reference volume of a low pressure phase  
222 at a reference pressure. The  $g$ – $G$  plots for hcp Fe–2.8wt.%Si and hcp Fe–6.1wt.%Si are  
223 shown in Fig. 4(a). Here,  $V_{01}$  was set as 23.44(2)  $\text{\AA}^3$  for bcc Fe–2.8wt.%Si at ambient  
224 pressure for hcp Fe–2.8wt.%Si and 21.89(6)  $\text{\AA}^3$  for bcc Fe–6.1wt.%Si at 11.5 GPa. The  
225 average values of  $V_0$  from the 2nd and 3rd  $g$ – $G$  plots were used as  $V_0$  for hcp  
226 Fe–2.8wt.%Si and hcp Fe–6.1wt.%Si, respectively.  $K_0$  and  $K'$  were obtained from  
227 fitting using the constant  $V_0$ . The  $V_0$  values from  $g$ – $G$  plots were used for the VEOS fits  
228 as a reference (e.g., Mao et al. 1990). Although we observed anomalous behavior in the  
229  $c/a$  ratio, we did not see any change in the  $P$ – $V$  plots, as shown in Fig. 4(b). The  
230 volumes of hcp Fe–Si alloys coexisting with a bcc phase were also used for EOS fitting  
231 as the volumes did not show any anomalies.

232 Following this procedure, we fitted 3BM and VEOS to the data over the  
233 explored conditions, where the obtained parameters are listed in Table 3 along with  
234 those reported in previous studies for comparison.  $V_0$  was also treated as a free  
235 parameter during the fits and obtained  $V_0$ ,  $K_0$ , and  $K'$  values are also listed in Table 3.  
236 The present  $V_0$  values obtained from  $g$ – $G$  plots are consistent with  $V_0$  obtained as the  
237 free parameter during the fits in this study. Obtained  $K_0$  and  $K'$  values for a fixed  $V_0$

238 were also consistent (within uncertainties) to those obtained with the free  $V_0$  values. The  
239 compression curves with fixed  $V_0$  are shown in Fig. 4(b). The black and gray solid  
240 curves represent 3BM for Fe–2.8wt.%Si and Fe–6.1wt.%Si, respectively. The black and  
241 gray dashed curves are VEOS for Fe–2.8wt.%Si and Fe–6.1wt.%Si, respectively. The  
242  $K_0$  values for Fe–2.8wt.%Si were found to be 196 GPa for 3BM and 195 GPa for VEOS.  
243 Those of Fe–6.1wt.%Si were 187 GPa for both 3BM and VEOS. These obtained values  
244 are similar to those reported by Asanuma et al. (2011) and Hirao et al. (2004), while  
245 larger than those reported by Lin et al. (2003) and Tateno et al. (2015), although these  
246 differences can be attributed to the different  $V_0$  values. As shown in Table 3, we  
247 recalculated  $V_0$  of Asanuma et al. (2011) and Tateno et al. (2015) based on g–G plots,  
248 where it is clear that the obtained  $V_0$  values were consistent with our values within error  
249 bars. The small  $K_0$  reported by Lin et al. (2003) and Fischer et al. (2014) can be  
250 explained by their large  $V_0$  values (which result in small  $K_0$ ). The hcp Fe–6.1wt.%Si has  
251 a larger unit cell volume than that of hcp Fe–2.8wt.%Si below 40 GPa, while above 40  
252 GPa, these volumes are similar. This suggests that Si in Fe slightly increases the unit  
253 cell volumes at relatively low pressures, as reported by Fischer et al. (2014).

#### 254 **Implication for Planetary cores**

255 We fitted the  $c/a$  data using a Boltzmann sigmoidal function, as follows  
256 (Boehler et al. 2008):

$$257 \quad \frac{c}{a} = \left(\frac{c}{a}\right)_f + \frac{\left(\frac{c}{a}\right)_i - \left(\frac{c}{a}\right)_f}{1 + \exp\left(\frac{P - P_{tr}}{dx}\right)}, \quad (3)$$

258 where  $(c/a)_f$  and  $(c/a)_i$  are final and initial  $c/a$  ratio values, respectively,  $P$  is the pressure  
259 in GPa and  $P_{tr}$  is the pressure at the inflection point of the eq. (3) in GPa, and  $dx$  is

260 related to the slope at  $P=P_{tr}$ . Eq. (3) was fitted to the experimental data, which yielded  
261  $P_{tr}$  values of 37.7(1.8) and 39.8(1.9) for Fe–2.8wt.%Si and Fe–6.1wt.%Si, respectively,  
262 with  $(c/a)_f$  and  $(c/a)_i$  values of 1.6054(3) and 1.6105(2) for Fe–2.8wt.%Si and 1.6108(8)  
263 and 1.6128(2) for Fe–6.1wt.%Si, which were obtained by averaging the initial and final  
264 data in Fig. 3. A similar fitting was performed for Fe data by Glazyrin et al. (2013),  
265 where  $(c/a)_f$  and  $(c/a)_i$  showed a proportional relationship to pressure, yielding  
266  $P_{tr}=34.1(4.2)$ . If  $P_{tr}$  is considered as the transition pressure where changes in the  $c/a$   
267 ratio occur, the pressure increased with increasing Si content in the Fe–Si alloy. The  
268 relationship between the transition pressure  $P_{tr}$  and Si content can be expressed as  
269 follows if a linear relationship is assumed:

$$270 \quad P_{tr} = 0.196(92) \times C_{\text{Si in Fe}} + 34.4(7), \quad (4)$$

271 where  $C_{\text{Si in Fe}}$  is the Si content in Fe [at%]. As reported by Ono (2015),  $P_{tr}$  may increase  
272 with increasing temperature. If the gradient of temperature–pressure relationship in Ono  
273 (2015) is applicable to the Fe–Si alloys, the effect of Si addition on  $P_{tr}$  under high  
274 temperature can be estimated. The estimated transition pressure–temperature  
275 relationships are shown in Fig. 5 along with Mercury, Venus, Earth, and Mars core  
276 conditions and Fe phase relationship. The transition  $P$ – $T$  relationships for pure Fe,  
277 Fe–6wt.%Si, and Fe–10wt.%Si were obtained using  $P_{tr}$  at room temperature and the  
278 temperature gradient from Ono (2015). The temperature profiles for Mercury and Mars  
279 are based on data from Rivoldini et al. (2009) and Rivoldini et al. (2011), respectively,  
280 while the core conditions for Venus are from Aitta (2012) and Lodders and Fegley  
281 (1998). Although we do not know exactly whether the terrestrial planets have solid  
282 inner cores due to lack of seismological data, our results indicate that only Venus’s core  
283 could undergo a change in  $c/a$  ratio, as shown in Fig. 5. For Mercury and Mars, the core

284 conditions are too hot compared to the transition  $P$ – $T$ . It should be noted that the  
285 Fe–alloys in the Mercury and Mars cores have an fcc structure rather than an hcp  
286 structure. Therefore, the transition probably does not occur under the conditions  
287 relevant for the cores of Mercury and Mars. On the other hand, the core temperature of  
288 the Earth is lower than the transition temperature. Therefore, a change in the  $c/a$  ratio  
289 might not be observed. In Venus, the temperature profile of the core intersects the  
290 transition conditions. The intersect pressures are between 237 and 251 GPa for the  
291 hot-core model and 173 and 187 GPa for the cold-core model, which corresponds to  
292 solid inner cores with radii of approximately 1300 and 2300 km, respectively. If Venus  
293 has a solid inner core and the inner core intercept boundary of the  $c/a$  changes as shown  
294 in Fig. 5, the inner core may have slower seismic wave velocities compared to those  
295 estimated under lower pressure conditions than an ETT pressure (e.g., Glazyrin et al.  
296 2013). However, changes in the  $c/a$  ratios of Fe–Si alloys were not investigated in the  
297 previous and this study under such high temperatures and pressures, and this point  
298 should be clarified in a future study. The present results show that the change in  $c/a$   
299 becomes unclear as increasing Si contents; it might be difficult to detect such changes in  
300  $c/a$  in Fe–Si alloys for high Si contents.

301

### 302 **Acknowledgments**

303 S.K. was supported by a Grant-in-Aid for Young Scientists (B) (No. 25800291) and E.O.  
304 was supported by a Grant-in-Aid for Scientific Research from the Ministry of Education,  
305 Culture, Science, Sport, and Technology of the Japanese Government (Nos. 22000002  
306 and 15H05748). The synchrotron radiation experiments were performed at the SPring-8  
307 facility with the approval of the Japan Synchrotron Radiation Research Institute

308 (Proposal Nos. 2013A1496, 2013A3513, 2013B0104, 2013A3517, 2014A0104,  
309 2014A1910, 2014A3516, 2014B0104, 2014A3519, 2015A0104, and 2015B0104) and  
310 Japan Atomic Energy Agency (Proposal Nos. 2013A-E01, 2013B-E07, 2014A-E06, and  
311 2014B-E09).

312

### 313 **References**

314 Aitta, A. (2012) Venus's internal structure, temperature and core composition. *Icarus*,  
315 218, 967–974.

316 Antonangeli, D., Siebert, J., Badro, J., Farber, D.L., Fiquet, G., Morard, G., and Ryerson,  
317 F.J. (2010) Composition of the Earth's inner core from high-pressure sound  
318 velocity measurements in Fe-Ni-Si alloys. *Earth and Planetary Science Letters*,  
319 295, 292–296.

320 Anzellini, S., Dewaele, A., Mezouar, M., Loubeyre, P., and Morard, G. (2013) Melting  
321 of iron at Earth's inner core boundary based on fast X-ray diffraction. *Science*, 340,  
322 464–466.

323 Asanuma, H., Ohtani, E., Sakai, T., Terasaki, H., Kamada, S., Hirao, N., and Ohishi, Y.  
324 (2011) Static compression of Fe<sub>0.83</sub>Ni<sub>0.09</sub>Si<sub>0.08</sub> alloy to 374 GPa and Fe<sub>0.93</sub>Si<sub>0.07</sub>  
325 alloy to 252 GPa: Implications for the Earth's inner core. *Earth and Planetary*  
326 *Science Letters*, 310, 113–118.

327 Badro, J., Fiquet, G., Guyot, F., Gregoryanz, E., Ocelli, F., Antonangeli, D., and  
328 d'Astuto, M. (2007) Effect of light elements on the sound velocities in solid iron:  
329 Implications for the composition of Earth's core. *Earth and Planetary Science*  
330 *Letters*, 254, 233–238.

- 331 Birch, F. (1952) Elasticity and constitution of the Earth's interior, Journal of  
332 Geophysical Research, 57, 227–286.
- 333 Birch, F. (1964) Density and composition of mantle and core, Journal of Geophysical  
334 Research, 69(20), 4377–4388.
- 335 Birch, F. (1978) Finite strain isotherm and velocities for single-crystal and  
336 polycrystalline NaCl at high pressures and 300 K. Journal of Geophysical Research,  
337 83(B), 127–1268.
- 338 Boehler, R., Santamaria-Perez, D., Errandonea, D., and Mezouar, M. (2008) Melting,  
339 density, and anisotropy of iron at core conditions: new X-ray measurements to 150  
340 GPa. Journal of Physics: Conference Series, 121, 022018.
- 341 Dewaele, A., Belonoshko, A. B., Garbarino, G., Occelli, F., Bouvier, P., Hanfland, M.,  
342 and Mezouar, M. (2012) High-pressure-high-temperature equation of state of KCl  
343 and KBr. Physical Review B, 85, 214105.
- 344 Dewaele, A., Loubeyre, P., Occelli, F., Mezouar, M., and Torrent, M. (2006)  
345 Quasihydrostatic equation of state of iron above 2 Mbar. Physical Review Letters,  
346 97, 215504.
- 347 Dewaele, A., Torrent, M., Loubeyre, P., and Mezouar, M. (2008) Compression curves of  
348 transition metals in the Mbar range: Experiments and projector augmented-wave  
349 calculations. Physical Review B, 78, 104102.
- 350 Fischer, R.A., Campbell, A.J., Caracas, R., Reaman, D.M., Dera, P., and Prakapenka,  
351 V.B. (2012) Equation of state and phase diagram of Fe-16Si alloy as a candidate  
352 component of Earth's core. Earth and Planetary Science Letters, 357–358,  
353 268–276.

- 354 Fischer, R.A., Campbell, A.J., Reaman, D.M., Miller, N.A., Heinz, D.L., Dera, P., and  
355 Prakapenka, V.B. (2013) Phase relations in the Fe-FeSi system at high pressures  
356 and temperatures. *Earth and Planetary Science Letters*, 373, 54–64.
- 357 Fischer, R.A., Campbell, A.J., Caracas, R., Reaman, N.A., Heinz, D.L., Dera, P., and  
358 Prakapenka, V.B. (2014) Equations of state in the Fe-FeSi system at high pressures  
359 and temperatures. *Journal of Geophysical Research*, 119, 2810–2827.
- 360 Fiquet, G., Badro, J., Guyot, F., Requardt, H., and Krisch, M. (2001) Sound velocities in  
361 iron to 110 gigapascals. *Science*, 291, 468–471.
- 362 Glazyrin, K., Pourovskii, L. V., Dubrovinsky, L., Narygina, O., McCammon, C.,  
363 Hewener, B., Schünemann, V., Wolny, J., Muffler, K., Chumakov, A. I., Crichton,  
364 W., Hanfland, M., Prakapenka, V.B., Tasnádi, F., Ekholm, M., Aichhorn, M.,  
365 Vildosola, V., Ruban, A. V., Katsnelson, M. I., and Abrikosov, I. A., (2013)  
366 Importance of correlation effects in hcp iron revealed by a pressure-induced  
367 electronic topological transition. *Physical Review Letters*, 110, 117206.
- 368 Hirao, N., Ohtani, E., Kondo, T., and Kikegawa, T. (2004) Equation of state of  
369 iron-silicon alloys to megabar pressure. *Physics Chemistry Minerals*, 31, 329–336.
- 370 Jeanloz, R. (1981) Finite-strain equation of state for high-pressure phases. *Geophysical*  
371 *Research Letters*, 8(12), 1219–1222.
- 372 Kamada, S., Ohtani, E., Terasaki, H., Sakai, T., Miyahara, M., Ohishi, Y., and Hirao, N.  
373 (2012) Melting relationships in the Fe–Fe<sub>3</sub>S system up to the outer core conditions.  
374 *Earth and Planetary Science Letters*, 359–360, 23–33.
- 375 Lifshitz, I.M. (1960) Anomalies of electron characteristics of a metal in the high  
376 pressure region. *Soviet Physics JETP*, 11(5), 1130–1135.



- 377 Lin, J.-F., Heinz, D. L., Campbell, A.J., Devine, J.M., and Shen, G. (2002) Iron-silicon  
378 alloy in Earth's core? *Science*, 295, 313–315.
- 379 Lin, J.-F., Campbell, A.J., and Heinz, D. L. (2003a) Static compression of iron-silicon  
380 alloys: Implications for silicon in the Earth's core. *Journal of Geophysical*  
381 *Research*, 108(B1), 2045.
- 382 Lin, J.-F., Struzhkin, V.V., Sturhahn, W., Huang, E., Zhao, J., Hu, M.Y., Alp, E.E., Mao,  
383 H.-K., Boctor, N., and Hemley, R.J. (2003b) Sound velocities of iron-nickel and  
384 iron-silicon alloys at high pressures. *Geophysical Research Letters*, 30(21), 2112.
- 385 Lin, J.-F., Scott, H.P., Fischer, R.A., Chang, Y.-Y., Kantor, I., and Prakapenka, V.B.  
386 (2009) Phase relations of Fe-Si alloy in the Earth's core. *Geophysical Research*  
387 *Letters*, 36, L06306.
- 388 Lodders, K., and Fegley, Jr., B. (1998) *The planetary scientist's companion*. 371 p.  
389 Oxford University Press, New York.
- 390 Mao, H.K., Wu, Y., Chen, L. C., Shu, J. F., and Jephcoat, A. P. (1990) Static  
391 compression of iron to 300 GPa and Fe<sub>0.8</sub>Ni<sub>0.2</sub> alloy to 260 GPa: Implications for  
392 composition of the core. *Journal of Geophysical Research*, 95(B13),  
393 21737–21742.
- 394 Mao, Z., Lin, J.-F., Liu, J., Alatas, A., Gao, L., Zhao, J., and Mao, H.-K. (2012) Sound  
395 velocities of Fe and Fe-Si alloy in the Earth's core. *Proceedings National*  
396 *Academy of Science*, 109, 10239–10244.
- 397 Mitsui, T., Hirao, N., Ohishi, Y., Masuda, R., Nakamura, Y., Enoki, H., Sakai, K., and  
398 Seto, M. (2009) Development of an energy-domain <sup>57</sup>Fe-Mössbauer spectrometer  
399 using synchrotron radiation and its application to ultrahigh-pressure studies with a  
400 diamond anvil cell, *Journal of Synchrotron Radiation*, 16, 723–729.

- 401 Ohishi, Y., Hirao, N., Sata, N., Hirose, K., and Takata, M. (2008) Highly intense  
402 monochromatic X-ray diffraction facility for high-pressure research at SPring-8.  
403 High Pressure Research, 28(3), 163–173.
- 404 Ohtani, E., Shibazaki, Y., Sakai, T., Mibe, K., Fukui, H., Kamada, S., Sakamaki, T., Seto,  
405 Y., Tsutsui, S., and Baron, A.Q.R. (2013) Sound velocity of hexagonal  
406 close-packed iron up to core pressures. Geophysical Research Letters, 40,  
407 5089–5094.
- 408 Ono, S., Kikegawa, T., Hirao, N., and Mibe, K. (2010) High-pressure magnetic  
409 transition in hcp-Fe. American Mineralogist, 95, 880–883.
- 410 Ono, S. (2015) Relationship between structural variation and spin transition of iron  
411 under high pressures and high temperatures. Solid State Communications, 203,  
412 1–4.
- 413 Ringwood, A.E. (1959) On the chemical evolution and densities of the planets.  
414 Geochimica et Cosmochimica Acta, 15, 257–283.
- 415 Rivoldini, A., Van Hoolst, T., and Verhoeven, O. (2009) The interior structure of  
416 Mercury and its core sulfur content. Icarus, 201, 12–30.
- 417 Rivoldini, A., Van Hoolst, T., Verhoeven, O., Mocquet, A., and Dehant, V. (2011)  
418 Geodesy constraints on the interior structure and composition of Mars. Icarus, 213,  
419 451–472.
- 420 Sakai, T., Takahashi, S., Nishitani, N., Mashino, I., Ohtani, E., and Hirao, N. (2014)  
421 Equation of state of pure iron and Fe<sub>0.9</sub>Ni<sub>0.1</sub> alloy up to 3 Mbar. Physics of Earth  
422 and Planetary Interiors, 228, 114–126.
- 423 Sakamaki, T., Ohtani, E., Fukui, H., Kamada, S., Takahashi, S., Sakairi, T., Takahata, A.,  
424 Sakai, T., Tsutsui, S., Ishikawa, D., Shiraishi, R., Seto, Y., Tsuchiya, T., and Baron,

- 425 A.Q.R. (2016) Constraints on Earth's inner core composition inferred from  
426 measurements of the sound velocity of hcp-iron in extreme conditions. *Science*  
427 *Advances*, 2, e1500802.
- 428 Seto, Y., Hamane, D., Nagai, T., and Sata, N. (2010). Development of a software suite  
429 on X-ray diffraction experiments. *The Review of high pressure science and*  
430 *technology*, 20(3), 269–276 (in Japanese).
- 431 Stixrude, L., Cohen, R.E., and Singh, D.J. (1994) Iron at high pressure:  
432 Linearized-augmented-plane-wave computations in the generalized-gradient  
433 approximation. *Physics Review B*, 50(9), 6642–6445.
- 434 Tateno, S., Hirose, K., Ohishi, Y., and Tatsumi, Y. (2010) The structure of iron in Earth's  
435 inner core. *Science*, 330, 359–361.
- 436 Tateno, S., Kuwayama, Y., Hirose, K., and Ohishi, Y. (2015) The structure of Fe-Si alloy  
437 in Earth's inner core. *Earth and Planetary Science Letters*, 418, 11–19.
- 438 Takemura, K., Sahu, P.Ch., and Toma, Y. (2001) Versatile gas-loading system for  
439 diamond-anvil cells. *Review of Scientific Instruments*, 72, 3873–3876.
- 440 Uchida, T., Wang, Y., Rivers, M.L., Sutton, S.R. (2001) Stability field and thermal  
441 equation of state of  $\epsilon$ -iron determined by synchrotron X-ray diffraction in a  
442 multianvil apparatus. *Journal of Geophysical Research*, 106(B10), 21799–21810.
- 443 Williams, Q., Jeanloz, R., Bass, J., Svendsen, B., and Ahrens, T.J. (1987) The melting  
444 curve of iron to 250 gigapascals: a constraint on temperature at Earth's core.  
445 *Science*, 236, 181–182.
- 446 Yamazaki, D., Ito, E., Yoshino, T., Yoneda, A., Guo, X., Zhang, B., Sun, W., Shimojuku,  
447 A., Tsujino, N., Kunimoto, T., Higo, Y., Funakoshi, K. (2012) P-V-T equation of  
448  $\epsilon$ -iron up to 80 GPa and 1900 K using the Kawai-type high pressure apparatus

449 equipped with sintered diamond anvils. Geophysical Research Letters, 39,  
450 L20308.  
451

452 **Figure captions**

453

454 Figure 1. A typical high-pressure synchrotron Mössbauer spectrum from Fe–2.8wt.%Si  
455 at 21.4 GPa.

456

457 Figure 2. CS values as a function of pressure. The open circles and squares represent CS  
458 values for Fe–2.8wt.%Si and Fe–6.1wt.%Si measured in this study. The cross and plus  
459 symbols are data for pure iron and Fe<sub>0.9</sub>Ni<sub>0.1</sub>, respectively, from Glazyrin et al. (2013).

460

461 Figure 3. The *c/a* ratios for the metals studied here compared to those from previous  
462 studies. The triangles and inverted triangles are the present results for Fe–2.8wt.%Si and  
463 Fe–6.1wt.%Si, respectively. The present P–V data are plotted for conditions where a  
464 single hcp phase was observed. The open circles and diamonds are previous data for Fe  
465 from Glazyrin et al. (2013) and Dewaele et al. (2006), respectively.

466

467 Figure 4. (a) *g*–*G* plots of Fe–Si alloys. The P–V data coexisting with bcc Fe–Si alloys  
468 were plotted and used to obtain *V*<sub>0</sub>. The black symbols show data for Fe–2.8wt.%Si,  
469 while the dark gray symbols show data for Fe–6.1wt.%Si. The black and dark gray solid  
470 lines are 2nd order *g*–*G* plots, and the dashed curves are 3rd order *g*–*G* plots. (b)  
471 Compressional behavior of Fe–2.8wt.%Si and Fe–6.1wt.%Si plotted with data from  
472 previous studies. The black triangles and dark gray inverted triangles show the volumes  
473 of Fe–2.8wt.%Si and Fe–6.1wt.%Si samples, respectively. The P–V data coexisting  
474 with bcc Fe–Si alloys are also plotted. The black solid and dashed curves show  
475 compression curves from the 3rd order Birch-Murnaghan equation of state and the Vinet

476 equation of state for Fe–2.8wt.%Si and dark gray solid and dashed curves show those  
477 for Fe–6.1wt.%Si.

478

479 Figure 5. Fe phase diagram with terrestrial planet core conditions and the transition  
480 boundary of  $c/a$  changes boundary based on this study and those of Glazyrin et al.  
481 (2013) and Ono et al. (2015). The Mercury and Mars core temperature and pressure  
482 conditions are based on Rivoldini et al. (2009) and Rivoldini et al. (2011), respectively.  
483 The Venus core conditions are from Aitta (2012) and Lodders and Fegley (1998). The  
484 Earth inner core conditions are based on Fe melting temperatures and studies of the  
485 effects of light element on the melting temperatures (e.g. Anzellini et al. 2013; Kamada  
486 et al. 2012).

487

488 **Tables**

489 Table 1. Hyperfine parameters of the samples based on SMS measurements.

<b>P [GPa]</b>	<b>CS [mm/s]</b>	<b>P [GPa]</b>	<b>CS [mm/s]</b>
<b>Run 1</b>	<b>Fe–2.8wt.%Si</b>	<b>Run 2</b>	<b>Fe–6.1wt.%Si</b>
21.4(1)	-0.295(19)	30.9(4)	-0.061(62)
23.4(2)	-0.320(22)	35.9(3)	-0.066(68)
25.3(3)	-0.334(24)	39.3(1)	-0.184(44)
27.7(4)	-0.354(24)	41.3(3)	-0.167(42)
29.5(4)	-0.376(26)	43.3(1)	-0.199(40)
31.3(4)	-0.413(28)	45.8(2)	-0.237(44)
33.7(4)	-0.441(29)	49.5(1)	-0.261(51)
35.6(3)	-0.454(27)	51.4(3)	-0.278(58)
36.9(3)	-0.487(35)	52.5(2)	-0.304(50)
39.4(6)	-0.492(27)	54.3(4)	-0.286(59)
41.2(3)	-0.488(40)	57.0(3)	-0.02(16)
43.3(4)	-0.525(36)	59.4(1)	-0.194(91)
45.3(3)	-0.545(38)	62.20(3)	-0.12(13)

The numbers in parentheses show experimental uncertainties.

Pressure errors were obtained from multiple measurements of ruby chips in the chamber using Dewaele's ruby scale (Dewaele et al. 2008).

490

491 Table 2. Lattice parameters of Fe–2.8wt.%Si and Fe–6.1wt.%Si alloys at high pressure.

P [GPa]	a [Å]	c [Å]	V [Å <sup>3</sup> ]	c/a	P [GPa]	a [Å]	c [Å]	V [Å <sup>3</sup> ]	c/a
<b>Run 3</b>					<b>Fe–2.8wt.%Si hcp</b>				
18.95(4) <sup>a</sup>	2.4466(10)	3.9421(22)	20.44(5)	1.6113(11)	20.46(42) <sup>a</sup>	2.4425(3)	3.9394(7)	20.35(2)	1.6129(3)
20.38(20) <sup>a</sup>	2.4422(9)	3.9337(19)	20.32(5)	1.6107(10)	22.39(41) <sup>a</sup>	2.4356(7)	3.9342(15)	20.21(4)	1.6153(8)
21.15(9)	2.4398(8)	3.9292(16)	20.26(5)	1.6105(8)	25.06(62) <sup>a</sup>	2.4296(5)	3.9198(13)	20.04(3)	1.6134(6)
22.40(16)	2.4360(8)	3.9223(16)	20.16(5)	1.6101(8)	27.05(48) <sup>a</sup>	2.4249(5)	3.9129(11)	19.92(3)	1.6137(6)
23.77(27)	2.4321(9)	3.9168(19)	20.06(5)	1.6104(10)	28.90(39)	2.4200(7)	3.9029(15)	19.79(4)	1.6128(8)
25.10(23)	2.4286(10)	3.9111(22)	19.98(6)	1.6105(11)	30.90(54)	2.4155(5)	3.8963(11)	19.69(3)	1.6130(5)
26.33(21)	2.4256(11)	3.9068(23)	19.91(6)	1.6107(12)	32.89(35)	2.4106(4)	3.8872(10)	19.56(3)	1.6125(5)
27.93(26)	2.4215(13)	3.9004(28)	19.81(8)	1.6107(14)	34.45(30)	2.4058(5)	3.8802(10)	19.45(3)	1.6128(5)
29.58(26)	2.4174(13)	3.8924(27)	19.70(8)	1.6102(14)	37.79(89)	2.3967(6)	3.8632(14)	19.22(4)	1.6119(7)
31.52(23)	2.4128(13)	3.8842(26)	19.58(7)	1.6098(14)	39.69(27)	2.3944(5)	3.8594(12)	19.16(3)	1.6118(6)
33.02(21)	2.4097(13)	3.8782(26)	19.50(7)	1.6094(14)	42.22(34)	2.3881(6)	3.8480(15)	19.01(4)	1.6113(7)
35.05(34)	2.4047(14)	3.8690(29)	19.38(8)	1.6089(15)	44.34(29)	2.3833(6)	3.8410(15)	18.90(4)	1.6116(7)
37.28(21)	2.3999(12)	3.8588(25)	19.25(7)	1.6079(13)	46.53(37)	2.3792(6)	3.8330(13)	18.79(3)	1.6111(7)
39.15(20)	2.3964(14)	3.8523(29)	19.16(8)	1.6076(15)	48.92(18)	2.3739(7)	3.8232(16)	18.66(4)	1.6105(8)
41.28(19)	2.3916(15)	3.8432(32)	19.04(9)	1.6069(17)	51.10(27)	2.3657(6)	3.8158(15)	18.49(4)	1.6130(7)
42.99(13)	2.3889(15)	3.8375(30)	18.97(8)	1.6064(16)	53.08(32)	2.3644(5)	3.8089(12)	18.44(3)	1.6109(6)
45.33(24)	2.3842(16)	3.8278(34)	18.84(9)	1.6055(18)	55.35(28)	2.3596(5)	3.8011(11)	18.33(3)	1.6109(6)
47.29(25)	2.3794(17)	3.8205(35)	18.73(9)	1.6057(18)	57.07(32)	2.3563(4)	3.7954(10)	18.25(3)	1.6107(5)
49.61(28)	2.3756(18)	3.8131(37)	18.64(10)	1.6051(19)	59.41 (41)	2.3512(3)	3.7886(8)	18.14(2)	1.6114(4)
					61.37(28)	2.3475(5)	3.7814(13)	18.05(3)	1.6108(7)
					59.74(1)	2.3506(5)	3.7864(11)	18.12(3)	1.6108(6)
					58.29(3)	2.3539(6)	3.7905(14)	18.19(4)	1.6103(7)
<b>Run 4</b>					<b>Fe–6.1wt%Si hcp</b>				
11.44(3) <sup>a</sup>	2.4759(32)	3.9944(75)	21.21(20)	1.6133(37)	55.06(6)	2.3607(7)	3.8010(17)	18.34(4)	1.6101(9)
11.49(4) <sup>a</sup>	2.4734(28)	3.9867(65)	21.12(18)	1.6118(32)	52.22(4)	2.3678(8)	3.8112(17)	18.50(5)	1.6096(9)
12.85(10) <sup>a</sup>	2.4721(26)	3.9892(61)	21.11(17)	1.6137(30)	46.84(143)	2.3787(17)	3.8278(33)	18.76(4)	1.6104(8)
14.75(9) <sup>a</sup>	2.4639(24)	3.9816(56)	20.93(15)	1.6160(27)	42.40(39)	2.3921(16)	3.8499(30)	19.05(4)	1.6114(8)
16.93(15) <sup>a</sup>	2.4570(22)	3.9615(51)	20.71(14)	1.6123(25)	38.79(23)	2.3979(7)	3.8654(16)	19.25(4)	1.6120(8)
19.02(15) <sup>a</sup>	2.4466(23)	3.9682(54)	20.57(14)	1.6219(27)	35.05(53)	2.4076(8)	3.8797(18)	19.48(5)	1.6114(9)

The numbers in parentheses show errors. Pressure errors were obtained from multiple measurements of ruby chips in the chamber using Dewaele's ruby scale (Dewaele et al. 2008).

a: hcp Fe–Si alloys coexisted with bcc Fe–Si alloys.



493 Table 3. The equation of state parameters for Fe-Si alloys.

Composition	P [GPa]	$V_0$ [Å <sup>3</sup> ]	$K_0$ [GPa]	$K'$	EOS <sup>a</sup>	ref.
Fe-2.8wt.%Si	19-50	22.19(7)	186(11)	4.80(38)	3BM	This study
		22.21(7)	183(11)	5.11(40)	VEOS	
		22.13(10) <sup>b</sup>	196.4(10)	4.45(7)	3BM	
		22.13(10) <sup>b</sup>	195.2(11)	4.64(8)	VEOS	
Fe-6.1wt.%Si	11-61	22.35(6)	190.7(90)	3.84(27)	3BM	This study
		22.37(7)	187.9(99)	4.04(34)	VEOS	
		22.38(1) <sup>b</sup>	187.2(20)	3.94(11)	3BM	
		22.38(1) <sup>b</sup>	186.1(21)	4.10(13)	VEOS	
Fe-3.4wt.%Si	23-252	22.4(28)	196(20)	4.3(2)	3BM	Asanuma et al. (2011)
		22.28(32) <sup>b</sup>	203(4)	4.29(9)	3BM <sup>c</sup>	
Fe-7.9wt.%Si	4.5-54	22.9(1)	141(10)	5.7(6)	3BM	Lin et al. (2003)
Fe-8.7wt.%Si	23-196	22.3(8)	198(9)	4.7(3)	3BM	Hirao et al. (2004)
Fe-9.0wt.%Si	47-197	23.9(2)	129.1(14)	5.29(8)	3BM	Fischer et al. (2014)
		24.19(8)	111.4(12)	6.08(8)	VEOS	
Fe-9.0wt.%Si	13-305	22.7(2)	168(12)	5.5(2)	VEOS	Tateno et al. (2015)
		22.24(32) <sup>b</sup>	206(4)	5.03(9)	VEOS <sup>d</sup>	
Fe	35-301	22.35(3)	164.8(36)	5.33(9)	3BM	Mao et al. (1990)
		22.35(3)	160.2(21)	5.82(8)	VEOS	
	6-15	22.7(3)	135(19)	6.0(4)	3BM	Uchida et al. (2001)
		22.468(24)	165(fixed)	4.97(4)	3BM	
	18-205	22.428(98)	163.4(79)	5.38(16)	VEOS	Dewaele et al. (2006)
		22.07(30) <sup>b</sup>	192.8(15)	4.93(5)	VEOS <sup>d</sup>	
	20-83	22.15(5)	202(7)	4.5(2)	3BM	Yamazaki et al. (2012)
		22.17(6)	196(8)	4.8(2)	VEOS	
	24-279	22.16(19) <sup>b</sup>	184.2(22)	4.78(12)	3BM	Sakai et al. (2014)
		22.16(19) <sup>b</sup>	179.2(22)	5.24(12)	VEOS	
		22.07(20) <sup>b</sup>	195.1(2.4)	4.93(12)	3BM	
		22.07(20) <sup>b</sup>	189.9(2.4)	5.39(13)	VEOS	

a: 3BM and VEOS represent the 3rd order Birch-Murnaghan and Vinet EOS, respectively.

b:  $V_0$  was fixed to the values obtained from g-G plots.

c: EOS parameters re-calculated with a fixed  $V_0$  from a g-G plot and pressures based on NaCl B2.

d: EOS parameters re-calculated with a fixed  $V_0$  from a g-G plot.

**Figures**

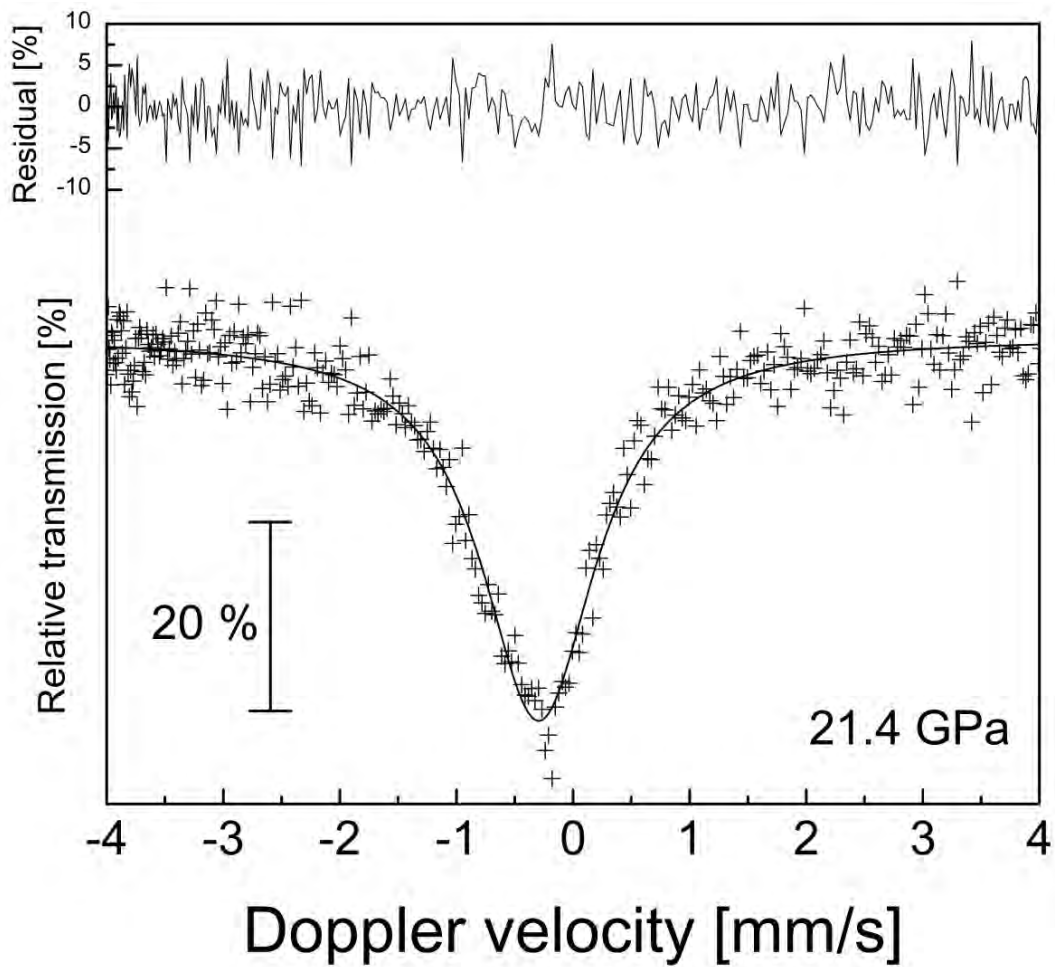


Fig. 1

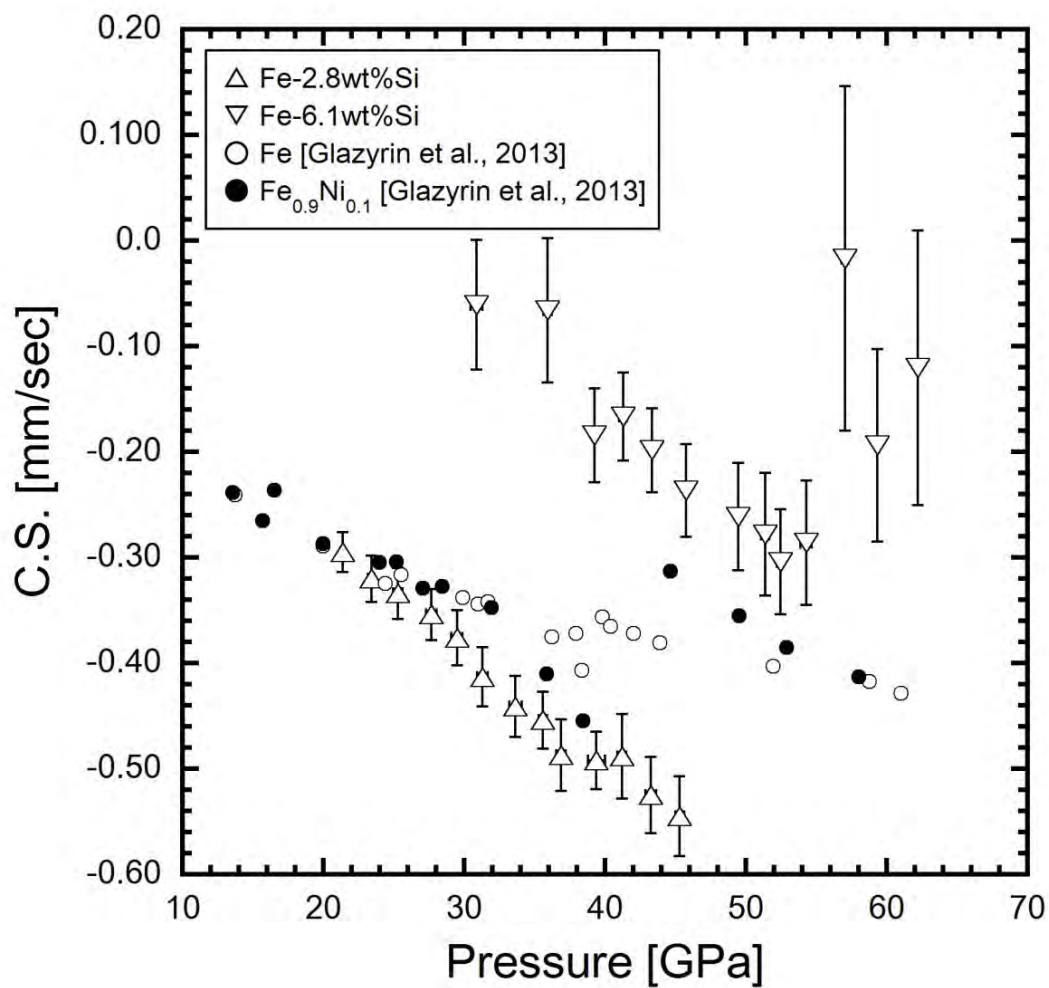


Fig. 2

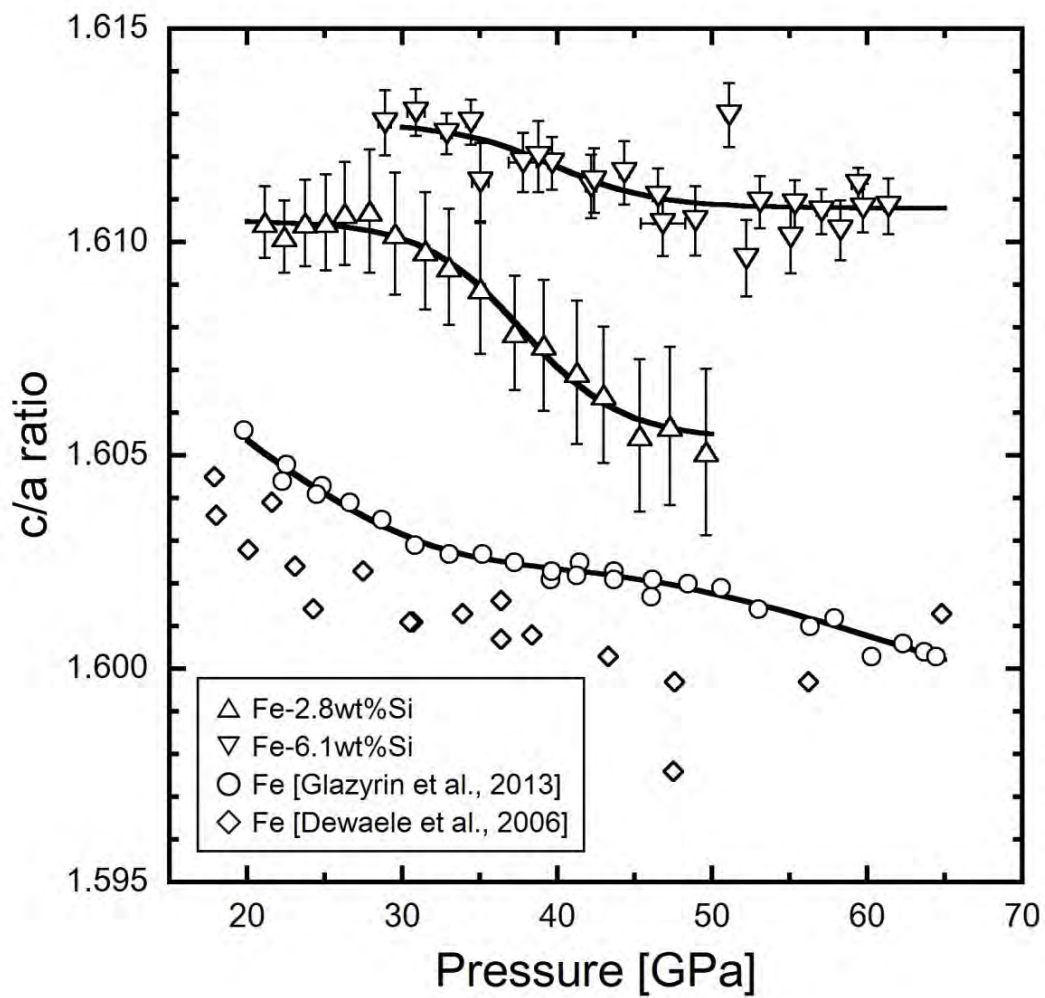


Fig. 3

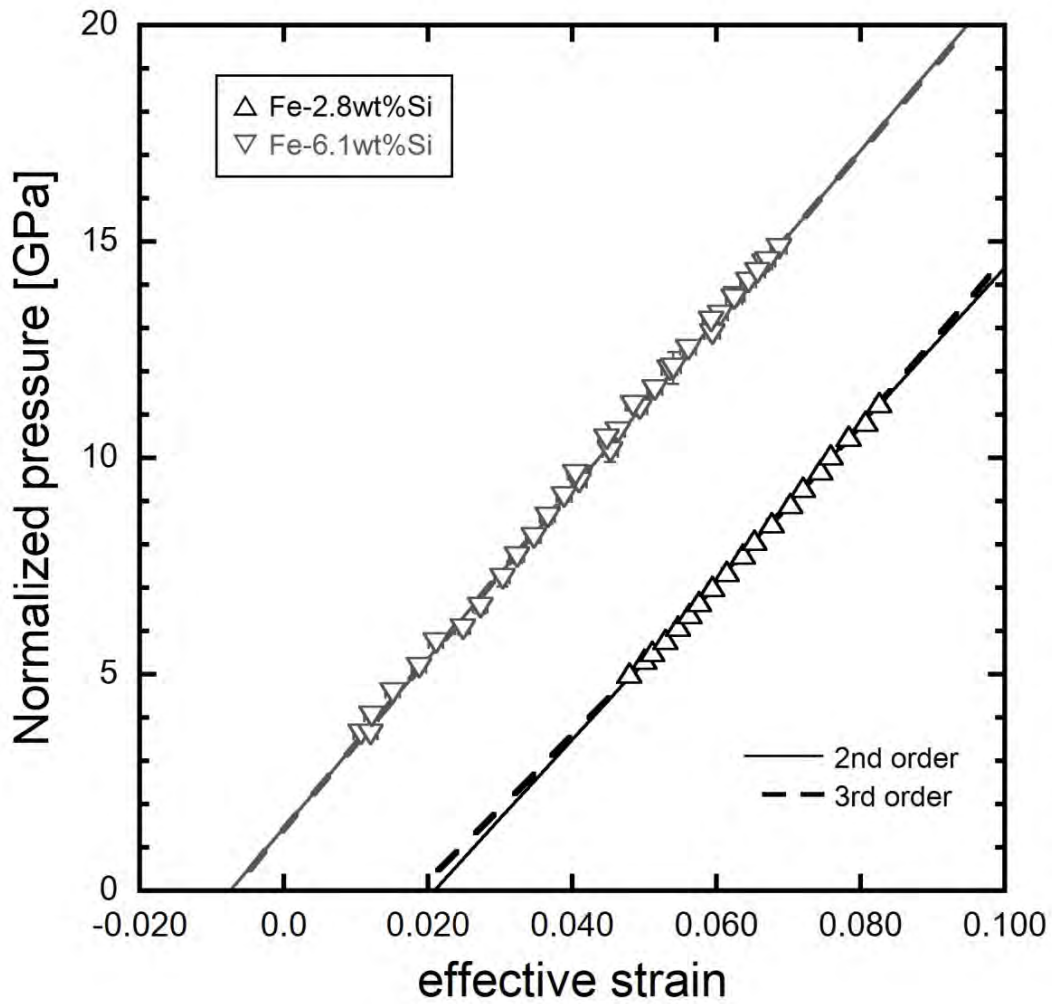


Fig. 4(a)

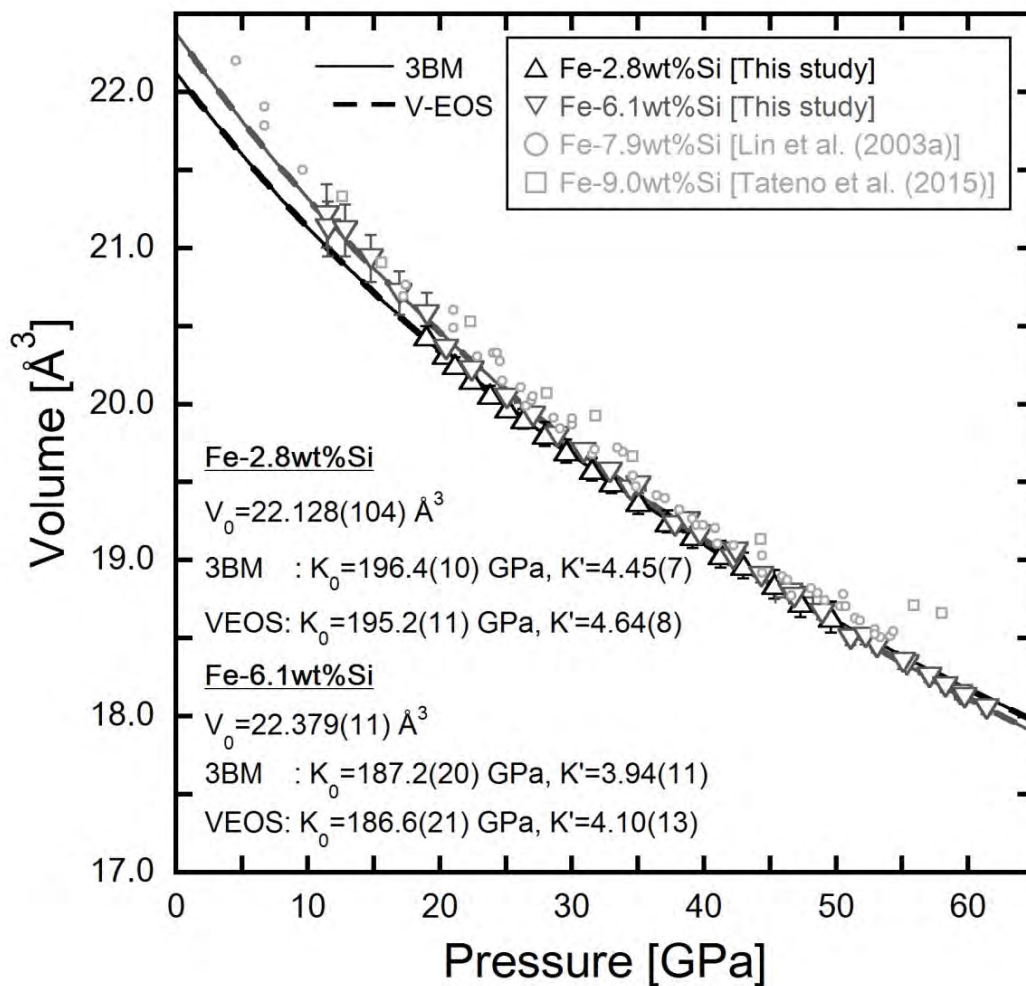


Fig. 4(b)

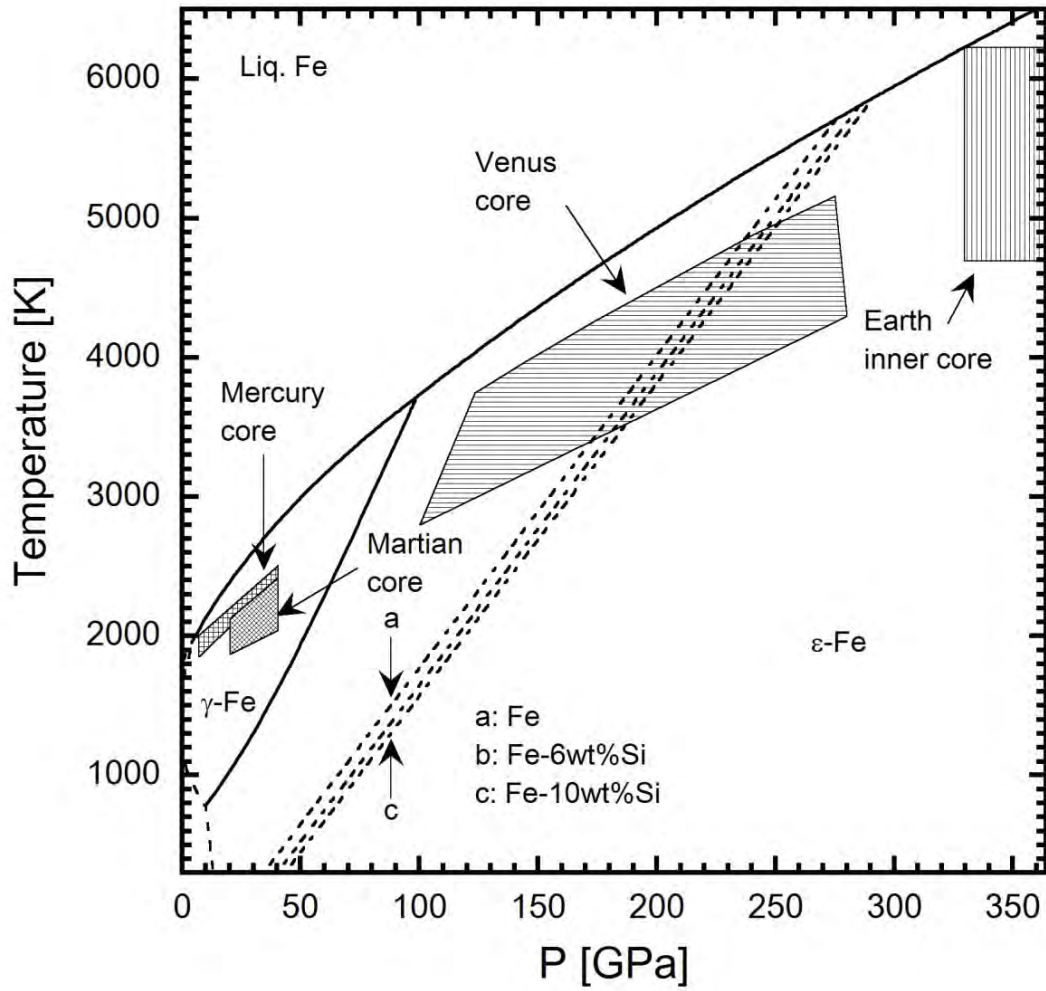


Fig. 5