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3	Structural incorporation of W ⁶⁺ into hematite and goethite: a
4	combined study of natural and synthetic iron oxides developed from
5	precursor ferrihydrite
6	Stefan Kreißl ^{*a} , Ralph Bolanz ^b , Jörg Göttlicher ^c , Ralph Steininger ^c , Mihail Tarassov ^d , Gregor
7	Markl ^a
8	
9	* corresponding autor; e-mail: stefan.kreissl@uni-tuebingen.de; phone: +49 70712973166
10	^a Eberhard Karls University, Faculty of Science, Department of Geosciences, Petrology and Mineral Raw Materials, Wilhelmstraße 56, 72074
11	Tübingen, Germany
12	^b Friedrich Schiller University, Institute of Geosciences, Department Mineralogy and Geochemistry, General and Applied Mineralogy, Carl-Zeiss-
13	Promenade 10, 07745 Jena, Germany
14	^c Karlsruhe Institute of Technology, ANKA Synchrotron Radiation Facility, Hard X-ray Spectroscopy, Hermann-von-Helmholtz-Platz 1, 76344
15	Eggenstein-Leopoldshafen, Germany
16	^d Bulgarian Academy of Sciences, Institute of Mineralogy and Crystallography, Department Mineralogy and Mineral Raw Materials, Acad. Georgi
17	Bonchev Str., bl. 107, 1113 Sofia, Bulgaria
18	
19	E-mail list:
20	Stefan Kreißl: stefan.kreissl@uni-tuebingen.de
21	Ralph Bolanz: ralph.bolanz@uni-jena.de
22	Jörg Göttlicher: joerg.goettlicher@kit.edu
23	Ralph Steininger: ralph.steininger@kit.edu
24	Mihail Tarassov: mptarass@dir.bg
25	Gregor Markl: gregor.markl@uni-tuebingen.de
26	

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Abstract

Hematite $(\alpha$ -Fe₂O₃) and goethite $(\alpha$ -FeOOH) can incorporate considerable amounts of 28 tungsten. Although tungsten concentrations up to several wt% in hematite and goethite have been 29 30 reported in the literature, none of the proposed models for a structural incorporation has been generally accepted yet. Here, the first combination of X-ray absorption fine structure (XAFS) 31 measurements with X-ray diffraction (XRD), Raman spectroscopy (RS), electron microprobe 32 33 (EMPA) and total reflection X-ray fluorescence (TXRF) provides a general relation between W 34 content and its structural incorporation into hematite and goethite. Botryoidal specimens of goethite and hematite, obtained from the Schwarzwald ore district, 35 36 Black Forest, SW Germany and from the Grantcharitza tungsten deposit, Bulgaria, display W concentrations of up to 5.5 and 2.15 mol% W for goethite and hematite, respectively. In addition 37 to these natural specimens, goethite and hematite were synthesized in the presence of W and 38 39 incorporate up to 7 and 1.3 mol% W, respectively. X-ray diffraction analysis does not indicate the presence of separate W-phases, supporting 40 the structural incorporation of W into the hematite and goethite. Refined unit cell parameters 41 indicate no changes with increasing W concentration in the hematites' but a rising structural 42 disorder within the structure of the synthetic goethites. Raman spectroscopy, however, shows an 43 increasing structural disorder for both synthetics, indicating an increase of Fe vacancies in both 44 45 hematite and goethite. A deprotonation mechanism for the goethite structure is unlikely according

- to the Raman results.
- 47 XAS near edge spectra indicate a strong distortion of the WO₆ octahedra in both hematite
 48 and goethite. Extended XAFS spectra of the natural and synthetic goethites and hematites show

49 striking similarities and suggest that W^{6+} resides in all samples on the Fe³⁺ position, again 50 without developing separate tungsten phases.

Calculations of the Fe-loss related to W incorporation reach mean values of ~2.9 and ~2.8 for goethite and hematite, respectively. The formation of two Fe^{3+} -vacancies in close proximity to the newly incorporated W⁶⁺ in addition to a protonation of the structures achieves charge balance within the hematite and goethite structure.

Hematite and goethite record the presence of W in fluids even in the absence of visible W 55 minerals. After W adsorption to ferrihydrite (the hematite and goethite precursor phase) and after 56 its transformation to either hematite or goethite, only hematite with up to 0.4 wt.% W is clearly 57 able to continuously monitor a changing W signature as a record of the fluid history within its 58 oscillatory growth zones. In contrast, goethite is probably not a good monitor of a primary W 59 fluid history. Their combination, however, could be particularly useful, as hematite records the W 60 concentrations in a fluid during ferihydrite precipitation, while goethite records W concentrations 61 during later ferrihydrite maturation. Botryoidal Fe-ores have never been considered for W 62 recovery but could play an important role to fight a potential supply risk of W as a high-63 technology metal. 64

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66 Keywords: tungsten, iron oxides, iron hydroxides, substitution, hydrothermal,
67 crystallography, spectroscopy, XAFS

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Introduction

Tungsten is a vital metal for our modern industry and is used in many modern applications 69 70 reaching from high-temperature and mechanically resistant electrodes, as an additive in metal alloys for high-temperature resistant steel to cathodes for microwave magnetrons and as 71 substitute for lead and uranium in specific ammunition types (Koutsospyros et al. 2006). In 2010, 72 73 the European Union rated W as a critical resource due to its economic importance and foreseeable supply risk (EU Commission 2010). In addition to its economic value, tungsten 74 variations in oscillatory zonation patterns of katathermal garnets, were used to interpret the 75 chemical evolution of hydrothermal systems by recording distinct fluid batches (Jamtveit et al. 76 1993). 77

Besides the most important W ore minerals, wolframite ((Fe,Mn)WO₄) and scheelite 78 (CaWO₄), iron oxides formed under hypogene and supergene conditions can exhibit high 79 concentrations of W with up to ~0.5 mol% (hypogene; Benvenuti et al. 2013) and 5.5 mol% 80 (supergene; Tarassov et al. 2002). Hematite and goethite are suitable candidates to be 81 investigated for fluid evolution studies, because they show an intense oscillatory zonation of W, 82 which is essential for such studies (cf. e.g. Haase et al. 1980; Manning and Bird 1990; Jamtveit 83 1991; Yardley et al. 1991; Putnis et al. 1992; Jamtveit and Hervig 1994; Jamtveit et al. 1993, 84 1995). 85

At low temperatures and under near-surface conditions, hematite and goethite are both transformation products of ferrihydrite $(Fe_{10}^{3+}O_{14}(OH)_2)$; Michel et al. 2007), a poorly crystalline ferric oxide, which forms during the early stages of Fe^{2+} oxidation, its hydrolysis and a rapid supply of Fe-monomers and -dimers (Cornell and Schwertmann 2003). Jambor and Dutrizac (1998) and references therein summarize structural observations, the transformation of ferrihydrite to hematite and goethite and its composition. During the formation of ferrihydrite,

other chemical elements such as W become immobilized in significant quantities, either by 92 surface adsorption or by co-precipitation (Kashiwabara et al. 2013). Depending on temperature, 93 94 pH, and the presence of other ions, ferrihydrite then transforms to goethite (via a dissolutionreprecipitation process) or to hematite (by a topotactic solid-state transformation; Cornell and 95 Schwertmann 2003; Cudennec and Lecerf 2006). These two formation mechanisms are highly 96 97 significant for the structural incorporation of minor elements into hematite and goethite, since Wbearing hematite forms from W-bearing ferrihydrite, while W-bearing goethite forms during the 98 polymerization of it's $Fe^{3+}O_6$ -octahedra in W enriched solutions. 99

Hence, for the structural incorporation of W into hematite and goethite, the structural 100 characteristics of both phases are of great importance. The structures of goethite and hematite 101 consist of hexagonal close packing (hcp) arrays of anions (O_2^- and OH^- for goethite and O_2^- for 102 hematite) stacked along the [001] direction with Fe^{3+} ions occupying 1/2 (for goethite) and 2/3 103 (for hematite) of the octahedral sites. Goethite is isostructural with diaspore (α -AlOOH) and 104 consists of FeO₆ double chains oriented parallel to [010]. The edge-sharing FeO₆-octahedra form 105 double chains connected by corner-sharing among each other (Alvarez et al. 2008). Hematite, on 106 the other hand, is isostructural with corundum. The FeO₆-octahedra in hematite are connected by 107 corner-, edge-, and face-sharing which results in significantly shorter Fe-Fe distances for the 108 face-sharing FeO₆-octahedra. 109

For a successful substitution of Fe^{3+} , W^{6+} has to be octahedrally coordinated by oxygen. At pH >4.4, tungsten is predominantly present as dissociated, tetrahedral coordinated WO_4^{2-} . At pH 112<4.4, W occurs as $WO_3(H_2O)_3$ in octahedral coordination (Cruywagen 2000; Smith et al. 2001).113Kashiwabara et al. (2013) showed that W adsorbs as distorted, octahedral coordinated inner-114sphere surface complex on ferrihydrite, goethite and hematite Besides the structural characteristics of the host phases, the strong Jahn-Teller distortion of the WO₆-octahedron (Hsin et al. 2008) may also affect the structural incorporation of W⁶⁺. For example, in the tungstite structure, the highly asymmetric WO₆-octahedron displays one short W=O bond of 1.69 Å, four intermediate W-O bonds with 1.83 Å and 1.93 Å, as well as one long bond distance of 2.34 Å (Szymański and Roberts 1984).

Although homovalent substitutions of Fe^{3+} by trivalent ions were extensively investigated 120 (cf. summary in Cornell and Schwertmann 2003), heterovalent substitutions like $Fe^{3+} \rightarrow W^{6+}$ 121 122 received significantly less attention (e.g. Sieber et al. 1985; Duff et al. 2002; Scheinost et al. 2006; Nico et al. 2009; Mitsunobu et al. 2010; Bolanz et al. 2013a, 2013b; Marshall et al. 2014; 123 Brinza et al. 2015). The structural incorporation of W^{6+} into hematite and goethite by 124 heterovalent substitution (2 Fe³⁺ \rightarrow W⁶⁺ + ⁰) has already been proposed in previous 125 investigations, the exact modes of incorporation, however, were never investigated by close-126 127 range order techniques like EXAFS (Kolitsch 1998; Tarassov et al. 2002; Ciobanu et al. 2013). In addition, it was not conclusively shown that W is part of the hematite and goethite crystal 128 129 structure and not only present as microinclusions of a different phase or adsorbed to the iron 130 oxide surface. This question of structural incorporation vs. adsorption is particularly important, as adsorbed ions can easily be released by a slight change in pH, while structurally bound elements 131 are immobilized on the long-term. Therefore, in order to decipher the history of a fluid by 132 fluctuations in W content, detailed information about the speciation of W^{6+} in iron oxides is 133 essential and will be addressed in this study. 134

By using a combination of chemical and spectroscopic techniques, the potential substitution of Fe^{3+} by W^{6+} and the exchange and charge balance reactions for this substitution within the hematite and goethite structure will be discussed.

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Samples

Natural specimens of tungsten-rich botryoidal hematite and goethite were obtained from 139 140 various iron ore deposits from the Schwarzwald ore district, Germany and the Grantcharitza tungsten ore deposit, Bulgaria. Samples originating from the Schwarzwald are mostly 141 uncorrelated to W-mineralization and were denoted with the abbreviation BF (Black Forest = 142 143 english for Schwarzwald). Samples from Grantcharitza come from the weathering zone of a Wdeposit and are denoted with GR. The investigation of these natural specimens is complemented 144 by an investigation of precisely synthesized iron oxides with the denotation SYN. For all 145 synthesized samples and hematite from sample BF-26, powdered specimens with different W-146 concentrations were prepared to correlate refined lattice parameters and their chemistry (cf. Table 147 1). In the natural samples of the GR deposit and in BF-26, hematite and goethite occur intimately 148 intergrown with each other, while the sample BF-11 consists of hematite and BF-20 of goethite 149 150 exclusively (cf. Table 2).

Although different physicochemical conditions prevailed during their formation, textural features of all natural samples and the method of the synthesis suggest that all hematite and goethite samples developed from ferrihydrite as precursor phase

The natural samples represent secondary assemblages of primary Fe- or W-mineralizations. Tarassov et al. (2002) already described the GR samples; text and Fig. B1 in Appendix B1 describe the BF samples and include information about the formation ages, tectonic settings, primary mineral assemblage, and linkage to minor W-mineralizations of both the GR and BF samples.

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Methods

Synthesis of W-bearing hematite and goethite (SYN samples)

The precursor for all synthetic W-hematite samples was two-line ferrihydrite $(Fe_{10}^{3+}O_{14}(OH)_2;$ Michel et al. 2007), synthesized according to Schwertmann and Cornell (2000). 162 Appendix B2 reports the preparation of suspended ferrihydrite used for the synthesis of hematite 163 (pH 4) and goethite (pH 12). For W-bearing goethite and hematite, the respective suspensions 164 were spiked with different concentrations of sodium tungstate (Na_2WO_4), calculated to arrive at 165 molar W/Fe ratios of 0, 0.05, 0.1, 0.2, 0.4, 0.8, and 1.6. 166 Subsequently, samples for hematite synthesis were filtered and washed with 500 mL of 167 deionized water, air-dried, and heated in an oven for 72 h at 500 °C. After the transformation of 168 ferrihydrite to hematite, the samples were finely ground and washed for about 15 minutes with 169 HCl (15 %) in order to remove remaining ferrihydrite. Afterwards, the hematite samples were 170 washed with deionized water for HCl removal, filtrated, air-dried, and sealed in glass vials. 171 172 For the synthesis of W-bearing goethite, the ferrihydrite slurry was split into two equal portions, the volume of the slurry was adjusted to 250 mL each, and filled in 250 mL 173 polyethylene bottles. One aliquot was aged in an oven at 70 °C for 60 days, whereas the other 174 was aged at 25 °C for the same time. In order to remove remaining ferrihydrite, all goethite 175 samples were washed with 250 mL of a 0.2 M ammonium oxalate $((NH_4)_2C_2O_4)$ solution with a 176 177 pH of 3 (Schwertmann 1964). The pH of the ammonium oxalate solution was adjusted with a 1 M HNO₃. After the ferrihydrite extraction, all samples were rinsed with deionized water, filtrated, 178 air-dried, and sealed in glass vials. Denotations, corresponding to the initial molar W/Fe ratios, 179 are SYN-hem_{0.00} to SYN-hem_{1.6} for the hematite samples, SYN-goe_{0.00:25°C} to SYN-goe_{1.6:25°C} of 180

- the goethite samples aged at 25 °C and SYN-goe_{0.00[:]70°C} to SYN-goe_{1.6[:]70°C} for samples aged at 70 181
- °C accordingly. 182

183 Electron microprobe analysis (EMPA)

Elemental composition of the natural and synthetic samples was determined using a JEOL 184 185 JXA-8900RL electron microprobe in wavelength-dispersive mode at the Fachbereich 186 Geowissenschaften of the University of Tübingen, Germany; quantitative element distribution maps of W were additionally obtained on a JEOL JXA-8230 at the Institute of Geosciences, 187 188 Friedrich-Schiller-University Jena, Germany. Natural and synthetic standards were used for calibration, the exact specifications can be found in Table A1 of Appendix A. For focused beam 189 measurements, a beam current of 20 nA and an acceleration voltage of 20 kV were used. For 190 191 element maps, a beam current of 50 nA, an acceleration voltage of 20 kV, and a dwell time of 192 300 ms/pixel were used to reach comparable count rates.

Although most samples showed no problems during the WDX analyses, the synthetic iron oxides showed low totals due to their high porosity. The chemical composition of all synthetic samples was therefore additionally investigated by total reflection X-ray fluorescence spectrometry (TXRF) (see below).

Table 1 shows the EMPA data of the synthetic and natural powder samples, which were also used for refinement of crystallographic parameters; Table 2 shows the EMPA data of the natural hematite and goethite samples BF and GR, respectively.

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Total reflection X-ray fluorescence (TXRF)

Concentrations of W and Fe in the synthetic samples and the natural sample BF-26 were additionally analyzed by total reflection X-ray fluorescence using a TXRF Bruker S2 PICOFOX at the Fachbereich Geowissenschaften of the University of Tübingen, Germany, equipped with a Mo-anode operating at 50 kV and 600 μ A. This method was chosen to confirm the EMPA analyses of the porous samples. To compare refined crystallographic parameters with the W- 206 concentrations of the respective samples, the same samples were also used for powder X-ray207 diffraction measurements (next chapter).

208 Parts of the natural hematite sample BF-26 showing different W concentrations with EMPA were measured as solids to avoid undesired precipitates during chemical extraction. 209 About 2 mg of powdered sample and 20 μ L of a 100 μ g/g Sr-solution as internal standard were 210 211 dispersed with 1 mL Triton X-100 solution and dried for 5 min at 70 °C on quartz discs. The reliability of this method has been shown by Cantaluppi et al. (2013). For each zone three 212 aliquots were prepared and analyzed with a measurement time of 300 s, respectively. The 213 214 observed data of the different zones are listed and compared to refined lattice parameters in Table 1. 215

According to the initial W/Fe molar solution ratios during the iron oxide synthesis, about 1-216 20 mg powder was digested in 15 mL HCl (32%) and 10 mL MilliQ water to reach comparable 217 218 count rates, respectively. At these low W concentrations in HCl solution, no precipitation was 219 observed. To determine the W-concentration, 450 µL of the undiluted sample solutions were 220 mixed with 50 μ L of a 5 μ g/g Ga-solution as internal standard. For Fe determination the sample solutions were diluted 1:10 with MilliQ water to the chloride concentration (Stosnach, 2010), 221 222 which could cause coagulation of crystallites, resulting in additional disturbing background 223 effects (De La Calle et al., 2013). A volume of 400 μ L of these diluted solutions were mixed with 100 µL of a 100 µg/g Ga-solution as internal standard. All solutions were dried for 5 min at 70 224 °C on quartz discs and measured for 600 s each. The spectra were evaluated with the Bruker 225 Spectra 6.2.0.0 software. 226

For all synthetic W-bearing samples, the molar loss of Fe for each mole of W can be calculated using Eq. 1:

229 Eq. 1:
$$\frac{Fe - loss}{W} = \frac{(Fe_{wt}^{W_0} - Fe_{wt}^{W_\chi}) \cdot M_W}{W_{wt}^{W_\chi} - K_{e}}$$

where $Fe_{wt\%}^{W_0}$ is the Fe content of the W-free goethite or hematite, $Fe_{wt\%}^{W_X}$ is the Fe content of W-bearing goethite or hematite, M_W and M_{Fe} are the molecular weights of W and Fe and $w_{wt\%}^{W_X}$ is the W content of W-bearing goethite or hematite. All Fe and W concentrations need to be corrected for the measurement total. The obtained chemical data of the synthetics are listed and compared to the Fe-loss and refined lattice parameters in Table 1.

For comparing the W concentrations between hematite/goethite and between synthetic/natural samples, the molar fraction of W in hematite and goethite $x_W^{hematite / goethite}$ was calculated using Eq. 2:

238 Eq. 2:
$$\mathbf{x}_{W}^{hematite / goethite} = \left(\frac{n_{W}}{n_{W} + \sum n_{i}}\right)_{hematite / goethite}$$

where n_W stands for the moles of W and $\sum n_i$ for the sum of moles of all other elements present within hematite or goethite. For the synthetic samples, $\sum n_i$ corresponds to Fe only, for the natural samples to the elements incorporated besides W according to Table 2.

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Powder X-ray diffraction (pXRD)

For phase identification and crystallographic parameters, powder X-ray diffraction (pXRD) patterns of all synthetic samples, the natural sample BF-26, and the reference compounds were collected with a Bruker D8 AXS Advance diffractometer with Cu K α radiation ($\lambda = 1.54058$ Å) at the Friedrich-Schiller-University Jena, Germany. The patterns were collected at 20 °C between 5-130 °20 with a step size of 0.02 °20 and a dwell time of 1 second. For refinement, the patterns of all samples were processed with the program TOPAS®, where a fitting range between 15-130 °20 was chosen. For each fit, the statistical measures from the obtained data are given including

the residuals for the weighted profile (R_{wp}) and the goodness-of-fit (GOF) (see Table 1). For sample BF-26, predefined regions of the botryoidal hematite, previously investigated by EMPA, were ground in an agate mortar (cf. Table 1; sample aliquots BF-26 I-VII). However, due to the small sample size and volume, all other goethite samples had to be analyzed by micro X-ray diffraction (see next chapter).

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Micro-resolved X-ray diffraction (µXRD)

256 Due to dilution effects during the preparation of the natural powder samples, resulting in low W contents, additional micro X-ray diffraction analyses were performed in regions of the 257 thin sections with the highest W concentrations to exclude a potential intergrowth of discrete W-258 bearing impurities. For the regions of interest, the spots with maximum W concentration, 259 previously determined by EMPA, were chosen. All uXRD measurements were performed on thin 260 261 sections of sample BF-11, BF-20, and BF-26 with a Bruker D8 Discover GADDS theta-theta micro-diffractometer at the Fachbereich Geowissenschaften, Universität Tübingen, Germany. To 262 prevent iron fluorescence, a Co-anode with K α radiation ($\lambda = 1.79$ Å) was used at 30 kV and 30 263 mA with a HOPG monochromator and monocapilarry optics of 500 µm with a 200 µm pinhole. 264 The incident angle to the samples was fixed to 10°. For the general area detector diffraction 265 system (GADDS) a Bruker VÅNTEC500 detector was used and the patterns were integrated 266 between 5-68 °2O. Selected µXRD-patterns can be found in Fig. A1 of Appendix A. 267

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Micro-Raman spectroscopy (µRS)

 μ RS was preformed to determine the purity of the natural and synthetic hematite/goethite samples as well as to identify the possible effects of the structural incorporation of W on the vibrational modes of both phases. In addition, μ RS was used to distinguish between the two possible charge balance mechanisms, Fe³⁺-vacancy formation or oxygen (de)protonation by H⁺. 273 For the measurements, a confocal Raman spectrometer Renishaw InVia Reflex with a laser wavelength of 532 nm was used at the Fachbereich Geowissenschaften of the University of 274 275 Tübingen, Germany. A laser power of ~ 0.2 mW at the sample surface was used to avoid thermal degradation of the samples. The significant part of the recorded Raman spectra were 276 characterized by strong fluorescence signals. Therefore, the recorded spectra were baseline 277 278 corrected by polynomial and linear fits for full spectra and detail spectra, respectively. Mixed Gaussian and Lorentzian curves were used for the deconvolution of the peak fits. Measurements, 279 baseline subtraction and peak fits performed by using the Renishaw software package WiRE 3.0. 280

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X-ray absorption near edge structure (XANES) spectroscopy

X-ray absorption near edge (XANES) spectra were measured at the W L_1 and L_3 edges at 282 the SUL-X beamline of the Angströmquelle Karlsruhe (ANKA, Karslruhe, Germany) using a 283 fixed exit double crystal monochromator. W L_3 edge XANES spectra were additionally recorded 284 for the natural hematite sample BF-26 using the Si(311) monochromator crystal pair or 285 286 decreasing of the slit 4 (intermediate focus) width at Si(111) monochromator crystals to 287 investigate the effect of energy resolution of spectral features (for details see Appendix B3). 288 Additionally, XANES spectra at the Fe K-edge were measured using the Si(111) monochromator crystal pair. 289

Beam sizes for XANES measurements on the thin sections were about 100 μ m x 100 μ m at sample position to ensure data recording of the area of interest. Only for the tests with higher energy resolution using the Si(111) monochromator crystals, a smaller beam size of 50 μ m x 50 μ m was required.

W L_3 - and Fe *K*-edge spectra were measured on natural and synthetic powder samples (BF-26, BF-20, SYN-goe_{0.05;25°C}, SYN-goe_{0.05;70°C}, SYN-goe_{1.6;70°C}, SYN-hem_{0.05} and SYN-hem_{1.6}) in transmission and fluorescence mode on cellulose pellets and in fluorescence mode only on the

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thin sections (BF-26 and BF-20). W L_1 -edge spectra were measured except for W-rich hematite BF-26 and (due to low W concentration in natural goethites) for the synthetic sample SYN_{goe0.05;70°C} (cf. Appendix B3).

Reference spectra at the W L_3 and W L_1 edge were measured for Na₂WO₄·2H₂O and CaWO₄ (scheelite) representing W in tetrahedral O coordination as well as (Fe,Mn)WO₄ (wolframite) and MnWO₄ (hübnerite) where W⁶⁺ is in octahedral coordination.

Energy calibrations, pre- and post-edge background corrections of W L_3 , W L_1 and Fe Kspectra, the used transmission and fluorescence methods, energy resolution, self absorption correction of the Fe K spectra as well as the extraction of the W L₃-edge spectra energy gap and the W L1 pre-edge area to identify WO₆ octahedra distortions after Yamazoe et al. (2008) are reported in Appendix B3.

308 Extended X-ray absorption fine-structure (EXAFS) spectroscopy

309 For all natural and synthetic samples, extended X-ray absorption fine-structure (EXAFS) 310 spectra were collected at the W L₃ edge at the SUL-X beamline of the Angströmquelle Karlsruhe (ANKA, Karlsruhe, Germany) using a Si(111) crystal pair with a fixed beam exit as 311 312 monochromator. The appropriate mass of each sample and reference compound, calculated with the program XAFSmass (Klementiev 2012), was mixed with cellulose powder, homogenized, 313 and pressed to a pellet. Measurement parameters of the EXAFS region are presented in Appendix 314 315 B3. The spectra were used for the EXAFS evaluation up to 12.8k (640.8 eV above the absorption 316 edge).

Samples measured as thin sections were collected in fluorescence mode, whereas all samples and reference compounds prepared as pellets were additionally measured in transmission mode (cf. chapter above). Depending on the quality of the measurement, either the fluorescence or transmission signal was chosen. All EXAFS data were processed and fit by the Athena and Artemis software package (Ravel and Newville 2005). In order to reduce the noise in higher k-range, the spectra of all synthetic goethite samples were merged and denoted SYN-goe_{0.05-1.6;70°C}. Equally, all spectra of synthetic hematite were merged and denoted SYN-hem_{0.05-1.6}. The paths, generated for the fitting, were calculated from modified models of hematite (Blake et al. 1966) and goethite (Gualtieri and Venturelli 1999) for all paths. The paths were generated using the software program FEFF6 (de Leon et al. 1991), embedded in the Artemis software package.

For each EXAFS fit, the statistical measures of the deviation of the fit from the obtained data are given, including the number of variables allowed to float in the fit (N_{var}), the number of independent data points (N_{idp}), as well as the R-factor. The N_{idp} is equal to $2\Delta kR/\pi$ where k and R are the fitting ranges. The R factor is defined by Eq. 3,

332 Eq. 3:
$$R = \frac{\sum (x_i^{data} - x_i^{mod \ el}(x))^2}{\sum (x_i^{data})^2}$$

where χ is the magnitude of the EXAFS oscillations and x is the set of variables to be refined. EXAFS spectra of the individual reference compounds were fit with the respective structures in order to identify problems that could be encountered when fitting the model of the structurally unknown samples. Additional information about the fitting procedure can be found in Appendix B5.

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Results

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9 Homogeneity and purity of the natural samples

Botryoidal hematites and goethites show a high porosity (see Fig. 1a, c and e). Thus, distinct W-phases could have formed after iron mineral formation, due to later W-rich fluid infiltration. To avoid any possible misinterpretation of our data, the absence of distinct W-phases, the homogeneity and purity of the natural samples was tested by EMPA element distribution maps, 2D-resolved μ XRD and μ RS on regions with the highest W concentrations of the BF samples.

The results of the μ XRD for the BF samples did not show any other diffraction reflexes besides those of hematite or goethite at their highest W concentrations of 1.1-1.8 wt% and 0.3 wt%, respectively (Fig. A1). μ RS of the same spots show bands very similar to those observed for pure, synthesized α -Fe₂O₃ and α -FeOOH by De Faria et al. (1997) and Bersani et al. (1999). Only the synthetic goethites show an additional band at 417 cm⁻¹, which has not been published for goethite yet (see chapter below).

The EMPA element distribution maps show, that W is generally homogeneously distributed 352 (Fig. 1). The count rates of the WL α fluorescence (Fig. 1; right side) do not show any correlation 353 to the lowest count rates of the back-scatter electron (BSE) images (Fig 1; left side), which are 354 355 decisive for the porosity and interstitial space between the single crystallites. In general, the $WL\alpha$ fluorescence maps indicate no abnormal W-rich clusters, which would indicate a separate W 356 phase. Only Fig. 1b and d show rhythmic variations of the WL α fluorescence, which can be 357 attributed to different W concentrations within the iron oxides during separate growth stages. E.g. 358 initial, globular aggregates in Fig. 1a and b show much higher W concentrations (green colors; up 359 to ~1.24 wt% W) than the following feather-like aggregates (blue colors; $\leq 0.2 \text{ wt}\% \text{ W}$). 360

The samples from Grantsharitza ($GR_{G-I,G-II,H-I}$) were already analyzed for phase purity in Tarassov et al. (2002), showing hematite, goethite, quartz and microcline reflexes only (their Fig. 1). The measurement of a W-rich phase F_w (cf. Appendix B1), microscopically not identified as hematite or goethite with concentrations of up to ~20±0.26 wt% W was avoided during the XAFS measurements.

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Chemistry of the natural and synthetic W-rich hematites and goethites

The W- and Fe-composition of the powdered synthetic samples and BF-26, quantified by TXRF and EMPA is shown in Table 1. Additionally, the amount of Fe loss for each incorporated W is given for the synthetic samples. Representative chemical compositions of all natural samples, obtained by EMPA are shown in Table 2.

In general, hematite samples from the Schwarzwald exhibit significantly higher concentrations of W compared to their goethite counterparts (Fig. 2b). Hematite (BF-26), completely unrelated to any W-mineralization, exhibits the highest W concentration with 2.16 wt% WO₃ (0.8 mol% W), whereas hematite sample BF-11, correlated with scheelite mineralization, shows only a maximum of 1.61 wt% WO₃ (0.6 mol% W). However, goethite samples from the Schwarzwald exhibit only low W contents with a maximum of 0.33 wt% WO₃ (0.12 mol% W) in sample BF-20.

In contrast, the samples from the Grantcharitza tungsten deposit exhibit their highest concentrations in goethite with WO₃ up to 13 wt% (5.5 mol% W) in sample GR_{G-I} and 5.6 wt% (2.15 mol% W) in hematite GR_{H-I} (Fig. 2b). The synthetic W-doped samples also show higher W concentrations in goethite compared to hematite, reaching maximum WO₃ of about 8.9 wt% (7 mol% W) in goethite and 3.0 wt% (1.2 mol% W) in hematite (Fig. 2a).

According to Fig. 2a, TXRF and EMPA, show comparable values of W only at molar W/Fe ratios ≤ 0.1 within the synthesis solution. The synthetics, which precipitated at higher W/Fe ratios show considerable variations between the TXRF and EMPA methods, whereas hematite shows a higher scatter of the EMPA data. Except for samples SYN-goe_{0.1;70°C} and SYN-goe_{1.6;70°C}, the TXRF data show a general underestimation with respect to the EMPA mean values. Excluding sample SYN-goe_{0.2;70°C} and SYN-hem_{1.6}, TXRF and EMPA represent overlapping values in case of the TXRF analysis relative error and the variability of the EMPA measurements (Fig. 2a, grey shaded areas).

Besides these deviations, however, the mean EMPA values of W in hematite and goethite 391 indicate a linear increase with respect to initially low W concentrations used during the synthesis 392 (Fig. 2a, dotted lines). For both, hematite and goethite, this linearity deviates from the observed 393 data with a further increase of W in the synthesis solution. For goethite, the trend changes to a 394 flatter slope, but continues as a different linear correlation. In contrast, hematite shows an 395 abruptly higher spread with the increase of W in the synthesis fluids at molar W/Fe-ratios ≥ 0.2 . 396 397 These different changes occur earlier for goethite (~25 mmol/L W) than for hematite (~40 mmol/L W), and only concentrations in hematite display a plateau at $\sim 1.2 X_W^{\text{hematite}}$ (Fig. 2a). The 398 goethite sample SYN-goe0.4:70°C shows a much higher W concentration than the illustrated 399 linearity by both EMPA and TXRF, the hematite sample SYN-hem_{1.6} is under-determined only 400 by TXRF. According to the Fe-loss calculations (see Table 1), each W atom replaces ~2.9 Fe 401 402 atoms of the goethite and ~ 2.8 of the hematite structure.

403

Powder X-ray diffraction

Powder X-ray diffraction patterns of the synthetic samples and sample BF-26 only display
reflections characteristic of hematite or goethite, respectively. The refined lattice parameters and
crystallite sizes of the full-profile fits are given in Table 1. It seems obvious that the lattice
parameters do not correlate with the W concentrations for hematite samples (SYN-hem).
However, in the synthetic goethites (SYN-goe), the unit cell parameter *c* and their crystallite sizes

correlate with their W content. With rising W concentration the unit cell parameter c increases as
well as the crystallite size (cf. Fig. A2). Goethite SYN-goe_{0.4;70°C}, grown from solution, does not
follow this trend as well as sample hematite SYN-hem_{1.6}, which shows a crystallite size about 1.5
times higher than the others. These deviations also coincide with their chemistry (cf. section
above).

414

X-ray absorption near-edge spectroscopy

415 **XANES at the W L₃ edge.** The normalized, W $L_3 \mu$ XANES spectra of the natural, W-rich 416 goethite BF-20 and hematite BF-26, as well as two synthetic goethites (SYN-goe_{0.05:25°C}, SYN-417 goe_{0.05;70°C}) and synthetic hematite (SYN-hem_{0.05}) are shown in Fig. 3. The whiteline of the goethite BF-20 spectrum is slightly smaller than the one of hematite BF-26. Similar is valid for 418 419 the spectra of the synthetic samples: The whiteline of goethite SYN-goe_{0.05:25°C} is smaller than the 420 one of hematite (SYN-hem_{0.5}) whereas the whiteline width of goethite SYN-goe_{0.05} is close to 421 the one of hematite. A better energy resolved spectrum is additionally shown for the natural 422 hematite BF-26 (Fig. 3, dotted line of spectra a) and demonstrates the composite nature of the 423 whiteline consisting of two individual peaks respectively at ~ 10211.7 eV and 10215 eV. Spectra 424 with higher energy resolution have not been measured for other samples because the low flux 425 conditions for higher energy resolution measurements were contrary to the beam requirements for 426 collection EXAFS data of good signal to noise ratio. The hematites, the natural goethite BF-20 and the synthetic goethite SYN-goe_{0.05:70°C} show an asymmetric whiteline with lowering of its 427 428 intensity at the higher energy flank. The widths of all whitelines of the sample spectra are 429 unambiguously broader and less sharp than those of the reference compounds CaWO₄ and 430 $Na_2WO_4 \cdot 2H_2O$ (Fig. 3) where W has a tetrahedral coordination. The whitelines shapes of the 431 octahedral coordinated W references MnWO₄ and (Fe,Mn)WO₄ (Fig. 3) are matching better with 432 those of hematite and goethite but asymmetry is absent or weak. The double whiteline features in MnWO₄ and (Fe,Mn)WO₄ are not visible because they were measured with Si(111) 433 434 monochromator crystal pair and wide slit gaps. However, the split character of the W L_3 whiteline has besides the peak deconvolution been proved by the double peak structure of the second 435 derivatives of the XANES spectra (Fig. A3). Crystal orientation effects, as the origin of different 436 437 whiteline shapes, have been excluded because spectra measured on material from BF-20 goethite and BF-26 hematite areas, separated, grinded and pressed to cellulose pellets (resulting in 438 randomly oriented crystals). They show the same spectral features as described for the 439 440 measurements on the polished sections in Fig. 3 where crystallites show a common orientation (Fig. 1c and e). Appendix B4 and Fig. A4.1 report results of the W L₃ whiteline split 441 442 quantification by peak deconvolution of selected samples and reference compounds.

443 **XANES at the W L_1 edge.** Natural hematite (BF-26) and the synthetic W containing 444 goethite (SYN-goe_{0.05.70°C}) has been measured. Sample spectra are compared to the reference compounds CaWO₄, Na₂WO₄·2H₂O, MnWO₄ and (Fe,Mn)WO₄ in Fig. 4. Both W L_1 XANES 445 spectra of hematite BF-26 and goethite SYN-goe_{0.05:70°C} are similar and show a double peak 446 shaped XANES region besides the pre-edge. Both pre-edge peaks are located at ~12107.3 eV. 447 The overall spectral shapes of the sample spectra are closest, among the reference spectra of this 448 449 study, to hübnerite (MnWO₄) and wolframite ((Fe,Mn)WO₄) that represent W in octahedral coordination. The pre-edge peak for goethite is more distinctive than for hematite. The tetrahedral 450 coordinated W of the references CaWO₄ and Na₂WO₄·2H₂O show well developed and high 451 intensity pre-edge peaks at ~12105.1 eV (Fig. 4). Fits of the pre-edge peaks are shown in Fig. 452 A4.2. The XANES region of the sample spectra is characterized by a double peak structure 453 whereas spectra of the reference substances show three peaks well visible for CaWO₄ and 454

455 Na₂WO₄·2H₂O and less pronounced for the wolframite and hübnerite spectra. Peaks of the 456 reference spectra are assigned with A (pre-edge) and B, C, D for the XANES region (Fig. 4).

457 **XANES at the Fe K edge.** After self absorption correction the Fe K XANES spectra of 458 natural goethite (BF-20) and hematite (BF-11) measured on thin sections are compared to a 459 goethite and hematite reference spectra (Fig. A5.1 and A5.2). The spectrum of goethite BF-20 460 matches well with the goethite reference spectrum, the one of hematite BF-11 with the hematite 461 reference.

462 Extended X-ray absorption fine structure spectroscopy at the W L₃ edge

EXAFS measurements were performed for detailed information about the local 463 environment of W⁶⁺ in W-bearing goethite and hematite. All Fourier-transformed EXAFS spectra 464 of W-bearing goethite and hematite are shown in Fig. 5a and 5b, respectively. Appendix B5 465 summarizes the fitting procedure used for the W-O and W-Fe paths for both phases, W-bearing 466 goethite and hematite, which were fit with the respective structures (Gualtieri and Venturelli, 467 1999; Blake et al., 1966), in which one Fe was replaced by W. The exact W-O and W-Fe 468 distances obtained from the fit for the different models, including Δr and σ^2 for all paths, are 469 given in Table 3.1 (goethite) and 3.2 (hematite). 470

For the goethites' first and second W-O path, the absorber atom (W) is surrounded by six oxygen atoms with W-O distance between 1.774 ± 0.009 and 2.10 ± 0.01 Å for the natural, and 1.781 ± 0.001 to 2.12 ± 0.01 Å (all corrected for phase shift) for the synthetic samples. This significant distortion of the WO₆-octahedron is consistent with our observation in the near-edge region of the X-ray absorption spectrum and was reported in previous investigations (Cid-Dresdner and Escobar 1968; Macavei and Schulz 1993).

477 Beyond the first six oxygen atoms, goethite shows a relatively broad and split feature at 478 about 2.3-3.6 Å (uncorrected for phase shift) (see Fig. 5a). The splitting of this feature can be attributed to the difference in distance between the edge (2 Fe at \sim 3.01 and 2 Fe at \sim 3.29 Å) and corner-sharing (4 Fe at 3.43 Å) Fe atoms adjacent to the absorber. Typically, for pure goethite measured at the Fe *K* edge, the first segment of the split feature at about 2.6 Å (uncorrected for phase shift) displays higher amplitude compared to the second segment at about 3.2 Å (uncorrected for phase shift) (Jiang et al. 2013).

In the case of our W-bearing goethite, however, this proportion is inverse. The first segment possesses significantly less amplitude compared to the second (see Fig. 5a). This decrease in amplitude, best visible for sample SYN-goe_{0.05-1.6;70°C} and GR_{G-I}, suggests the formation of Fe vacancies close to the absorber and supports our chemical analyses. Accordingly, during the fitting of sample SYN-goe_{0.05-1.6;70°C} and GR_{G-I}, the removal of one of the closest adjacent Fe atoms (typically located at 3.01 Å) improved the fit significantly, while the best fit was achieved by removing one additional Fe atom edge-sharing to W (located at 3.29 Å).

For the hematites' first and second W-O path, the absorber atom (W) is also coordinated by six O atoms with unequal W-O distances, like observed for goethite, with 1.822 ± 0.004 to 2.100 ± 0.008 Å for the natural, and 1.845 ± 0.004 to 2.087 ± 0.009 Å (both corrected for phase shift) for the synthetic samples.

In the hematite structure, $Fe^{3+}O_6^{9-}$ -octahedra are either connected by face-, edge-, or corner-495 sharing. Since no natural or synthetic compound with face-sharing W^{6+} or $Fe^{3+}-W^{6+}$ -octahedra is 496 known, the closest face-sharing Fe^{3+} -atom, next to the W⁶⁺-octahedron, was removed from the 497 model, which significantly improved the fit. According to our chemical data, W replaces about 3 498 Fe atoms. During the fitting, we therefore included and excluded one additional Fe atom adjacent 499 to W, typically located in a distance of about 2.97 Å from the substituted Fe^{3+} (Blake et al. 1966). 500 Similarly to goethite, the best fit could be achieved by removing two Fe atoms adjacent to W, 501 typically located at about 2.90 and 2.97 Å (Blake et al. 1966). However, the removed Fe atom at 502

503 2.97 Å belongs to a group of 3 Fe which all share the same distance to W. The loss of amplitude 504 by removing one of these 3 Fe atoms can therefore easily be compensated by decreasing the 505 value for σ^2 . In conclusion, the formation of one Fe vacancy close to W is strongly supported by 506 the results of the EXAFS fitting, but the formation of a second Fe vacancy in hematite cannot be 507 confirmed or ruled out with absolute certainty.

508 Raman spectroscopy

Raman spectroscopy was performed for the synthetic goethite samples to obtain information about possible changes in the Fe-O and O-H vibrations and to infer the W exchange mechanism, as deprotonation or Fe-vacancy formation within goethite. Raman spectra of all synthetic goethite samples are given in Fig. 6a-e.

All spectra show characteristic bands of goethite published by Cornell and Schwertmann (2003); only one single Raman-active band at ~417 cm⁻¹ occurs which has not already been published for goethite (see Fig. 6d)

The Raman spectrum of the W-free sample (SYN-goe_{$0.0,70^{\circ}C$}) consists of a series of narrow, well-resolved peaks at wavelengths of 247, 300, 482, 551, 640 (weak), 683, 1000, 1110 cm⁻¹ and several very broad bands centered at about 1275 and 1454 cm⁻¹ (Fig. 6a). In the range 330-450 cm⁻¹ are well-resolved peaks at 387 (strongest peak), 401 (shoulder) and 417 cm⁻¹ (Fig. 6d).

With increasing tungsten content, all peaks become broader, the major peak at 387 slightly shifts to 392 cm⁻¹ and the intensities and asymmetries of the bands in the range 580-780 cm⁻¹ and in the very broad band at 1200-1400 cm⁻¹ slightly increase (Fig. 6a; fits in Fig. 6b-e). In Fig. 6b three bands can be distinguished for the 1200-1400 cm⁻¹ region at 1350, 1310 and 1255 cm⁻¹ of the SYN-goe_{0.0;70°C} spectrum. Besides the nonsystematic bands at 1350 and 1255 cm⁻¹ the third one's intensity relatively rises and shifts to slightly higher wavenumbers. Besides the 689 cm⁻¹ band of W-free goethite in Fig. 6c, additional bands at 612 and 657 cm⁻¹ occur and increase with

Page 23

527	increasing W. Raman spectra of natural W-enriched goethites (GR _{G-I} and GR _{G-II} ; Fig. A6) agree
528	well with the spectrum of the artificial sample SYN-goe _{1.6;70°C} with the highest W content.
529	The spectra of synthetic hematites with different W contents show no significant
530	differences in the positions, intensity and FWHM of the peaks (Fig. A7). These spectra nicely
531	resemble the spectrum of natural hematite GR_{H-1} , besides a shoulder at ~700 cm ⁻¹ seen in the
532	spectrum of the natural material (Fig. A6). However, band fits of the 550-750 cm ⁻¹ region in Fig.
533	A7 show this \sim 700 cm ⁻¹ band and also show that the 660 cm ⁻¹ band intensity rises relatively to
534	the 612 cm ⁻¹ band at higher W concentrations.

536

Discussion

537

Chemistry of W-bearing hematite and goethite

W-bearing hematites and goethites show different chemical compositions according to their 538 geo-environmental appearance. Samples from an environment without W mineralization show 539 higher W concentrations in hematite than in goethite, even if they formed paragenetically (see 540 samples BF-26; Fig. 2b and Table 2). Also, in the presence of minor W-mineralization (see 541 542 sample BF-11; Fig. 2b and Table 2), W is preferably associated with hematite rather than goethite. On the contrary, synthetic iron oxides and GR samples formed in the constant presence 543 of high W. They display an opposite trend, where goethite incorporates significantly more W (up 544 545 to 5.5 mol%) than hematite.

Both the natural and synthetic formation of hematite and goethite can be linked to the 546 precursor phase ferrihydrite (Cudennec and Lecerf 2006). If both natural sample types formed 547 around neutral pH, the adsorption of W to ferrihydrite can be assumed to be of similar magnitude 548 (Gustafsson 2003). Hence, the opposing behavior in W-poor (BF samples) and W-rich (GR and 549 550 SYN samples) environments cannot be explained by the initial adsorption of W to the precursor 551 phase ferrihydrite. A possible explanation is the difference in the transformation processes of ferrihydrite to hematite or goethite. According to Cudennec and Lecerf (2006) hematite forms 552 from ferrihydrite by a topotactic solidified transformation simply accompanied by dehydration in 553 554 the gel during the ageing process, whereas goethite forms via a dissolution and reprecipitation 555 process.

Goethite, forming in an open system for W, could not retain all earlier adsorbed or absorbed W and hence, tungsten could get lost during the dissolution of ferrihydrite and reprecipitation of goethite. Analogous to octahedrally coordinated Fe of the hematite structure, ferrihydrite has already adsorbed hydrated inner sphere surface W-complexes in distorted

Page 25

octahedral coordination prior to its transformation to W-hematite (Gustafsson 2003; Kashiwabara et al. 2013). The formation of W-goethite directly from the fluid is more complex because W predominantly occurs as tetrahedral coordinated WO_4^{2-} and its coordination number has to be increased first (Takeno 2005). This requires a two times protonation of this species (Cruywagen 2000).

If the system is closed for W or a rapid, high and constant supply of W exists, goethite crystallites are able to incorporate high amounts of W. Hence, if the concentration of W in the transformation fluid is high enough, goethite could incorporate the high W concentrations in situ, whereas hematite would only retain the previously adsorbed W amount from ferrihydrite.

The different slopes of X_w^{hematite/goethite} with respect to increasing molar concentration of W 569 within the synthesis fluids (see Fig. 2a) can be explained, since at low ratios the solution is 570 diluted within dimensions where the Henry's law can be assumed and the phase stability is not 571 affected (Mysen 1978). Accordingly, concentrations below ~25 mmol/L W in goethite and below 572 \sim 40 mmol/L W in hematite indicate thermodynamically stable solid solutions. The flattening of 573 574 the slope for goethite and the large variance for hematite at higher W concentrations indicate the deviation from an ideal solution behavior, where the scope of Henry's law is not applicable. We 575 suggest that this disequilibrium between W supply and incorporation is an important pre-requisite 576 577 for the formation of distinct W-phases.

Fluids which strongly vary in W content precipitate ferrihydrite with variable W concentrations. Over time, these newly formed ferrihydrite crusts, which adsorbed or incorporated W from their formation solution, will transform to hematite with distinct domains, enriched or depleted in W. In contrast, during goethite formation, W previously associated with this ferrihydrite, can be homogeneously redistributed during the dissolution-reprecipitation process and the new goethite develops a homogeneous W distribution.

584 It can also be speculated, that the formation pathway has a direct control on the maximum W content in hematite and goethite. In our experiments, hematite reached a limit of $\sim 1.2 X_w^{hematite}$ 585 586 in the synthetic samples, even at concentrations of 50 mmol/L W and higher (Fig. 2a). W concentrations in hematite are therefore limited by the W previously associated with ferrihydrite, 587 even if a later W-enriched solution would arrive. W concentration in goethite, on the other hand, 588 589 is, to a certain extent, correlated to the concentration of W in solution. Obviously, due to the formation of Fe-vacancies, the W content in goethite is limited by the stability of the goethite 590 structure to the increasing number of vacancies. 591

Based on the phenomena discussed above, the secondary oxi/hydroxides hematite and 592 goethite can track the fluid history of W in W-poor geo-environments. If the synthetic 593 594 experiments (Fig. 2a) are applied to the mean W contents of the Schwarzwald samples (BF, Fig. 2b), the hematite sample BF-26 formed in the presence of solutions with 21 mmol/L W, sample 595 BF-11 at 7 mmol/L W, and the goethites BF-26 at 0.1 mmol/L W, BF-20 at 0.4 mmol/L W. For 596 597 the sample BF-26 this would indicate that this goethite formed at significantly lower W 598 concentrations than the hematite. If the results of the synthetic iron oxides (Fig. 2a) are applied to the mean W contents of the samples from Grantcharitza (GR, Fig. 2b), they would correspond to 599 W concentrations within fluids of ~200 mmol/L W for goethite GR_{G-I}, 8 mmol/L W for goethite 600 GR_{G-II} and >200 mmol/L W for hematite GR_{H-I} . This implies that goethite GR_{G-I} and hematite 601 602 GR_{H-I} originate from fluids with similar W concentrations. Goethite GR_{G-II}, however, crystallized at significantly lower concentrations, although showing the same mean W content as hematite. 603 The presence of two different fluids is also supported by microscopic observations of the phases 604 (cf. Appendix B1). 605

Because of the large variability of W in hematites grown from synthesis solutions with >20
 mmol/L W, the interpretation of fluid history should be reliable for hematites containing less than

0.3-0.4 mol% W (i.e. grown in fluids <20 mmol/L W). The interpretation for goethites should be
possible for much higher W concentrations, such as the ones reached by our synthesis
experiments, but their use is restricted because of potential W redistribution during the
ferrihydrite-goethite transformation.

The synthesis experiments showed that naturally occurring hematites and goethites from the Schwarzwald and the goethites from Bulgaria could easily have incorporated their highest Wconcentrations due to former precipitation of ferrihydrite. The hematites from Bulgaria, however, exceed the maximum possible W incorporation revealed by our synthesis experiments, an observation, for which we lack an explanation.

617 Crystallographic parameters of W-doped hematite and goethite

The increase of W in goethite causes an increase of the cell parameter c (cf. Table 1 and Fig. A2) which happens, if the structural disorder becomes higher (Cornell and Schwertmann 2003). Increasing structural disorder is also shown by our μ RS data (Fig. 6). This result is unequivocal evidence that the tungsten atoms are incorporated into the structure of goethite.

For hematite, there are no obvious correlations between cell parameters and W content (Table 1), either, because of low concentrations or because of the similarity of the ionic radii of Fe^{3+} and W^{6+} (Shannon 1976). Also Kolitsch (1998) showed that an incorporation of relatively large amounts of W is possible while no significant changes of the unit-cell parameters are caused.

627 XANES W L₁

XANES W L₁ and W L₃ spectra

The nature of the W octahedra can be inferred from the pre-edge intensity from the W L_1 edge spectra and the splitting of the whiteline in the W L_3 edge spectra, shown by Yamazoe et al. (2008). For a detailed description of the phenomena causing different splits and pre-edge shapes in Fig. A8 see Appendix B6. In Fig. A8 the positions of hübnerite (C), wolframite (D), the synthetic goethite SYN-goe_{0.05;70°C} and natural sample hematite BF-26 are characterized by a distorted octahedral structure because they are located about halfway between regular octahedral (lower right corner; compounds f and g) and tetrahedral coordination (upper left corner; compounds a and b) (cf. Yamazoe et al. 2008).

The W L₁ pre-edge area and the W L₃ whiteline split values of natural hematite BF-26 and goethite SYN-goe_{0.05;70°C} are located close to each other. Hence, the distortion of their octahedra should be similar. The natural goethite sample BF-20 is expected to possess a somewhat higher octahedral distortion due to a ~0.5 eV smaller W L₃ whiteline split. For a detailed explanation and temperature effects accounting the distortion of the WO₆ octahedra, see Appendix B6.

641 XANES Fe K spectra

Self absorption-corrected Fe K XANES spectra are almost identical to a pure goethite reference spectrum for sample BF-20 and to a pure hematite reference spectrum for sample BF-11 (Fig. A5.1 and A5.2). A Fe K XANES spectrum of wolframite where all Fe is Fe^{2+} is also shown in Fig. A5.1 and A5.2 to demonstrate the edge shift for a bare Fe^{2+} reference. If per W⁶⁺ three Fe^{3+} are reduced to three Fe^{2+} , such a fraction is potentially too small to cause a recognizable edge shift in the Fe K-edge towards lower energy. Thus, an appreciable edge shift is linked to very high W concentrations as it is in goethite samples SYN-goe_{1.670°C} and GR_{G-I}.

649

Raman spectroscopy

According to group theory analysis, goethite has 24 Raman active modes (Rousseau et al. 1981). 18 modes of vibration of O and Fe (FeO₆ octahedron modes) fall into the spectral range below 700 cm⁻¹. The remaining 6 modes (for H) are attributed to different vibrations of OHgroups. In analogy with the isostructural diaspore (AlOOH), at least 8 modes of OH bending vibrations can be expected in the range 1000-1300 cm⁻¹ (Demichelis et al. 2007). Hence, we interpret the 6 bands from ~ 1000 cm⁻¹ to ~ 1350 cm⁻¹ in Fig. 6b as the corresponding OH bending vibrations of goethite.

The presence of tungsten in the structure of goethite increases the structural disorder in 657 goethite visible in the Raman spectra as broadening/displacement of the all peaks below 700 cm⁻¹ 658 (Fig. 6a,c and d) as well as in the formation of poorly resolved bands at ~880 cm⁻¹ (Fig. 6a; 659 marked by arrows) and in the 150-275 cm⁻¹ region (Fig. 6e). An increasing content of tungsten in 660 Fig. 6a does not indicate a decrease of the intensity of Raman bands in the interval 1100-1400 661 cm^{-1} . Moreover, the relative intensity of this spectral interval, especially for the ~1310 cm⁻¹ band, 662 is relatively higher when the tungsten content is larger (Fig. 6b). This means that a protonation 663 mechanism for the structural incorporation of W into goethite by H⁺ seems to be more probable 664 than deprotonation. 665

The situation with hematite is rather different. The presence of tungsten in the hematite 666 structure is not well visible in the Raman spectra due to the masking effect of other factors. High 667 concentrations of vacancies in the positions of Fe atoms is an inherent feature of hematite formed 668 under natural and laboratory conditions during topotactical transformation of ferrihydrite to 669 hematite. In the Raman spectrum of hematite, the defect structure of the Fe-sublattice is 670 manifested by an additional peak at about 660 cm⁻¹, besides the others allowed by group theory 671 672 analysis (Bersani et al. 1999). A similar effect on Raman spectra is expected by the structural incorporation of tungsten that increases the number of vacancies in the position of Fe. According 673 to Fig. A7 this is observed at increased W incorporation, but a clear correlation with increasing 674 W concentrations is at first glance missing. However, Chernyshova et el. (2007) showed a 675 particle size effect. If samples with identical particle sizes are compared (SYN-hem_{0.00} to SYN-676 hem_{0.4}; both 22 nm; and SYN-hem_{0.05} to SYN-hem_{0.8}; both 29 nm; Table 1) a positive correlation 677

between the W concentration and the 660 cm⁻¹ band intensity emerges. Obviously, the incorporation of W into hematite enhances the Fe-sublattice distortion by the production of vacancies.

681 Mechanisms for the structural incorporation of W^{6+} in goethite and hematite

Goethite is built by double-chains of FeO_6 -ocahedra connected by edge-sharing running parallel to *b*. Four double-chains, connected by corner sharing, form a pseudo-tunnel structure filled with hydrogen atoms for charge balance. The substitution of Fe^{3+} by W^{6+} creates three positive excess charges, which have to be compensated.

Obviously, the release of three hydrogen atoms could compensate the excess charge. At the 686 highest W concentration in goethite GR_{G-I}, however, the hydrogen content would then be reduced 687 significantly. Analogous to the structural incorporation of Sb⁵⁺, this deprotonation mechanism 688 689 only works for low concentrations of highly charged cations (Bolanz et al. 2013b). If the Sb concentration rises, the goethite structure would probably collapse due to missing hydrogen 690 bonds. In analogy, temperature-induced dehydroxylation experiments show a breakdown of the 691 692 goethite structure long before the depletion of the hydroxyl groups (Gualtieri and Venturelli 1999). 693

Based on our chemical and spectroscopic analyses, the formation of Fe^{3+} vacancies seems 694 to be the most likely balance reaction. This would also reduce the number of edge-sharing Fe-695 octahedra with highly charged W^{6+} . In all natural and synthetic ferric tungstate structures 696 (Escobar et al. 1971; Sensgas and Galy 1974; Pinto et al. 1977; Yang et al. 2003), a maximum of 697 two octahedra are edge-sharing with the W-octahedron, and only for Fe₂WO₆ (Sensgas and Galy 698 1974; Pinto et al. 1977), these edge-sharing octahedra are Fe^{3+} . In the goethite structure, however, 699 all Fe-octahedra have four edge-sharing Fe partners forming the octahedral double chains. In 700 701 analogy to the above-mentioned natural and synthetic ferric tungstate structures, we assume, that for the goethite structure the number of four hypothetic edge-sharing Fe partners adjacent to W has to be reduced to a maximum of two. This results in two vacant Fe sites in goethite adjacent to W^{6+} . Accordingly, one W would replace three Fe atoms, exactly as suggested by our chemical analyses. Obviously, this substitution produces three excess negative charges, which would have to be balanced by the protonation of the adjacent oxygen atoms, previously connected to Fe³⁺, or by the formation of oxygen vacancies. For goethite, μ RS demonstrates that the protonation seems to be the actual valence balancing process (see Eq. 4).

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Eq. 4:
$$3Fe^{3+}$$
 protonatio n & vacancies $W^{6+} + 3H + 2[]^0$

The hematite structure does not possess structurally bound H which could compensate forthe excess charge.

Due to the isotructural relationship between hematite (Fe₂O₃) and ilmenite (Fe²⁺Ti⁴⁺O₃), a reduction of surrounding Fe³⁺ to Fe²⁺ was suggested for the structural incorporation of Ti⁴⁺ (Morin 1951; Balko and Clarkson 2001) and Ge⁴⁺ (Sieber et al. 1985; Bernstein and Waychunas 1987), but no Fe²⁺ could be detected by XANES.

Furthermore, as for W-bearing goethite, our chemical analyses suggest, that for each W 716 717 nearly three Fe atoms leave the structure. In addition to the specific atomic position where W substitutes Fe, also the next closest Fe atom with a distance of 2.9 Å is highly unlikely to stay in 718 the hematite structure. To our knowledge, no natural or synthetic structure displays face-sharing 719 $Fe^{3+}-W^{6+}$ or even $W^{6+}-W^{6+}$ octahedra. Additionally, as in goethite, the number of edge-sharing 720 721 Fe-octahedra would have to be reduced by one. This results in a total loss of three Fe atoms for one W^{6+} . As in the case of goethite, either the protonation of O^{2-} atoms by H⁺ or the formation of 722 O²-vacancies could compensate for the excess negative charge. As already obtained for hematite 723 grown from precursor ferrihydrite, protonated oxygen and Fe³⁺-vacancies are more likely 724 (Wolska 1981, 1988; Wolska and Szajda 1985). 725

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Implications

728 The combination of chemical and spectroscopic techniques as well as the consideration of 729 the different processes involved during maturation of ferrihydrite to hematite or goethite imply 730 that oscillatory-zoned, W-bearing hematite records the chemical evolution of a fluid and allows 731 to distinguish different fluid batches. Botryoidal hematite formed as solid transformation from 732 the precursor phase ferrihydrite can record the fluid history of a W-bearing fluid (both in W-733 enriched and W-poor environments), as hematite inherits the W content of the precursor 734 ferrihydrite. Thus, hematite is able to record differences in W content in a fluid and retain this 735 information over geologically long periods of time. In contrast, goethite only records the W concentrations, which prevailed at the time of its crystallization, i.e., during goethite formation 736 737 from ferrihydrite. In combination: hematite records the W concentrations in a fluid during ferihvdrite precipitation, goethite during ferrihydrite maturation. 738

The W fraction incorporated into goethite depends on the W concentration in the system. This involves mixing processes of W from dissolving ferrihydrite, W present in the actual fluid and a complex relationship between the adsorption potential of goethite, its crystal size/morphology and its precipitation rate. These interactions cannot be evaluated yet. For hematite containing less than 0.4 mol% W, the adsorption potential of W to ferrihydrite appears to be the most important parameter for W incorporation.

Our investigations indicate that the incorporation of W into hematite and goethite takes place as a heterovalent substitution of one W for three Fe atoms into their crystal structure. Hence, W is, once incorporated, immobilized for as long as these phases are stable. Lastly, it should be noted that W-bearing iron-oxi(hydroxi)des, formed in the oxidation zones of W deposits, represent potentially economic W resources.

751

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988 Figure captions:

989 Fig. 1: Heterogeneous and homogeneous W-distribution in hematite from the Schwarzwald. (a-b) 990 Typical heterogeneous W-distribution between W-rich hem-I and hem-II, BF-11. Note relic siderite morphology with porous hematite inside (marked in blue). Bright green-red regions of 991 992 secondary Y(\pm REE)-phosphate are misinterpret cause of W L α and Y K α overlap (b upper right; colored about 300-440 cps). Regions of homogeneous distributions used for XAFS measurements 993 (c-d) hematite in BF-11, (e-f) hematite in BF-26 and (g-h) goethite in BF-20. (a) 994 995 Microphotograph, reflected light, slightly dejusted polarizers, (b, d, f, h) W-distribution maps of the W Lα fluorescence, (c, e, g) BSE-images in COMPO-mode. White arrows indicate growth 996 directions of hematite or goethite crystallites. 997

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Fig. 2: W concentrations within hematite (red) and goethite (brown) used in this study as molar 999 fraction χ of W in hematite and goethite, respectively. For calculation of χ see Eq. 2. a) synthetic 1000 1001 samples are given as EMPA data (crosses) and TXRF data (squares) versus their initial molar W/Fe ratios (lower scale; 0.00, 0.05, 0.1, 0.2, 0.4, 0.8 and 1.6) and appropriate W concentrations 1002 1003 (upper scale) within the fluids during their synthesis. Error bars in black indicate the relative error 1004 of the TXRF analysis of 10%, the crosses the variation of the single EMPA measurements. b) 1005 EMPA data of the natural samples as variation of each single measurement (crosses). In a) and b) mean values marked in black and median values in white. Samples description in the lower part: 1006 1007 BF samples from W-free/minor W deposits, GR samples from scheelite vein. Indexes of the SYN samples express the molar W/Fe ratios during synthesis. The grey shaded areas represent the 1008 interpolated overlap of TXRF and EMPA data. Dotted lines show the trends of W incorporation 1009 by EMPA mean values, indicating different slopes of hypothetic solid solutions at highly diluted 1010 W synthesis fluids. 1011

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Fig. 3: W L₃ µ-XANES spectra of natural (a) hematite (BF-26), (b) goethite (BF-20), (c) 1013 1014 synthetic hematite (SYN-hem_{0.05}), (d) and (e) synthetic goethites (solid line SYN-goe_{0.05:70°C}, dashed line SYN-goe_{0.05:25°C}, respectively). Samples are compared to XANES spectra of 1015 reference (f) MnWO₄ (hübnerite), (g) (Fe,Mn)WO₄ (wolframite): WO₆ octahedra; and (h) 1016 1017 CaWO₄ (scheelite), and (i) Na₂WO₄·2H₂O: WO₄ tetrahedra. Natural and synthetic samples were measured at thin sections and at pellets in fluorescence mode and the reference materials at 1018 pellets in transmission. The spectra of the hematite sample BF-26 with higher energy resolution 1019 1020 is given a dotted line.

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Fig. 4: W $L_1 \mu$ -XANES spectra of (a) natural hematite (BF-26) and (b) synthetic goethite (SYN-1022 goe_{0.05:70°C}). Samples compared to XANES transmission spectra of reference (c) MnWO₄ 1023 (hübnerite), (d) (Fe,Mn)WO₄ (wolframtie), both WO₆ octahedra and (e) CaWO₄ (scheelite), (f) 1024 $Na_2WO_4 \cdot 2H_2O_5$, both WO₄ tetrahedra. Samples were mesured at thin section and pellet in 1025 1026 fluorescence mode and the reference materials at pellets in transmission. For natural goethite and SYN-goe_{0.05:25°C} and SYN-hem_{0.5} fluorescence emission was weak to measure L_1 edge spectra in 1027 reasonable time. (A) indicates the pre-edge 2s(W) to 5d(W) + 2p(O) transition (2p(O)-5d(W))1028 1029 mixing), (B) the edge (2s(W) to 6p(W) + 2p(O) transition to the continuum), (C) and (D) multiple scattering. 1030

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Fig. 5: k^3 -weighted W L₃-edge EXAFS spectra (left), real part Fourier transform W L₃-edge EXAFS spectra (middle) and magnitude of the Fourier transform (right). (a) goethites SYNgoe_{0.05-1.6;70°C} (synthesized at 70°C), SYN-goe_{0.05;25°C} (synthesized at 25°C), BF-20, and GR_{G-I}. (b) hematites SYN-hem_{0.05-1.6} (synthesized at 500°C), BF-26 and BF-11. The measured data is

- indicated by the dotted lines, whereas the corresponding fit is indicated by the solid lines. R-space uncorrected for phase shift.
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- 1039 Fig. 6: Raman spectra of the synthetic goethite samples with increasing W concentration from
- 1040 bottom to top: SYN-goe_{0.00;70°C}, SYN-goe_{0.05;70°C}, SYN-goe_{0.1;70°C}, SYN-goe_{0.2;70°C}, SYN-
- 1041 goe_{0.4;70°C}, SYN-goe_{0.8;70°C} and SYN-goe_{1.6;70°C}. (a) whole spectra; (b) curve fit of the OH bending
- 1042 modes at 1500-900 cm⁻¹ range; (c) curve fit of 780-580 cm⁻¹ range; (d) curve fit of 450-330 cm⁻¹
- 1043 range; (e) range of 275-150 cm⁻¹. For the spectra fits, dashed lines represent the fitting peaks, red
- 1044 solid lines their accumulation and bold solid lines the fitted data.

Table 1: Lattice parameters of powder samples hematite (BF-26 and SYN-hem_{0.00-1.6}) and goethite (SYN-goe_{0.00-1.6}) compared to their respective W-concentrations. Additionally the table compares Fe_2O_3 and WO_3 chemistry determined by TXRF and EMPA (cf. Fig. 2). The Fe-loss per incorporated W, calculated by Eq. 1, is shown for the TXRF results of the synthetic samples. Table 2 shows the detailed chemistry of all natural samples.^{a, b}

			L	attice parar	neters				ТХ	RF - Iron-	loss			EMP	A	
Sample	phase	a	b	с	cryst.	Rwp	GOF	Fe ₂ O ₃	WO ₃	total	\mathbf{N}^{g}	Fe-loss	Fe ₂ O ₃	WO ₃	total	\mathbf{N}^{g}
		[Å]	[Å]	[Å]	[nm]			[wt%]	[wt%]	[wt%]			[wt%]	[wt%]	[wt%]	
BF-26 I	hem	5.0357		13.7569	61	4.26	1.13	NM	bdl ^c	-	3	-	NM	NM	-	-
BF-26 II	hem	5.0366		13.7601	75	5.82	1.56	NM	0.23 ^c	-	3	-	NM	NM	-	-
BF-26 III	hem	5.0362		13.7599	79	6.66	1.79	NM	0.60°	-	3	-	NM	NM	-	-
BF-26 IV	hem	5.0362		13.7577	77	5.92	1.57	NM	0.62 ^c	-	3	-	NM	NM	-	-
BF-26 V	hem	5.0365		13.7591	76	6.00	1.59	NM	0.76 ^c	-	3	-	NM	NM	-	-
BF-26 VI	hem	5.0367		13.7583	79	5.58	1.49	NM	1.19 ^c	-	3	-	95.87^{f}	1.05 ^f	100.27	21
BF-26 VII	hem	5.0364		13.7571	82	6.00	1.57	NM	1.53°	-	3	-	95.47^{f}	$1.37^{\rm f}$	99.55	20
SYN-hem _{0.00}	hem	5.0391		13.7613	22	6.38	1.30	99.6 ^d	bdl ^d	99.6	2	-	67.88 ^e	bdl ^e	68.26	10
SYN-hem _{0.05}	hem	5.0387		13.7575	29	6.21	1.35	97.8 ^d	0.2 ^d	98.0	2	2.69	66.63 ^e	0.24 ^e	67.19	10
SYN-hem _{0.1}	hem	5.0392		13.7578	25	6.43	1.38	100.4 ^d	0.5 ^d	100.9	2	2.81	63.84 ^e	0.43 ^e	64.56	10
SYN-hem _{0.2}	hem	5.0400		13.7605	26	6.51	1.34	99.8 ^d	1.2 ^d	101.0	3	2.86	76.89 ^e	1.11 ^e	78.39	10
SYN-hem _{0.4}	hem	5.0400		13.7652	22	6.37	1.27	94.4 ^d	1.8 ^d	96.2	3	2.88	75.51 ^e	1.69 ^e	77.64	10
SYN-hem _{0.8}	hem	5.0409		13.7637	29	6.44	1.36	97.7 ^d	2.2 ^d	99.9	3	2.88	79.21 ^e	1.99 ^e	81.55	20
SYN-hem _{1.6}	hem	5.0398		13.7620	38	6.50	1.36	101.2 ^d	1.7 ^d	103.0	3	2.87	80.45 ^e	2.69 ^e	83.40	10
SYN-goe _{0.00;70°C}	goe	9.9606	3.0240	4.6101	33	6.63	1.36	93.7 ^d	bdl ^d	93.7	3	-	48.86 ^e	bdl ^e	49.08	11
SYN-goe _{0.05;25°C}	goe	9.9653	3.0241	4.6122	31	6.98	1.47	83.5 ^d	3.0 ^d	86.5	3	2.90	48.30 ^e	1.56 ^e	50.08	28
SYN-goe _{0.05;70°C}	goe	NM	NM	NM	NM	NM	NM	NM	NM	NM	-	-	53.08 ^e	1.86 ^e	55.20	14
SYN-goe _{0.1;70°C}	goe	9.9644	3.0238	4.6145	47	6.36	1.38	83.6 ^d	4.9 ^d	88.5	3	2.90	45.46 ^e	2.54 ^e	48.22	10
SYN-goe _{0.2;70°C}	goe	9.9652	3.0238	4.6172	66	6.22	1.35	83.1 ^d	5.0 ^d	88.1	2	2.90	53.56 ^e	3.96 ^e	57.74	13
SYN-goe _{0.4;70°C}	goe	9.9665	3.0237	4.6210	102	5.92	1.33	78.8 ^d	8.8 ^d	87.6	3	2.90	60.50 ^e	8.19 ^e	68.94	10
SYN-goe _{0.8;70°C}	goe	9.9658	3.0232	4.6211	65	6.83	1.22	83.2 ^d	7.7 ^d	90.9	1	2.90	66.31 ^e	6.65 ^e	73.24	17
SYN-goe _{1.6;70°C}	goe	9.9630	3.0235	4.6243	45	7.08	1.21	68.8 ^d	10.3 ^d	79.0	2	2.90	61.47 ^e	8.51 ^e	70.33	21

^a bdl = signal detected but below detection limit; ^bNM = not measured ; ^cTXRF bulk analysis of solid powder samples. Here a quantification of Fe₂O₃ is not possible due to self-absorption effects; ^dTXRF bulk analysis of solved powder samples; ^cEMPA data of resinated and polished powder samples; ^fEMPA data of polished thin sections. Deficit to total according to oxide components not presented (cf. Table 2); ^gNumber of measurements.

Table 2: Representative chemical results of the natural hematite/goethite samples by EMPA characterization. The data is given as mean values in wt% of the respective element oxides. The variation of the chemistry of each sample in respect to W and the comparison between the EMPA and TXRF data is shown in Fig. 2

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							EMI	PA [wt%	-]					
Sample	Phase	WO ₃	Fe ₂ O ₃	MnO	Al_2O_3	MgO	Sb_2O_5	P_2O_5	As_2O_5	MoO ₃	CaO	SiO_2	total	Ν
BF-20	goe	0.20	79.29	0.19	0.24	0.025	0.17	0.98	0.34	bdl	0.05	4.12	85.52	53
BF-11	hem	0.45	93.79	0.15	0.37	0.017	0.19	0.02	1.17	0.10	0.10	0.86	97.17	255
BF-26	hem	1.38	92.64	0.30	0.60	0.16	0.05	0.32	0.70	bdl	0.10	0.98	97.33	229
BF-26	goe	0.04	79.71	0.62	0.09	0.037	bdl	0.42	0.06	bdl	0.03	2.92	83.88	60
GR _{G-II}	goe	3.65	79.40	0.01	0.29	0.023	NM	1.45	0.40	NM	0.04	1.94	86.80	18
GR _{G-I}	goe	11.81	74.71	0.01	0.26	0.025	NM	1.14	bdl	NM	0.07	1.32	89.33	26
GR _{H-I}	hem	3.84	86.53	bdl	0.26	0.018	NM	0.46	bdl	NM	0.03	2.09	93.23	8

bdl = signal detected but below detection limit; NM = element not measured; hem = hematite; goe = goethite; N = number of measurements

		GR _{G-I}				BF-20		SYN-goe _{0.05;70°C}					SYN-goe _{0.05;25°C}					
path	Ν	R [Å]	σ^2	path	Ν	R [Å]	σ^2	pa	ath	Ν	R [Å]	σ^2	path	Ν	R [Å]	σ^2		
W-O	4	1.928±0.006	0,008	W-O	4	1.774±0.009	0,007	W	/-0	4	1.812±0.005	0,007	W-O	4	1.781±0.001	0,007		
W-O	2	2.10±0.01	0,008	W-O	2	2.10±0.02	0,007	W	/-0	2	2.12±0.01	0,007	W-O	2	2.12±0.01	0,007		
W-Fe	1	2.94±0.03	0,012	W-Fe	1	2.857±0.009	0,015	W	/-Fe	1	2.98±0.02	0,010	W-Fe	1	2.87±0.001	0,015		
W-O	1	3.36±0.02	0,012	W-O	1	3.35±0.02	0,015	W	/-0	1	3.36±0.01	0,010	W-O	1	3.37±0.02	0,015		
W-Fe	1	3.43±0.02	0,012	W-Fe	1	3.43±0.02	0,015	W	/-Fe	1	3.44±0.01	0,010	W-Fe	1	3.44±0.02	0,015		
W-Fe	4	3.57±0.02	0,012	W-Fe	4	3.57±0.02	0,015	W	/-Fe	4	3.58±0.01	0,010	W-Fe	4	3.58±0.02	0,015		
$\Delta E_0 = 1.7$, N _{var} = 15, N _{ind} = 20.2, R = 0.025, k range = 3.6-11.6, R range = 1-4, window type: Hanning				= 1.7, $N_{var} = 15$, $N_{ind} = 20.2$, 0.025, k range = 3.6-11.6, e = 1-4, window type: Hanning $\Delta E_0 = -0.7$, $N_{var} = 14$, $N_{ind} = 18.77$, $R = 0.017$, k range = 2.5-10, R range = 1-4, window type: Hanning					ΔE <i>R</i> R rang	o = 2.8 ? = 0.0 ge = 1	5, N _{var} = 15, N _{ind} = 22, <i>k</i> range = 3- -4, window type	= 21.3, 11.5, : Hanning	$\Delta E_0 = 1.2$, N _{var} = 14, N _{ind} = 21.3, R = 0.017, k range = 2.5-11, R range = 1-4, window type: Hanning					

Table 3.1: Interatomic distances, Δr , and σ^2 of W-bearing goethite fit to sample GR_{G-I}, BF-20, SYN-goe_{0.05-1.6;70°C} and SYN-goe_{0.05;25°C}.

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Table 3.2: Interatomic distances, Δr , and σ^2 of W-bearing hematite fit to sample BF-11, BF-26 and SYN-hem_{0.05-1.6}.

		BF-11					BF-26			SYN-hem _{1.6}						
path	N	R [Å]	σ^2		path	Ν	R [Å]	σ^2		path	Ν	R [Å]	σ^2			
W-O	4	1.822±0.004	0,007		W-O	4	1.823±0.005	0,008		W-O	4	1.845±0.004	0,006			
W-O	2	2.100±0.008	0,007		W-O	2	2.09±0.01	0,008		W-O	2	2.087±0.009	0,006			
W-Fe	2	3.058±0.006	0,010		W-Fe	2	3.071±0.005	0,010		W-Fe	2	3.059±0.005	0,009			
W-Fe	3	3.451±0.006	0,010		W-Fe	3	3.464±0.005	0,010		W-Fe	3	3.452±0.005	0,009			
W-O	2	3.382±0.008	0,010		W-O	2	3.38±0.01	0,010		W-O	2	3.487±0.005	0,009			
W-O	1	3.382±0.008	0,010		W-O	1	3.38±0.01	0,010		W-O	1	3.487±0.005	0,009			
W-O	3	3.473±0.004	0,010		W-O	3	3.480±0.005	0,012		W-O	3	3.587±0.009	0,013			
W-Fe	3	3.581±0.004	0,010		W-Fe	3	3.590±0.005	0,012		W-Fe	3	3.605±0.004	0,013			
W-Fe	3	3.689±0.008	0,010		W-Fe	3	3.68±0.01	0,012		W-Fe	3	3.676±0.009	0,013			
W-O	3	3.769±0.008	0,010		W-O	3	3.670±0.005	0,012		W-O	3	3.685±0.004	0,013			
W-Fe	1	3.969±0.008	0,010		W-Fe	1	3.96±0.01	0,012		W-Fe	1	3.885±0.004	0,013			
W-O	3	4.119±0.008	0,010		W-O	3	4.11±0.01	0,012		W-O	3	4.107±0.009	0,013			
W-O	2	4.271±0.004	0,010		W-O	2	4.279±0.005	0,012		W-O	2	4.366±0.009	0,013			
W-O	1	4.272±0.004	0,010		W-O	1	4.279±0.005	0,012		W-O	1	4.294±0.004	0,013			
ΔE₀ R = R rang	$\Delta E_0 = 5.2$, N _{var} = 14, N _{ind} = 22.5, R = 0.015, k range = 2.5-11.5, R range = 1-4, window type: Hanning					$\Delta E_0 = 6.6$, N _{var} = 15, N _{ind} = 21.3, R = 0.019, k range = 2.5-11, R range = 1-4, window type: Hanning						$\Delta E_0 = 8.9, N_{var} = 15, N_{ind} = 23.8,$ R = 0.018, k range = 2.5-12, R range = 1-4, window type: Hanning				

Fig. 1: (1 column; 90 mm width)



Fig. 2: (2 column; 180 mm width)



Fig. 3: (1 column; 90 mm width)



Fig. 4: (1 column; 90 mm width)





Fig. 6: (1 column; 90 mm width)

