1	Revised version 3
2	Ca L <sub>2,3</sub> -edge near edge X-ray absorption fine structure of tricalcium aluminate,
3	gypsum and calcium (sulfo)aluminate hydrates
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#### Abstract

25 Tricalcium aluminate (cement clinker phase), gypsum, katoite, ettringite and calcium monosulfoaluminate hydrate (abbreviated as kuzelite) are the major minerals in the 26 27 hydration reaction of tricalcium aluminate in the presence of gypsum and have critical 28 impacts on the kinetics and thermodynamics of early-age cement hydration mechanisms. 29 Here, spectroscopic analysis of these minerals is conducted using scanning transmission X-ray microscopy (STXM). Their Ca  $L_{2,3}$ -edge near edge X-ray absorption fine structure 30 31 (NEXAFS) spectra are measured and correlated to the known Ca coordination environments. The results indicate that these minerals have unique Ca environments that 32 33 can be differentiated from one another based on the intensities and positions of the 34 absorption peaks at 346.5-348.5 eV and 350.5-351.5 eV. It is concluded that Ca in 35 tricalcium aluminate (cubic and orthorhombic polymorphs) and katoite is in cubic-like 36 coordination with negative 10Dq, whereas Ca is in an octahedral-like coordination with positive 10Dq in ettringite, gypsum and kuzelite. For tricalcium aluminate, the Ca atoms 37 38 in both polymorphs are in similar chemical environments with slightly more distortion in 39 the orthorhombic polymorph. As a common issue in STXM experiments, absorption 40 saturation of NEXAFS spectra is also investigated. It is demonstrated that the optical 41 density difference between pre- and post-edge absorption levels provides a reliable 42 indication of the sample thickness in the systems studied. The present work provides a reference for STXM study of the calcium (sulfo)aluminate reaction system in cement 43 44 hydration and natural aqueous environments.

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Keywords: Tricalcium aluminate, calcium aluminate hydrate, calcium sulfoaluminate
hydrate, STXM, NEXAFS, Ca L<sub>2,3</sub>-edge, absorption saturation

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## Introduction

51 Calcium (sulfo)aluminate hydrates are found in geological formations (Bell and Rossman 1992; Nelson 1982), wet near-surface environments (Puppala et al. 2005; van 52 Aardt and Visser 1977), and complex composite materials such as hydrated Portland 53 54 cement (PC) (Taylor 1997; Mehta and Monteiro 2006). Recent studies suggest that heavy metal oxyanions, such as chromate (Hashem and Amin 2014), arsenate (Sasaki et al. 55 2014) and selenate (Baur and Johnson 2003), can be immobilized using calcium 56 (sulfo)aluminate hydrates, e.g., ettringite (Chrysochoou and Dermatas 2006; Zhou et al. 57 2006; Hillier et al. 2007), as these elements can be directly substituted into the structures 58 of these minerals. The substitution extent of heavy metal oxyanions has been shown to 59 depend on the coordination environment of Ca with respect to sulfate. It has also been 60 suggested that the coordination environment of Ca is a critical parameter in determining 61 62 the reactivity of minerals (Davidovits 1994; Skibsted et al. 1993; Black et al. 2003). The Ca coordination symmetry of many natural and synthetic solid phases has been studied 63 by Ca  $L_{2,3}$ -edge NEXAFS (Solomon et al. 2012; Ha et al. 2012; Jackson et al. 2013; Geng 64 65 et al. 2015; Fleet and Liu 2009). The peak structures and separations at the absorption edges are affected by the crystal field splitting and spin-orbit effects (Stöhr 2013). 66 Nanometer-resolved identification of Ca coordination environments in complex 67 composite materials (e.g., hydrated cement) has been reported but seems challenging due 68 to the lack of direct links between NEXAFS spectra and coordination configurations 69

(Solomon et al. 2012; Jackson et al. 2013). The efficient assignment of spectra to mineral
phases greatly relies on understanding the reference spectra of known minerals and semiempirical prediction of coordination environment from the spectra details.

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74 Scanning transmission X-ray microscopy (STXM) combines spatially resolved 75 measurements with X-rays of tunable energy at synchrotron radiation facilities (Warwick 76 et al. 2002; Kilcoyne et al. 2003). The bonding and coordination environments of Ca in 77 calcium (sulfo)aluminate hydrates intermixed on the nano- and microscales can be spatially determined by STXM using coupled near edge X-ray absorption fine structure 78 (NEXAFS) measurements (Stöhr 2013). This technique has gained significant interest in 79 80 geology (Wan et al. 2007), environmental and material research applications (Chen et al. 2014), photovoltaics (McNeill et al. 2008) and cosmological chemistry (Sandford et al. 81 82 2006). Cement hydration systems have submicron chemical and structural heterogeneities 83 that greatly influence their macroscopic performances, e.g., rheology, hydration heat 84 release and volumetric stability. STXM provides a powerful analytical tool for measuring 85 both spectroscopic and spatial information down to the micro- or nanometer scales for cementitious materials; however, collecting reference spectra of cement-based minerals is 86 critical for analyzing the spatially resolved spectroscopic data. 87

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Tricalcium aluminate ( $3CaO \cdot Al_2O_3$ , abbreviated as  $C_3A$ ) is a clinker mineral whose hydration determines the early-age performance of fresh concrete mixtures. It is cubic ( $C_3A_{cub}$ ) when the Na<sub>2</sub>O mass content is less than 1.9 wt% and orthorhombic ( $C_3A_{orth}$ ) when the Na<sub>2</sub>O doping is between 3.5 wt% and 4.2 wt% (Barnes and Bensted 2002). Its

direct reaction with water rapidly forms metastable calcium aluminate hydrates, which 93 94 quickly become katoite (3CaO·Al<sub>2</sub>O<sub>3</sub>·6H<sub>2</sub>O, a garnet structure type mineral). In the presence of gypsum (CaSO<sub>4</sub>·2H<sub>2</sub>O), the hydration speed is adjusted by the slow 95 formation of ettringite  $(3CaO \cdot Al_2O_3 \cdot 3CaSO_4 \cdot 32H_2O)$ , which is later transformed into 96 97 calcium monosulfoaluminate hydrate, i.e. kuzelite (3CaO·Al<sub>2</sub>O<sub>3</sub>·CaSO<sub>4</sub>·12H<sub>2</sub>O) through reaction with remnant C<sub>3</sub>A after the gypsum is completely consumed (Barnes and 98 Bensted 2002). In this project, STXM is used to collect the reference Ca  $L_{2,3}$ -edge 99 100 NEXAFS spectra of several calcium (sulfo)aluminate hydrate mineral phases. The 101 correlations between the spectral features and Ca coordination environments are 102 discussed. The influence of the absorption saturation on the correct interpretation of the 103 Ca  $L_{2,3}$ -edge NEXAFS spectra is also studied. This work provides reference spectra for 104 the STXM study of calcium (sulfo)aluminate reaction systems in cement hydration and 105 natural aqueous environments.

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### Ca *L*<sub>2,3</sub>-edge NEXAFS theory

The crystal field splitting of  $d^0$  compounds such as  $Ca^{2+}$  has been investigated 108 experimentally and theoretically (Figure 1) to understand the link between coordination 109 110 symmetry and near-edge adsorption fine structure (de Groot et al. 1990; Burns 1993; 111 Albright et al. 2013). The lowest unoccupied molecular orbitals (LUMOs) in the  $CaO_x$ 112 complex are  $t_{2g}$  and  $e_g$ . They are close in energy but distinct in symmetry. The  $t_{2g}$  state is mainly composed of  $d_{xy}$ ,  $d_{yz}$  and  $d_{xz}$  orbitals and therefore has the same symmetry as these 113 orbitals, whereas  $e_g$  is mainly composed of  $d_z^2$  and  $d_x^2 d_{y}^2$  orbitals. In the isolated Ca<sup>2+</sup> ion, 114 the  $t_{2g}$  and  $e_g$  states share the same energy, but when approached by coordinating oxygens, 115

116 their energies become significantly different. The  $t_{2g}$  state is at a higher energy than the  $e_g$ 117 state when Ca coordination is in cubic symmetry whereas the opposite is true for octahedral symmetry. Therefore, the energy gap  $(e_g - t_{2g})$ , equivalently termed 10Dq, is 118 119 negative for cubic symmetry and positive for octahedral symmetry. A typical Ca  $L_{2,3}$ -120 edge NEXAFS spectra has two major peaks  $(a_2 \text{ and } b_2)$ , two minor peaks  $(a_1 \text{ and } b_1)$ , and 121 several leading peaks (1, 2 possibly). The spin-orbital interaction splits the Ca  $L_{2,3}$ -edge 122 NEXAFS spectra into two doublets. The lower-energy doublet (a1 and a2) corresponds to 123 excitation from  $2p_{3/2}$  orbitals whereas the higher-energy doublet (b1 and b2) corresponds 124 to excitation from  $2p_{1/2}$  orbitals. In each doublet, the major peak corresponds to a final 125 state of the higher energy state in  $t_{2g}$  and  $e_g$ , while the lower one corresponds to the minor 126 peak. The peak position differences  $a_2-a_1$  and  $b_2-b_1$  are determined by, but not identical to, 127 10Dq (de Groot et al. 1990). The leading peaks 1, 2 and  $b_{10}$  are part of the multiplet 128 spectrum and can be accounted for by the Coulomb repulsion and exchange terms (de 129 Groot et al. 1990). In cubic coordination, the minor peak 2 occurs between the energies of 130 the  $a_1$  and  $a_2$  peaks, which can be used to distinguish between cubic and octahedral 131 coordination symmetries.

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In a Ca  $L_{2,3}$  NEXAFS spectrum, peak broadening is caused by the solid-state broadening effect when the sample becomes thicker and disorder appears in the crystal structures, which increases the number of inter-peak absorption states (Obst et al. 2009). The intensities of the minor peaks also increase more rapidly than those of the major peaks with increasing sample thickness. A previous study concluded that peak heights should not exceed  $0.30\pm0.05$  OD (optical density) to obtain reliable spectra with no absorption

saturation artifacts, corresponding to a sample thickness of 30 nm (Hanhan et al. 2009). However, conditions vary from beamline to beamline. Quantitative NEXAFS spectra with no saturation artifacts can be obtained using beamline 5.3.2.2 at the ALS of the Lawrence Berkeley National Laboratory for OD values of ~1.0, corresponding to a sample thickness of ~200 nm. Saturation effects are generally significant in the analysis of NEXAFS spectra of cement systems, as the typical grain size distribution in cement varies from ~100 nm to ~100  $\mu$ m.

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### Materials and Methods

148 To ensure purity, all samples studied here were lab-prepared phases, rather than from 149 natural sources. Katoite, ettringite,  $C_3A_{cub}$  and  $C_3A_{orth}$  powders were purchased from 150 Mineral Research Processing (http://www.mineralresearchprocessing.fr/). The crystal 151 structures and chemical compositions of these minerals were verified by powder X-ray diffraction (XRD) (Electronic Supporting Information). Gypsum was purchased from 152 Fisher Scientific (#S76764). Ca L<sub>2,3</sub>-edge NEXAFS spectra of katoite, ettringite, C<sub>3</sub>A<sub>cub</sub> 153 154 and  $C_{3}A_{orth}$  were measured on samples prepared by drop-casting small amounts of these precursors directly onto Si<sub>3</sub>N<sub>4</sub> windows. Ca  $L_{2,3}$ -edge NEXAFS spectra of kuzelite was 155 156 obtained by spectromicroscopic analysis of samples produced by mixing gypsum,  $C_3A_{cub}$ 157 and deionized water by hand at a mass ratio of 0.4:1:14 in sealed plastic vials under ambient conditions (1 atm,  $25 \pm 2$  °C). Small volumes (~0.2 µL) of the hydrated mixtures 158 159 were drop-cast onto 100-nm-thick  $Si_3N_4$  windows (Norcada) after 9 days of reaction. The reactions were stopped by removing excess water immediately after drop-casting by 160 161 gently contacting the mixtures with Kimwipes (Kimtech Science), leaving a thin layer of

hydrated solids (Figure 2) (Kilcoyne et al. 2003). Ca  $L_{2,3}$ -edge NEXAFS spectra of katoite and C<sub>3</sub>A<sub>cub</sub> were measured on samples prepared by drop-casting small amounts of these precursors directly onto Si<sub>3</sub>N<sub>4</sub> windows.

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166 The STXM and Ca  $L_{2,3}$ -edge NEXAFS measurements were conducted at beamline 167 5.3.2.2 of the Advanced Light Source (Lawrence Berkeley National Laboratory), which 168 operates at 1.9 GeV and 500 mA using a bending magnet source and an active servo-169 stabilized toroidal pre-mirror. Two computer-controlled slits are placed on each side of a 170 low dispersion spherical grating monochromator. Within the energy range of 250-780 eV, 171 the resolving power of the radiation source at beamline 5.3.2.2 is  $E/\Delta E \sim 5000$  when 172 using slit dimensions of approximately 20 x 10 x 10  $\mu$ m and decreases to  $E/\Delta E \sim 2000$ with slit dimensions of approximately 60 x 30 x 30  $\mu$ m, where E is the incident beam 173 174 energy and  $\Delta E$  is the resolvable energy step size. The scanning transmission X-ray microscope installed on beamline 5.3.2.2 has a spatial scanning resolution limit of ~31 175 176 nm utilizing a 240 µm diameter zone plate, with the transmission signal recorded using a 177 phosphor-coated photomultiplier tube (PMT) (Stöhr 2013; Collins and Ade 2012). The photon energy is calibrated by setting the position of the Rydberg 3s peak for CO<sub>2</sub> to 178 179 292.74 eV as detailed in reference (Collins and Ade 2012).

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Samples were measured under 0.3 atm of helium. The energy scanning range was set to 340-360 eV, with a step size of 0.1 eV between 345 and 356 eV and a step size of 0.2 eV used at the other sampled energies. The measured data were processed using aXis2000

184 software (Hitchcock et al. 2012).

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# **Results and Discussion**

## **187** Ca L<sub>2,3</sub>-edge NEXAFS Spectra and Crystal Chemistry

The Ca  $L_{2,3}$ -edge NEXAFS spectra of all the studied minerals show fine structures within the energy range of 346 to 354 eV (Figure 3), consistent with Ca  $L_{2,3}$ -edge NEXAFS spectra of many Ca-containing minerals (Solomon et al. 2012; Ha et al. 2012; Jackson et al. 2013; Geng et al. 2015; Fleet and Liu 2009). Unique spectra were repeatedly observed for each mineral phase on different sampling regions of similar thickness, indicating that the chemical environment of Ca in each mineral is relatively homogeneous at the spatial resolution of ~30 nm.

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**Katoite.** The shape of the Ca  $L_{2,3}$ -edge NEXAFS spectrum of katoite (Figure 3) is similar 196 to that of CaF<sub>2</sub> (de Groot et al. 1990; Naftel et al. 2001; Miedema et al. 2011), which 197 contains Ca coordinated to eight F atoms in cubic symmetry. The cubic-like symmetry of 198 199 katoite is confirmed by the larger peak energy differences of  $a_2-a_1$  (1.70 eV) than  $b_2-b_1$ 200 (1.21 eV) in this phase. A similar  $a_2$ - $a_1$  energy difference is found for CaF<sub>2</sub> (1.60-1.70 eV), although the  $b_2$ - $b_1$  energy difference in CaF<sub>2</sub> (1.40-1.44 eV) is larger. According to 201 202 Lager et al. (1987), all Ca and O atoms are in equivalent positions in katoite. Figure 4a shows that each Ca atom is coordinated to eight O atoms, half with Ca-O bond lengths of 203 204 2.360 Å and half with Ca-O bond lengths of 2.514 Å. Along the two-fold rotation axis, 205 the top four O atoms are labelled  $O_t$  and the bottom four are labelled  $O_b$ . The coordination symmetry of Ca in katoite is distorted from perfect cubic symmetry in two 206 207 ways: (1) the different Ca-O bond lengths modify the positions of O atoms in the top and

bottom planes of the structure to rectangular rather than square shapes; and (2) there is aslight misalignment of the top and bottom rectangles (Figure 4a).

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211 **Ettringite.** The Ca  $L_{23}$ -edge NEXAFS spectrum of ettringite (Figure 3a) resembles that 212 of octahedrally coordinated Ca both because the peak energy differences of a2-a1 and b2-213 b1 are comparable and because no peak is observed between a1 and a2. The leading peak 214 2 is almost indistinguishable in this spectrum. The energy differences between the peak positions  $a_2$  and  $a_1$  at the  $L_3$  edge and  $b_2$  and  $b_1$  at the  $L_2$  edge are 1.05 eV and 1.14 eV, 215 216 respectively, which are low relative to the corresponding values for calcite and calcium hydroxide (Geng et al. 2015). The relatively low splittings of the  $L_3$  and  $L_2$  edges and low 217 218 intensities of the minor a<sub>1</sub> and b<sub>1</sub> peaks indicate a weak crystal field.

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220 So far three ettringite structures have been reported where Ca environments are highly 221 comparable (Moore and Taylor 1970; Hartman and Berliner 2006; Goetz and Neubauer 222 2006). Ettringite contains eight-fold coordinated Ca in two positions (Figure 4b), with Ca-O bond lengths for Ca1 ranging from 2.356 Å to 2.739 Å, average of 2.464 Å, and 223 those for Ca2 ranging from 2.317 Å to 2.747 Å, average of 2.491 Å (Moore and Taylor 224 225 1970). Viewing Ca1 along the (100) axis, the corners of the rectangle formed by two O1 226 atoms and two O3 atoms are misaligned with the corners of the rectangle formed by O6, 227 O8, O10 and O12 atoms, and thus, the  $t_{2g}$  orbital is less destabilized by the ligands. Instead, the O8 atom and two O1 atoms can destabilize the  $d_z^2$  orbital from the axial 228 229 direction and the five equatorial oxygen atoms, i.e., O6, O10, O12 and two O3 atoms can destabilize the  $d_x^{2,2}$  and  $d_{xy}$  orbitals (note that O6 and O12 are slightly off the equatorial 230

plane to the O8 side). In contrast, the  $d_{yz}$  and  $d_{xz}$  orbitals are less likely to be destabilized because their symmetry poorly matches that of the ligand configuration. Ca2 has a very similar configuration to Ca1, with O7 and O2 in axial positions and O5, O9, O11, O4s in equatorial positions. Based on the above reasoning, we predict that Ca in ettringite is in a weak crystal field with positive 10Dq. This matches very well with the Ca  $L_{2,3}$ -edge NEXAFS spectrum of ettringite in Figure 3a.

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Kuzelite. The Ca  $L_{2,3}$ -edge NEXAFS spectra for kuzelite and ettringite are similar, except that the positions of the  $a_2$ ,  $b_1$  and  $b_2$  peaks in the former phase are shifted to lower energy, which also has a relatively smaller  $L_3$  splitting (0.93 eV) and a relatively larger  $L_2$ splitting (1.18 eV). The Ca  $L_{2,3}$ -edge NEXAFS spectrum for kuzelite contains very well resolved leading (1, 2) and minor (a1, b1) peaks and has a relatively high signal to noise ratio, which can be attributed to the analysis of large, uniformly thick and near-perfect hexagonal crystals (Figure 3c).

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246 In kuzelite, there is one equivalent Ca position (Allmann 1977) (Figure 4c). The oxygen 247 in water  $(O_w)$  is located on the (001) axis (threefold rotation axis) 2.497 Å from the central Ca atom; three hydroxyl O atoms in hydroxyl groups, labelled O<sub>hd</sub>, are 248 249 equatorially coordinated with bond lengths of 2.375 Å; another three O<sub>hd</sub> atoms are 250 coordinated to the central Ca atom from the opposite side of O<sub>w</sub> with a bond length of 251 2.455 Å (referred as top O<sub>hd</sub> atoms as viewed from the perspective shown in Figure 4c, where the side that the rotation axis is pointing toward is denoted as the top side). There 252 is a misalignment of 43.7° in the orientations of the equatorial O<sub>hd</sub> regular triangle (blue 253

triangle in Figure 4c inset) and the top  $O_{hd}$  regular triangle (red triangle in Figure 4c inset). In this configuration, only the  $d_z^2$  orbital is likely to be significantly destabilized, axially by  $O_w$ , on top by the  $O_{hd}$  triangle, and equatorially by the equatorial  $O_{hd}$  triangle. Compared with the strong destabilization of the perfect octahedral ligand configuration (Moore and Taylor 1970), we expect the Ca  $L_{2,3}$ -edge NEXAFS spectra for kuzelite to be similar to those with small positive 10Dq, which matches the result shown in Figure 3a.

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261 **Gypsum.** The Ca  $L_{2,3}$ -edge NEXAFS spectrum measured for gypsum is similar to the 262 spectra obtained for ettringite and kuzelite (Figure 3a), although the  $L_3$  edge peak 263 splitting  $(a_2-a_1)$  of 0.85 eV and the  $L_2$  edge peak splitting  $(b_2-b_1)$  of 0.90 eV for gypsum 264 are smaller, indicating a weaker crystal field. Values of  $L_3$  and  $L_2$  edge splitting of ~0.95 265 eV have previously been reported for CaSO<sub>4</sub>, indicating a stronger crystal field with 266 emission of crystalliferous water (Patel and Aswath 2012). The Ca  $L_{2,3}$ -edge NEXAFS 267 spectrum for gypsum also contains weak leading 1 and 2 peaks and shows a small, 268 positive 10Dq value because peak 2 is located between peak 1 and  $a_1$ . There is an 269 additional weak b<sub>10</sub> peak at 350.82 eV, which was not identified in the NEXAFS of 270 perfectly octahedral coordinated compounds nor in theoretical calculations (de Groot et al. 271 1990). Therefore, the NEXAFS results for gypsum indicate that this phase contains Ca in 272 distorted coordination symmetry with 3d orbitals destabilized in a way similar to 273 octahedral coordination.

The analysis for gypsum presented above is consistent with the crystal chemistry of Ca in this phase (Comodi et al. 2008). In gypsum, Ca is coordinated to eight oxygen atoms in

277	one equivalent position (Figure 4d), with bond lengths ranging from 2.360 Å to 2.547 Å.
278	The relative orientation of the two O2 atoms and two $O_{\rm w}$ atoms to the four O1 atoms in
279	gypsum and katoite are similar if viewed along the (010) axis (i.e., the rotation axis),
280	although neither the O2, $O_w$ nor O1 atoms lie in the same plane. From the geometric
281	point of view, the coordination of Ca in gypsum can be classified into neither octahedral-
282	like nor cubic-like symmetry but rather a spherical-like symmetry, with 8 ligand O atoms
283	distributing randomly within short range of distances from the center Ca. This leads to a
284	much weaker splittings at the $L_3$ and $L_2$ edges than in ettringite and kuzelite.
285	
286	Cubic and orthorhombic C <sub>3</sub> A. The Ca $L_{2,3}$ -edge NEXAFS spectrum C <sub>3</sub> A <sub>cub</sub> exhibits a
287	cubic-like symmetry pattern with peak 2 located between the positions of peaks $a_1$ and $a_2$
288	with $L_3$ and $L_2$ edge splittings of 1.57 eV and 1.28 eV, respectively.
289	
290	There are six Ca sites in the $C_3A_{cub}$ unit cell (Mondal and Jeffery 1975) (Figure 5). Cal
291	and Ca2 have identical site symmetry and are in a distorted octahedral configuration with
292	Ca-O bond lengths of 2.338 Å. All three oxygen pairs in Ca1 and Ca2 take Ca as the
293	center, yet the angles between these pairs are 79.2° instead of 90°. Ca3 is also distorted
294	from octahedral coordination with three 2.352 Å-long Ca-O bonds and three 2.356 Å-
295	long Ca-O bonds. The angle of each O3-Ca-O4 bond is 162.52°. The four equatorial O
296	atoms are not exactly coplanar. The coordination structures of Ca4, Ca5 and Ca6 are

more complex, with Ca-O bond lengths varying between 2.258 Å and 3.013 Å. Therefore,

299 octahedrally coordinated Ca (Ca1, Ca2 and Ca3) and irregularly coordinated Ca (Ca4,

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Ca L<sub>2,3</sub>-edge NEXAFS spectrum for C<sub>3</sub>A<sub>cub</sub> is a superposition of contributions from

Ca5 and Ca6) and is not expected to show strong  $L_2$  and  $L_3$  edge splitting because the percentage of octahedral-like sites is relatively low (11.1% for summation of Ca1 and Ca2, 11.1% for Ca3, as shown in Table 1). The Ca  $L_{2,3}$ -edge NEXAFS spectrum for C<sub>3</sub>A<sub>cub</sub> is consistent with this line of reasoning, and the complex coordinations of Ca4, Ca5 and Ca6 produce cubic-like crystal field splitting effects with negative 10*Dq*.

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306 The symmetry and size of the unit cell (Table 1) as well as the configurations of Ca 307 atoms (Figure 6) in  $C_3A_{orth}$  and  $C_3A_{cub}$  are different. In  $C_3A_{orth}$ , only Ca3 is in distorted 308 octahedral coordination, with Ca-O bond lengths ranging from 2.31 to 2.409 Å 309 (averaging at 2.355 Å). The other Ca atoms have irregular symmetries but higher 310 coordination numbers (8 for Ca1, Ca2 and Ca4, and 7 for Ca5). Each of the 8-coordinated Ca atoms has four short Ca-O bonds (2.2-2.6 Å) and four long Ca-O bonds (2.6-3.5 Å), 311 312 with average bond lengths of 2.668 Å for Ca1-O, 2.637 Å for Ca2-O and 2.598 Å for Ca4-O. The Ca5 atom has four short Ca-O bonds (2.4-2.5 Å) and three long Ca-O bonds 313 314 (2.6-3.1 Å), averaging 2.616 Å.

The previously mentioned symmetries of Ca atoms in  $C_3A_{orth}$  explains the cubic-like Ca  $L_{2,3}$  edge NEXAFS spectrum that was measured for this phase (Figure 3). This spectrum differs from that for  $C_3A_{cub}$  by a shift in the location of peak 2 to lower energy and a more broadened peak b1, with otherwise minor differences in spectral line shape and peak positions. The energy differences between peaks a2 and a1 and b2 and b1 are slightly smaller in the Ca  $L_{2,3}$  edge NEXAFS spectrum for  $C_3A_{orth}$  relative to  $C_3A_{cub}$ , indicating a slightly weaker crystal field of Ca in  $C_3A_{cub}$ . The doping of Na is not uniform

323 on the unit cell scale, leading to a fractional occupancy of Ca1, which breaks the perfect 324 periodicity over long ranges (Nish and Takeuchi 1975). This may explain the broadening of peak b1 in the C<sub>3</sub>A<sub>orth</sub> spectrum. The sharpening of peak 2 in the spectrum of C<sub>3</sub>A<sub>orth</sub> 325 326 could then be explained by the higher fraction of octahedral-like coordinated Ca3 atoms 327 in  $C_3A_{orth}$  (23.5%) relative to  $C_3A_{cub}$  (11.1% for each of the two octahedral-like 328 coordinated sites), whose al peak is sharp (de Groot et al. 1990). Therefore, the Ca atoms 329 in  $C_3A_{orth}$  are in cubic-like coordination with slightly weaker crystal fields than those in 330  $C_3A_{cub}$ .

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### 332 Absorption saturation

333 In a transmission X-ray spectroscopy experiment, NEXAFS spectra become 'saturated' 334 once the sample thickness reaches a limiting value. Increasing sample thickness beyond 335 this limit results in significantly broadened absorption peaks and increases the peak intensity ratio (PIR) of the minor peaks relative to the major peaks. The Ca  $L_{2,3}$ -edge 336 337 NEXAFS spectra collected for ettringite, gypsum and kuzelite samples of different 338 thickness show that the PIRs of the leading and minor peaks typically increase with 339 increasing sample thickness (Figure 7 and Figure 8b). Peaks 2 and  $b_{10}$  in the Ca  $L_{2,3}$ -edge 340 NEXAFS spectrum for gypsum, as well as peak 2 for ettringite, become increasingly 341 difficult to resolve as the sample thickness increases. Most of the major peaks and some 342 of the minor peaks in the Ca  $L_{2,3}$ -edge NEXAFS spectra for ettringite and gypsum become significantly broadened with increasing sample thickness, although the minor 343 peaks  $(1, 2, a_1, a_2)$  in the Ca  $L_{2,3}$ -edge NEXAFS spectra for kuzelite do not display peak 344 345 broadening. This is explained by the perfect or near-perfect crystal structure of the

kuzelite crystallites sampled (Figure 3c), such that relatively low amounts of structuraldisorder-induced interpeak states are encountered for this phase. Unsurprisingly, peak positions remain unchanged during absorption saturation for all measured phases, meaning that individual phases in composite geochemical systems can be resolved by STXM using this parameter. The features located between 356 and 358 eV are artifacts caused by dynamic focusing of the zone plate.

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Using maximum OD values to estimate sample thickness is a common approach (Hanhan et al. 2009). However, it could be impractical when the background absorption is comparable to the resonance absorption, which frequently occurs in thick sampling regions. Here, we propose another estimation of sample thickness. Sample thickness can be estimated using the Beer-Lambert Law (eqs.(1-2)):

$$I = I_0 e^{-\mu t} \tag{1}$$

$$OD = \mu t = -\ln(I/I_0) \tag{2}$$

360 where  $I_0$  and I are the incident and transmitted beam intensities, respectively, t is the 361 sample thickness, and  $\mu$  is the attenuation coefficient (Hubbell and Seltzer 1995). 362 Although  $\mu$  is a function of the incident beam energy, it can be treated as a constant if the 363 incident beam energy ranges within several tens of eV outside of an absorption edge 364 (Figure 8a). At the absorption edge,  $\mu$  changes drastically with a small range of variation 365 in beam energy. Beyond the absorption edge,  $\mu$  once again changes immeasurably with 366 beam energy, as illustrated in Figure 8a. The difference between the pre-edge and post-367 edge OD, i.e., the optical density difference ( $\Delta$ OD), is related to t and  $\Delta \mu$ , i.e., the 368 difference in  $\mu$  over this same energy range (eq.(3)):

$$\Delta OD = \Delta \mu t \tag{3}$$

Values of minor to major PIRs in unnormalized Ca  $L_{2,3}$ -edge NEXAFS spectra with different  $\triangle OD$  are shown in Figure 8b. Given the same  $\triangle OD$ , the PIRs of ettringite, gypsum and kuzelite are similar despite differences in attenuation coefficients and sample densities. For example, using  $I(b_1)/I(b_2)$  at  $\triangle OD=0.4$ , the PIR of these minerals vary within the range 0.75-0.8. These results indicate that the  $\triangle OD$  parameter may provide a universal measurement of sample attenuation thickness in STXM studies of heterogeneous microscale systems containing these phases.

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378 The  $I(b_1)/I(b_2)$  curve for kuzelite approaches its lower limit asymptotes at  $\Delta OD < 0.1$  and 379 upper limit asymptotes at  $\Delta OD \ge 0.6$ . The same observation applies to all three curves of 380 ettringite, where the lower limit is at  $\Delta OD \sim 0.08$  and the upper limit is  $\Delta OD \sim 0.43$ . The 381  $I(a_1)/I(a_2)$ ,  $I(2)/I(a_2)$  and  $I(1)/I(a_2)$  curves of kuzelite display only lower asymptotes at  $\Delta OD < 0.1$  because the  $\Delta OD$  does not become large enough to cause full saturation in the 382 383 minor peaks  $a_1$ , 1 and 2. For gypsum, a lower asymptote is not clear due to lack of 384 sampling in the low  $\Delta OD$  region but an upper asymptote is obtained for  $\Delta OD > 0.5$ . In 385 the medium  $\triangle OD$  range (approximately 0.1~0.4), the slope of all PIR curves for these 386 phases increase approximately linearly.

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### Implications

In this paper, Ca  $L_{2,3}$ -edge NEXAFS spectra of katoite, ettringite, gypsum, kuzelite, C<sub>3</sub>A<sub>cub</sub> and C<sub>3</sub>A<sub>orth</sub> were reported using scanning transmission X-ray microscopy (STXM). The unique feature of each absorption spectra was described, and the absorption

saturation phenomena was quantitatively studied. The current work provides fundamental 392 393 evidence which can be used to differentiate the aforementioned phases in STXM study of 394 calcium sulfoaluminate aqueous system. Besides providing reference for studying such 395 system in natural system, our work has implication on cement chemistry scientific 396 community by demonstrating novel methods to investigate unsolved topics, for example, 397 the retardation mechanism of calcium sulfate (hydrates) on the rapid hydration of  $C_{3}A_{3}$ , 398 which is critical to understanding the early age performance of fresh concrete mixture. 399 Another potential application of the current methodology lies in the hydration dynamics 400 of calcium sulfoaluminate cement, which is a promising alternative of ordinary Portland 401 cement to reduce the heavy carbon footprint of the cement industry (Justnes 2012). In 402 summary, we expect our work to raise general interest from the cement and concrete 403 research community, since STXM can be a powerful chemical and morphological probe 404 to study a rapidly evolving aqueous system given that the reference spectra are well 405 resolved.

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# **Figure captions**

549	Figure 1. Representation of the crystal field splitting of 3 <i>d</i> orbitals in cubic and
550	octahedral coordination states in Ca <sup>2+</sup> . Here, the central atoms (solid blue circles) are Ca
551	and the coordinated atoms (hollow circles) are oxygen. Descriptions of the various
552	parameters depicted are described in the text. Adapted from Burns (1993).
553	<b>Figure 2.</b> Schematic representation of a sample in the instrument chamber for Ca $L_{2,3}$ -
554	edge NEXAFS and STXM analysis (not to scale).
555	Figure 3. (a) Ca $L_{2,3}$ -edge NEXAFS spectra of ettringite, kuzelite, katoite, gypsum,
556	$C_3A_{cub}$ and $C_3A_{orth}$ , collected from the regions in (b)-(g). The absorption signals of the
557	dashed circle or straight lines in each image were averaged to yield the spectra. Insets are
558	spectra of $CaF_2$ (de Groot et al. 1990) and $Ca(OH)_2$ (Geng et al. 2015) as examples of
559	cubic and octahedral coordination, respectively.
560	Figure 4. Coordination environments of Ca in (a) katoite (Lager et al. 1987), (b)
561	ettringite (Moore and Taylor 1970), (c) kuzelite (Allmann 1977) and (d) gypsum (Comdi
562	et al. 2008)0, with rotation axes marked by dashed lines. The insets show color blocks
563	that illustrate the orientations of coplanar oxygen atoms viewed down the rotation axes.
564	Filled triangles represent three-fold axes. Arrow ends and filled ellipses represent two-
565	fold axes. Bond lengths are in Å.
566	Figure 5. Coordination environments of the six Ca sites in the C <sub>3</sub> A <sub>cub</sub> unit cell (Mondal

- and Jeffery 1975): (a) Ca1; (b) Ca2; (c) Ca3; (d) Ca4; (e) Ca5; and (f) Ca6. Angles are
- 568 marked with red labels.

569	Figure 6. Coordination environments of the five Ca sites in the C <sub>3</sub> A <sub>orth</sub> unit cell (Nishi
570	and Takeuchi 1975): (a) Ca1; (b) Ca2; (c) Ca3; (d) Ca4 and (e) Ca5.
571	Figure 7. Ca $L_{2,3}$ -edge NEXAFS spectra of (a) ettringite, (b) gypsum and (c) kuzelite in
572	samples of different thickness. The features located between 356 and 358 eV are
573	measurement artifacts.
574	<b>Figure 8.</b> (a) Sample thickness parameters indicated for a Ca $L_{2,3}$ -edge NEXAFS
575	spectrum of calcium monosulfoaluminate hydrate. (b) PIRs for ettringite, gypsum and
576	calcium monosulfoaluminate hydrate as functions of $\Delta OD$ .
577	
578	Tables
579	<b>Table 1.</b> Numbers and locations of Ca and Na atoms in the $C_3A_{cub}$ (Mondal and Jeffery

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1975) and C<sub>3</sub>A<sub>orth</sub> unit cells (Nishi and Takeuchi 1975).

		Ca1	Ca2	Ca3	Ca4	Ca5	Ca6
C <sub>3</sub> A <sub>cub</sub>	Number of atoms	4	4	8	8	24	24
	View from (-0.4, 0.7, 1.5) $z = \frac{x}{y}$				• <b>8</b> • <b>8</b> •	000000 000000 000000	
		Ca1	Ca2	Ca3	Ca4	Ca5	Na
-	Number of atoms	2 (occupancy 0.25)	8	8	8	8	4
C <sub>3</sub> A <sub>orth</sub>	View from (-1.5, 1, -0.24) $z^{Z}$ y		000				
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Figures





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Figure 3. (a) Ca  $L_{2,3}$ -edge NEXAFS spectra of ettringite, kuzelite, katoite, gypsum, C<sub>3</sub>A<sub>cub</sub> and C<sub>3</sub>A<sub>orth</sub>, collected from the regions in (b)-(g). The absorption signals of the dashed circle or straight lines in each image were averaged to yield the spectra. Insets are spectra of CaF<sub>2</sub> (de Groot et al. 1990) and Ca(OH)<sub>2</sub> (Geng et al. 2015) as examples of cubic and octahedral coordination, respectively.

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are in Å.



Figure 5. Coordination environments of the six Ca sites in the C<sub>3</sub>A<sub>cub</sub> unit cell cell
(Mondal and Jeffery 1975)0: (a) Ca1; (b) Ca2; (c) Ca3; (d) Ca4; (e) Ca5; and (f) Ca6.



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**Figure 6.** Coordination environments of the five Ca sites in the C<sub>3</sub>A<sub>orth</sub> unit cell (Nishi

617 and Takeuchi 1975): (a) Ca1; (b) Ca2; (c) Ca3; (d) Ca4 and (e) Ca5.

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**Figure 7.** Ca  $L_{2,3}$ -edge NEXAFS spectra of (a) ettringite, (b) gypsum and (c) kuzelite in samples of different thickness. The features located between 356 and 358 eV are

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measurement artifacts.



**Figure 8.** (a) Sample thickness parameters indicated for a Ca  $L_{2,3}$ -edge NEXAFS

627 spectrum of calcium monosulfoaluminate hydrate. (b) PIRs for ettringite, gypsum and

628 calcium monosulfoaluminate hydrate as functions of  $\Delta$ OD.

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