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3 **The influence of OH content on elastic constants of topaz**  
4 **[Al<sub>2</sub>SiO<sub>4</sub>(F,OH)<sub>2</sub>]**

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22 **ABSTRACT**

23 Topaz,  $\text{Al}_2\text{SiO}_4(\text{OH})_x\text{F}_{(2-x)}$ , might play a significant role in transporting water and fluorine  
24 into the Earth's interior at subduction zones. Seismological detection of topaz gives us  
25 insights into the transport of water and fluorine, and requires a thorough understanding  
26 of its elastic properties. The influence of OH content on elastic constants of topaz has not  
27 been fully understood, though experimental and theoretical studies have been done on  
28 topaz with various OH contents. We thus determined elastic constants of topaz for 5  
29 natural single-crystal specimens with different OH content ( $x=0.28\sim 0.72$ ) via the sphere  
30 resonance method at an ambient condition. Determined  $C_{11}$ ,  $C_{22}$ ,  $C_{44}$ ,  $C_{66}$ ,  $C_{12}$ ,  $C_{23}$  and  
31  $C_{31}$  increase with OH content while  $C_{33}$  and  $C_{55}$  decrease. For the change in OH molar  
32 content from 0.0 to 1.0, relatively large changes ( $>3.0\%$ ) are seen in  $C_{33}$  (8.0(6)%),  $C_{55}$   
33 (4.9(6)%) and  $C_{22}$  (3.1(7)%). The OH content dependence of elastic constants is  
34 qualitatively similar to that of theoretically determined values except for  $C_{11}$ . The  
35 theoretical value of  $C_{11}$  decreases as the OH molar content increases from 0.0 to 1.0  
36 whereas the experimental value of  $C_{11}$  slightly increases. Our determined elastic constants  
37 are significantly higher ( $>3\%$ ) than theoretically determined values, especially in  
38 diagonal components ( $C_{ii}$ ). The theoretical lower values must be related to the lattice  
39 parameters used in the theoretical study, which are systematically larger than measured

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40 lattice parameters. The theoretical approach should be modified to reproduce measured  
41 lattice parameters and lead to the agreement of theoretical and experimental elastic  
42 constants. Our results provide a clue to a better understanding of elasticity of topaz and a  
43 basis for the seismological detection of subducted oceanic sediments.

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45 **Key words:** topaz, elastic constants, sphere resonance method, OH content dependence

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## INTRODUCTION

48 Topaz is an orthosilicate mineral with ideal formula  $\text{Al}_2\text{SiO}_4(\text{OH})_x\text{F}_{(2-x)}$ . It is usually  
49 found as an accessory mineral in fluorine-rich granitic rocks related to the formation of  
50 pneumatolithic/hydrothermal deposits (Pichavant and Manning, 1984; Taylor, 1992;  
51 Taylor and Fallick, 1997), or in rock formations that experience ultra-high-pressure  
52 metamorphism (Zhang et al., 2002; Alberico et al., 2003). High pressure and temperature  
53 experiments on  $\text{Al}_2\text{O}_3$ - $\text{SiO}_2$ - $\text{H}_2\text{O}$  ternary system, which represents subducted oceanic  
54 sediments, have shown that hydrous aluminosilicate minerals like phase pi  
55 ( $\text{Al}_2\text{Si}_2\text{O}_7(\text{OH})_3$ ) and topaz-OH can retain water up to ~8 GPa (Wunder et al., 1993a, b;  
56 Schreyer, 1995). Later, high pressure experiments on a synthetic oceanic sediment  
57 revealed that topaz-OH is stable at pressures from 8 to 12 GPa and transforms to another

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58 hydrous aluminosilicate mineral phase Egg ( $\text{AlSiO}_3(\text{OH})$ ) at pressures beyond 12 GPa,  
59 which can carry water to the mantle transition zone (Ono, 1998). The difference in the  
60 stability field of topaz-OH reflects the difference in the chemical composition of their  
61 starting materials. The hydroxy-rich topaz, thus, might play a significant role in  
62 transporting water into the Earth's interior (Mookherjee et al., 2016; Ulian and Valdrè,  
63 2017; Sema and Watanabe, 2017; Tennakoon et al., 2018).

64 Topaz might also be a major carrier of fluorine into the Earth's interior (Ulian and  
65 Valdrè, 2017; Sema and Watanabe, 2017). Halogens, especially fluorine (320 ppm) and  
66 chlorine (850 ppm), are relatively abundant in oceanic sediments (Carpenter, 1969;  
67 Muramatsu and Wedepohl, 1998). Fluorine ions can substitute for hydroxy groups in  
68 many hydrous minerals because of their similar ionic charges and radii (Pyle and Mather,  
69 2009). Seismological detection of topaz gives us insights into the transport of water and  
70 fluorine in subduction zones. It requires a thorough understanding of elastic properties of  
71 topaz with a wide range of OH content.

72 Elastic constants of topaz were firstly determined by Voigt (1888) through bending  
73 and twisting of single-crystal prisms. Later, Haussühl (1993), Sema and Watanabe (2017)  
74 and Tennakoon et al. (2018) determined elastic constants of topaz with various chemical  
75 compositions through the resonance method. Though the temperature dependence of

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76 elastic constants was investigated by Sema and Watanabe (2017) and Tennakoon et al.  
77 (2018), these measurements have been conducted only at atmospheric pressure. The  
78 pressure dependence of elastic constants has been studied through the *first principles*  
79 calculation by Mookherjee et al. (2016) and Ulian and Valdrè (2017). Mookherjee et al.  
80 (2016) determined the elastic constants of topaz-OH while Ulian and Valdrè (2017)  
81 evaluated those of topaz with systematically changed compositions ( $x=0.0\sim 2.0$ ).

82 The influence of OH content on elastic constants of topaz has not been fully  
83 understood, though both experimental and theoretical studies have been done on elastic  
84 constants of topaz with various OH contents (Table 1). Determined elastic constant  $C_{22}$  is  
85 shown in Figure 1 as a function of OH molar content. Experimental works were conducted  
86 at ambient temperatures ( $\sim 300$  K) while theoretical works determined elastic constants at  
87 0 K. It is difficult to see a systematic relationship between the experimentally determined  
88 elastic constant and OH molar content. Though the results of Ulian and Valdrè (2018)  
89 show a systematic change in  $C_{22}$  with OH molar content, there is a remarkable  
90 discrepancy between two theoretical studies on topaz-OH and there are significant  
91 discrepancies between experimental and theoretical values. These discrepancies cannot  
92 be explained by the temperature difference between experimental and theoretical works.  
93 Lower elastic constants are generally expected at higher temperatures.

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94           Theoretical and experimental studies should be combined to understand the  
95 influence of OH content on elasticity of topaz. Theoretical studies are useful for  
96 understanding the influence of OH content on elasticity, since the OH content in topaz  
97 can be relatively easily controlled. On the other hand, it is difficult to find natural topaz  
98 single-crystal samples with wide range of the OH content, which depends on the mineral  
99 formation environment (Zhang et al., 2002). However, the experimental determination of  
100 elastic constants is still needed to examine theoretical studies. We have thus conducted  
101 experimental determination of elastic constants of topaz for 5 different OH contents  
102 ( $x=0.28\sim 0.72$ ) by the sphere resonance method and compared them with results of  
103 previous works.

104

105

## SPECIMENS

106           Five natural single-crystals of topaz were selected ( $x = 0.28, 0.38, 0.45, 0.66$  and  $0.72$ ).  
107 Locality, chemical formula, lattice parameters and unit cell volume are listed in Table 2.  
108 Chemical compositions were determined by using an EPMA (JOEL, JXA-8500F) at  
109 JAMSTEC Yokosuka Headquarters. Topaz has edge-sharing Al-octahedral units and  
110 corner-sharing Si-tetrahedral units (Figure S1). The orthorhombic symmetry with *Pbnm*  
111 space group was confirmed and lattice parameters were determined by using a single

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112 crystal X-ray diffractometer (Rigaku, VariMax RAPID-DW). Lattice parameters  $a$  and  $b$   
113 increase with increasing OH content, while  $c$  decreases. The unit cell volume increases  
114 with OH content.

115 Each crystal was shaped into a sphere (Figure 2) by the two-pipe method (Bond, 1954).  
116 The average and standard deviation of diameter for 30 measurements are shown in Table  
117 2. The ratio of the standard deviation to the average was smaller than 0.04% for all  
118 specimens. The density was calculated from the mass and volume of spheres. It decreases  
119 with OH content, reflecting the increase in unit cell volume.

120 Sample FKO is the sphere sample used in Sema and Watanabe (2017). Measurements  
121 were remade on this sample and the diameter was corrected to be 6.487(3) mm by using  
122 a calibrated micrometer. The OH molar content,  $x$  was redetermined to be 0.45.

123

124

#### SPHERE RESONANCE METHOD

125 The sphere resonance method (e.g., Soga and Anderson, 1967) was employed to  
126 determine elastic constants. The measurement system is schematically shown in Figure  
127 S2. A sphere specimen was held between two ultrasonic transducers (OLYMPUS,  
128 V156RM). A function synthesizer (NF Corporation, WF1944B) applied a sinusoidal  
129 signal ( $V_{pp} = 10.0$  V) to the driving transducer and the excited vibration in the specimen

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130 was picked up by the receiving transducer. The lower transducer was placed on a balance  
131 to control the specimen-holding force. Measurements were conducted at an ambient  
132 temperature (22°C). The transducers, specimen and balance were placed in a container to  
133 reduce the temperature change. The amplitude of the excited vibration was measured by  
134 a lock-in-amplifier (Stanford Research Systems, SR884). The frequency range of the  
135 sinusoidal signal is shown in Table S1. The frequency was changed by 20 Hz.

136 Elastic constants were determined to minimize the difference between measured and  
137 calculated resonance frequencies, which was characterized by the root mean square  
138 relative error given by

$$139 \quad D = \sqrt{\frac{1}{n} \sum_i \left( \frac{f_i^{meas} - f_i^{cal}}{f_i^{cal}} \right)^2},$$

140 where  $f_i^{meas}$  and  $f_i^{cal}$  are measured and calculated resonance frequencies, respectively. The  
141 xyz algorithm (Vissher et al., 1991) was applied to calculate resonance frequencies for a  
142 given set of elastic constants. Uncertainties in elastic constants are determined by the  
143 least-square method as described in Sema and Watanabe (2017).

144

#### 145 **ELASTIC CONSTANTS OF TOPAZ**

146 A typical resonance spectrum is shown in Figure S3. It should be noted that the  
147 vibration of a specimen is not a free vibration. The specimen-holding force  $F$  slightly



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148 deforms the specimen to change the resonant frequencies from those of free vibration. In  
149 addition, the frequency of a maximum amplitude of excited vibration is different from the  
150 resonant frequency of the deformed specimen with normalized difference of order  $1/(2Q^2)$ ,  
151 where  $Q$  is the quality factor of the resonance (e.g., Migliori and Sarrao, 1997). The  
152 dissipation of acoustic energy is characterized by  $1/Q$ . In our measurements,  $Q$  is  
153 evaluated by the ratio of the frequency of a maximum to the full width at half maximum  
154 of the square of amplitude. It is  $(1\sim 5)\times 10^4$  for each specimen and comparable to the values  
155 reported by Tennakoon et al. (2018). Thus, the difference between the frequency of a  
156 maximum amplitude and the resonant frequency of the deformed specimen is negligible.  
157 The frequency of a maximum amplitude can be treated as a resonant frequency of the  
158 deformed specimen. Resonance frequencies were measured for 5 different specimen-  
159 holding forces (2.0-10.0 gf), and the resonance frequency of free vibration was estimated  
160 by extrapolating measured frequencies to  $F=0$  as Sema and Watanabe (2017). Estimated  
161 resonance frequencies of free vibration are summarized in Table S3. Elastic constants  
162 were determined by using 42 resonance frequencies. Tennakoon et al. (2018) noted that  
163 the elastic constants are determined stably if the number of resonance modes considered  
164 is more than 35.

165 Elastic constants of five topaz specimens are shown in Figure 3 as a function of OH

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166 molar content and listed in Table S4. The difference between measured and calculated  
167 resonance frequencies were smaller than 0.2% for all vibration modes (Table S3). Our  
168 determined  $C_{11}$ ,  $C_{22}$ ,  $C_{44}$ ,  $C_{66}$ ,  $C_{12}$ ,  $C_{23}$  and  $C_{31}$  increase with OH content, while  $C_{33}$  and  
169  $C_{55}$  decrease. The elastic constants of Sample FKO ( $x=0.45$ ) differ from those in Sema  
170 and Watanabe (2017) by 0.4% at most. A linear equation  $y = a_1 + a_2x$  was fitted to the  
171 relation between the elastic constant ( $y$ ) and OH content ( $x$ ), and estimated  $a_1$  and  $a_2$  are  
172 shown in Table S5. The ratio  $a_2/a_1$  is the relative change in an elastic constant for the  
173 increase in OH molar content from 0.0 to 1.0. Relatively large changes (>3.0%) are seen  
174 in  $C_{33}$  (8.0(6)%),  $C_{55}$  (4.9(6)%) and  $C_{22}$  (3.1(7)%).

175 Elastic constants reported in previous studies are also shown in Figure 3 for  
176 comparison. Haussül (1993) and Tennakoon et al. (2018) determined elastic constants of  
177 topaz single-crystals by the resonance method for  $x=0$  and 0.58, respectively. Though  
178 there are some deviations, experimentally determined elastic constants show consistent  
179 changes:  $C_{11}$ ,  $C_{22}$ ,  $C_{44}$ ,  $C_{66}$ ,  $C_{12}$ ,  $C_{23}$  and  $C_{31}$  increase, while  $C_{33}$  and  $C_{55}$  decrease. Ulian  
180 and Valdrè (2017) determined elastic constants for a wide range of OH content by *first*  
181 *principles* calculation. Though their theoretical values are significantly lower than  
182 experimental values, especially in diagonal components, the OH content dependence of  
183 theoretical values is similar to that of experimental values, except for  $C_{11}$ . A linear

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184 equation was also fitted to the relation between the theoretical elastic constant ( $\gamma$ ) and OH  
185 molar content ( $x$ ) for  $x \leq 1.0$ . The slope  $a_2$  for the theoretical values is also listed in Table  
186 S5 for comparison. Similar slopes are obtained for theoretical and experimental values of  
187  $C_{22}$ ,  $C_{44}$ ,  $C_{55}$ ,  $C_{66}$  and  $C_{12}$ . The largest change is seen in  $C_{33}$  both in theoretical and  
188 experimental values. The magnitude of change in the theoretical value of  $C_{33}$  is almost  
189 twice as large as that in the experimental value. The theoretical value of  $C_{11}$  decreases as  
190 the OH molar content increases from 0.0 to 1.0, whereas the experimental value of  $C_{11}$   
191 slightly increases.

192

193

## DISCUSSION

### 194 **Discrepancy between experimental and theoretical elastic constants**

195 Elastic constants determined by the *first principles* calculation (Ulian and Valdrè,  
196 2017) are significantly lower than experimentally determined elastic constants, though  
197 theoretical and experimental values show similar dependence on OH content except for  
198  $C_{11}$  (Figure 3). Theoretical values are determined at the athermal condition ( $T=0$  K), as  
199 already pointed out in Introduction, while experimental values at ambient temperature  
200 ( $T=295$  K). The difference in temperature cannot explain the discrepancies in elastic  
201 constants, since elastic constants will be lower at higher temperatures due to

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202 anharmonicity. The theoretical lower elastic constants must be related to the lattice  
203 parameters in the *first principles* calculation. Lattice parameters of topaz in previous and  
204 our studies are shown in Figure 4 as a function of OH content. The lattice parameters in  
205 Ulian and Valdrè (2017) are 1% larger than measured values in *a*, 0.8% in *b* and 1% in *c*.  
206 The larger lattice parameters are attributed to the computational approach used in the *first*  
207 *principles* calculation. The hybrid exchange functional (B3LYP), which was used through  
208 the density functional theory, generally overestimates unit-cell volumes (Ulian and Valdrè,  
209 2017). The theoretical approach should be modified to reproduce measured lattice  
210 parameters. It will lead to the agreement between theoretically and experimentally  
211 determined elastic constants.

212 Different trends of  $C_{11}$  are seen between experimental and theoretical values. Our  
213 experimental value slightly increases ( $1 \pm 2$  GPa/mol) for the change in OH content from  
214 0.28 to 0.72, whereas the theoretical value significantly decreases (-11 GPa/mol) for the  
215 change in OH content from 0.0 to 1.0. Previous experimental values also show an increase  
216 with OH content. This difference might also be reconciled by decreasing lattice  
217 parameters in theoretical studies.

218

219 **Anisotropy in compressibility**

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220 Linear compressibility is calculated from determined elastic constants and compared  
221 with results of previous studies. Komatsu et al. (2003) and Gatta et al. (2006) determined  
222 isothermal linear compressibility through hydrostatic compression tests. Ulian and Valdrè  
223 (2017) calculated isothermal linear compressibility from the lattice parameters at different  
224 pressures which were obtained through the *first principles* simulation. Tennakoon et al.  
225 (2018) evaluated adiabatic linear compressibility from the elastic constants determined  
226 by the resonance method. Compressibility along the  $i$ -th axis  $\beta_i$  is related to elastic  
227 compliance  $S_{ij}$  as

$$228 \quad \beta_i = S_{i1} + S_{i2} + S_{i3}$$

229 Adiabatic elastic compliance is calculated from determined elastic stiffness. Determined  
230 adiabatic linear compressibility is listed in Table S6 and shown in Figure 5. Adiabatic  
231 linear compressibilities which were calculated from the elastic constants determined by  
232 Haussühl (1993) are also shown in Figure 5.

233 The results on linear compressibility are qualitatively consistent with those in  
234 previous studies, though there are small discrepancies in magnitude between previous and  
235 our studies. The most compressible axis is the  $c$ -axis, and the least the  $b$ -axis. The  
236 compressibility along the  $a$ -axis is slightly lower than that along the  $c$ -axis. The  
237 compressibility along the  $c$ -axis increases with OH content  $((0.22 \pm 0.06) \times 10^{-3} \text{ GPa}^{-1} \text{ mol}^{-1}$

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238 <sup>1</sup>) while that along the *b*-axis decreases  $((-0.11\pm 0.06)\times 10^{-3} \text{ GPa}^{-1}\text{mol}^{-1})$ . That along the *a*-  
239 axis shows no significant change  $((-0.04\pm 0.08)\times 10^{-3} \text{ GPa}^{-1}\text{mol}^{-1})$ . These trends are  
240 broadly consistent with those of the *first principles* calculation (Uljan and Valdrè, 2017).  
241 The compressibility ratio  $\beta_a/\beta_b$  is 1.41~1.45, and the ratio  $\beta_c/\beta_b$  1.48~1.60. Both ratios  
242 increase with OH content. The approaching of  $\beta_c$  to  $\beta_a$  at lower OH content is also seen  
243 in theoretical results. The higher compressibility along the *c*-axis is basically due to the  
244 compressible “tetrahedra-free zones” which are perpendicular to the *c*-axis (Gatta et al.,  
245 2006). They reported that the SiO<sub>4</sub> tetrahedron is less compressible than the AlO<sub>4</sub>F<sub>2</sub>  
246 octahedron. There are other tetrahedra-free weak zones which are subperpendicular to the  
247 *a*-axis (Figure 6). These weak zones contribute to higher compressibility along the *a*-axis.

248 The isothermal compressibility obtained from hydrostatic compression tests is quite  
249 close to our adiabatic compressibility. It is due to the low temperature ( $T < 300 \text{ K}$ ) where  
250 the oscillation of atoms around their equilibrium positions is nearly harmonic not to cause  
251 significant thermal expansion (e.g., Poirier, 1991).

252

### 253 **Anisotropy in seismic velocities**

254 Seismic velocities in a topaz single crystal (MC,  $x=0.28$ ) at an ambient condition are  
255 shown in Figure 7. Velocities in various propagation directions are calculated as solutions

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256 of the Christoffel equation (e.g., Mainprice, 2007). Compressional-wave velocity ( $V_P$ )  
257 shows relatively large directional variation (Figure 7(a)). The anisotropy parameter  $AV_P$   
258 is 11.8%, which is comparable to that of enstatite (Chai et al., 1997b). The highest velocity  
259 (9.95 km/s) is in the direction between the  $a$ - and  $b$ -axes, and the lowest (8.84 km/s) in  
260 the  $a$ -axis direction. Compressional-wave velocity is relatively low (<9.5 km/s) in the  $ca$ -  
261 plane.

262 In addition to the velocity variation with propagation direction, shear-wave velocity  
263 changes with the polarization direction. In a given propagation direction, the highest and  
264 lowest velocities are called the fast shear-wave velocity and slow shear-wave velocity,  
265 respectively. The fast shear-wave velocity ( $V_{S1}$ ) is high in the  $bc$ -plane and the  $a$ -axis  
266 direction, while low in the direction between the  $a$ - and  $b$ -axes and that between the  $a$ -  
267 and  $c$ -axes (Figure 7(c)). The slow shear-wave velocity ( $V_{S2}$ ) is high in the  $a$ -axis  
268 direction, while low in the direction between  $a$ - and  $b$ -axes (Figure 7(d)). The polarization  
269 anisotropy, which is characterized by  $AV_S$ , is remarkable (12~14%) in the direction  
270 between the  $a$ - and  $b$ - axes (Figure 7(b)).

271 Topaz single crystals with higher OH content show similar directional variations in  
272 seismic velocities. The directions of the highest and lowest velocities ( $V_P$ ,  $V_{S1}$ ,  $V_{S2}$ ) and  
273 the maximum  $AV_S$  show no changes with OH content. The anisotropy parameters,  $AV_P$

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274 and the maximum  $AV_s$  are shown in Figure S4 as a function of OH content.  $AV_P$  and the  
275 maximum  $AV_s$  show only slight changes for the increase of OH content from 0.28 to 0.72,  
276 reflecting the weak OH dependence of elastic constants. The increase in  $AV_P$  and decrease  
277 in the maximum  $AV_s$  are attributed to the increase in  $C_{22}$  and decrease in  $C_{55}$ , respectively  
278 (Figure 3). Similar anisotropy parameters are reported for the topaz with the OH content  
279 of 0.58 (Tennakoon et al., 2018) and calculated for the fluorine end member topaz  
280 (Hausühl, 1993). The elastic constants determined by Ulian and Valdrè (2017) give also  
281 similar anisotropy parameters except for the OH content of 1.0. The large value of  $AV_P$  is  
282 attributed to the decrease in  $C_{11}$  (Figure 3).

283

#### 284 **Isotropic Moduli**

285 The bulk ( $K$ ) and shear ( $G$ ) moduli that an isotropic polycrystalline aggregate will  
286 show can be evaluated from single-crystal elastic constants through the Voigt and Reuss  
287 averaging schemes. Voigt (1928) assumed uniform strain throughout the aggregate, while  
288 Reuss (1929) uniform stress. The Voigt average gives an upper bound, and the Reuss  
289 average yields a lower bound for an isotropic modulus. Calculated isotropic moduli are  
290 shown in Figure 8 and Table S7. The Hill average, which is the arithmetic mean of the  
291 Voigt and Reuss averages, is also shown. Figure 8 also shows bulk and shear moduli



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292 calculated from the previously reported elastic constants (Hausühl, 1993; Ulian and  
293 Valdrè, 2017; Tennakoon et al., 2018). The lower theoretically determined elastic  
294 constants of Ulian and Valdrè (2017) yield lower isotropic moduli.

295 The bulk and shear moduli slightly decrease with increasing OH content (Figure 8(a),  
296 (b)). The OH content dependence of the bulk modulus is dominated by that of  $C_{33}$ , which  
297 shows the strongest dependence on the OH content among elastic constants related to the  
298 bulk modulus (Figure 3). On the other hand, the OH dependence of  $C_{33}$  and  $C_{55}$  dominates  
299 that of the shear modulus.

300 The difference between the Voigt and Reuss averages is less than 2% for all single-  
301 crystal samples. The magnitude of the difference between the two averages reflects the  
302 degree of anisotropy in elasticity of a single crystal. For strongly anisotropic olivine  
303 ( $AV_P=24.3\%$ ; Webb, 1989), the difference is 3.2% and 3.9% for the bulk and shear moduli,  
304 respectively. For moderately anisotropic enstatite ( $AV_P=13.6\%$ ; Chai et al., 1997b), the  
305 difference is 1.5% and 1.7% for the bulk and shear moduli. There is little difference  
306 between the two averages for nearly isotropic pyrope (Chai et al., 1997a).

307 Compressional- and shear-wave velocities calculated from isotropic moduli are  
308 shown in Figure 8 as a function of OH content. Seismic velocities calculated from the  
309 previously reported elastic constants are also shown for comparison. Owing to the lower

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310 density, the lower isotropic moduli of Ulian and Valdrè (2017) yield slightly higher  
311 compressional-wave velocity. The shear-wave velocity is almost the same as those  
312 calculated from experimentally determined elastic constants. A linear equation is fitted to  
313 the relation between the velocity calculated from the Hill averages and OH content. Based  
314 on this study, the compositional dependence of velocities is  $dV_P/dx_{OH} = 0.041(1)$  km/s  
315 and  $dV_S/dx_{OH} = 0.03(1)$  km/s. On the other hand, the compositional dependence of  
316 velocities of Ulian and Valdrè (2017) for  $0.0 \leq x \leq 1.0$  is  $dV_P/dx_{OH} = -0.072(6)$  km/s and  
317  $dV_S/dx_{OH} = -0.03(1)$  km/s. The difference in the compositional dependence of velocity is  
318 mostly attributed to that in the compositional dependence of  $C_{11}$  and  $C_{33}$  between Ulian  
319 and Valdrè (2017) and this study (Figure 3).

320

321

## IMPLICATIONS

322 Our study has confirmed the discrepancy between experimentally and theoretically  
323 determined elastic constants that was described in Introduction. The discrepancy should  
324 be a clue to a better understanding of elasticity of topaz. The theoretical lower elastic  
325 constants must be related to the lattice parameters, which are systematically larger than  
326 measured values. The theoretical approach should be modified to reproduce the measured  
327 lattice parameters. It will lead to the agreement between experimentally and theoretically

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328 determined elastic constants and the improvement of the *first principles* calculation. This  
329 study will boost our understanding of elasticity of minerals.

330 Our study will be a basis for the seismological detection of subducted oceanic crust.  
331 Sema and Watanabe (2017) calculated isotropic seismic velocities in topaz (FKO) of  
332  $x=0.45$  at high pressure (10 GPa) and temperature (300~1400 K), and compared them  
333 with velocities in olivine and garnet, which are major components of the overlying wedge  
334 mantle and the underlying subducted oceanic crust, respectively. The compressional-  
335 wave velocity in topaz is distinctly higher ( $\sim 1$  km/s) than that in olivine. On the basis of  
336 the weak dependence of isotropic seismic velocities on OH content (Fig.12(b)), at least  
337 for the OH content of  $x=0.0\sim 1.0$ , topaz should show distinctly higher compressional-wave  
338 velocity than olivine. There might be a strong velocity contrast between the overlying  
339 mantle and the thin sediment-origin layer at the depth around 300 km. A seismological  
340 technique such as the receiver function technique (e.g., Kawakatsu and Watada, 2007)  
341 should be applied to detect a thin layer of topaz. It will give us insights into the transport  
342 of water and fluorine in subduction zones.

343

344

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439

440

441 **List of figure captions**

442 Figure 1

443 Reported elastic constant  $C_{22}$  of topaz as a function of OH molar content. Experimental  
444 values (Hausühl, 1993; Sema and Watanabe, 2017; Tennakoon et al., 2018) were  
445 determined at ambient temperatures (~300 K) while theoretical values (Mookherjee et al.,  
446 2016; Ulian and Valdrè, 2017) at 0 K. The zero-pressure elastic constant of Mookherjee et  
447 al. (2016) was interpolated from reported values of orthorhombic topaz-OH at -3.5 and  
448 1.2 GPa.

449

450 Figure 2

451 Five specimens of topaz single crystal. (a) Original shape. (b) Sphere specimen.

452

453 Figure 3

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454 Determined elastic constants of five topaz specimens as a function of OH molar content.  
455 Results of previous works are also shown for comparison. Experimental values (Hausühl,  
456 1993; Tennakoon et al., 2018) were determined at ambient temperatures (~300 K) while  
457 theoretical values (Mookherjee et al., 2016; Ulian and Valdrè, 2017) at 0 K. The zero-  
458 pressure elastic constant of Mookherjee et al. (2016) was interpolated from reported values  
459 of orthorhombic topaz-OH at -3.5 and 1.2 GPa.

460

461 Figure 4

462 Lattice parameters as a function of OH molar content. This study, Komatsu et al. (2003),  
463 Gatta et al. (2006) and Tennakoon et al. (2018) determined lattice parameters by X-ray  
464 diffraction method at ambient temperatures while Mookherjee et al. (2016) and Ulian and  
465 Valdrè (2017) by the *first principles* calculation at athermal condition ( $T = 0$  K). A linear  
466 equation was fitted to the relation between the lattice parameter and OH content in this  
467 study and Ulian and Valdrè (2017) for  $0 \leq x \leq 1$ . For this study,  $a = 4.636(2) + 0.036(4)$   
468  $x_{\text{OH}}$ ,  $b = 8.758(7) + 0.10(1) x_{\text{OH}}$ ,  $c = 8.3962(9) - 0.014(2) x_{\text{OH}}$ . For Ulian and Valdrè (2017),  
469  $a = 4.701(1) + 0.020(2) x_{\text{OH}}$ ,  $b = 8.8698(6) + 0.094(1) x_{\text{OH}}$ ,  $c = 8.504(1) - 0.031(2) x_{\text{OH}}$ .

470

471 Figure 5

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472 Linear compressibility calculated from determined elastic constants as a function of OH  
473 molar content. The compressibility reported in previous studies (Komatsu et al., 2003;  
474 Gatta et al., 2006; Ulian and Valdrè, 2017; Tennakoon et al., 2018) are also shown. The  
475 values of Haussühl (1993) were calculated from the reported elastic constants. The solid,  
476 grey and open symbols are for the directions of  $a$ ,  $b$  and  $c$ -axis, respectively.

477

478 Figure 6

479 Crystal structure of topaz viewed along the  $c$ -axis. There is a tetrahedra-free zone  
480 subperpendicular to the  $a$ -axis, which contributes to relatively high compressibility along  
481 the  $a$ -axis. This figure was created with a free software VESTA (Momma and Izumi,  
482 2011) by using structural parameters determined by Ulian and Valdrè (2017).

483

484 Figure 7

485 Compressional-wave velocity ( $V_P$ ) and shear-wave velocities ( $V_{S1} > V_{S2}$ ) in a topaz  
486 single-crystal MC ( $x=0.28$ ) calculated from determined elastic constants. The magnitude  
487 of polarization anisotropy ( $AV_s$ ) and polarization plane of the fast shear-wave are also  
488 shown. Careware software developed by Mainprice (1990) was used for calculation and  
489 stereographic projection.

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490

491 Figure 8

492 (a) Bulk and (b) shear moduli as a function of OH molar content. Isotropic moduli are  
493 calculated from the determined elastic constants through the Voigt, Reuss and Hill  
494 averaging schemes. The solid, open and grey symbols are for the Voigt, Reuss and Hill  
495 averaging schemes, respectively. The values calculated from previous studies (Haussühl,  
496 1993; Ulian and Valdrè, 2017; Tennakoon et al., 2018) are also shown for comparison. (c)  
497 Compressional- and (d) shear-wave velocities calculated from isotropic moduli. Symbols  
498 are used as in (a) and (b).

**Table 1. Previous studies for detemining elatsic constants of topaz**

	Reference	Method
Experimental	Hausühl (1993)	Rectangular-resonance method
	Sema and Watanabe (2017)	Sphere-resonance method
	Tennakoon et al. (2018)	Rectangular-resonance method
Theoretical	Mookherjee et al. (2016)	<i>First principles</i> calculation
	Ulian and Valdrè (2017)	<i>First principles</i> calculation

Pressure (GPa)	Temperature (K)	Chemical formula
0	293	$\text{Al}_2\text{SiO}_4\text{F}_2$
0	291.9	$\text{Al}_2\text{SiO}_4\text{F}_{1.56}(\text{OH})_{0.42}$
0	293~973	$\text{Al}_2\text{SiO}_4\text{F}_{1.42}(\text{OH})_{0.58}$
-4.9~34.1	0	$\text{Al}_2\text{SiO}_4(\text{OH})_2$
0~55	0	$\text{Al}_2\text{SiO}_4\text{F}_2$ $\text{Al}_2\text{SiO}_4\text{F}_{1.75}(\text{OH})_{0.25}$ $\text{Al}_2\text{SiO}_4\text{F}_{1.50}(\text{OH})_{0.50}$ $\text{Al}_2\text{SiO}_4\text{F}_{1.00}(\text{OH})_{1.00}$ $\text{Al}_2\text{SiO}_4\text{F}_{0.50}(\text{OH})_{1.50}$ $\text{Al}_2\text{SiO}_4(\text{OH})_2$

**Table 2. Locality, chemistry and physical properties of specimens**

Specimen	MC	MGK
Locality	Maynard's Claim, Thomas Range, Juab county, Utah state, USA	Mogok township, Pyin-Oo-Lwin district, Mandalay region, Myanmar
Chemical formula	$\text{Al}_{2.00}\text{SiO}_4\text{F}_{1.72}(\text{OH})_{0.28}$	$\text{Al}_{2.00}\text{SiO}_4\text{F}_{1.72}(\text{OH})_{0.38}$
Lattice parameters and unit-cell volume		
<i>a</i> (Å)	4.64719(8)	4.64982(8)
<i>b</i> (Å)	8.79034(16)	8.79869(16)
<i>c</i> (Å)	8.39215(15)	8.39070(15)
<i>V</i> (Å <sup>3</sup> )	342.823(11)	343.283(11)
Diameter of sphere		
Average (N=30) (mm)	4.422(3)	4.588(3)
Standard deviation (N=30) (mm)	0.002	0.002
Standard deviation/Average (%)	0.04	0.04
Density (g/cm <sup>3</sup> )	3.582(8)	3.572(7)

FKO	OP	KTL
Fukuoka, Nakatsugawa city, Gifu prefecture, Japan	Ouro Preto, Minas Gerais state, Brazil	Katlang, Tharparkar district, Sindh province, Pakistan
$\text{Al}_{1.97}\text{SiO}_4\text{F}_{1.55}(\text{OH})_{0.45}$	$\text{Al}_{1.97}\text{SiO}_4\text{F}_{1.34}(\text{OH})_{0.66}$	$\text{Al}_{1.97}\text{SiO}_4\text{F}_{1.34}(\text{OH})_{0.72}$
4.64956(8)	4.66016(8)	4.66228(14)
8.79758(16)	8.82724(16)	8.8346(3)
8.39106(15)	8.38715(15)	8.3863(2)
343.235(11)	345.017(11)	345.427(17)
6.487(3)	2.666(3)	4.078(3)
0.002	0.001	0.001
0.02	0.04	0.03
3.568(5)	3.562(12)	3.536(8)





Figure 1

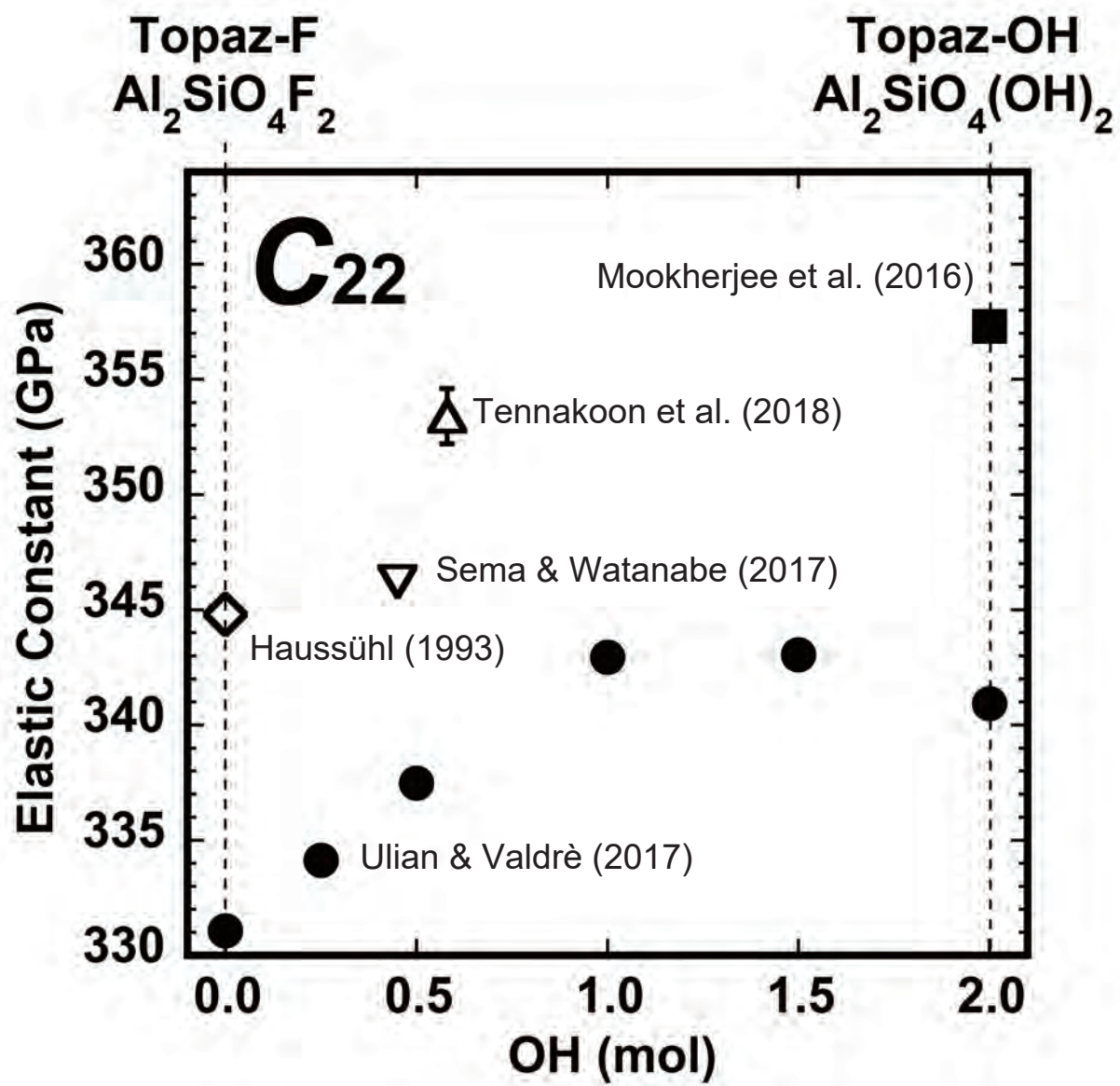
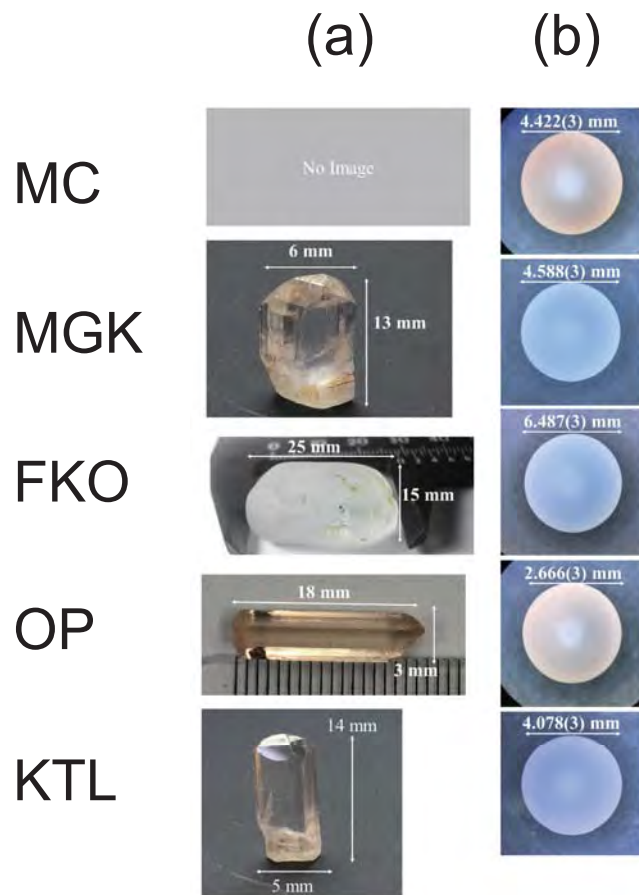


Figure 2



# Figure 3

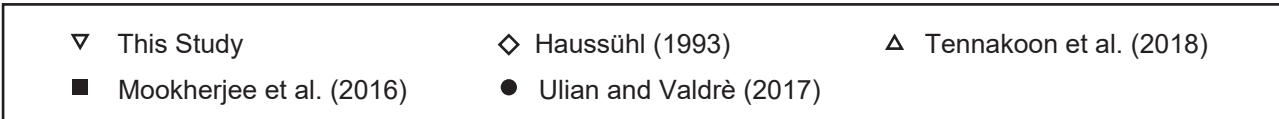
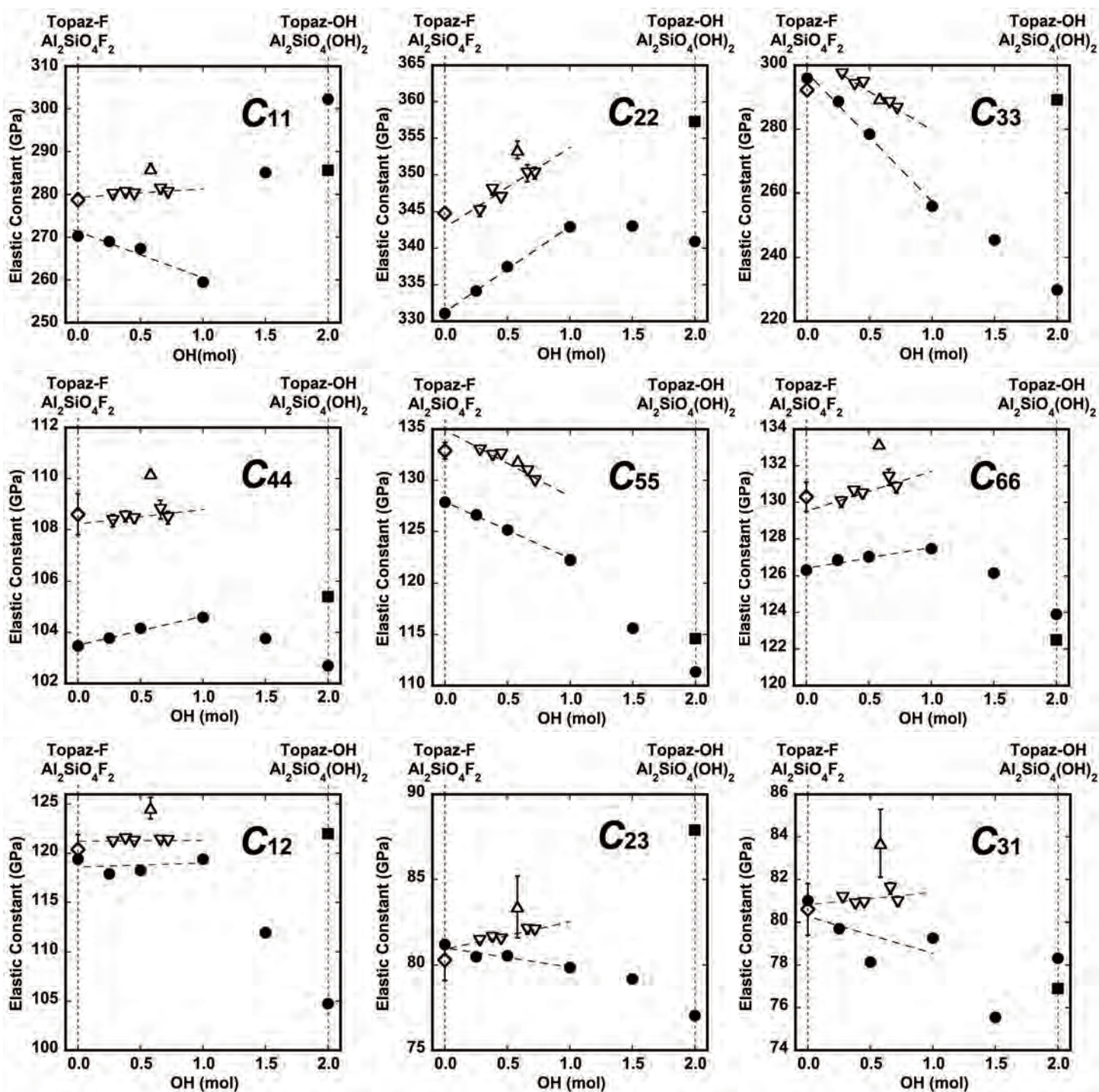


Figure 4

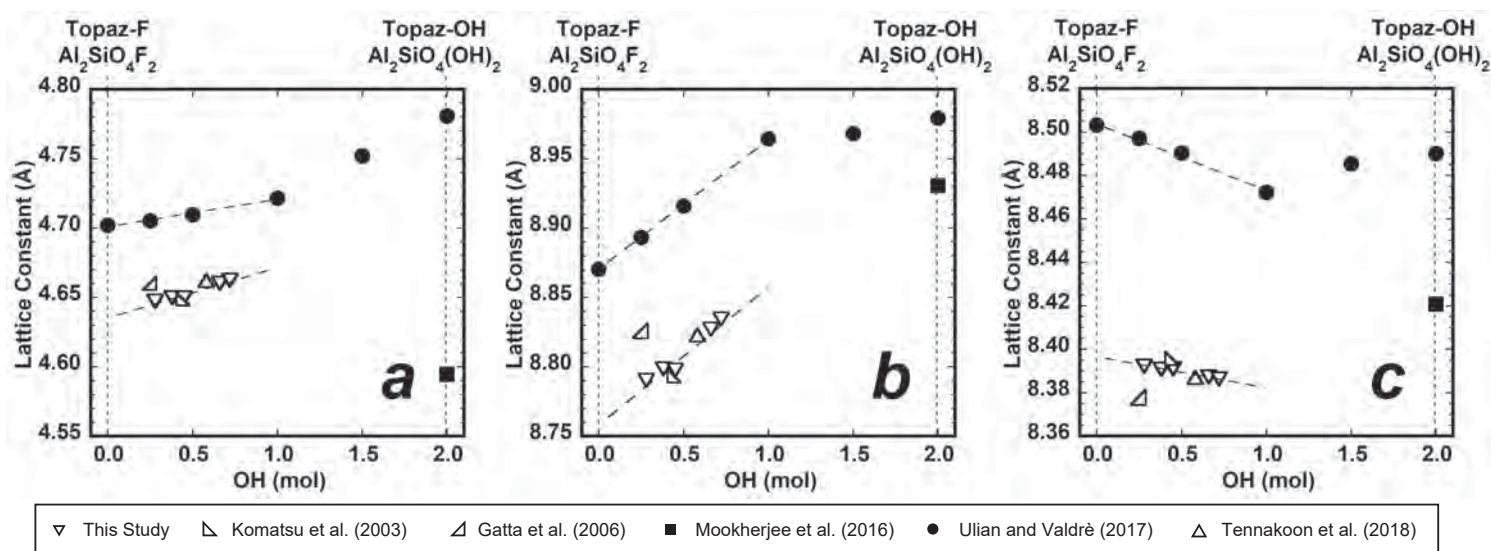
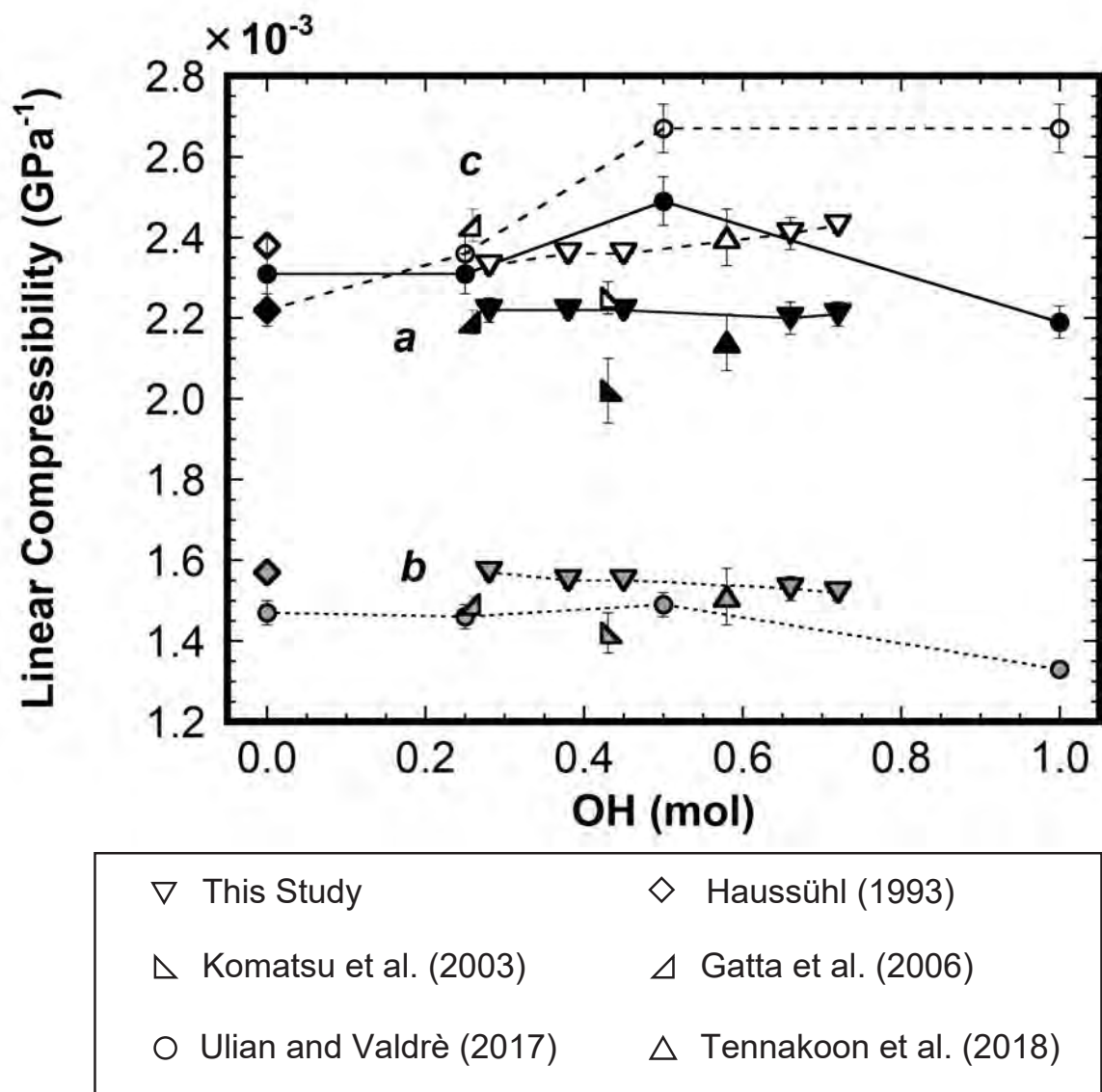
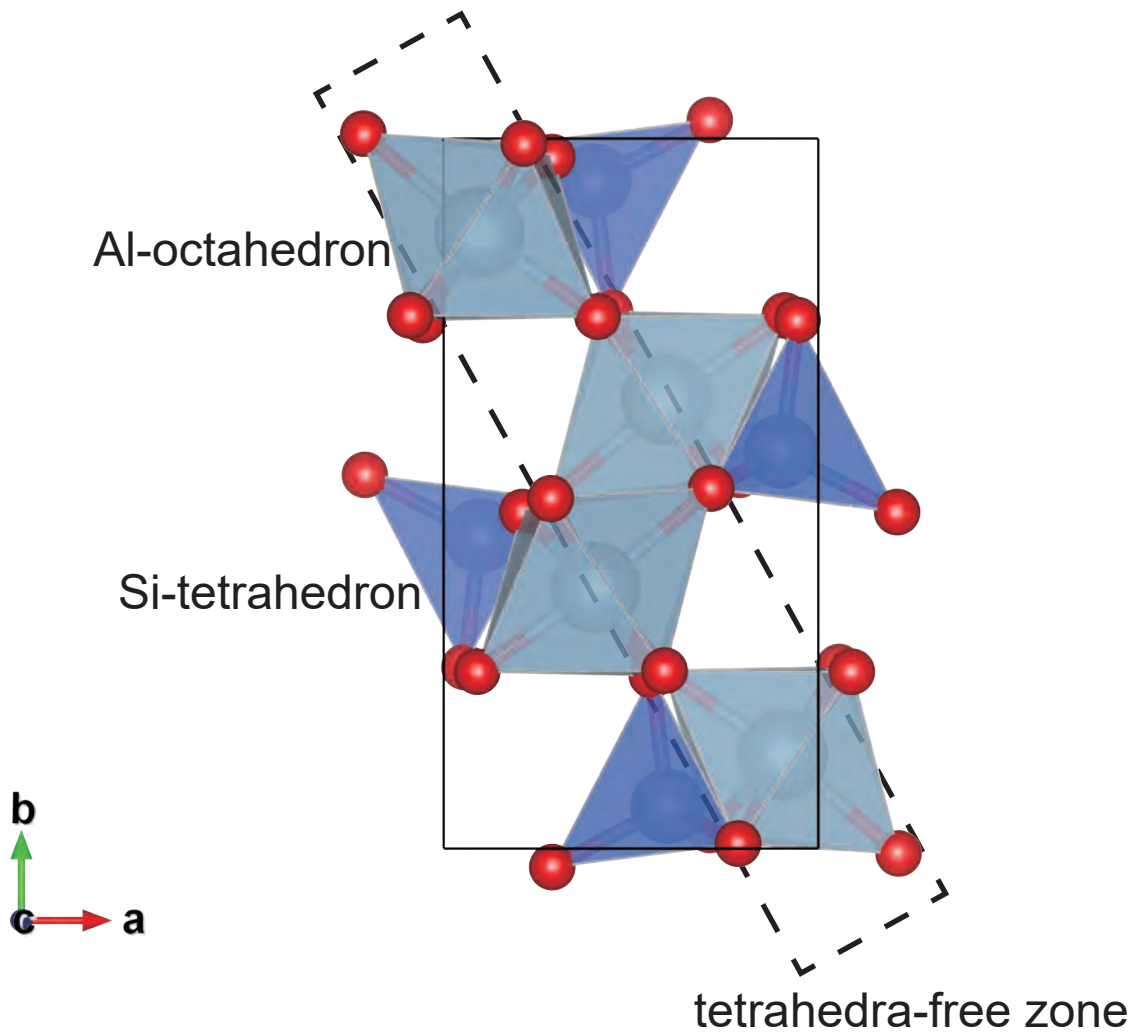


Figure 5



# Figure 6



# Figure 7

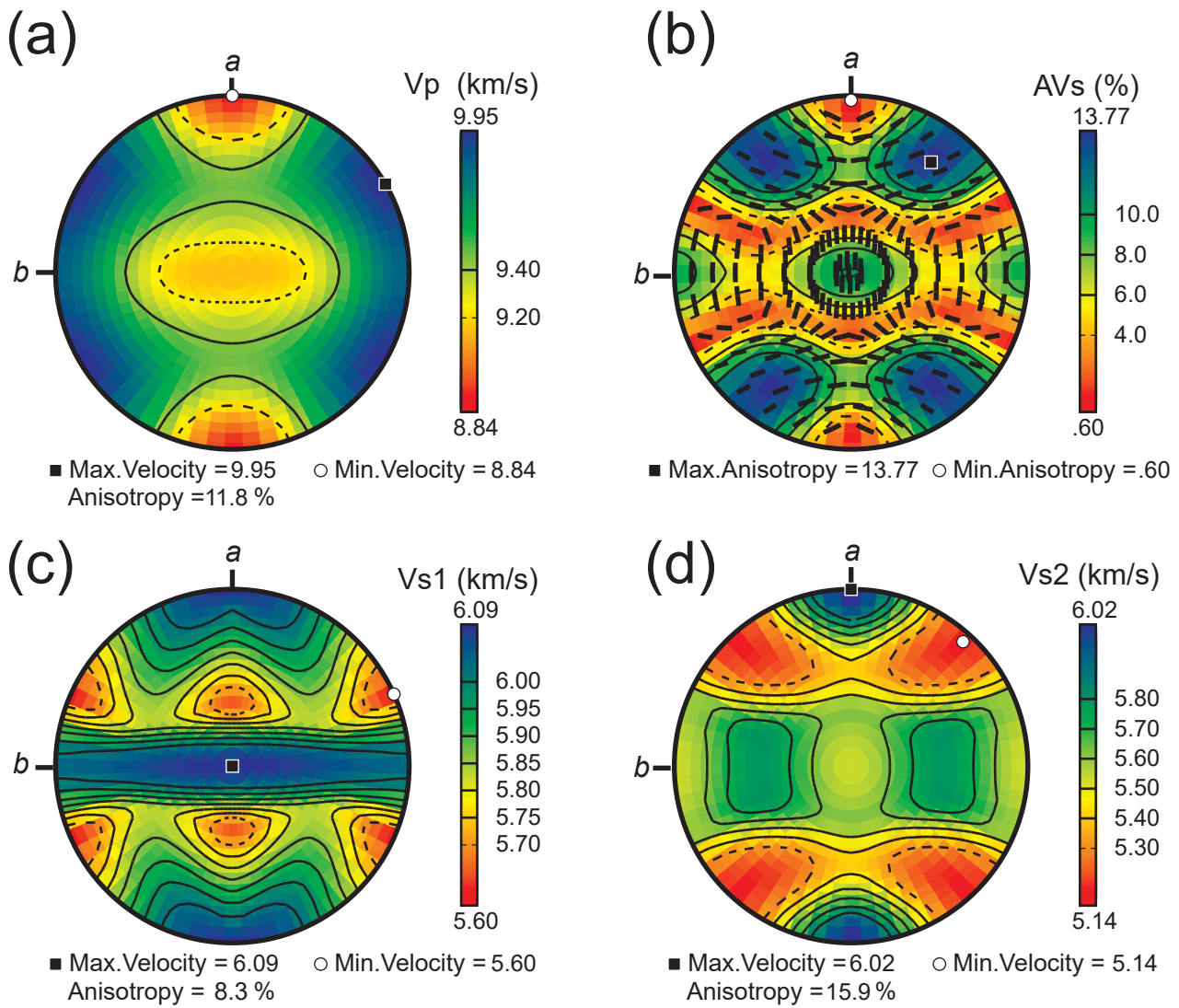




Figure 8

