1	Revision 1
2	Raman scattering and Cr ³⁺ luminescence study on the structural behavior of
3	δ-AlOOH at high pressures
4	Baoyun Wang ^{1,2} , Dayong Tan ^{3,4} , Wansheng Xiao ^{3,4} , Xing Ding ^{1,4} , Li Li ^{1,4} and
5	Maoshuang Song ^{1,4*}
6	¹ State Key Laboratory of Isotope Geochemistry, Guangzhou Institute of Geochemistry,
7	Chinese Academy of Sciences, Guangzhou 510640, China;
8	² College of Earth and Planetary Sciences, University of Chinese Academy of Sciences,
9	Beijing, 100049, China;
10	³ Key Laboratory of Mineralogy and Metallogeny, Guangzhou Institute of
11	Geochemistry, Chinese Academy of Sciences, Guangzhou 510640, China;
12	⁴ CAS Center for Excellence in Deep Earth Science, Guangzhou, 510640, China
13	*Corresponding author: M. Song (msong@gig.ac.cn)
14	Abstract
15	δ -AlOOH is regarded as a potential water carrier that is stable in the Earth's lower
16	mantle down to the core-mantle boundary along the cold slab geotherm; thus,
17	knowledge of its structural evolution under high pressure is very important for
18	understanding water circulation in the Earth's interior. In this work, we conducted
19	Raman scattering and luminescence spectroscopic experiments on δ -AlOOH at high

pressures up to 34.6 GPa and 22.1 GPa, respectively. From the collected Raman 20 spectra, significant changes in the pressure dependence of the frequencies of 21 Raman-active modes were observed at ~8 GPa with several modes displaying 22 softening behaviors. In particular, the soft A_1 mode, which corresponds to a lattice 23 vibration of the AlO₆ octahedron correlated to OH stretching vibration, decreases 24 rapidly with increasing pressure and shows a trend of approaching 0 cm⁻¹ at \sim 9 GPa 25 according to a quadratic polynomial extrapolation. These results provide clear 26 Raman-scattering spectroscopic evidence for the P2₁nm-to-Pnnm structural transition. 27 Similarly, the phase transition was also observed in the luminescence spectra of Cr^{3+} 28 in both powder and single-crystal δ -AlOOH samples, characterized by abrupt changes 29 in the pressure dependences of the wavelength of the R-lines and sidebands across the 30 $P2_1nm$ -to-Pnnm transition. The continuous decrease in R_2 - R_1 splitting with pressure 31 indicated that the distortion of the AlO₆ octahedron was suppressed under 32 compression. No abnormal features were clearly observed in our Raman or 33 34 luminescence spectra at ~ 18 GPa, where the ordered symmetrization or fully-centered state with hydrogen located at the midpoint of the hydrogen bond was observed by a 35 neutron diffraction study. However, some subtle changes in Raman spectra indicated 36 37 that the ordered symmetrization state might form at around 16 GPa.

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Keywords: δ-AlOOH, phase transition, Raman spectra, luminescence, high pressure

39 1. Introduction

40 Hydrous minerals in subducted slabs are widely accepted as the dominant water carriers into the Earth's interior (Ohtani 2020). Their dehydration can significantly 41 42 influence the evolution of the mantle by altering the physical and chemical properties of the surrounding materials (Hirschmann 2006; Mei and Kohlstedt 2000; Wang et al. 43 2006). To date, various hydrous minerals, such as antigorite, lawsonite and dense 44 hydrous magnesium silicate phases, have been suggested to transport water from the 45 Earth's surface to the deep mantle (Iwamori 2004; Ono 1998). Among those hydrous 46 minerals, δ-AlOOH stands out due to its wide P-T stability field. High-pressure and 47 high-temperature experiments have indicated that δ -AlOOH could survive P-T 48 conditions up to ~140 GPa and 2500 K; hence this mineral is regarded as a potential 49 water carrier to the Earth's core-mantle boundary (Duan et al. 2018; Ohtani et al. 2001; 50 Piet et al. 2020). Since Suzuki et al. (2000) first synthesized it by the dehydration 51 reaction of Al(OH)₃ at 21 GPa and 1000 °C, several routes have been proposed to 52 produce δ -AlOOH using various starting materials including diaspore, phase egg and 53 hydrous peridotite (Fukuyama et al. 2017; Ohtani et al. 2001; Sano et al. 2004; 54 Yoshino et al. 2019; Zhang et al. 2019). These results have further confirmed the 55 possible existence of δ -AlOOH in the deep interior of the Earth. 56

 δ -AlOOH belongs to orthorhombic system with a *P*2₁*nm* space group under ambient conditions (Komatsu et al. 2006). It consists of infinite edge-sharing AlO₆ octahedral chains along the *c*-axis that are cross-linked by hydrogen bonds in the *a-b* plane (Figure 1a). Due to its significance as an important water carrier at lower mantle

61 depths in subduction zones, the structural behavior of δ -AlOOH at high pressure has attracted much attention. X-ray diffraction studies observed a significant change in the 62 axial ratios of a/c and b/c at ~10 GPa, which is likely to correspond to the structural 63 transition from $P2_1nm$ to Pnnm (Kuribayashi et al. 2014; Sano-Furukawa et al. 2009; 64 Simonova et al. 2020). An analysis of the unit-cell volume versus pressure data 65 showed that the bulk modulus of the high-pressure *Pnnm* phase was ~52% higher than 66 that of the low pressure $P2_1nm$ phase, indicating a stiffened unit cell after the phase 67 transition (Simonova et al. 2020). Moreover, significant changes in the frequency 68 69 slope with pressure for the vibrational modes related to hydrogen bonds were also indicated at ~10 GPa by IR spectroscopic measurements (Kagi et al. 2010). A recent 70 neutron diffraction study further revealed that this structural transition is accompanied 71 72 with the order-to-disorder transformation of hydrogen along hydrogen bonds as a 73 result of the tunneling effect of hydrogen, and the high-pressure disordered phase further transitions into an ordered phase with symmetrized hydrogen bonds or 74 75 fully-centered phase at ~ 18 GPa without changing the *Pnnm* structure; in this ordered phase, hydrogen atoms are located at the midpoints of two neighboring 76 hydrogen-bonded oxygen atoms (Sano-Furukawa et al. 2018). However, the transition 77 to the ordered symmetrization phase was not observed to relate to any unusual feature 78 in previous X-ray diffraction or spectroscopic works (Kagi et al. 2010; Simonova et al. 79 2020). The structures of the disordered and ordered *Pnnm* phases are shown in Figure 80 81 1 (b) and Figure 1 (c), respectively. Additionally, first-principle calculations were used to understand these structural behaviors and their effect on the physical properties of 82

83	$\delta\text{-AlOOH}.$ These calculations have reproduced the primary experimental phenomena,
84	such as changes in the compressibility of the axial ratios and pressure dependences of
85	elastic moduli across the phase transition (Cedillo et al. 2016; Cortona 2017; Li et al.
86	2006; Pillai et al. 2018; Tsuchiya and Tsuchiya 2009). However, the critical pressure
87	of \sim 30 GPa for hydrogen-bond symmetrization predicted in most theoretical studies
88	was much higher than the experimental value of ~18 GPa (Sano-Furukawa et al. 2018;
89	Tsuchiya et al. 2008). Moreover, the calculations completely missed the disordered
90	phase in the structural transition path and hence ascribed the abnormal behaviors to
91	the symmetrization of hydrogen bonds, which was not consistent with the neutron
92	diffraction result of Sano-Furukawa et al. (2018). Although a comprehensive
93	theoretical model has not yet been well depicted, one recent theoretical work showed
94	that the disordered $Pnnm$ phase of δ -AlOOH can be successfully described when
95	thermal and nuclear quantum effects are considered (Bronstein et al. 2017).
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It is well known that Raman and luminescence spectroscopies are sensitive to 96 local bonding environments; hence it is possible to characterize the structural 97 evolution of δ -AlOOH from a spectroscopic perspective. Indeed, one previous study 98 by Mashino et al. (2016) already showed the disappearance and appearance of Raman 99 peaks at 6.5 GPa, which were taken as the signatures for the $P2_1nm$ -to-Pnnm phase 100 transition. The observed critical pressure of 6.5 GPa in their study was slightly lower 101 than that determined by X-ray and neutron diffraction studies; this discrepancy may 102 103 be due to the existence of deviatoric stress in the sample because no pressure-transmitting medium was used in the experiments. Additionally, the 104

first-principle calculations of Tsuchiya et al. (2008) also showed that some lattice 105 modes in the $P2_1nm$ phase exhibited softening behaviors with increasing pressure. 106 However, no detailed Raman scattering study at high pressure has yet to be reported 107 for δ-AlOOH. A previous Raman study recorded only a limited number of Raman 108 modes (Mashino et al. 2016), and no softening behavior was clearly documented. 109 Therefore, a further Raman study is still essential for better understanding the 110 structural behavior and vibrational dynamics of δ -AlOOH under high pressure. In 111 addition to Raman spectroscopy, luminescence spectroscopy of Cr³⁺ provides another 112 effective method to probe the local structural information in minerals under high 113 pressure. Ruby is a typical material of Cr^{3+} luminescence and is widely used as a 114 pressure calibration in diamond-anvil cells by utilizing the pressure-induced shift of 115 its R₁ emission line (Syassen 2008). Several recent studies have further certified that 116 the luminescence of Cr³⁺ can be used as an indicator for subtle structural modification 117 in minerals under high pressure (O'Bannon III and WIllIams 2019; O'Bannon and 118 Williams 2016a, 2016b). To the best of our knowledge, this method has not been 119 applied to investigate the compression behavior of δ -AlOOH thus far. 120

To further explore the structural evolution of δ -AlOOH upon compression by spectroscopic methods, we performed high pressure Raman scattering and luminescence spectroscopic measurements on single-crystal δ -AlOOH and Cr³⁺-bearing powder δ -AlOOH. We mainly focused on the proposed phase transitions in the process of hydrogen-bond symmetrization and tried to identify the unique signatures of these phase transitions in the spectroscopy results.

127 **2.** Experimental methods

Two samples, single-crystal δ -AlOOH and Cr³⁺-bearing powder δ -AlOOH, were 128 synthesized under high pressure and high-temperature conditions on the Sakura 129 2500-ton multi-anvil apparatus at the Guangzhou Institute of Geochemistry, Chinese 130 Academy of Sciences. The typical sample assembly for the synthesis of δ -AlOOH is 131 132 the same as that described by Liu et al. (2019). For single-crystal δ -AlOOH, reagent grade Al(OH)₃ of high purity was used as the starting material and was placed in a 133 welded gold capsule; the synthesis experiment was performed at 20 GPa and 1000 °C 134 with a duration of 22 hours (run number U795). The recovered sample was composed 135 of transparent crystals with maximum dimensions of approximately 300 µm. To 136 synthesize Cr^{3+} -bearing powder δ -AlOOH, we utilized ground natural diaspore 137 powder with strong Cr^{3+} luminescence as the starting material, and the P-T conditions 138 for the synthesis were ~19 GPa and ~1000 °C (run number U765; the temperature was 139 estimated from the heating power due to thermocouple cutoff during the experiment). 140 The recovered sample contained fine-grained aggregates of δ -AlOOH. Both samples 141 were confirmed to be a single phase of δ -AlOOH with a $P2_1nm$ structure by X-ray 142 diffraction measurements at the 13IDD beamline of GSECARS. The lattice 143 parameters were a = 4.7093(8) Å, b = 4.2271(1) Å, c = 2.8302(1) Å, and $V_0 = 56.34(5)$ 144 Å³ for the single-crystal sample and a = 4.7136(1) Å, b = 4.2255(4) Å, c = 2.8329(1)145 Å, and $V_0 = 56.42$ (4) Å³ for the powder sample, which are very close to the values 146 147 reported in a recent single-crystal X-ray diffraction study (Simonova et al. 2020). The chemical compositions determined by electron probe microanalysis (EPMA) indicate 148

that the single-crystal sample is of high purity, containing AlOOH with an ideal formula, while the powder sample contained small amounts of Fe_2O_3 (0.57 wt. %) and Cr_2O_3 (0.05 wt. %).

Raman scattering and luminescence spectroscopy measurements were performed 152 for both the single-crystal and powder δ -AlOOH samples at ambient conditions and 153 154 under high pressures. The high pressures were generated by a short symmetric diamond-anvil cell with a culet size of 400 μ m. A piece of double-polished δ -AlOOH 155 single-crystal or a flat powder δ -AlOOH pellet was loaded into the sample chamber 156 drilled in a pre-indented Re gasket. Argon was loaded to serve as a 157 pressure-transmitting medium by the cryogenic method. The pressure was determined 158 by the traditional ruby fluorescence method (Mao et al. 1986). The Raman scattering 159 and luminescence spectra were collected with a resolution of 1-2 cm⁻¹ using an InVia 160 Renishaw Raman spectrometer for powder δ -AlOOH and a WITec Raman 161 spectrometer for single-crystal δ-AlOOH, respectively. The wavelength of the laser 162 for both Raman systems was 532 nm. The obtained spectra were fitted by Horiba 163 Scientific LabSpec 5 software to extract the peak positions. 164

165 **3. Results and Discussions**

166 **3.1 Raman and luminescence spectra at ambient conditions**

167 The Raman spectra of the single-crystal and powder δ -AlOOH collected under 168 ambient conditions are shown in Figure 2. The spectral profiles of the two samples are 169 quite similar in the lattice vibration region ranging from 100-1400 cm⁻¹. A factor

group analysis predicted 21 Raman-active modes ($\Gamma = 7A_1 + 4A_2 + 3B_1 + 7B_2$) for 170 δ -AlOOH (Tsuchiya et al. 2008). However, we only observed 9 modes due to the 171 weak intensities of the other modes or due to vibrational peak overlapping. The peak 172 fitting results indicated that the observed modes for both samples were almost the 173 same in frequency within errors; therefore, we listed the frequency values in Figure 2 174 without distinguishing the samples. The symmetry of each mode was also labeled 175 according to the first-principle calculations by Tsuchiya et al. (2008). As it was 176 difficult to assign symmetry accurately based only on the calculated frequency when 177 178 several peaks were very close together in frequency, we carefully traced the evolution of the modes with pressure and compared them with the simulated results to avoid 179 potential mistakes. The B_2 mode at 1340 cm⁻¹ can be assigned as originating from the 180 bending vibration of the O-H···O hydrogen bonds. The modes between 276 and 620 181 cm^{-1} can be attributed to the vibrations of the AlO₆ octahedron (Xue et al. 2006). In 182 particular, the A₁ mode at 276 cm⁻¹ and the B₂ mode at 382 cm⁻¹ correspond to lattice 183 vibrations that are correlated with the O-H stretching vibrations of hydrogen bonds 184 (Tsuchiya et al. 2008). 185

In the O-H stretching vibration region, no peaks can be distinguished for the powder δ -AlOOH sample due to the strong luminescence of Cr³⁺ (Figure 2). However, broad multibands existed for the single-crystal δ -AlOOH sample. According to the correlation between the O-H stretching frequency and O-H…O bond distance proposed by Libowitzky (1999), we derived that the O-O distance ranges from 2.52 Å to 2.63 Å, which is close to the value of 2.55 Å determined by X-ray diffraction measurements (Kuribayashi et al. 2014). To date, the origin of these multibands is still
under debate, and has been either ascribed to either Fermi resonance by Xue et al.
(2006) or disorder superstructure by Tsuchiya et al. (2008). We found that the disorder
model of Tsuchiya et al. (2008) can reproduce the Raman spectra collected in our
present experiments well.

In the luminescence spectra (Figure 3), the most prominent feature of the powder 197 δ -AlOOH was two sharp and strong peaks at 684.2 nm and 686.5 nm, which can be 198 undoubtedly assigned to the R_1 and R_2 lines of Cr^{3+} luminescence. The R-lines 199 originate from the spin-forbidden ${}^{2}E{}^{-4}A_{2}$ transition of Cr^{3+} and imply a relatively 200 strong octahedral crystal field of CrO_6 in δ -AlOOH (Tanabe and Sugano 1954). Even 201 though no Cr^{3+} was detected by EPMA in the crystal δ -AlOOH, the sample displays a 202 luminescence spectrum very similar to that of the powder δ -AlOOH sample, implying 203 that a minor amount of Cr^{3+} was incorporated in the crystal δ -AlOOH sample. We 204 speculate that the minor Cr^{3+} in the crystal δ -AlOOH was likely introduced by the 205 LaCrO₃ heater during the high-pressure and high-temperature synthesis experiment 206 207 using the multi-anvil apparatus, which was also observed in the synthesis experiment of Si-rich Mg-sursassite (Bindi et al. 2020). Based on the EPMA data, a small amount 208 of Fe_2O_3 is present in the powder δ -AlOOH; however, it seems that no obvious 209 luminescence peaks result from this Fe^{3+} due to the similarity of the luminescence 210 spectra of the two samples. Additionally, several sidebands are observed on both sides 211 of the R-lines. The sidebands may be associated with Cr³⁺pairs or vibronic peaks 212 (O'Bannon and Williams 2016a, 2016b). All the luminescence peak positions are 213

given in Table 1.

215 **3.2 High pressure Raman spectra**

Raman spectra were collected at high pressures up to 34 GPa for the powder 216 δ -AlOOH and up to 34.6 GPa for the single-crystal δ -AlOOH. The stacked 217 representative Raman spectra are shown in Figure 4. Due to the strong first-order 218 Raman peak of diamond at ~1300 cm⁻¹, our Raman scattering measurements at high 219 pressures were restricted to a wavenumber region from $\sim 100 \text{ cm}^{-1}$ to 1200 cm⁻¹. 220 Although the OH stretching vibrational modes for single-crystal δ -AlOOH were 221 clearly recorded under ambient conditions, they were unable to be observed at high 222 pressures due to the interference of the second-order Raman peaks of diamond at 223 ~2600 cm⁻¹. Therefore, the Raman-active O-H…O bending and OH stretching 224 vibrational modes, which are in the wavenumber ranges of 1200–1400 cm⁻¹ and 225 2000–3000 cm⁻¹ respectively, were not investigated at high pressures in this study. 226 227 Fortunately, we were able to clearly observe at least six lattice vibrational modes in the collected high-pressure Raman spectra, allowing us to trace the compression 228 behavior of δ -AlOOH. 229

The powder and single-crystal δ -AlOOH samples exhibit similar spectral features and evolution with pressure. The lowest A₁ mode at 276 cm⁻¹ shifts quickly to a low wavenumber with the broadening peak width and gradually decreasing intensity as the pressure increases. The B₂ mode at 382 cm⁻¹ also shows a negative pressure dependence but disappears at 6.5 GPa. In contrast to the softening behavior of the 276 cm⁻¹ A₁ and 382 cm⁻¹ B₂ modes, the B₂ modes at 390 cm⁻¹ and 416 cm⁻¹ shift toward high wavenumbers with increasing pressure. The A₁ mode at ~515 cm⁻¹ is not sensitive to pressure and the B₂ mode at 626 cm⁻¹ shows a weak negative response to pressure. At pressures of 8.1 GPa and above, only two of six initial peaks can be observed, and several new weak peaks gradually appear. The detailed evolution trends of these modes are marked by red dashed lines in Figure 4.

The frequencies of the vibrational modes are plotted as a function of pressure in 241 242 Figure 5, together with the previous data of Mashino et al. (2016). Clearly, the B_2 modes at 390 cm⁻¹ and 416 cm⁻¹ show abrupt changes in their pressure dependences at 243 ~8 GPa, indicating the $P2_1nm$ -to-Pnnm phase transition observed in previous 244 diffraction studies (Sano-Furukawa et al. 2018; Sano-Furukawa et al. 2009; Simonova 245 et al. 2020). In the pressure range of 0-8 GPa, these two modes increase rapidly with 246 pressure and display nonlinear behavior. Above 8 GPa, the modes exhibit less 247 sensitivity to pressure with smaller linear slopes. In the case of the four softening 248 modes, the A_1 mode at 276 cm⁻¹ and B_2 mode at 382 cm⁻¹, which correspond to the 249 lattice vibrations of AlO₆ correlated with the OH stretching vibration, are found to be 250 highly sensitive to pressure with respect to the other two B₂ softening modes at 515 251 cm^{-1} and 626 cm^{-1} . The frequency of the B₂ mode decreases from 382 cm^{-1} under 252 ambient pressure to 346 cm⁻¹ at 5.5 GPa and the A_1 mode decreases from 276 cm⁻¹ at 253 ambient conditions to \sim 74 cm⁻¹ at 7.4 GPa. It is obvious that the negative slopes 254 255 become larger with increasing pressure. Given the many pressure points, the shift of the A_1 soft mode at 276 cm⁻¹ with pressure is fitted using the quadratic polynomial 256

function. The best fitting shows that the frequency of the A_1 mode reaches 0 cm⁻¹ at 257 ~ 9 GPa (see the inset of Figure 5), which is very close to the pressure of the 258 $P2_1nm$ -to-Pnnm phase transition. We noticed that one new peak at 701 cm⁻¹ appears at 259 9 GPa for powder δ -AlOOH; this peak may be related to the phase transition. 260 Additionally, another new weak peak at 616 cm⁻¹ was observed at 13.1 GPa for crystal 261 δ -AlOOH, and some new peaks are more clear at pressures of 16.5 GPa and above. A 262 recent neutron diffraction study indicated that the hydrogen-bond symmetrization of 263 δ -AlOOH took place at ~18 GPa. We speculate that the appearance of these news 264 265 peaks might correlate with the process of hydrogen-bond symmetrization from the 266 disordered state to the ordered state of hydrogen.

267 **3.3 High pressure luminescence spectra**

Luminescence spectroscopic measurements were performed at pressures up to 268 22.1 GPa for both the powder and single-crystal δ-AlOOH samples (Figure 6 and 269 Figure 7). Because the sidebands of the single-crystal δ -AlOOH were very weak due 270 to the low Cr^{3+} content, only the R-lines were clearly discerned for the spectra at high 271 pressures (Figure 7). In contrast, the sidebands and R-lines of the powder δ -AlOOH 272 could be traced up to the maximum pressure. The luminescence spectra of δ -AlOOH 273 at selected pressures are depicted in Figure 6. The relative intensity of the R-lines to 274 the sidebands gradually deceases with increasing pressure and the R₁ and R₂ lines 275 276 gradually converge into a single peak. At pressures of 13.4 GPa and above, it became difficult to deconvolve the R-lines with two peaks. The obtained peak positions for 277 the emission lines at different pressures for both samples are shown in Figure 8. One 278

key feature is the nonlinear behavior that shows kinks at ~ 8 GPa. For example, R₁ 279 remains at a nearly constant wavelength with increasing pressure below 8 GPa and 280 then increases rapidly at pressures above 8 GPa. The slopes or pressure dependences 281 of the sidebands and R-lines were calculated for the pressure regions below and above 282 8 GPa respectively, and are listed in Table 1. The anomalous variation in the slopes of 283 284 the sidebands and R-lines is not surprising since a phase transition has been suggested to take place at the same pressure (Kuribayashi et al. 2014; Sano-Furukawa et al. 285 2018). Therefore, our luminescence spectra provide additional clear evidence for the 286 287 structural transition from $P2_1nm$ to Pnnm in δ -AlOOH. Similar to our Raman experiments, no obvious change in the pressure dependence of the wavelengths of the 288 sidebands or R-lines was observed at ~18 GPa. 289

The R-lines of Cr^{3+} correspond to the environment of the Al site in the crystal 290 structure of δ -AlOOH, and R₂-R₁ splitting is regarded as an indicator of the distortion 291 of the AlO₆ octahedron (O'Bannon III and WIllIams 2019; O'Bannon and Williams 292 2016a, 2016b). Generally, a highly-distorted AlO₆ octahedron leads to a large R₂-R₁ 293 splitting. Here, we plotted the R_2 - R_1 splitting as a function of pressure in Figure 9 (a). 294 The R_2 - R_1 separation decreases continuously with pressure, implying that the AlO₆ 295 octahedron becomes less distorted under compression. From previous X-ray 296 diffraction studies on the ordered asymmetrical $P2_1nm$ phase under ambient 297 conditions, two different oxygen sites exist in a selected AlO₆ octahedron of 298 299 δ -AlOOH based on their relation to hydrogen. The O1 is linked to hydrogen by a weak H…O hydrogen bond, and the O2 forms covalent bonds with hydrogen 300

(Komatsu et al. 2006). As shown in Figure 9 (b), we label O1a and O2a for oxygen 301 atoms in the axial positions and O1e and O2e for oxygen atoms in the equatorial 302 positions. In the high-pressure phase with symmetrical hydrogen-bonds (the ordered 303 304 symmetrization phase), Al shifts to the center of an AlO₆ octahedron, and four Al-O bonds in the equatorial plane become equal in bond length (Figure 9 (c)); hence, the 305 distortion of the AlO₆ octahedron is expected to be suppressed. According to the 306 method proposed by Robinson et al. (1971), quadratic elongation (QE), the distortion 307 factor of the AlO₆ octahedron, was calculated using VESTA software with the 308 structural data reported by previous diffraction studies (Momma and Izumi 2008; 309 310 Sano-Furukawa et al. 2018; Simonova et al. 2020). As shown in the inset of Figure 9 (a), the QE value, which is 1.0071 at ambient conditions, drops to 1.0033 at \sim 8 GPa 311 after the phase transition from the $P2_1nm$ to the Pnnm structure, indicating that 312 pressure-induced suppression of distortion in the AlO₆ octahedron occurs during the 313 phase transition and further verifying further the conclusion inferred from R-lines 314 splitting. 315

316 3.4 Structural behavior and hydrogen-bond symmetrization in δ-AlOOH

317 δ -AlOOH possesses a distorted rutile-type structure (P2₁nm) with a strong and asymmetrical hydrogen bond under ambient conditions (Komatsu et al. 2006) and is 318 predicted to transition into the *Pnnm* structure with symmetrical hydrogen bonds 319 under compression (Tsuchiya et al. 2008). The abnormal change in the compressibility 320 321 of δ -AlOOH that has been observed at ~10 GPa by X-ray diffraction experiments in the past decade was correlated with hydrogen-bond symmetrization 322 bv 15

Sano-Furukawa et al. (2009). However, most theoretical studies have predicted a 323 much higher critical pressure of ~30 GPa for hydrogen-bond symmetrization in 324 δ-AlOOH (Bronstein et al. 2017; Cedillo et al. 2016; Cortona 2017; Pillai et al. 2018; 325 326 Tsuchiya and Tsuchiya 2009; Tsuchiya et al. 2008). Later X-ray diffraction studies indicated that the abnormal change in the compressibility of δ -AlOOH is related to the 327 P2₁nm-to-Pnnm phase transition in fact (Kuribayashi et al. 2014; Simonova et al. 328 2020). More recently, a neutron diffraction study revealed that δ -AlOOH undergoes 329 the $P2_1nm$ -to-Pnnm transition at ~9 GPa, that this transition is coupled with the 330 order-disorder transition of hydrogen along the hydrogen bonds due to tunneling 331 332 effect, and that the transition to the ordered symmetrization phase occurs at ~18 GPa (Sano-Furukawa et al. 2018). The difference in pressure of the hydrogen bond 333 symmetrization between the neutron diffraction study (Sano-Furusawa et al. 2018) 334 and the theoretical study by Tsuchiya et al. (2008) can be explained well as a result of 335 the temperature effect (Kang et al. 2017). 336

The Raman and luminescence spectral changes of δ -AlOOH at ~8 GPa observed 337 in this study agree well with the $P2_1nm$ -to-Pnnm structure transition reported by 338 previous X-ray and neutron diffraction studies (Kuribayashi et al. 2014; 339 Sano-Furukawa et al. 2018; Simonova et al. 2020). In the low-pressure $P2_1nm$ phase 340 with ordered asymmetrical hydrogen bonds, the softening behavior of the four Raman 341 lattice modes reflects the pressure-induced evolution of the hydrogen-bond geometry 342 in the process of hydrogen-bond symmetrization, i.e., the covalent O-H bondlength 343 344 slightly increases while the O-H···O distance becomes shortened under compression.

In contrast to the softening behavior, the B_2 modes at 390 cm⁻¹ and 416 cm⁻¹ show 345 rapid increases with pressure that are consistent with the high compressibility or low 346 bulk modulus of the P21nm phase. For the Pnnm phase, all the observed Raman 347 modes exhibit less sensitivity in frequency to pressure with small linear slopes (Figure 348 5), which is consistent with the low compressibility or high bulk modulus of this 349 phase (Sano-Furukawa et al. 2009; Simonova et al. 2020). It is interesting that the A₁ 350 soft mode at 276 cm⁻¹ was extrapolated to approach 0 cm⁻¹ at ~9 GPa. Tsuchiya et al. 351 (2008) indeed predicted by first-principle calculations that this mode will become 0 352 cm⁻¹ when hydrogen-bond symmetrization takes place at 30 GPa. It is also noteworthy 353 that the A₁ softening mode at 276 cm⁻¹ vanishes before reaching 0 cm⁻¹ at ~9 GPa, but 354 the $P2_1nm$ -to-Pnnm transition takes place at ~8 GPa. We speculate that the vanishing 355 of the A_1 soft mode at 276 cm⁻¹ is due to the much longer timespan of the Raman 356 measurements than the quantum tunneling effect of hydrogen (proton) along the 357 hydrogen bonds in δ -AlOOH; hence, the P2₁nm-to-Pnnm structure transition is 358 359 predicted to be coupled with an order-disorder transition of hydrogen due to tunneling effect (Benoit et al. 1996; Sano-Furukawa et al. 2018). 360

Additionally, we did not notice significant changes in the pressure dependence (slope) of frequency for the Raman-active lattice modes or the emission R-lines or sidebands of δ -AlOOH at ~18 GPa, at which the hydrogen bonds change from a disordered state to an ordered symmetrization state with a single-well proton (hydrogen) potential, as observed by a recent neutron diffraction study (Sano-Furukawa et al. 2018). Similarly, previous IR spectroscopy and single-crystal

diffraction studies also did not show any unusual behavior at approximately 18 GPa 367 (Kagi et al. 2010; Simonova et al. 2020). However, our Raman spectroscopic results 368 do show some subtle changes that may be related to the transition from the disordered 369 to ordered symmetrization state of hydrogen. In comparison with the 10.5 GPa 370 spectrum of single-crystal δ -AlOOH, which can be assigned to be in the 371 372 hydrogen-disordered state, several new but very weak peaks appear in the 16.5 GPa spectrum, and these new peaks become more clear at pressures of 24.4 GPa and above 373 (Figure 4b). However, in comparison with the $P2_1nm$ -to-Pnnm transition, we noticed 374 375 that the disordered-ordered symmetrization transition shows no obvious changes in the Raman scattering spectra. This might be determined by the nature of this phase 376 transition (second-order), which is only related to the motion of hydrogen (proton) 377 378 along hydrogen bonds from the disordered state to the fully-centered state with hydrogen located at the midpoint of a hydrogen bond. Both the neutron diffraction 379 and Brillouin scattering evidence support the existence of a disordered-to-ordered 380 symmetrization transition at pressures of approximately 15-20 GPa (Mashino et al. 381 2016; Sano-Furukawa et al. 2018; Su et al. 2021b). Theoretical studies predicted that 382 thermal and quantum effects may play important roles in the process of 383 384 hydrogen-bond symmetrization (Bronstein et al. 2017; Kang et al. 2017). The ab initio molecular dynamics calculations with finite-temperature statistics yielded that the 385 transition pressure to the ordered hydrogen-bond symmetrization state was 17.0 GPa 386 at 300 K, being in good agreement with the neutron diffraction study and our present 387 study (Kang et al. 2017; Sano-Furukawa et al. 2018). 388

389 4. Implications

390 The Hydrogen-bond geometry and its evolution with pressure have profound effect on the structural behaviour and physical property of δ -AlOOH and other 391 hydrous minerals, especially elasticity (Mashino et al. 2016; Tsuchiya and Tsuchiya 392 2009). First-principles calculations by Tsuchiya and Tsuchiya (2009) predicted that 393 the elastic constants of δ -AlOOH with asymmetrical hydrogen bonds (the ordered 394 asymmetrical $P2_1nm$ phase) are significantly smaller than those of δ -AlOOH with 395 symmetrical hydrogen bonds. In the process of hydrogen-bond symmetrization of 396 δ -AlOOH, the diagonal elastic constants (especially C₁₁ and C₂₂), the bulk and shear 397 moduli, and the acoustic velocities (V_P, V_S) show an anomalous increase or harden 398 rapidly with pressure, and then become stiffened after achieving the ordered 399 symmetrization state. These theoretical predictions have been confirmed by 400 high-pressure Brillouin scattering experiments on δ -AlOOH polycrystalline aggregate 401 (Mashino et al. 2016; Su et al. 2021b). Besides, according to the phase diagram of 402 AlOOH (Yoshino et al. 2019), the hydrogen-bond symmetrization is inferred to 403 promote the stability field (or dehydration temperature) of δ -AlOOH. 404

Phase relation and stability experiments revealed that δ-AlOOH is an important
hydrous mineral as the decomposition product of phase egg in the sedimentary layer
of subducted slabs (Ono et al. 1998; Sano et al. 2004; Fukuyama et al. 2017), and may
even exist in the basaltic and ultramafic layers (Suzuki et al. 2000; Ohtani et al. 2001;
Yuan et al. 2019). It is stable under the P-T conditions of the lower part of the mantle
transition zone and the lower mantle (Sano et al. 2004; Fukuyama et al. 2017; Duan et

al. 2018; Yuan et al. 2019). According to our Raman and luminescence results, 411 δ-AlOOH under compression undergoes hydrogen-bond symmetrization and achieves 412 the ordered symmetrization state at ~ 16 GPa. Because quantum effects of hydrogen is 413 predicted to dominate the process of hydrogen-bond symmetrization (Bronstein et al. 414 2017), a weak temperature dependence of the transition pressure to the ordered 415 symmetrization phase is expected. Therefore, it can be estimated that δ -AlOOH is 416 generally in the ordered symmetrization state in the Earth's deep interior. The 417 compressional-wave (V_P) and shear-wave (V_S) velocities of δ -AlOOH in the ordered 418 symmetrization phase are significantly greater than most major minerals in the mantle 419 transition zone and low mantle bridgemanite and hence may contribute to the 420 high-velocity seismic anomalies at depths of the mantle transition zone and upmost 421 422 low mantle (Su et al. 2021b).

In contrast to δ -AlOOH, other subduction-related hydrous minerals, such as 423 ε-FeOOH, phase D and ice VII, transit to the disordered phase and the ordered 424 symmetrization phase at pressures higher than their lower pressure limits of stability 425 fields (Guthrie et al. 2019; Thompson et al. 2020; Tsuchiya et al. 2005; Yoshino et al. 426 2019). These hydrous minerals may vary from ordered asymmetrical, disordered and 427 ordered symmetrization states of hydrogen bond with increasing depth in the Earth's 428 deep interior. To constrain well geophysical observations so as to understand the water 429 storage and circulation in subduction zones, it is essential to investigate systematically 430 431 the effect of hydrogen-bond symmetrization on the physical property of subduction-related hydrous minerals by both high-pressure experiments and 432

433 theoretical calculations.

Under nature circumstances, δ -AlOOH in the basaltic and ultramafic layers of 434 subducting slabs may contain some amount of iron (Yuan et al. 2019). The 435 incorporation of iron can significantly increase the P2₁nm-Pnnm transition and 436 hydrogen-bond symmetrization pressures of δ -AlOOH, decrease its acoustic velocities 437 and affect other physical properties such as electrical conductivity and thermal 438 conductivity (Su et al. 2021a, 2021b). Especially, the iron in δ -AlOOH can induce the 439 high spin to low spin transition at high pressure. Therefore, it is also necessary to 440 figure out in details the effect of iron on the physical property of δ -AlOOH in future. 441

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Table 1. The Cr^{3+} luminescence peak positions of δ -AlOOH under ambient conditions

614 and their slopes under compression.

No.	Ambient	$P2_1nm$	Pnnm
	λ (nm)	(0-8 GPa)	(8-22
			GPa)
p1	661.3	0.17	0.12
p2	666.2	0.29	0.18
р3	669.1	0.12	-
p4	672.4	0.09	0.14
p5	673.4	0.35	0.09
p6	675.4	-	-
R_2	684.2	0.17	0.23

R_1	686.5	0.00	0.22
p7	698.4	0.08	0.35
p8	702.0	-	-
р9	705.0	0.10	0.33
p10	709.0	-	-
p11	711.6	0.21	0.49
p12	714.3	-	-
p13	717.4	0.45	0.49
p14	720.4	-	-

616

617 Figure captions

Figure 1. Crystal structures of δ -AlOOH. (a) The low pressure ordered $P2_1nm$ phase with asymmetrical hydrogen bonds from ambient pressure to 8 GPa,, (b) high-pressure disordered *Pnnm* phase with symmetrical hydrogen bonds from 8 to 18 GPa, and (c) high-pressure ordered *Pnnm* phase with symmetrical hydrogen bonds above 18 GPa. The large silver, medium-red and small white spheres represent Al, O and H atoms, respectively. The structures were drawn using VESTA software (Momma and Izumi 2008). Structural data are from Sano-Furukawa et al. (2018).

625

Figure 2. Raman spectra of powder and crystal δ -AlOOH under ambient condition.

627

Figure 3. Luminescence spectra of powder and crystal δ-AlOOH under ambient

	DOI: https://doi.org/10.2138/am-2022-8005. http://www.minsocam.org/
629	conditions. The inset shows the R_1 and R_2 values of Cr^{3+} luminescence in δ -AlOOH.
630	
631	Figure 4. Representative high-pressure Raman spectra of δ -AlOOH in the (a) powder
632	sample and (b) crystal sample. The red dashed lines trace the evolution of Raman
633	peaks with increasing pressure.
634	
635	Figure 5. The Raman mode frequencies of δ -AlOOH as a function of pressure. The
636	black vertical dashed line indicates the phase transition at ~8 GPa from $P2_1nm$ to
637	Pnnm. The black dashed lines below 8 GPa show the nonlinear evolution of the mode
638	with pressure. The black solid lines above 8 GPa show the linear fitting results of the
639	compressional data and the slopes are shown near the lines. The inset shows the
640	quadratic polynomial extrapolation of the A_1 mode at 276 cm ⁻¹ .
641	
642	Figure 6. Stacked high-pressure Cr^{3+} luminescence spectra of powder δ -AlOOH. The
643	# denotes the R-line for ruby.
644	
645	Figure 7. Stacked high-pressure Cr^{3+} luminescence spectra of crystal δ -AlOOH.
646	
647	Figure 8. The Cr^{3+} luminescence peak positions of δ -AlOOH as a function of pressure.
648	The black vertical dashed line indicates the phase transition from $P2_1nm$ to $Pnnm$. The
649	blue solid lines are the linear fitting results of the compressional data.
650	

- **Figure 9.** (a) Cr^{3+} R-line splitting (R₁-R₂) of δ-AlOOH as a function of pressure. The
- inset shows the quadratic elongation (QE) values at high pressure. (b) The AlO_6
- 653 octahedron in $P2_1nm$ -phase δ -AlOOH. (c) The AlO₆ octahedron in *Pnnm*-phase
- δ -AlOOH. The large silver, medium-red and small white spheres represent Al, O and
- 655 H atoms, respectively.



Figure 1 Always consult and cite the final, published document. See http://www.minsocam.org or GeoscienceWorld



Figure2

Always consult and cite the final, published document. See http://www.minsocam.org or GeoscienceWorld





Figure4





Figure6

Wavenumber (cm



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Figure8



Figure9