

Revision 1

Elasticity and high pressure behavior of Mg₂Cr₂O₅ and CaTi₂O₄-type phases of magnesiochromite and chromite

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Abstract

In situ high-pressure and high-temperature X-ray diffraction studies on magnesiochromite, MgCr₂O₄ and a natural chromite, (Mg,Fe)(Al,Cr)₂O₄ using a laser-heated diamond anvil cell technique were performed at pressures to about 45 GPa. Our results on MgCr₂O₄ at ~15 GPa showed temperature-induced dissociation of MgCr₂O₄ to Cr₂O₃ + MgO below ~1500 K and formation of modified Ludwigite (mLd)-type Mg₂Cr₂O₅ + Cr₂O₃ above ~1500 K. Above 20 GPa, only a single phase with CaTi₂O₄-type structure of MgCr₂O₄ was observed at 1400-2000 K. A second-order Birch-Murnaghan fit to the pressure-volume data for the CaTi₂O₄-type phase of MgCr₂O₄ yields zero-pressure volume (V_0) = 264.4(8) Å³ and bulk modulus (K_0) = 185.4(4) GPa, and for the CaTi₂O₄-type structure of natural (Mg,Fe)(Al,Cr)₂O₄ yields V_0 = 261(1) Å³ and K_0 = 175.4(2) GPa. A second-order Birch-Murnaghan fit to the pressure-volume data for mLd-type Mg₂Cr₂O₅ yields V_0 = 338.9(8) Å³ and K_0 = 186.5(6) GPa. Our high-pressure phase study of the

chromite spinels can be used as an indicator for shock pressure in impact rocks and meteorites. The bulk modulus of the high pressure phases of MgCr_2O_4 and FeCr_2O_4 can help develop a thermodynamic model for Mg and Fe endmember spinels in the upper mantle and transition zone.

Keywords

Magnesiochromite, chromite, CaTi_2O_4 phase, modified Ludwigite phase, equation of state.

Introduction

Chromite spinels $(\text{Mg,Fe})\text{Cr}_2\text{O}_4$ are found in peridotite from the Earth's mantle. They also occur in layered ultramafic intrusive rocks (Gu and Wills 1988) and in metamorphic rocks such as serpentinites (Pearre and Heyl 1960). Chromite is commonly associated with olivine, magnetite, and corundum (Guilbert et al. 1986). Although typically present as accessory phases, chromite spinels are widely considered to be important petrogenetic indicators (Irvine 1965, 1967; Evans and Frost 1975; Sack and Ghiorso 1991; Bosi et al. 2008), and can point to certain regions in garnet lherzolite as targets for diamond exploration (Griffin and Ryan 1995, Stachel and Harris 2008). Yamamoto et al. (2008) found numerous exsolution lamellae of diopsidic clinopyroxene and coesite in podiform chromitites from the Luobusa ophiolite. To produce such exsolution features, it is necessary to incorporate SiO_2 and CaO components in the host chromite or its precursor. In fact, natural chromites contain only 0.6 wt.% SiO_2 and the CaO content is less than several tens of ppm (Arai and Yurimoto 1994). However, MgCr_2O_4 -rich CaTi_2O_4 -type (CT)/ CaFe_2O_4 -type (CF) phase can contain Ca, Si, Ti, and Fe as CaFe_2O_4 and CaTi_2O_4 solid solutions (Chen et al. 2003a, 2003b). Those exsolution lamellae were explained by the association of high pressure minerals (diamond, clinopyroxene and coesite) with chromites in the podiform chromitites of the Luobusa

ophiolite by the inverse transformation from CT/CF phase to chromite by a mantle upwelling process (Yang et al. 2007; Yamamoto et al. 2009; Arai, 2010, 2013; Satukawa et al. 2015; Zhang et al. 2017). It has been suggested that high pressure and high temperature phases of chromite spinel are likely present in the deep mantle (Yamamoto et al. 2008; Satukawa et al. 2015; Zhang et al. 2017). This makes it geologically important to determine the equation of state (EOS) of these high pressure and high temperature phases of chromite spinel. Furthermore, studying the equations of state of high pressure phases of chromite is essential for understanding the bulk modulus and potentially for thermodynamic models of deep Earth and planetary materials. It is known that natural chromites exhibit solid solution in the FeCr_2O_4 - MgCr_2O_4 system. Synthetic chromites, on the other hand, can readily be prepared as pure end-member phases such as MgCr_2O_4 , eliminating the compositional variation in the final products.

Chromite spinel with the general formula ACr_2O_4 (where A is divalent metal) has a normal spinel structure with a tetrahedral coordination of the divalent metal ions and an octahedral coordination of chromium ions (Romeijn 1953; Miyahara and Ohnishi 1956; Sawaoka et al. 1971). The divalent metal ions can be Fe, Mg, Zn or Mn. Natural chromite is also a common mineral in many meteorites (Rubin 1997). More importantly, it has high degree of resistance against weathering and diagenesis, and is the only mineral surviving in a meteorite embedded within Middle Ordovician limestones (Thorslund et al. 1984). In a recent study of the shock veins of the Suizhou meteorite, Chen et al. (2003a, 2003b) reported that chromite spinel transforms to the CF-type structure at 12.5 GPa and to the CT-type structure above 20 GPa.

Due to their importance in the geosciences, physics, materials science, and crystallography, properties of chromite spinels have been studied extensively both theoretically and experimentally.

Ab initio calculation has been performed to examine the phase transformation of chromite (Catti et al. 1999). The simulation showed that chromite can dissociate to constituent oxides at high pressures and room temperature. Wang et al. (2002) first studied the magnesiochromite (MgCr_2O_4) spinel at high pressure using Raman and reported that this spinel underwent a phase transformation to CF or CT structure at 14.2 GPa. The complete transition took place at 30.1 GPa. However, the transformation of MgCr_2O_4 to CF or CT phase at room temperature was not supported by a recent in situ experimental study (Yong et al. 2012). Instead, a polymorphic phase transition from cubic to tetragonal structure was reported for MgCr_2O_4 at 23.5 GPa and room temperature (Yong et al. 2012) and at a much lower pressure (12 GPa) for chromite (FeCr_2O_4) (Kyono et al. 2012). A recent study on chromite (FeCr_2O_4) showed that chromite dissociates to modified Ludwigite-type $\text{Fe}_2\text{Cr}_2\text{O}_5$ phase + corundum-type Cr_2O_3 at 12-16 GPa. The assemblage is then transformed at 17-18 GPa to CF (space group *Pnma*)-type structure below 1573 K, or to CT (space group *Cmcm*)-type structure above 1573 K (Ishii et al. 2014). Another recent study on magnesiochromite (MgCr_2O_4) showed different phase behaviors (Ishii et al. 2015). MgCr_2O_4 dissociates into a mixture of new modified Ludwigite (mLd)-type $\text{Mg}_2\text{Cr}_2\text{O}_5$ phase (space group *Pbam*) + corundum-type Cr_2O_3 at 1273–1873 K, but, at temperature < 1273 K, it decomposes into MgO periclase + corundum-type Cr_2O_3 at 12-15 GPa. At about 17–19 GPa, the mixture of mLd-type $\text{Mg}_2\text{Cr}_2\text{O}_5$ phase + corundum-type Cr_2O_3 transforms to a CT-type single phase. Note that the CF-type structure of MgCr_2O_4 was not observed in that study (Ishii et al. 2015).

Therefore, in this study we report and compare phase diagram of MgCr_2O_4 and the equations of state of CaTi_2O_4 -type structure of MgCr_2O_4 and of natural $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$, and mLd-type $\text{Mg}_2\text{Cr}_2\text{O}_5$ phase at mantle pressures.

Experiments

Synthetic magnesiochromite (MgCr_2O_4) and natural chromite $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$ were used as starting materials. The MgCr_2O_4 was obtained from the same batch used in our previous study (Yong et al. 2012). Natural chromite $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$ (Sudbury, Ontario) with a chemical formula of $\text{Mg}_{0.52(2)}\text{Fe}_{0.61(2)}\text{Mn}_{0.01(0)}\text{Ti}_{0.01(0)}\text{Al}_{0.58(1)}\text{Cr}_{1.30(1)}\text{O}_4$ was determined by electron microprobe at University of Western Ontario. The chromite sample was first ground to mm-sized grains and then dried in an oven at 473 K for at least 12 h. In order to separate the chromite from other accessory minerals, pure chromite grains were hand-picked with a needle under a microscope.

The mm-sized samples were ground to 1-3 μm fine powder and mixed with ~10 wt% gold for use as a pressure calibrant. The pressure was determined by the equation of state of gold (Fei et al. 2007). Mixtures of the samples and gold were then loaded into a ~120 μm hole drilled in a rhenium gasket. Neon was also loaded as a pressure medium in the diamond-anvil cell (DAC) using the GSECARS high-pressure gas loading system (Rivers et al. 2008). In situ high-pressure and high-temperature X-ray diffraction (XRD) experiments were performed at 13-ID-D, GSECARS, Advanced Photon Source (APS) of Argonne National Laboratory (Prakapenka et al. 2008). A double-sided laser heating method was employed with synchrotron XRD. The temperature of the sample heated with focused spot in a diameter of about 20 μm was measured from both sides by spectroradiometry. The samples were probed with a monochromatic X-ray beam with a wavelength of 0.3344 Å and a 3×2 μm cross-section. The X-ray beam was ensured to be in the center of the heating spot by carefully aligning optical paths using X-ray induced fluorescence on the sample. A Mar-165 CCD detector was used to collect the XRD images. The sample-to-detector distance and orientation of the detector were calibrated using CeO_2 and LaB_6 powder. The

DIOPTAS software was used for calibration and integration of the diffraction images (Prescher and Prakapenka 2015).

Results and Discussion

In the high pressure and high temperature experiments conducted on synthetic MgCr_2O_4 , we observed dissociation of MgCr_2O_4 to $\text{Cr}_2\text{O}_3 + \text{MgO}$ at ~ 15 GPa and temperature below 1440 K, whereas mLd-type $\text{Mg}_2\text{Cr}_2\text{O}_5 + \text{Cr}_2\text{O}_3$ was observed when the sample was heated above 1600 K. At pressure above 20 GPa, only a single phase, CaTi_2O_4 -type (CT) structure was observed at the temperature of 1400-2000 K (Fig. 1). The tetragonal phase of MgCr_2O_4 was also observed above 23.5 GPa and ambient temperature (Fig. 1). Our synthesis result is in broad agreement with previously reported high pressure and high temperature experiments and the obtained phase diagram is shown in Fig. 2 (Kyono et al. 2012, Yong et al. 2012, Ishii et al. 2015). The discrepancy of phase boundary between $\text{MgO} + \text{Cr}_2\text{O}_3$ and mLd-type $\text{Mg}_2\text{Cr}_2\text{O}_5 + \text{Cr}_2\text{O}_3$ could be resulted from different temperature determination method. In order to determine the EOS of the polymorphs of MgCr_2O_4 , the syntheses were performed within the stability field of the desired polymorph. The CT phase of MgCr_2O_4 was synthesized at ~ 45 GPa and ~ 1800 K, natural $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$ was compressed to ~ 40 GPa and heated at ~ 1800 K, and mLd-type $\text{Mg}_2\text{Cr}_2\text{O}_5$ structure was produced at ~ 16 GPa and ~ 1500 K, respectively. The phases synthesized at high pressure and high temperature were then decompressed to ambient conditions over a period of ~ 6 -8 hours and a dense set of XRD patterns was collected during decompression.

The cubic spinel phase of MgCr_2O_4 transformed completely to an orthorhombic CT structure at ~ 45 GPa and ~ 1800 K under hydrostatic conditions. The observed and calculated peak positions of CT structure of MgCr_2O_4 retained at ambient conditions are shown in Table 1. The unit-cell

lattice parameters of the CT structure of MgCr_2O_4 at ambient conditions are $a_0 = 2.86464(8) \text{ \AA}$, $b_0 = 9.51504(5) \text{ \AA}$, and $c_0 = 9.6981(6) \text{ \AA}$. The obtained unit-cell lattice parameters at ambient conditions are close to those determined by Ishii et al. (2015) ($a_0 = 2.85107(2) \text{ \AA}$, $b_0 = 9.48930(8) \text{ \AA}$ and $c_0 = 9.67853(8) \text{ \AA}$), which confirms the synthesis of CT phase. The unit-cell lattice parameters and the volumes of synthesized CT phase of MgCr_2O_4 at different pressures are tabulated in Table S1.

Similarly, natural $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$ transformed completely to CT-type structure (*Cmcm*) at ~ 40 GPa and ~ 1800 K. The unit-cell lattice parameters for the CT phase of natural $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$ at ambient conditions are $a_0 = 2.8575(6) \text{ \AA}$, $b_0 = 9.4506(6) \text{ \AA}$, and $c_0 = 9.6641(7) \text{ \AA}$, which are slightly lower than those of the CT phase of MgCr_2O_4 (see above) and CT phase of FeCr_2O_4 ($a_0 = 2.8845(1) \text{ \AA}$, $b_0 = 9.5207(2) \text{ \AA}$, and $c_0 = 9.7532(2) \text{ \AA}$) (Ishii et al. 2014). The unit-cell lattice parameters and the volumes of the CT phase of natural $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$ at different pressures are listed in Table S2.

To evaluate the pressure derivative of the bulk modulus (K_0') of CT-type phases of both the compounds we plotted the normalized pressure versus Eulerian strain. Our results show a horizontal relationship for both compounds, suggesting a value of 4 for the pressure derivative of the isothermal bulk modulus (Fig. 3). This allows us to fit the pressure-volume data with a second-order Birch-Murnaghan EOS and the results are shown in Fig. 3 and Table 2. Comparing the EOS fitting curves of the CT phase of MgCr_2O_4 , natural chromite $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$ and FeCr_2O_4 (Ishii et al. 2014) indicate that natural chromite $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$ has the lowest volume whereas FeCr_2O_4 has the highest volume (Fig. 3). In the CT phase the coordination number of Cr is suggested to be 6 (i.e. it occupies octahedral sites) whereas the coordination number of Mg and Fe

is 6+2 (they occupy bicapped trigonal prism sites) based on the results of Ishii et al. (2014). Furthermore, our high-temperature data of CT phase of Mg_2CrO_4 fitted to the second-order EOS yield $V_0 = 266.5(2) \text{ \AA}^3$ at 1200 K, $V_0 = 268.4(2) \text{ \AA}^3$ at 1400 K, and $V_0 = 270.2(2) \text{ \AA}^3$ at 1550 K (Fig. 4). High-temperature data of CT phase of natural $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$ fitted to the second-order EOS yield $V_0 = 262.8(1) \text{ \AA}^3$ at 1600 K (Fig. 4).

From our experiments, the mLd-type $\text{Mg}_2\text{Cr}_2\text{O}_5 + \text{Cr}_2\text{O}_3$ phase that we synthesized at high pressure and high temperature conditions is mixed with original chromite spinel phase and sometimes with CaTi_2O_4 -type phase. The persistence of original spinel structure was attributed to slow kinetics and short heating duration (the duration of each heating cycle was around 1 hour). On the other hand, growth of minor amounts of CaTi_2O_4 -type structure may be attributed to brief temperature fluctuations during the synthesis process. However, the limited amounts of CT phase at this pressure and temperature range did not affect the data processing. The mLd-type $\text{Mg}_2\text{Cr}_2\text{O}_5$ is stable in a narrow P-T range (Fig. 2) and, therefore, it is difficult to obtain a pure phase of mLd-type $\text{Mg}_2\text{Cr}_2\text{O}_5$. For our purposes, the data quality of the mixed phase is sufficient to determine the EOS. Observed and calculated peak positions of mLd-type $\text{Mg}_2\text{Cr}_2\text{O}_5$, Cr_2O_3 and CT-type structure of MgCr_2O_4 at ambient conditions is shown in Table S3.

In addition, MgCr_2O_4 dissociates into $\text{MgO} + \text{Cr}_2\text{O}_3$ at similar pressures but lower temperatures than those at which mLd-type $\text{Mg}_2\text{Cr}_2\text{O}_5 + \text{Cr}_2\text{O}_3$ form (Fig. 2). With the large number of pressure-volume data collected for Cr_2O_3 , we were also able to determine the EOS of Cr_2O_3 phase and compare with the EOS determined by Dera et al. (2011). Fig. 5 shows the isothermal compression curve of Cr_2O_3 phase from the same XRD patterns that are used to determine the EOS of mLd-type $\text{Mg}_2\text{Cr}_2\text{O}_5$ phase. The Birch-Murnaghan EOS fitting of our Cr_2O_3 phase is in very good

agreement with previously reported EOS of Cr₂O₃ (Dera et al. 2011). This confirms that our data collected from a mixture of phases are of high quality.

The variation of volume with pressure for mLd-type Mg₂Cr₂O₅ phase is tabulated in Table S4, and the EOS is presented in Fig. 6. The plot of normalized pressure versus Eulerian strain indicates that the value of K_0' is 4 as the data points have a linear distribution with a derivative equal to zero. Pressure-volume data fitted to a second order Birch-Murnaghan EOS yield $V_0 = 338.9(8) \text{ \AA}^3$ and $K_0 = 186.5(6) \text{ GPa}$ at room temperature. Our limited high temperature data fitted to the same second order Birch-Murnaghan EOS yield $V_0 = 348.5(8) \text{ \AA}^3$.

The rates of change in unit-cell lattice parameters for two CT and one mLd phases are slightly different from each other (Fig. 7), suggesting that axial compression of the orthorhombic phases of these compounds is anisotropic. The CT structure has similar compressibility in *a*- and *b*-axis but different in *c*-axis, whereas mLd-type Mg₂Cr₂O₅ phase has same compressibility in *a*- and *c*-axis but different in *b*-axis.

The density profiles of FeCr₂O₄, MgCr₂O₄, (Mg,Fe)(Al,Cr)₂O₄ and their high pressure phases including mLd and CT phases calculated at room temperature and high temperatures are plotted in Fig. 8. Low and high pressure FeCr₂O₄ phases show significantly higher densities than MgCr₂O₄ phases (Fig. 8). The CT phase of MgCr₂O₄ has higher density than the mLd-type Mg₂Cr₂O₅ phase. This is reasonable as the mLd phase is the lower pressure phase of MgCr₂O₄ and may be found at shallower depth in Earth's mantle compared to the CT phase of MgCr₂O₄. A similar trend is also observed for FeCr₂O₄ phases. It is interesting to note that the CT phase of (Mg,Fe)(Al,Cr)₂O₄ is denser than the CT phase of MgCr₂O₄, but still less dense than the CT phase of FeCr₂O₄. This

relationship can be attributed to the substitution of Al from Fe in the CT phase of $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$ compared to the Fe-free CT phase of MgCr_2O_4 , and the Fe-rich CT phase of FeCr_2O_4 . Furthermore, the presence of Al makes the high-pressure phase of chromium spinel more compressible. As a result, the density of CT phase of $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$ with the presence of Al is close to that of chromite (FeCr_2O_4). Therefore, the appearance of appreciable amounts of the $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$ polymorphs inside the planetary interiors should have an observable effect on physical properties due to their high density contrast. Furthermore, the high-temperature data of CT phase of MgCr_2O_4 and $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$ and the mLd-type $\text{Mg}_2\text{Cr}_2\text{O}_5$ phase show about 2-3.0% density drop, compared to their room-temperature data.

To evaluate the effects of bulk sound velocity of FeCr_2O_4 , MgCr_2O_4 , $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$ and their high pressure phases at high pressures, we used the obtained densities and derived EOS of those phases at room and high temperatures to calculate their bulk sound velocities and the results are shown in Fig. 9. The bulk sound velocities of MgCr_2O_4 spinel show highest values whereas CT phase of FeCr_2O_4 and $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$ are among the lowest ones. In addition, transformation of MgCr_2O_4 to mLd-type $\text{Mg}_2\text{Cr}_2\text{O}_5$ phase at ~15 GPa and to CT phase of MgCr_2O_4 at ~20 GPa would drop the velocity significantly. Note that the mLd phase of $\text{Mg}_2\text{Cr}_2\text{O}_5$ has comparable bulk sound velocity as FeCr_2O_4 spinel at room temperature and at upper mantle pressures. Transformation to CT phase of FeCr_2O_4 , MgCr_2O_4 , and $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$ all result in much lower bulk sound velocities. Importantly, high-temperature bulk sound velocity of mLd phase of $\text{Mg}_2\text{Cr}_2\text{O}_5$ and CT phase of MgCr_2O_4 and $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$ show higher values than their room-temperature ones, suggesting that high temperature (1550-1700 K) enhances the bulk sound velocity about 0.7-1.4%. In summary, our study suggests distinct density variations on FeCr_2O_4 ,

MgCr₂O₄, and (Mg,Fe)(Al,Cr)₂O₄ polymorphs without significant differences in their bulk sound velocities.

Implications

High-pressure phases of chromite phases (CF and CT) were found in Suizhou meteorite (Chen et al. 2003a, 2003b) and Luobusa ophiolite (Satukawa et al. 2015; Zhang et al. 2017). Our high-pressure synthesis study show that the chromite spinels transform to modified Ludwigite (mLd)-type phase + Cr₂O₃ at ~15 GPa and then to CaTi₂O₄-type phase above 20 GPa, in good agreement with previous reports (Ishii et al. 2014, 2015). The experimentally calibrated high-pressure phase transformations of FeCr₂O₄, MgCr₂O₄, and (Mg,Fe)(Al,Cr)₂O₄ series minerals provide a pressure gauge for shock-metamorphosed rocks and meteorites and also for mantle-derivative rocks. Previous EOS studies on spinels showed compositional dependence on the bulk moduli of Fe and Mg end-member of spinels and post-spinel phases (Siersch et al. 2017). In this study, it is extended beyond the post-spinel phases and even compared to the aluminum-bearing chromite spinel. The Fe-rich CT phase is indeed less compressible than Mg-rich and Al-bearing CT phases. The substitution of Mg and Al for Fe in the CT phase not only influences the bulk modulus but also the volumes. It is therefore an important implication for modeling the thermodynamic properties of high-pressure chromium spinel phases at high pressure and temperature conditions. In addition, determination of the isothermal bulk modulus is critical for obtaining the other thermodynamic parameters such as vibrational Grüneisen parameter. Eventually, the newly determined bulk moduli for two high pressure phases of MgCr₂O₄, and one high pressure phase of (Mg,Fe)(Al,Cr)₂O₄ in this study will help to facilitate the determination of the physical properties of the high pressure forms of chromite spinels obtained from deep mantle such as the Tibetan ophiolites (Zhang et al. 2017) and/or meteorites (Chen et al. 2003a, b).

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List of figure captions

Figure 1: X-ray diffraction patterns of high pressure phases synthesized at various pressure and temperature conditions from MgCr_2O_4 as a starting material. The synthesized phases are: (I) MgCr_2O_4 (II) Tetragonal MgCr_2O_4 (III) $\text{Cr}_2\text{O}_3+\text{MgO}$ (IV) $\text{Cr}_2\text{O}_3+\text{Mg}_2\text{Cr}_2\text{O}_5$ (V) CaTi_2O_4 -type structure of MgCr_2O_4 . Chr denotes MgCr_2O_4 , mLd denotes $\text{Mg}_2\text{Cr}_2\text{O}_5$, Cr denotes Cr_2O_3 , Au denotes gold, Ne denotes neon, CT denotes CaTi_2O_4 -type phase and TT denotes tetragonal MgCr_2O_4 . Vertical bars are the calculated peak positions for the different phases present.

Figure 2: Phase diagram of MgCr_2O_4 . The solid lines are phase boundaries from this study.

The red dotted lines are from Ishii et al. (2015).

Figure 3: Volumes as a function of pressure for the CT phase of MgCr_2O_4 , natural $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$ and synthetic FeCr_2O_4 . (i) Natural $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$: ambient volume $V_0 = 261(1) \text{ \AA}^3$, Bulk Modulus $K_0 = 175.4(2) \text{ GPa}$, and $K_0' = 4$; (ii) MgCr_2O_4 : $V_0 = 264.4(8) \text{ \AA}^3$, $K_0 = 185.4(4) \text{ GPa}$, and $K_0' = 4$; and (iii) synthetic FeCr_2O_4 (Ishii et al. 2014) : $V_0 = 266.95 \text{ \AA}^3$, $K_0 = 199 \text{ GPa}$, and $K_0' = 4$. Pressure derivative of the isothermal bulk modulus (K_0') was determined using normalized pressure and Eulerian Strain relationship for both CT phase of natural $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$ (lower left inset a) and MgCr_2O_4 (upper right inset b).

Figure 4: Comparison between the Birch-Murnaghan EOS fit to volume of Cr_2O_3 as a function of pressure (this study) and that of Dera et al. (2011).

Figure 5: Pressure-volume relationship of modified Ludwigite-type $\text{Mg}_2\text{Cr}_2\text{O}_5$ phase. The inset shows linear relationship of normalized pressure and Eulerian strain for modified Ludwigite-type $\text{Mg}_2\text{Cr}_2\text{O}_5$ phase.

Figure 6: Compressibility of the (top) CT phase of MgCr_2O_4 , (middle) CT phase of natural $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$ and (bottom) modified Ludwigite-type $\text{Mg}_2\text{Cr}_2\text{O}_5$ at high pressures. The three phases all show weakly anisotropic behavior.

Figure 7: Density as a function of pressure for MgCr_2O_4 , FeCr_2O_4 , mLd-type $\text{Mg}_2\text{Cr}_2\text{O}_5$, CT phase of MgCr_2O_4 , CT phase of FeCr_2O_4 and CT phase of natural $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$. N chromite stands for natural $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$.

Figure 8: Bulk sound velocity as a function of pressure for MgCr_2O_4 , FeCr_2O_4 , mLd-type $\text{Mg}_2\text{Cr}_2\text{O}_5$, CT phase of MgCr_2O_4 , CT phase of FeCr_2O_4 and CT phase of natural $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$. N chromite stands for natural $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$.

Tables

Table 1: Calculated and observed d -spacings of synthesized CT phase of MgCr_2O_4 at ambient conditions.

$d_{\text{obs.}}$ (Å)	hkl	$d_{\text{calc.}}$ (Å)	Δd (Å)
4.8454 (2)	002	4.8491 (1)	-0.0037
4.2731 (1)	021	4.2713 (1)	0.0018
3.3979 (1)	022	3.3959 (1)	0.0019
2.7401 (3)	110	2.7431 (2)	-0.0029
2.6736 (1)	023	2.6738 (1)	-0.0002
2.6372 (1)	111	2.6394 (1)	-0.0021
2.3808 (2)	040	2.3788 (2)	0.0020
2.1631 (0)	024	2.1602 (2)	0.0028
2.1243 (1)	130	2.1259 (1)	-0.0016
2.0886 (2)	113	2.0913 (1)	-0.0027
2.0749 (2)	131	2.0765 (1)	-0.0020

1.9464 (2)	132	1.9469 (3)	-0.0005
1.8166 (1)	114	1.8164 (1)	0.0002
1.7971 (1)	025	1.7961 (1)	0.0009
1.7765 (3)	133	1.7761 (1)	0.0004
1.6169 (1)	006	1.6164 (3)	0.0005
1.5846 (1)	150	1.5851 (1)	-0.0006
1.5641 (2)	151	1.5643 (3)	-0.0002
1.5066 (4)	152	1.5066 (2)	-0.0005
1.5036 (1)	045	1.5032 (1)	0.0003
1.4321 (1)	135	1.4328 (1)	-0.0006
1.4243 (2)	063	1.4237 (1)	0.0006
1.3931 (2)	116	1.3925 (1)	0.0007
1.3775 (4)	202	1.3736 (1)	0.0039
1.3374 (2)	046	1.3369 (2)	0.0005
1.3262 (2)	154	1.3267 (1)	-0.0005
1.2832 (1)	136	1.2866 (1)	-0.0034
1.2612 (2)	223	1.2625 (1)	-0.0013
1.2269 (2)	240	1.2271 (1)	-0.0006
1.2164 (1)	241	1.2173 (1)	-0.0003
1.1955 (1)	047	1.1972 (2)	-0.0020
1.1901 (1)	080	1.1894 (2)	0.0007
1.1807 (4)	081	1.1805 (2)	0.0002
1.1611 (4)	137	1.1606 (1)	0.0004
1.1205 (1)	225	1.1197 (1)	0.0008

1.1099 (1) 118 1.1087 (1) 0.0012

*Numbers in the parenthesis are the errors at the last digit.

Table 2: Comparison of volumes, bulk moduli and their pressure derivatives of CT phases, mLd-type $\text{Mg}_2\text{Cr}_2\text{O}_5$, and Cr_2O_3 obtained from this study and previous reports.

Polymorphs	V_0 (\AA^3)	K_0 (GPa)	K_0'
CT phase of MgCr_2O_4	264.4(8)	185.4(4)	4.0
CT phase of natural $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$	261.0(1)	175.4(2)	4.0
mLd-type $\text{Mg}_2\text{Cr}_2\text{O}_5$	338.9(8)	186.5(6)	4.0
CT phase of FeCr_2O_4 (Ishii et al. 2014)	285.22(1)	199.0(0)	4.0
Cr_2O_3	289.1(8)	214.0 (5)	4.7(2)
Cr_2O_3 (Dera et al. 2011)	288.6(2)	220.0 (4)	4.7(2)

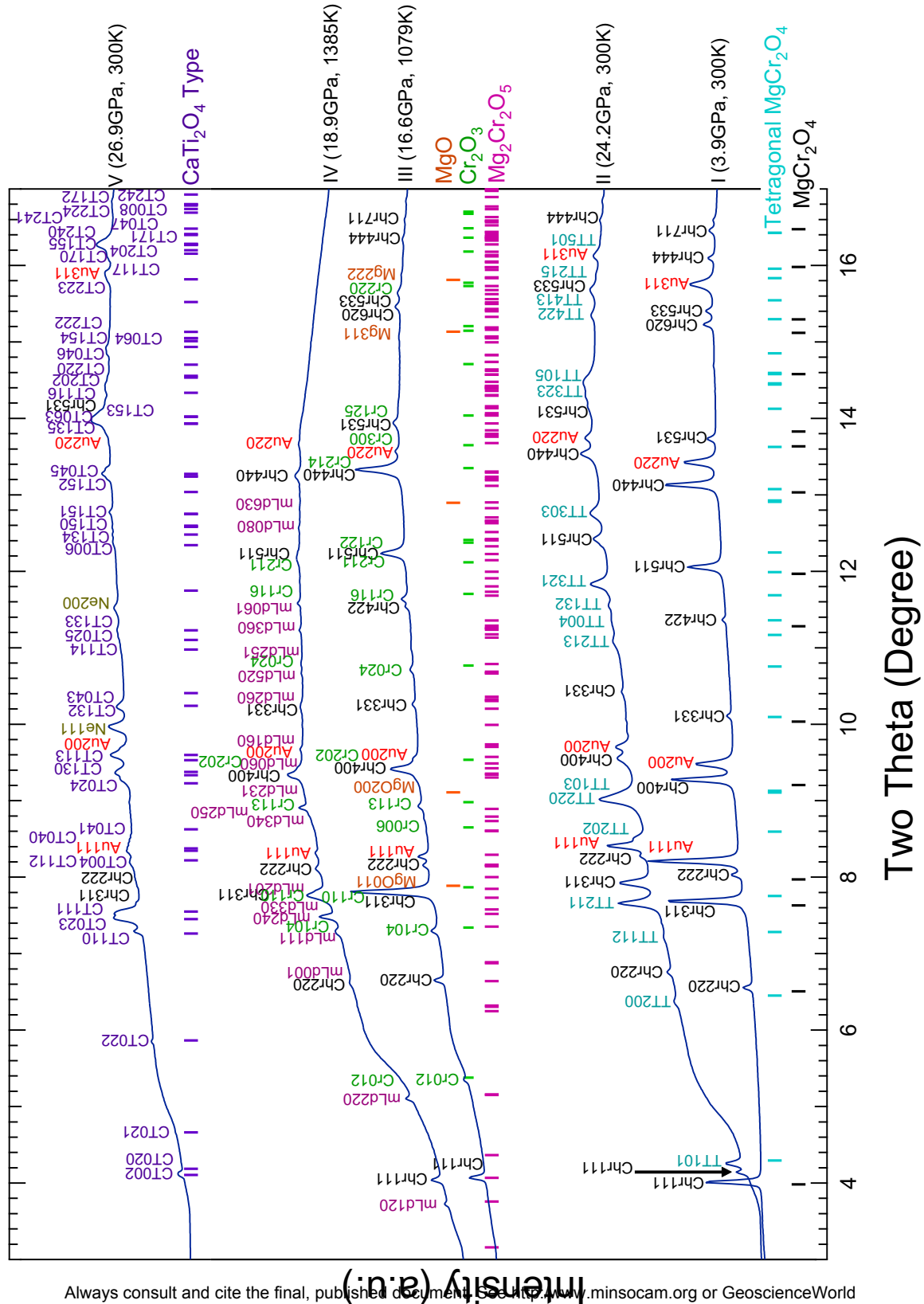


Figure 1: X-ray diffraction patterns of high pressure phases synthesized at various pressure and temperature conditions from MgCr_2O_4 as a starting material. The synthesized phases are: (I) MgCr_2O_4 (II) Tetragonal MgCr_2O_4 (III) $\text{Cr}_2\text{O}_3+\text{MgO}$ (IV) $\text{Cr}_2\text{O}_3+\text{Mg}_2\text{Cr}_2\text{O}_5$ (V) CaTi_2O_4 -type structure of MgCr_2O_4 . Chr denotes MgCr_2O_4 , mLd denotes $\text{Mg}_2\text{Cr}_2\text{O}_5$, Cr denotes Cr_2O_3 , Au denotes gold, Ne denotes neon, CT denotes CaTi_2O_4 -type phase and TT denotes tetragonal MgCr_2O_4 . Vertical bars are the calculated peak positions for the different phases present.

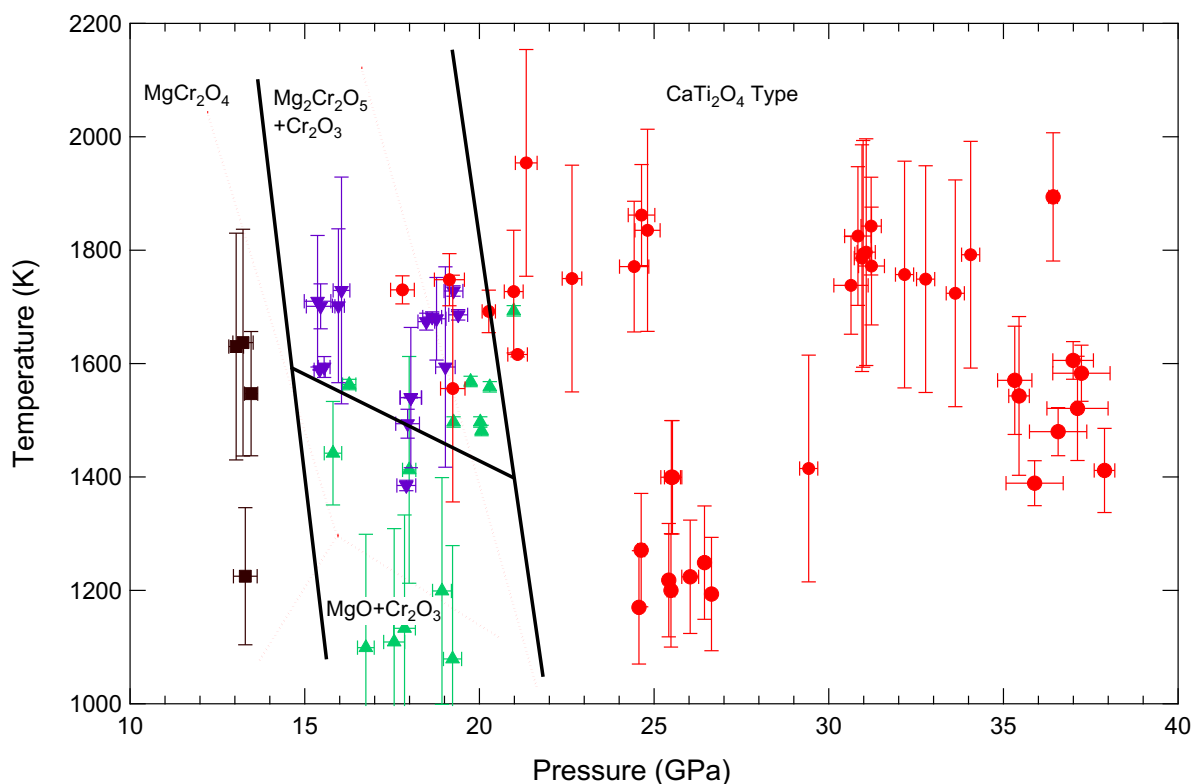


Figure 2: Phase diagram of MgCr_2O_4 . The solid lines are phase boundaries from this study. The red dotted lines are from Ishii et al. (2015).

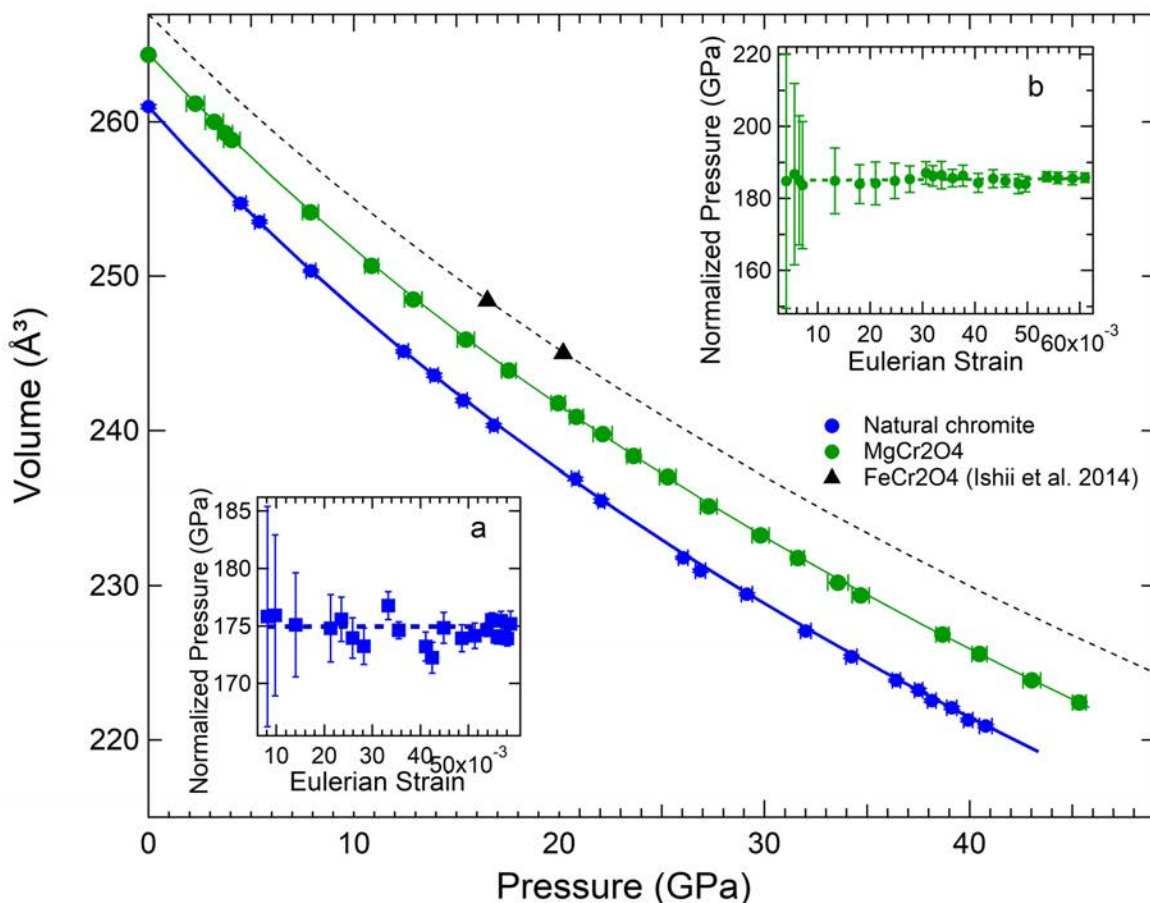


Figure 3: Volumes as a function of pressure for the CT phase of MgCr_2O_4 , natural $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$ and synthetic FeCr_2O_4 . (i) Natural $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$: ambient volume $V_0 = 261(1) \text{ \AA}^3$, Bulk Modulus $K_0 = 175.4(2) \text{ GPa}$, and $K_0' = 4$; (ii) MgCr_2O_4 : $V_0 = 264.4(8) \text{ \AA}^3$, $K_0 = 185.4(4) \text{ GPa}$, and $K_0' = 4$; and (iii) synthetic FeCr_2O_4 (Ishii et al. 2014) : $V_0 = 266.95 \text{ \AA}^3$, $K_0 = 199 \text{ GPa}$, and $K_0' = 4$. Pressure derivative of the isothermal bulk modulus (K_0') was determined using normalized pressure and Eulerian Strain relationship for both CT phase of natural $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$ (lower left inset a) and MgCr_2O_4 (upper right inset b).

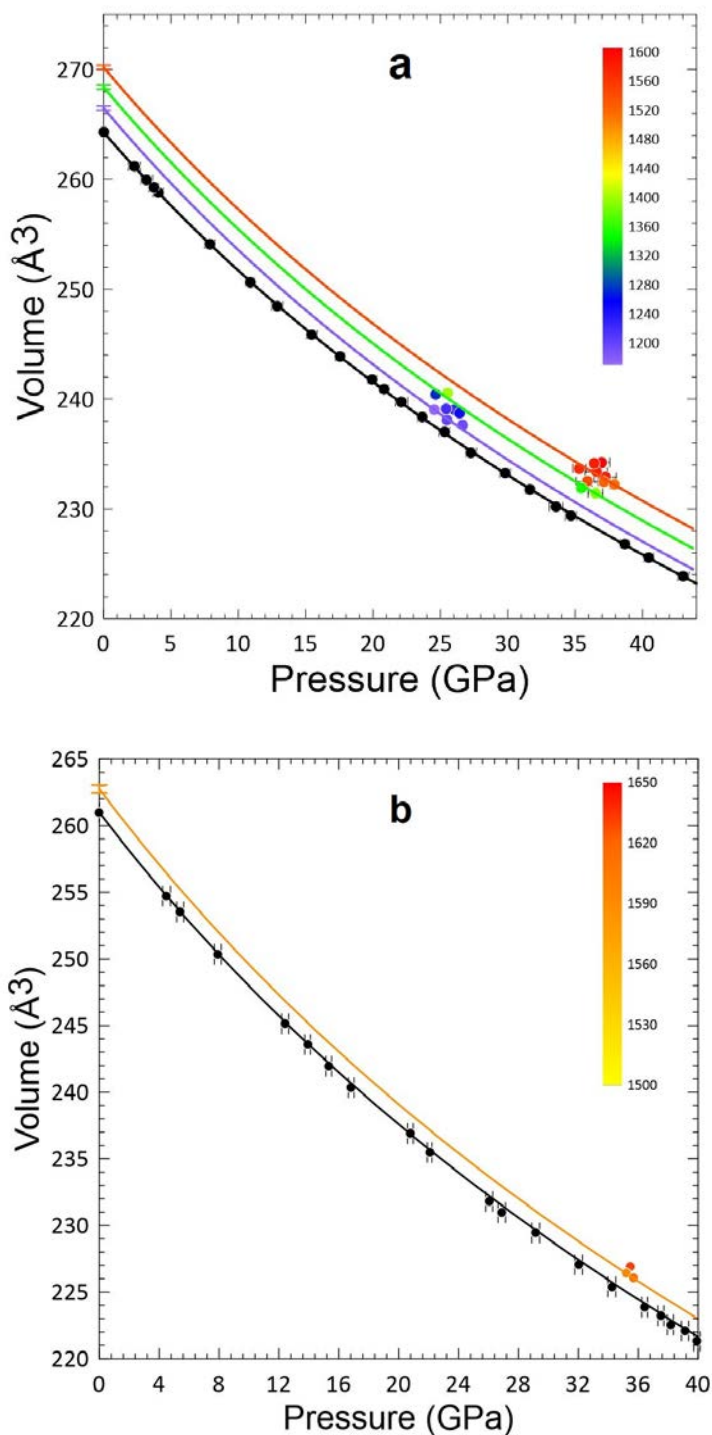


Figure 4: Pressure-volume-temperature data and isotherms of 1200, 1400 and 1550 K for the CT phase of MgCr_2O_4 (a) and 1600 K for the CT phase of natural $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$ (b). Black circles are room-temperature data and color symbols are high-temperature data.

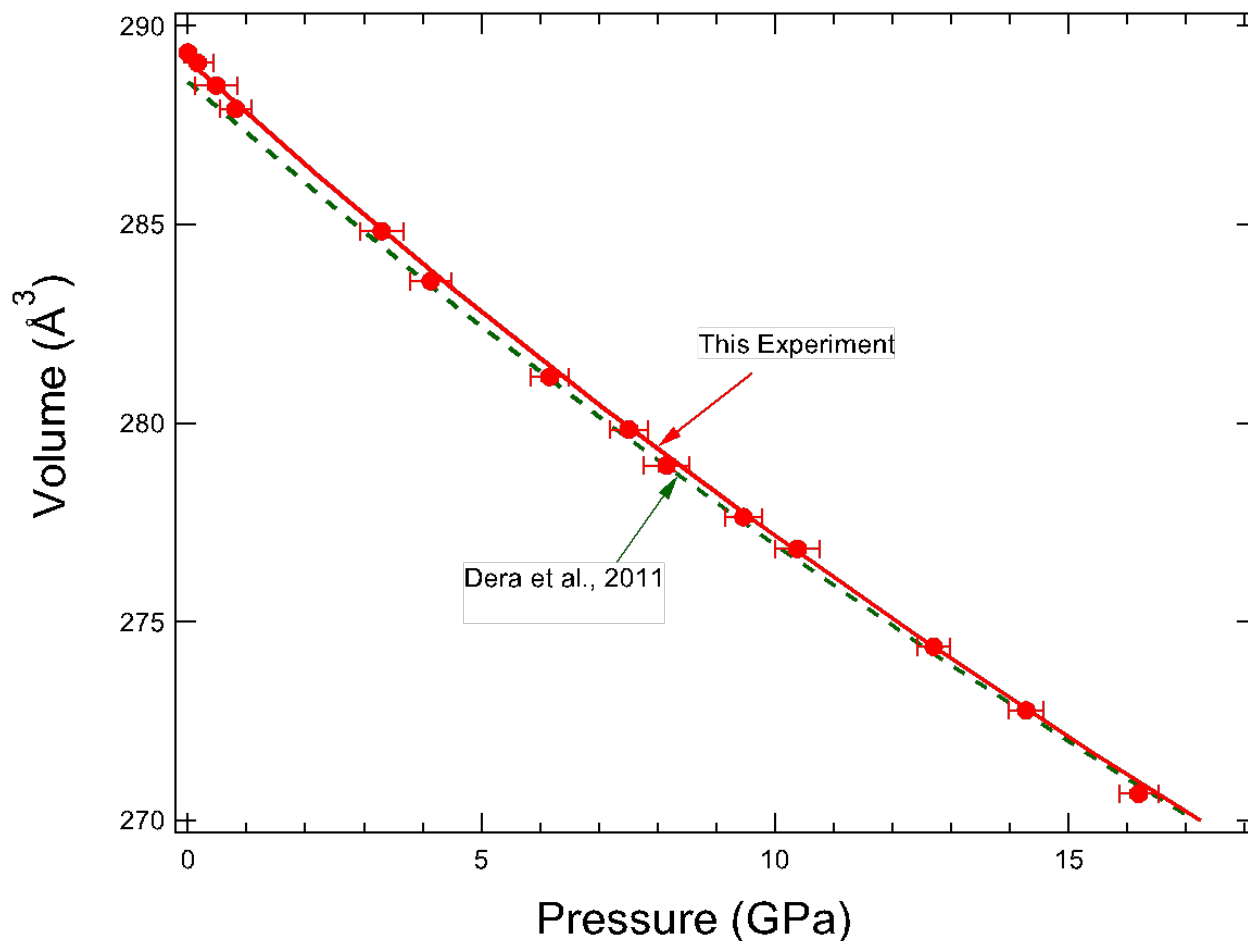


Figure 5: Comparison of the Birch-Murnaghan EOS fitting curve of Cr_2O_3 between this study and Dera et al. (2011).

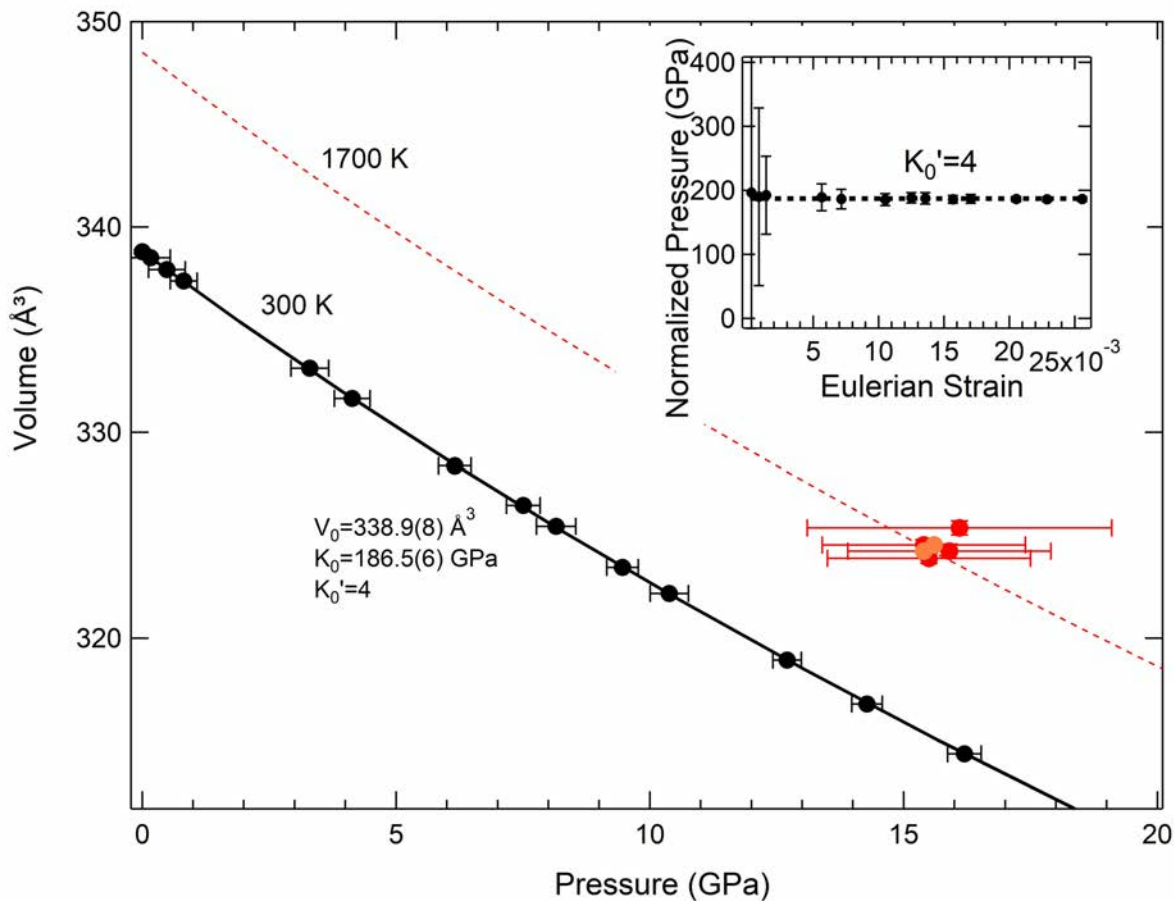
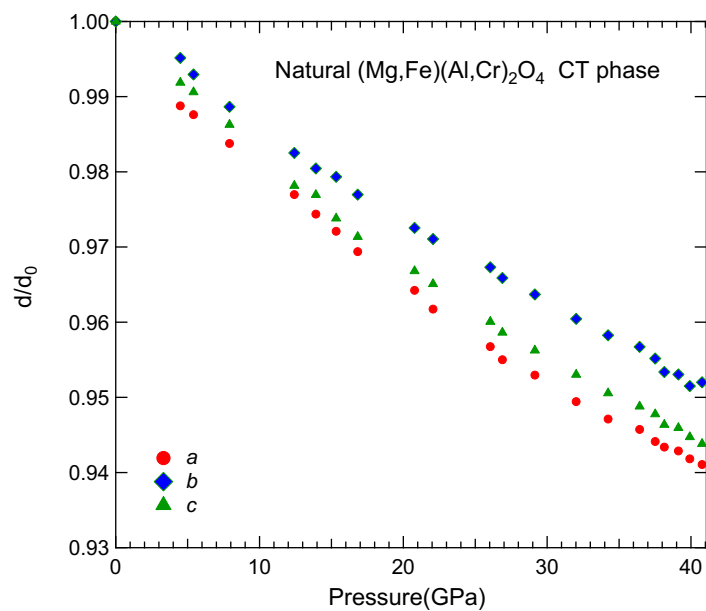
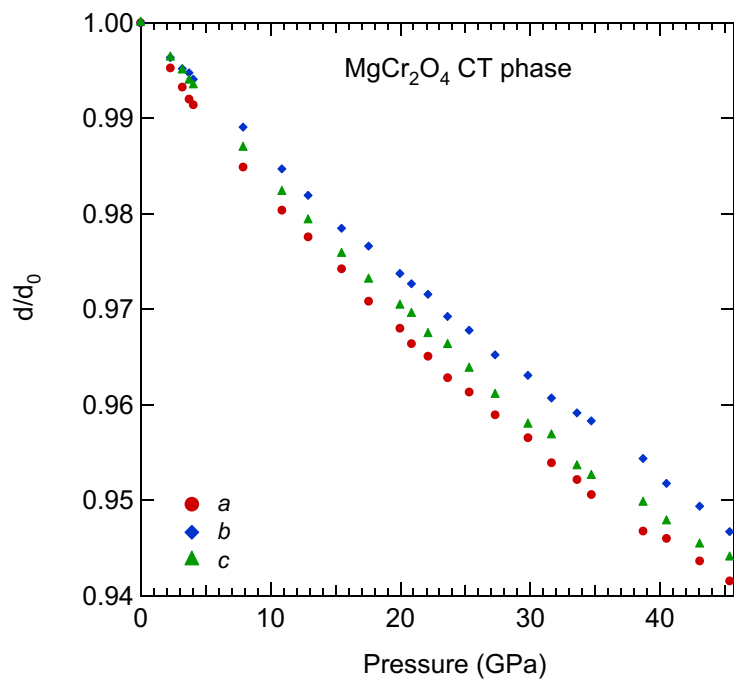


Figure 6: Pressure-volume-temperature data of modified Ludwigite-type $\text{Mg}_2\text{Cr}_2\text{O}_5$ phase. The inset shows a linear relationship of normalized pressure and Eulerian strain for modified Ludwigite-type $\text{Mg}_2\text{Cr}_2\text{O}_5$ phase. Black circles are data from room temperature, orange circles are data from 1590 K and red circles are data from 1700-1730 K.



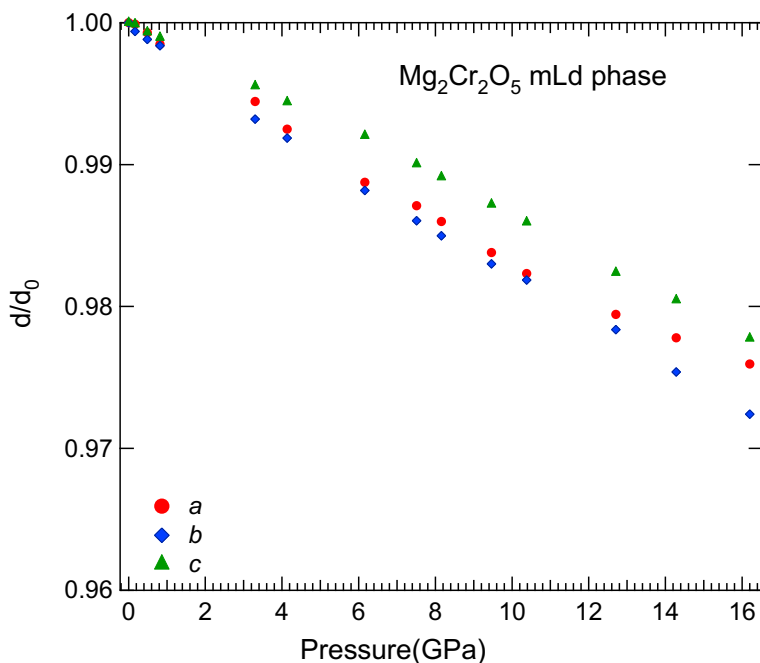


Figure 7: Axial compressibility of (top) the CT phase of MgCr₂O₄, (middle) the CT phase of natural (Mg,Fe)(Al,Cr)₂O₄ and (bottom) the modified Ludwigite-type Mg₂Cr₂O₅ at high pressures. Three phases all show weak anisotropic behavior.

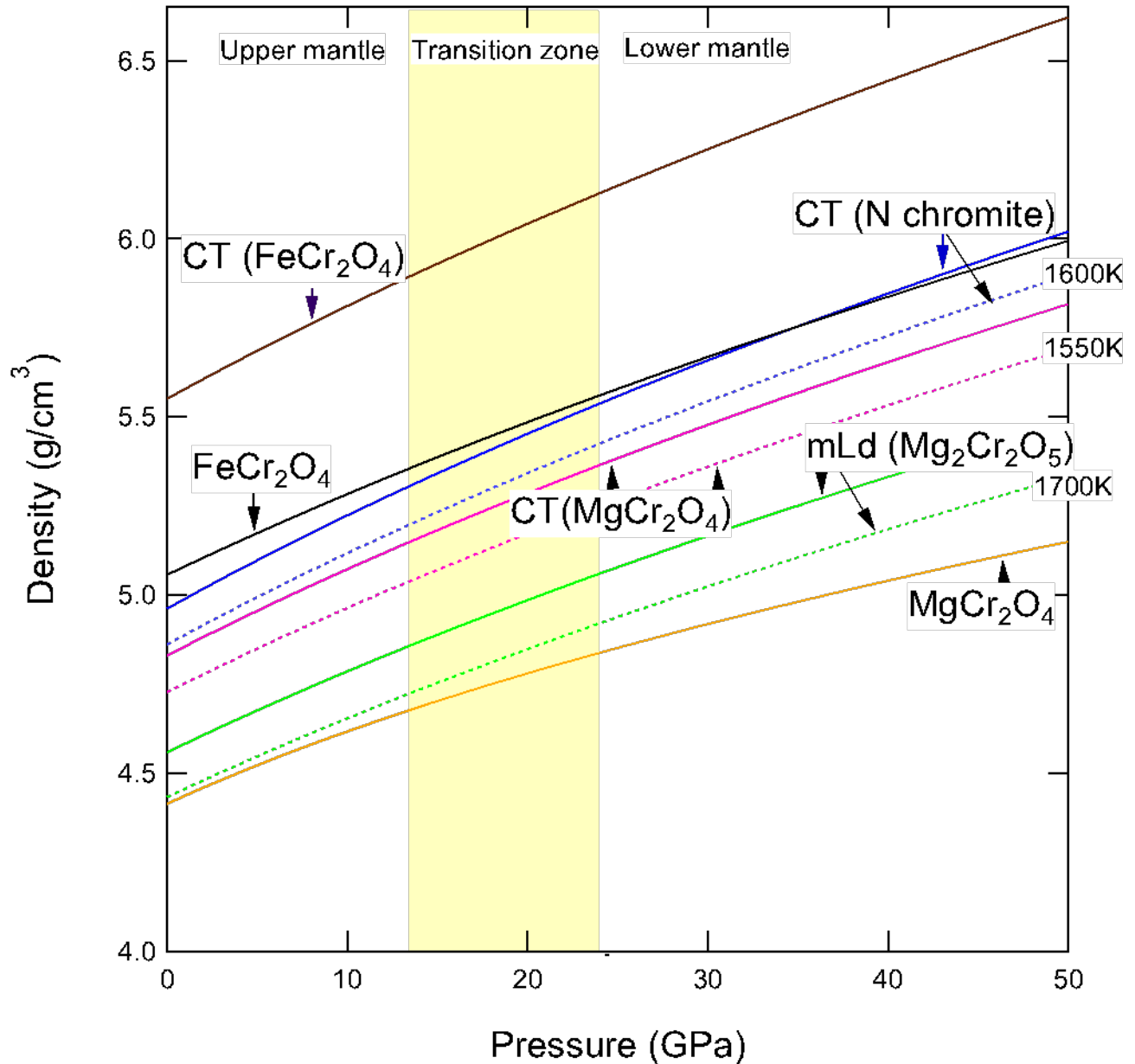


Figure 8: Density as a function of pressure for MgCr₂O₄, FeCr₂O₄, mLd-type Mg₂Cr₂O₅, CT phase of MgCr₂O₄, CT phase of FeCr₂O₄ and CT phase of natural (Mg,Fe)(Al,Cr)₂O₄. N chromite stands for natural (Mg,Fe)(Al,Cr)₂O₄. Solid curves are data from room temperature and dashed curves are data from high temperatures.

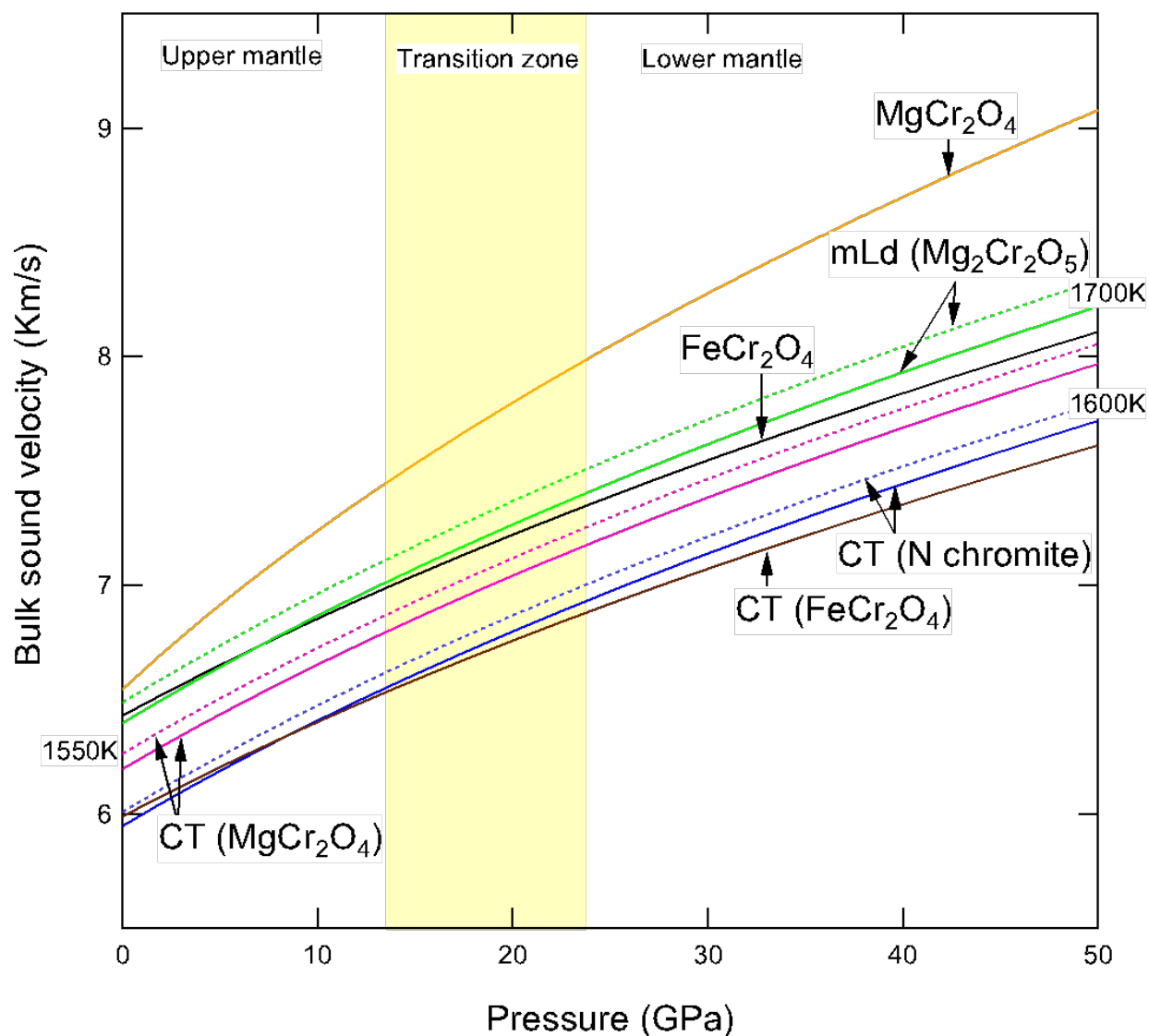


Figure 9: Bulk sound velocity as a function of pressure for MgCr_2O_4 , FeCr_2O_4 , mLd-type $\text{Mg}_2\text{Cr}_2\text{O}_5$, CT phase of MgCr_2O_4 , CT phase of FeCr_2O_4 and CT phase of natural $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$. N chromite stands for natural $(\text{Mg,Fe})(\text{Al,Cr})_2\text{O}_4$. Solid curves are data from room temperature and dashed curves are data from high temperatures.