

1 **REVISION R**

2 **THERMOELASTIC PROPERTIES OF ZIRCON: IMPLICATIONS FOR**
3 **GEOtherMOBAROMETRY**

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18 **ABSTRACT**

19 A thermal-pressure equation of state has been determined for zircon (ZrSiO₄) that characterizes
20 its thermoelastic behavior at metamorphic conditions. New pressure-volume (*P-V*) data from a
21 “Mud-Tank” zircon have been collected from 1 bar to 8.47(1) GPa using X-ray diffraction, and
22 elastic moduli were measured from room temperature up to 1172 K by resonance ultrasound
23 spectroscopy. These data were fitted simultaneously with temperature-volume (*T-V*) data from
24 the literature in EosFit7c using a new scaling technique. The parameters of a third order Birch-
25 Murnaghan EoS with a Mie-Grüneisen-Debye model for thermal pressure has compressional
26 EoS parameters $K_{0T} = 224.5(1.2)$ GPa, $K_{0T}' = 4.90(31)$ with a fixed initial unit-cell volume $V_0 =$
27 39.26 cm³/mol and thermal parameters $\gamma_0 = 0.868(15)$, $q = 2.37(80)$, and $\Theta_D = 848(38)$ K. EoS
28 parameters that describe the variation of unit-cell parameters with pressure and temperature were
29 determined using an isothermal-type EoS. This new EoS confirms that zircons are stiffer than
30 garnets and exhibit a much lower thermal expansion. This results in steep isomekes between

31 zircon and garnets, which makes zircon trapped as inclusions in garnets at metamorphic
32 conditions a good piezothermometer.

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INTRODUCTION

35 Zircon (ZrSiO_4) is an important and widespread mineral in the Earth's crust and upper
36 mantle, commonly used to date geologic events using the U-Th-Pb geochronometer (e.g.,
37 Hanchar and Hoskin 2003). Because it is highly refractory, zircon is a common detrital
38 component in many sedimentary deposits (e.g., Fedo et al. 2003) and can also be found as an
39 accessory mineral in sedimentary, metamorphic, and igneous rocks (e.g., Finch and Hanchar
40 2003). Zircon can also be produced during prograde metamorphism as a result of the breakdown
41 of minerals bearing Zr as a minor or trace component. It is therefore common for zircon crystals
42 to be found trapped as inclusions, frequently within garnet hosts, as a result of garnet growth
43 during prograde metamorphism. Zircon inclusions in garnet therefore have the potential to be
44 used in piezobarometry in which the residual stress or pressure in the inclusions, arising from the
45 contrast in the elastic properties of garnet and zircon, can be used to infer entrapment conditions
46 (e.g., Angel et al. 2015). A reliable equation of state (EoS) for zircon is required for these
47 calculations. However, zircon EoS parameters are poorly constrained. Reported isothermal bulk
48 modulus values at room conditions vary substantially between $K_{0T} = 198$ GPa (Ono et al. 2004)
49 and 227 GPa (Hazen and Finger 1979). A redetermination of the P - V - T EoS of zircon from the
50 data available in the literature yields $K_{0T} = 233$ GPa and a pressure derivative of the bulk
51 modulus of $K_{0T}' = -0.56$ (Zaffiro 2019), while the most recent ab-initio calculations report $K_{0T}' =$
52 4.71 (Stangarone et al. 2019).

53 In this paper, we present new data to resolve discrepancies between reported zircon EoS
54 and determine a reliable thermal-pressure EoS. P - V data were measured using single-crystal X-
55 ray diffraction, and values of the adiabatic bulk modulus, K_S , were obtained from the elastic
56 tensor of a non-metamict zircon at high temperatures determined using resonant ultrasound
57 spectroscopy (RUS). A Mie-Grüneisen-Debye (MGD) thermal-pressure EoS was determined
58 from this new data plus literature data using a new scaling method in the fitting to remove bias
59 and ensure consistency. The moduli values determined from the fitting and discussed in this
60 paper are Reuss bound values, appropriate for describing the properties of zircon under

61 hydrostatic pressure. In this paper, we show that not only does our thermal-pressure EoS for
62 zircon fit these data well, but the isobaric heat capacity C_p calculated from our EoS closely
63 matches the experimentally-determined C_p values from the literature.

64

65 **EXPERIMENTAL METHODS**

66 ***P-V* study**

67 A portion of the standard sample UWZ-1, originating from the Mud Tank carbonatite
68 complex near Alice Springs, Australia, was kindly provided by John Valley (University of
69 Wisconsin). This sample is characterized by very low U/Th substitution and low amorphization,
70 and has an estimated age of 732 Ma (e.g., Jackson et al. 2004; Yuan et al. 2018). Samples from
71 the UWZ-1 bulk crystal were optically colorless which indicated low trace element abundances
72 and low $^{176}\text{Hf}/^{177}\text{Hf}$ isotope ratios (Woodhead and Hergt 2005; Gain et al. 2019). SEM imaging,
73 SIMS analyses of δO^{18} and OH/O, and δO^{18} laser fluorination measurements on the UWZ-1
74 sample by John Valley (pers. comm.) are consistent with chemical homogeneity.

75 Single-crystal X-ray diffraction experiments were conducted to determine the isothermal
76 equation of state of Mud-Tank zircon. A single crystal of dimensions $160\ \mu\text{m} \times 110\ \mu\text{m} \times 40\ \mu\text{m}$
77 was cut from this sample and loaded into an ETH-type diamond-anvil cell (Miletich et al. 2000)
78 along with a single crystal of quartz as a pressure calibrant (Scheidl et al. 2016). Room-pressure
79 unit-cell parameters were collected at 296 K, after which a pressure medium of 4:1
80 methanol:ethanol solution was loaded into the diamond-anvil cell and the cell was increased to
81 higher pressures. Unit-cell parameters were collected using a Huber four-circle X-ray
82 diffractometer with $\text{MoK}\alpha$ radiation, run by SINGLE software (Angel and Finger 2011). Unit-
83 cell parameters of zircon were determined based on the 8-position centering method (King and
84 Finger 1979) using 8-9 reflections, with $12.45 \leq 2\theta \leq 28.03$. In total, 18 measurements were
85 collected in increasing pressure increments from 1 bar to 8.47(1) GPa. Three additional data
86 points collected upon decompression of the cell were consistent within uncertainties with those
87 collected upon compression (Table 1).

88 **RUS measurements**

89 A second sample of the Mud-Tank zircon, sourced independently from a mineral dealer,
90 was prepared in the form of a rectangular parallelepiped with polished faces parallel to (100),
91 (010) and (001) within 0.5° as determined by X-ray diffraction measurements. It had dimensions
92 of $4.637 \times 4.628 \times 3.166 \text{ mm}^3$ and a mass of 0.3167 g, which corresponds to a density of 4.661
93 g/cm^3 , in comparison with a theoretical density calculated from the measured lattice parameters
94 of 4.663 g/cm^3 . There were chips out of the edges of the crystal but the fraction of the total
95 volume and mass of the sample that these represented was less than 2 parts per million. No
96 cracks, inclusions, or other imperfections were visible inside the crystal, which was optically
97 clear. Laser ablation ICP-MS measurements on a fragment of this specimen showed that, despite
98 some zoning visible by cathodoluminescence, the trace elements were homogeneous with
99 concentrations, including Hf^{177} , mostly below the median trace element abundances of Mud
100 Tank zircon as given in Gain et al. (2019). The sole exception was Nb^{93} with an average of
101 $17.25(0.3) \text{ ppm Nb}$ in our sample compared with 7.99 ppm Nb as a global average in Mud Tank
102 zircons (Gain et al. 2019).

103 The RUS technique has been described in detail by Migliori and Sarrao (1997). The
104 Cambridge equipment makes use of DRS Modulus II electronics for data collection at room
105 temperature and Stanford electronics (Migliori and Maynard 2005) for data collection at high
106 temperatures. Measurements at room temperature were performed with the crystal resting
107 directly between two PZT piezoelectric transducers. For measurements at high temperatures the
108 crystal was held lightly across a pair of opposite corners between the tips of a pair of horizontal
109 alumina buffer rods which are inserted into a Netzsch resistance furnace (McKnight et al. 2008).
110 The driving and detecting transducers were attached to the ends of the buffer rods, outside of the
111 furnace. Temperature was measured with a thermocouple placed within a few millimeters of the
112 sample. A further small adjustment of the measured temperature scale was made by calibration
113 against the α - β transition in quartz which gives a clear and sharp minimum in elastic moduli at
114 846 K. The estimated accuracy of measured temperatures was considered to be better than $\pm 2 \text{ K}$.
115 High-temperature spectra were collected in an automated heating and cooling cycle with nominal
116 temperature steps of 100 K up to $\sim 1200 \text{ K}$. A settle time of 20 minutes was allowed for thermal
117 equilibration of the sample before the data collection at each temperature.

118 Values of the six independent elastic constants for crystallographic point group $4/\text{mmm}$ at
119 room temperature were determined by fitting to the resonance frequencies of 52 peaks between

120 0.3 and 1.5 MHz using the DRS software (Migliori and Sarrao 1997). Not all the same
121 resonances could be detected in the spectra collected at high temperatures primarily due to
122 attenuation of the signal by the buffer rods. As a consequence, the frequencies of between 41 and
123 49 resonance peaks were used for fitting of the high-temperature single-crystal elastic moduli. In
124 order to obtain an internally consistent data set, the highest temperature data were fit first and the
125 results were used as the starting values at the next temperature down. Changes in the shape of the
126 crystal at each high temperature were calculated from a preliminary determination of the thermal
127 expansion coefficient of zircon from literature data that is indistinguishable from the final P - V - T
128 EoS described below. Root-mean-squared errors from the fitting were in the range of 0.31 to
129 0.37%. Values of the inverse mechanical quality factor, Q^{-1} , taken as $\Delta f/f$ where Δf was the peak
130 width at half maximum height for a resonance peak with frequency $f \sim 1.0$ MHz, were close to
131 10^{-4} at each temperature. This low value is consistent with the sample being a high-quality single
132 crystal. There was a slight dependence of final values of the elastic moduli on the starting values
133 used in each case, signifying that the fitting surface has local minima. Uncertainties of the
134 individual moduli were derived from the curvature of the solution surface in the vicinity of the
135 minimum point (Migliori et al. 1990; Migliori and Maynard 2005) and therefore do not include
136 uncertainties due to specimen shape, size, or orientation.

137

138

RESULTS

139 Compressional Study

140 The unit-cell volume of the Mud-Tank zircon was found to smoothly decrease as a
141 function of pressure (Figure 1), up to a maximum hydrostatic pressure of 8.47(1) GPa. A fit of a
142 third-order Birch-Murnaghan equation of state (EoS) with EosFit-GUI (Gonzalez-Platas et al.
143 2016) with full weights (Angel et al. 2014a) yielded the coefficients given in Table 2. Statistics
144 improved marginally with a fit of a fourth-order Birch-Murnaghan EoS, with a slightly lower χ_w^2
145 and a minimal change in the values of the EoS parameters. P - V data points display significant
146 curvature (Figure 1) and the data in the f - F plot exhibit a significantly positive slope implying
147 that $K_{0T}' > 4$ and eliminating the possibility of a second-order Birch-Murnaghan EoS fitting the
148 data. A third- or fourth-order Birch-Murnaghan EoS therefore provides the best statistical fit to
149 our data.

150 The variation of the Mud-Tank zircon's unit-cell parameters with pressure (Figure
151 **1**) displays an anisotropic axial compressibility, with a/a_0 being more compressible than
152 c/c_0 . These results are consistent with axial compressibility data derived from the elastic
153 tensor of zircon (Özkan et al. 1974). The axial data in the f - F plots displayed a linear
154 trend, with a positive slope for a and a negative slope for c ; therefore, these data were fit
155 with third-order Birch-Murnaghan EoS to obtain the axial parameters given in Table **2**.

156

157 **RUS results**

158 The elastic moduli measured before heating are reported in Table **3** together with the
159 high-temperature results, along with the adiabatic Reuss bulk (K_S) and shear moduli (G_S) and the
160 adiabatic linear moduli, M_{aS} and M_{cS} , for the a - and c -axes calculated from the individual tensor
161 components. The estimated standard deviations (e.s.d.'s) associated with the bulk and linear
162 moduli derived from the e.s.d.'s of the C_{ij} values are on the order of $\sim 1\%$. The elastic moduli do
163 not show any anomalous behavior that would indicate significant decomposition or a displacive
164 phase transition of the crystal; the room-temperature values of the moduli at the end of the run
165 are within the experimental uncertainties of those at the start, except for C_{66} which was slightly
166 stiffer at the end. If this difference is real, it may indicate some slight change in crystallinity in
167 the sample as a consequence of heating.

168 Our values of the individual shear moduli C_{44} and C_{66} at room temperature, and the value
169 of the Reuss average shear modulus G_S , agree with those derived from ultrasonic wave velocity
170 measurements of non-metamict zircon by Özkan et al. (1974). But our values of compressional
171 moduli C_{11} and C_{33} are 2% higher, and C_{12} and C_{13} about 7% higher, than those previous
172 measurements. This leads to a bulk modulus at room conditions (Figure **2**) that is approximately
173 3.5% higher, although our measured temperature dependence dK_S/dT is in good agreement with
174 the data of Özkan and Jamieson (1978) as re-evaluated by Özkan (2008). This difference cannot
175 be due to radiation damage which softens the bulk modulus (Binvignat et al. 2018) because the
176 diffraction peak widths of the sample used for the compression experiment indicate a very low
177 degree, if any, of radiation damage in that sample. This suggests that the offset in bulk moduli
178 values may be a systematic error in our data that arises from the fact that the resonances of a
179 millimeter-sized sample primarily involve shearing motions and relatively little breathing motion

180 so the shear elastic constants are constrained more tightly than those which contribute to the bulk
181 modulus. The small misorientation errors and slight damage to the edges of the sample may also
182 contribute to the offset in bulk moduli values.

183 ***P-V-T EoS***

184 Pressure-volume data from the compressional study and the RUS data were combined
185 with temperature-volume data reported in the literature to determine the *P-V-T* EoS of zircon.
186 Only the *T-V* data in the range 100–1200 K were considered in the current analysis since the
187 unit-cell parameters can be affected by the decomposition of zircon or a proposed displacive
188 structural change at about 1200 K (e.g., Mursic et al. 1992). In order to allow for the different
189 calibrations of diffractometers used to collect the published *T-V* data, each dataset was first
190 scaled by the measured volume at room conditions to obtain V/V_0 (Figure 3), and then
191 recalculated as molar volumes by using $V_0 = 39.260 \text{ cm}^3/\text{mol}$ (Holland and Powell 2011). This
192 means that published datasets without measurements at room conditions had to be excluded
193 from fitting. Individual data points that are significant outliers from the general trends of the
194 literature data, whether in volume or cell parameters (e.g., Subbarao and Gokhale 1968; Bayer
195 1972), were also excluded. All data used in fitting the EoS are listed in Table 4. Fits were
196 performed with EosFit7c (Angel et al. 2014a), using the methods of Milani et al. (2017) to fit
197 the EoS to both volume and the adiabatic bulk moduli data simultaneously. When converted to
198 molar volumes, the new *P-V* data implied a very slightly different value of V_0 than the value
199 from Holland and Powell (2011). Rather than scaling these data in advance of the fitting, which
200 would bias the final results and parameter values, we have implemented the refinement of
201 dataset scale factors in EosFit7c, and separate scale factors were refined for the *V-T*, *P-V*
202 datasets and the dataset of bulk moduli from the RUS measurements.

203 A thermal pressure EoS was employed to fit the data listed in Table 4, in which the
204 pressure at any V and T is considered as the sum of the reference pressure P_{ref} needed to reach a
205 volume V at a reference temperature T_0 , and the thermal pressure ΔP_{th} necessary to travel along
206 an isochor to reach a final temperature T . The thermal pressure induced by heating along the
207 isochor is given by the thermodynamic identity (e.g., Anderson 1995):

$$\Delta P_{th} = \int_{T_0}^T (\alpha_V K_T)_V dT. \quad (1)$$

208 Different thermal-pressure EoS are distinguished by the method used to calculate ΔP_{th}
209 through (1). The application of the Debye model in the MGD EoS is advantageous as it presents
210 a simple technique to model ΔP_{th} using relatively few parameters and assumptions. The MGD
211 EoS uses the Grüneisen relation to define the relationship between the elastic properties of a
212 material and its heat capacity:

$$\alpha_V K_T = \frac{\gamma C_{vm}}{V_m}, \quad (2)$$

213 where C_{vm} is the molar heat capacity at constant volume, V_m is the molar volume, and γ is
214 the dimensionless Grüneisen coefficient (e.g., Grüneisen 1912; Anderson 1995). It follows from
215 (2) that ΔP_{th} can also be expressed in terms of γ and C_{vm} :

$$\Delta P_{th} = \int_{T_0}^T \left(\frac{\gamma C_{vm}}{V_m} \right)_V \partial T, \quad (3)$$

216 where T and T_0 are the final and initial reference temperature conditions. The MGD EoS uses the
217 Debye model of the phonon density of states to define C_{vm} as:

$$C_{vm} = 9NR \left(\left(\frac{T}{\theta_D} \right)^3 \int_0^{\theta_D/T} \frac{x^4 e^x}{(e^x - 1)^2} dx \right), \quad (4)$$

218 where R is the gas constant, θ_D is the Debye temperature, and N is the number of atoms in the
219 formula unit. The quasi-harmonic approximation (QHA) assumes that γ is only a function of
220 volume, allowing γ to be removed from the integral in (3). We can incorporate (4) into a
221 simplified (3) to define the thermal pressure (ΔP_{th}):

$$\Delta P_{th} = \frac{3N\gamma}{V_m} R \left(TD \left(\frac{\theta_D}{T} \right) - T_0 D \left(\frac{\theta_D}{T_0} \right) \right), \quad (5)$$

222 where $D \left(\frac{\theta_D}{T} \right)$ is the Debye function (Debye 1912). The volume dependence of the Grüneisen
223 coefficient γ consistent with the QHA is given by:

$$\gamma = \gamma_0 \left(\frac{V}{V_0} \right)^q, \quad (6)$$

224 where q is the Anderson-Grüneisen parameter and γ_0 is the Grüneisen parameter at reference
225 conditions (Anderson 1968). Lastly, the Debye temperature (Θ_D) is expressed as:

$$\theta_D = \theta_{D0} \exp\left(\frac{\gamma_0 - \gamma}{q}\right), \quad (7)$$

226 where Θ_{D0} is the Debye temperature at reference conditions.

227 Because the thermal pressure from an MGD EoS involves the molar volume (e.g.,
228 Equation (3)), we kept this fixed at the literature value of 39.26 cm³/mol (Holland and Powell
229 2011) and we refined the dataset scale factors. A full refinement of a Mie-Grüneisen-Debye
230 thermal-pressure EoS with a third-order Birch-Murnaghan EoS was performed with full weights
231 and yielded $\chi_w^2 = 0.70$. Refined EoS parameters from this fitting are reported in Table 5. It is
232 important to note that compressibility parameters from the P - V - T EoS agree with the EoS
233 coefficients from P - V fitting within 1 σ (see Table 2). The scale factors (Table 5) for the two
234 volume datasets are close to unity and account for small differences in instrument calibrations
235 and laboratory “room conditions”.

236 The value of $K_{0S} = K_{0T}(1 + \alpha_0\gamma_0T_0)$ calculated from our thermal-pressure EoS is
237 225.0(1.2) GPa, which agrees well with independent measurements at room conditions (Özkan et
238 al. 1974; Özkan and Jamieson 1978). Note that the scale factor for the RUS data (Table 5) means
239 that our experimental values of K_S are consistently 3% higher than the calculated values from the
240 EoS (Figure 2). As discussed above, this may be attributed to the combination of lack of
241 constraints on the compressional moduli by the RUS data from such a small sample and the
242 effects of imperfections in the sample including misorientation errors of the sample faces, and
243 the damage to corners and edges.

244 The value of K_{0T}' has been poorly constrained within the literature, with reported values
245 ranging from 3.9 to 6.61 (Özkan and Jamieson 1978; Van Westrenen et al. 2004), while a re-
246 analysis of all literature data together yields $K_{0T}' = -0.56$ (Zaffiro 2019). The value of K_{0T}' from
247 our thermal-pressure EoS is 4.9(3), in the middle of this range and in good agreement with $K_{0T}' =$
248 4.71(4) from a recent series of DFT simulations (Stangarone et al. 2019). Additionally, the
249 dK_S/dT value from our thermal-pressure EoS at 300 K and ambient pressure is -0.0156 GPa/K,

250 which is in good agreement with the numerically-calculated values of -0.0152 and -0.0164
251 GPa/K at 300 K and ambient pressure (Özkan 2008).

252 The isochoric heat capacity of an MGD EoS for zircon is given directly at any pressure or
253 temperature for which we know the molar volume V_m , by Equation (4). The isobaric heat
254 capacity C_p follows from:

$$C_p = C_v + TV_m\alpha^2K_T. \quad (8)$$

255 The C_p values from our EoS as a function of temperature are in reasonable agreement with the
256 least-squares fitting of C_p derived from calorimetric zircon data (O'Neill 2006; Figure 4). This
257 curve fits our calculated data closely but not exactly, probably because higher-frequency
258 vibrational modes, including the Si-O stretching band, are not represented in the Debye model.

259 A C_p curve derived from DFT calculations of the calorimetric and electronic properties of
260 zircon (Terki et al. 2005) also shows good agreement with the data in Figure 4. Additionally,
261 Terki et al. (2005) calculated a Debye temperature $\Theta_D = 887$ K at 0 K using a quasi-harmonic
262 Debye model, which falls within 1σ of the calculated Debye temperature 849(38) K from our
263 thermal-pressure EoS at 0 K. This is also remarkably similar to the Debye temperature $\Theta_D = 870$
264 K extrapolated from a neutron-weighted phonon density of states map of a polycrystalline zircon
265 (Nipko and Loong 1997; Chaplot et al. 2006). The rate of change with temperature of the Debye
266 temperature of our MGD EoS is, however, about one-half that calculated by Terki et al. (2005),
267 which may be a consequence of the different methods used to calculate the Debye temperature
268 (as in McLellan 1980).

269 **Cell parameter equations for PT**

270 In EosFit, the parameters to describe the variation of the unit-cell parameters are obtained
271 by fitting the cubes of the unit-cell parameters and treating them as volumes (Angel et al. 2014a).
272 This yields linear moduli and thermal expansion coefficients that agree with independent
273 determinations. However, it is not clear how to modify this approach so as to be able to treat the
274 cell parameters as quantities equivalent to molar volumes that would be required to fit them with
275 a linearized MGD EoS, nor what the refined parameters such as Debye temperature or γ_0
276 physically represent. Therefore we use an “isothermal type” of EoS (Angel et al. 2018) for
277 describing the unit-cell parameter variation and, for internal consistency, we also report in Table

278 **6** the corresponding parameters for this kind of EoS to describe the volume. In the absence of a
279 physical model for this type of EoS, unlike the MGD, there are more parameters, some of which
280 such as γ_0 , q , and δ' are, either individually or collectively, not constrained by the data available.
281 The values of γ_0 and q have therefore been fixed to those refined for the MGD EoS, and δ' has
282 been given a value to reproduce the variation of K_{0T}' of the MGD EoS. Up to 5 GPa and 1200 K
283 this isothermal EoS gives volumes within 0.003%, bulk moduli within 0.1% and K_{0T}' within 0.02
284 of the values predicted by the refined MGD EoS.

285 In order to obtain parameters to describe the cell parameter variation in P and T (Table
286 **6**), we fixed the value of γ_{i0} for each axis to the value of γ_0 for the volume, multiplied by the ratio
287 $M_{i0}/3K_0$ (Milani et al. 2017), and the values of q , and δ' to those for the volume EoS. These
288 constraints are sufficient to allow refinement of the other EoS parameters to describe the a -axis
289 (Table **6**) but the c -axis is so stiff that the value of δ , which controls dK/dT , cannot be refined
290 and a value was chosen that reproduces the general trend of the data. These sets of EoS
291 parameters are internally consistent in that the more uncertain values for the properties of the c -
292 axis given by its own EoS, are in good agreement within 10^{-4} in strain and 0.5 GPa in modulus
293 with those calculated from the ratio V/a^2 over metamorphic ranges of P and T . The refined axial
294 moduli correspond to adiabatic values at room conditions of 573(3) and 1042(13) GPa,
295 respectively, for the a - and c -axes respectively, in reasonable agreement with the values from
296 ultrasonic wave velocity measurements of 580 and 1012 GPa (Özkan et al. 1974).

297

298

IMPLICATIONS

299 The combined fit to our new RUS and P - V data together with the data available in the
300 literature yields a MGD EoS (Table **5**) and isothermal-type EoS (Table **6**) that are in good
301 agreement with previous determinations of the elastic tensor of zircon (Özkan 2008) and its
302 variation with temperature (Özkan 2008). But all of these measurements are significantly stiffer
303 than the bulk modulus obtained from powder diffraction data by van Westrenen et al. (2004), for
304 reasons that cannot be determined from the available published information. In particular, the
305 new P - V data resolve the previous discrepancy between the wide range of values of K_{0T}' reported
306 in the literature, including a value of -0.56 from a global fit of the literature P - V data (Zaffiro

307 2019), and provide a value of K_{0T} that is in good agreement with the recent DFT simulation of
308 zircon (Stangarone et al. 2019). Our data therefore confirm experimentally that the displacive
309 transition to the high-pressure phase of zircon above 20 GPa (Stangarone et al. 2019; Mihailova
310 et al. 2019) is not accompanied by significant elastic softening, at least up to 8.5 GPa. The EoS
311 parameters are provided in .eos files that can be read by the EosFit suite of programs both as
312 supplementary material to this paper, and as files for free download from www.rossangel.net.

313 Pyrope garnets are common hosts for zircon inclusions and have EoS parameters $\alpha_H =$
314 $2.54 \times 10^{-5} \text{ K}^{-1}$ and $\beta_H = 1/K_{0T} = 0.0061 \text{ GPa}^{-1}$ at room conditions (Milani et al. 2017), with the
315 subscript 'H' indicating here the host mineral. These parameters are significantly larger than
316 those of zircon; thus for zircon inclusions in a garnet host $\alpha_H > \alpha_I$ and $\beta_H > \beta_I$. As a consequence,
317 the isomekes (Rosenfeld and Chase 1961), which define lines of equal fractional volume change
318 of the two phases, have steep positive slopes given by $\left(\frac{\partial P}{\partial T}\right)_{isomeke} = \frac{\alpha_I - \alpha_H}{\beta_I - \beta_H}$. Figure 5 shows
319 that the isomekes of zircon in garnet calculated with both the MGD and isothermal EoS for
320 zircon reported in this work are indistinguishable. The significance of the isomekes is that a
321 zircon trapped in a garnet at any point along a single isomeke will exhibit the same final
322 inclusion pressure, P_{inc} , measured when the garnet is at room conditions (e.g., Rosenfeld and
323 Chase 1961; Angel et al. 2014b, 2017). These P_{inc} values are indicated on the isomekes shown in
324 Figure 5. It is clear from the spacing of the isomekes that P_{inc} of zircon in garnet is more
325 sensitive to temperature rather than pressure, and thus zircon inclusions in garnets are better
326 piezothermometers than piezobarometers.

327 Normally, soft inclusions in stiffer hosts (such as quartz in garnet) yield positive
328 inclusion pressures P_{inc} at room conditions (e.g., Angel et al. 2014b), whereas stiff inclusions in
329 softer hosts such as zircon in garnet might be expected to have either negative or zero P_{inc} at
330 room conditions. However, Figure 5 shows that the considerable contrast between the thermal
331 expansion coefficients of zircon and garnet (α_I and α_H) that results in steep isomekes also places
332 room pressure and temperature conditions above the isomekes that run through metamorphic
333 conditions. As a consequence, room conditions lie in the region where P_{inc} is greater than the
334 external pressure (e.g., Ferrero and Angel 2018), resulting in positive residual pressures in zircon
335 inclusions trapped under metamorphic conditions.

336 The new EoS has a significantly lower thermal expansion coefficient and bulk modulus
337 than the Holland and Powell (2011) EoS for zircon, whereas the EoS that can be obtained by
338 fitting (Zaffiro 2019) previously published data also exhibits a high bulk modulus similar to that
339 of Holland and Powell (2011), but has a thermal expansion coefficient similar to the one
340 determined here. The smaller β_I determined by Zaffiro (2019) results in isomekes steeper than
341 those shown in Figure 5 while the larger α_I and smaller β_I from Holland and Powell (2011) give
342 isomekes with significantly shallower slopes. The consequence is that using the EoS from
343 Zaffiro (2019) for zircon leads to inferred entrapment pressures at 700°C that are roughly 0.3
344 GPa greater than those calculated with our new EoS, and those calculated with the EoS from
345 Holland and Powell (2011) can be up to 0.8 GPa lower.

346 In this study we have also introduced the refinement of scaling of datasets during the
347 fitting of EoS with the EoSFit program. This allows different datasets, whether of volume or bulk
348 moduli, to be used together without biasing the final results by scaling of the data prior to fitting.
349 In particular, this can accommodate the small differences in volumes frequently found between
350 datasets from diffraction data that arise from both the different calibrations of diffractometers
351 and uncharacterized differences in laboratory temperatures, frequently simply reported as “room
352 temperature”. We have shown that this rescaling can also accommodate the differences in
353 absolute values of the bulk moduli arising, for example in this study, from the necessity of using
354 a sample that was half the ideal size required for RUS measurements. Such scaling could also
355 accommodate the differences in bulk moduli of single crystal and polycrystalline specimens,
356 allowing data from both types of elasticity measurements to be fitted together in a self-consistent
357 manner. Lastly, we note that the reasonable agreement (Figure 4) between the heat capacity
358 obtained from our EoS and from measurements (O’Neill 2006) suggests the possibility of
359 refining EoS parameters not only to volume and bulk moduli data, but also simultaneously to
360 experimentally-determined C_p data.

361

362

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- 498

499

TABLES

500 Table 1: Unit-cell volume and axial parameters of the Mud-Tank zircon as a function of pressure
501 collected in this study.

<i>P</i> (GPa)	<i>a</i> (Å)	<i>c</i> (Å)	<i>V</i> (Å ³)
0.000100(1)	6.60633(11)	5.98228(20)	261.088(12)
0.186(10)	6.60431(09)	5.98114(17)	260.879(10)
0.861(09)	6.59647(12)	5.97703(20)	260.081(12)
1.746(12)	6.58654(14)	5.97216(23)	259.087(14)
2.351(10)	6.57984(20)	5.96947(34)	258.444(21)
3.188(12)	6.57111(15)	5.96386(25)	257.516(15)
3.481(08)	6.56815(12)	5.96236(24)	257.220(14)
4.309(08)	6.55982(12)	5.95687(23)	256.331(14)
4.710(14)	6.55561(13)	5.95498(28)	255.921(16)
5.176(09)	6.55088(11)	5.95262(21)	255.451(13)
6.205(09)	6.54085(11)	5.94629(21)	254.398(12)
6.465(09)	6.53828(12)	5.94497(24)	254.143(14)
6.673(08)	6.53635(17)	5.94386(33)	253.945(19)
6.956(10)	6.53367(13)	5.94209(24)	253.661(14)
7.174(09)	6.53141(12)	5.94092(23)	253.435(13)
7.706(08)	6.52640(17)	5.93752(33)	252.902(19)
7.952(12)	6.52411(13)	5.93655(25)	252.683(15)
8.465(11)	6.51936(13)	5.93353(25)	252.187(14)
8.294(10)*	6.52100(15)	5.93455(29)	252.357(17)
5.737(08)*	6.54513(17)	5.94955(32)	254.871(18)
2.832(10)*	6.57486(12)	5.96620(22)	257.912(13)

502 *Data collected during decompression.

503

504 Table 2: Equation of state volume and unit-cell parameters for the Mud-Tank zircon.

	EoS	V_0 (Å ³)/ L_0 (Å)	K_{0T}/M_{0T} (GPa)	$K_{0T'}/M_{0T'}$	$K_{0T''}/M_{0T''}$ (GPa ⁻¹)	χ_w^2
Volume	BM3	261.08(1)	224.9(1.2)	4.76(30)	-0.0233*	0.25
Volume	BM4	261.09(1)	222.8(2.8)	6.2(1.8)	-0.41(50)	0.23
<i>a</i> -axis	BM3	6.60632(10)	572.2(3.0)	16.80(0.78)	-0.127*	0.51
<i>c</i> -axis	BM3	5.98224(13)	1039(13)	-0.8(2.9)	-0.159*	1.04

505 *Value implied.

Table 3: Adiabatic elastic moduli of zircon.

T (K)	C₁₁	C₃₃	C₁₃	C₁₂	C₄₄	C₆₆	K_s	M_{aS}	M_{cS}	G_s
293(3)	431.4 (2.9)	500.4 (4.5)	160 (4.3)	75.3 (3.1)	113.46 (6)	48.92 (2)	233.3	594.5	1083.8	98.5
428(3)	424.47 (3.99)	489.88 (7.15)	154.48 (6.21)	72.77 (4.01)	112.57 (10)	48.78 (2)	228.0	584.0	1040.3	97.8
512(3)	421.93 (4.01)	490.66 (7.02)	157.02 (6.12)	74.63 (4.04)	111.89 (10)	48.56 (2)	228.7	582.5	1064.7	97.1
607(3)	415.24 (3.57)	482.96 (5.84)	153.5 (5.40)	74.77 (3.81)	110.93 (9)	48.65 (2)	225.1	575.3	1035.6	96.6
711(3)	413.99 (2.24)	484.97 (3.78)	157.15 (3.44)	76.46 (2.45)	109.79 (5)	48.3 (1)	226.6	574.9	1069.9	95.7
823(3)	409.02 (3.89)	479 (6.90)	155.23 (5.98)	75.83 (3.97)	108.89 (10)	48.03 (3)	223.9	568.5	1055.4	94.9
939(3)	404.64 (1.94)	477.84 (4.16)	156.87 (3.2)	75.44 (1.98)	107.72 (4)	47.84 (1)	222.9	561.4	1083.2	94.0
1056(3)	397.8 (4.10)	469.61 (7.37)	152.95 (6.35)	73.94 (4.19)	106.56 (10)	47.5 (3)	218.7	551.8	1053.7	93.1
1172(3)	392.82 (3.97)	463.89 (7.10)	152.08 (6.08)	74.09 (4.02)	105.42 (9)	47.2 (3)	216.6	546.3	1046.6	92.1

All values are adiabatic, in GPa. Uncertainties from fitting of the resonance frequencies are given in parentheses. Uncertainties in bulk, shear and linear moduli are estimated to be 1%.

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Table 4: Sources of data used in P - V - T EoS calculations.

Source	Data Type	P max (GPa)	T range (K)	Ndata
This study	Single-crystal XRD	8.47	Ambient	21
This study	RUS to determine K_s	Ambient	293-1172	9
Chaplot et al. (2002)	Powder XRD	Ambient	100-280	15
Mursic et al. (1992)	Neutron powder diffraction	Ambient	500-1200	8

509 Note: Ndata is the number of data points used from each source.

510 Table 5: P - V - T EoS parameters and scale factors using a 3rd-order Birch-Murnaghan and Mie-Grüneisen-Debye thermal-pressure EoS.
511

<i>Compressibility</i>				
EoS	V_0 (cm ³ /mol)	K_{0T} (GPa)	K_{0T}'	K_{0T}'' (GPa ⁻¹)
BM3	39.2600	224.5(1.2)	4.9(3)	-0.025*
<i>Thermal expansion</i>				
EoS	Θ_D (K)	Atoms/formula unit	γ_0	q
MGD	849(38)	6**	0.868(15)	2.37(80)
<i>Scale factors</i>				
	P - V data	T - V data	RUS data	
	1.00154(4)	0.99983(3)	1.030(8)	

512 *Value implied.

513 **Fixed value.

514

515 **Table 6:** Refined parameters for “isothermal-type” EoS to volume and cell parameters of zircon.

	V	a	c
K_{0T}/M_{i0T} (GPa)	224.4(1.2)	571(3)	1036(13)
K_{0T}'/M_{i0T}'	4.9(3)	17.1(8)	-0.1(2.0)
α_{V0}/α_{iV0} (K^{-1})	$1.02(2) \times 10^{-5}$	$0.26(1) \times 10^{-5}$	$0.49(1) \times 10^{-5}$
Θ_E (K)	642(25)	709(43)	566(30)
δ	6.5(8)	9.9(1.2)	3.8*
δ'	3*	3*	3*
γ_0	0.868*	0.736*	1.337*
q	2.37*	2.37*	2.37*

516 *Fixed value.

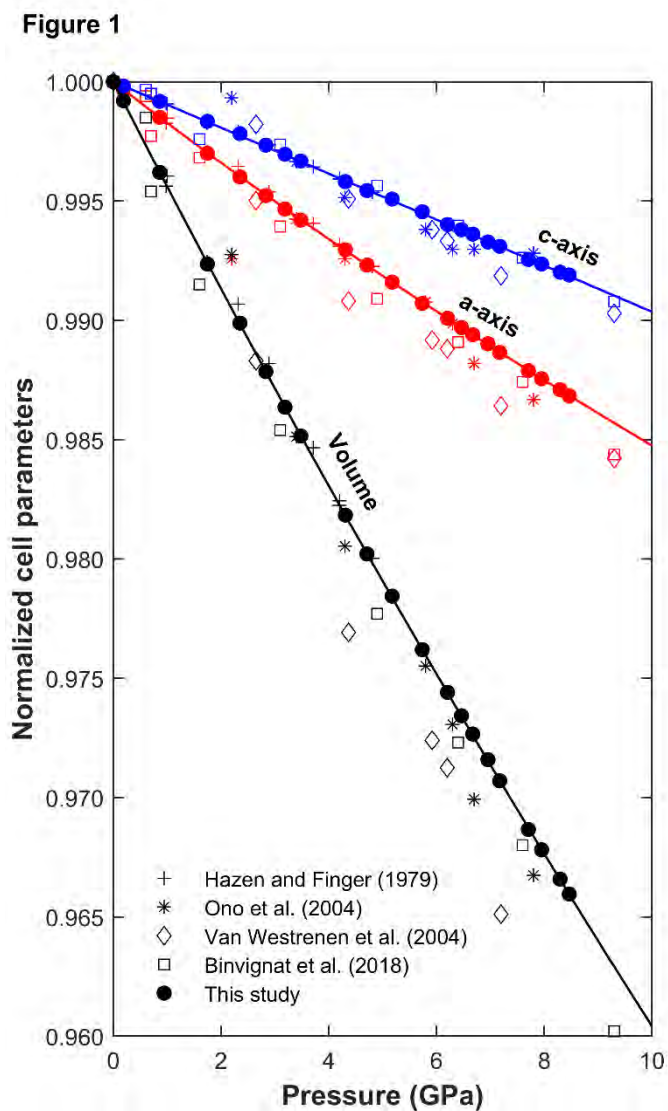
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FIGURES

Figure 1: Pressure variation of unit-cell volume (black), *a*-axis length (red), and *c*-axis length (blue). The line for *P-V* is the MGD thermal-pressure EoS determined in this study and the lines for *P-a* and *P-c* are the isothermal-type EoS normalized to 298 K. These EoS are indistinguishable from those obtained from BM3 EoS fits to the new *P* data alone.

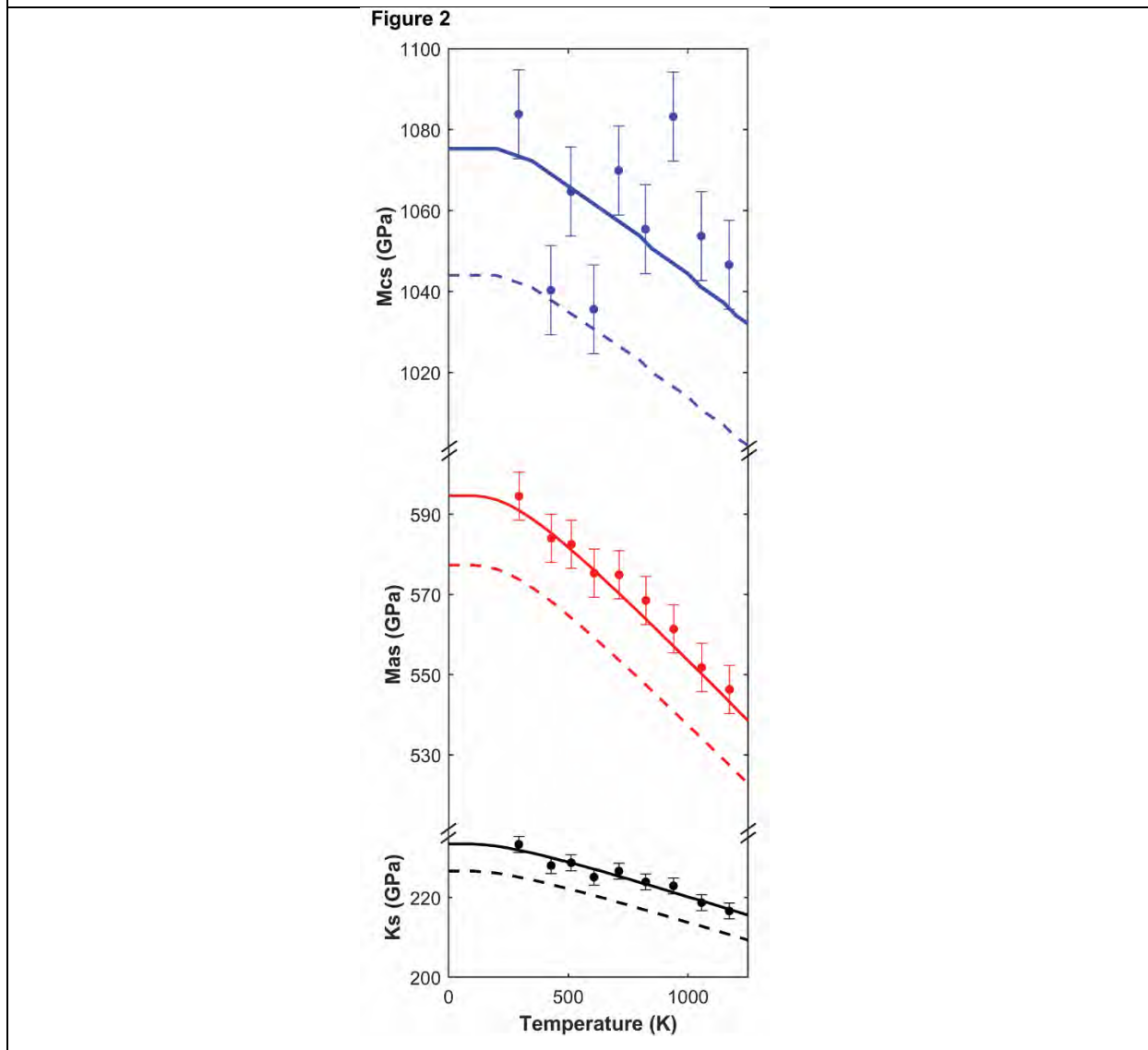


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Figure 2: Variation with temperature of experimental moduli data from the Mud-Tank zircon with the bulk modulus (lines) calculated from the MGD (Table 5) and linear moduli from isothermal-type EoS (Table 6). Dashed lines are moduli calculated without the scale factor and solid lines with the scale factor.

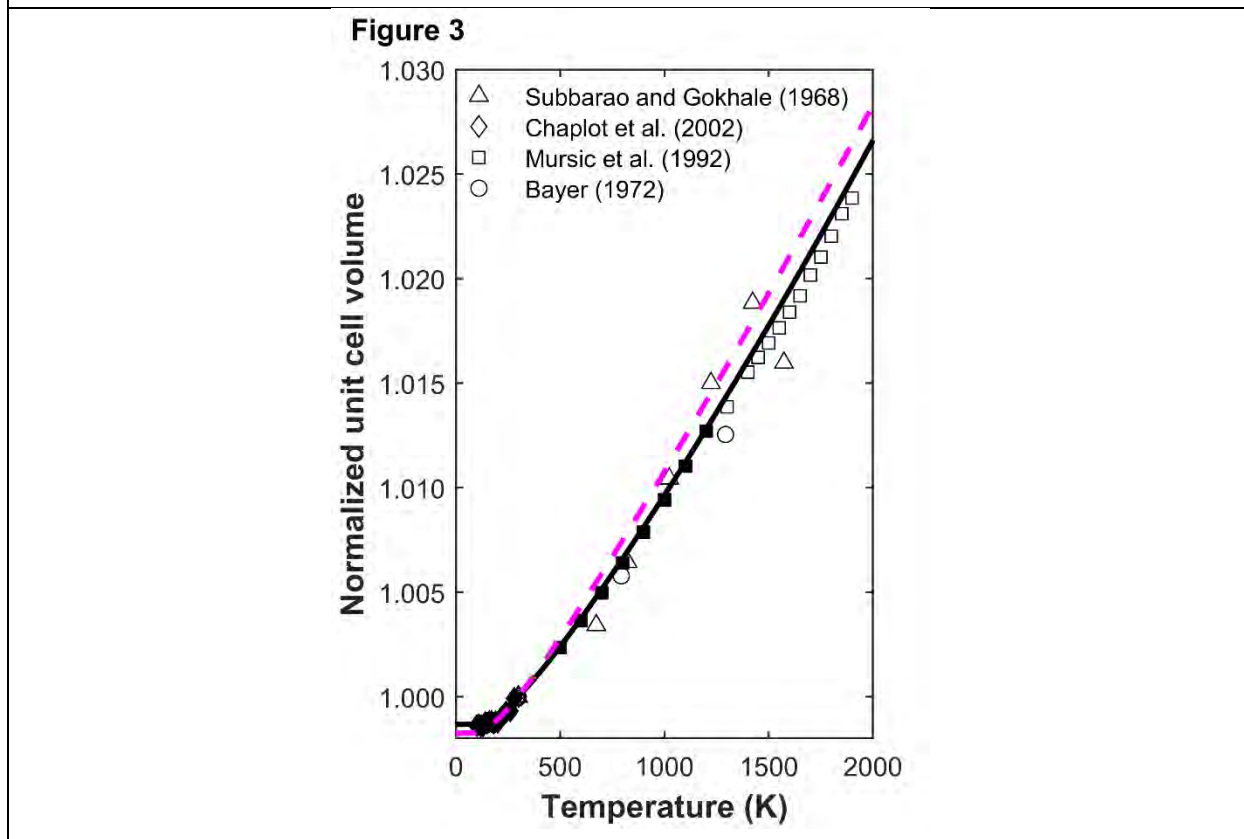


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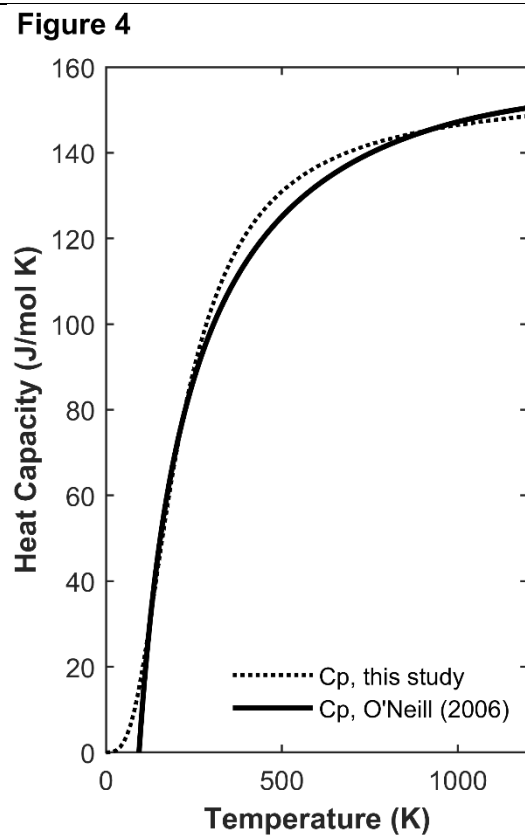
Figure 3: The refined MGD thermal pressure EoS from this study (solid line) with the thermal pressure EoS from Holland and Powell (2011) (dashed line) normalized to $V_0 = 39.26 \text{ cm}^3/\text{mol}$ at 298 K plotted with T - V data. Data points with solid symbols were used in the fit of the EoS, while data with open symbols were excluded from the fit.



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Figure 4: Variation with temperature of the isobaric heat capacity of zircon (C_p). Dotted line: calculated C_p values from the MGD thermal-pressure EoS from this study; solid line: weighted least-squares fitting of zircon calorimetric data (O'Neill 2006)



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Figure 5: Isomekes for zircon inclusions in pyrope for $P_{inc} = 0-0.9$ GPa. The solid black isomekes were calculated with the zircon MGD EoS (Table 5) and the blue dashed isomekes with the zircon BM3-isothermal EoS (Table 6). The parameters used for a BM3-MGD EoS for pyrope were $K_{0T} = 169.9$ GPa, $K_{0T}' = 4.4$, $\gamma_0 = 1.19$, $q = 0$, and $\Theta_D = 650$ K.

Figure 5

