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Revision 1

Crystal chemistry of schreibersite, (Fe,Ni)₃P

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Abstract

Schreibersite, $(\text{Fe,Ni})_3\text{P}$, the most abundant cosmic phosphide, is a principal carrier of phosphorus in the natural Fe–Ni–P system and a likely precursor for prebiotic organophosphorus compounds at the early stages of Earth’s evolution. The crystal structure of the mineral bears three metal sites allowing for unrestricted substitution of Fe for Ni. The distribution of these elements across the structure could serve as a tracer of crystallization conditions of schreibersite and its parent celestial bodies. However, discrimination between Fe ($Z = 26$) and Ni ($Z = 28$) based on the conventional X-ray structural analysis was for a long time hampered due to the proximity of their atomic scattering factors. We herein show that this problem has been overcome with the implementation of area detectors in the practice of X-ray diffraction. In this paper we report on previously unknown site-specific substitution trends in schreibersite structure. The composition of the studied mineral encompasses a Ni content ranging between 0.03 and 1.54 Ni atoms per formula unit (*apfu*): the entire Fe-dominant side of the join Fe_3P – Ni_3P . Of 23 schreibersite crystals studied, 22 comprise magmatic and non-magmatic iron meteorites and main group pallasites. The near end-member mineral (0.03 Ni *apfu*) comes from the pyrometamorphic rocks of the Hatrurim Basin, Negev desert, Israel. It was found that Fe/Ni substitution in schreibersite follows the same trends in all studied meteorites. The dependencies are non-linear and can be described by 2nd order polynomials. However, substitution across the *M2* and *M3* sites within the most common range of compositions ($0.6 < \text{Ni} < 1.5$ *apfu*) is well approximated by a linear regression: $\text{Ni}(M2) = 0.84 \times \text{Ni}(M3) - 0.30$ *apfu* (standard error 0.04 Ni *apfu*). The analysis of the obtained results shows a strong divergence between the behavior of unit-cell parameters of natural schreibersite and those of synthetic $(\text{Fe,Ni})_3\text{P}$. This indicates that Fe/Ni substitution trends in the mineral and its synthetic surrogates are different. A plausible explanation might be related to the differences in the system equilibration time of meteoritic schreibersite (millions of years) and synthetic $(\text{Fe,Ni})_3\text{P}$ (~100 days). However, regardless of the reason for the observed difference, synthetic $(\text{Fe,Ni})_3\text{P}$ can not be considered a

42 structural analogue of natural schreibersite, and this has to be taken into account when using
43 synthetic $(\text{Fe,Ni})_3\text{P}$ as an imitator of schreibersite in reconstructions of natural processes.

44

45 **Keywords:** phosphide; schreibersite; Fe_3P , Fe–Ni–P system; meteorite; planetary interiors; crystal
46 structure; solid solution

47

48

Introduction

49 Schreibersite, $(\text{Fe,Ni})_3\text{P}$ (Berzelius 1832) is the most common natural phosphide and a general
50 carrier for reduced phosphorus in meteoritic matter (Doan and Goldstein 1969). The mineral may
51 constitute for up to 14 % of metal matrix in iron and stony-iron meteorites (Buchwald 1975; Wasson
52 and Choe 2009). Being a dominant phosphide phase in the metal-rich part of the Fe–Ni–P(S)
53 system, schreibersite is a regular subject of studies devoted to evolution of planetary interiors (e.g.,
54 Clarke and Goldstein 1978; Scott et al. 2007; Goldstein et al. 2009; Gu et al. 2014, 2016; He et al.
55 2019; Khisina et al. 2019; Chabot et al. 2020; Chornkrathok et al. 2020). The mineral is a typical
56 constituent of enstatite chondrites and achondrites (Wasson and Wai 1970); it was encountered in
57 carbonaceous chondrites (Zolensky et al. 2002), acapulcoites, lodranites and winonaites (Li et al.
58 2011; Keil and McCoy 2018) and in Lunar rocks (Smith and Steele 1976; Gleißner and Becker
59 2017). Thermodynamic calculations evidence that schreibersite could be a primary phosphorus-
60 bearing phase formed during solar nebula condensation (Pasek 2008). The mineral is considered a
61 likely source of low-valent phosphorus which was required for initiation of prebiotic
62 phosphorylation processes on early Earth (e.g., Bryant et al. 2013; Pasek 2020).

63 Investigation of synthetic compounds has showed that phosphides Fe_3P and Ni_3P are
64 isostructural and form a continuous series of solid solutions (Rundqvist 1962; Rundqvist et al. 1962;
65 Spriggs 1970). The composition of rarely occurring terrestrial schreibersite approaches pure Fe_3P
66 (e.g., Britvin et al. 2017b). On the contrary, meteoritic mineral is always enriched in Ni, forming a
67 tie-line towards the Ni-dominant counterpart, nickelphosphide $(\text{Ni,Fe})_3\text{P}$ (Britvin et al. 1999). In
68 spite of a simple composition, schreibersite has a complex crystal structure which contains one 9-
69 coordinated phosphorus site and three cage-like metal-centered polyhedra: $M1$ (coordination number
70 15), $M2$ (14) and $M3$ (13) (Figure 1). The polyhedra, which have no analogues among other Fe-Ni
71 phosphides (Britvin et al. 2020a), are arranged into the three-dimensional framework. The reliable
72 determination of Fe/Ni distribution across the metal sites was the main challenge of schreibersite

73 crystal chemistry. Doenitz (1970) has demonstrated on the mineral from the North Chile (Tocopilla)
74 meteorite that Ni in schreibersite preferably occupies $M3$ and $M2$ structural sites in the ratio of $\sim 2/1$,
75 avoiding $M1$ site. This finding might have promising implications, as distribution of Fe and Ni
76 across the different sites of schreibersite structure could be related to crystallization pathways of
77 metal-rich meteorites and their parent bodies. A serious obstacle towards exploring this hypothesis
78 was a proximity of X-ray scattering factors of Fe ($Z = 26$) and Ni ($Z = 28$), that makes quantitative
79 X-ray refinement of Fe/Ni populations a difficult task (e.g., Hawthorne 1983; Warner et al. 1995).
80 Because of that, Doenitz (1970) and the successors (Moretzki et al. 2005) have used various
81 techniques based on the effect of anomalous scattering of X-rays at the Fe- K absorption edge.
82 However, the experimental limitations allowed only semi-quantitative estimation of Fe/Ni site
83 populations and restricted the number of studied crystals to a few samples.

84 In the course of an ongoing research of natural phosphides, we have found that Fe/Ni
85 population across the metal sites of schreibersite structure can be determined with high precision
86 using a conventional X-ray structural analysis, due to implementation of high-redundancy and high-
87 intensity datasets obtained using area-detector diffractometers. The results reported herein represent
88 the crystal chemistry of the entire Fe-dominant side of natural solid solutions $\text{Fe}_3\text{P}-\text{Ni}_3\text{P}$.

89

90

Materials and Methods

91 Schreibersite

92 Meteorite samples (Table 1) were kindly provided by the Mining Museum, Saint Petersburg
93 Mining University, except for a slice of the Mont Dieu octahedrite which was courteously donated
94 by Dr. Alain Carion (carionmineraux.com). Schreibersite grains were either hand-picked from
95 polished sections using a tungsten carbide needle or extracted from meteorite chips by the
96 dissolution of metal matrix in a diluted HCl.

97

98 **Electron microprobe analysis (EMPA)**

99 EMPA was performed from polished sections coated with a carbon film. Fe, Ni and P
100 contents were determined by means of a Hitachi S-3400N SEM equipped with an Oxford
101 Instruments AzTec Energy X-Max 20 EDX spectrometer (20 kV, 1 nA). Cobalt content was
102 determined with an INCA WAVE 500 WDX spectrometer (20 kV, 15 nA). Schreibersite of known
103 composition and Co metal were used as analytical standards. Neither silicon nor sulfur was detected
104 in the mineral. No compositional zoning was observed. The complete EMPA results are provided in
105 Supplemental Tables S1 and S2.

106

107 **X-ray single-crystal study**

108 The datasets were collected by means of two instruments, Bruker Smart APEXII CCD
109 (conventional sealed tube) and Bruker Kappa APEX DUO (kappa-geometry, microfocus mirror
110 optics). Both devices were equipped with the identical 1024K APEXII CCD detectors and were
111 operated using MoK α -radiation. The choice of the instrument was determined by the crystal size and
112 the time available for current measurement. For each crystal, a full Ewald sphere up to $2\Theta = 64^\circ$
113 ($\sin\Theta/\lambda = 0.746 \text{ \AA}^{-1}$) was scanned with a frame width of 1° . The collection times were adjusted to
114 attain the average $I/\sigma(I)$ ratio not less than 30. Data processing and integration included standard
115 routines incorporated into a Bruker APEX2 software. In order to avoid the problem of low accuracy
116 of unit-cell measurements with area detectors (Duisenberg et al. 2000), we used the unit-cell
117 parameters refined from the Rietveld data (see below). The absorption correction was performed
118 with a SADABS multi-scan procedure (Sheldrick 2015a). The HKL files were generated unmerged
119 in HKLF4 format. Crystal structures were solved by an intrinsic phasing method as implemented in
120 *SHELXT*-2018, using Olex2 operational environment (Sheldrick 2015b; Dolomanov et al. 2009).
121 Because schreibersite crystallizes in a non-centrosymmetric space group $\bar{I}4$ (# 82), atomic
122 coordinates (but not HKL source files) were transformed when necessary to either standard (Doenitz

123 1970) or inverse setting. Structure refinements were carried out using full-matrix least-square
124 approach with a *SHELXL*-2018 software (Sheldrick 2015a). All sites in the crystal structure were
125 treated as fully occupied, based on the results of EMPA and previous reports on schreibersite
126 stoichiometry. Metal sites were first checked for mixed Fe/Ni population using standard *SHELXL*
127 procedures. In case when population of a site by a minor substituent (either Fe or Ni) exceeded 3σ
128 (*apfu*), the site was refined as having mixed Fe/Ni occupancy. Otherwise, site population was fixed
129 to 1 (Fe or Ni, respectively). A brief summary of data collection and refinement parameters prepared
130 using a *publCIF* software (Westrip 2010) is given in Supplemental Table S3. Full details can be
131 retrieved from crystallographic information file (CIF) included in Supplemental Materials.

132

133 **Rietveld refinements.**

134 Powder X-ray diffraction patterns were recorded from the same crystal fragments which
135 were used for the single-crystal study. A Rigaku RAXIS Rapid II diffractometer (semi-cylindrical
136 imaging plate, Debye-Scherrer geometry, $r = 127.4$ mm) was equipped with a rotating anode $\text{CoK}\alpha$ -
137 radiation source (40 kV, 15 mA) and microfocus X-ray optics. The imaging plate was calibrated
138 with a NIST silicon standard. Powdered schreibersite was mixed with an epoxy resin into the balls
139 of 0.15-0.2 mm in diameter. The balls were pinned onto the glass fibers and exposed to X-rays for
140 30 min with a simultaneous ϕ -axis rotation. The imaging plate-to-profile conversion was performed
141 by use of an *osc2xrd* program (Britvin et al. 2017a). Full-profile Rietveld refinements were carried
142 out using a Bruker TOPAS v.5.0 software (Coelho 2018). The two-stage refinement procedure was
143 employed. At first, the unit-cell parameters alone were refined by Rietveld method using the input
144 data (including site occupancies) taken from the initial single-crystal refinements. These refined
145 unit-cell parameters (Table 2) were used in final refinements of all single-crystal datasets. At the
146 second stage, the powder diffraction patterns were re-refined by Rietveld method based on the input
147 atomic coordinates, site occupancies and displacement parameters acquired from final single-crystal

148 data. During this, last Rietveld refinement stage, all parameters were freely refined, except for the
149 displacement values of the metal sites which were treated as being linked via the same B_{eq} (based on
150 the nearly the same U_{iso} obtained in the single-crystal datasets). The only sample where we did not
151 refine atomic coordinates was the Fe_3P end-member from Hatrurim (Htr). The full-profile
152 refinement allowed to achieve a precision of 5×10^{-4} Å for the a parameter and 3×10^{-4} Å for the c
153 parameter, as it was confirmed by comparison of imaging plate data with the Si-calibrated test
154 profile of GM sample obtained using a Bruker Phaser II diffractometer (Bragg-Brentano geometry).
155 Data collection and Rietveld refinement details are summarized in Supplemental Table S4. The
156 snapshots of refinement plots can be viewed in Supplemental Figures S4-S26. Further details of the
157 Rietveld refinements, including observed, calculated and background profile listings, can be
158 retrieved from the powder CIF files included into Supplemental Materials.

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Results

161 Occurrence and chemical composition

162 Of 23 schreibersite crystals subjected to this study (Table 1), 22 originate from 13 iron
163 meteorites belonging to magmatic and non-magmatic groups (e.g., Goldstein et al. 2009) and 6
164 main-group pallasites (e.g., Boesenberg et al. 2012). The terrestrial mineral (Figure 2) comes from
165 the pyrometamorphic rocks of the Halamish Wadi, Hatrurim Basin, Israel (e.g., Britvin et al. 2015).
166 Meteoritic schreibersite is represented by 0.1–0.5 mm single-crystal fragments extracted from the α -
167 (Fe,Ni) (kamacite) matrix, except for the sample Crn2 (Cranbourne) comprised of so-called
168 “rhabdite” (schreibersite variety having needle-like habit). The Ni-depleted schreibersite from
169 Zacatecas (1792) (Table 1) represents the rim surrounding the troilite nodule. The grain morphology
170 of the mineral from the listed meteorites (except for Mont Dieu) was described by Buchwald (1975)
171 and Buseck (1977). Terrestrial schreibersite is of interest due to its association: the mineral is being

172 replaced by the polycrystalline rims of barringerite, Fe₂P (Figure 2), exactly in the same manner as it
173 has been reported in meteoritic assemblages (Buseck 1969; Britvin et al. 2020b).

174 Nickel contents in the studied mineral cover a range from 0.03 to 1.54 *apfu* (Table 1) and
175 thus encompass the entire schreibersite half (Fe > Ni) of the join Fe₃P–Ni₃P. All analyses
176 correspond to a stoichiometry of M₃P (Supplemental Table S2), indicating for the absence of
177 partially vacant sites in the crystal structure, in agreement with the previously reported data. One of
178 the aims of the present work was an investigation of Fe for Ni substitution that required X-ray
179 refinement of site populations. The precision of conducted X-ray refinements can be estimated via
180 comparison of total Ni contents determined by electron microprobe and those derived from the X-
181 ray structural analysis (Table 1, Figure 3). It can be seen that the sums of Ni site populations
182 obtained from single-crystal and Rietveld refinements quantitatively match the results of EMPA
183 within the standard errors of 0.04 and 0.01 Ni *apfu*, respectively (Figure 3).

184

185 **Absolute structure**

186 Schreibersite crystallizes in a non-centrosymmetric space group $\bar{I}4$ that implies the existence
187 of two spatial atomic arrangements called *absolute structures*. These arrangements, being related by
188 an inversion operator, are designated as a *standard* absolute structure and an *inverse* one. The
189 assignment of a crystal to a certain absolute structure can be done based on the differences in the
190 intensities of Friedel reflection pairs, using a measure known as an absolute structure (Flack *x*)
191 parameter (Flack and Bernardinelli 1999). The latter can vary between 0 (standard setting) and 1
192 (inverse one); the intermediate values indicate the occurrence of inversion (racemic) twinning. The
193 possibility of absolute structure determination for any given pair of X-ray wavelength/crystal
194 composition can be evaluated using a *Friedif* coefficient (Flack and Shmueli 2007). The chemical
195 composition of schreibersite (*Friedif* > 200) allows convincing determination of absolute structure
196 using MoK α -radiation. Of 23 schreibersite crystals studied herein, 13 were found to adopt standard

197 setting relative to that of Doenitz (1970), 7 have inverse absolute structure, and 3 represent inversion
198 twins (Table 2). These results show significant predominance of the crystals having standard
199 absolute structure, in contrast to the results by Skála and Císařová (2005) who reported the
200 prevalence of inverse setting (6) among 8 schreibersite crystals from 5 iron meteorites. The overall
201 count of schreibersite having the confirmed standard setting (including the crystal reported by
202 Doenitz (1970) is 16 crystals. 13 crystals have an inverse absolute structure, and in 3 cases the
203 mineral exhibits significant degree of racemic twinning. Therefore, based on the known data, one
204 can not postulate the predominance of either absolute structure among natural schreibersite.

205

206 **Unit-cell parameters of schreibersite and synthetic analogues**

207 The unit-cell parameters of studied crystals are listed in Table 2 and plotted in Figure 4.
208 Several authors reported lattice parameters of natural schreibersite (e.g., Skála and Císařová 2005;
209 Geist et al. 2005; He et al. 2019 and the references cited therein). However, all published results
210 were obtained with the use of single-crystal diffractometers whose precision corresponds to the 3rd
211 decimal place, whereas the 4th digit reflects the tightness of least-squares convergence. Besides, not
212 all papers provide electron microprobe data for the mineral. This has led to disparate results well
213 illustrated by the diffuse character of a/c ratio scatter plots (e.g., Geist et al. 2005). The
214 discrepancies between chemical composition and unit-cell metrics of schreibersite were also
215 discussed by Skála and Císařová (2005). Because of apparent heterogeneity of the published single-
216 crystal data, we herein compare only those reference parameters which were refined by the whole-
217 pattern profile fitting and thus have a precision of the 4rd decimal place. All these results belong to
218 the synthetic solid solutions $\text{Fe}_3\text{P-Ni}_3\text{P}$ (Skála and Drábek 2000; 2003).

219 The a parameter of natural schreibersite and its synthetic analogues follows the same trend
220 (Figure 4a). However, consolidation of the present results with the data by Skála and Drábek (2000;
221 2003) demonstrates that this dependence is not linear as it was suggested previously but obeys very

222 flat 2nd order polynomial regression. The latter gives an intercept of 9.1084(12) Å, coincident with
223 the a parameter of 9.1085(3) Å experimentally determined for pure Fe₃P (Skála and Drábek 2000).
224 The dependence of unit-cell volume (Figure 4b) also complies with the 2nd order polynomial as it
225 was noticed by Spriggs (1970) and discussed by Skála and Drábek (2000). The intercept of the
226 volume/composition fit, 370.3(2) Å³, coincides with the experimental value of 370.22(2) Å³ for Fe₃P
227 (Skála and Drábek 2000). In contrast to the a parameter and cell volume, the behaviour of the c
228 parameter of natural schreibersite is completely distinct from that of its synthetic analogues (Figure
229 4c). The latter were shown to obey the 2nd order polynomial regression (Spriggs 1970; Skála and
230 Drábek 2000). On the contrary, our data clearly demonstrate that the c parameter of natural mineral
231 follows quasi-linear dependence up to the Ni-richest composition (1.54 Ni *apfu*). The observed
232 divergence is emphasized further by plotting the c/a parameter ratios (Figure 4d). These values are
233 nearly the same for the Fe₃P and Ni₃P end-members, and because of that, c/a ratio is a very sensitive
234 indicator of any cell distortions, though requires high-precision data. The consequences rising from
235 the observed differences are discussed below.

236

237 **Site populations**

238 A quantitative match between Ni contents determined by EMPA and the values obtained
239 from the X-ray single-crystal data evidences that the uncertainties in the values of Ni population
240 (Table 3) represent the real precision of the refinements. The precision of the Rietveld data in Table
241 3 can be overestimated, as it is typical for the Rietveld refinement results. Nevertheless, cross-
242 plotting of site populations determined by the two methods (Figure 5) ensures the data consistency.

243 Single-crystal refinements of the $M1$ site did not reveal Ni occupancy factors exceeding 3σ
244 *apfu*, confirming the previous observations that Ni avoids incorporation in this position (e.g.,
245 Doenitz 1970). The formerly obscured substitution trends for the $M2$ and $M3$ sites are illustrated by
246 Figure 6a. Both sites exhibit non-linear dependencies of Ni incorporation upon the linear increase of

247 total Ni content. The *M3* trend demonstrates significant offset from the theoretical tie-line Fe₃P–
248 Ni₃P. In spite of the general nonlinearity, Fe for Ni substitution across the wide range of
249 compositions, from 0.6 to 1.5 total Ni *apfu*, can be well approximated by two virtually parallel linear
250 trends separated by statistically significant difference (Figure 6b). The overall interdependence can
251 be expressed by linear equation:

$$252 \quad \text{Ni}(M2) = 0.84 \times \text{Ni}(M3) - 0.30 \text{ apfu} [R^2 = 0.91, \text{ standard error } 0.04 \text{ Ni apfu}] \quad (1)$$

253 The equality of both regression slopes to the value of 1/2 (Figure 6b) evidences that incorporation of
254 Ni across this range proceeds through a highly ordered pathway, most probably – through a coupled
255 substitution of 2Fe for 2Ni, according to a scheme: Fe(*M2*) + Fe(*M3*) → Ni(*M2*) + Ni(*M3*).

256 The important consequence raised from equation (1) is that the *M3* site in natural
257 schreibersite attains full Ni saturation (Ni₃ = 1 *apfu*) exactly at the borderline with nickelphosphide
258 (total Ni = 1.5 *apfu*) (Figure 6a). The significant departures of both substitution trends from the
259 straight tie-line Fe₃P–Ni₃P (Figure 6a) explain the observed violations of lattice parameters from
260 Vegard's law (e.g., Zen 1956) (*cf.* Figure 4). The dependencies of *c/a* parameter ratio on Ni site
261 population are illustrated in Figure 7. The influence of Ni population at the *M3* site can be described
262 by a linear regression, whereas substitution at the *M2* site follows 2nd order polynomial.
263 Consequently, the quasi-linear dependence of *c/a* ratio on the chemical composition (Figure 4d)
264 represent superposition of two regressions plotted in Figure 7, with predominant influence of
265 substitution at the *M3* site.

266

267 **Interatomic bonds**

268 The complex shape of coordination polyhedra in schreibersite (Figure 1) determines the large
269 number of interatomic interactions, whereas the metallic type of bonding (La Cruz et al. 2016) and
270 high metal/phosphorus ratio implies the occurrence of both intermetallic and metal-phosphorus
271 interactions. Of 26 symmetrically independent bonds in the schreibersite structure, 9 comprise the

272 *M*–*P* linkages within the phosphorus-centered polyhedron [*PM*₉] (Figure 1a). The remaining 17
273 metal-metal bonds maintain the integrity of non-convex metal-centered polyhedra (Figure 1b-d).
274 The lengths of *M*–*P* bonds spread within a narrow range of values, between 2.23–2.35 Å. In
275 contrast, the lengths of intermetallic bonds vary considerably from 2.41 (*M*₁–*M*₁) to 3.02 Å (*M*₂–
276 *M*₂). The high precision of conducted experiments and ability to discriminate between Fe and Ni
277 allowed plotting the site-specific dependencies for every bond (Supplementary Figure S1–S3).
278 However, the revealed trends were found to be rather complex and have no simple relations to
279 crystallographic directions in the structure. Therefore, we herein restrict the comments to a few of
280 the most representative examples. The intermetallic linkage *M*₂–*M*₂^{ix} is being most sensitive to Ni
281 incorporation, both in the *M*₂ and *M*₃ site (Figure 8a). Although both sites show steep negative
282 trends of bond length upon increase of Ni content, these dependencies are described by two
283 completely different curves. Another intermetallic bond, *M*₃–*M*₃^{vii}, exhibits the maximal steepness
284 of bond-length ramp with an increasing Ni site population (Figure 8b). However, contrary to the first
285 example, the variations in bond length are very flat and thus much less sensitive to changes in the
286 chemical composition. The “knee-fold” shapes of *M*₁–*M*₃ⁱⁱ trends (Figure 8c) indicate for some yet
287 obscured interatomic interactions which abruptly turn the observed substitution dependencies. The
288 *M*₂–*P* bond is quite interesting because of almost complete insensitivity to Fe for Ni substitution
289 (Figure 8d). The overall change in the *M*₂–*P* bond length is 0.004 Å. The indifference of *M*₂–*P* bond
290 to compositional changes might make it useful in practice, e.g. in revealing the obvious outliers in
291 the bond length which would worth of additional investigation.

292

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Discussion

294

Refinement of similar X-ray scatterers: a Fe/Ni case

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In spite of a progress in X-ray diffraction techniques, the proper determination of similar X-
296 ray scatterers in solid solutions remains a challenge of mineralogical crystallography (Hawthorne

1983; Angel and Nestola 2016). The case of Fe ($Z = 26$) and Ni ($Z = 28$) is an example of the
element pair whose difference in (non-anomalous) scattering power is considered insufficient for the
site population refinement (e.g., Warner et al. 1995; Moretzki et al. 2005). We herein provided the
statistically proven evidence that X-ray single-crystal refinement using $\text{MoK}\alpha$ -radiation can be
regarded a quantitative method for discrimination between similar scatterers such as Fe and Ni
(Figure 3, 5-8). This result might look somewhat unexpected, in view of not using the effect of
anomalous dispersion at the Fe- K absorption edge (e.g., Doenitz et al. 1970; Moretzki et al. 2005).
However, a calculation of the difference between scattering factors of Fe and Ni gives a value of
7.3% for $\text{MoK}\alpha$ -radiation (Supplementary Table S5). This value is twice smaller than the difference
attainable using anomalous scattering with $\text{CoK}\alpha$ (15.4%), but it is not small. As for comparison, it
was shown that the good quality X-ray data allow determination of very subtle (down to 0.4 %) differences between the intensities of Friedel opposites (e.g., Flack and Shmueli 2007). The X-ray-based refinement of similar scatterers has become possible with the implementation of area detectors in practice of X-ray diffraction. The problems of redundancy, intensity, resolution, treatment of absorption and extinction were the factors critically limiting the reliability of site occupancy analysis with the data collected using point-counting detectors (e.g., Kroll et al. 1997). All these problems are nowadays diminished to a degree of routine procedures due to usage of area detectors. The detailed discussion of experimental conditions required for reliable discrimination between similar X-ray scatterers lies beyond the scope of the present paper. However, the main conditions can be formulated as: (1) High-redundancy datasets (full Ewald sphere is preferable), (2) High-brightness data (average $I/\sigma(I)$ ratio between 30 and 100), (3) Data-to-parameter ratio not less than 8 (a standard CheckCIF requirement).

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322 **Fe/Ni substitution trends: natural schreibersite vs synthetic (Fe,Ni)₃P**

323 The important consequence arising from the present work is that the trends in ordered Fe/Ni
324 substitution are the same for schreibersite originated from different meteorite groups, i.e., from
325 magmatic and non-magmatic irons and pallasites. This evidences for the similarity of schreibersite
326 crystallization conditions, most probable – for the time required for intersite equilibration between
327 Fe and Ni.

328 The precise refinements of unit-cell parameters and site populations have showed that the *c*
329 parameter and *c/a* ratio of schreibersite are sensitive indicators of site-specific ordering of Fe and Ni
330 (Figure 4cd, 7). Although both natural mineral and its synthetic counterparts exhibit substantial
331 violations from Vegard's law, there is an apparent difference in their parameters' behaviour. The
332 abrupt divergence in the *c/a* ratio trends (Figure 4d) evidences that Fe for Ni substitution in natural
333 schreibersite proceeds through a pathway different from that of synthetic (Fe,Ni)₃P. Therefore, one
334 can conclude that natural schreibersite is not structurally identical to synthetic (Fe,Ni)₃P having the
335 same bulk composition. The latter does not depend on the route used for the synthesis of (Fe,Ni)₃P:
336 Spriggs (1970) investigated the fused Fe₃P–Ni₃P samples whereas Skála and Drábek (2000; 2003)
337 studied the sintered ones, and both authors demonstrate the similar trends in variations of unit-cell
338 parameters. Unfortunately, the absence of site-specific Fe/Ni distribution data for synthetic
339 (Fe,Ni)₃P precludes further comparison with the results reported in the present work. However, one
340 can speculate that the observed differences may relate to the Fe/Ni ordering schemes, which in turn
341 can be dependent on different system equilibration times of meteoritic schreibersite (millions of
342 years, e.g., Goldstein et al. 2009) and synthetic (Fe,Ni)₃P (up to 110 days, Skála and Drábek 2000).
343 The discrepancy between meteoritic schreibersite and synthetic (Fe,Ni)₃P may be important in view
344 of a wide usage of schreibersite surrogates as simulants of natural mineral (Pasek et al. 2007; Pech
345 et al. 2011; Pirim et al. 2014; La Cruz et al. 2016; Parker et al. 2016). It was shown previously that
346 the crystal structures of phosphides substantially affect their surface chemistry and hence – the

347 pathways of surface reactions (e.g., Wexler et al. 2017; Li et al. 2019). The latter should not be
348 ignored in the experiments aimed at mimicking the natural processes, such as biomimetic
349 phosphorylation reactions. Besides, structural differences may considerably affect the behaviour of
350 studied phases in the high-pressure/high-temperature experiments imitating the conditions
351 characteristic of planetary interiors.

352

353

Implications

354 The results disclosed in the present work might have both fundamental and applied consequences.
355 The knowledge of structural and compositional trends in the dominant cosmic phosphide may serve
356 a basement for the further studies aimed at understanding crystallization histories of metal-rich
357 interiors of celestial bodies. The obvious structural differences revealed between natural
358 schreibersite and synthetic (Fe,Ni)₃P may warn for the caution that should be undertaken while
359 working with the synthetic schreibersite surrogates. The important applied result of this study is that
360 we have showed the possibility of reliable X-ray refinements of structural sites occupied by similar
361 X-ray scatterers, such as Fe and Ni.

362

363

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533

534 **List of figure captions**

535 **Figure 1.** Coordination polyhedra in the crystal structure of schreibersite. (a) Phosphorus-centered
536 polyhedron [PM₉]. (b) Coordination cage of M1 site (CN 15). (c) Coordination cage of M2 (CN 14).
537 (d) Coordination cage of M3 (CN 13). Legend: yellow, P; brown, M1; blue, M2; green, M3. Thermal
538 displacement ellipsoids are drawn at 50% probability level. Created based on the data for
539 schreibersite from the Bischtübe octahedrite, using CCDC Mercury software (Macrae et al. 2006).

540

541 **Figure 2.** Terrestrial schreibersite. (a) Single-crystal grain of schreibersite surrounded by the thick
542 rims of polycrystalline barringerite, (Fe_{1.98}Ni_{0.02})_{2.00}P. Phosphides are embedded into the matrix of
543 hydrous Ca-Mg silicates. Reflected light. (b) The same grain, SEM BSE image. The worm-like
544 channels in schreibersite are filled with an unknown Na-Fe phosphate. Halamish Wadi, Hatrurim
545 Basin, Negev desert, Israel. Schr, schreibersite; Brr, barringerite; Tr, troilite.

546

547 **Figure 3.** Plots of Ni contents as determined by EMPA vs. Ni contents obtained by summation of X-
548 ray refined site populations (Table 1). The linear fits are marked by red lines. (a) Single-crystal data:
549 $R^2 = 0.97$; intercept $-0.00(4)$ Ni *apfu*; slope 1.00(4). (b) Rietveld refinement results: $R^2 = 0.98$;
550 intercept of $-0.03(1)$ Ni *apfu*; slope 1.00(3). The dashed lines trace the theoretical tie-line Fe₃P–
551 Ni₃P.

552

553 **Figure 4.** Dependencies of unit-cell parameters of natural schreibersite and synthetic (Fe,Ni)₃P upon
554 total Ni content. (a) The *a* parameter follows a flat 2nd order polynomial curve ($R^2 = 0.995$). (b)
555 Unit-cell volume can also be fitted with a 2nd order polynomial ($R^2 = 0.993$). (c) Divergence in
556 behaviour of the *c* parameter of natural schreibersite and synthetic (Fe,Ni)₃P. (d) Divergence in the
557 *c/a* parameter ratio between natural schreibersite and synthetic (Fe,Ni)₃P. The errors (esd's) are

558 smaller than size of the symbols. Data for natural schreibersite (red circles, this work) were taken
559 from Table 2. Data for synthetic $(\text{Fe,Ni})_3\text{P}$: Skála and Drábek (2000, 2003).

560

561 **Figure 5.** The site-specific match between Ni populations determined by single-crystal and Rietveld
562 refinements. Linear fit for the $M2$ site: $R^2 = 0.95$; intercept $0.002(7)$ Ni *apfu*; slope $1.06(5)$. Linear
563 fit for the $M3$ site: $R^2 = 0.99$, intercept $0.001(4)$ Ni *apfu*; slope $0.93(2)$.

564

565 **Figure 6.** Plots of Ni site population versus total Ni content in natural schreibersite (Table 3). (a)
566 2nd order polynomial fits for the entire range of compositions. (b) Linear regressions for $0.58 \leq$
567 $[\text{total Ni}] \leq 1.54$ *apfu*. Single-crystal refinement results. The dashed lines trace a theoretical join
568 $\text{Fe}_3\text{P}-\text{Ni}_3\text{P}$.

569

570 **Figure 7.** Dependencies of c/a ratio of natural schreibersite on Ni site population (Table 3). Single-
571 crystal refinement results.

572

573 **Figure 8.** Examples of site-specific dependencies between bond length and Ni site population of
574 natural schreibersite. (a) The $M2-M2^{\text{ix}}$ bond showing the most considerable decrease in length
575 upon increase of Ni population. (b) The $M3-M3^{\text{vii}}$ bond showing the strongest opposite behavior.
576 (c) Very complex variations for $M1-M3^{\text{ii}}$ bond. (d) The $M2-P$ bond is virtually insensitive to
577 variations in Ni content. Vertical scale is the same in all plots. Single-crystal refinement results.
578 Symmetry codes: (ii) $y, -x, -z+1$; (vii) $-x+1/2, -y-1/2, z-1/2$; (ix) $y+1/2, -x+1/2, -z+3/2$.

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580 **Tables**

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Table 1. Total Ni contents in schreibersite as determined by three methods ^a

Source ^b	Classification ^c	EMPA	Single-crystal ^d	Rietveld ^d
Hatrum Basin (Htr)	Terrestrial	0.03(1)	n.d. ^e	n.d. ^e
Zacatecas (1792) (Zct)	Iron-ungr	0.40(1)	0.35(2)	0.31(1)
São Julião de Moreira (SJM)	IIB (Ogg)	0.50(1)	0.54(3)	0.58(1)
Madoc (Mad)	IIIAB (Om)	0.58(1)	0.59(2)	0.62(1)
Bischtübe (Bst)	IAB-sLL (Og)	0.58(1)	0.58(3)	0.57(1)
Toluca (Tol)	IAB-sLL (Og)	0.62(1)	0.64(3)	0.68(1)
Glorieta Mountain (GM)	PMG-an	0.64(1)	0.61(3)	0.70(1)
Augustinovka (Aug)	IIIAB (Om)	0.72(1)	0.79(2)	0.59(1)
Mont Dieu (MD)	IIE (Om)	0.92(2)	0.89(4)	0.83(1)
Lazarev (Lzr)	Iron-ungr (Pal?)	0.99(2)	0.96(4)	0.89(1)
Krasnojarsk (Krs)	PMG-an	1.00(2)	0.96(3)	1.08(1)
Seymchan (Sey)	PMG	1.00(2)	1.08(3)	1.07(1)
Brenham (Brh)	PMG-an	1.01(2)	1.01(4)	1.02(1)
Cranbourne (Crm1)	IAB-MG (Og)	1.03(2)	1.05(2)	1.00(1)
Magura (Mgr)	IAB-MG (Og)	1.08(2)	1.03(6)	0.84(1)
Canyon Diablo (CD1)	IAB-MG (Og)	1.13(2)	1.19(3)	1.07(1)
Cosby's Creek (Csb)	IAB-MG (Og)	1.13(2)	1.05(5)	1.15(1)
Petropavlovsk (Ptr)	Iron-ungr (Om)	1.13(2)	1.05(4)	0.86(1)
Springwater (Spr)	PMG	1.14(2)	1.11(4)	1.22(1)
Finmarken (Fnm2)	PMG	1.25(2)	1.17(4)	1.13(1)
Canyon Diablo (CD2)	IAB-MG (Og)	1.28(3)	1.24(2)	1.23(1)
Finmarken (Fnm3)	PMG	1.48(3)	1.59(3)	1.62(1)
Cranbourne (Crm2)	IAB-MG (Og)	1.54(3)	1.55(3)	1.60(3)

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^a Calculated on 4 atoms per formula unit; ordered by Ni content (ascending). ^b Meteorites, except for the mineral from the Hatrum Basin. Sample codes are given in parentheses. ^c Meteorite classification according to Buchwald (1975) and Grady (2000). ^d Sums of Ni site populations obtained from X-ray structure refinements. ^e Not determined.

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Table 2. Absolute structure and unit-cell parameters of schreibersite ^a

Code	Flack <i>x</i> ^b	<i>a</i> (Å)	<i>c</i> (Å)	<i>V</i> (Å ³)	<i>c/a</i>
Htr	0.07(3)	9.1092	4.4621	370.25	0.4899
Zct	0.96(2)	9.0852	4.4661	368.64	0.4916
SJM	0.06(2)	9.0782	4.4655	368.02	0.4919
Mad	0.98(1)	9.0749	4.4654	367.74	0.4921
Bst	0.04(2)	9.0732	4.4651	367.58	0.4921
Tol	0.03(1)	9.0720	4.4651	367.48	0.4922
GM	0.03(3)	9.0698	4.4647	367.27	0.4923
Aug	0.39(2)	9.0683	4.4660	367.26	0.4925
MD	0.38(3)	9.0571	4.4660	366.35	0.4931
Lzr	0.97(1)	9.0532	4.4656	366.00	0.4933
Krs	0.04(1)	9.0487	4.4646	365.56	0.4934
Sey	0.97(2)	9.0509	4.4647	365.74	0.4933
Brh	0.03(1)	9.0495	4.4642	365.59	0.4933
Crm1	0.03(2)	9.0471	4.4647	365.44	0.4935
Mgr	0.03(2)	9.0s470	4.4644	365.40	0.4935
CD1	0.97(2)	9.0409	4.4638	364.86	0.4937
Csb	0.92(3)	9.0435	4.4649	365.16	0.4937
Ptr	0.03(2)	9.0424	4.4642	365.02	0.4937
Spr	0.96(1)	9.0426	4.4641	365.02	0.4937
Fnm2	0.06(3)	9.0416	4.4646	364.98	0.4938
CD2	0.01(1)	9.0363	4.4644	364.54	0.4941
Fnm3	0.54(3)	9.0279	4.4637	363.80	0.4944
Crm2	0.01(1)	9.0242	4.4620	363.37	0.4944

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^a Estimated standard deviations: *a* 5×10⁻⁴ Å; *c* 3×10⁻⁴ Å; *V* 0.05 Å³. ^b Absolute structure parameter determined from the single-crystal data (Flack and Bernardinelli 1999), relative to standard setting by Doenitz (1970).

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Table 3. Refined Ni populations at the metal sites of schreibersite

Code	Single-crystal data		Rietveld refinement	
	<i>M2</i>	<i>M3</i>	<i>M2</i>	<i>M3</i>
Htr	0	0	0	0
Zct	0.08(2)	0.27(1)	0.11(1)	0.20(1)
SJM	0.09(2)	0.44(2)	0.15(1)	0.43(1)
Mad	0.13(2)	0.46(2)	0.21(1)	0.41(1)
Bst	0.11(2)	0.47(2)	0.14(1)	0.43(1)
Tol	0.10(2)	0.54(2)	0.17(1)	0.51(1)
GM	0.11(2)	0.50(2)	0.19(1)	0.51(1)
Aug	0.21(2)	0.58(1)	0.12(1)	0.47(1)
MD	0.22(3)	0.67(2)	0.23(1)	0.60(1)
Lzr	0.24(3)	0.72(2)	0.28(1)	0.61(1)
Krs	0.26(2)	0.70(2)	0.40(1)	0.68(1)
Sey	0.33(2)	0.75(2)	0.33(1)	0.74(1)
Brh	0.26(3)	0.75(2)	0.34(1)	0.68(1)
Cm1	0.33(2)	0.72(1)	0.31(1)	0.69(1)
Mgr	0.31(5)	0.72(4)	0.26(1)	0.58(1)
CD1	0.39(2)	0.80(2)	0.34(1)	0.73(1)
Csb	0.32(4)	0.73(3)	0.42(1)	0.73(1)
Ptr	0.38(3)	0.67(3)	0.27(1)	0.59(1)
Spr	0.32(3)	0.79(2)	0.42(1)	0.80(1)
Fnm2	0.37(3)	0.80(2)	0.42(1)	0.71(1)
CD2	0.41(2)	0.83(1)	0.44(1)	0.79(1)
Fnm3	0.59(3)	1	0.62(1)	1
Cm2 ^a	0.55(3)	1	0.67(1)	0.88(2)

^a Besides, Rietveld refinement gave 0.05(1) Ni in the *M1* site.

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604 **Figures**

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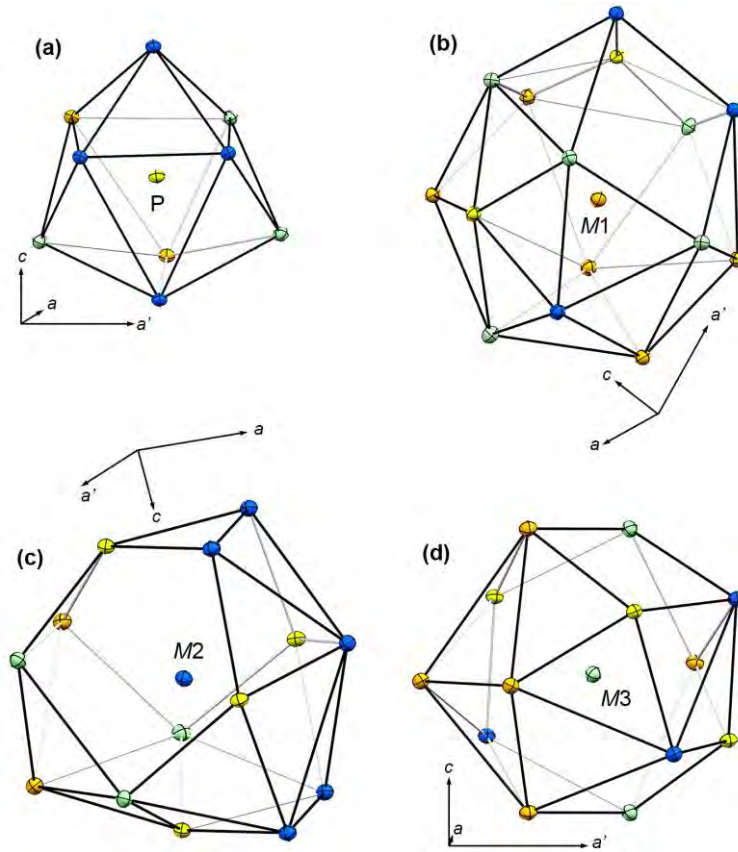
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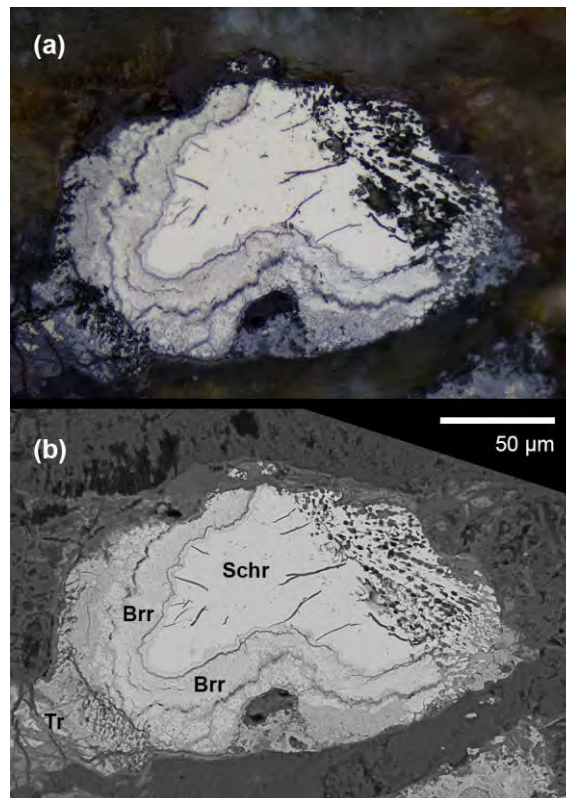
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Figure 1

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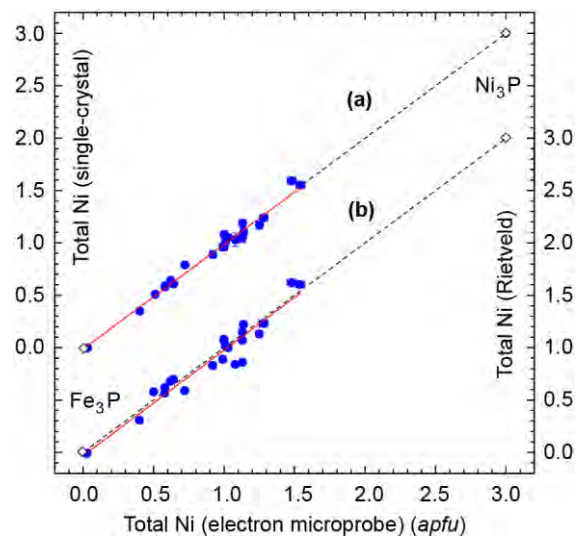
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Figure 2

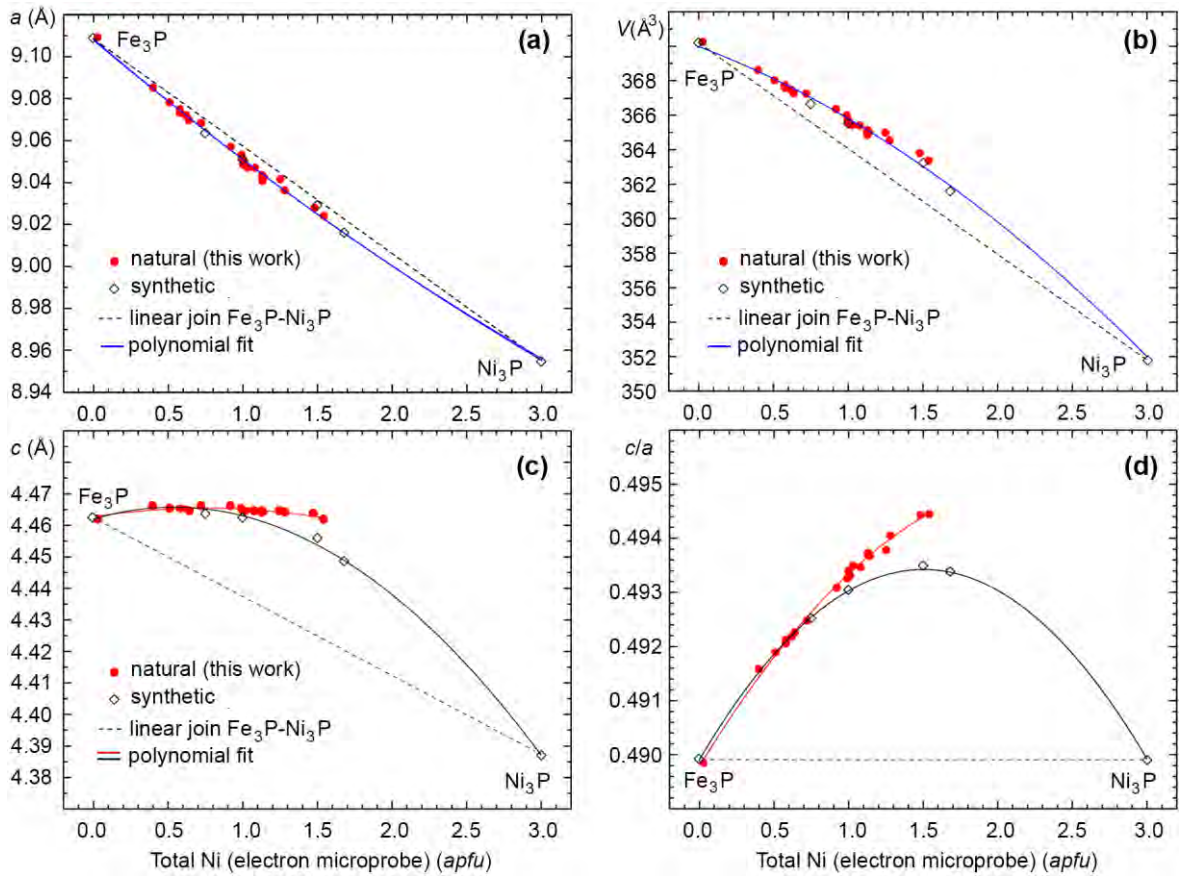


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Figure 3



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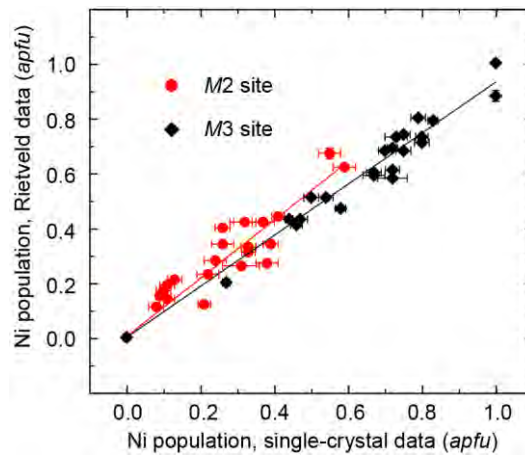
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Figure 4



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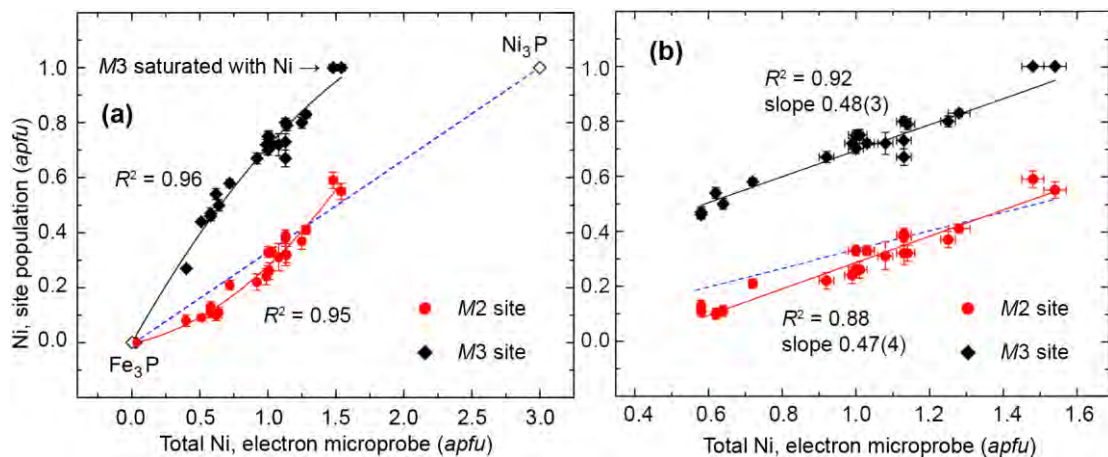
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Figure 5

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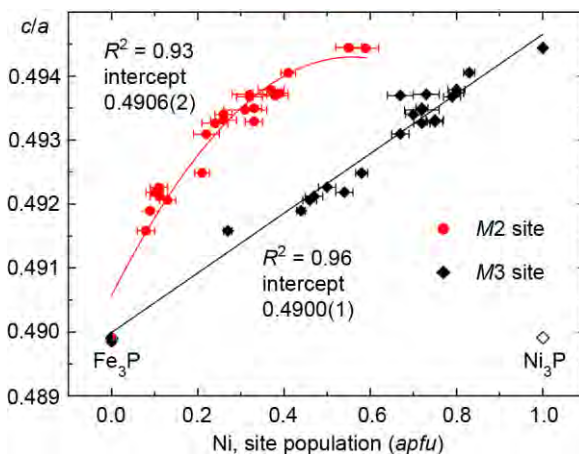
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Figure 6



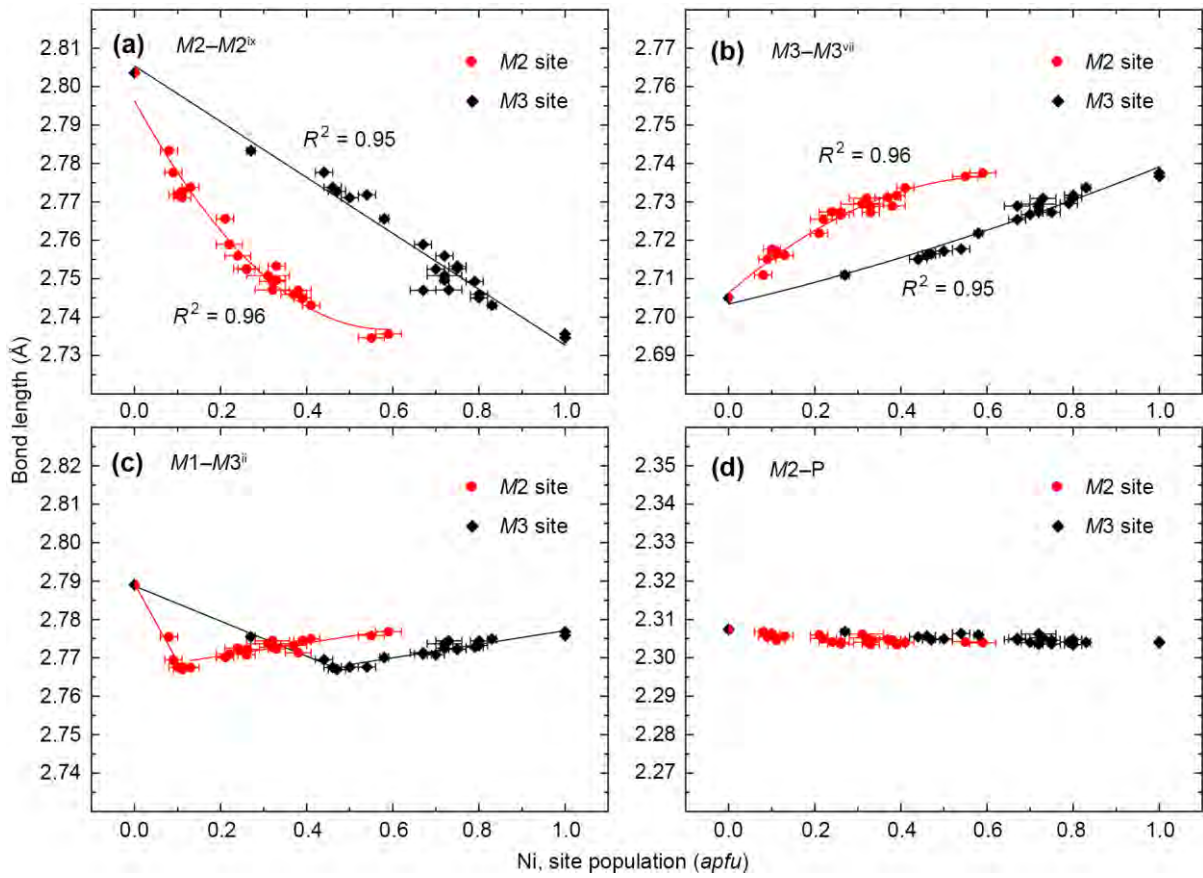
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Figure 7

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Figure 8