

## Revision 1

# Phase relationships in the system ZnS-CuInS<sub>2</sub>: insights from a nanoscale study of indium-bearing sphalerite

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### ABSTRACT

Micron- to submicron-scale indium-rich domains are preserved within sphalerite included in hornfels-hosted pyrrhotite from the Dulong polymetallic skarn, Yunnan, China. The nano-mineralogy of the ZnS-bearing blebs was investigated using scanning transmission electron microscopy on thinned foils extracted *in-situ* from pyrrhotite. Indium incorporation in sphalerite occurs via the coupled substitution  $2\text{Zn}^{2+} \leftrightarrow \text{Cu}^{+} + \text{In}^{3+}$ , the results thus allow insights into phase relationships in the system ZnS-CuInS<sub>2</sub> in which solubility limits are debated with respect to a cubic to tetragonal phase transition. The highest

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21 concentrations of In are measured in basket-weave domains from the smallest ZnS blebs or from un-  
22 patterned areas in coarser, irregular ZnS inclusions in pyrrhotite. Indium-rich domains contain 17-49  
23 mol.% CuInS<sub>2</sub>, whereas In-poor sphalerite contains <5 mol.% CuInS<sub>2</sub>. Atomic-scale metal ordering  
24 observed in In-(Cu)-rich ZnS domains was modelled as mixed sites in a cubic structure with  $P4\bar{3}m$   
25 symmetry and empirical formula [(Cu,In,Zn)<sub>3</sub>(Zn<sub>0.5</sub>Fe<sub>0.5</sub>)<sub>4</sub>S<sub>4</sub>. This sphalerite modification is distinct  
26 from the cubic-tetragonal phase transition reported elsewhere for analogous, synthetic phases with  
27 abundant planar defects. The Zn<sub>1.5</sub>Fe<sub>0.5</sub>CuInS<sub>4</sub> nanophase described here potentially represents a Fe-  
28 bearing polymorph of Zn<sub>2</sub>CuInS<sub>4</sub>, considered as an endmember in the sakuraiite solid solution series.  
29 At ≤50 mol.% CuInS<sub>2</sub> in the ZnS-CuInS<sub>2</sub> system, incorporation of In via coupled In+Cu substitution is  
30 promoted within a cubic ZnS phase with lower symmetry than sphalerite rather than into the spatially  
31 co-existing chalcopyrite of tetragonal symmetry. Solid state diffusion accounts for phase re-  
32 equilibration resulting in the basket-weave textures typical of In-(Cu)-rich domains in the smallest  
33 blebs, whereas fluid percolation assists grain coarsening in the irregular inclusions. The results show  
34 evidence for existence of more complex phase transition than previously recognized from experimental  
35 studies, and intriguingly, also to a potential eutectic in the system ZnS-CuInS<sub>2</sub>. Pyrrhotite-bearing  
36 hornfels in skarns may concentrate In and other critical metals hosted in sphalerite and related sulfides  
37 due to the efficient scavenging from fluid by these minerals and the subsequent preservation of those  
38 included phases by sealing within the pyrrhotite matrix.

39 **Keywords:** Sphalerite, indium, system ZnS-CuInS<sub>2</sub>, HAADF STEM

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## 40 INTRODUCTION

41 Sphalerite is the dominant source of indium (In), a critical element of economic interest due to  
42 expanding applications in photoconductors, semi-conductors and other advanced technologies (e.g.,  
43 [Werner et al. 2017](#)). Knowledge about how critical metals occur in nature is of importance for

44 identifying new resources and optimizing recovery from existing operations. This is particularly  
45 relevant in the case of In which seldom forms independent minerals.

46 Zinc sulfide (ZnS) crystallizes as close-packed structures, of which the most common are the cubic  
47 *3C* and hexagonal *2H* structures in sphalerite, (Zn,Fe)S, and its polymorph wurtzite, respectively  
48 (Pósfai and Buseck 1997, and references therein; Table 1). These are referred to as ZnS<sub>3C</sub> and ZnS<sub>2H</sub>  
49 hereafter. Indium is a chalcophile element, almost exclusively occurring in sulfides (Cook et al. 2011a)  
50 and, in most base metal sulfide ores, will be preferentially partitioned into sphalerite over co-existing  
51 Cu-Fe-sulfides (George et al. 2016). Unlike isovalent substitutions, e.g., Zn<sup>2+</sup> ↔ Fe<sup>2+</sup>, incorporation of  
52 trivalent atoms such as In<sup>3+</sup> requires a coupled substitution, of which the most common is: 2Zn<sup>2+</sup> ↔  
53 Cu<sup>+</sup> + In<sup>3+</sup> (e.g., Cook et al. 2012). Other substitution mechanisms have, however, been proposed based  
54 on measured inter-element correlations, including those involving available monovalent ions (e.g.,  
55 Ag<sup>1+</sup>), and Sn commonly considered of tetravalent charge. Such substitutions are more complex and  
56 stipulate presence of vacancies (□) or different oxidation state for Sn, e.g., 3Zn<sup>2+</sup> ↔ In<sup>3+</sup> + Sn<sup>3+</sup> + □  
57 (Belissont et al. 2014), 3Zn<sup>2+</sup> ↔ In<sup>3+</sup> + (Cu,Ag)<sup>+</sup> + Sn<sup>2+</sup> (Frenzel et al. 2016), and 4Zn<sup>2+</sup> ↔ In<sup>3+</sup> + Sn<sup>4+</sup>  
58 + (Cu,Ag)<sup>+</sup> + □ (Frenzel et al. 2016).

59 Conversion between the *3C* and *2H* structures, as well as formation of numerous ZnS polytypes, is  
60 readily promoted by screw dislocations and twin mechanisms as shown by transmission electron  
61 microscopy (TEM) and X-ray diffraction (e.g., Akizuki 1981; Fleet 1977). Phase transformations  
62 between ZnS species and substitution mechanisms can be accommodated by polysynthetic twinning  
63 along <111> planes in *3C* ZnS, as documented by TEM studies of Fe-rich, Ge-(Fe)-rich, In-Cu-(Fe)-  
64 bearing sphalerite, and analogous ZnS-derived compounds (e.g., Pósfai et al. 1988; Bente et al. 1998;  
65 Wagner et al. 2000a, b; Šrot et al. 2003; Ciobanu et al. 2011; Cook et al. 2015).

66 Studies of the system ZnS-CuInS<sub>2</sub> (Sombuthawee et al. 1978; Schorr et al. 2004) show extensive  
67 solubility but incomplete solid solution due to structural transition between cubic ZnS and the

68 tetragonal (chalcopyrite-type) structure of  $\text{CuInS}_2$  (roquesite). [Schorr and Wagner \(2005\)](#) observed a  
69 miscibility gap between  $\sim 40$  and  $\sim 90$  mol.%  $\text{CuInS}_2$  based upon a neutron diffraction study of metal  
70 ordering in the solid solution  $2\text{ZnS}-\text{CuInS}_2$ . Nonetheless, complete solid solution in the system  $\text{ZnS}-$   
71  $\text{CuInS}_2$  is supported by TEM studies of  $\text{Zn}_{2-2x}\text{Cu}_x\text{In}_x\text{S}_2$  (ZCIS;  $0.78 < x < 1$ ) alloys showing that the  
72 transition between tetragonal chalcopyrite-type ( $x > 0.78$ ) and cubic sphalerite-type ZCIS ( $x < 0.78$ ) is  
73 accommodated by formation of twinned nanodomains with ‘CuAu-I’ type ordering ([Wagner et al.](#)  
74 [2000a](#)). *Ab initio* calculations and synthesis of  $\text{CuInS}_2$  alloys with non-chalcopyrite ordering show that  
75 the  $\text{Cu}^{1+}$  and  $\text{In}^{3+}$  atoms are placed on alternating (100) planes (‘CuAu-I’ type substitution) rather than  
76 at (201) planes (chalcopyrite-type substitution) in the  $\text{ZnS}_{3C}$  structure, leading to compounds with  
77  $P\bar{4}m2$  symmetry ([Wei et al. 1992](#); [Su et al. 1998](#)). Metastable phases with ‘CuAu-I’ ordering occur  
78 spontaneously in the systems  $\text{ZnS}-\text{CuInX}_2$  ( $X=\text{S}, \text{Se}$ ) during polymorphic transition between  $\text{ZnS}_{3C}$  and  
79 chalcopyrite-type compounds at temperatures in the range 800-1300 K ([Wei et al. 1992](#), and references  
80 therein).

81 Advances in electron microscopy techniques, and particularly Z-contrast imaging using high-angle  
82 annular dark field scanning transmission electron microscopy (HAADF STEM) on thinned foils  
83 extracted *in-situ* from samples previously characterized with respect to composition, allows  
84 unparalleled insights into crystal-chemistry and metal ordering in ore minerals at the nanoscale  
85 ([Ciobanu et al. 2016a](#)).

86 Understanding of the crystal-structures of In-bearing ZnS and related phases draws largely on the  
87 results of investigation of synthetic samples and remains largely unstudied in natural materials.  
88 Although In-bearing sphalerite has been reported from a variety of polymetallic ores, occurrences  
89 lacking other sulfides into which the same trace elements may partition, are rare. The present HAADF  
90 STEM study is carried out on In-rich sphalerite occurring as inclusions within pyrrhotite, a matrix in  
91 which In is incompatible, thus offering an excellent opportunity to assess phase transitions among

92 natural phases in the system ZnS-CuInS<sub>2</sub>. This material derives from the In-bearing Dulong Sn-  
93 polymetallic skarn deposit, Yunnan Province, China (Xu et al. 2020). In this contribution, we aim to  
94 assess (i) In-distribution and concentration in sphalerite inclusions with various textures, and (ii)  
95 identify metal order or disorder throughout In-rich domains.

#### 96 **BACKGROUND: SPHALERITE AND RELATED SULFIDE GROUPS**

97 The debate surrounding crystal chemistry of species within the system Zn-Cu-In-Sn-Fe-S is useful  
98 in understanding phase transition from cubic to tetragonal and vice versa between related sulfide  
99 groups. Microanalysis of natural sphalerites have shown that In concentrations can extend up to wt.%  
100 levels (e.g., Cook et al. 2009 and references therein). There are, however, In-bearing, ZnS-related  
101 sulfides, i.e., sakuraiite and ishiharaite (Table 1), in which the sum of substituting metals can make up  
102 to 80-90 wt.%. Whereas ishiharaite (Cu,Ga,Fe,In,Zn)S was defined as a  $F\bar{4}3m$  sphalerite-type structure  
103 in which all metals are placed in a single crystallographic site (Márquez-Zavalía et al. 2014), the crystal  
104 chemistry of sakuraiite is still open to debate (see discussion in Momma et al. 2017). Sakuraiite was  
105 reported either as (Cu,Zn,Fe)<sub>3</sub>(In,Sn)S<sub>4</sub>, as tetragonal phase with extended In for Sn substitution (Kato  
106 1965; Shimizu et al., 1986; Momma et al. 2017), or cubic (Cu,Zn,In,Fe,Sn)S, with three possible space  
107 groups suggested ( $P432$ ;  $P\bar{4}3m$  and  $Pm\bar{3}m$ ; Kissin and Owens 1986) (Table 1).

108 A parallel can be drawn to the related stannite group whereby the join between stannite and  
109 k esterite, Cu<sub>2</sub>FeSnS<sub>4</sub> – Cu<sub>2</sub>ZnSnS<sub>4</sub> (Table 1) has also been subject to much debate in terms of  
110 continuous or discontinuous solid solution, phase stability and crystal structures (Table 1; Bonazzi et al.  
111 2003, and references therein). The Cu<sub>2</sub>FeSnS<sub>4</sub> phase had been considered as the mineral isostannite but  
112 was formally discredited (Kissin and Owens 1989), based on new analysis of type material, which was  
113 found to consist of stannite and k esterite. Kissin and Owens (1989) noted, however, that the study of  
114 type material did not preclude the potential existence of a cubic polymorph of stannite.

115 Two models, with different distribution of cations among positions at  $(0,0,0)$ ,  $(0, \frac{1}{2}, \frac{1}{4})$  and  $(0, \frac{1}{2}, \frac{3}{4})$   
116 coordinates, have been proposed for structurally-related yet nevertheless distinct minerals, i.e.,  $I\bar{4}2m$   
117 and  $I\bar{4}$  space groups for stannite and k esterite, respectively (Hall et al. 1978). These were later re-  
118 evaluated within a single, higher symmetry  $I\bar{4}2m$  structural model but with distinct cation occupancies  
119 for the two endmember minerals (Bonazzi et al. 2003). A cubic modification of stannite,  $\text{Cu}_{2-x}\text{Fe}_{1-x}\text{SnS}_4$ ,  
120 was identified by Rietveld methods as a ZnS-derivative with lower symmetry ( $P\bar{4}3m$ ; Table 1)  
121 and this is characterized by mixed sites comprising Sn + Fe (position 1a:  $0,0,0$ ) and Cu + Fe + Sn  
122 (position 3c:  $0, \frac{1}{2}, \frac{1}{2}$ ) (Evstigneeva and Kabalov 2001). This phase is likely the same phase  
123 synthesized earlier by Springer (1972). Re-examination of Springer's experimental products and  
124 Guinier films (Kissin 1989) confirmed the presence of such a phase. Kissin and Owens (1979) and  
125 Kissin (1989) showed, however, that end-member stannite and k esterite are synthesized at 800  C. One  
126 might therefore conclude that if a high-temperature, cubic polymorph of stannite exists, it must be  
127 unquenchable and that different synthetic products may be obtained depending on temperature and  
128 annealing steps during cooling, consistent with the conclusion of Kissin (1989) that at lower  
129 temperature conditions, Fe-rich compositions do not yield equilibrium stannite structures and  
130 compositions. In a subsequent publication, Evstigneeva et al. (2001) discussed a larger stannite family  
131 based on Cu-Fe isomorphism in these sulfides.

132 The latest reinvestigation of sakuraiite type material by single crystal X-Ray diffraction led to a  
133 pseudo-tetragonal phase ( $P4\bar{2}m$ ) considered as an intermediate member of the stannite- $\text{Zn}_2\text{CuInS}_4$   
134 solid solution (Momma et al. 2017). These authors distinguished between Cu, Zn and Fe atoms  
135 randomly distributed in two crystallographic sites (1a and 2f) and Sn which is placed in a distinct site  
136 (1d), concluding that this type of metal ordering is different from that of other known sulfides, e.g.,  
137 sphalerite ( $F\bar{4}3m$ ), stannite ( $I\bar{4}2m$ ), k esterite ( $I\bar{4}2m$  or  $I\bar{4}$ ) and chalcopyrite ( $I\bar{4}2d$ ) (Momma et al.  
138 2017).

139 The complexity of phases in the Zn–Cu–(Ag)–(Fe)–(In)–Sn–S system is further emphasized by the  
140 variety of minerals that have chalcopyrite- or stannite-like structures (Table 1). Some of these have  
141 been reported as nanoparticles (NPs) within sphalerite (e.g., Ciobanu et al. 2011) or have been  
142 synthesized as multi-component  $(\text{ZnS})_x(\text{Cu}_2\text{SnS}_3)_{1-x}$  and  $(\text{CuInS}_2)_x(\text{Cu}_2\text{SnS}_3)_{1-x}$  nanocrystals with  
143 metastable ZnS structures (e.g., Liu et al. 2011).

## 144 SAMPLES AND METHODOLOGY

145 The sample of pyrrhotite-bearing hornfels (biotite schist) contains some of the In-richest sphalerite  
146 in the Dulong deposit (In contents between 0.21 and 14.75 wt.% based on EPMA data; Xu et al. 2020).  
147 Five thin foils for nanoscale study (Supplemental<sup>1</sup> Fig. S1) were prepared using focused ion beam  
148 (FIB)-SEM by cutting across sphalerite blebs or boundaries to pyrrhotite (Fig. 1). Each foil was placed  
149 on a copper grid and thinned.

150 HAADF STEM imaging was performed using an ultra-high resolution, probe-corrected, FEI Titan  
151 Themis S/TEM. This instrument is equipped with the X-FEG Schottky source and Super-X EDS  
152 geometry. The Super-X EDS detector provides geometrically symmetric EDS detection with an  
153 effective solid angle of 0.8 Sr. Probe correction delivered sub-Ångstrom spatial resolution and an inner  
154 collection angle greater than 50 mrad was used for HAADF experiments using the Fischione HAADF  
155 detector. Diffraction measurements were performed using DigitalMicrograph™ 3.11.1 and Winwulff©  
156 1.4.0 software. Data from the American Mineralogist Crystal Structure Database were used for  
157 indexing of the electron diffractions. Crystal structure simulations were carried out using  
158 CrystalMaker® version 9.2.7 and STEM™ for xHREM software.

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<sup>1</sup> Deposit Item

159

## RESULTS

### 160 *Micron to nanoscale textures and indium distribution in sphalerite*

161 The studied sphalerite occurs as irregular, bleb-like, or lobate inclusions in pyrrhotite (Fig. 1). On the  
162 surface of the polished block, In-rich areas have patchy distribution within the irregular, coarser  
163 sphalerite inclusions (Fig. 1a, b) or make up the larger part of the smaller sphalerite blebs (Fig. 1c, d).  
164 Single grains of pyrrhotite can also occur within the coarser lobate sphalerite (Fig. 1e). The FIB cuts  
165 were placed across all three types of textures exposing the sphalerite inclusions at depth, as well as  
166 phase boundaries and the presence of chalcopyrite and other mineral NPs, particularly Bi-(Ag)-phases  
167 (Fig. 2).

168 HAADF STEM imaging shows In-rich and -poor domains with scalloped or concave mutual  
169 boundaries (Fig. 2a, b) and the presence of phyllosilicates that crosscut the sphalerite blebs (Fig. 2c).  
170 Nanometer-wide fractures are marked by the presence of chalcopyrite NPs, often along sets of trails  
171 (Fig. 2d). Relative enrichment in indium occurs along some mutual boundaries between sphalerite and  
172 pyrrhotite (Fig. 2e). In contrast, the In-rich domains in the sphalerite blebs display a basket-weave  
173 texture formed by nm-wide lamellae of In-poor sphalerite (Fig. 2f). In detail, the boundaries between  
174 In-rich and -poor domains from the coarser inclusions are very irregular whereas those between  
175 sphalerite and pyrrhotite are sharp (Fig. 2g). Similar enrichment in In is also observed surrounding NPs  
176 of native bismuth (Fig. 2h). Some of the domains richest in In tend to develop a euhedral morphology,  
177 e.g. as hexagonal platelets (Fig. 2i). Chalcopyrite NPs along fractures in the lobate sphalerite display  
178 euhedral to sub-rounded morphology (Fig. 2j). The margins between the sphalerite in pyrrhotite are  
179 marked by enrichment in Bi and Ag and by formation of discrete NPs of Bi-Ag-bearing minerals  
180 (possibly sulfosalts; Fig. 2k). Other fractures, no more than a few nm in width, are marked by  
181 enrichment in Cu, Si, and O (Supplemental<sup>1</sup> Fig. S2a, b).



182 EDS-STEM mapping and spot analysis shows further details of the distribution of In and other trace  
183 elements in sphalerite (Figs. 3-6; Supplemental<sup>1</sup> Figs. S2, 3). Indium is present along the entire length  
184 of the sphalerite inclusion exposed at depth in foil #1, albeit with a patchy, irregular distribution (Fig.  
185 3a). EDS maps at higher resolution show that In concentrations correlate positively with Cu (Fig. 3b).  
186 Notably, chalcopyrite inclusions within the same area display no In-content (Fig. 3c), whereas the  
187 correlation between Cu and In is clearest in sphalerite from an adjacent domain in the same area (Fig.  
188 3d). The assemblage also includes NPs of native copper, observed within both chalcopyrite and  
189 sphalerite (Fig. 3c, e).

190 Maps of the sub-rounded sphalerite inclusion in pyrrhotite from foil #2 (Fig. 4) display conspicuous  
191 In-enrichment around NPs of native bismuth and also around boundaries to adjacent pyrrhotite (Fig. 4b,  
192 c). The illustrated inclusion is crosscut and displaced by lamellae of phyllosilicates, most likely an  
193 intergrowth of chlorite and phlogopite (Supplemental<sup>1</sup> Fig. S3). The smallest sphalerite blebs in  
194 pyrrhotite show In-enrichment in the domain with basket-weave texture (foil#3; Fig. 5a). In both the  
195 irregular and bleb-like sub-types of sphalerite inclusions in pyrrhotite, Fe is concentrated within the In-  
196 rich domains rather than in the In-poor sphalerite (Figs. 3d and 5a). Chalcopyrite NPs mapped from  
197 trails within sphalerite (Fig. 5b, Supplemental<sup>1</sup> Fig. S2c, d) show negligible In content.

198 Representative EDS spectra for sphalerite, chalcopyrite and various NPs are shown in  
199 Supplemental<sup>1</sup> Figure S4. Sphalerite from In-rich areas in different foils/textures show comparable Fe  
200 to In ratios (spectra #2-5; 7 shown on Supplemental<sup>1</sup> Figure S4). Spectra obtained from adjacent In-  
201 poor sphalerite still show the presence of Fe and minor In (spectra #1 and 6 on Supplemental<sup>1</sup> Figure  
202 S4). The lobate sphalerite from foil #5 shows Fe at comparable concentration with the other areas but  
203 the lowest content of In (spectrum #8 in Supplemental<sup>1</sup> Figure S4). Chalcopyrite from the irregular  
204 sphalerite (Fig. 3c) differs from the chalcopyrite along trails in lobate sphalerite (Fig. 5b) in that it  
205 contains measurable Zn. Neither contain other minor/trace elements at concentration levels measurable

206 by EDS STEM analysis (spectra #9 and 10 in [Supplemental<sup>1</sup> Figure S4](#)). The ‘trail chalcopyrite’,  
207 however, hosts Ag- and Ag-Bi-bearing NPs, comparable to those present along margins between  
208 sphalerite and pyrrhotite in the same sample (foil#5; spectra #11-13 in [Supplemental<sup>1</sup> Figure S4](#)).

209 EDS spot analyses obtained from the mapped areas ([Table 2](#); [Supplemental<sup>1</sup> Table S1](#)) have been  
210 plotted in terms of mol.% CuInS<sub>2</sub> versus lattice parameter using data given by [Wagner et al. \(2000\)](#)  
211 ([Fig. 6](#)). Since very small volumes were analyzed without use of standards, these data must be  
212 considered semi-quantitative. Mol.% CuInS<sub>2</sub> was calculated by assigning equal atom% Cu as In (to  
213 correct for overreading of Cu from the grid) and normalizing to 100%; Fe content was added to Zn.  
214 The plot shows a compositional spread for In-rich sphalerite between 17-49 mol.% CuInS<sub>2</sub>, whereas In-  
215 poor sphalerite contains <5 mol.% CuInS<sub>2</sub>.

#### 216 *High-resolution imaging – defects and phase transformation*

217 In order to assess the identity of substituted ZnS phases mapped as inclusions in pyrrhotite, high-  
218 resolution HAADF STEM imaging of the five foils was undertaken by tilting the specimen up to four  
219 main zone axes in each area or sample. Fast Fourier transform (FFT) patterns obtained from images  
220 were indexed, and STEM simulations on these orientations show a good match with the images and  
221 crystal structure models ([Fig. 7](#)). Simulations undertaken for roquesite as an analog for a tetrahedral  
222 structure ([Supplemental<sup>1</sup> Fig. S5](#)) show similar patterns as cubic ZnS on the [001] zone axis but distinct  
223 ‘dumbbell’ In and Cu atom pairs for the [100], [101], [111] and [1 $\bar{2}$ 1] zone axes. Such ‘dumbbell’  
224 patterns were however observed only across small strips at phase boundaries between In-rich and -poor  
225 areas (see below).

226 Simple stacking faults occur as stepwise defects along {111} directions ([Fig. 8a](#)), but the continuity  
227 of the atomic arrays across the boundaries between In-poor and -rich domains, is generally preserved in  
228 all inclusion types ([Fig. 8b, c](#)). Nonetheless, In-rich sphalerite at the boundary with native bismuth

229 displays quasi-regular cation ordering expressed as alternating brighter and dimmer atoms along  $b$  and  
230  $c$  axes (Fig. 8d, e; see also below).

231 In further detail, the  $\{111\}$  defects are associated with lattice distortion leading to irregular atom  
232 displacements (Fig. 9a-f). These defects can form a regular network defining euhedral shapes (Fig. 9a),  
233 resembling those shown for In-rich areas on Figures 2i and 3d. More often, however, these defects  
234 form a zig-zag pattern and their length is less than 50 nm (Fig. 9b). In either case the FFT patterns on  
235 the  $[0\bar{1}1]$  zone axis show disorder on  $\langle 111 \rangle^*$  lattice vectors rather than twinning (Fig. 9c). Atom  
236 arrangements change along the stacking faults from simple kinks to several dumbbell arrays (Fig. 9d-f).  
237 Screw dislocations along  $\{111\}$  fault planes are associated with variable width and atomic disorder  
238 (Fig. 9g). Structural disorder is also observed as local defects shown as spotted darker areas on  $[1\bar{1}1]$   
239 (Fig. 9h). In detail, such areas are associated with a decrease in total signal intensity, suggesting the  
240 presence of atom vacancies (Fig. 9i).

241 Locally, a transformation from  $[100]$ -cubic ZnS to  $[100]$ -tetragonal roquesite-type is observed along  
242 10-20 nm-wide stripe at the In-enriched boundary between sphalerite and pyrrhotite (foil #2; Fig. 10a).  
243 In contrast, preservation of the  $[100]$  cubic ZnS structure across boundaries of In-enrichment is  
244 associated with quasi-regular ordering of brighter/larger atoms as a  $\sim 5\text{\AA} \times 5\text{\AA}$  pattern, which is easy to  
245 visualize at the direct contact to ZnS<sub>3C</sub> (Fig. 10b). The FFT pattern (inset, Fig. 10b) shows the structure  
246 with quasi-ordering of brighter atoms within the ZnS substructure. In closer detail, the roquesite-type  
247 pattern displays a dumbbell atomic arrangement with satellite reflections at  $\frac{1}{2} \langle 011 \rangle^*$  (Fig. 10c, d).  
248 This is different from the square,  $\sim 5\text{\AA} \times 5\text{\AA}$ , pattern formed by brighter atoms attributable to columns  
249 of In-(Cu, Zn) mixed sites (Fig. 10e). Corresponding FFT pattern is distinct from the one shown as  
250 Figure 10d in that it features additional satellite reflections along  $b^*$  and  $c^*$ , which are forbidden for the  
251 space group  $F\bar{4}3m$  of ZnS<sub>3C</sub> (compare Fig. 10f with Fig. 7a). The HAADF STEM image and FFT for

252 this type of ZnS transformation are also distinct from the [001] roquesite-type structure ([Supplemental<sup>1</sup>](#)  
253 [Fig. S5](#)).

254 Pyrrhotite ([Supplemental<sup>1</sup> Fig. S6](#)) and sphalerite are not observed to be in epitaxial relationship in  
255 any of the samples. Selected area of electron diffraction (SAED) obtained from pyrrhotite images as in  
256 [Supplemental<sup>1</sup> Figure S6](#) were indexed using the monoclinic Fe<sub>7</sub>S<sub>8</sub> polytype with *C2/c* space group of  
257 [Powell et al. \(2004\)](#) ([Supplemental<sup>1</sup> Fig. S6](#)). In detail, the satellite reflections that express metal  
258 vacancy ordering are faint and comparable with those known for 1C-type pyrrhotite ([Posfai et al.](#)  
259 [2000](#)).

## 260 DISCUSSION

### 261 *Trace element incorporation and defects*

262 The studied In-rich ZnS from Dulong is one among many natural specimens showing enrichment in  
263 In and other accompanying elements via metal substitution for Zn ([Fig. 11](#)). For example,  
264 compositionally towards the ZnS end of the ZnS-CuInS<sub>2</sub> join is the exceptional In-bearing sphalerite  
265 from Toyoha (Japan) sphalerite, which displays exceptional complexity of micron-scale growth  
266 rhythms with respect to In and Cu ([Cook et al. 2009](#); [Ciobanu et al. 2011](#); [Shimizu and Morishita](#)  
267 [2012](#)). Increased structural complexity, polytypism and planar defects such as twins are associated with  
268 oscillatory zoning patterns and extend down to the nanoscale ([Ciobanu et al. 2011](#)), yet crystal-  
269 structural continuity across In-poor and In-rich boundaries is recognized, as indicated by HAADF  
270 STEM imaging ([Ciobanu et al. 2016b](#)). This observation implies random metal substitution on a single  
271 site, thus preserving the cubic sphalerite symmetry, as in the high-substituted ZnS structure of  
272 ishiharaite ([Márquez-Zavalía et al. 2014](#)).

273 Assessment of HAADF STEM images, as well as FFT patterns produced by defects, is necessary to  
274 better constrain the changes within domains with distinct patterns of metal ordering. Although the In-

275 bearing sphalerite from Dulong lacks any chemical zoning or abundant planar defects, the latter, where  
276 present (Fig. 9a-g), clearly support formation of high-In domains in the sphalerite. Transition between  
277  $ZnS_{3C}$  and tetragonal roquesite-type ( $I\bar{4}2m$  space group) structures is limited to such narrow defects  
278 (Fig. 10a, c).

279 Evidence for more complex In substitution mechanisms could be associated with the presence of  
280 localized defects (Fig. 9h, i), in which metal loss can be attributed to vacancies, as, for example, those  
281 suggested for Sn incorporation into ZnS (e.g., Frenzel et al. 2016). Although Sn concentrations are  
282 below the detection limit of the STEM EDX detectors, it is likely this element, otherwise abundant in  
283 the Dulong deposit (Xu et al., 2020) is also present in the In-rich ZnS phase at concentrations of least  
284 some to tens of ppm.

#### 285 *Cubic modification of In-bearing ZnS: isomorphism and polymorphism in the system Zn-Cu-In-Sn-Fe*

286 The In-rich ZnS from Dulong (up to ~50 mol.%  $CuInS_2$ ; Fig. 6) that displays metal ordering shows  
287 FFT patterns comparable with electron diffractions calculated for tetragonal  $Zn_{2-2x}Cu_xIn_xS_2$  (ZCIS;  
288  $0.78 < x < 1$ ) alloys but with numerous nanotwins viewed on [001] direction (Wagner et al. 2000b).  
289 However, the  $\frac{1}{2} c^*$  and  $\frac{1}{2} b^*$  satellite reflections observed in the present study (Fig. 10f) cannot be  
290 attributed to twins and their superpositions, as had been discussed for the alloys, since such planar  
291 defects are not present (HAADF STEM images shown as Figs. 8c and 10b). The HAADF STEM  
292 images and corresponding FFT patterns can be simulated if we consider a model comparable to the  
293 cubic modification of the  $Cu_2FeSnS_4$  compound reported by Evstigneeva and Kabalov (2001). In our  
294 case, the In-rich ZnS with  $P\bar{4}3m$  symmetry and empirical formula  $[(Cu,In,Zn)_3(Zn_{0.5}Fe_{0.5})]_4S_4$  was  
295 modelled as a phase with two mixed sites: Me1a (In, Cu, and Zn); and Me3c (Zn and Fe) (Fig. 12a).  
296 The simulated electron diffraction and STEM images (Fig. 12b, c) show a good fit with the measured  
297 FFT patterns and HAADF STEM images (Fig. 10e, f). HAADF signal intensity profiles obtained  
298 across selected directions in the In-rich ZnS displaying metal ordering show variation, as predicted by

299 the proposed crystal model (Fig. 12d, e). Importantly, these profiles are distinct from those recorded in  
300 adjacent In-poor ZnS (Fig. 12f, g).

301 As a result, we conclude that partial metal ordering occurring in the In-rich ZnS from Dulong  
302 represents a cubic modification of sphalerite that is distinct from the cubic-tetragonal phase transition  
303 reported for ZCIS compounds (Fig. 6). The nanophase described here could be considered as a Fe-  
304 bearing cubic polymorph of endmember  $\text{Zn}_2\text{CuInS}_4$  in the sakuraiite solid solution series (Shimizu et  
305 al. 1986; Momma et al. 2017), assuming the latter is proven to have tetragonal symmetry. Isomorphism  
306 between the synthetic cubic modification of the  $\text{Cu}_2\text{FeSnS}_4$  phase (Evstigneeva and Kabalov 2001) and  
307  $\text{Zn}_{1.5}\text{Fe}_{0.5}\text{CuInS}_4$  (the In-rich ZnS phase from Dulong) extends the ‘larger stannite family’ of  
308 Evstigneeva et al. (2001) to the sphalerite-related structures. The [Cu,Sn] for [(Zn,Fe)In] substitution  
309 discussed for sakuraiite-k esterite-stannite along the trend of In/(In+Sn) in phases shown on the Cu-Fe-  
310 Zn plot obtained for assemblages from Ikuno Mine, Japan (Shimizu et al. 1986; Momma et al. 2017),  
311 may be only one tie-line among several others expressing co-crystallization of phases at specific  
312 conditions and not a generic trend.

### 313 *Formation of In-bearing sphalerite in the Dulong skarn*

314 The diversity of textures involving sphalerite present in the Dulong samples can be interpreted in  
315 terms of evolving In-bearing mineral speciation derived from the same early, prograde fluid interacting  
316 with host pyrrhotite, via solid state mineral re-equilibration in the smallest blebs to fluid percolation  
317 becoming more substantial from the irregular to the larger, lobate sphalerite (Fig. 1). The basket-weave  
318 textures observed for the smallest In-rich blebs (Figs. 1c, 2f, and 5a) are comparable to those obtained  
319 for the pseudo-binary system  $2(\text{ZnTe})\text{-CuInTe}_2$  (Roussak et al. 2005), where they were attributed to  
320 sub-solidus transition from cubic to tetragonal symmetry with cation type ordering of a CuPt-type  
321 structure. In the studied samples, cubic to tetragonal transition (roquesite-type) is observed only in the  
322 larger, irregular inclusions. The coexistence of sphalerite, containing low levels of In, and near

323 endmember roquesite is observed wherever bulk Zn/In ratios in the assemblage are low (e.g., [Cook et](#)  
324 [al. 2011b](#)), and may be indicative of either non-ideal solid solution or the impact of superimposed  
325 diffusion-driven processes.

326 The basket-weave textures are directly attributable to solid state diffusion during which phase re-  
327 equilibration takes place, but the curvilinear boundary between In-rich and -poor domains in the blebs  
328 ([Fig. 2b](#)) more likely represents an initial unmixing or phase-separation corresponding to a binary  
329 eutectic at ca. 50 mol. % CuInS<sub>2</sub> in the system Zn-Cu-In-S (if Fe is ignored).

330 The irregular, coarser inclusions with In-bearing sphalerite show more varied chemistry, with  
331 nucleation of Cu-, and Bi-, or Bi-Ag-NPs either within chalcopyrite inclusions or along grain  
332 boundaries with host pyrrhotite ([Figs. 3-5](#)). Coarsening of blebs to irregular inclusions predates  
333 formation of phyllosilicates given the crosscutting relationships illustrated in [Fig. 4a](#), indicating that  
334 precipitation from In-bearing fluids must have been prior to such retrograde skarn alteration. This  
335 episode must nonetheless have followed pyrrhotite formation considering the lack of coherent  
336 intergrowths with sphalerite.

337 The low concentrations of In in the lobate sphalerite indicates that formation of larger pockets of  
338 base metal mineralization ([Fig. 1e](#)) effectively dilute rather than concentrate In within sphalerite. The  
339 tendency for In to be preferentially incorporated into sphalerite over coexisting chalcopyrite at  
340 equilibrium ([George et al. 2016](#)) is demonstrated here down to the nanoscale ([Fig. 5b](#)). The  
341 concentration of Cu within In-rich ZnS rather than in the adjacent chalcopyrite NPs is an important  
342 observation, in that it corroborates the partitioning of In and Cu into cubic ZnS relative to tetragonal  
343 CuFeS<sub>2</sub>. This is assuming, however, that the chalcopyrite NPs form from locally-generated  
344 concentrations of Cu-Fe-rich fluid source, or they post-date initial crystallisation of the In-Cu-Fe-  
345 bearing ZnS phase. This shows that the cubic to tetragonal transition in the system ZnS-CuInS<sub>2</sub> is not

346 as easy to constrain from natural assemblages, which inherently include Fe, among other cations that  
347 could play a role in phase stability and solid-state diffusion responsible for metal ordering.

348 The presence of Ag- and Bi-NPs indicates, however, a comparable fluid signature with that which  
349 precipitated the highest In-sphalerite, thus indicating a continuous mineralizing process rather than  
350 discrete stages each featuring distinct fluids. Since bismuth is not easily incorporated into sphalerite,  
351 the bismuth NPs provide a diffusion barrier for In-enrichment in ZnS at the contact to such NPs (Fig.  
352 4a, b). Moreover, this barrier also allows for metal ordering leading to the cubic In-rich ZnS with  
353 mixed ordering on tetrahedral sites (Figs. 8d and 10b).

## 354 **IMPLICATIONS**

355 The concentration of In in the initial skarn stage at Dulong is probably not higher than in many other  
356 skarn systems, but the retention of unusually high concentrations in early sphalerite relative to later,  
357 Sn-rich, base metal sulfide assemblages can be attributed to locking within the pyrrhotite matrix.

358 Pyrrhotite-bearing hornfels associated with polymetallic skarns may thus represent a critical  
359 minerals target. Recent studies have shown enrichment of In, among other critical metals, in basaltic  
360 lava (Greaney et al. 2017), the same concept could potentially be applied to pyrrhotite from igneous  
361 mafic rocks, in which ZnS and related sulfides would act as scavengers for critical metals from the  
362 melt.

363 Nanoscale studies of natural sphalerite have a critical role to play in gaining a better understanding  
364 of the mechanisms of minor element incorporation in sulfides. The research described here also  
365 represents a contribution on the fundamental character of solid solution series in common minerals  
366 such as sphalerite and related sulfides.



367

## CONCLUDING REMARKS

- 368 1. High-In sphalerite (17-49 mol.% CuInS<sub>2</sub>) is preserved as micron- to submicron-sized areas within  
369 ZnS blebs and irregular inclusions in a hornfels-hosted pyrrhotite matrix. Since the incorporation of  
370 In within ZnS is coupled with Cu, nanomineralogy of ZnS-bearing assemblages allows insights into  
371 phase relationships in the system ZnS-CuInS<sub>2</sub>.
- 372 2. HAADF STEM imaging of In-rich ZnS domains shows metal ordering modelled as mixed sites in a  
373 sphalerite-type structure with  $P\bar{4}3m$  symmetry and empirical formula [(Cu,In,Zn)<sub>3</sub>(Zn<sub>0.5</sub>Fe<sub>0.5</sub>)<sub>4</sub>S<sub>4</sub>.  
374 This cubic modification of sphalerite is distinct from the cubic-tetragonal phase transition reported  
375 for analogous, synthetic phases elsewhere.
- 376 3. The Zn<sub>1.5</sub>Fe<sub>0.5</sub>CuInS<sub>4</sub> nanophase is isomorphous with the cubic modification of Cu<sub>2</sub>FeSnS<sub>4</sub> and  
377 potentially represents a Fe-bearing polymorph of Zn<sub>2</sub>CuInS<sub>4</sub>, considered as an endmember in the  
378 sakuraiite solid solution series.
- 379 4. Indium enrichment, including transformation between [100]-cubic ZnS to [100]-tetragonal  
380 roquesite-type structure, is only promoted locally along planar defects (stacking faults and screw  
381 dislocations along {111}).
- 382 5. Indium concentration in Cu-bearing ZnS domains rather than in adjacent chalcopyrite NPs  
383 corroborates partitioning of In and Cu into cubic ZnS relative to tetragonal CuFeS<sub>2</sub>.
- 384 6. Comparable degrees of In-enrichment in sphalerite are achieved during (i) phase  
385 separation/unmixing of In-(Cu)-rich domains in ZnS in the smallest, micron-sized In-Cu-Zn-S-  
386 bearing blebs with curved domain boundaries; and (ii) fluid percolation resulting in bleb-coarsening  
387 and irregular morphology. Solid-state diffusion accounts for phase re-equilibration in the basket-  
388 weave texture typical of In-(Cu)-rich domains in the smallest blebs. Metal ordering, leading to the  
389 cubic In-rich ZnS with mixed ordering on tetrahedral sites, is assisted by the presence of diffusion

390 barriers such as Bi-NPs in the coarsest irregular blebs. An increase in fluid/rock ratio dissipates In  
391 within the largest, lobate sphalerite containing the lowest content of this element.

392 7. Pyrrhotite-bearing hornfels in a skarn deposit is a potential host for In and other critical metals  
393 hosted in sphalerite and related sulfides due to efficient scavenging from fluid by these minerals  
394 and subsequent locking within the pyrrhotite matrix.

#### 395 ACKNOWLEDGEMENTS

396 This work was carried out while J.X. was a visitor at the University of Adelaide. Funding from the National  
397 Key RandD Program of China (2017YFC0602500), Key Laboratory of Mineral Resources (KLMR2017-13),  
398 China Postdoctoral Science Foundation, and the China Scholarship Council. We gratefully acknowledge  
399 Microscopy Australia for access to instrumentation. We acknowledge insightful comments two *American*  
400 *Mineralogist* reviewers, Associate Editor Daniel Harlov, as well as comments from two reviewers of an earlier  
401 version of this manuscript.

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522 **Figure captions**

523 **Figure 1.** BSE (a-d) and reflected light (e) images showing sphalerite inclusions in pyrrhotite and the location of  
524 FIB cuts (foils #1-5). (a, b) Irregular In-rich sphalerite (In-Sp) inclusions in pyrrhotite (Po). (c, d) Bleb In-Sp  
525 along trails in Po. (e) Lobate sphalerite (Sp) intergrown with Po and chalcopyrite (Cp). Note inclusions of Cp of  
526 variable size in Sp and euhedral platelet of Po. Qz=quartz.

527 **Figure 2.** HAADF STEM images showing details of In-rich sphalerite (Sp). (a) Irregular sphalerite (Sp)  
528 inclusion (Foil #1) displaying In-rich and -poor domains with scalloped mutual boundaries. Note chalcopyrite  
529 (Cp) in sphalerite hosts a nanoparticle (NP) of native copper (Cu). (b) Bleb-like Sp inclusion with In-rich ZnS  
530 showing concave boundary (Foil #3). (c) Coarser ZnS inclusion in pyrrhotite (Po) crosscut by phyllosilicates  
531 (phlogopite+chlorite; Phl/Chl). Note native copper and native Bi-NPs within ZnS (Foil #2). (d) Trails of Cp NPs  
532 within lobate Sp (Foil #5). (e) Indium-rich ZnS at the boundary to pyrrhotite (inclusion shown in (c)). (f) Basket-  
533 weave texture (arrowed) in In-rich ZnS from bleb in (b). (g) Detail of In-rich ZnS margin displaying  
534 scalloped/irregular and sharp morphology towards Sp and Po, respectively (Foil #1). (h) Scalloped boundary  
535 between In-rich ZnS in Sp around a native bismuth NP (Foil #2). (i) Euhedral In-Cu-rich ZnS inclusions in Sp  
536 (Foil #1). (j) Sub-rounded chalcopyrite inclusion along trails in sphalerite (Foil #5). (k) Margins between  
537 sphalerite and pyrrhotite marked by enrichment in Bi and Ag and by formation of discrete NPs of Ag-Bi-bearing  
538 minerals, possibly sulfosalts (Ag-Bi-ss; Foil #5).

539 **Figure 3.** STEM EDS maps for selected elements (as labelled) for one of the largest, irregular sphalerite  
540 inclusions displaying In enrichment (Foil #1). (a) shows composite maps (In-Zn-Cu) giving an overview of the  
541 detailed areas shown in (b)-(e). Note the obvious correlation between Cu and In distribution in ZnS (b-d).  
542 Cp=chalcopyrite; Sp=sphalerite; Po=pyrrhotite. Circles and # numbers correspond to EDS-STEM spectra in  
543 [Supplemental<sup>1</sup> Fig. S4](#).

544 **Figure 4.** STEM EDS maps for selected elements (as labelled) showing In-enrichment in one of the coarser,  
545 bleb-like sphalerite (Sp) inclusions with marginal corrosion and displacement relative to a lamellae of  
546 phlogopite/chlorite (Phl/Chl) (Foil #2). (a) shows a composite map (Si-Fe-Cu-In-Zn) giving an overview of the  
547 detailed areas with In-enrichment shown in (b) and (c). Note In-enrichment around native bismuth NPs in (b)

548 and the Cu-, In-rich margin between sphalerite and pyrrhotite (Po) in (c). Circles and # numbers correspond to  
549 EDS-STEM spectra in [Supplemental<sup>1</sup> Fig. S4](#).

550 **Figure 5.** HAADF STEM images (right) and STEM EDS maps for selected elements (as labelled) for the  
551 smallest sphalerite (Sp) blebs (a; Foil #3) and the larger, lobate sphalerite with chalcopyrite (Cp) inclusions (b;  
552 Foil #5). In (a), note the basket-weave texture for the In-rich domain (arrowed). In (b), In-richer margins are  
553 present at the contacts between chalcopyrite and host, In-bearing sphalerite. Green arrows show Ag-bearing NPs.  
554 Circles and # numbers correspond to EDS-STEM spectra in [Supplemental<sup>1</sup> Fig. S4](#).

555 **Figure 6.** Plot showing compositional variation among the analyzed phases in the system ZnS-CuInS<sub>2</sub>. Lattice  
556 parameters are from [Wagner et al. \(2000a\)](#). ZCIS<sup>1</sup>: intermediate compound Zn<sub>2-2x</sub>Cu<sub>x</sub>In<sub>x</sub>S<sub>2</sub> (0.78<x<1) plotted for  
557 x=0.78 (from [Wagner et al. 2000b](#)).

558 **Figure 7.** From left to right, FFT patterns, HAADF STEM images, simulations and structural models for  
559 sphalerite viewed on (a) [100], (b) [01 $\bar{1}$ ], (c) [1 $\bar{1}$ 1], and (d) [1 $\bar{2}$ 1] zone axes. The main structural motifs are  
560 marked as overlays on the images, simulations and crystal models.

561 **Figure 8.** High-resolution HAADF STEM images showing details of sphalerite (Sp); (a-b) Specimen tilted on  
562 [0 $\bar{1}$ 1]<sub>ZnS<sub>3</sub>C</sub> and (c-e) [100]<sub>ZnS<sub>3</sub>C</sub>. (a) Stepwise, stacking fault defects along {111} directions in ZnS<sub>3</sub>C (b)  
563 Continuous boundary between In-poor and -rich domains (dashed line; Foil #1). Note that lattice distortions  
564 (dotted line) can be locally observed in the latter. (c) In-depleted Sp lamella from the basket-weave texture in In-  
565 rich Sp (Foil #3) Note structural continuity across the In-rich matrix. (d) Sharp contact between native bismuth  
566 and In-rich Sp in Foil #2. (e) Detail of (d) showing alternating bright and less-bright atoms with  $\frac{1}{2} b \times \frac{1}{2} c$   
567 periodicity. This is different from ZnS<sub>[100]</sub> showing atoms of equal HAADF intensity (compare with [Figure 7a](#)).

568 **Figure 9.** High-resolution HAADF STEM images showing defects in sphalerite (Sp); (a-g) specimen tilted on  
569 [0 $\bar{1}$ 1]<sub>ZnS<sub>3</sub>C</sub>, and (h, i) on [1 $\bar{1}$ 1]<sub>ZnS<sub>3</sub>C</sub>. (a, b) Stacking fault defects along conjugate {111} directions (arrowed)  
570 occurring close to areas of In-enrichment such as those shown in [Figure 3d](#) (Foil #1). (c) FFT pattern  
571 corresponding to Sp with such conjugate {111} defects. Arrows show streaking along defect directions. (d-f)  
572 Image cropped from (a) and (b) showing atom displacements along the defect highlighted as red and yellow  
573 circles. Atom arrangements change along the stacking faults from simple kinks in (d) to several dumbbell arrays



574 in (e, f). (g) Screw dislocations along {111} fault planes (dashed lines) display atomic disorder. (h). Structural  
575 disorder is also observed as local defects of darker intensity (arrowed). (i) Image cropped from (h) showing the  
576 decrease in total signal intensity and the disappearance of atoms along {220}, suggesting the presence of metal  
577 vacancies.

578 **Figure 10.** High-resolution HAADF STEM images showing crystal structural modifications across boundaries  
579 of In-enrichment in ZnS. (a) Transformation from [100]-cubic ZnS to [100]-tetragonal roquesite-type along 10-  
580 20 nm-wide stripes at the In-enriched boundary between sphalerite and pyrrhotite (Foil #2). (b) Preservation of  
581 the [100] cubic ZnS structure across boundaries of In-enrichment associated with quasi-regular ordering of  
582 brighter and larger atoms as a  $\sim 5\text{\AA} \times 5\text{\AA}$  pattern easy to visualize at the direct contact (dashed line) with ZnS.  
583 FFT pattern (inset) shows the lattice with quasi-ordering of brighter atoms within the ZnS substructure. (c)  
584 Image cropped from (a) showing the roquesite-type pattern with a dumbbell atomic arrangement on [100] as  
585 shown in STEM simulation in the inset. (d) FFT pattern corresponding to (c) with satellite reflections at  $\frac{1}{2}$   
586  $\langle 011 \rangle^*$  (yellow circles; T=tetrahedral). (e) Image cropped from (b) showing an area with more ordered  
587 distribution of brighter atoms attributable to columns of In-(Cu, Zn) mixed sites (yellow circles). (f) FFT pattern  
588 corresponding to image in (e) displaying satellite reflections (yellow circles) along  $b^*$  and  $c^*$  forbidden in the  
589 space group  $F\bar{4}3m$  of  $\text{ZnS}_{3C}$  (arrowed). The pattern and FFT are distinct from the roquesite-type structure  
590 (Supplemental<sup>1</sup> Fig. S4) and correspond to ZnS with  $P\bar{4}3m$  symmetry (see Figure 12 for interpretation).

591 **Figure 11.** Plot of (Fe+Cu+Ag+In+Sn+Ga) a.p.f.u. vs. Zn a.p.f.u. for In-rich ZnS from Dulong and other  
592 localities. <sup>1</sup>Capillitas Mine, Argentina (Marquez-Zavalia et al. 2014); <sup>2</sup>Ikuno Mine, Japan) (Kissin and Owens  
593 1986); <sup>3</sup>Toyoha, Japan (Ciobanu et al. 2011).

594 **Figure 12.** (a-c) From left to right, crystal model, SAED and STEM simulations for In-rich ZnS with  $P\bar{4}3m$   
595 symmetry and empirical formula  $(\text{Cu,In,Zn})_3(\text{Zn}_{0.5}\text{Fe}_{0.5})\text{S}_4$  modelled as a phase with two mixed sites: Me1a (In,  
596 Cu, and Zn); and Me3c (Fe, Zn). (d, e) Image and total intensity profile of In-rich ZnS along directions showing  
597 variation in signal intensity according to the considered model for In-rich ZnS. (f, g) Sphalerite image and total  
598 intensity profiles along two of the three directions shown for In-rich ZnS. Note the uniform signal intensity for  
599 Zn sites on both profiles.

**Table 1.** Overview of selected published data for phases in the system ZnS-CuInS<sub>2</sub> and related species

Mineral	Locality	Formula	System	Space group	a (Å)	b (Å)	c (Å)	Reference
<i>Sphalerite Group</i>								
Sphalerite	Jachymov, Czech Rep.	Zn <sub>0.9</sub> Fe <sub>0.11</sub> S <sub>0.99</sub>	Cubic	$F\bar{4}3m$	5.4174	5.4174	5.4174	Ondruš et al. (2003)
Sphalerite	Iwodake, Japan	Zn <sub>0.968</sub> Fe <sub>0.023</sub> Cu <sub>0.005</sub> In <sub>0.002</sub> Cd <sub>0.002</sub> ) <sub>1</sub> S	Cubic	$F\bar{4}3m$	5.4159	5.4159	5.4159	Nitta et al. (2008)
Ishiharaite	Capillitas, Argentina	(Cu,Ga,Fe,In,Zn)S	Cubic	$F\bar{4}3m$	5.368	5.368	5.368	Márquez-Zavalía et al. (2014)
<i>Related to sphalerite group</i>								
Sakuraiite	Ikuno, Japan	(Cu,Zn,Fe) <sub>3</sub> (In,Sn)S <sub>4</sub>	Tetragonal	(not given)	5.455	5.455	10.91	Kato (1965)
Sakuraiite	Japan	(Cu,Zn,Fe) <sub>3</sub> (In,Sn)S <sub>4</sub>	Tetragonal	$P\bar{4}2m$	5.45	5.45	5.4691	Momma et al. (2017)
Sakuraiite	Ikuno, Japan	(Cu,Zn,Fe,In,Sn)S	Cubic	3 alternatives*	5.4563	5.4563	5.4563	Kissin and Owens (1986)
<i>Chalcopyrite Group</i>								
Chalcopyrite		CuFeS <sub>2</sub>	Tetragonal	$I\bar{4}2d$	5.289	5.289	10.423	Hall and Stewart (1973)
Roquesite	Synthetic	CuInS <sub>2</sub>	Tetragonal	$I\bar{4}2d$	5.52279	5.52279	11.13295	Abrahams and Bernstein (1973)
Laforetite	Montgros, France	AgInS <sub>2</sub>	Tetragonal	$I\bar{4}2d$	5.88	5.88	11.21	Meisser et al. (1999)
<i>Stannite Group</i>								
Stannite	Oruro, Bolivia	Cu <sub>2</sub> (Fe,Zn)SnS <sub>4</sub>	Tetragonal	$I\bar{4}2m$	5.449	5.449	10.757	Hall et al. (1978)
Kësterite	Oruro, Bolivia	Cu <sub>2</sub> (Zn,Fe)SnS <sub>4</sub>	Tetragonal	$I\bar{4}$	5.427	5.427	10.8751	Hall et al. (1978)
Kësterite	Synthetic	Cu <sub>2</sub> ZnSnS <sub>4</sub>	Tetragonal	$I\bar{4}2m$	5.434	5.434	10.856	Bonazzi et al. (2003)
Petrukite	Multiple localities	(Cu,Fe,Zn) <sub>3</sub> (Sn,In)S <sub>4</sub>	Orthorhombic	$Pmn2_1$	7.6671	6.4399	6.2605	Kissin and Owens (1989)
Cubic Cu <sub>2</sub> FeSnS <sub>4</sub>	Synthetic	Cu <sub>2</sub> FeSnS <sub>4</sub>	Cubic	$F\bar{4}3m$	5.4179	5.4179	5.4179	Evstigneeva and Kabalov (2001)

\* Three possible space groups considered:  $P432$ ,  $P\bar{4}3m$  and  $Pm\bar{3}m$

**Table 2.** STEM EDS compositional data for sphalerite grains in studied foils

	*In wt.% (measured)			**Mol.% CuInS <sub>2</sub> (calculated)		
	Mean	Min.	Max.	Mean	Min.	Max.
<i>Foil #1</i>						
Sphalerite (n=11)	0.92	0.58	1.59	1.6	1	2.7
In-rich sphalerite (n=15)	17.95	10.41	24.85	36.5	19.1	48.8
<i>Foil #2</i>						
Sphalerite (n=8)	0.67	0.38	1	1.2	0.7	1.7
In-rich sphalerite (n=10)	14.58	9.17	20.77	29.9	17.2	45.5
<i>Foil #3</i>						
Sphalerite (n=3)	1	0.66	1.54	1.8	1.1	2.8
In-rich sphalerite (n=5)	20.26	18.04	21.09	42.9	41.9	45.3
<i>Foil #5</i>						
Sphalerite (n=10)	0.39	0.24	0.63	0.7	0.4	1.1

\*Measured Cu contains excess Cu from TEM grid thus measured In, Zn, S etc, are lower.

\*\*Calculated assuming Cu=In (atom.% basis) and substitution 2(Zn,Fe) ↔ Cu + In, ignoring <1.0 wt.% concentrations of Ag and Sn in some analyses.

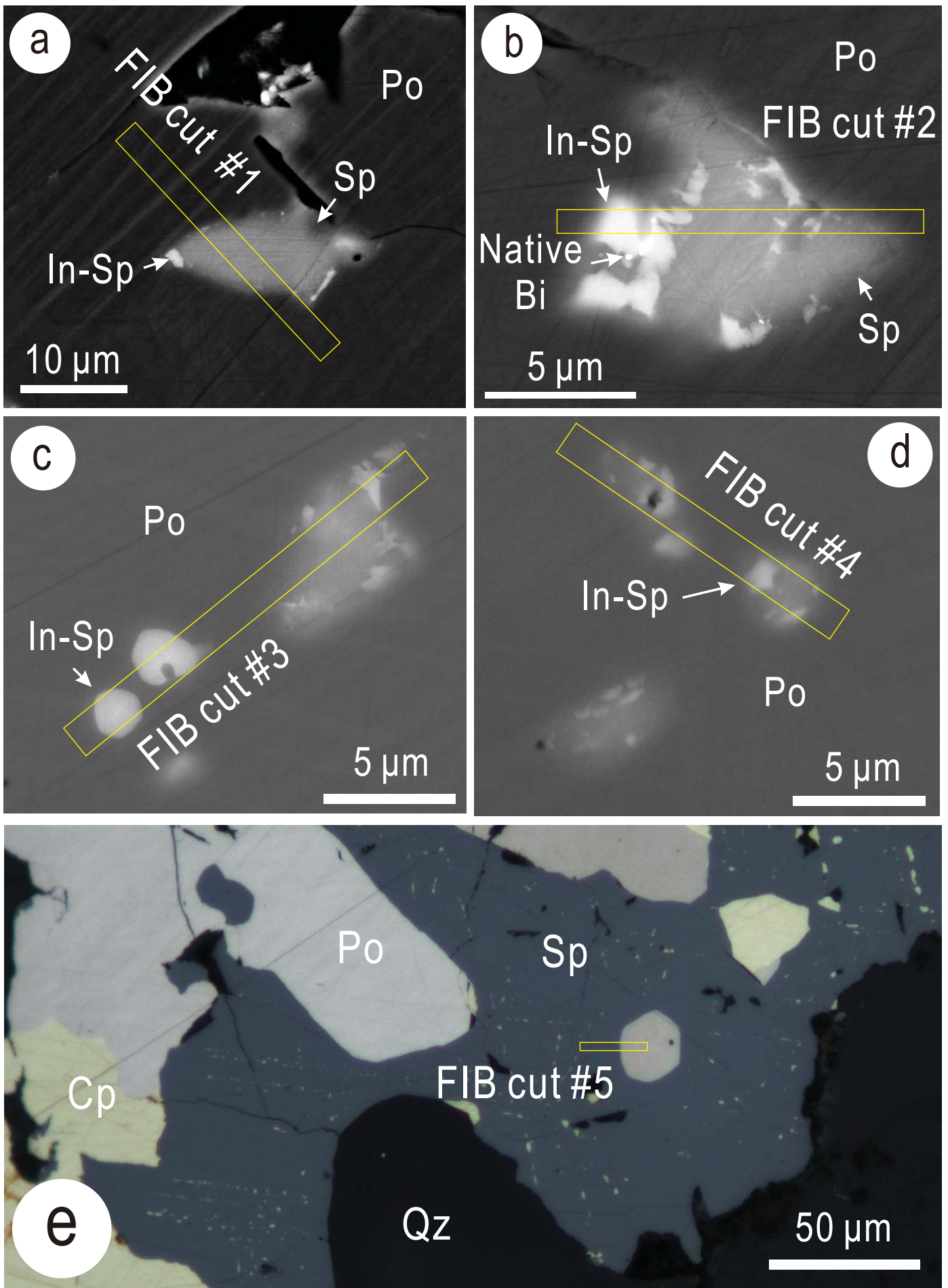


Figure 1. Xu et al.

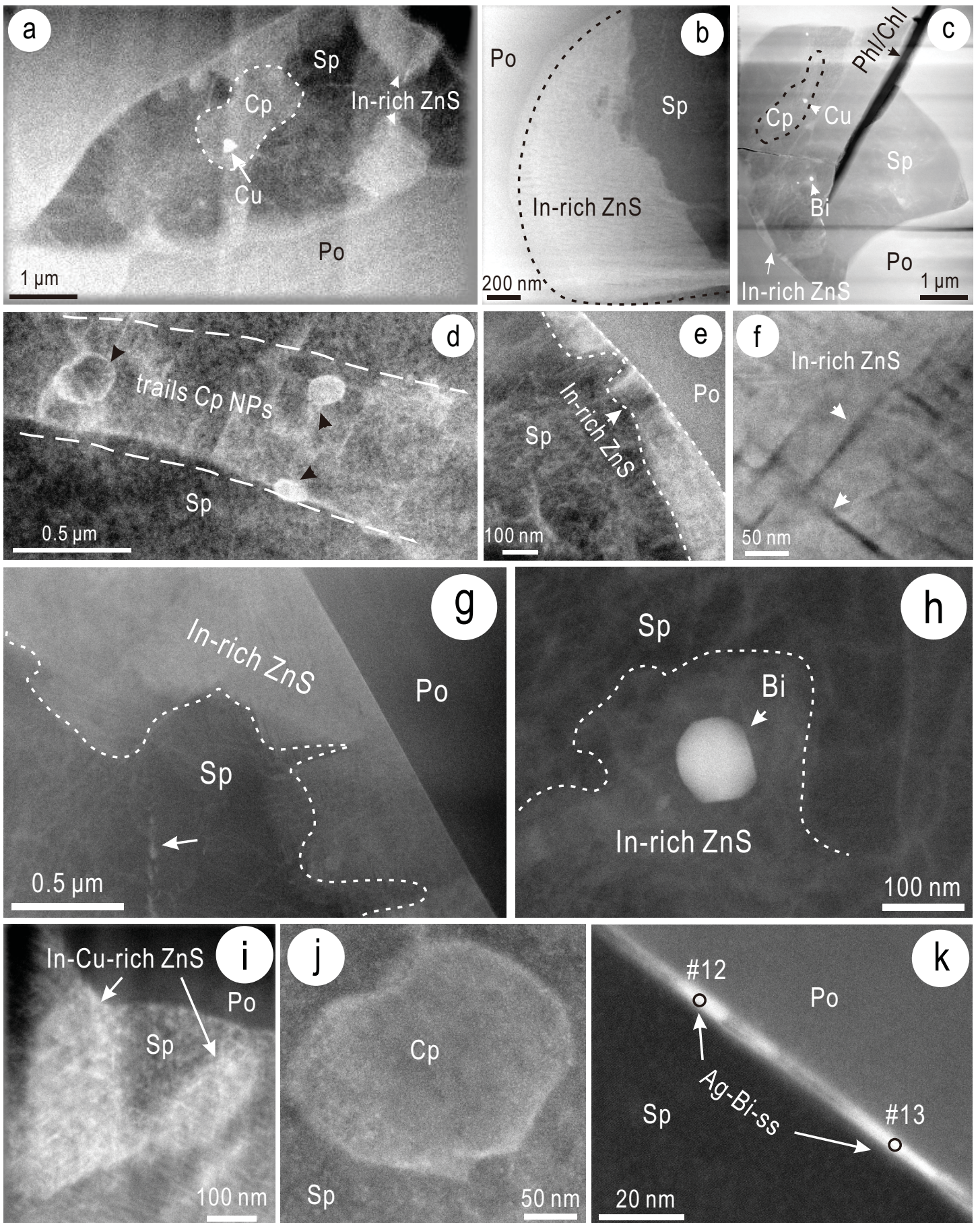


Figure 2. Xu et al.

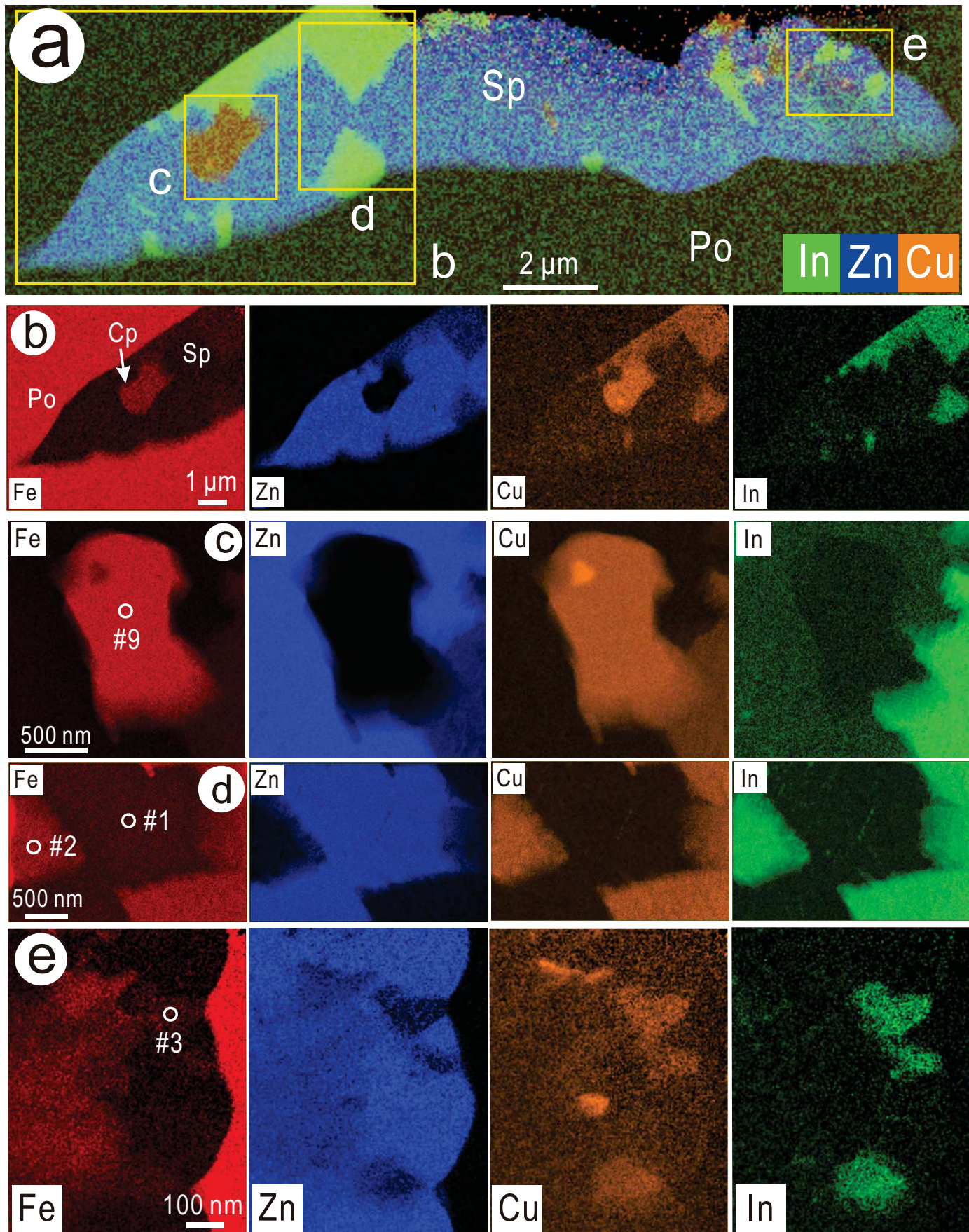


Figure 3. Xu et al.

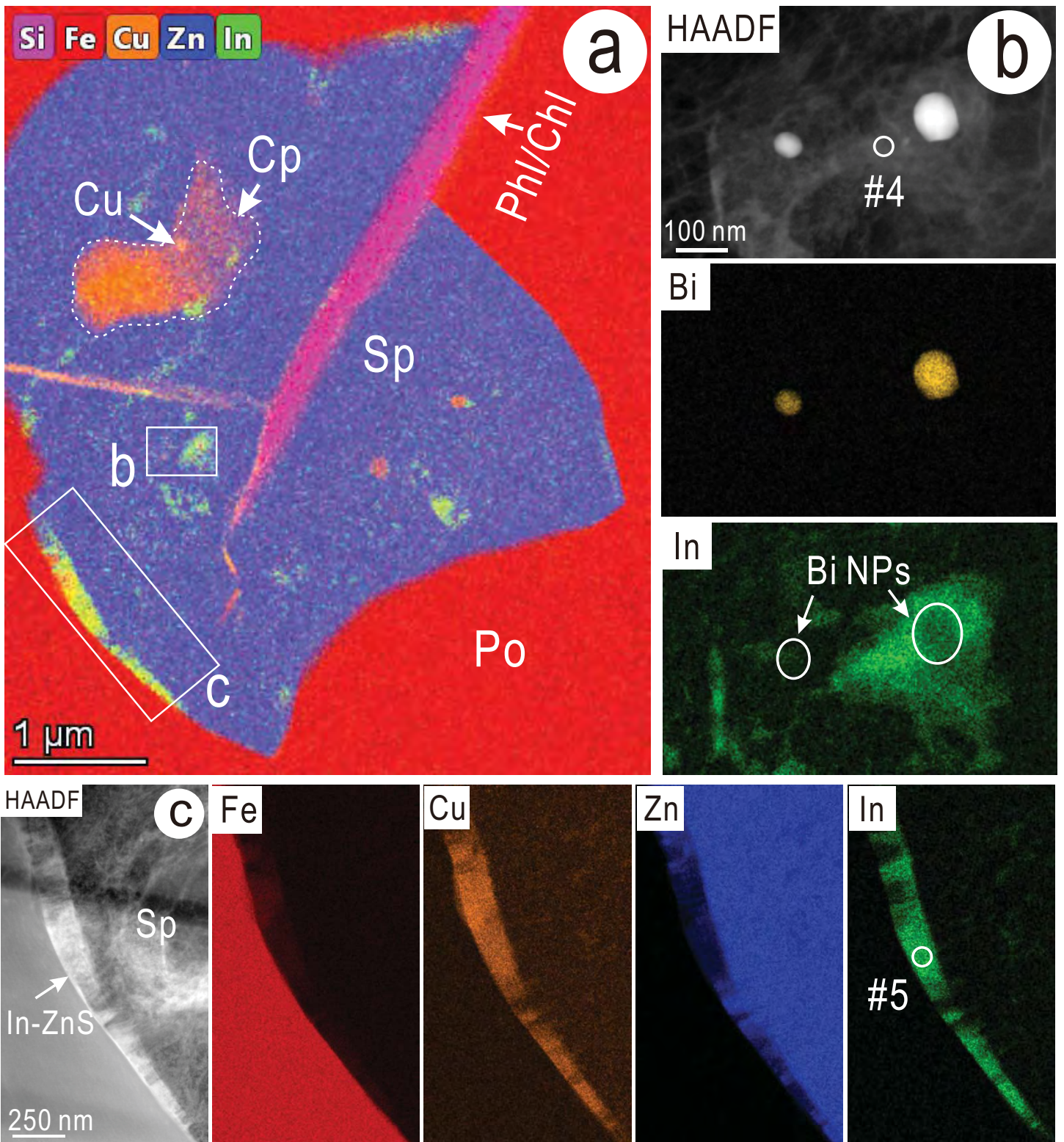


Figure 4. Xu et al.

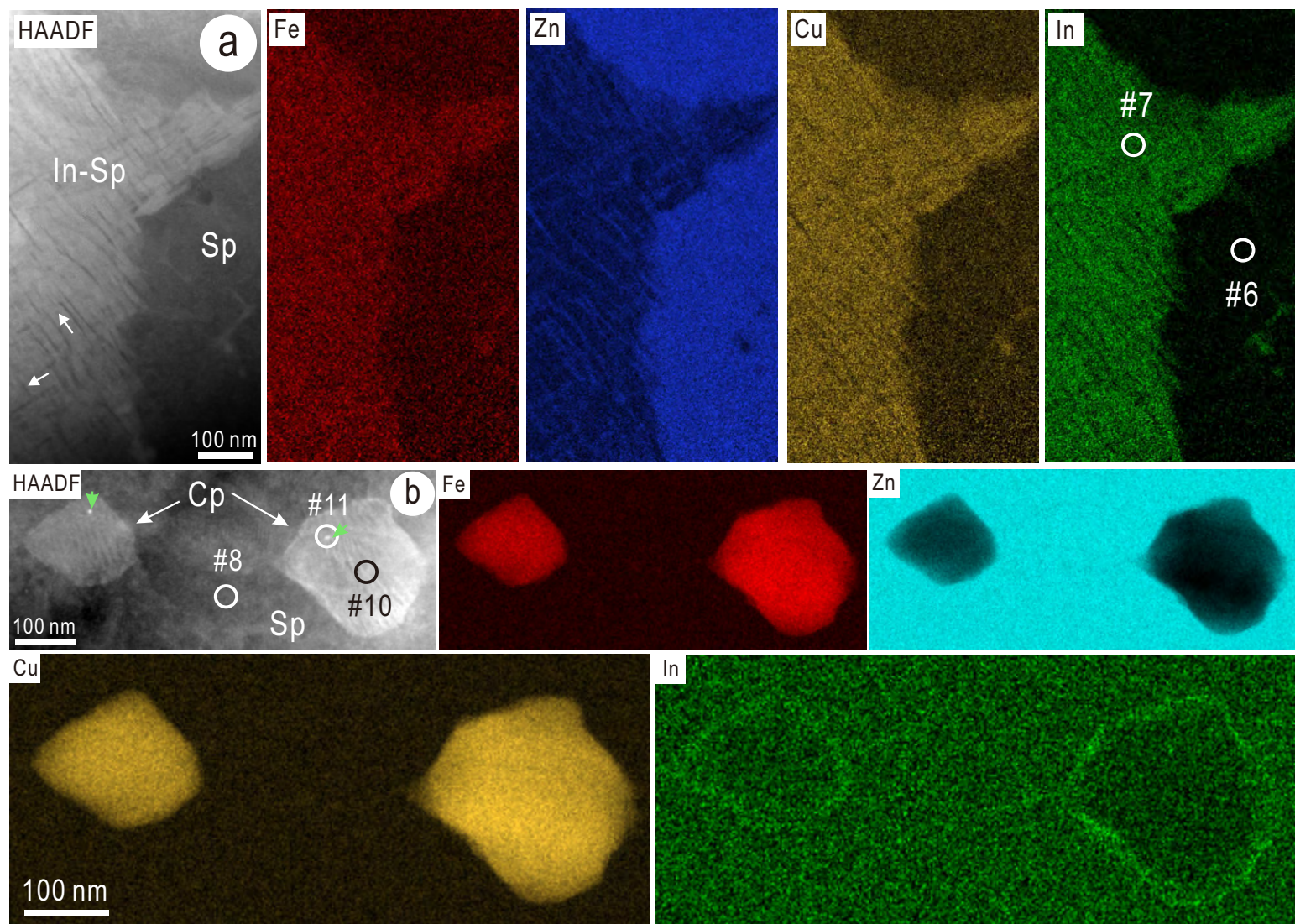


Figure 5. Xu et al.



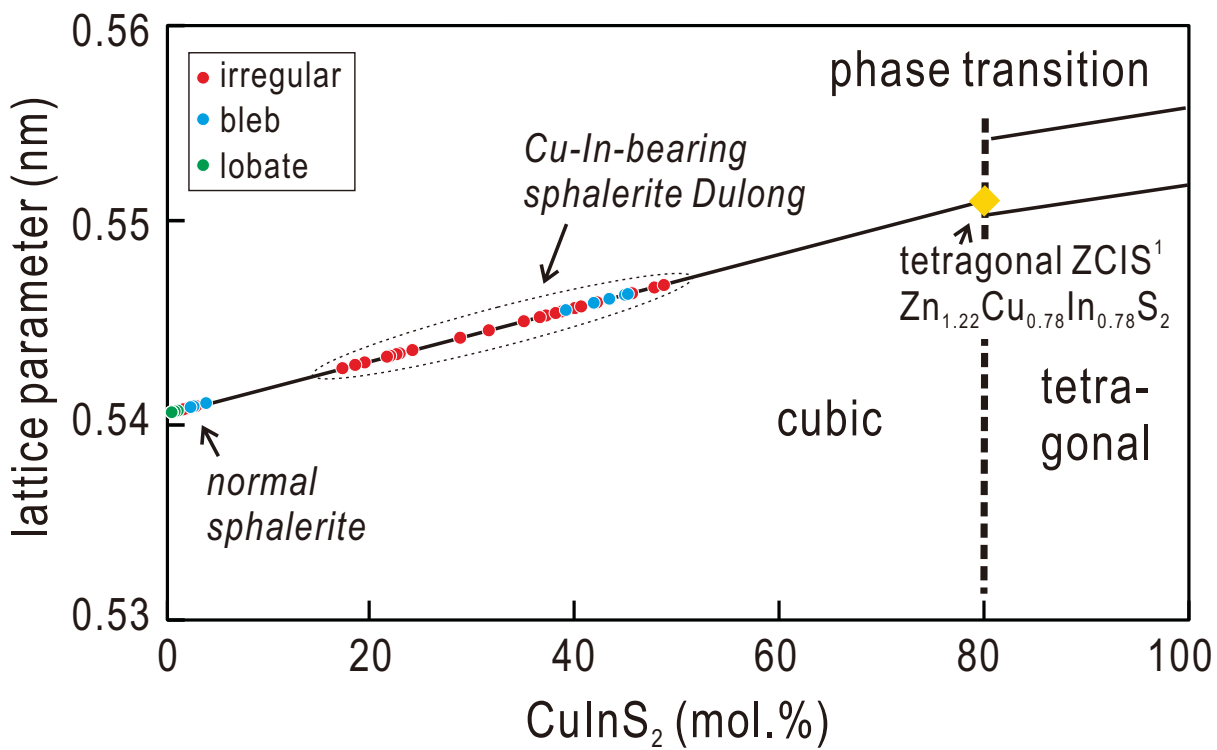


Figure 6. Xu et al.

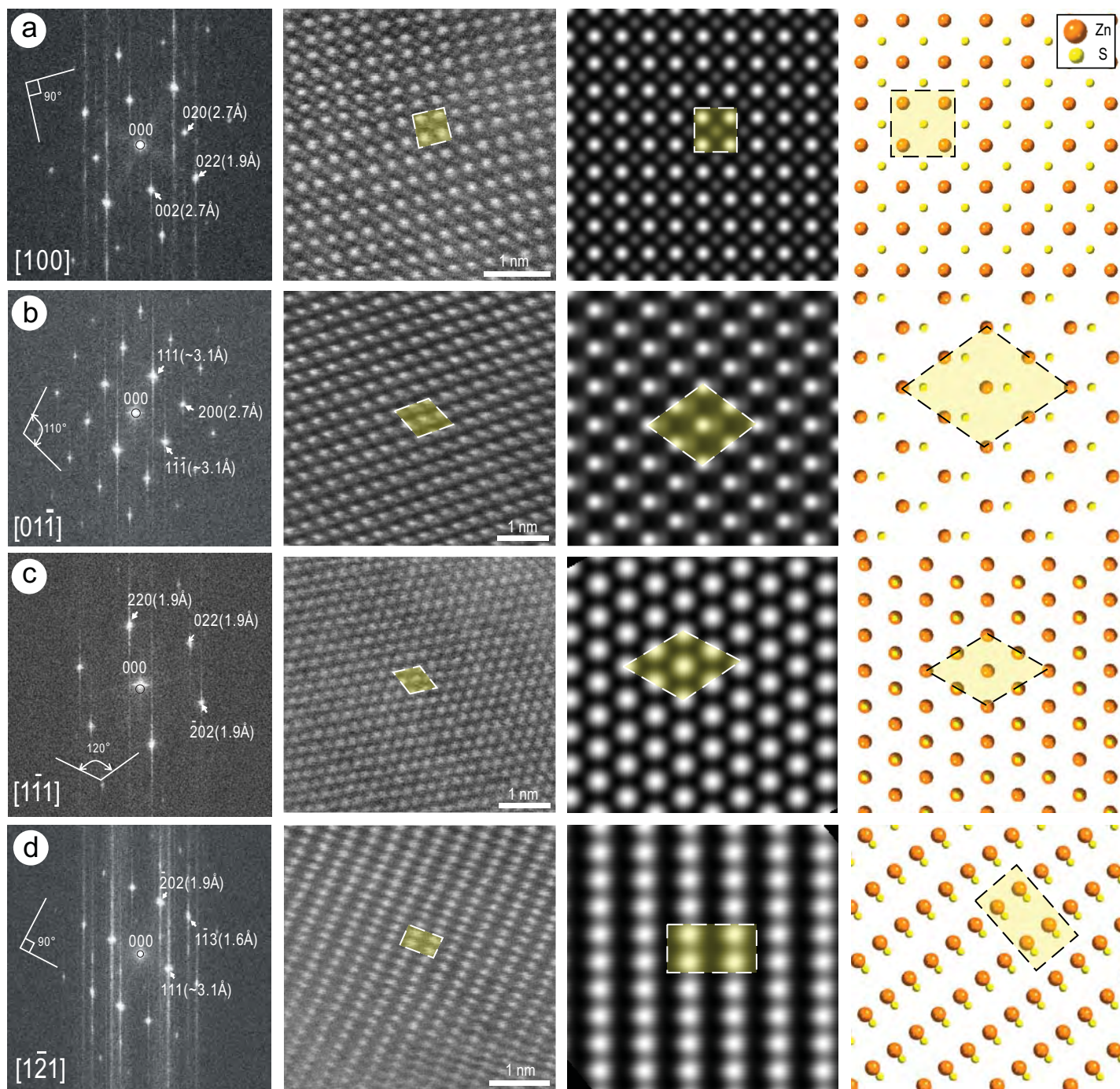


Figure 7. Xu et al.

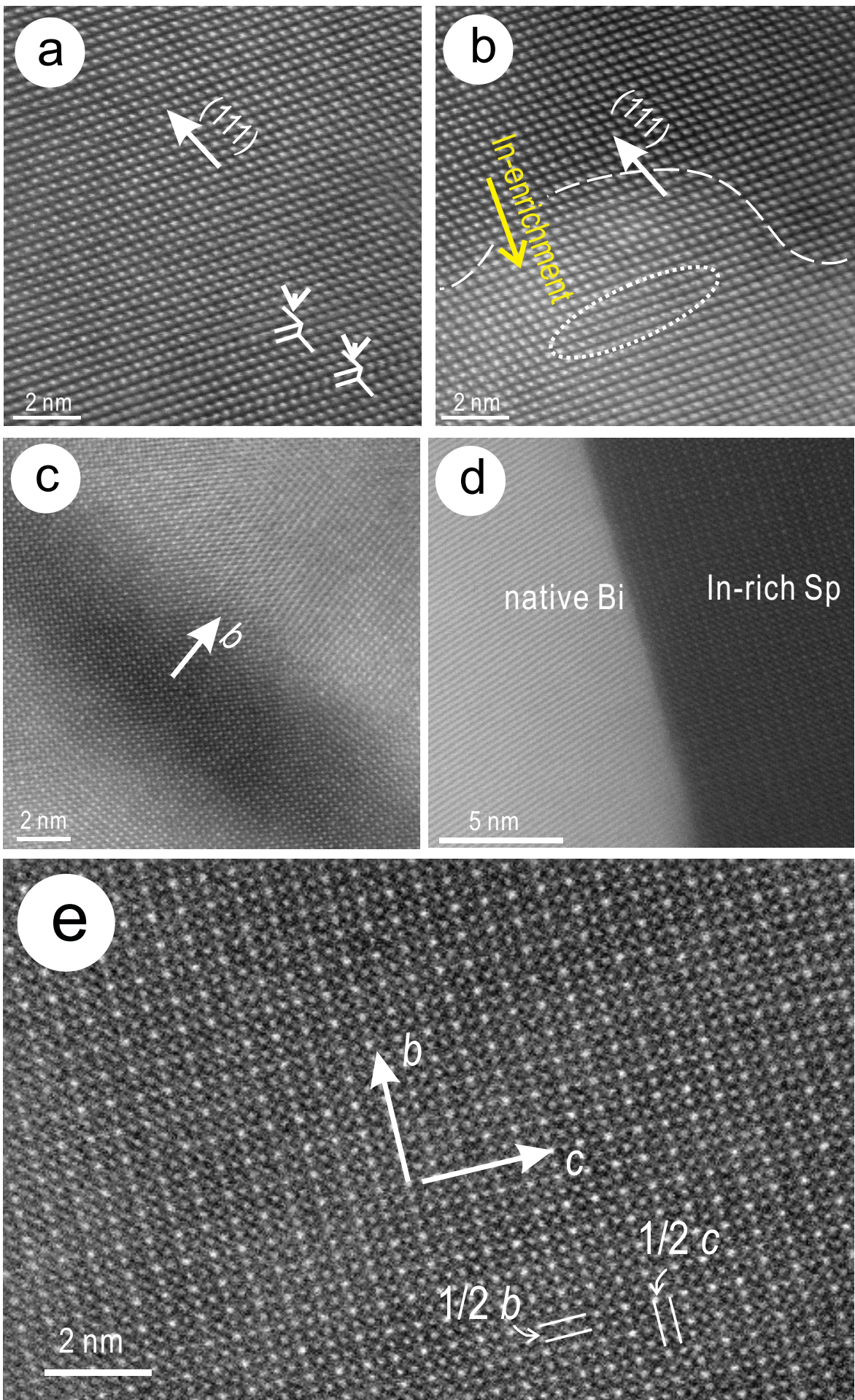


Figure 8. Xu et al.

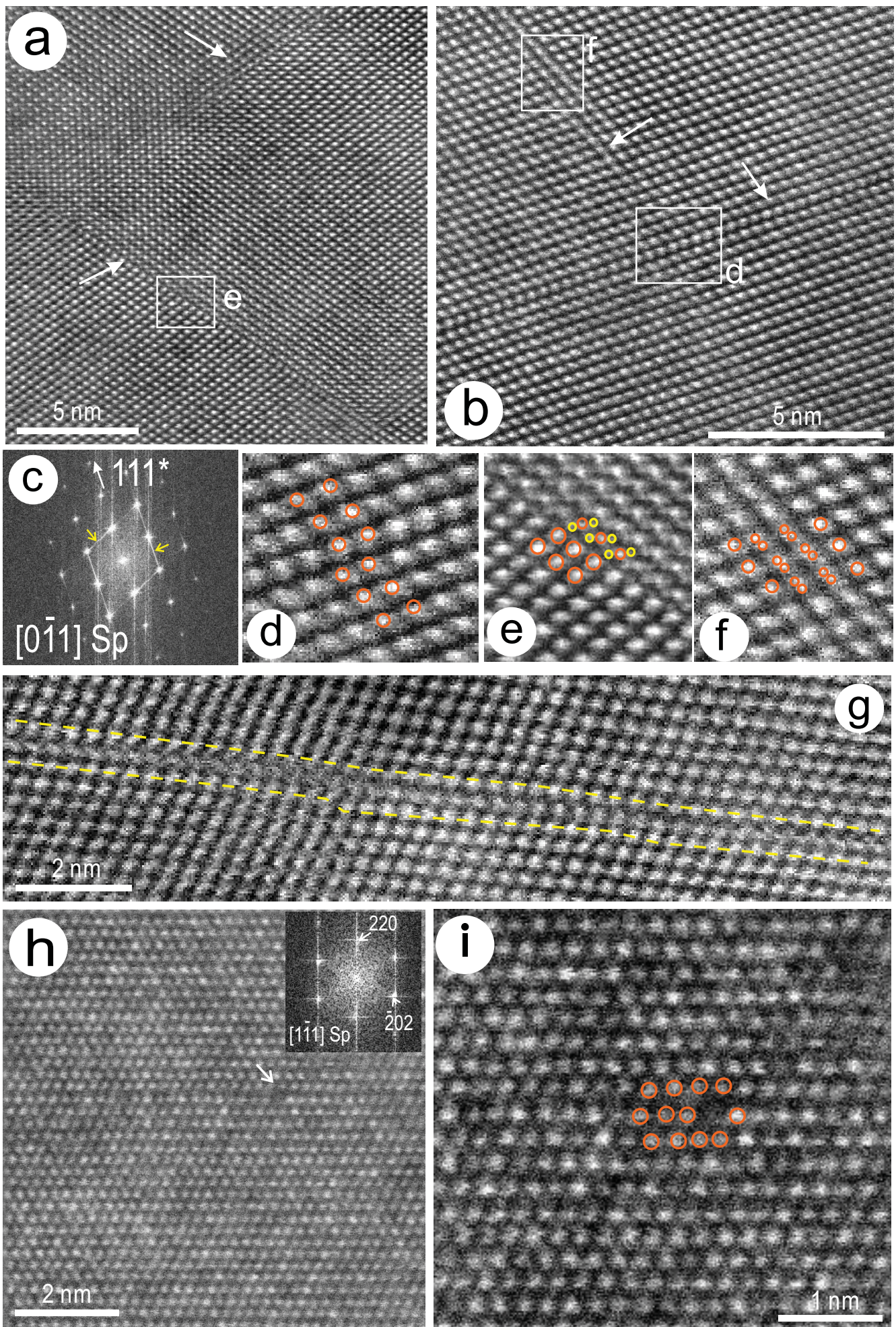


Figure 9. Xu et al.

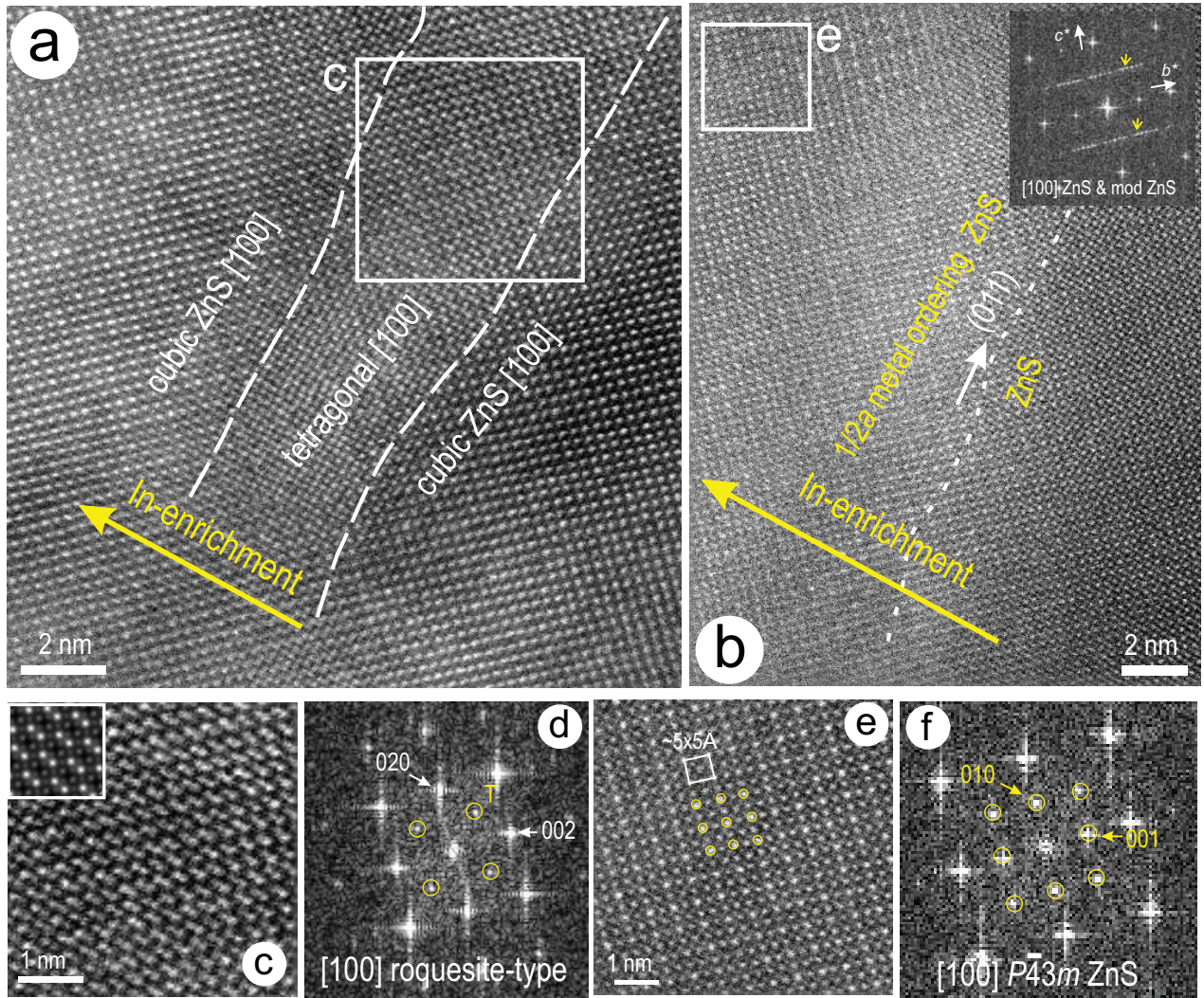


Figure 10. Xu et al.

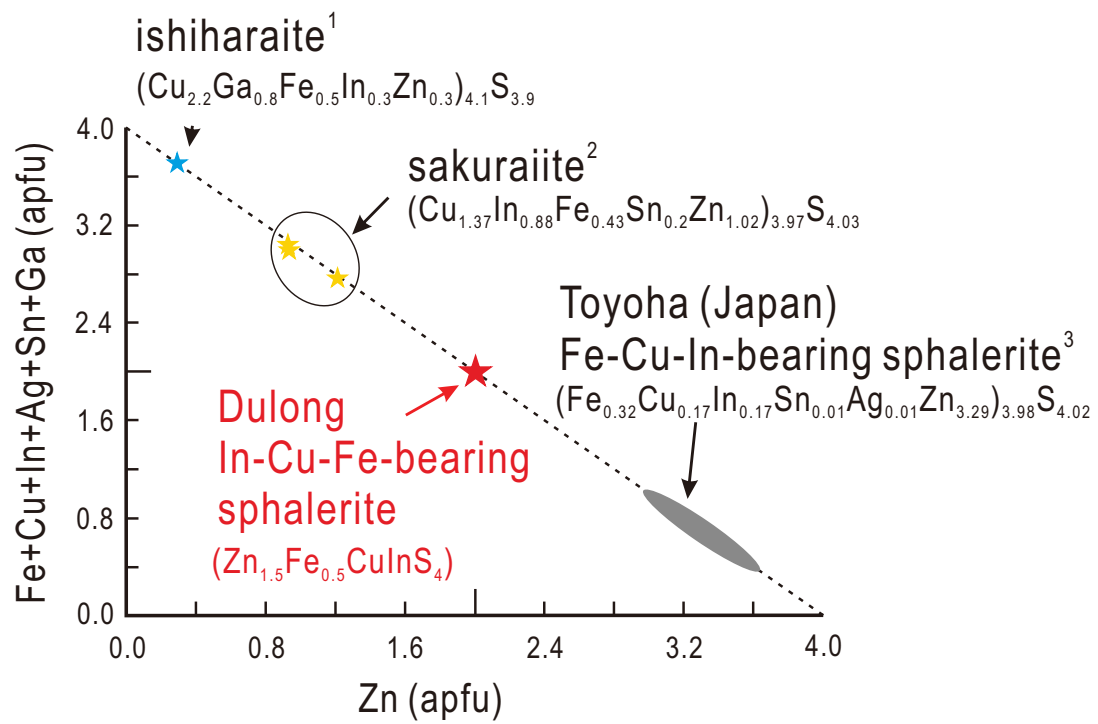


Figure 11. Xu et al.

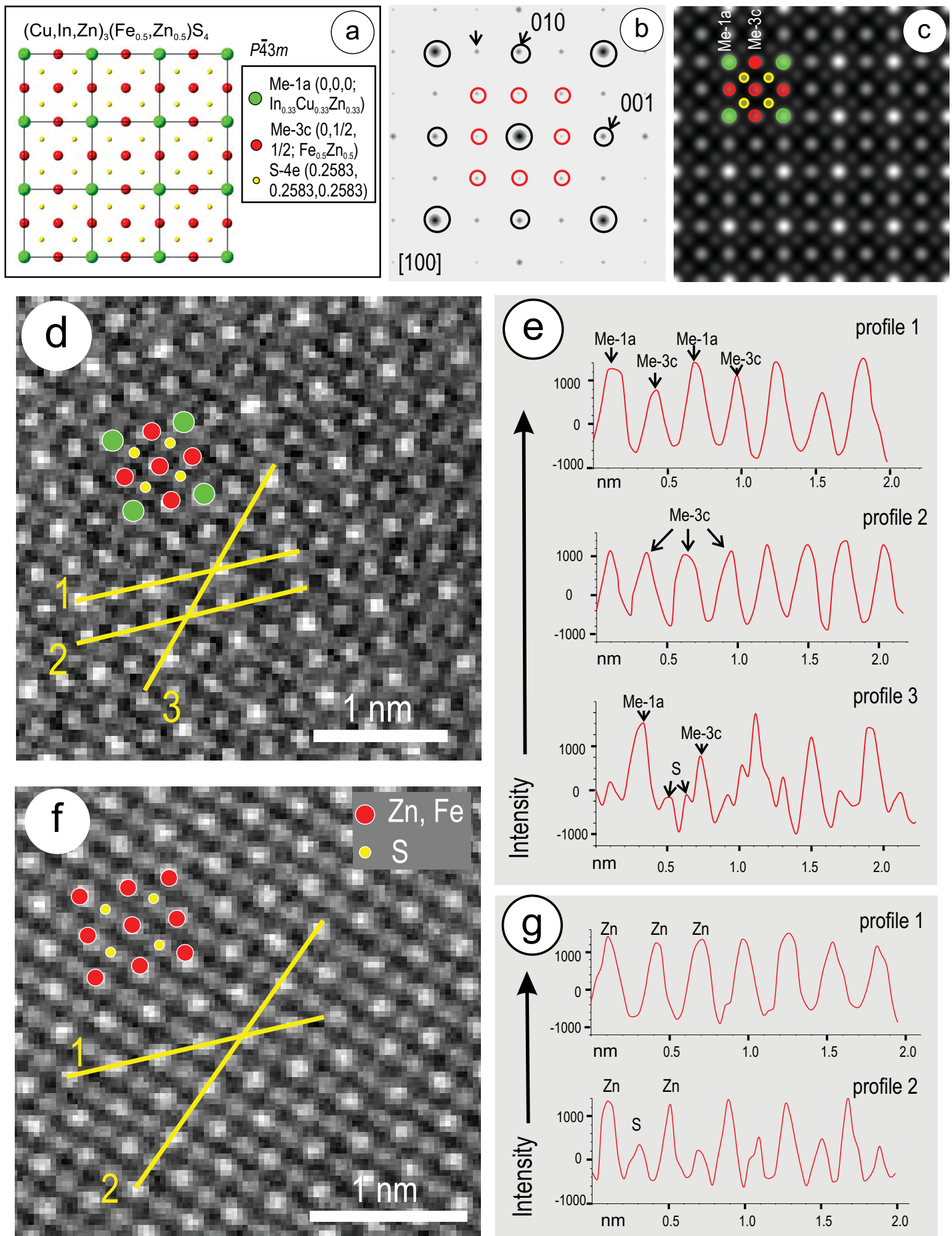


Figure 12. Xu et al.