Effects of composition and pressure on

electronic states of iron in bridgmanite

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Abstract

- Electronic states of iron in the lower mantle's dominant mineral, (Mg,Fe,Al)(Fe,Al,Si)O₃
- 26 bridgmanite, control physical properties of the mantle including density, elasticity, and electrical
- and thermal conductivity. However, determination of electronic states of iron has been
- controversial, in part due to different interpretations of Mössbauer spectroscopy results used to
- 29 identify spin state, valence state, and site occupancy of iron. We applied energy-domain
- 30 Mössbauer spectroscopy to a set of four bridgmanite samples spanning a wide range of
- compositions: 10-50% Fe/total cations, 0-25% Al/total cations, 12-100% Fe³⁺/total Fe.
- 32 Measurements performed in the diamond anvil cell at pressures up to 76 GPa below and above
- the high to low spin transition in Fe³⁺ provide a Mössbauer reference library for bridgmanite and
- demonstrate the effects of pressure and composition on electronic states of iron. Results indicate
- that although the spin transition in Fe³⁺ in the bridgmanite B-site occurs as predicted, it does not
- 36 strongly affect the observed quadrupole splitting of 1.4 mm/s, and only decreases center shift for
- 37 this site to 0 mm/s at \sim 70 GPa. Thus center shift can easily distinguish Fe³⁺ from Fe²⁺ at high
- pressure, which exhibits two distinct Mössbauer sites with center shift ~1 mm/s and quadrupole
- splitting 2.4-3.1 and 3.9 mm/s at \sim 70 GPa. Correct quantification of Fe³⁺/total Fe in bridgmanite

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- is required to constrain effects of composition and redox states in experimental measurements of seismic properties of bridgmanite. In Fe-rich, mixed-valence bridgmanite at deep-mantle-relevant pressures, up to ~20% of the Fe may be a Fe^{2.5+} charge transfer component, which should enhance electrical and thermal conductivity in Fe-rich heterogeneities at the base of
- Keywords: bridgmanite, Mössbauer spectroscopy, iron oxidation state, lower mantle

Introduction

Earth's mantle.

Iron-bearing bridgmanite, the most abundant material in the Earth's interior, is the dominant mineral responsible for chemical and physical behavior of the lower mantle. Redox of iron in bridgmanite buffers the mantle and thus has implications for formation of the atmosphere and habitability of the planet, storage and transport of volatiles including water and carbon in the deep interior, and interpretation of structures and dynamic processes observed by seismic tomography (e.g. Frost and McCammon 2008; Gu et al. 2016; Liu et al. 2018). Heterogeneous structures at the base of the mantle including large low shear velocity provinces and ultra-low velocity zones have been suggested to be iron-rich relative to surrounding mantle based on inversion of normal mode data (Ishii and Tromp 1999), tidal tomography (Lau et al. 2017), and travel times of seismic waves reflected off the core-mantle boundary (Rost et al. 2005); identifying these features and their role in the differentiation and mixing of the mantle depends on accurate constraints on the physical properties of iron-rich bridgmanite. Because bridgmanite samples from the lower mantle have not been recovered, redox conditions are inferred based on

60 remote geophysical observations and experimental and theoretical modeling of chemistry at 61 extreme pressures and temperatures. 62 Depth and composition both affect redox states of iron incorporated in the mantle's dominant 63 phase, but efforts to measure these changes have been complicated by the crystal chemistry of 64 (Mg,Fe,Al)(Fe,Al,Si)O₃ bridgmanite (e.g. McCammon et al. 2013; Shim et al. 2017). 65 Bridgmanite adopts the orthorhombic GdFeO₃-type perovskite structure, with a larger 8-12-fold pseudo-dodecahedral "A" site and smaller 6-fold octahedral "B" site. Both Fe²⁺ and Fe³⁺ 66 substitute for Mg in the A-site, while only the smaller Fe³⁺ ion may substitute for Si in the B-site. 67 68 At pressure/temperature conditions corresponding to the top of the lower mantle, bridgmanite 69 typically exhibits Fe³⁺/total Fe ratio at least 50%, even under reducing conditions in contact with metallic iron (Frost et al. 2004; Shim et al. 2017). The Fe³⁺/total Fe ratio in bridgmanite 70 synthesized at pressures corresponding to the deep lower mantle has been observed to decrease 71 72 with pressure to \sim 15%, then increase again to \sim 50% (Shim et al. 2017). Variations in valence 73 states of iron with depth are inferred to be due to energetics of substitution mechanisms as local 74 structure and spin state of iron change with depth. Distortion of the perovskite structure due to 75 pressure and/or composition divides the A-site into A1- and A2-sites (Hsu et al. 2010; Hummer and Fei 2012) and occurs at similar conditions as the observed decrease in Fe³⁺/total Fe. A spin-76 pairing transition in Fe³⁺ in the B-site at ~48 GPa, corresponding to the mid-lower-mantle, may 77 also drive partitioning of Fe³⁺ into the B-site (Badro et al. 2004; Catalli et al. 2010, 2011; Hsu et 78 al. 2011; Liu et al. 2018). The presence of Al may also modify incorporation of Fe³⁺ in 79 80 bridgmanite through paired substitution of trivalent cations for Mg and Si (e.g. Frost and 81 Langenhorst 2002; Piet et al. 2016) and formation of vacancies in cation (Sinmyo et al. 2014;

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Kupenko et al. 2015) or oxygen sites (Grüninger et al. 2019; Liu et al. 2019). Based on experiments (e.g. Potapkin et al. 2013) and computational modeling using density functional theory (DFT) (Caracas 2010), paired substitution places Al in the B-site and Fe³⁺ in the A-site at equilibrium, though some observations support migration of Al and Fe³⁺ between the two sites (Catalli et al. 2011) and/or migration of Fe³⁺ from A to B site due to vacancy redistribution (Kupenko et al. 2015). In bridgmanite compositions with mixed valence states, Fe²⁺–Fe³⁺ electronic charge transfer may also create Fe^{2.5+} states (Mattson and Rossman 1987, Burns 1993). Because the separation distance between A and B sites is smaller than between A and adjacent A sites, the activation energy related to the distance of electron migration was suggested to be lower for A-B Fe²⁺–Fe³⁺ charge transfer (Fei et al. 1994). For the same reason, charge transfer is generally expected to be promoted by pressure and impeded by temperature (Mattson and Rossman 1987; Fei et al. 1994; Xu and McCammon 2002; Lobanov et al. 2017). In total, Fe in bridgmanite in Earth's mantle may be distributed among a multitude of different combinations of valence state (Fe²⁺, Fe^{2.5+}, Fe³⁺), crystallographic site (A1, A2, or B), and spin state (high or low). Controversy over the electronic states of iron in bridgmanite has also been drawn out by different interpretations of a key observational technique, Mössbauer spectroscopy. The Mössbauer effect allows each valence state, coordination environment, and spin state to be distinguished by characteristic values for center shift (CS) and quadrupole splitting (OS) (Dyar et al. 2006). Interpretation of CS and QS obtained at high pressures requires deconvolving poorly-constrained effects of structural and electronic changes. Pressure generally distorts crystallographic sites of bridgmanite, leading to higher OS with pressure (e.g. Lin et al. 2013; McCammon et al. 2013).

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Although the existence of the spin transition in Fe³⁺ in the bridgmanite B-site is now wellaccepted (Lin et al. 2013; Badro 2014), the published values for Mössbauer parameters of lowspin Fe³⁺ in bridgmanite range from CS~1 mm/s and OS ~3 mm/s similar to high-spin Fe²⁺ (Catalli et al. 2010; Gu et al. 2012) to CS<0.5 mm/s and QS ~1-1.5 mm/s similar to high-spin Fe³⁺ (Jackson et al. 2005; Kupenko et al. 2015; Liu et al. 2018). Predictions by DFT support higher QS values 2-3 mm/s due to asymmetrical electronic field gradient generated by spin-down electrons in the B-site, but have not addressed CS of low-spin Fe³⁺ (Hsu et al. 2011). Depending on which of these interpretations is correct, low-spin Fe³⁺ may be misidentified as high-spin Fe²⁺ or high-spin Fe³⁺, or vice versa. The conditions at which the spin transition is reported based on these observations range from 13-24 GPa (Mao et al. 2015) to 50-70 GPa (Catalli et al. 2010, 2011). Other phenomena suggested to impact Mössbauer spectra of bridgmanite and related compounds at high pressures include charge-transfer, which should result in enhanced electrical and thermal conductivity in silicates/perovskites with a mixture of Fe²⁺ and Fe³⁺ (Fei et al. 1994: Xu and McCammon 2002; Keppler et al. 2008; Long et al. 2009), and changes in glasses in recoil-free fraction of Fe²⁺ and Fe³⁺, causing shifts in the relative contributions of Fe²⁺ and Fe³⁺ to spectral area (Prescher et al. 2014). In addition, spectra for complex materials such as bridgmanite suffer from overlapping peaks and non-unique fitting. These issues may be addressed by complementary techniques such as X-ray emission spectroscopy and X-ray diffraction to independently constrain spin state and structure (e.g. Catalli et al. 2010) and systematic analysis of a range of compositions (e.g. McCammon et al. 2013). Previous attempts to prepare well-characterized bridgmanite rely on samples synthesized in the multianvil press, which is typically limited to ~25 GPa. At these conditions, bridgmanite is stable

with only up to ~15-20% Fe (Fei et al. 1996; Tange et al. 2009), and typically exhibits a mixture of Fe²⁺ and Fe³⁺ depending on composition and f_{O2} (e.g. Frost and Langenhorst 2002; Frost et al. 2004). At pressures ~75-100 GPa in the diamond anvil cell, a wider range of compositions with up to at least 75-90% FeSiO₃ and lower Fe³⁺/total Fe becomes stable (Tateno et al. 2007; Dorfman et al. 2012, 2013; Ismailova et al. 2016). The goal of this study is to systematically analyze electronic states of iron-rich bridgmanite using Mössbauer spectroscopy of well-characterized end-member samples. These experiments resolve discrepancies in interpretation of high-pressure Mössbauer spectra of bridgmanite by firmly constraining valence states of iron at 1 bar, where Mössbauer parameters CS and QS are unambiguous. Results address the conditions and observable characteristics of possible spin transitions and charge transfer, and will be important to future characterization of bridgmanite relevant to Earth's deep mantle heterogeneities.

Methods

- Four bridgmanite compositions were examined ranging from 12-100% Fe³⁺/total Fe and from 10-50% Fe/total cations. Bridgmanites were synthesized from the following starting materials:
 - 50% enstatite, 50% ferrosilite (En50Fs50) pigeonite was synthesized at the Crystal Growth Facility at EPFL in a gas-mixing furnace. A stoichiometric mixture of Fe₂O₃ (enriched to 48.3% ⁵⁷Fe), MgO and SiO₂ was ground in an agate mortar until homogenous and cold-sintered in a 3-ton press at room temperature for 1 minute. The resulting pellet was placed in an alumina crucible on a bed of unenriched powder mixture of the same composition. The sample was first heated in air at 1000°C, then reduced

under an N_2/H_2 mixture bubbled through H_2O at 1100° - $1200^\circ C$ for 3 days, then rapidly quenched to room temperature. Mössbauer analysis of the starting material was consistent with complete reduction of ferric iron to ferrous iron, with a detection limit of a few percent. X-ray diffraction and electron microprobe analysis (Table 1) confirmed that the oxides transformed to monoclinic $P2_1/c$ pigeonite with composition ($Mg_{0.52}Fe_{0.48}^{2+}$)SiO₃ (0% ferric before bridgmanite synthesis).

- 90% enstatite, 10% hematite (En90Hem10) glass starting material was synthesized from intimately mixed Fe₂O₃ (enriched to 96.6% ⁵⁷Fe), MgO and SiO₂ powders by containerless laser-levitation (Weber et al. 1994) under compressed air at IPGP, Paris. Mössbauer spectroscopy indicates total ferric iron content of the glass before bridgmanite synthesis is ~30%. The homogeneity and composition of the glass were determined by electron microprobe analysis (Table 1), yielding formula for this sample as follows:
 Mg_{0.97}Fe²⁺_{0.14}Fe³⁺_{0.06}Si_{0.90}O₃. Oxygen deficiency in measured formula may be due to use of Fe²⁺-rich standards and few-% uncertainty indicated by 97-98% oxide totals during microprobe measurements.
- 50% enstatite, 50% hematite (En50Hem50) akimotoite synthesis and characterization were described in (Liu et al. 2018). Based on microprobe analysis and Mössbauer spectroscopy, the formula obtained for this sample is Mg_{0.46}Fe_{1.04}Si_{0.49}O₃ (100% ferric).
- 75% ferrosilite, 25% corundum (Fs75Co25, or almandine-composition) glass synthesis and characterization were described in Dorfman et al. (2016). The composition of the glass, normalized to 3 oxygens, is Fe²⁺_{0.55}Fe³⁺_{0.12}Al_{0.54}Si_{0.73}O₃ (20% ferric before bridgmanite synthesis).

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Starting materials were loaded as crystalline powder or glass chip in diamond anvil cells with 100-150-micrometer culet and 300-micrometer bevel. Each sample was contained in a 30-50micrometer hole drilled in a Re gasket. To insulate the samples during laser heating and transmit quasi-hydrostatic stress, En50Fs50 and Fs75Co25 were loaded sandwiched between platelets of NaCl; En90Hem10 was loaded in Ar gas at the EPFL; and En50Hem50 was loaded in Ne gas using the COMPRES/GSECARS gas loading system (Rivers et al. 2008). Pressure during experiments up to ~100 GPa was measured by Raman spectroscopy of the stressed diamond anvil culet center (Akahama and Kawamura 2006), or for En50Hem50 a ruby sphere loaded with the sample (Mao et al. 1986). Bridgmanite was synthesized from Fs75Co25, En50Fs50, and En90Hem10 compositions as in previous studies (Dorfman et al. 2012, 2013) by laser heating to 2000-2500 K for ~20-60 min after compression at 300 K directly to ~75-100 GPa. The transformation and homogeneity of each sample was confirmed by X-ray diffraction at beamlines ID27 and ID09 (λ=0.3738 Å and 0.4117 Å or 0.4155 Å, respectively) of the ESRF (Figure 1). Minor CaCl₂-type SiO₂ diffraction peaks are also observed in some samples after laser heating. As described by Liu et al. (2018), the En50Hem50 akimotoite starting material transforms reproducibly and reversibly to bridgmanite at 300 K and 22-26 GPa, so this sample was simply compressed at 300 K. Energy-domain Mössbauer spectroscopy was performed at the Nuclear Resonance beamline ID18 of the European Synchrotron Radiation Facility (Rüffer and Chumakov 1996; Potapkin et al. 2012) and the COMPRES/sector 3 offline Mössbauer spectroscopy laboratory at the Advanced Photon Source at Argonne National Laboratory. At ID18, incident light is synchrotron X-rays monochromatized by a ⁵⁷FeBO₃ single crystal to the ⁵⁷Fe resonant energy of 14.4 keV 9

with energy resolution of \sim 5.5 neV. The X-ray beam is focused to \sim 9x14 micrometers full width at half maximum. At the sector 3 offline Mössbauer laboratory, incident gamma-rays are provided by a 400-micrometer-diameter radioactive 57 Co point source. To generate a range of energies for absorption spectroscopy measurements, both the 57 FeBO₃ monochromator at ID18 and the 57 Co at sector 3 are oscillated +/- 5mm/s (1 bar measurement at sector 3 used +/- 10 mm/s). Source velocities are calibrated with an α -Fe foil standard at 300 K. Synchrotron source line width and center shift are calibrated with a K_2Mg^{57} Fe(CN)₆ standard before and after each measurement. The linewidth for the conventional measurements was set to match a 6-month-old radioactive point source. Synchrotron Mössbauer spectra were collected for \sim 2-8 hrs at each pressure. Conventional Mössbauer spectra were obtained for \sim 2 weeks each. Spectra were fit to pseudo Voigt doublets using MossA software (Prescher et al. 2012). All spectra were fit using a full transmission integral to account for thicknesses of radiation sources and samples.

Results and discussion

To ensure no change in valence states of Fe across the measured pressure range, all Mössbauer spectroscopy measurements were performed at 300 K and no additional heating was applied to samples after synthesis of bridgmanite. Upon laser heating, both structural transformation and redox may occur. Although some previous studies assume Fe³⁺/total Fe does not change during synthesis (e.g. Nishio-Hamane et al. 2007; Lundin et al. 2008; Dorfman et al. 2013), redox during laser heating is confirmed by our data to result in differences in Fe³⁺/total Fe in synthesized bridgmanite relative to starting materials. During 300 K compression and decompression, we assume that kinetics do not permit site-site diffusion and redox reactions. Therefore, Fe³⁺/total Fe for bridgmanites synthesized at high pressure may be fixed to values 10

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observed after 300 K decompression to 1 bar (these Fe³⁺/total Fe values are reported here in 213 214 figures and tables). Mössbauer spectra were measured for Fs75Co25, En50Fs50, and 215 En90Hem10 bridgmanites on decompression from ~75-100 GPa in ~25 GPa steps to 1 bar 216 (Figures 2-4 a-c). En50Hem50 akimotoite was observed upon compression from 1 bar to 217 bridgmanite at 32 and 70 GPa (Figures 2-4 d). Changes in the Mössbauer parameters of each 218 bridgmanite composition during 300 K compression/decompression (Table 2, Figure 5) may be 219 due to 1) electronic spin transitions, 2) pressure-induced structural changes including site 220 distortion or amorphization, or 3) charge transfer between nearby Fe ions. The spin transition in Fe³⁺ 221 Effects of the high-to-low spin transition on Mössbauer parameters of Fe³⁺ in bridgmanite may 222 223 be clearly observed in En50Hem50 bridgmanite, for which the spin transition in the octahedral 224 site has been documented at ~48 GPa at 300 K by X-ray diffraction and X-ray emission spectroscopy (Liu et al. 2018). En50Hem50 bridgmanite is 100% Fe³⁺, with 50% Fe in the 225 226 pseudo-dodecahedral site, and 50% Fe in the octahedral site. Mössbauer spectra at high pressure 227 exhibit two closely-overlapping doublets, with absorption at \sim -0.4-0.0 mm/s and \sim 0.6-1 mm/s 228 (Figure 3d, Figure 4d). Across the pressure-induced spin transition from 32 to 70 GPa, spectra 229 change slightly: average CS decreases by 0.15 mm/s and QS increases by 0.35 mm/s. Two reasonable fits to these data are possible for a sample with pure Fe³⁺: either Fe³⁺ in A- and B-230 231 sites adopt CS differing by <0.05 mm/s with contrasting QS, 0.6-0.9 and 1.4-1.8 mm/s, or the 232 difference in CS is larger, ~0.4 mm/s, while QS is identical within ~0.5 mm/s. Neither option is 233 consistent with a large change across the spin transition to QS ~3 mm/s predicted in bridgmanite 234 by DFT (Hsu et al. 2011). At 32 GPa (Figure 3d), the fit with contrasting CS would produce

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CS=0.45 mm/s for the higher CS component, which would be high for Fe³⁺ (Figure 5). We therefore favor a fit with contrasting QS for high-spin Fe³⁺ in A-site vs. B-site, consistent with the different asymmetry of the two sites in high-spin state. At 70 GPa (Figure 4d), the overall decrease in CS allows a fit with contrasting CS with acceptable Mössbauer parameters for Fe³⁺ for both sites, where the low-spin B-site Fe³⁺ exhibits lower CS and higher OS relative to highspin B-site Fe³⁺ at lower pressure. In addition, recent data for well-characterized samples indicate that pressure tends to increase QS (i.e. level of structural distortion around the iron ion increases), and the principal change in Mössbauer parameters of octahedrally-coordinated Fe³⁺ across the spin transition is a decrease in CS (i.e. density of electrons around the iron nucleus increases) (Pasternak et al. 2002; Kupenko et al. 2015; Liu et al. 2018) (Table 3). We thus obtain CS ~0.35 mm/s for high-spin Fe³⁺ in both sites at 32 GPa below the spin transition, and a lower CS -0.07 mm/s for low-spin Fe³⁺ at 70 GPa (Table 2, Figure 5). These results can be used to determine whether spin transitions occur in other compositions with mixed valence states of iron. For less Fe³⁺-rich En50Fs50 and En90Hem10 compositions, Fe³⁺ doublets can be interpreted based on values observed at similar conditions for En50Hem50. The fit uncertainty for Fe³⁺ doublets in En50Fs50 at 75 GPa is large (as much as ~1 mm/s) due to overlap with other doublets, but the refined value 0.4 mm/s is consistent with high-spin Fe³⁺ in the A-site, as expected with a (Mg+Fe)/Si ratio ~1 and B-site filled with Si. For En90Hem10, with ~50% Fe³⁺, stronger, better-resolved Fe³⁺ exhibits a slight decrease in CS, from 0.34 mm/s to 0.29 mm/s at 36 and 67 GPa, respectively. While this CS indicates a dominantly HS Fe³⁺ component, based on an assumption of linear mixing of CS values as much as ~20% of the Fe³⁺ (~10% total Fe) may be low-spin Fe³⁺ in the B-site at 67 GPa (Figure 6a). This partial spin

transition is consistent with the more Si-depleted composition with space available for Fe³⁺ in the 257 258 B-site. 259 Site distortion All Fe²⁺-bearing bridgmanite compositions exhibit splitting at lower-mantle pressures consistent 260 with predictions of distortion of the A-site (Hsu et al. 2010). At pressures above 1 bar for Fe²⁺-261 rich Fs75Co25 and En50Fs50 compositions, absorption at ~2-3 mm/s requires two Fe²⁺ sites 262 263 (Figure 3a,b, Figure 4a,b). OS of both sites increases with pressure (Figure 5) and the relative 264 weight of the higher-QS site increases, in accord with increasingly asymmetric A-sites. At ~75 GPa in these compositions, $\sim 1/3$ of the Fe²⁺ sites exhibit OS=3.8-3.9 mm/s, with the remaining 265 Fe²⁺ at QS=2.1-2.5 mm/s. For Fe³⁺ in the A-site, QS remains relatively constant with pressure. 266 267 However, spectral resolution does not allow us to either confirm or disprove development of multiple Fe³⁺ A-sites due to site distortion (Hummer and Fei 2012). 268 Charge transfer (Fe^{2.5+}) 269 270 The signature of charge transfer, i.e. increased delocalization of electrons between iron ions, is 271 observed in Fe compositions Fs75Co25 and En50Fs50 as an absorption peak at ~1.6 mm/s (Figure 2a, Figure 3a, Figure 4a-b). This peak can be fitted with a doublet with CS ~0.8 and QS 272 ~1.6. intermediate between typical parameters observed for Fe²⁺ and Fe³⁺ and similar to Fe^{2.5+} as 273 previously identified in bridgmanite (Fei et al. 1994; Xu and McCammon 2002) (Figure 7). The 274 Fe^{2.5+} doublet intensity is interpreted to represent 50% Fe²⁺ and 50% Fe³⁺ when calculating 275 Fe³⁺/total Fe (Figure 6b). This assumption yields constant Fe³⁺/total Fe with 300 K 276 277 compression/decompression for all compositions within ~10\% error entailed in fitting site 278 weights, indicating no change in total redox state with pressure. 13

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In general, the favorability of charge transfer would be expected to depend on whether Fe²⁺ and Fe³⁺ occupy neighboring sites (note that Fe-rich En50Hem50 does not exhibit charge transfer because all Fe is Fe³⁺) and how much energy is required for electrons to hop between these sites. As noted in previous work (Fei et al. 1994), the shortest site-site distances for electron migration in the bridgmanite structure are between face-sharing B-sites and A-sites. However, the iron-rich compositions examined in this study incorporate at least enough Al and Si to fill the B-site, and are expected to accommodate both Fe²⁺ and Fe³⁺ in the A-site. The face-sharing A-A site distance in Fe-rich bridgmanite, based on lattice parameters determined by X-ray diffraction (Dorfman et al. 2012, 2013), is ~10% larger than the face-sharing A-B site distance, but comparable to edge-sharing octahedra in chain silicates at 1 bar known to exhibit Fe²⁺-Fe³⁺ charge transfer (Mattson and Rossman 1987). In the Fs75Co25 composition, Fe^{2.5+} completely replaces Fe³⁺ at all pressures at 300 K. Even at 1 bar, ~20% of the spectral weight is a component with CS=0.6 mm/s, too high for Fe³⁺ and consistent with Fe^{2.5+} (Figure 7). With increasing pressure, the intensity of the Fe^{2.5+} component is constant within uncertainty, though the Mössbauer parameters are more uncertain due to overlap. Rietveld refinement of atomic positions (Dorfman et al. 2012) confirms that all Fe in this composition resides in the A-site. As a result, charge transfer in this composition must take place between face-sharing A-site Fe²⁺ and A-site Fe³⁺. For En50Fs50-composition bridgmanite, Fe²⁺-Fe³⁺ neighboring A-sites will be diluted by Mg in the A-site, and charge transfer appears to be possible but less favorable. A charge transfer component accounting for $\sim 1/3$ of the Fe³⁺ is needed to fit data obtained at ~ 75 GPa (Figure 4b). Although Fe²⁺-Fe³⁺ neighboring A-sites would be expected to be nearly as common in En50Fs50 14

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as in Fs75Co25, a separate Fe³⁺ doublet is still observed, and upon decompression to pressures 25 GPa and below the Fe^{2.5+} component disappears (Figure 3a.b). Weights of Fe^{2.5+} and Fe³⁺ sites at all conditions are consistent with the 1 bar Fe³⁺/total Fe (Figure 6b). These observations suggest a pressure-driven crossover in the stability of charge transfer in this composition. Compression of adjoining sites is expected to lower energy required for electron transfer (Fei et al. 1994; Xu and McCammon 2002). The mobility of electrons between A-sites appears to increase accordingly between 25 and 75 GPa. Observations of charge transfer in En50Fs50 relative to Fs75Co25 bridgmanite these samples indicate that charge transfer is promoted by pressure and high Fe concentration. Evidence for charge transfer in En90Hem10-composition bridgmanite, which is also mixedvalence but relatively Fe-poor and has (Mg+Fe)/(Al+Si) ratio greater than 1, is not clear (Figure 4c). A Fe²⁺ site with CS~0.8 at high pressures is observed in En90Hem10 composition, but broad peaks, high QS, and no change in intensity of Fe³⁺ doublets when this doublet appears (Figure 6b, Table 2) all suggest the CS 0.8 mm/s doublet is more likely to represent Fe²⁺ in untransformed glass starting material. In contrast to the Fe²⁺-rich compositions Fs75Co25 and En50Fs50, in the En90Hem10 composition the \sim 50% Fe²⁺ and 50% Fe³⁺ are expected to occupy the A- and B-sites respectively due to the higher (Mg+Fe)/(Al+Si) ratio, so charge transfer would take place between A- and B-sites rather than two A-sites. Even for this lower Fe concentration. Fe²⁺-Fe³⁺ neighboring sites would be expected to be common: the probability that none of the 8 B-sites surrounding Fe^{2+} in the A-site is Fe^{3+} should be ~50%. Under these conditions A-B site charge transfer is not evident, though the presence of an overlapping Fe^{2.5+} doublet cannot be ruled out. In contrast to intuition based on electron migration distance (Fei et al. 1994), charge

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transfer appears to be more favorable between face-sharing A-sites than between face-sharing A- and B-sites. Although the distance between A-sites is longer, the enthalpy cost of moving an electron into the highly-compressed B-site (effectively, Fe²⁺ in B and Fe³⁺ in A) may provide an explanation for a failure to observe A-B charge transfer.

Implications

If previous studies of physical properties of bridgmanite have misidentified or missed spin transitions based on incorrect assignment of sites observed by Mössbauer spectroscopy. predicted seismic velocities for iron-bearing lower mantle phase assemblages may be systematically offset. The greatest effects of valence states of iron on geophysical properties are likely to be observed in the shallow lower mantle due to softening during the spin transition for Fe³⁺-bearing compositions (Shukla et al. 2016) and higher contrast in bulk compressibility and density between Fe²⁺- and Fe³⁺-rich bridgmanite at pressures at and below the Fe³⁺ spin transition (Liu et al. 2018). At greater depths above the spin transition pressure range, seismic velocities of Fe²⁺- and Fe³⁺-rich bridgmanite converge. While speciation of Fe will also affect partitioning of Fe between bridgmanite and ferropericlase in the mantle (e.g. Piet et al. 2016), the net effect of multiple orders of magnitude difference in bridgmanite-ferropericlase partition coefficient on density contrast is only 0.1% (Ricolleau et al. 2009; Dorfman and Duffy 2014), suggesting that effects of partitioning on elastic properties are too subtle to be observed via seismology. Attempts to use seismology to map the redox conditions of the lower mantle should focus on depths below the spin transition pressure, between 660 and 1000 km.

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The dependence of charge transfer in mantle bridgmanite on composition and pressure may not significantly affect elastic properties, but could result in regional- and depth-variation of transport properties including electrical and thermal conductivity. A-A site charge transfer should have no effect on density of the bridgmanite structure as the overall site occupancy remains the same. A-B site charge transfer would decrease density by expanding the B-site, but based on our data, we infer that this mechanism is not significant in Earth's mantle. Either mechanism of charge transfer, along with other mechanisms including conductivity of vacancies and ions, will contribute to electrical and thermal conductivity (Xu and McCammon 2002). Our observations do support enhancement of charge transfer by Fe-enrichment and high pressures corresponding to the deep lower mantle—both likely characteristics of dense heterogeneities (Ishii and Tromp 1999; Lau et al. 2017) and basalt-rich slabs near the core-mantle boundary (e.g. Grand 2002). In addition, the very deep lower mantle has been suggested to host mixedvalence bridgmanite, even under reduced conditions in contact with metallic iron from the outer core (Shim et al. 2017). Regions rich in subducted basalt will have higher Al- and Si-content, which is more likely to stabilize mixed-valence iron in the bridgmanite A-site. Basalt graveyards could thus more efficiently conduct heat from the outer core, with corresponding effects on the formation of thermochemical plumes and entrainment of subducted material. As the densest heterogeneities identified in the lower mantle, ULVZs resting on the core-mantle boundary are likely Fe-rich (e.g. Rost et al. 2005; Brown et al. 2015), but their redox states are not wellconstrained by available geophysical data. Proposed compositions for ULVZs include enrichment in metallic melt (Williams and Garnero 1996; Liu et al. 2016) or iron-rich oxide (Wicks et al. 2010; Hu et al. 2016; Liu et al. 2017) or silicate (Mao et al. 2006). However, the

high temperatures expected at the base of the mantle may impede charge transfer (Mattson and Rossman 1987, Xu and McCammon 2002), and additional mechanisms such as conductivity of vacancies and ions will be increasingly important at high temperatures (Xu and McCammon 2002). Future experimental studies will also need to quantify charge transfer in lower-mantle post-perovskite. If observed, enhanced electrical and thermal conductivity of ULVZs may indicate either the presence of Fe metal or charge transfer due to mixed-valence Fe-rich bridgmanite. Distinguishing these hypotheses is important to constraining redox evolution at the core-mantle boundary.

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acknowledges funding support from the Marie Heim-Vögtlin program of the Swiss National Science Foundation through project PMPDP2 151256 and the National Science Foundation project EAR-1664332. Figure and table captions Figure 1: X-ray diffraction patterns for bridgmanites obtained after laser heating and before decompression with energy-domain SMS, with diffraction angle converted to lattice spacing (d) to account for differences in diffraction wavelength. Diffraction peaks from medium (N=NaCl, and Ar for En90Hem10 sample) and minor stishovite (St) are identified, and black sticks below comprise bridgmanite reference pattern. Figure 2: Observed Mössbauer spectra at 1 bar. For compositions in a) b) and c), synchrotron Mössbauer spectra are obtained at ESRF ID18 from iron-bearing bridgmanite samples recovered to 1 bar after 300 K decompression from high-pressure synthesis. In d), conventional ⁵⁷Co Mössbauer spectrum is obtained at APS sector 3 offline lab from synthetic akimotoite before 300 K compression. Blue doublets=Fe²⁺. Red doublets=high spin Fe³⁺. Purple doublets=Fe^{2.5+} charge transfer component. Original data are black points, with total fit indicated by red curve and misfit by red points. Figure 3: Observed Mössbauer spectra for iron-bearing bridgmanite at high pressure ~27 GPa below the spin transition in Fe³⁺. Spectra shown in a) b) and c) use synchrotron Mössbauer source at ESRF ID18 (total Fe³⁺/total Fe listed for each composition is assumed to be same as 1 bar recovered sample), while d) uses conventional ⁵⁷Co Mössbauer source at APS sector 3 offline lab. Blue doublets=Fe²⁺ (where darker blue indicates distorted A-site with high OS, and

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lighter blue Fe²⁺ in untransformed glass in En90Hem10 composition in c)). Red doublets=high 408 spin Fe³⁺. Purple doublets=Fe^{2.5+} charge transfer component. Original data are black points, with 409 total fit indicated by red curve and misfit by red points. 410 411 Figure 4: Observed Mössbauer spectra for iron-bearing bridgmanite at high pressure ~72 GPa above the spin transition in Fe³⁺. Spectra shown in a) b) and c) use synchrotron Mössbauer 412 source at ESRF ID18 (total Fe³⁺/total Fe listed for each composition is assumed to be same as 1 413 bar recovered sample), while d) uses conventional ⁵⁷Co Mössbauer source at APS sector 3 414 offline lab. Blue doublets=Fe²⁺ (where darker blue indicates distorted A-site with high QS, and 415 lighter blue Fe²⁺ in untransformed glass in En90Hem10 composition in c)). Red doublets=high 416 spin Fe³⁺, and pink=low spin Fe³⁺. Purple doublets=Fe^{2.5+} charge transfer component. Original 417 418 data are black points, with total fit indicated by red curve and misfit by red points. 419 Figure 5: Mössbauer parameters a) quadrupole splitting (OS) and b) center shift (CS) observed 420 for bridgmanite as a function of pressure for all sites in all compositions. Symbols indicate bulk 421 composition of each bridgmanite sample (diamonds: Fs75Co25, squares: En50Fs50, circles: 422 En90Hem10, triangles: En50Hem10), while colors indicate spin and valence state of iron (blue: Fe²⁺, purple: Fe^{2.5+} (charge transfer), red: high-spin Fe³⁺, light pink: low-spin Fe³⁺, magenta: 423 mixed-spin Fe³⁺ (overlap between doublets impedes resolution of separate high- and low-spin 424 425 doublets). 426 Figure 6: a) Relative total amount of high-spin iron in each sample as a function of pressure, based on sum of area of all high-spin Fe²⁺, Fe^{2.5+}, and Fe³⁺ doublets. b) Relative total amount of 427 Fe³⁺ in each sample as a function of pressure, based on area of Fe³⁺ doublets plus half the area of 428

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429 charge transfer component. Error bars represent 2 σ obtained from fit of Mössbauer spectral 430 intensity. 431 Figure 7: Mössbauer parameters observed for Fe in bridgmanite and other compounds. Results from this study for Fe²⁺=blue circles, high-spin Fe³⁺=red up-pointing triangles, low-spin 432 Fe³⁺=pink down-pointing triangles, charge transfer (Fe^{2.5+}) component=purple diamonds. Error 433 bars represent fit uncertainty for each doublet. Predicted values for QS from density functional 434 theory for Fe²⁺ from (Hsu et al. 2010) and for Fe³⁺ from (Hsu et al. 2011). Shaded regions 435 436 comprise envelope of observations from previous studies: (Greenwood and Gibb 1971; Pasternak 437 et al. 2002; Hummer and Fei 2012; McCammon et al. 2013). 438 Table 1: Microprobe composition analysis of starting materials performed at the University of 439 Lausanne. Standards used to quantify SiO₂, MgO, and FeO were forsterite, fayalite and 440 orthopyroxene. Oxide totals for glass were consistently less than 100%, potentially due to use of Fe²⁺-rich standards, higher Fe³⁺-content in glass, and matrix effects. Measurements with totals 441 442 less than 97% were dropped. FeO vs. FeO_{1.5} was determined by Mössbauer spectroscopy at 443 ambient conditions. 444 Table 2: Mössbauer parameters observed for bridgmanite. CS and QS in units of mm/s. * indicates parameters constrained to equal values to aid fit convergence. Note that Fe^{2.5+} fraction 445 can be assumed to represent 50% $Fe^{2+} + 50\% Fe^{3+}$. 446 Table 3: Hyperfine parameters of high- and low-spin Fe³⁺ in perovskites obtained by energy-447 448 domain Mössbauer spectroscopy.

Tables and figures

450 Table 1

	En50Fs50 (pigeonite)		En90Hem10 (glass)		
	Weight % oxides	Cations normalized to 3 O (3% Fe ³⁺)	Weight % oxides	Cations normalized to 3 O (30% Fe ³⁺)	
SiO ₂	52.0(2)	0.996	49.2(2)	0.900	
MgO	18.1(2)	0.519	35.4(4)	0.965	
FeO	30.3(2)	0.478	13.31(18)	0.143	
FeO _{1.5}		0.015		0.061	
Total	100.4(4)	1.993	97.8(6)	2.069	

452 Table 2

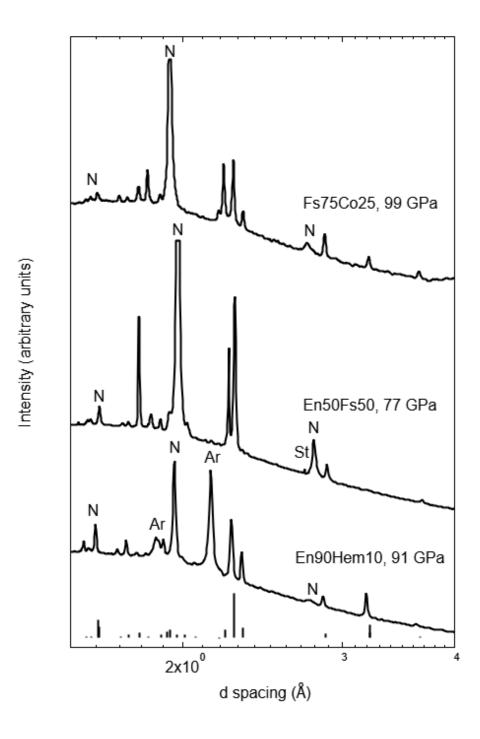
	Fs75Co25			En50Fs50				
	1 bar Fe ²⁺		Fe ^{2.5+}	1 bar Fe ²⁺		Fe ³⁺ HS		
CS	1.07(4)		0.60(8)	1.06(3)			0.39(3)	
(mm/s)								
QS	2.01(7)		1.06(20)	1.96(6)		0		
(mm/s)								
Weight	80(11)		20(11)	72(6)		28(6)		
	15 GPa			25 GPa				
	Fe ²⁺		Fe ^{2.5+}	Fe ²⁺		Fe ³⁺ HS		
CS	1.089*	1.089*	0.84(15)	1.21(5)	1.08(3)	0.2(3)		
	(18)	(18)						
QS	3.66(8)	2.34(2)	1.2(3)	2.69(11)	1.73(9)	1.0(4)		
Weight	6(9)	76(17)	17(16)	52(27)	26(21)	22(21)		
	76 GPa			75 GPa				
	Fe ²⁺		Fe ^{2.5+}	Fe ²⁺		Fe ^{2.5+}	Fe ³⁺ HS	
CS	0.998(9)	1.0(3)	0.6(2)	0.985(8)	1.00(4)	0.9(9)	0.4 (10)	
QS	3.926	2.4(4)	1.7(7)	3.90(2)	3.1(3)	1.6	0.7 (19)	
	(19)					(1.9)		
Weight	28(14)	54(20)	18(19)	20(27)	43(28)	18(17)	18(18)	
	En90Hem10			En50Hem50				
	1 bar Fe ²⁺			1 bar				
			Fe ³⁺ HS	Fe ³⁺ HS				
CS	1.02(7)		0.42(9)	0.36(3)				
QS	1.96(19)		0.95(16)	0.73(5)				
Weight			53(9)	100				
	36 GPa			32 GPa				
	Fe ²⁺		Fe ³⁺ HS	Fe ³⁺ A HS		Fe ³⁺ B HS		
CS	1.2(8)	0.8(7)	0.34(3)	0.35(2)		0.32(3)		
QS	3.1(1.7)	2.3(1.4)	1.02(6)	1.37(11) 0.59(10)				
Weight	22(14)	25(18)	52(16)	0.5*				
	67 GPa 70 GPa							
	Fe ²⁺		Fe ³⁺ mixed	Fe ³⁺ A HS Fe ³⁺ B LS		S		
CS	1.0(3)	0.81(13)	0.29(3)	0.38(9) -0.07(10))		
QS	3.3(3)	2.3(7)	1.13(9)	1.35(4) 1.30(5)				
Weight	20(17)	31(16)	50(15)	0.5* 0.5*				

455 Table 3

	HS CS (mm/s)	HS QS (mm/s)	LS CS (mm/s)	LS QS (mm/s)	Reference
Bridgmanite (Mg,Fe,Al) (Fe,Al,Si)O ₃	0.0-0.5	0.3-1.7	N/A	N/A	(Hummer and Fei 2012)
	0.4	1.2-1.5	N/A	N/A	(Potapkin et al. 2013)
	0.4	0.8-1.3	-0.1	0.9	(Kupenko et al. 2015)
	0.1-0.15 (ΔIS A-B)	1.1-1.2	0.3 (ΔIS A- B)	1.3-1.4	(Liu et al. 2018)
	0.2-0.4	0.6-1.4	0.0	1.3	This work
Perovskite- structured rare-earth orthoferrites	0.22-0.24	0.24-0.59	0.04-0.23	0.55-1.42	(Pasternak et al. 2002)

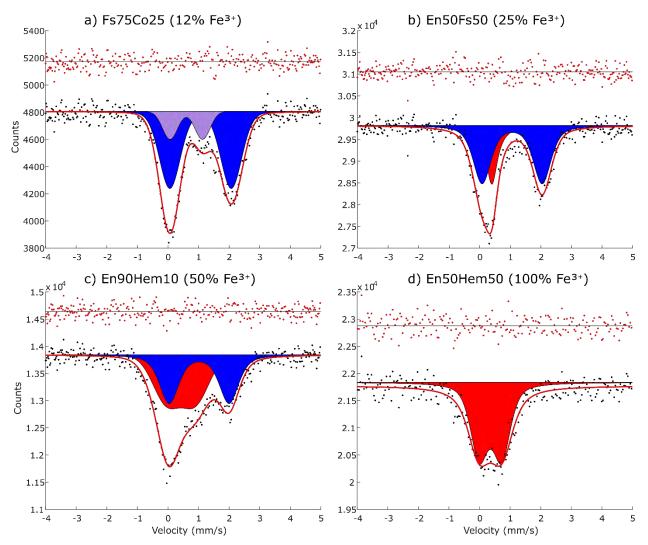
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458 Figure 1



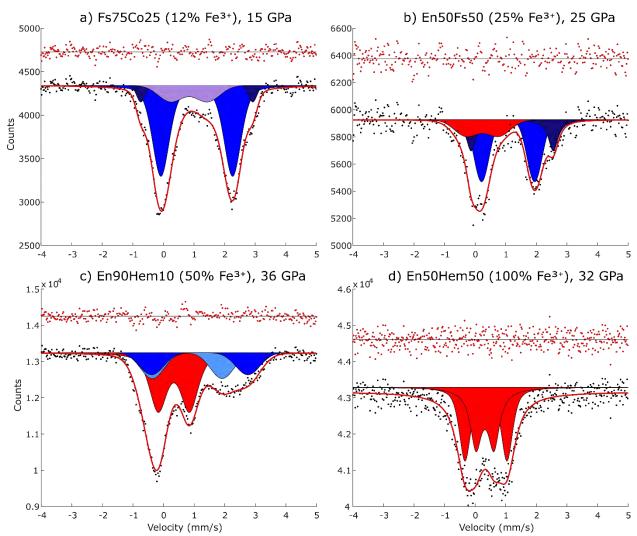
460 Figure 2

1 bar



462 Figure 3

Below spin transition

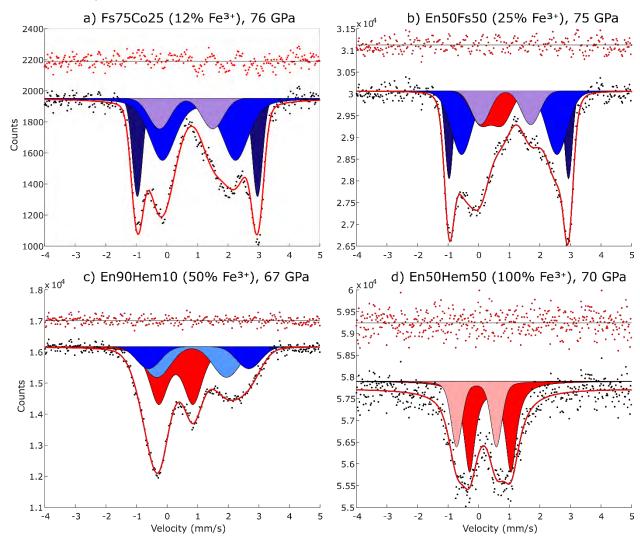


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465 Figure 4

Above spin transition



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470 Figure 5

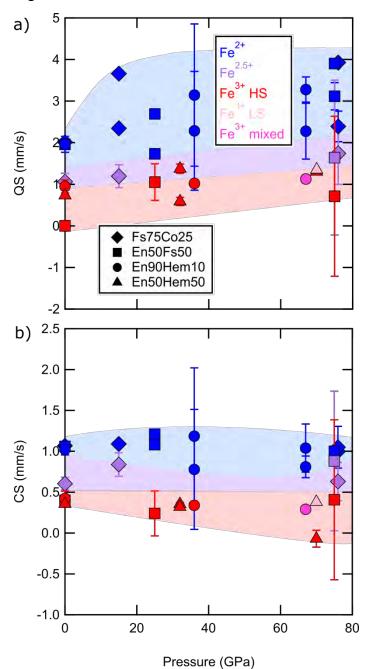
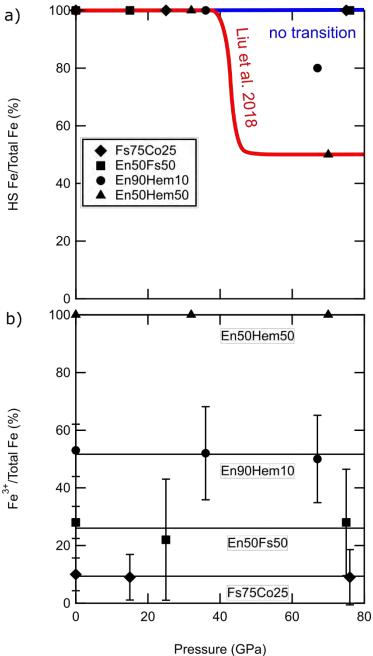


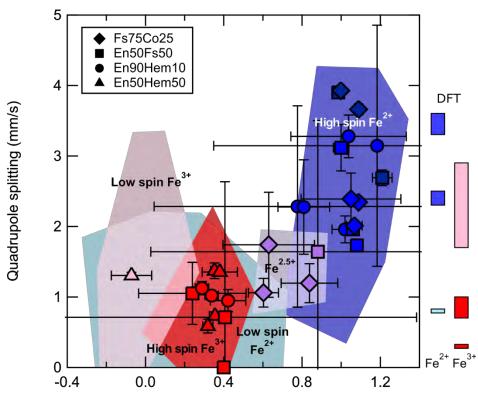
Figure 6 472



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476 Figure 7



477 Center shift (mm/s)

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