



22 magnetite measured on a micron-scale is caused by three different effects: Al solid  
23 solution, Al-rich nanometer-sized lamellae and zinc spinel nanoparticles in the host  
24 magnetite. Here, we propose a genetic relationship among the three different phases  
25 mentioned above. At first, a continuous increase of the Al concentration in the interfacial  
26 fluid can be incorporated into the crystal lattice of magnetite forming a solid solution.  
27 During cooling in a later stage, aluminum in magnetite is oversaturated and exsolution of  
28 hercynite (Al-rich lamellae) occurs from the host magnetite. If the Al concentration at the  
29 fluid-magnetite interface still increases during further growth of magnetite, the  
30 substitution of Fe by Al has gradually reached saturation so that aluminum cannot be  
31 incorporated in the magnetite crystal structure any longer. Using the magnetite lattice as a  
32 template, nucleation of abundant zinc spinel nanoparticles occurs. This will in turn lead to  
33 a gradual depletion of Al concentration in the interfacial fluid until the available ions for  
34 zinc spinel nucleation and growth have been used up. As a result, the migrating  
35 fluid-magnetite interface will enrich the Al concentration in the interfacial fluid until the  
36 available ion concentration is sufficient for nucleation of zinc spinel phase again. The  
37 fluid-mineral interface in this mechanism has been repeatedly utilized during crystal  
38 growth, providing an efficient way for the uptake of trace element from a related  
39 undersaturated bulk fluid.

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41 **Keywords:** fluid-mineral interface, mineral nanoparticle, hydrothermal magnetite, uptake  
42 of trace element

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## Introduction

45 The partition coefficient of a certain trace element between the bulk fluid and the  
46 bulk crystal can be determined in controlled experiments under a steady-state condition.  
47 The incompatibility or compatibility of such a trace element in a mineral can thus be used  
48 to determine its geochemical behavior in natural systems. However, the fluid-mineral  
49 interfaces generally have a composition that differs from that of the bulk fluid or the bulk  
50 crystal (Brown, 2001; Fenter and Sturchio, 2004; Stipp, 1999). The interfacial fluid  
51 composition could be recorded by the newly formed crystal lattice when the crystal  
52 growth rate is fast enough, leading to deviations from thermodynamic partitioning  
53 equilibrium (Watson, 1996; Watson, 2004). This kinetic control on the uptake of trace  
54 elements induced by the crystal growth rate is so called the growth entrapment model  
55 (Lanzillo et al., 2014).

56 The formation of a sharp compositional boundary between a trace element-poor  
57 domain and a trace element-rich domain is suggested to form in this model when the  
58 crystal growth rate is fast enough. This signature will be further preserved when it is not  
59 eliminated by the intracrystalline diffusion, which is a function of the temperature.  
60 Specifically, a trace element-rich domain in a host mineral can contain some mineral  
61 nanoparticles when the local fluid is supersaturated with respect to a new phase (Deditius  
62 et al., 2018).

63 To better understand the role of mineral nanoparticles at a fluid-mineral interface,

64 this study focuses on a well-documented magnetite grain with oscillatory zoning from the  
65 Baishiya skarn iron deposit, East Kunlun orogenic belt, northern Qinghai-Tibet Plateau.  
66 Using high-resolution transmission electronic microscopy (TEM), we have observed  
67 Al-rich nanometer scale lamellae, Ti-rich magnetite nanoparticles and zinc spinel  
68 nanoparticles in the host magnetite. An integrated study from a micron scale to a  
69 nanometer scale is focused on: 1) genesis of mineral nanoparticles in hydrothermal  
70 magnetite; 2) explanation why zinc prefers to form zinc spinel nanoparticles rather than  
71 be incorporated into the magnetite structure thus forming a solid solution; 3) a potential  
72 mechanism for the uptake of trace elements at a fluid-mineral interface in hydrothermal  
73 deposits.

74

### 75 **Geological setting and sample description**

76 The East Kunlun orogenic belt in the northern Qinghai-Tibet Plateau consists of  
77 multistage granitoid batholiths (Fig. 1a), and is dominated by Late Paleozoic to early  
78 Mesozoic granitoids (Fig. 1b). The magmatic-hydrothermal deposits are mainly  
79 associated with the Triassic magmatism induced by the northward subduction of the  
80 Songpan-Ganzi-Bayan Har block. The Baishiya skarn iron deposit, located in the uplift  
81 edge of the Dulan-Elashan belt, is hosted by the Triassic granodiorite and the limestone in  
82 the Dagangou Formation (Fig. 1c). There is a discontinuous mineralized contact zone  
83 approximately 2000 m long and 50-300 m wide, with a proven reserve of 8 Mt Fe at an  
84 average grade of 35–51 % Fe.



85 The magnetite sample was collected from the Baishiya skarn iron deposit, providing  
86 a unique opportunity to investigate trace element behavior in hydrothermal magnetite at a  
87 nanometer scale. The magnetite ores in the Baishiya skarn deposit have previously been  
88 identified with four generations based on in-situ textures and trace elements analyses, as  
89 well as gangue mineral assemblage (Yin et al., 2017). In particular, the studied magnetite  
90 crystal coexisting with phlogopite and minor hedenbergite is characterized by a  
91 homogeneous core and an oscillatory rim in the backscattered electron image (Fig. 2).  
92 This special texture is a perfect record of the migrating fluid-mineral interface during  
93 non-equilibrium crystal growth.

94 Trace element concentrations have been determined in this magnetite crystal by  
95 electron microprobe and laser ablation inductively coupled plasma mass spectrometry  
96 (LA-ICP-MS) (Yin et al., 2017). They are plotted in the order of increasing compatibility  
97 in the magnetite structure (Fig. 3; Dare et al., 2014). In comparison with the  
98 homogeneous core, the oscillatory rim has significantly higher concentrations of Si, Al,  
99 Mg, Zn and lower concentrations of Ti, V, and Ni.

100

101

### **Analytical methods**

102 Electron-transparent foils for TEM were prepared by focused ion beam technique  
103 (FIB), utilizing a single-beam device (FEI FIB200 TEM) and a state-of-the-art  
104 DualBeam™ system (FEI Helios) operated at German Research Center for Geosciences,  
105 Potsdam. The foils have been cut by FIB along the growth direction of magnetite to

106 investigate the phase variation on a nanometer scale (Fig. 2d). The foil thickness is  
107 approximately less than ~100-150 nm. Details about the FIB sample preparation from  
108 magnetite can be found elsewhere (Wirth, 2009). The TEM was performed employing a  
109 TECNAI F20 X-Twin transmission electron microscope operated at 200 keV with a  
110 Schottky emitter as electron source. Using scanning transmission mode, annular-dark  
111 field (STEM-ADF) images and some EDS maps were acquired with a JEOL  
112 JEM-Arm300F at 300 keV acceleration voltage and a cold field emission gun (FEG)  
113 emitter as an electron source (JEOL Application laboratory, Tokyo, Japan). The TECNAI  
114 was equipped with a Gatan imaging filter, a Fishione high-angle annular dark-field  
115 detector (HAADF) and an energy dispersive X-ray (EDX) analyzer with ultrathin  
116 detector window. The HAADF detector can be operated as a Z-contrast detector using a  
117 short camera length (75mm) or as a combined Z-contrast plus diffraction contrast  
118 sensitive detector (camera length 330 mm). TEM bright-field and dark-field images, as  
119 well as high-resolution lattice fringe images, were routinely acquired as energy-filtered  
120 images applying a 20 eV window to the zero-loss peak. High-resolution images were  
121 used to calculate diffraction patterns by Fast Fourier Transform (FFT). EDX analyses  
122 usually have been carried out in the STEM thus avoiding mass loss during data  
123 acquisition. Lines scans and elemental mapping have been acquired using the  
124 corresponding X-Ray intensities.

125

126

## Results

#### 127 4.1 Nanometer-sized Al-Mg-rich lamellae in magnetite

128 The Al-Mg-rich lamellae (approximately 100 nm × 3 nm) are documented in  
129 STEM-ADF image (Fig. 4). Element mapping and line scans across the lamellae show  
130 that the thin lamellae are enriched in Al, Mg, Ti, and depleted in Fe compared with the  
131 host magnetite (Fig. 4). Calculated diffraction patterns from the HAADF images show a  
132 split of the second order reflection of the (111) reflections, suggesting the presence of two  
133 different spinel crystals with slightly different unit cell parameters ( $a_{0 \text{ hercynite}} = 0.8119 \text{ nm}$ ;  
134  $a_{0 \text{ magnetite}} = 0.8387 \text{ nm}$  (MinDat.org)). This scenario is in accordance with the separation  
135 of the (222) reflections of the reciprocal lattice in Fig. 7a, in which the inner spot is  
136 0.24211 nm for  $(222)_{\text{magnetite}}$  and the outer spot is 0.23438 nm for  $(222)_{\text{hercynite}}$ .  
137 Consequently, the Al-Mg-rich lamellae are a hercynite phase that has exsolved from the  
138 host magnetite, in which iron and aluminum have been partly substituted by Mg and Ti,  
139 respectively.

140

#### 141 4.2 Zinc spinel nanoparticles in magnetite

142 High-resolution TEM study revealed several crystalline inclusions up to 100 nm in  
143 size, and most of which have an irregular cross-section (Fig. 9). X-ray maps of these  
144 inclusions show a pronounced enrichment of Zn and Al and concomitant depletion of Fe  
145 compared with the host magnetite (Fig. 5). A slight enrichment of Mg content has also  
146 been observed in some cases (Fig. 6). In a few cases, a decrease of oxygen ( $K\alpha$ ) X-ray  
147 intensity together with a decrease of intensities from other elements suggests the

148 existence of pores once filled with a fluid (Fig. 6). The fluid phase was trapped in the  
149 pores and then released during FIB milling. The pore (~5 nm) seems to be inside the zinc  
150 spinel phase. It is much smaller than the thickness of the TEM foil (~100 nm). Again,  
151 there is a splitting in the third order reflections of the (111) diffraction spots in the  
152 diffraction pattern (FFT) between the Zn-Al-rich nanoparticle and the host magnetite (Fig.  
153 7b), indicating the same crystal structure with only slightly deviating lattice parameters of  
154 the zinc spinel phase from that of the host magnetite.

155

#### 156 4.3 Titanium-rich magnetite nanocrystals in magnetite

157 Occasionally, we have also observed some individual Ti-rich nanoparticles with  
158 5-30 nm in cross section (Fig. 4, 6). They are either Ti-rich magnetite or ulvöspinel due to  
159 undistinguishable lattice parameters (Fig. 7b). Host magnetite and ulvöspinel have the  
160 same cubic crystal structure and similar unit cell parameters ( $a_{0 \text{ ulvöspinel}} = 0.845 \text{ nm}$ ;  $a_{0 \text{ magnetite}} = 0.8387 \text{ nm}$  (MinDat.org)). The Ti concentration in the dark zone of magnetite  
161 approximately ranges from 10 ppm to 100 ppm on a micron scale by LA-ICP-MS (Fig. 2).  
162 Even in the trace element-rich domain on a nanometer scale, it is below the detection  
163 limit by EDX analysis in the magnetite matrix. Thus, the Ti-rich nanoparticles are  
164 unlikely to be exsolution of ulvöspinel from the host magnetite (Lilova et al., 2012). On  
165 the contrary, previous study suggested that Ti is an incompatible element in magnetite  
166 structure at middle to lower temperature (e.g., 390 °C; Nadoll et al., 2017). It will be  
167 enriched in the interfacial fluid due to migrating magnetite/fluid interface of the growing  
168

169 magnetite. Once the Ti concentration reaches to a supersaturated situation, nucleation of  
170 Ti-rich magnetite nanoparticles occurs.

171

## 172 **Discussion**

### 173 **The exsolution of a hercynite phase**

174 At a temperature below 600 °C, such as a skarn environment, the threshold value  
175 (Magnetite<sub>90</sub>Hercynite<sub>10</sub>) for hercynite exsolution in magnetite structure is that the Al  
176 content reaches approximately 1.2 wt.% (Fig. 8) (Turnock and Eugster, 1962). This  
177 critical value could be comparable to our “mixed” result in the magnetite matrix (Fig. 3).  
178 In fact, an exsolution of nanoscale hercynite from magnetite had long been observed in  
179 geological samples. For example, Al-Mn-Fe-bearing spinel nanoplates have been found  
180 in metamorphic magnetite crystals with the Al concentration up to 1.04 wt.% (Sitzman et  
181 al., 2000). Hydrothermal magnetite crystal with 2-3 wt.% of Al from the Los Colorados  
182 iron oxide-apatite deposit also contains nanoscale hercynite-rich domains (Deditius et al.,  
183 2018). Most of these hercynite lamellae have a thickness of a few nanometers and  
184 parallel to a certain orientation, which is consistent with our observation of the  
185 Al-Mg-rich lamellae in this study. Consequently, it is reasonable to conclude that the  
186 Al-Mg-rich lamellae are hercynite phases exsolved from the host magnetite.

187

### 188 **Nucleation of aluminous spinel nanoparticles at the fluid-magnetite interface**

189 Oscillatory zoning is a common feature of natural crystal growth. It can form by a

190 rapid depletion and subsequent increase of some chemical components at a fluid-mineral  
191 interface (Putnis et al., 1992). For magnetite precipitated from a hydrothermal fluid, it  
192 also requires that the magnetite growth rate is high enough relative to the intracrystalline  
193 diffusion rate (temperature-dependent) so that the sharp boundaries of the original zoning  
194 will be preserved (Deditius et al., 2018; Watson, 2004). Regardless of the magnetite  
195 growth rate, the Al concentration in the newly formed magnetite lattice are controlled by  
196 two effects. The first one is the intracrystalline diffusion rate of Al between the newly  
197 formed magnetite lattice and bulk magnetite crystal (Watson, 2004). Because the newly  
198 formed magnetite lattice precipitated directly from the interfacial fluid, the diffusion rate  
199 of Al between the interfacial fluid layer and the bulk fluid is also a crucial factor (Watson  
200 and Müller, 2009).

201 As an incompatible element in magnetite structure, aluminum prefers to remain in  
202 the fluid phase, leading to a continuous enrichment of the Al concentration at the moving  
203 fluid-magnetite interface. This will in turn result in an increased adsorption of  $\text{Al}^{3+}$  at the  
204 mobile magnetite surface, which will be incorporated into the newly formed magnetite  
205 lattice by rapid magnetite growth (Watson, 2004). If the Al concentration in the magnetite  
206 surface structure continues to increase, hercynite exsolution occurs during the late stage  
207 of cooling due to a limited solid solution for Al substitution in the magnetite structure  
208 (Phase 1; Fig. 9).

209 The radius of  $\text{Al}^{3+}$  ion (0.54 Å) is smaller than that of  $\text{Fe}^{3+}$  ion (0.65 Å) (Dare et al.,  
210 2012). An increase of this substitution by Al will lead to a corresponding increase of the

211 free energy of the magnetite structure due to this mismatch in crystal lattice. Once the  
212 replacement of Fe by Al in the magnetite surface structure has reached a maximum value  
213 (Fig. 8), additional  $\text{Al}^{3+}$  will not be incorporated into the newly formed magnetite lattice  
214 any longer, but enriched in the interfacial fluid. An aluminous spinel nucleus with a few  
215 nanometers in size will directly form at the magnetite surface using the magnetite crystal  
216 lattice as a template for nucleation. The reason is that the nucleating aluminous spinel  
217 phase closely matches with the atomic structure of the magnetite lattice so that the lattice  
218 strain energy is minimized by this way. In other words, in comparison with Al-spinel  
219 precipitated from a fluid phase (Fig. 10), the “critical size” for a stable nucleus that has  
220 used magnetite as a template for nucleation should be reduced to some extent. In fact, this  
221 chemically tailored substrate has been routinely used to modify energy landscape at the  
222 substrate/nucleus interface and thereby steer heterogeneous nucleation in the biological  
223 and artificial mineralization experiments (De Yoreo and Vekilov, 2003; Li et al., 2018).

224 The nucleation and growth of abundant Al-rich nanoparticles at the magnetite  
225 surface will not end until the available ions (e.g., Zn, Mg, Al) in the interfacial fluid have  
226 been used up. The concentration of the residual ions is not sufficient for a nucleus to  
227 exceed the “critical size” (Phase 2; Fig. 9). In other words, the nucleation process will  
228 gradually reduce the concentration of  $\text{Al}^{3+}$  in the interfacial fluid and finally lead to the  
229 absence of aluminous spinel at the moving fluid-magnetite interface, so as to a  
230 crystallization of relatively pure magnetite without aluminous spinel (Phase 3; Fig. 9).

231

## 232 **Nucleation of aluminous spinel directly from a local fluid**

233 In the model above, one may doubt that abundant aluminous nanoparticles are a  
234 direct result of the supersaturation of a local fluid, and then they have been enclosed by  
235 the growing magnetite. If this assumption is reasonable, they should have a random  
236 distribution of crystallographic orientation because they are isolated with each other,  
237 which is obviously contrary with our observation. Individual aluminous nanoparticles  
238 have been measured randomly in their electronic diffraction patterns. It turns out that they  
239 share a same crystallographic orientation with the host magnetite due to an extremely  
240 narrow d-spacings. For example, additional spots in the diffraction pattern of magnetite  
241 have never been observed using the TEM in GFZ, Potsdam. Even with a more advanced  
242 TEM (JEOL Application laboratory, Tokyo, Japan), a split can only be observed in the  
243 second order reflection (Fig. 7). This suggests that the atomic structure of aluminous  
244 spinel phase is closely matched with the host magnetite substrate.

245

## 246 **Trace element behavior at a fluid-magnetite interface**

247 Previous research generally considered that zinc is a compatible element in  
248 magnetite structure (Nadoll et al., 2014; Nadoll et al., 2017). However, the contrast of  
249 zinc concentration between the host magnetite and the aluminous spinel phase suggests a  
250 strongly preferential uptake of zinc by the aluminous spinel phase at a fluid-magnetite  
251 interface (Fig. 5-6). At a lower concentration of zinc in the host magnetite (Fig. 3), an  
252 intriguing question is why zinc prefers to form aluminous spinel nanoparticles rather than



253 being incorporated in magnetite structure as a solid solution.

254 The relationships among endmembers in the spinel group could account for this  
255 heterogeneous distribution of zinc at a fluid-magnetite interface. Specifically, franklinite  
256 ( $\text{ZnFe}_2\text{O}_4$ ) can only form a partial solid solution with magnetite (Valentino et al., 1990).  
257 The intergrowth of franklinite-magnetite lamellae has been observed in natural samples  
258 (Valentino et al., 1990). However, there is a significant substitution among Zn, Mg and Fe  
259 in aluminous spinel (Heimann et al., 2005), presumably indicating complete solid  
260 solutions. As a result, it is reasonable to consider that the free energy of lattice strain  
261 induced by the Zn substitution for Fe should be lower in the  $\text{ZnAl}_2\text{O}_4$ – $\text{FeAl}_2\text{O}_4$  series  
262 than that of the  $\text{ZnFe}_2\text{O}_4$ – $\text{FeFe}_2\text{O}_4$  series. Thus, we consider that zinc should be more  
263 compatible with aluminous spinel than with magnetite in a hydrothermal fluid.

264 Once the nucleation and growth of zinc spinel phase occurs, it may also have an  
265 influence on other trace elements. For instance, there is also a slight enrichment of Mg in  
266 the zinc spinel nanoparticle, which is less pronounced than that of Zn and Al (Fig. 6).  
267 Previous research considered that magnesioferrite ( $\text{MgFe}_2\text{O}_4$ ) phases can form a complete  
268 solid solution with magnetite (Dupuis and Beaudoin, 2011). If true, it means that the  
269 competition for Mg by aluminous spinel should be at the same level as that of magnetite.  
270 However, if the endmember of the aluminous spinel is zinc spinel, it turns out to be a  
271 competitive replacement by  $\text{Mg}^{2+}$  (0.72 Å) between  $\text{Fe}^{2+}$  (0.78 Å) and  $\text{Zn}^{2+}$  (0.74 Å)  
272 (Dare et al., 2012). The radius of  $\text{Mg}^{2+}$  is closer to  $\text{Zn}^{2+}$  rather than that of  $\text{Fe}^{2+}$ , which  
273 may be the reason for this slightly enrichment.

274

275

### Implications

276 The zinc concentration in the bulk fluid should be lower than that in magnetite  
277 because Zn is a relatively compatible element in magnetite structure (Nadoll et al., 2014).  
278 It is consistent with the fact that zinc minerals are absent in the magnetite ores and the  
279 related thin sections, indicating a zinc-undersaturated bulk fluid phase. However, at a  
280 migrating fluid-magnetite interface, a continuous increase of the Al concentration allows  
281 the nucleation of a zinc spinel phase by utilizing magnetite as a template. This local  
282 kinetic effect has changed the reaction pathway of zinc, leading to the uptake of zinc  
283 from a zinc-undersaturated bulk fluid.

284 The uptake of trace elements at fluid-mineral interfaces by nanoparticles can be  
285 comparable to some mineral replacement experiments (Li et al., 2015; Tooth et al., 2011).  
286 However, there are some prominent characteristics in this study. First and foremost, the  
287 trigger of this local kinetic effect is the higher crystal growth rate relative to  
288 intracrystalline diffusion rate at a fluid-mineral interface, causing a gradual enrichment of  
289 trace elements in the interfacial fluid. Furthermore, once the available  $Zn^{2+}$  and  $Al^{3+}$  for  
290 the nucleation of a stable zinc spinel nanoparticle are exhausted in the interfacial fluid, a  
291 relatively pure magnetite crystallization will consume Fe and enrich available  $Zn^{2+}$  and  
292  $Al^{3+}$  ions in the interfacial fluid again. The moving fluid-magnetite interface has been  
293 repeatedly utilized to nucleate zinc spinel during magnetite growth, which resembles a  
294 rhythmical distribution of pyrrhotite nanoparticles observed in a magmatic apatite

295 (Gottesmann and Wirth, 1997). Consequently, it is a potential mechanism for the uptake  
296 of trace element from an undersaturated bulk fluid. Finally, the host substrate could play a  
297 crucial role as a template in nucleating zinc spinel or mineral nanoparticles and thus have  
298 an influence on related trace element behavior, which might have been overlooked in  
299 hydrothermal systems. Such a new documentation of phenomena may be of broad  
300 relevance and importance for us to understand the growth of mineral nanoparticle (De  
301 Yoreo et al., 2015).

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303

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396

397    **Figure captions**

- 398    Figure 1. (a) A simplified tectonic outline of Qinghai-Tibet Plateau, modified from Xu et  
399    al. (2015); (b) The granitoid batholiths and related magmatic-hydrothermal deposits in

400 the East Kunlun orogenic belt, modified from Meng et al. (2013); (c) A simplified  
401 geological map of the Baishiya orefield, modified from Yin et al. (2017).

402

403 Figure 2. (a) The hand specimen of iron ores; (b) The selected magnetite crystals with  
404 euhedral morphology; (c) Photomicrograph of iron ores shows a mineral assemblage of  
405 phlogopite, magnetite, and hedenbergite; (d) BSE image of selected magnetite crystal  
406 consists of a homogeneous core and an oscillatory rim.

407

408 Figure 3. Trace element concentrations in the magnetite are determined by LA-ICP-MS  
409 and plotted with increasing compatibility into magnetite structure after Dare et al. (2014).  
410 The data source is from Yin et al. (2017).

411

412 Figure 4. STEM-ADF image and representative elemental mapping of the nanoscale  
413 lamellae using X-ray intensities of Mg-K $\alpha$ , Al-K $\alpha$ , O-K $\alpha$ , Fe-K $\alpha$  and Ti-K $\alpha$  lines.

414

415 Figure 5. STEM-ADF image and a representative elemental mapping of the mineral  
416 nanoparticles using X-ray intensity of Si-K $\alpha$ , Fe-K $\alpha$ , O-K $\alpha$ , Zn-K $\alpha$  and Al-K $\alpha$  lines.

417

418 Figure 6. STEM-ADF image and a representative elemental mapping of the mineral  
419 nanoparticles using X-ray intensity of O-K $\alpha$ , Fe-K $\alpha$ , Si-K $\alpha$ , Al-K $\alpha$ , Zn-K $\alpha$ , Ti-K $\alpha$  and  
420 Mg-K $\alpha$ . Note that there are three phases: 1) Zn-Al-Mg-rich spinel phase; 2) Ti-rich



421 magnetite phase; 3) A pore inside the zinc spinel phase characterized by a depletion of  
422 oxygen (white circle). The size of these phases may be much smaller than the foil  
423 thickness (~100 nm).

424

425 Figure 7. (a) Energy filtered high-resolution lattice fringe image of the Mg-Al-rich  
426 lamella in host magnetite and (b) the related indexed diffraction pattern (fast Fourier  
427 Transform FFT). Note that there is split of the second order of the (111) reflections,  
428 suggesting two different spinel crystals with slightly different unit cell parameters. (c)  
429 The energy filtered high-resolution lattice fringe image of the Zn-Mg-Al-rich  
430 nanoparticle and the Ti-rich nanoparticle in the TEM foil of Fig. 3. (d) The indexed  
431 diffraction pattern (fast Fourier Transform FFT) from the different zones of the lattice  
432 fringe image. Note that the lattice parameters of the Ti-rich nanoparticle resemble those  
433 of the host magnetite because no additional spots have been observed in the related zone.  
434 However, a split can be observed when the Zn-Mg-Al-rich nanoparticle and the host  
435 magnetite have been involved.

436

437 Figure 8. The hercynite-magnetite solvus (Turnock and Eugster, 1962). The value of Al  
438 content marked in the arrow represents a total concentration of Al in the  
439 hercynite-magnetite series.

440

441 Figure 9. The spatial distribution of the Al-rich lamellae bearing magnetite (Phase 1), the

442 spotted nanoparticles bearing magnetite (Phase 2) and the relatively pure magnetite  
443 (Phase 3) in the foil. Note that their alternative variation is responsible for the oscillatory  
444 zoning on a micron scale of the magnetite rim.

445

446 Figure 10. (a) A zinc spinel nucleus nucleated and grown completely within the fluid  
447 phase; (b) A zinc spinel nucleus nucleated at the magnetite substrate. Note that the  
448 interfacial energy between magnetite and spinel nucleus is much lower than that between  
449 fluid phase and spinel nucleus (De Yoreo and Vekilov, 2003).

450

# Figure 1

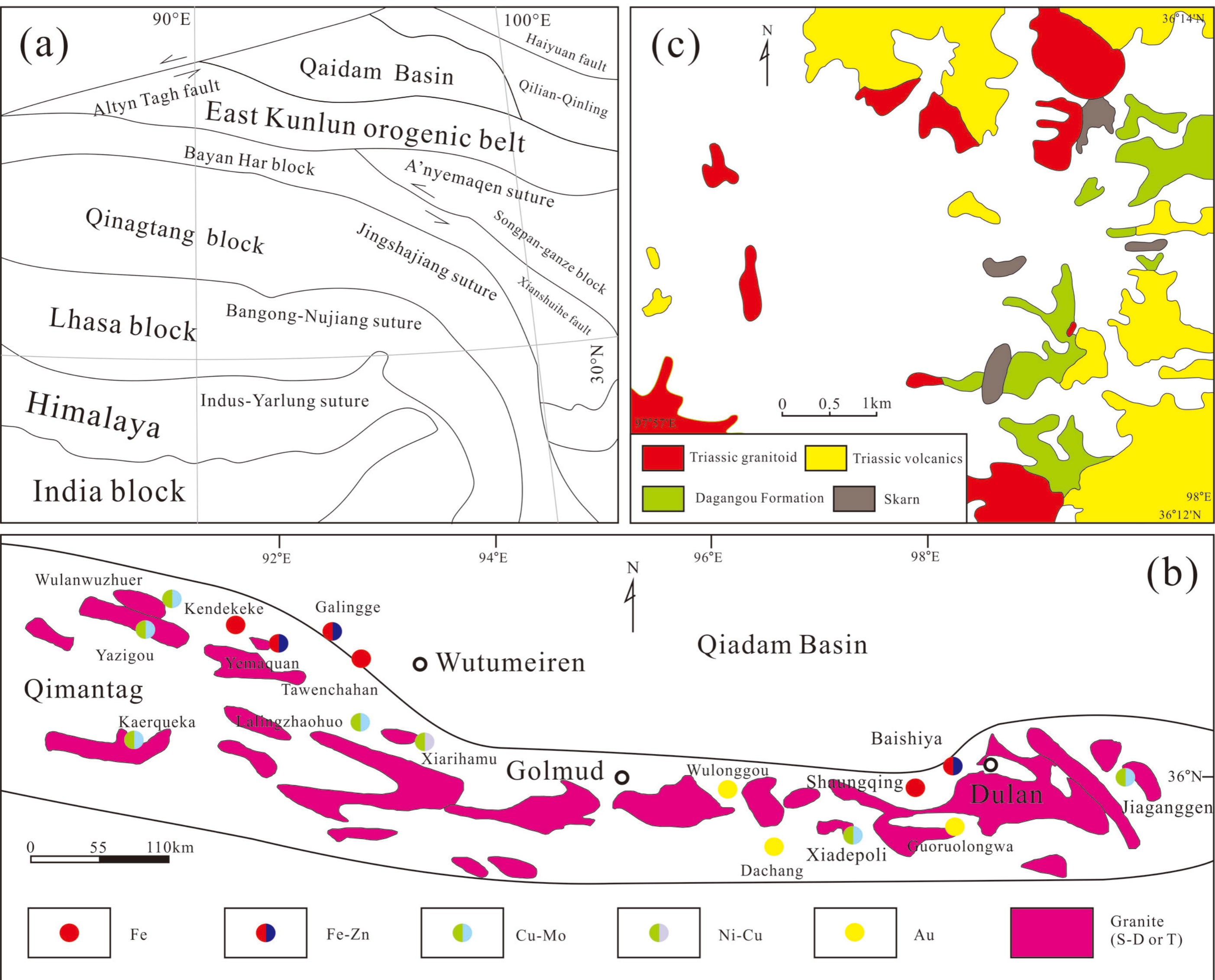
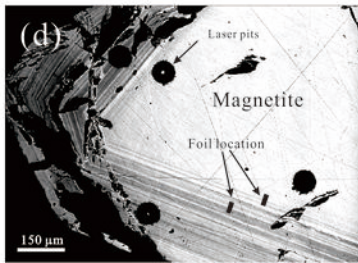
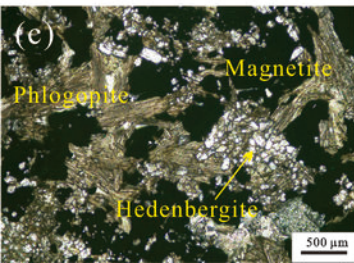
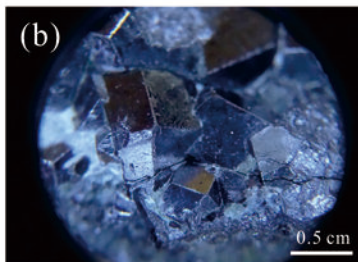
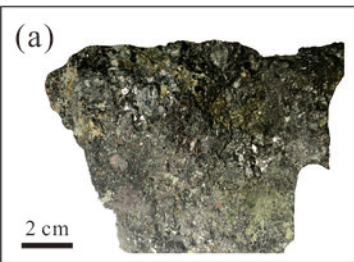


Figure 2



# Figure 3

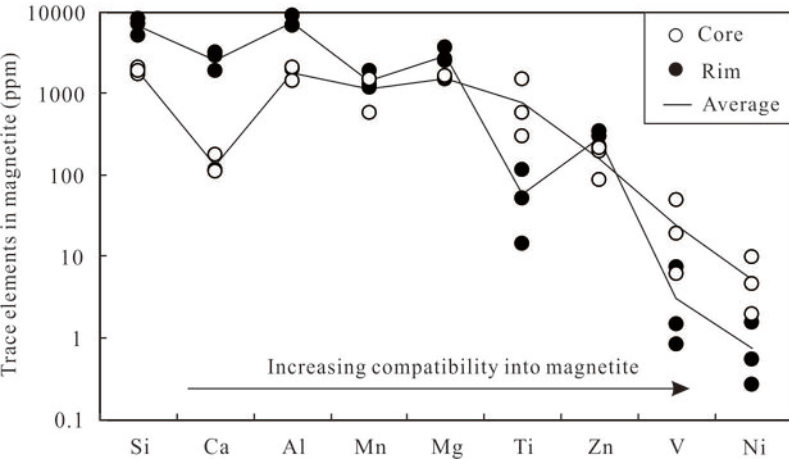




Figure 4

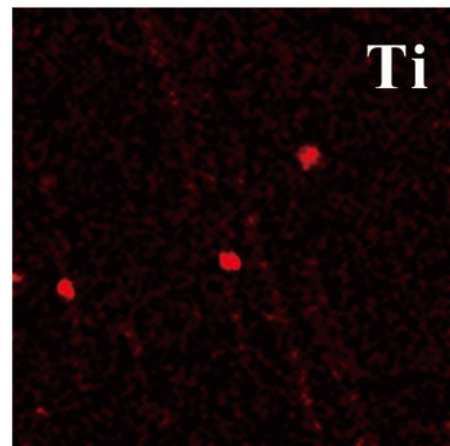
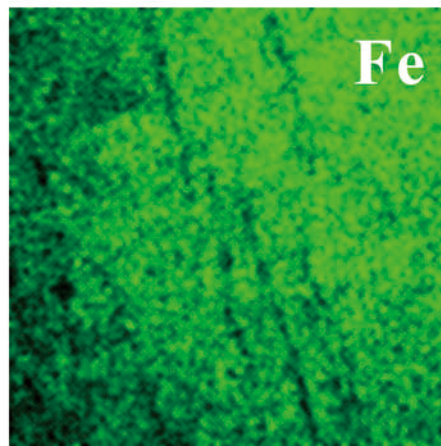
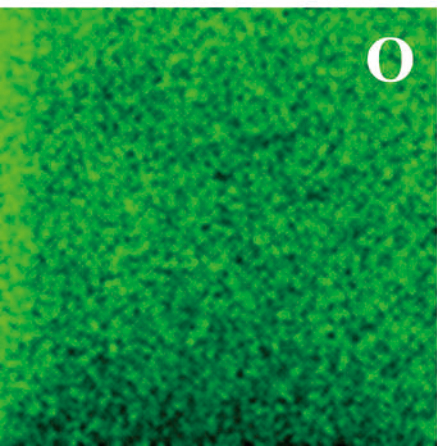
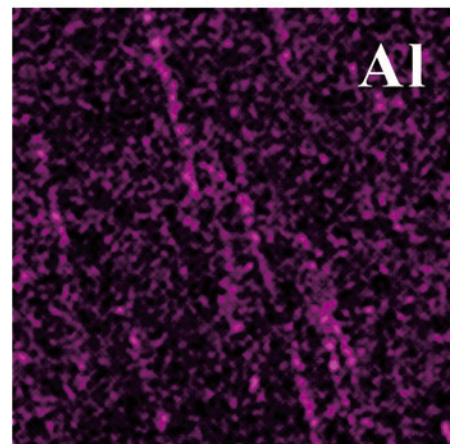
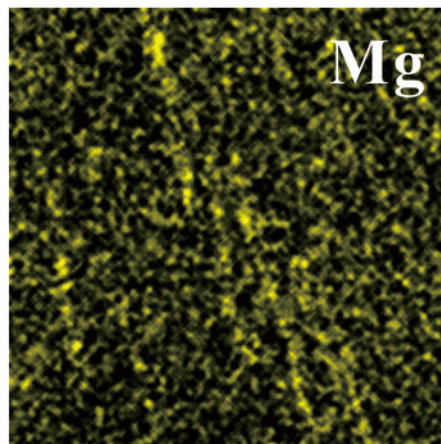
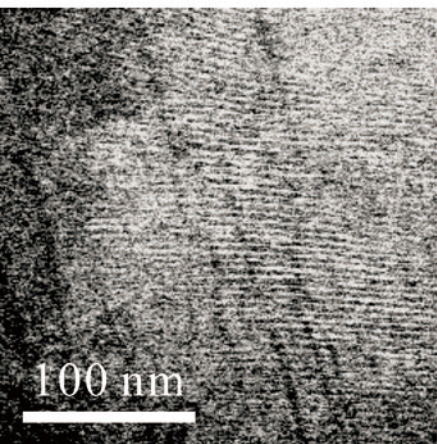
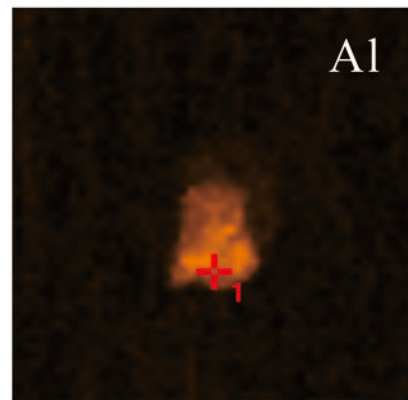
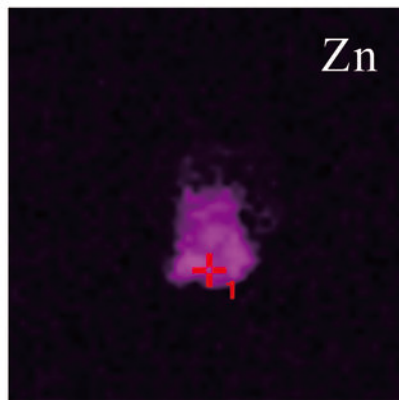
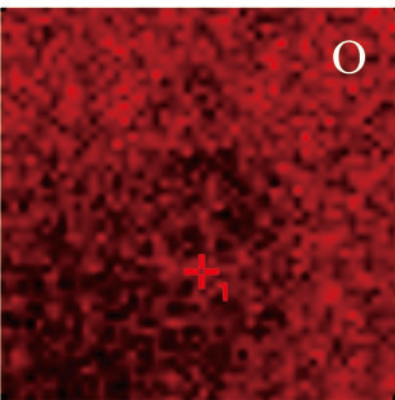
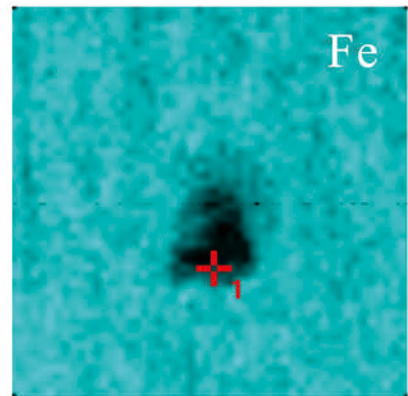
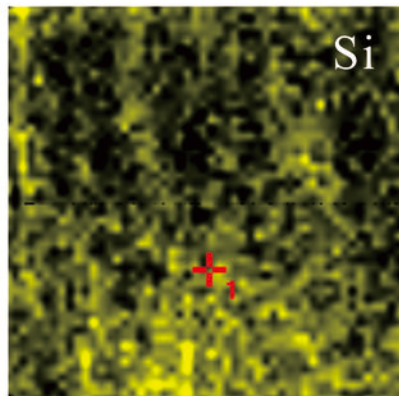
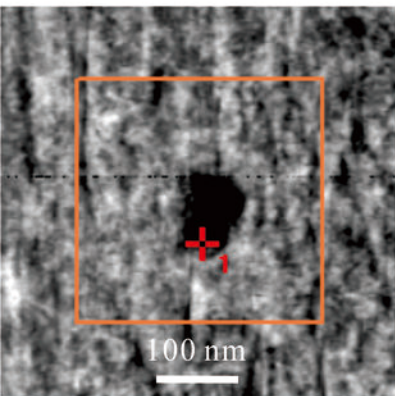
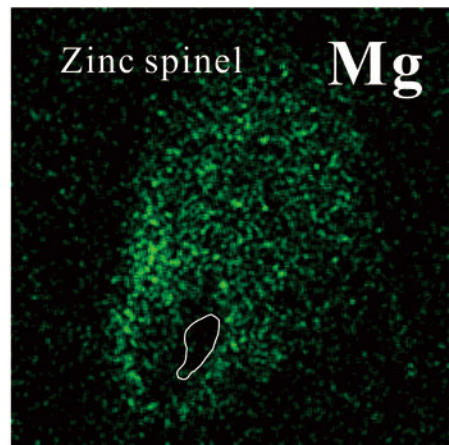
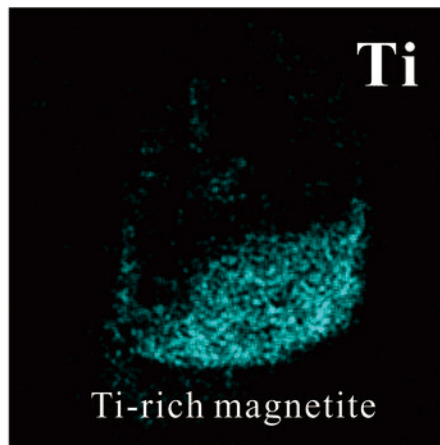
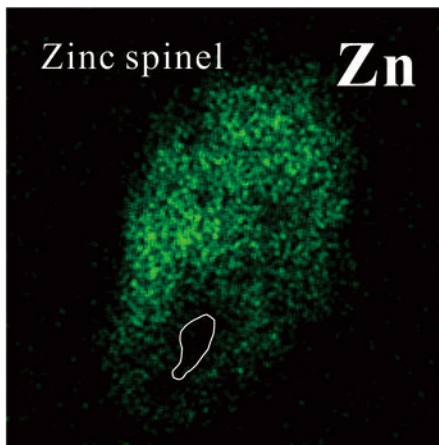
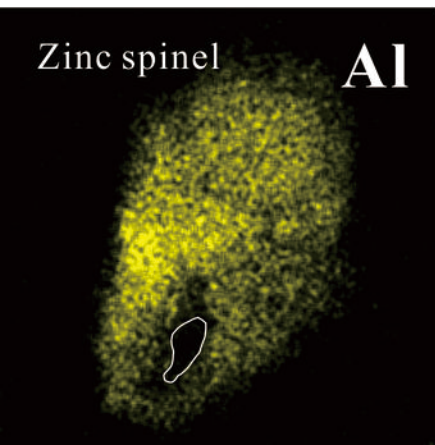
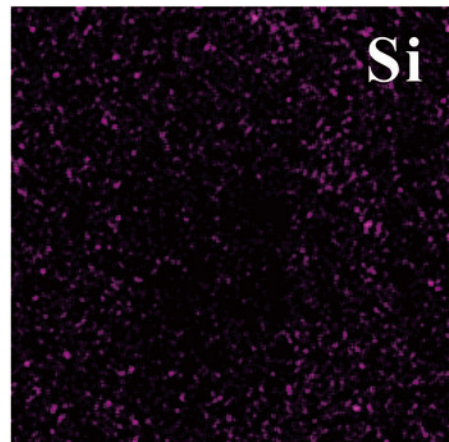
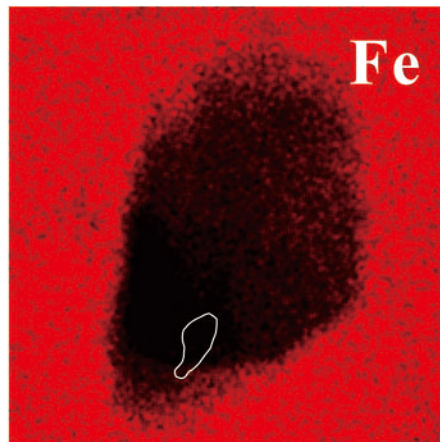
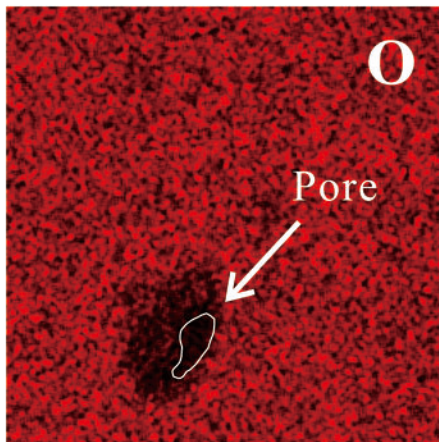
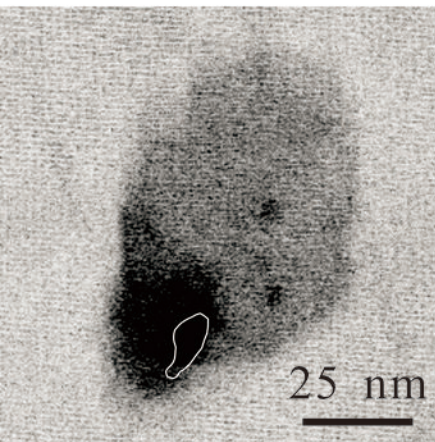


Figure 5





# Figure 6





# Figure 7

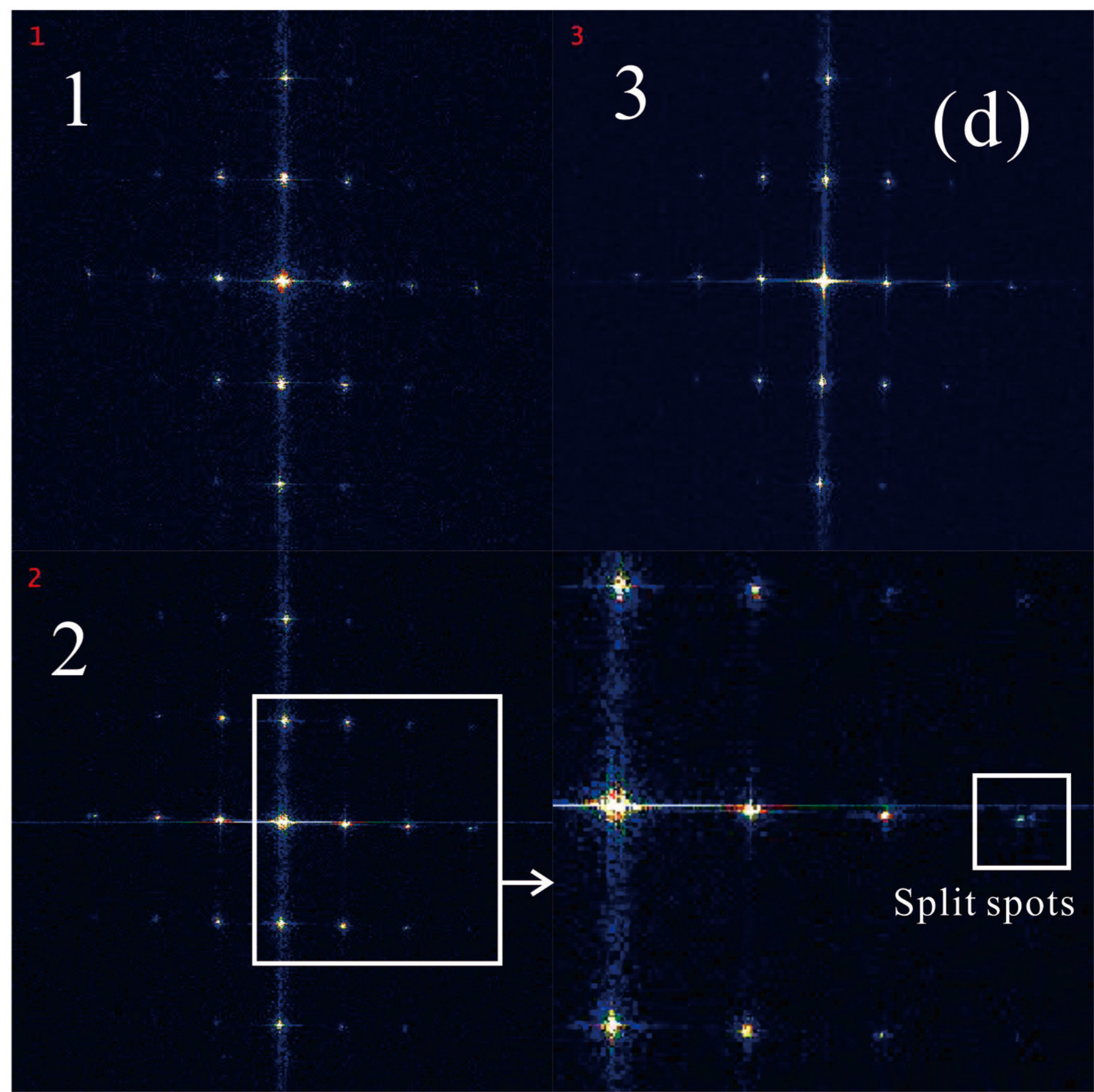
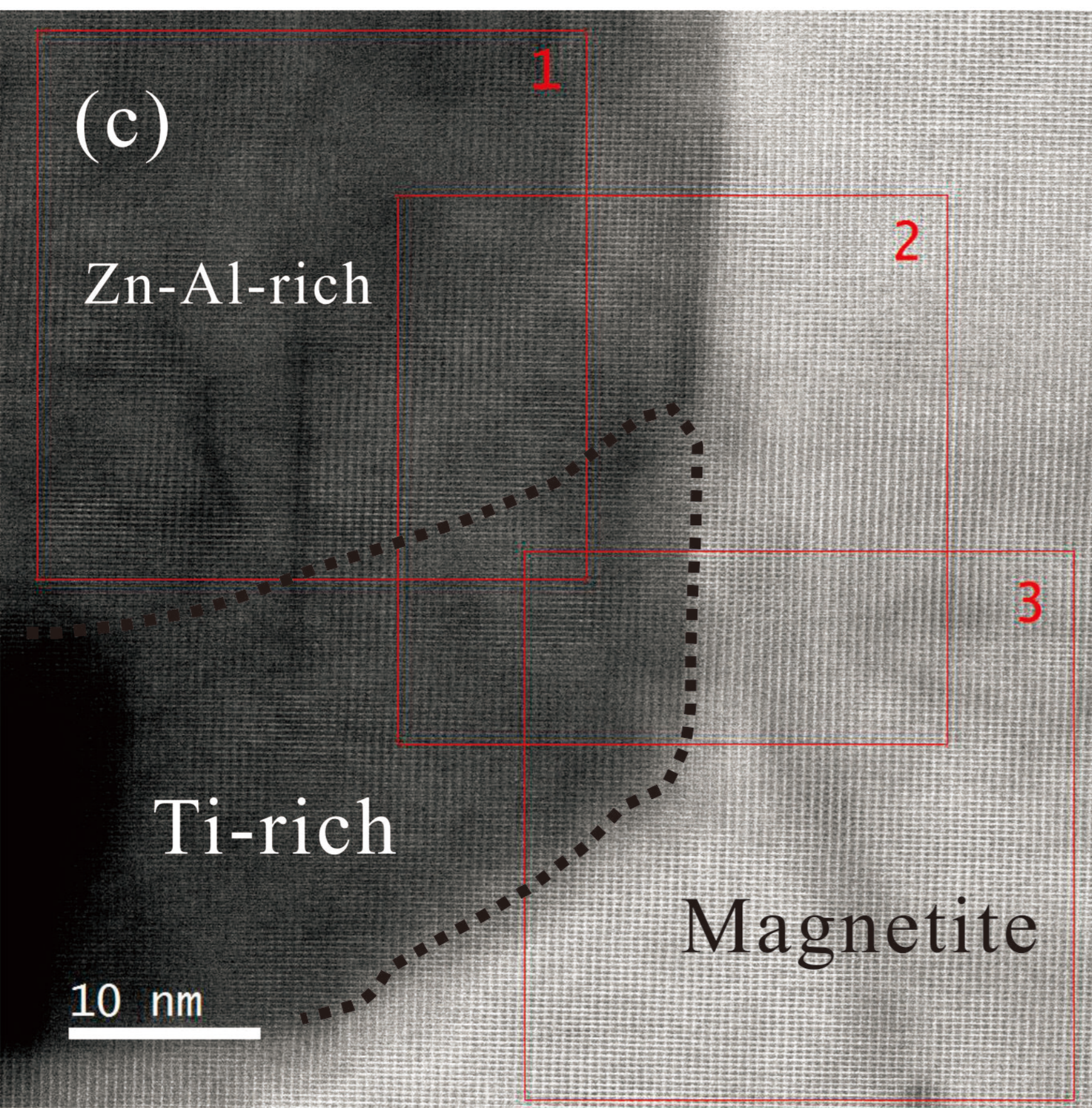
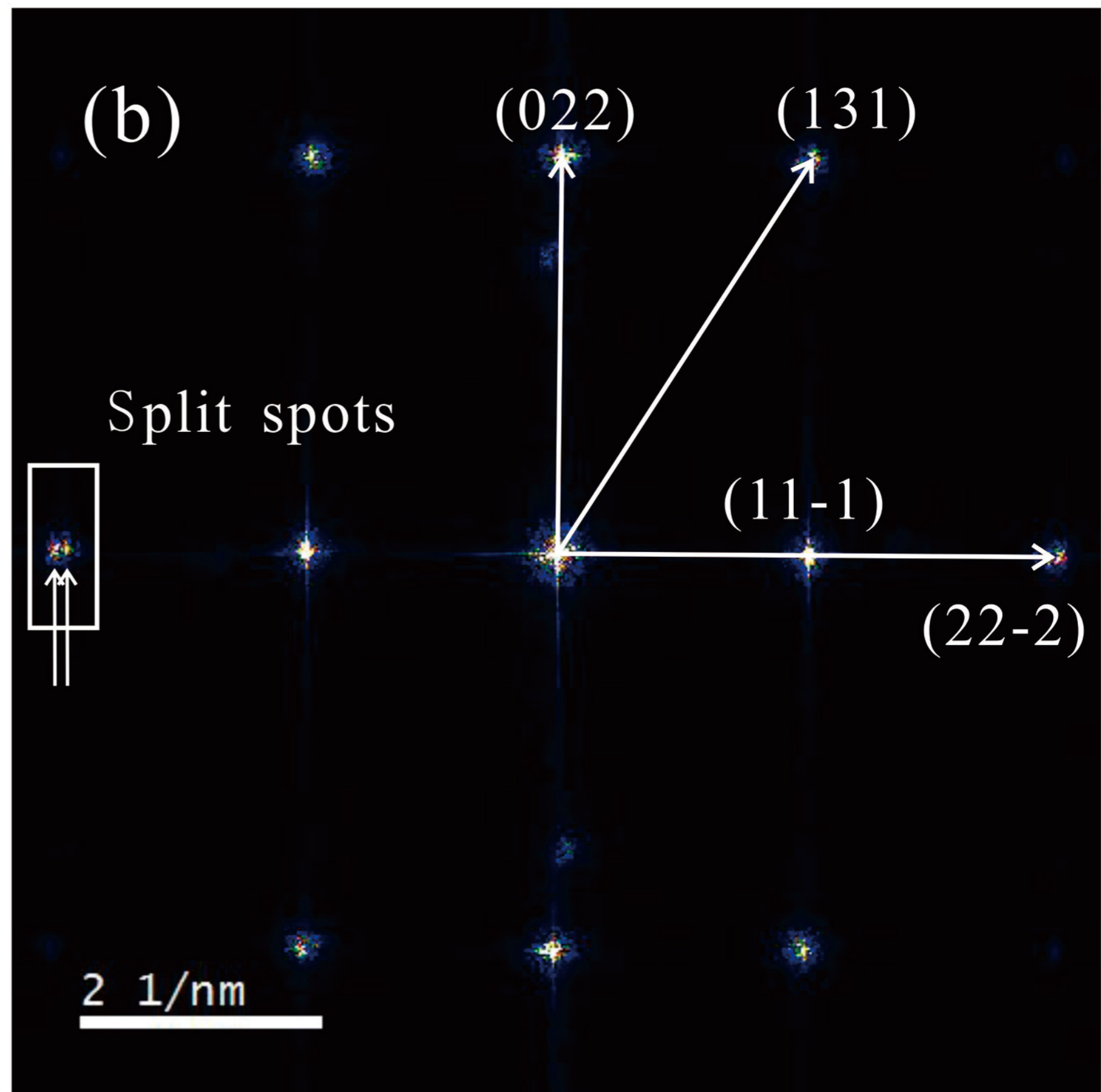
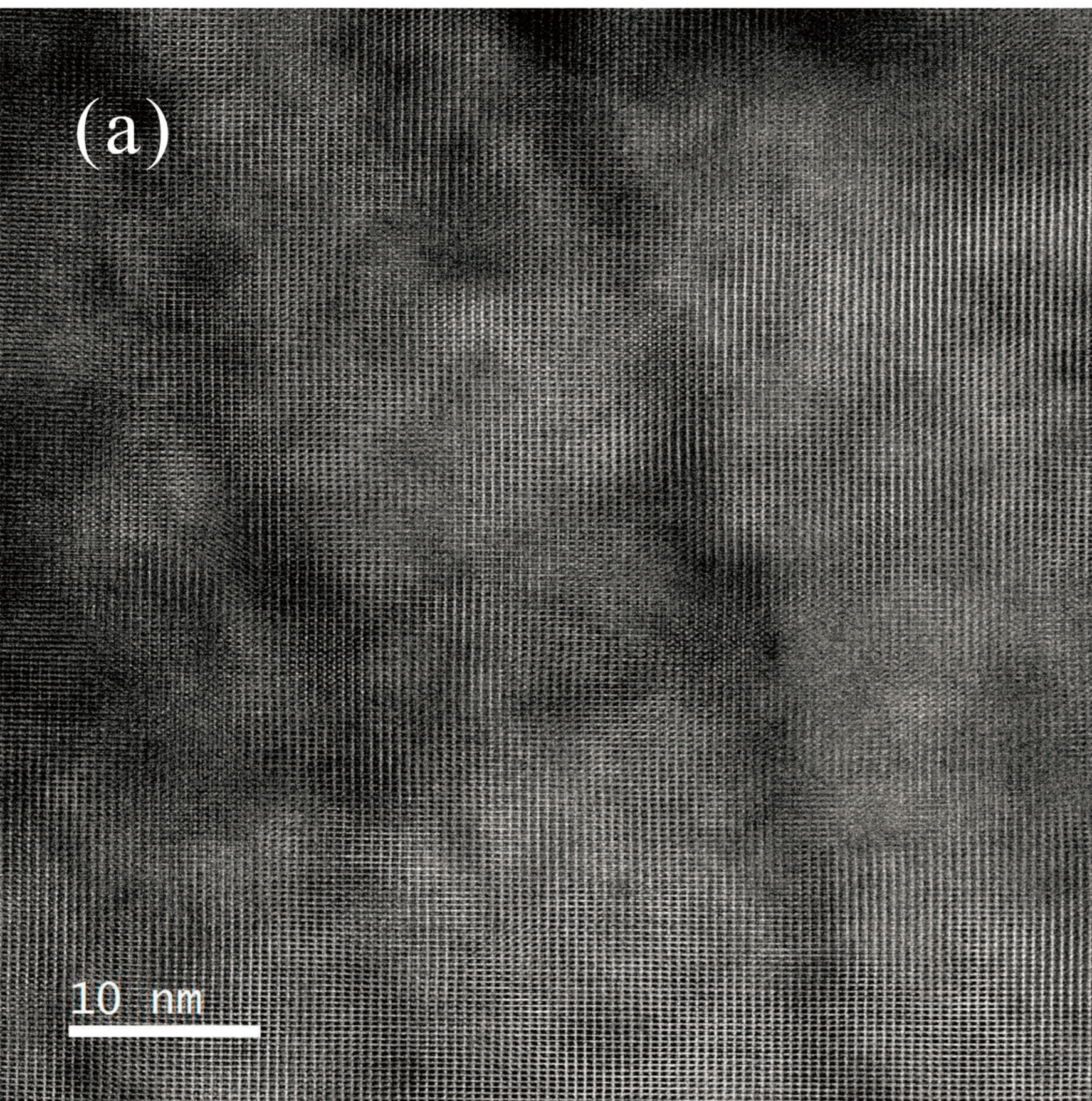




Figure 8

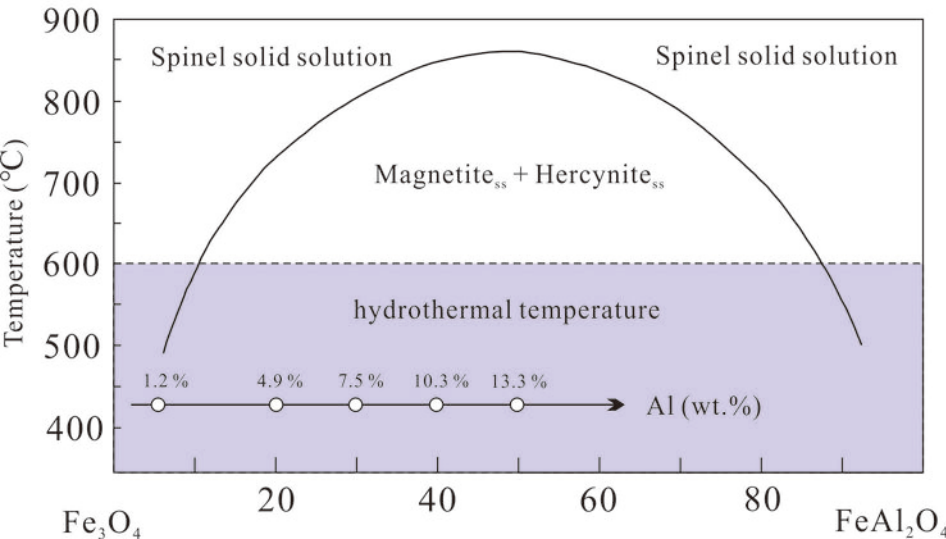
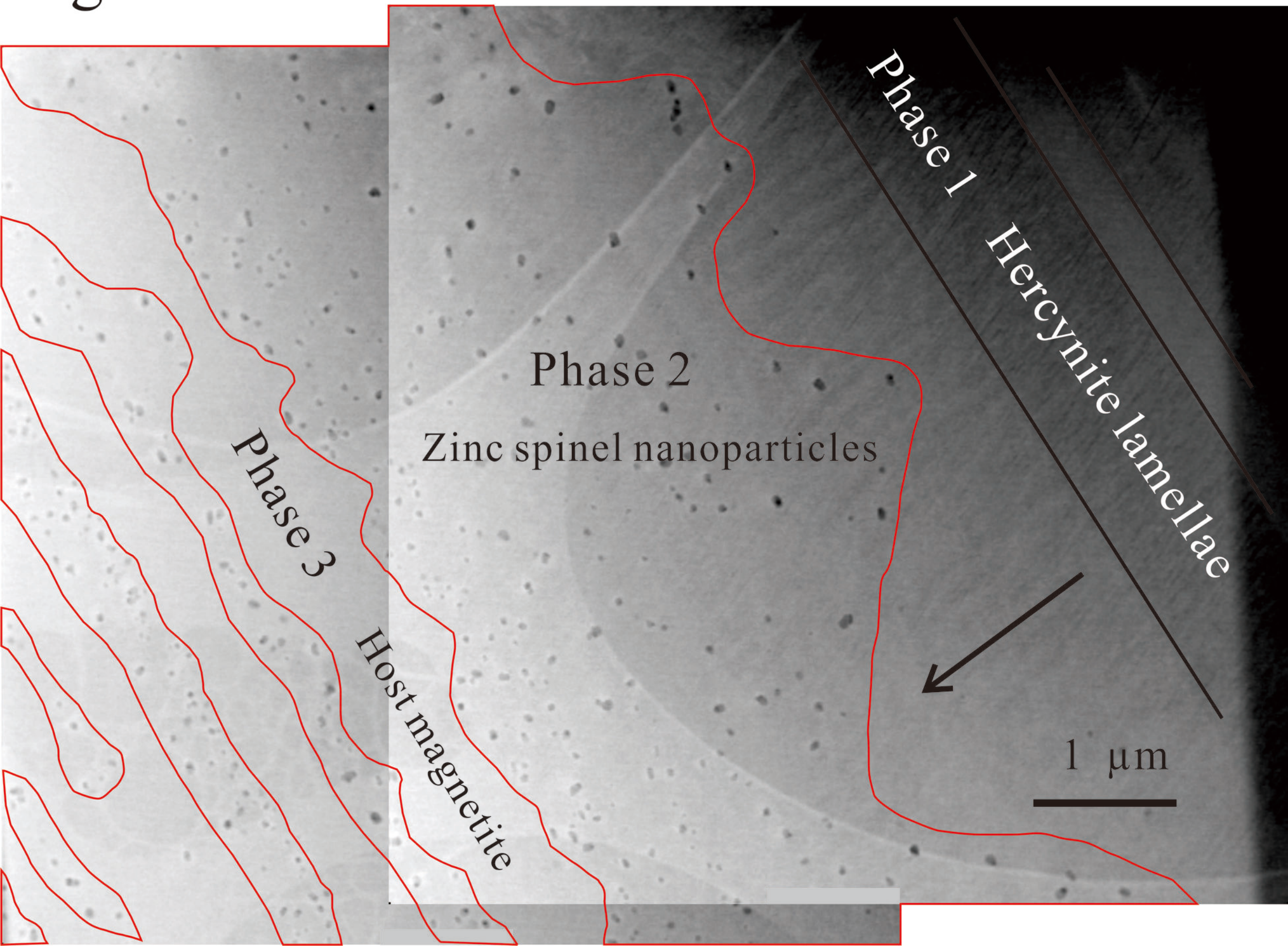
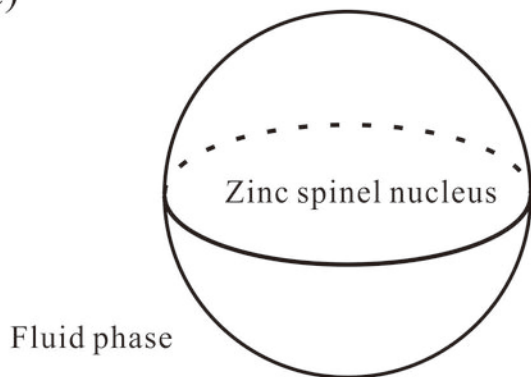


Figure 9



# Figure 10

(a)



(b)

