

1 **Revision 2**

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3 **High-pressure study of dravite tourmaline: Insights into the accommodating nature of the**
4 **tourmaline structure**

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12 **Abstract**

13 The high-pressure behavior of dravite tourmaline $[\text{Na}(\text{Mg}_3)\text{Al}_6(\text{Si}_6\text{O}_{18})(\text{BO}_3)_3(\text{OH})_3(\text{OH})]$
14 has been studied using luminescence spectroscopy and synchrotron based single-crystal
15 diffraction up to ~65 and ~24 GPa, respectively. Two emission bands associated with $\text{Cr}^{3+}/\text{V}^{2+}$
16 substitution are constant in energy up to ~9.0 GPa, and shift to longer wavelength at higher
17 pressures, suggesting that a change in compressional mechanism could occur at this pressure.
18 Single-crystal diffraction data show subtle changes in ring ditrigonality occur near 9.0 GPa which
19 could cause the observed change in luminescence. Near 15 GPa, a splitting of one of the
20 emission bands is observed, suggesting that a phase transition occurs at this pressure and that
21 two unique octahedral sites are present in the high-pressure phase. Hysteresis is not observed
22 on decompression, which indicates that this is a second order transition, and the high-pressure
23 structure appears to be metastable up to ~65 GPa. Single-crystal diffraction measurements
24 show that a phase transition from rhombohedral $R3m$ to rhombohedral $R3$ occurs at pressures
25 near 15.4 GPa. The high-pressure phase is characterized by a distorted Si_6O_{18} ring (e.g. the Si-Si-
26 Si angles deviate from 120°), and the Si, Al, O6, O7, and O8 sites of the low-pressure phase split,

27 implying that the high-pressure phase of tourmaline is a higher entropy phase. The large *X*-site
28 exerts the primary control on compressibility, and the substitution of larger cations into this site
29 will likely lower the pressure at which this transition occurs. Dravite tourmaline shows
30 anisotropic compression with the *c*-axis being more compressible than the *a*-axis. The pressure
31 volume data up to ~15.4 GPa were fit with 2nd and 3rd order Birch-Murnaghan equations of
32 state. We obtain a bulk modulus, $K_0 = 109.6(3.2)$ GPa, and a pressure derivative $K_0' = 4.6(8)$ GPa,
33 and with the pressure derivative set to 4, a bulk modulus of 112.0(1.0) GPa is derived.
34 Moreover, our high-pressure results show that massive overbonding of the *X* and *Y* sites can be
35 accommodated by the tourmaline structure. This unexpected result may explain the
36 extraordinary structural tolerance with respect to chemical substitution on the *X*, *Y*, and *Z* sites.

37 **Key Words:** dravite; High-pressure; Single-crystal diffraction, luminescence, phase transitions

38 **Introduction**

39 Tourmaline is a supergroup of minerals that is chemically complex and is divided into
40 different varieties based on chemical composition (Hawthorne and Henry 1999; Henry et al.
41 2011). This group of minerals is so complex and chemically variable that an entire
42 subcommittee on tourmaline nomenclature exists. These minerals are categorized using the
43 generalized tourmaline structural formula $XY_3Z_6(T_6O_{18})(BO_3)_3V_3W$ where *X* = Na, Ca, K; *Y* = Fe,
44 Mg, Mn, Al, Li; *Z* = Al, Fe, Mg, Cr; *T* = Si, Al; *B* = B; *V* = (OH), O; and *W* = (OH), O, F (Hawthorne
45 and Henry 1999; Henry et al. 2011). From a compositional perspective, most of the variability
46 occurs in the *X*, *Y*, and *Z* sites; more information on the classification of tourmaline supergroup
47 minerals can be found in Hawthorne and Henry (1999) and Henry et al. (2011). Tourmaline is
48 also highly sought after as a gemstone due to its high hardness and wide range of colors.

49 Tourmaline typically crystallizes in the rhombohedral crystal system with $R3m$ symmetry
50 (Hawthorne and Henry 1999; Henry et al. 2011), although other crystal systems have been
51 reported (Akizuri et al. 2001; Shtukenberg et al. 2007). It is typically considered a ring-silicate
52 with six corner-shared TO_4 tetrahedral sites that form a hexagonal ring, a nine-fold X -site, two
53 six-fold octahedral Y - and Z -sites, and a trigonal planar B -site (Figure 1). Although the T-sites are
54 dominantly occupied by Si, B and Li substitution into them has been reported (Rosenberg and
55 Foit 1979; Hughes et al. 2000). Interestingly, the first reported structure of tourmaline
56 presented the Si_6O_{18} ring as ditrigonal in symmetry (Hamburger and Buerger 1948). However, it
57 was later shown to be hexagonal in symmetry (Ito and Sadanaga 1951), and all subsequent
58 studies agree that the Si_6O_{18} ring is hexagonal at ambient conditions.

59 Tourmaline is the most widespread borosilicate mineral in natural rocks and the
60 dominant carrier of B in many rock types (Henry and Dutrow, 1996), and is found in subduction
61 zone environments (Nakano and Nakamura 2001; Bebout and Nakamura 2003; Ota et al. 2008).
62 B isotope systematics in tourmalines have been used to elucidate crust-mantle recycling
63 processes in subduction zones (Nakano and Nakamura, 2001). Petrologic studies have shown
64 that dravite tourmaline breaks down at ~ 7.0 GPa and ~ 900 °C (Werding and Schreyer 1996) and
65 ~ 3 -5 GPa and ~ 1000 °C (Krosse 1995) and that breakdown pressures and temperatures are
66 dependent on its precise composition (Ota et al. 2008). The decomposition products associated
67 with tourmaline breakdown are complex, but one decomposition product, phengite, is
68 important from a deep Earth boron and water cycling perspective; phengite could transport B
69 to depths >300 km (Domanik and Holloway 1996; Marschall et al. 2007). Higher pressure
70 studies (e.g. > 7.0 GPa) on tourmaline are scarce, and only two studies appear to have been

71 reported: Li et al., (2004) and Xu et al., (2016). Li et al. (2004) conducted high-pressure powder
72 diffraction experiments on schorl tourmaline $[\text{Na}(\text{Fe}_3)\text{Al}_6(\text{Si}_6\text{O}_{18})(\text{BO}_3)_3(\text{OH})_3(\text{OH})]$ up to 27.8
73 GPa, with no phase transitions being reported. Xu et al. (2016) reported high-pressure and
74 temperature powder diffraction experiments on uvite tourmaline
75 $[\text{Ca}(\text{Mg}_3)\text{MgAl}_5(\text{Si}_6\text{O}_{18})(\text{BO}_3)_3(\text{OH})_3(\text{OH})]$ up to ~ 18.4 GPa and 723 K and again, no phase
76 transitions were observed. These powder diffraction studies provide insights into how the unit
77 cell responds to compression, but they do not elucidate the positional changes of the individual
78 atoms in the unit cell (Li et al. 2004; Xu et al. 2016).

79 This is in contrast to other ring-silicates such as cordierite $[(\text{Mg}, \text{Fe})_2\text{Al}_3(\text{AlSi}_5\text{O}_{18})]$ and
80 beryl $[\text{Be}_3\text{Al}_2(\text{Si}_6\text{O}_{18})]$ which each show extensive high-pressure polymorphism (Prencipe et al.
81 2011; Miletich et al. 2014; Scheidl et al. 2014; Finkelstein et al. 2015; O'Bannon and Williams
82 2016b). Both cordierite and beryl undergo high-pressure phase transitions that involve
83 distortion of the Si_6O_{18} ring. The main structural difference between tourmaline and these
84 other ring silicates is that tourmaline does not have open channels in its structure like cordierite
85 and beryl. Thus, it is possible that the denser-packed configuration of the Si_6O_{18} rings in
86 tourmaline produces a structure that is more stable under compression than open channel ring
87 silicates like cordierite or beryl.

88 Tourmaline is also both pyroelectric and piezoelectric, and has been used in industrial
89 applications such as pressure gauges and air and water purifiers (Fron del 1948; Lameiras et al.
90 2010). Growing synthetic tourmalines >1 mm in diameter has not yet been achieved, and thus
91 industrial applications require natural tourmalines (Shekhar Pandey and Schreuer 2012). Thus,
92 despite its widespread use in industrial applications and as a geochemical probe, tourmaline's

93 high-pressure stability/polymorphism is poorly understood. Hence, our experiments are
94 oriented towards understanding the trace element behavior and the bulk structural response of
95 dravite tourmaline to compression by combining the results of luminescence and single-crystal
96 diffraction studies to provide an integrated picture of structural changes in dravite tourmaline
97 under compression.

98 **Methods**

99 *2.1 Sample characterization*

100 The natural light green gem quality tourmaline sample used in this study is from the
101 Lengenbach Quarry, Switzerland. We characterized the sample with single-crystal X-ray
102 diffraction, Raman, and luminescence spectroscopy, which all agree well with previous studies
103 (Hawthorne et al. 1993; Gasharova et al. 1997; Gaft et al. 2005). Ambient pressure single-crystal
104 X-ray measurements were conducted on Beamline 11.3.1 at the Advanced Light Source (ALS) at
105 Lawrence Berkeley National Lab in Berkeley, CA. We modeled our tourmaline as an endmember
106 dravite and our ambient structure refines with an R_1 of 1.85% when refined anisotropically and
107 2.99% when refined isotropically. Its crystal system, spacegroup and unit cell parameters are:
108 rhombohedral $R3m$, a -axis 15.9370(6) Å, c -axis 7.1749(3) Å, and volume 1578.19(14) Å³,
109 respectively. For the ambient structure refinement, we fixed the O1-H1 and O3-H3 bond
110 lengths to 0.958(8) and 0.972(2) Å, respectively. These bond lengths were determined from
111 previous neutron diffraction results on tourmaline (Gatta et al. 2014). Trace element
112 concentrations of dravite tourmaline were measured with a PhotonMachines Analyte 193H,
113 which is a 193-nm ArF excimer laser system coupled with a ThermoScientific ElementXR single-
114 collect or magnetic sector ICP-MS. The instrument was calibrated with a SRM 610 trace element

115 glass from NIST, and Al was used as the internal standard. The following trace elements were
116 measured (in ppm): Cr 215 (± 19), V 336 (± 32), Li 18 (± 4.0), Ti 474 (± 53), Ca 2183 (± 597), and K,
117 Fe, and Mn are below detection limits. The Subcommittee on Tourmaline Nomenclature (STN)
118 of the IMA Commission on New Minerals, Nomenclature and Classification recommends
119 nomenclature of tourmaline supergroup minerals (Novak et al., 2009; Henry et al., 2011;
120 Hawthorne and Dirlam, 2011). Using our single-crystal X-ray results combined with our trace
121 element results, and following STN recommendations, we determined that this tourmaline is of
122 the dravite variety. Dravite variety tourmaline has an ideal chemical formula of
123 $[\text{Na}(\text{Mg}_3)\text{Al}_6(\text{Si}_6\text{O}_{18})(\text{BO}_3)_3(\text{OH})_3(\text{OH})]$. Hence, the structures were refined isotropically and
124 modeled as endmember dravite tourmaline: this allowed direct comparisons of the results
125 obtained at each pressure.

126 *2.2 Luminescence spectroscopy*

127 Luminescence spectra were collected from 650-800 nm ($15380\text{-}12500\text{ cm}^{-1}$) with a
128 Horiba LabRAM HR Evolution Raman spectrometer with a spectrometer focal length of 800 mm.
129 Spectra were collected to a pressure of ~ 65 GPa and on decompression at 300 K using an
130 excitation wavelength of 532 nm. A symmetric type DAC with 200 μm type-Ia diamonds was
131 used in all high-pressure luminescence experiments, and single crystals of the sample and at
132 least two ruby spheres (Chervin et al. 2001) were loaded into the sample compartment. Either
133 4:1 methanol:ethanol mixtures or Ne was used as the pressure medium depending on the
134 target pressure range of the experiment. An Olympus BXF_M-ILHS microscope with a 50x long
135 working distance objective was used to focus the laser beam onto the sample. An 1800
136 lines/mm grating with a corresponding spectral resolution of $\sim 1\text{ cm}^{-1}$ (or, equivalently, ~ 0.05

137 nm) was utilized. Combinations of Gaussian and Lorentzian functions were fit to the
138 luminescence spectra with Horiba LabSpec6 software.

139 *2.3 High-pressure single crystal diffraction*

140 High-pressure single-crystal measurements were carried out using a (DESY) BX90
141 (Kantor et al. 2012) type diamond anvil cell (DAC) equipped with type Ia 500 μm culet Boehler-
142 Almax geometry diamond anvils with 85° angular access (Boehler and De Hantsetters 2004).
143 Rhenium gaskets with a 300 μm hole were used to contain the samples. Gasket holes were cut
144 using an Oxford Instruments laser mill at the Advanced Light Source (ALS), Berkeley, CA. Single
145 crystals of dravite tourmaline and two ruby spheres were loaded into the sample compartment
146 with neon as the pressure medium. The standard ruby fluorescence gauge was used to
147 determine pressure (Mao et al. 1986). Neon was loaded using the gas loading apparatus at the
148 ALS. Deviatoric stress becomes detectable above ~ 15 GPa in Ne, but the pressure gradients
149 remain small (Meng et al., 1993; Klotz et al. 2009).

150 High-pressure single-crystal X-ray diffraction measurements were conducted at
151 Beamline 12.2.2 at the ALS, which was recently described by Stan et al (2018). Measurements
152 were taken at various pressures up to ~ 23.6 GPa at room temperature. The DAC was mounted
153 on a Huber sample stage, and shutterless single-crystal diffraction data were collected on a
154 Perkin Elmer amorphous silicon detector using synchrotron radiation monochromated by
155 silicon(111) to a wavelength of 0.49594 Å (25 keV). Distance and wavelength calibrations were
156 done using a NIST single crystal ruby diffraction standard (Wong-Ng et al. 2001). Phi scans were
157 employed, from -40° to 40° and 140° to 220°, to measure across both diamonds with a 0.25°
158 image width for conventional data collection, and 1° to keep the most intense reflections

159 within detector dynamic range.

160 *2.4 Data processing*

161 Image masks, to avoid integrating signal from detector regions obscured by the DAC,
162 were created using the program ECLIPSE (Parsons 2010). The data were integrated using the
163 program SAINT v8.34A. A multi-scan correction for absorption was applied using the program
164 SADABS-2014/11. Structures were solved by dual space methods (SHELXS-97/ SHELXT) and
165 refined by full-matrix least-squares on F^2 (SHELXL-2014) (Sheldrick 2008) using the graphical
166 user interface *ShelXle* (Hübschle et al. 2011). All atoms in the room pressure structure and the
167 high-pressure structures were refined isotropically. Because of their small scattering cross-
168 sections, H atoms were not refined in the high-pressure structures. Moreover, evidence of Ne
169 diffusion into the tourmaline structure under high-pressure was not observed in our single-
170 crystal measurements: neither anomalous changes in volume nor excess electron density were
171 observed. Complete crystallographic information files (CIFs) for each structure can be found in
172 supplementary material, as well as a discussion on the selection of our unit cells.

173 **Results and discussion**

174 *Ambient pressure luminescence spectra*

175 The luminescence spectrum of dravite is characteristic of d^3 ions (e.g. Cr^{3+} , V^{2+}) in an
176 intermediate strength crystal field (Tanabe and Sugano 1954). Both sharp and broad emission
177 bands are observed at room temperature and 77 K at ambient pressures (Figure 2). Considering
178 the similar concentrations of Cr and V in our sample (e.g. Cr: 215 ± 19 ppm, V: 336 ± 32 ppm), it
179 is reasonable to assume that our steady state luminescence spectra contain overlapping bands
180 from the emission of both Cr and V. This is commonly observed in emerald, which is the Cr and

181 V rich variety of beryl (Ollier et al. 2015; O'Bannon and Williams 2016b). The intense and
182 relatively sharp emission bands are commonly referred to as the R-lines and are associated with
183 the spin-forbidden ${}^2E-{}^4A_2$ transition, with the 4A_2 state being the ground state and the 2E state
184 being split in non-cubic environments (Syassen 2008), and the broad band emission is assigned
185 to the spin-allowed ${}^4T_2-{}^4A_2$ transition (Sugano and Tanabe 1958; Burns 1993; Gaft et al. 2005).

186 At 300 K, three bands can be fit under the sharp emission bands and two can be fit
187 under the broad portion of the spectrum at room temperature. At 77 K, the broad thermally
188 populated 4T_2 -associated bands are not entirely quenched, and four bands can be fit under the
189 2E sharp emission bands. A detailed deconvolution of these sharp line emission bands at both
190 room temperature and 77 K is given in the supplementary material (Figure S1). Dravite has two
191 crystallographically unique octahedral sites, Al and Mg, and emission bands from both sites
192 would likely be observed in steady state luminescence spectra. The simplest way for Cr^{3+} and
193 V^{2+} to substitute into dravite is through isovalent substitutions (e.g. Cr^{3+} into Al^{3+} site and V^{2+}
194 into Mg^{2+}). The four deconvoluted components at 77 K thus can be attributed to two sets of R-
195 lines, one set from Cr^{3+} and one from V^{2+} . The Al site in dravite is less distorted than the Mg site
196 (Hawthorne et al. 1993, and this study), which would be expected to give rise to a narrower R-
197 line splitting. Given the likely relative intensities of R_1 and R_2 and the anticipated relative
198 splittings, we propose the following assignments: the Cr^{3+} R_1 band is likely located at 683.7 nm,
199 with its R_2 band at 681.4 nm, and the V^{2+} R_1 band is at 686.4 nm and its associated R_2 band at
200 679.8 nm. This results in an R-line splitting of $\sim 50\text{ cm}^{-1}$ for Cr^{3+} and $\sim 143\text{ cm}^{-1}$ for V^{2+} . For
201 comparison, ruby [$Al_2O_3:Cr$] has an Al-site that is close to an ideal octahedron with a slight
202 trigonal distortion, and has an R-line splitting of $\sim 29\text{ cm}^{-1}$ (Syassen 2008), while zoisite

203 [Ca₂Al₃Si₃O₁₂(OH)] which has a highly distorted Al-site has an R-line splitting of ~340 cm⁻¹
204 (Koziarska et al. 1994). The distortions of the Al and Mg sites in dravite fall intermediate
205 between the distortions of the Al sites in ruby and zoisite, so these R-line splittings are
206 plausible. To confirm the assignments of these emission bands, luminescence lifetime
207 measurements or variably substituted samples would be required.

208 *High-pressure luminescence spectra*

209 High-pressure luminescence spectra are shown in Figure 3. Under compression the
210 intensity of the R-lines decreases: this is likely primarily associated with the transition metal
211 absorption bands migrating away from our excitation wavelength of 532 nm. Moreover, the
212 disappearance of the broad ⁴T₂-associated band shows that a transition from an intermediate
213 to strong crystal field occurs in the first few GPa of compression, which is consistent with
214 previous observation in other Cr-bearing oxides (Dolan et al. 1986; de Viry et al. 1987;
215 Hommerich and Bray 1995; Grinberg and Suchocki 2007; O'Bannon and Williams 2016b). As at
216 300 K, it is difficult to fit four bands under the ²E region under compression, so we fit the
217 spectra with three bands: based on our 77 K assignments, these are associated with the R₁ and
218 R₂ peaks of Cr³⁺ and the R₁ peak of V²⁺. Near 6.0 GPa, the primary emission feature fit by three
219 bands clearly separates into individual peaks (Figure 3).

220 Of the original three bands that were fit at ambient pressure and temperature, two
221 show negligible pressure shifts--essentially by 0 nm/GPa up to ~9 GPa, while the other band
222 shifts by 0.50 ± 0.02 nm/GPa (-8.8 (± 0.3) cm⁻¹/GPa) up to the same pressure (Figure 4a). The
223 observation of effectively stationary emission bands under compression is without precedent
224 among Cr³⁺ emission in oxides (e.g., Dolan et al. 1986; Mao et al. 1986; de Viry et al. 1987;

225 Hommerich and Bray 1995; Grinberg and Suchocki 2007; O'Bannon and Williams 2016b). Above
226 ~9.0 GPa, the initially stationary bands begin to shift positively in wavelength, towards lower
227 energies (Figure 4b). The two bands that shift by 0 nm/GPa likely are associated with one
228 octahedral site while the other band is produced by substitution in the other octahedral site.
229 This is in accord with our assignment of the former two bands to emission from Cr³⁺
230 substituting into the Al³⁺ site.

231 Four separate experiments were conducted in the ≤ 10.5 GPa range to assess the
232 reproducibility of the unusual negligible pressure shift. It is completely reproducible, and no
233 hysteresis is observed on decompression from ~10.5 GPa. The ~0 nm/GPa pressure shift
234 suggests one of three possibilities. First, the crystallographic site that this band is emitting from
235 may not be changing in volume under compression. Second, there may be a complex trade-off
236 between electron density and volumetric change within the site. Third, an admixture of
237 electronic states, rather than a pure ${}^2E \rightarrow {}^4A_2$ transition, may generate the observed sharp-line
238 emission. As described in the single-crystal diffraction section below, there are no dramatic
239 structural "smoking guns" for this unusual pressure shift. However, there are some subtle
240 changes in the ring distortion that may be responsible. Admixture of states (probably with the
241 4T_2 state) is a possibility; however, the troubling aspect of this explanation is that a pressure-
242 induced shift in admixture must perfectly balance with the pressure shifts of the two states to
243 produce a net zero pressure shift. Stated another way, the separation of the two states, their
244 resulting admixture, and their pressure shifts must collectively sum to zero for ~9.0 GPa of
245 compression. This scenario is quite unlikely because of the dramatically different pressure-
246 dependences of the 4T_2 and 2E states (e.g., O'Bannon and Williams 2016a), but could be

247 testable with high-pressure lifetime measurements. A trade-off between changes in electron
248 density and volumetric compaction is possible but would require detailed insights into the
249 bonding environment/electron density within the octahedral site under compression.

250 The splitting between these two bands also does not change on compression,
251 suggesting that the site distortion is unchanged, as well. The change in slope at 9 GPa is, as
252 described below, likely not an indicator of a phase transition, but may represent a change in the
253 compression mechanism. Moreover, the other observed emission band shifts positively in
254 wavelength, consistent with the volume of this site decreasing with increasing pressure. Since
255 we cannot deconvolute the expected R_2 component of this band in this pressure range, it is
256 difficult to know from the emission spectra if this site becomes more or less distorted.
257 However, the peak width of the positively shifting band slightly narrows under compression to
258 ~ 9.0 GPa, which is an indicator that its site becomes less distorted under compression.

259 In the pressure range above 10 GPa, peak splitting is observed near 15 GPa, with new
260 emission bands observed; at pressures above 45 GPa, non-linear pressure shifts are observed
261 (Figure 3, 4b). No hysteresis is observed in the emission on decompression from 65 GPa, and
262 the Raman spectrum of the decompressed sample is identical to the starting material. Peak
263 splitting of emission bands under compression has been shown to be an indicator of phase
264 transitions (O'Bannon and Williams 2016a,b). This is particularly anticipated when multiple
265 octahedral environments are produced by a decrease in symmetry associated with a phase
266 transition. For example, in the ring silicate beryl, the Cr^{3+} R-lines split into two sets of R-lines
267 near 14.7 GPa (O'Bannon and Williams, 2016b) which agrees well with a theoretically predicted
268 ~ 15 GPa soft-mode transition (Prencipe et al. 2011), as well as high-pressure single crystal

269 diffraction studies which show that the Al-site splits into two crystallographically unique sites
270 near 15 GPa (M. Merlini, personal communication). The peak splitting observed in dravite
271 tourmaline under compression provides clear evidence that one of the octahedral sites splits
272 into two unique sites (Figure 5).

273 The additional new bands at longer wavelength that appear above 10 GPa (Figures 3,
274 4b) are likely neighbor lines that are due to paired Cr-Cr emission (e.g., O'Bannon and Williams,
275 2017); vibronic lines may also lie at these wavelengths, but these typically do not dramatically
276 exceed the zero-phonon lines in amplitude. The appearance of these new bands, and the
277 progressive increase in their intensity with pressure may reflect: (1) the intensity of the broad
278 4T_2 transition has decreased substantially by this pressure and alternate mechanisms for
279 emission are enhanced; (2) a change in the nearest neighbor configuration due to the splitting
280 of one of the octahedral sites into two unique sites that produces configurations that favor
281 energy transfer into and/or the magnetic interactions that generate emission from pairs;
282 and/or (3) a pressure-induced shift in absorption that results in preferential excitation of pair-
283 line emissions (e.g., O'Bannon and Williams 2017). Within the tourmaline structure, there are a
284 wide range of possible prospective pair-wise magnetic interactions that could occur between
285 substituents in the octahedral sites through an intervening anion, and it is well-known that the
286 strength of such exchanges depends strongly on both the inter-ion distance and on the angle
287 (e.g., McCarthy and Gudel 1988). Indeed, such pair-wise interactions between transition metals
288 have long been recognized to be important within the optical spectrum of tourmalines
289 (Mattson and Rossman 1987). Therefore, while the exact origin of the new bands and the shift

290 in intensity above 10 GPa within the emission spectra remain unclear, it is probable that they
291 can be attributed to pressure-enhanced pair-line emissions.

292 Thus, the major effects of compression on the luminescence spectrum are the change in
293 pressure shift near 9.0 GPa and the peak splitting and increased intensities of new peaks
294 observed at ~15 GPa. The lack of hysteresis on decompression from ~65 GPa indicates that any
295 phase transition associated with these changes is second order. We do not believe that the
296 change observed at 9.0 GPa is produced by a pressure-induced structural phase transition but is
297 likely generated by either a change in compression mechanism or a progressive change in the
298 electron density associated with the site. The changes observed at ~15 GPa are, however,
299 consistent with a phase transition that involves a subtle distortion of the overall structure of
300 dravite tourmaline that splits the Al-site into two unique sites. By analogy to high-pressure
301 phase transitions in cordierite and beryl (Prencipe et al. 2011; Miletich et al. 2014; Finkelstein
302 et al. 2015; O'Bannon and Williams 2016b), the structural change likely involves a modest
303 distortion of the Si₆O₁₈ ring away from hexagonal symmetry. Importantly, the overall topology
304 of the structure appears to be preserved, and a lowering of symmetry from rhombohedral is
305 unlikely. The presence of such a modest structural change implies that dravite tourmaline's
306 metastability under room temperature compression to ~65 GPa is generated by the topology of
307 the structure that stabilizes the Si₆O₁₈ ring at high pressures. Moreover, a structural transition
308 that involves a modest distortion of the Si₆O₁₈ ring with no change in crystal system would be
309 difficult to constrain with powder diffraction techniques (e.g., Li et al. 2004; Xu et al. 2016).

310 *High-pressure single-crystal diffraction*

311 To characterize the changes observed using luminescence spectroscopy, single crystal X-
312 ray diffraction measurements were done at various pressures up to ~23.6 GPa. Dravite
313 tourmaline indexes to a rhombohedral unit cell across the pressure range of the X-ray
314 measurements. Structures solved as rhombohedral $R3m$ up to ~15.4 GPa, and above this
315 pressure the space group changed to $R3$. In the following sections, we discuss the equation of
316 state (EoS) for the low pressure phase of dravite tourmaline, and the high-pressure crystal
317 structure of this material.

318 *Dravite tourmaline EoS*

319 Dravite tourmaline shows highly anisotropic compression (Figure 6), with the c -axis
320 being substantially more compressible than the a -axis, which is in good agreement with the
321 previous study of uvite tourmaline reported by Xu et al. (2016). Interestingly, Liu et al. (2004)
322 report that the c -axis of schorl tourmaline is less compressible than the a -axis, which is opposite
323 of what is observed in dravite and uvite tourmaline. But, the lattice parameters and unit cell
324 volumes reported by Liu et al. (2004) are quite scattered. For clarity, we do not include these
325 results in Figure 6, but they are included in supplementary Figure S2. The reason for the scatter
326 in their data, and the difference in compressibility of the a - and c -axes is not clear, but could be
327 a result of non-hydrostatic conditions of methanol:ethanol mixtures above ~10.5 GPa. We fit
328 our pressure volume data with both 2nd and 3rd order Birch-Murnaghan equations of state
329 (Table 1) using the EOSFit7GUI (Gonzalez-Platas et al., 2016), and our results are shown in Table
330 2. We used a fully weighted fit, and the room pressure and temperature volume determined in
331 our experiments is V_0 . An F - f plot based on the Birch-Murnaghan EoS fit of the volume data is
332 shown in Figure S3. Our results indicate that dravite tourmaline is more compressible than uvite

333 tourmaline. For comparison, the uvite sample measured by Xu et al. (2016) is not an
334 endmember but has ~10 % Na in the X-site, and pure uvite is likely less compressible than their
335 measurements. The relatively large value for K_0 reported for schorl tourmaline by Li et al.
336 (2004) does not appear compatible with our results, nor with those of Xu et al. (2016).

337 *High-pressure crystal structures*

338 Structures were solved at various high pressures, and results of the structure
339 refinements are shown in Table 3. The site volumes as a function of pressure show that the Na
340 site is the most compressible (Figure 7). The Si site is the least compressible, while the two
341 octahedral sites behave essentially identically to one another and are intermediate in their
342 compressibility between the Si and Na sites. Clearly, neither octahedral site volume is constant
343 under compression. So, the unusual observation of a negligible pressure shift of two of the
344 luminescence bands cannot be explained via a site volume argument alone. A more nuanced
345 explanation for the shift of the energy levels, involving a shift in covalency/electron density of
346 the site that counterbalances the volumetric compaction, appears necessary. The anisotropic
347 compression of dravite tourmaline can be straightforwardly explained by the location of the Na
348 site in the dravite structure and its large volume change under compression (Figure 7).
349 Speculatively, the large volume change in the Na site could produce a pressure-induced
350 reduction in the electron density around the Al-sites, such that their electron density remains
351 fairly constant under compression up to ~9.0 GPa.

352 We calculate the distortion parameters for the octahedral and tetrahedral sites as a
353 function of pressure (Supplementary Figure S4). Interestingly, the quadratic elongation (QE)
354 and angle variance (AV) of the Mg site both decrease under compression, indicating that this

355 site becomes less distorted under compression: this is in accord with our inference based on
356 the width of the emission peak that we have assigned to this site. The Al site distortion changes
357 by very little, and the Si site becomes slightly more distorted under compression. The Al-site
358 distortion change is negligible, which also provides confirmation for our assignment of the two
359 emission bands with invariant positions that show no change in their splitting up to ~9.0 GPa to
360 this site.

361 The X-site in tourmaline (the Na site in dravite) is typically considered a 9-fold
362 coordinated site. Among the bond lengths associated with this site as a function of pressure
363 (Figure 8), there are two unique bonds between the Na and the Si₆O₁₈ ring (O5 and O4), and
364 there is one unique bond between the Na and the Mg octahedra (O2). There is also a potential
365 10th coordinating oxygen atom (O1), and the distance to this oxygen atom is also shown in
366 Figure 8. The O1 atom is, however, one of the two proton sites in tourmaline, although the H1
367 site has a very low reported occupancy (Gatta et al. 2014). F substitution also occurs only at the
368 O1 site, and this site is influenced by cation occupancy of the X and Y sites (Henry and Dutrow
369 2011). Hence, O1 is not simply an oxygen site that appears to become bonded to the Na atom
370 at high-pressure: rather, it is a site that is partially occupied by O, OH, and F. Interestingly, the
371 two unique Na-O bonds to the Si₆O₁₈ ring show different behaviors. Under compression to
372 ~15.4 GPa, the Na-O4 bonds change by ~0.05 Å while the Na-O5 bonds change by ~0.25 Å. Over
373 the same pressure range, the Na-O2 bonds change by ~0.11 Å. In this sense, the Na site appears
374 to undergo the bulk of its compaction through six neighbors: three bonds into the Si₆O₁₈ ring
375 and three bonds into the Mg octahedra. This pseudo-six-fold behavior, with compacting

376 connections to both the Mg polyhedra and the Si-ring, also sets up a natural way to distort the
377 Si₆O₁₈ ring away from hexagonal symmetry.

378 The B-O distances are less well-resolved and change very little under compression but
379 show a trend towards being more equidistant up to ~15.4 GPa (Supplementary Figure S5). In
380 the high-pressure phase, B is surrounded by three crystallographically unique oxygen atoms (O8
381 splits into O8a and O8b). There is no indication that the BO₃ groups deviate from a planar
382 configuration up to at least 23.6 GPa, which is expected given the space groups of the low- and
383 high-pressure phases.

384 *Compression mechanism*

385 The compression mechanism of tourmaline involves a complex interplay of bond length
386 shortening and oxygen atom displacement/rotation relative to their neighbors. Under
387 compression the Y-site (MgO₆) volume decreases through Mg-O bond shortening and the
388 distortion parameters of this site also decrease. As the volume of the Y-site decreases, the
389 Si₆O₁₈ ring becomes more puckered due to the displacement of the O6 atom towards the
390 threefold rotation axis (Figure 9a). This increase in ring puckering has been observed in
391 tourmaline when smaller cations are substituted into the Y-site. Bosi and Lucchesi (2007) report
392 that the configuration of the six-membered ring is strongly affected by the average <Y-O>
393 distance and, as <Y-O> decreases due to the substitution of smaller cations, the tetrahedra
394 rotate around the O4-O5 edge and the O6 atom is displaced towards the threefold axis which
395 produces puckering of the tetrahedral ring. Thus, chemical substitution of smaller cations into
396 the Y-site and pressure appear to have the same effect on ring puckering.

397 As ring puckering increases under compression the O7 atom rotates towards the Z-site
398 (AlO₆), which shortens the Al-O7 bond, and compresses the Z-site. This displacement of the O7
399 atom contributes to the increase in observed ring crimping under compression (Figure 9b), and
400 the increase in ring crimping changes the O7-O8 edge distance. Crimping is defined as $\Delta Z =$
401 $[\frac{z_{O5} - (z_{O4} + z_{O7})}{2}] * c$, where z is the coordinate of the oxygen atoms and c is the length of the c-
402 axis (Gorskaya et al. 1982). From a chemical substitution perspective, it has been shown that
403 substitution into the X-site primarily affects the crimping and ditrigonality of the ring and
404 distortion of the tetrahedra, while substitution into the Y-site primarily affects ring puckering
405 (Foit, 1989). Ditrigonality increases across the pressure range of this single-crystal study (Figure
406 9c). It is defined as $\delta = (r_1 - r_s) / r_s$, where r_1 and r_s are the distances from O4 and O5,
407 respectively, to the threefold axis (Barton 1969). Interestingly, the distances from the three-fold
408 axis to O4 and O5 as a function of pressure (Supplementary Figure S6) indicate that there is a
409 change in the pressure dependence of the O4 to threefold axis distance near 9.0 GPa. This
410 change in compression mechanism may initiate the change in luminescence observed near 9.0
411 GPa (Figure 4). Under compression, the Na site moves towards the ring, which also contributes
412 to the increase in ring crimping and ditrigonality. Moreover, the overall distortion of the Z-site
413 is primarily controlled by the size of the Y-site cation (Foit 1989). Therefore, it is not surprising
414 that the distortion parameters of the Z-site show very little change under compression. Above
415 ~15.4 GPa, tetrahedral rotation and ring puckering continue to increase while ring crimping
416 decreases above the transition and then remains almost constant to the highest pressure of this
417 study.

418 *High-pressure phase transition*

419 At pressures above 15.4 GPa, the structure of dravite tourmaline is solved as
420 rhombohedral $R\bar{3}$. The only difference in symmetry associated with this transition is the loss of
421 a mirror plane in the high-pressure structure. The net result is that the Si, Al, O6, O7, and O8
422 sites all split into unique sites (Figure 10), the volume of the open space in the structure by the
423 X-site has decreased significantly by ~ 15.4 GPa, and the effective coordination of Na increases
424 (Fig. 8) which suggests that the high-pressure phase is a higher entropy phase (Navrotsky, 1980).
425 Since the 3-fold rotation axis is preserved across this phase transition there is no change in the
426 Mg, Na, or B sites. Importantly, this subtle crystallographic change is completely consistent with
427 our luminescence observations of a second order phase transition occurring near ~ 15.4 GPa.
428 Additionally, our assignment of the emission band that splits under compression to the Al site is
429 robust since this site splits into two unique sites above ~ 15.4 GPa. Interestingly, the slight
430 distortion of the Si_6O_{18} ring preserves the overall topology of the tourmaline structure, while
431 providing a new deformation mechanism for the high-pressure phase. A plot of the Si-Si-Si
432 angles shows that this angle deviates away from 120° above ~ 15.4 GPa and increases with
433 increasing pressure up to ~ 23.6 GPa (Figure 11a). There is also a discontinuous shift in
434 tetrahedral rotation, as well as a change in the pressure dependence of ring puckering and
435 crimping above ~ 15.4 GPa (Figures 9b, 11). The deformation mechanism of the high-pressure
436 phase primarily involves tetrahedral rotation and changes in the Si-Si-Si angles while ring
437 crimping and puckering play less of a role. These subtle changes in ring distortion demonstrate
438 the subtlety of this phase transition, and why it would be very difficult to resolve this type of
439 phase transition with high-pressure powder diffraction techniques. Our luminescence data

440 show that this high-pressure phase remains metastable under room temperature compression
441 up to at least 65 GPa.

442 Speculatively, the high-pressure transition appears to be driven by steric/geometric
443 effects caused by the decrease in volume of the Y-site and displacement of the X-site towards
444 the Si₆O₁₈ ring. As the Y-site volume decreases, the O6 atom is displaced towards the threefold
445 axis causing ring puckering, which displaces the O7 atom towards the Z-site (Al), causing the Al-
446 O7 distance to shorten which increases ring crimping. This causes a change in the tetrahedral
447 rotation of the SiO₄ tetrahedra, which breaks the mirror plane symmetry for O6 and O7. This
448 change in rotation induces an additional twist on the MgO₆ and AlO₆ octahedra, which in turn
449 breaks the mirror symmetry of the O8 atom.

450 *Na site coordination change?*

451 The possible coordination change of the Na site merits discussion, as the topology of the
452 tourmaline structure shows that the incipient Na1-O1 bond (Figure 8) provides the only
453 straightforward means to further polymerize this structure. The distance to the potential 10th
454 oxygen decreases by ~0.43 Å to 15.4 GPa, and the pressure dependence of its distance is
455 constant above this pressure. Multiple phenomena could be at work here in changing the
456 behavior of the Na-O bonds above the phase transition: (1) Na1 and O1 behave like they are
457 bonded near the phase transition pressure (e.g. a coordination change of Na takes place); (2)
458 repulsion increases between the O1 and O4, and O5 atoms; (3) an increase in repulsion
459 between H1 and Na1 occurs near 15.4 GPa, which results in a shortening of the O1-H1 bond,
460 and/or a change in H-bonding configuration; and/or (4) the compression mechanism of the
461 high-pressure phase is different than the low pressure phase.

462 Considering the mixed occupancy of the O1 site (e.g. Gatta et al., 2014; Henry and
463 Dutrow, 2011; Bosi, 2018), it is unlikely a bond is forming between Na and O1. A plot of O-O
464 distances reveals that there are no systematic trends that would suggest that the change in
465 pressure dependence of the Na-O1 distance is due to an increase in O-O repulsion
466 (supplementary Figure S7). It has been reported that an increase in size of the X-site cation
467 results in an expansion of the X-site coordination polyhedron, which impinges on the hydroxyl
468 at the O1 site, leading to a shortening of the O1-H1 bond (Berryman et al., 2016). It is possible
469 that above ~15.4 GPa, an increase in Na-H repulsion occurs that results in a shortening of the O-
470 H bond which changes the pressure dependency of the Na1-O1 distance. However, due to the
471 low reported occupancy of H1 this seems unlikely. As discussed above, ring puckering and
472 crimping also behave differently in the high-pressure phase, indicating that compression
473 mechanisms shift in the high-pressure phase. Thus, a change in compression mechanism is the
474 most plausible explanation for the change in Na1-O1 pressure dependence that is observed
475 above ~15.4 GPa.

476 *Bond Valence Analysis*

477 We quantitatively analyze the stability of the dravite tourmaline structure using the
478 bond valence approach (Brown, 2002). The bond valence approach has been extensively
479 deployed to examine the structural changes in tourmalines due to chemical variability at
480 ambient conditions (Fortier and Donnay, 1975; Hawthorne et al., 1993; Hawthorne, 1996;
481 Hawthorne, 2002; Bosi and Lucchesi, 2007). We first use the relative deviation of the bond
482 valence sum from the ideal value ($(\sum(S_{ij}) - v_i)/v_i$) for all cations and anions to quantify the steric
483 stress of the structure at ambient condition. Here, S_{ij} is the bond valence between atoms i and j ,

484 and v_i is the absolute value of atom i 's valence. The bond valence sums are shown in Figure 12
485 on the ordinate (i.e. the zero-pressure values). It is striking that even at ambient conditions, the
486 Mg cation is overbonded by almost 20 %. This is due to the MgO_6 octahedron sharing two edges
487 in cis-conformation with the AlO_6 octahedron. The relaxed cation-oxygen distances for Al^{3+} and
488 Mg^{2+} differ by about 10% (1.9 vs 2.1 Å). Thus, each MgO_6 shares two edges with an AlO_6
489 octahedron, whereas each AlO_6 octahedron shares one edge with a MgO_6 octahedra.
490 Consequently, Mg shows an almost 20% overbonding while AlO_6 is underbonded, albeit to a
491 smaller extent (~ 3%).

492 In terms of the bond valence sums as a function of pressure (Figure 12), two features
493 stand out: Na (which at ambient pressure is slightly underbonded) increases its overbonding by
494 a remarkable 70%. Mg, which starts out at almost 20% overbonded increases to about 40% by
495 15 GPa, at which pressure it then levels out. Notably, Na and Mg are the two cations with the
496 weakest individual bonds. All other cations show a shallow increase in overbonding, which
497 increase moderately to a maximum of 20% at the highest pressures. The same general increase
498 holds for the oxygen atoms, which are all tightly bunched together at values below 20 % (we
499 disregard O1 and O3 here because the poorly constrained hydrogen positions make their bond
500 valence sums ill defined).

501 The weakly bonded Na and Mg cations seem to be dramatically overbonded at high
502 pressures, without affecting significantly the stability of the tourmaline structure (beyond the
503 modest descent of symmetry near 15 GPa). This stability of the structure may be rooted in the
504 minor overbonding of the oxygen atoms, by no more than 20 % up to the highest pressure of
505 ~24 GPa. Consequently, we hypothesize that this high-pressure stability of the tourmaline

506 structure is closely tied to its extraordinary structural tolerance with respect to chemical
507 substitutions on the X, Y, and Z sites. The bond network of tourmaline seems to have the ability
508 to balance the modest overbonding of the oxygen atoms which hold the framework together,
509 in spite of the significant overbonding on the most weakly bonded metal ions.

510 Indeed, the subtle mirror plane breaking phase transition observed above ~15.4 GPa is
511 not reflected in the bond valence analysis. Instead, this phase transition is likely due to a
512 steric/geometric effect caused by a complex interplay of bond length shortening and oxygen
513 atom rotation/displacement which alter ring puckering, ring crimping, tetrahedral rotation, and
514 ditrignality. As the ring distortion parameters change under compression, mirror plane
515 symmetry is broken for O6, O7, and O8. Hence, there does not appear to be an obvious
516 villain(s) that drives the transition from $R3m$ to $R3$; it is perhaps best viewed as a cooperative
517 instability of the (initially) hexagonal ring configuration, via its interactions with the Y (Na)
518 cation.

519 **Implications**

520 Other compositions of tourmaline that are rhombohedral $R3m$ at room temperature
521 and pressure will likely behave similarly, and undergo a similar transition, under compression.
522 The main difference is likely to be the pressure at which the $R3m$ to $R3$ transition occurs.
523 Tourmaline is a complex mineral group, and the high-pressure behavior of tourmalines with
524 extensive Li, Cr, or V substitution into the Y and Z sites, Pb substitution into the X-site, or Al, Li
525 substitution into the T-sites could be quite different than that of tourmalines with
526 predominantly Al and Mg in their Y and Z sites. Nevertheless, the main chemical control over
527 the pressure at which this transition occurs is likely the identity of the cation in the X-site. Here,

528 simple systematics would indicate that a larger cation would generate this distortion at lower
529 pressure conditions. Thus, the pressures of this transition in uvite tourmaline
530 $[\text{Ca}(\text{Mg}_3)\text{MgAl}_5(\text{Si}_6\text{O}_{18})(\text{BO}_3)_3(\text{OH})_3(\text{OH})]$ and maruyamaite tourmaline
531 $[\text{K}(\text{MgAl}_2)(\text{Al}_5\text{Mg})(\text{BO}_3)_3(\text{Si}_6\text{O}_{18})(\text{OH})_3\text{O}]$ are expected to be higher and lower than in dravite
532 tourmaline, respectively. Moreover, our high-pressure results show that massive overbonding
533 of the X and Y sites can be accommodated by the tourmaline structure. This unexpected result
534 may explain the extraordinary structural tolerance with respect to chemical substitution on the
535 X, Y, and Z sites. Hence, any over/underbonding due to chemical substitutions is balanced by
536 the topology of the tourmaline structure, and perhaps in particular the oxygen atoms.

537 The crystallization conditions of metamorphic tourmaline are complex, but one means
538 of crystallization involves consuming B that is released during clay and mica mineral
539 dehydration reactions (Ota et al. 2007). Hence, tourmaline is likely present in subduction zones
540 where sedimentary rocks are being subducted. Petrologic studies have shown that dravite
541 tourmaline breaks down at ~ 7.0 GPa and ~ 900 °C (Werding and Schreyer 1996) and ~ 3 -5 GPa
542 and ~ 1000 °C (Krosse 1995). However, tourmaline stability is highly dependent on
543 compositional variations (e.g. its $\text{Mg}/[\text{Fe}+\text{Mg}]$ ratio, Na, Al and B contents), coexisting fluids and
544 the host rock (Henry and Dutrow, 1996; Dutrow et al., 1999; Schreyer, 2000; Ota et al. 2008).
545 The higher pressure/lower temperature stability of dravite tourmaline is not well constrained
546 (see figure 2b in Ota et al. 2007). Moreover, tourmaline has a greatly diminished stability range
547 in SiO_2 saturated systems with common pelitic compositions, and even in relatively cold
548 subduction zones, tourmaline breaks down by ~ 150 km depth (Ota et al. 2008). Hence, the
549 commonly invoked “cold subduction zone” argument for conducting high-pressure room

550 temperature experiments may be invalid in the case of tourmaline. A complete understanding
551 of the host rock chemistry, redox conditions, and pressure/temperature conditions are required
552 to truly constrain the stability field of not only tourmaline supergroup minerals, but other
553 subduction zone related mineral phases. The stability fields of dravite tourmaline and other
554 varieties of tourmaline are not well constrained, and neither the onset pressure of the
555 symmetry-lowering transition in other varieties of tourmaline, nor its temperature-dependence
556 is known. Thus, it is unclear whether the high-pressure phase of dravite or other compositions
557 of tourmaline would be found at depth within subduction zones.

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741 Figure 1. Our room pressure crystal structure diagram of dravite tourmaline. (a) looking down
742 the *c*-axis with oxygens removed for clarity. (b) oblique view showing the configuration of the X-
743 site within the Si₆O₁₈ ring. Modified after Hawthorne and Dirlam (2011). Diagrams were
744 generated in CrystalMaker[®] v8.7.6.

745 Figure 2. Steady state luminescence spectra of dravite tourmaline at 300 and 77 K using 532nm
746 excitation. Note that the thermally populated ⁴T₂ bands are not entirely quenched at 77K. The
747 modulation in the broad band emission band at 300 K is an artifact.

748 Figure 3. High-pressure luminescence spectra of dravite tourmaline up to 65 GPa in a Ne
749 pressure medium. The transition from intermediate to strong crystal field occurs between room
750 pressure and ~6.0 GPa, as manifested by the decline in intensity of the broad-band ⁴T₂ emission
751 centered near 725 nm.

752 Figure 4. Pressure shift of the observed luminescence bands in dravite (a) low pressure region,
753 showing the essentially 0 nm/GPa pressure shift of two of the emission bands; and (b) up to 65
754 GPa. The grey region indicates a possible change in compression mechanism. Closed symbols
755 are data collected on compression, and open symbols are on decompression (for clarity, these
756 are not included in b). Error bars are smaller than the symbols.

757 Figure 5. Detail of dravite R-related emission band illustrating the splitting observed under
758 compression.

759 Figure 6. Relative change of lattice parameters and unit cell volume for uvite (Xu et al. 2016),
760 and dravite (this study). The results of Li et al. (2004) are not plotted for clarity (see
761 Supplementary Information). Error bars for our study are smaller than the symbols.

762 Figure 7. Site volumes in dravite tourmaline as a function of pressure. There is a large difference
763 in compressibility between the Na site and the Si, Al, and Mg sites. Note that the Na site is
764 considered 9-fold across the pressure range of these measurements.

765 Figure 8. Na-O bond lengths of dravite tourmaline as a function of pressure. Note the different
766 behavior of the two sets of three Na-O-Si bonds, and the change in slope of the Na-O1 distance
767 above ~15.4 GPa.

768 Figure 9. Ring distortion parameters as a function of pressure: (a) O6-threefold axis distance as
769 a function of pressure (e.g. ring puckering). This distance decreases as pressure increases,
770 indicating that ring puckering increases as pressure increases. (b) Ring crimping (see text for
771 definition), showing the change in ring crimping in the high-pressure phase. Note, the ~7 GPa
772 data point appears to be an outlier. (c) Ditrigonality (see text). The change in puckering and
773 crimping in the high-pressure phase indicate that the compression mechanism changes in the
774 high-pressure phase. Lines are linear fits to guide the eye.

775 Figure 10. Our 23.6 GPa R3 structure of dravite tourmaline. (a) looking down the *c*-axis with
776 oxygens removed for clarity. (b) oblique view showing the configuration of the X-site within the
777 Si₆O₁₈ ring. Note that Al, Si, O6, O7, and O8 all split into unique sites in the high-pressure phase.
778 Diagrams were generated in CrystalMaker[®] v8.7.6.

779 Figure 11. (a) Si-Si-Si atom angles, it should be noted that the errors on bond angles are
780 typically quite large; however, by ~ 23.6 GPa the deviation away from 120° is larger than the
781 errors. Errors are not included on the low pressure data since the 120° angle is due to
782 symmetry constraints. (b) Tetrahedral rotation as a function of pressure. The discontinuous
783 shift in tetrahedral rotation near 15.4 GPa shows that the phase transition occurs near this
784 pressure, but the pressure dependence of rotation is only slight altered in the high-pressure
785 phase.

786 Figure 12. Bond valence calculations for different ions within the dravite tourmaline structure
787 as a function of pressure.

788 Table 1. Unit cell parameters of dravite tourmaline at various pressures. *Crystal structures
789 solved at these pressures

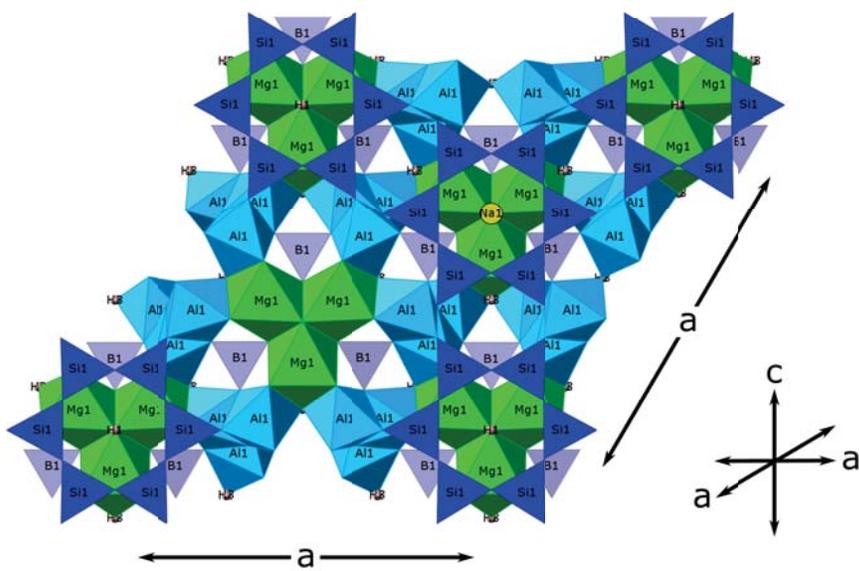
790

791 Table 2. Pressure-volume data fit up to ~ 15.4 GPa using both 2nd and 3rd order Birch-Murnaghan
792 equations of state. It should be noted that our refined K' is within one e.s.d. of the implied
793 value of 4 of the 2nd-order Birch-Murnaghan.

794 Table 3. Crystal structure refinement details for dravite tourmaline at various pressures.
795 Complete CIFs can be found in the supplementary material.

FIGURE 1

a



b

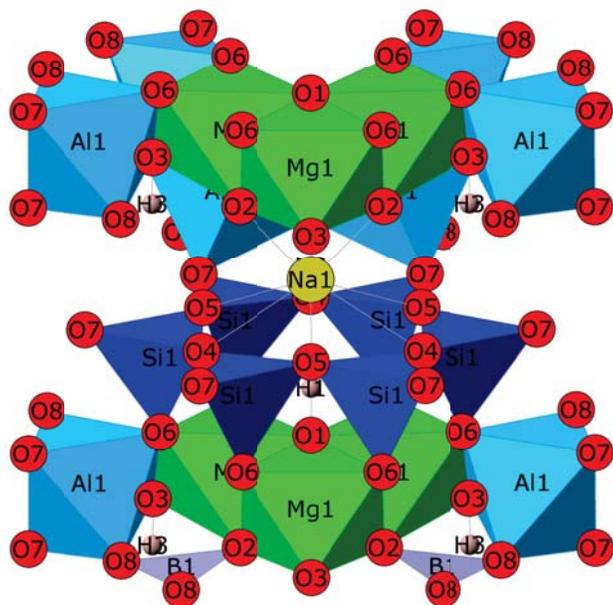


FIGURE 2

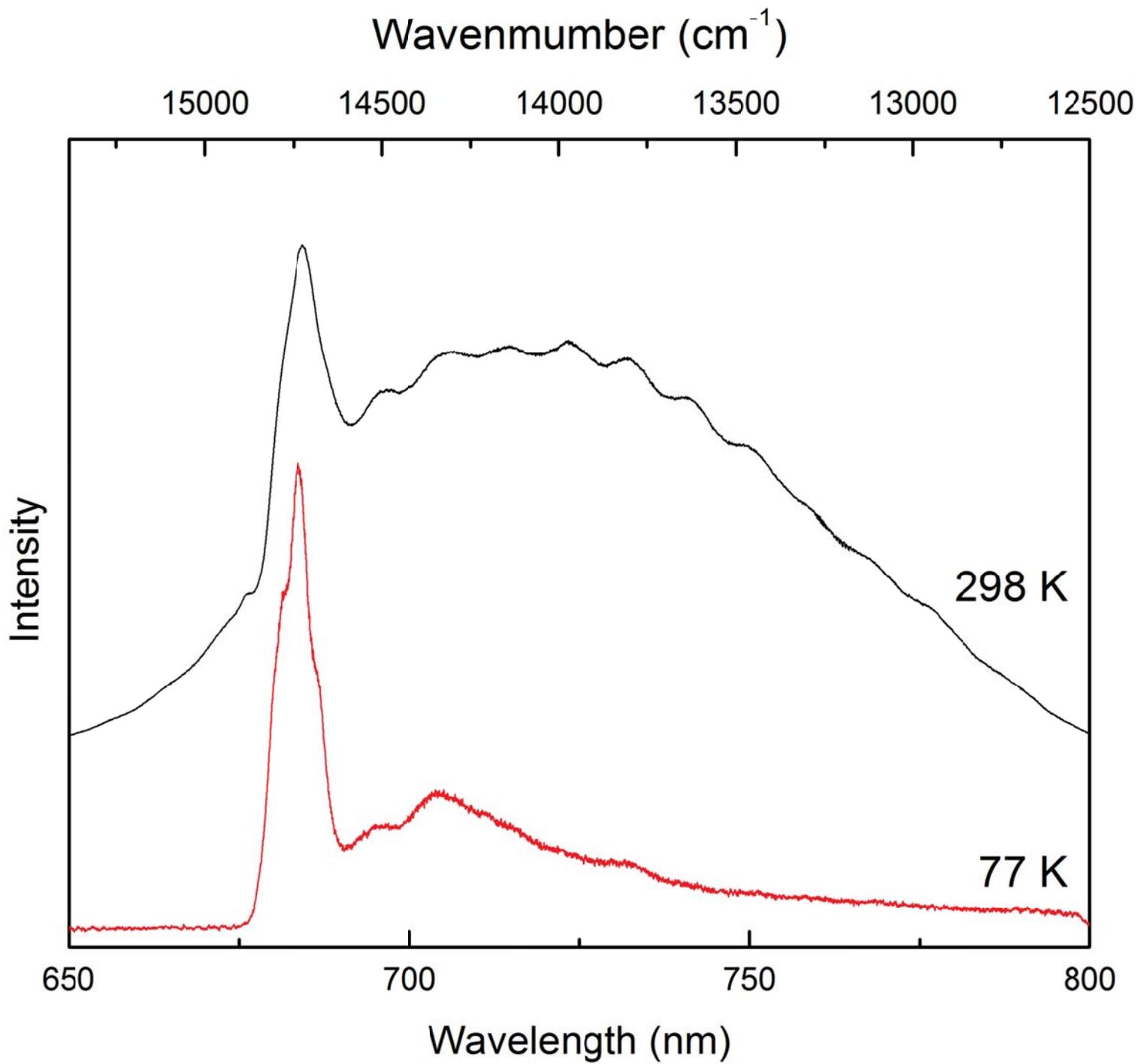


FIGURE 3

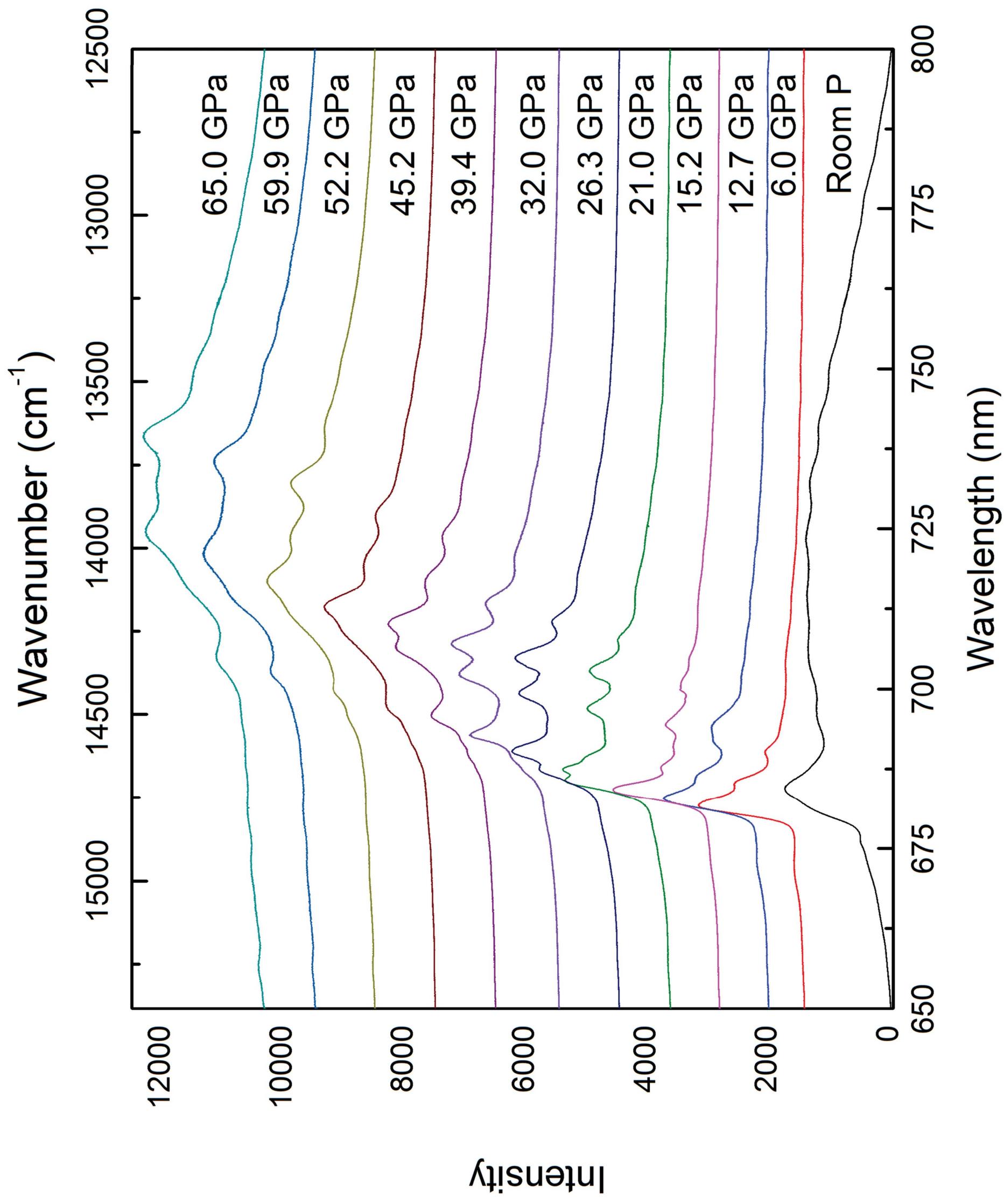


FIGURE 4a

a

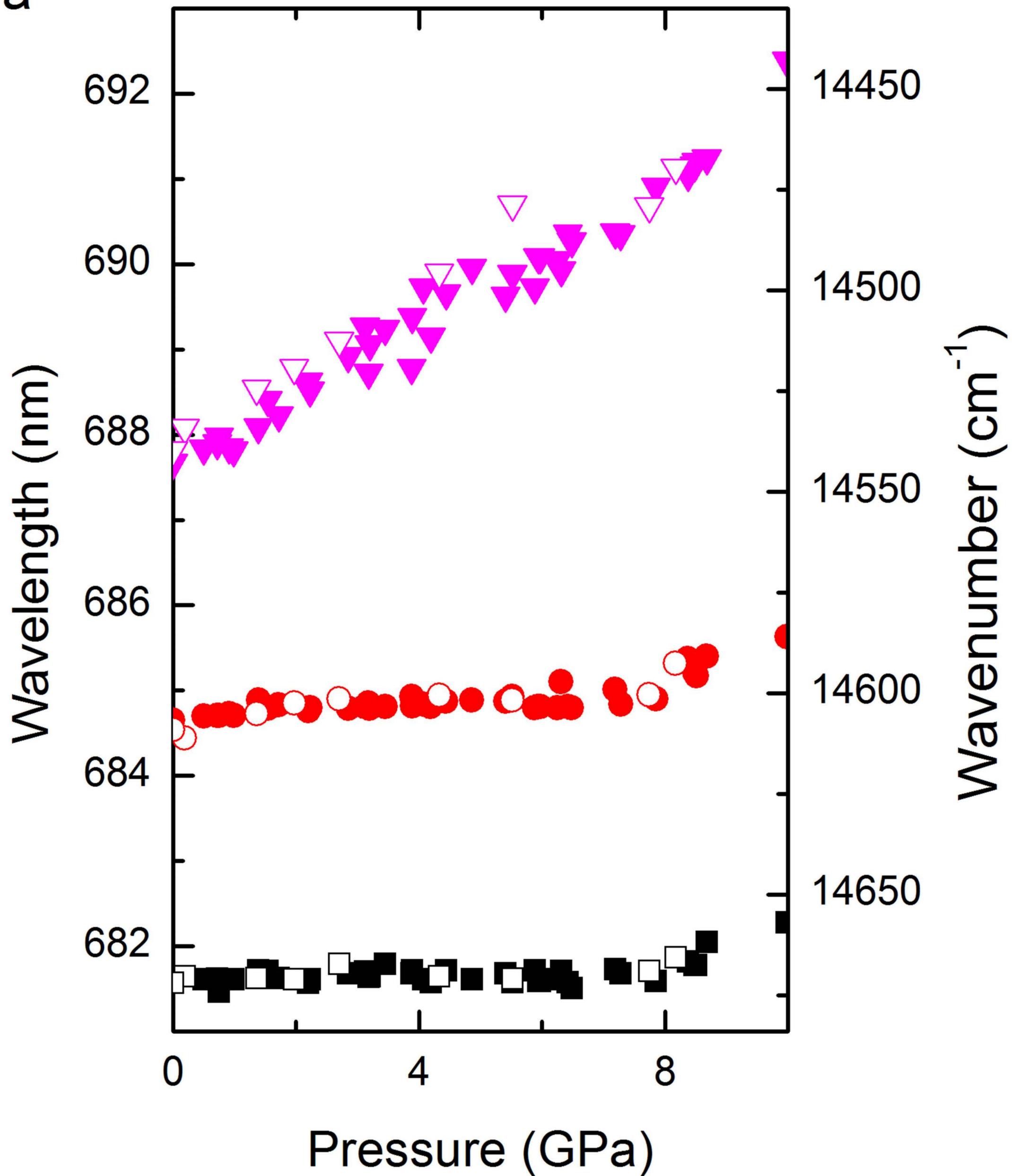


FIGURE 4b

b

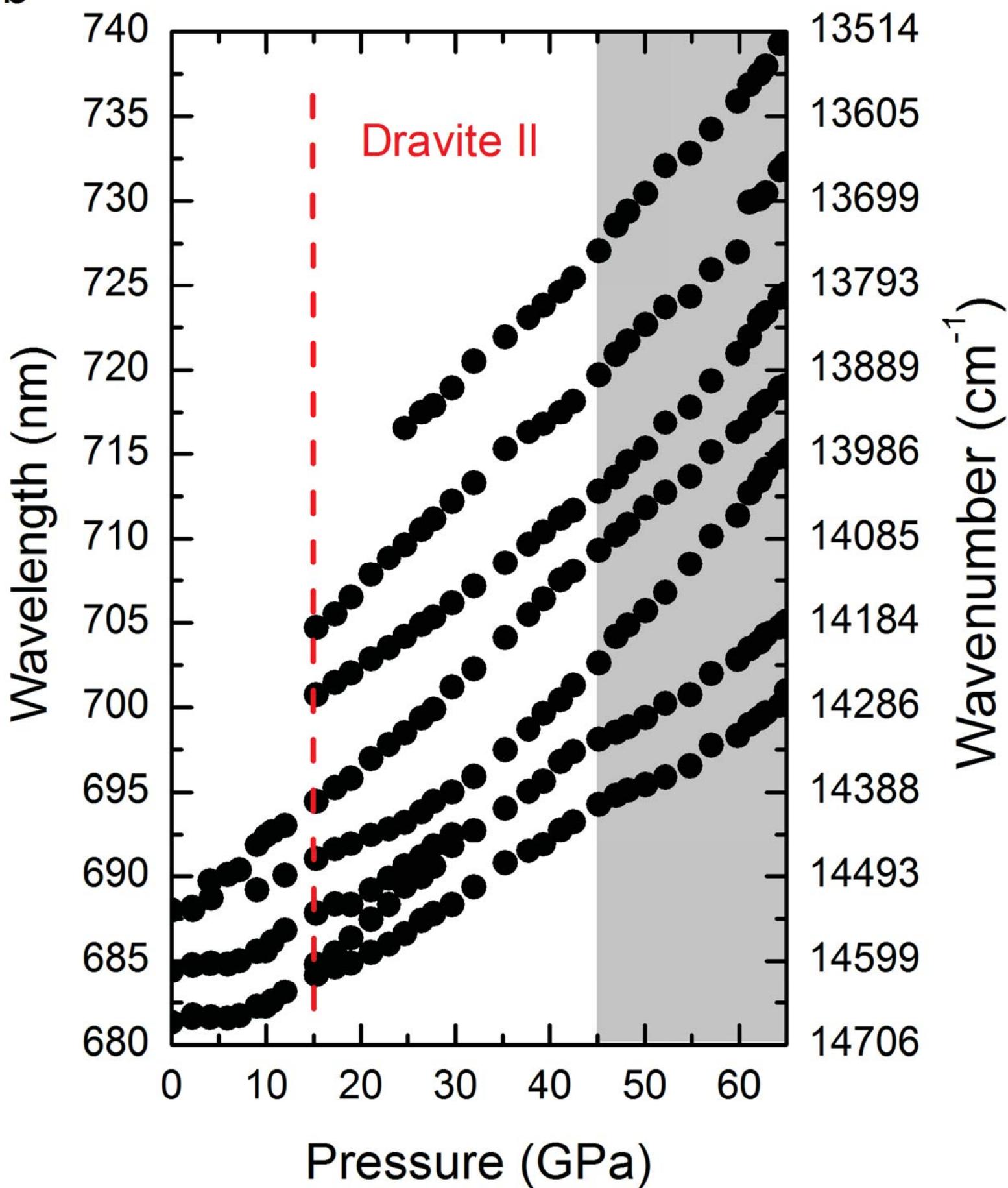


FIGURE 5

Dravite emission bands

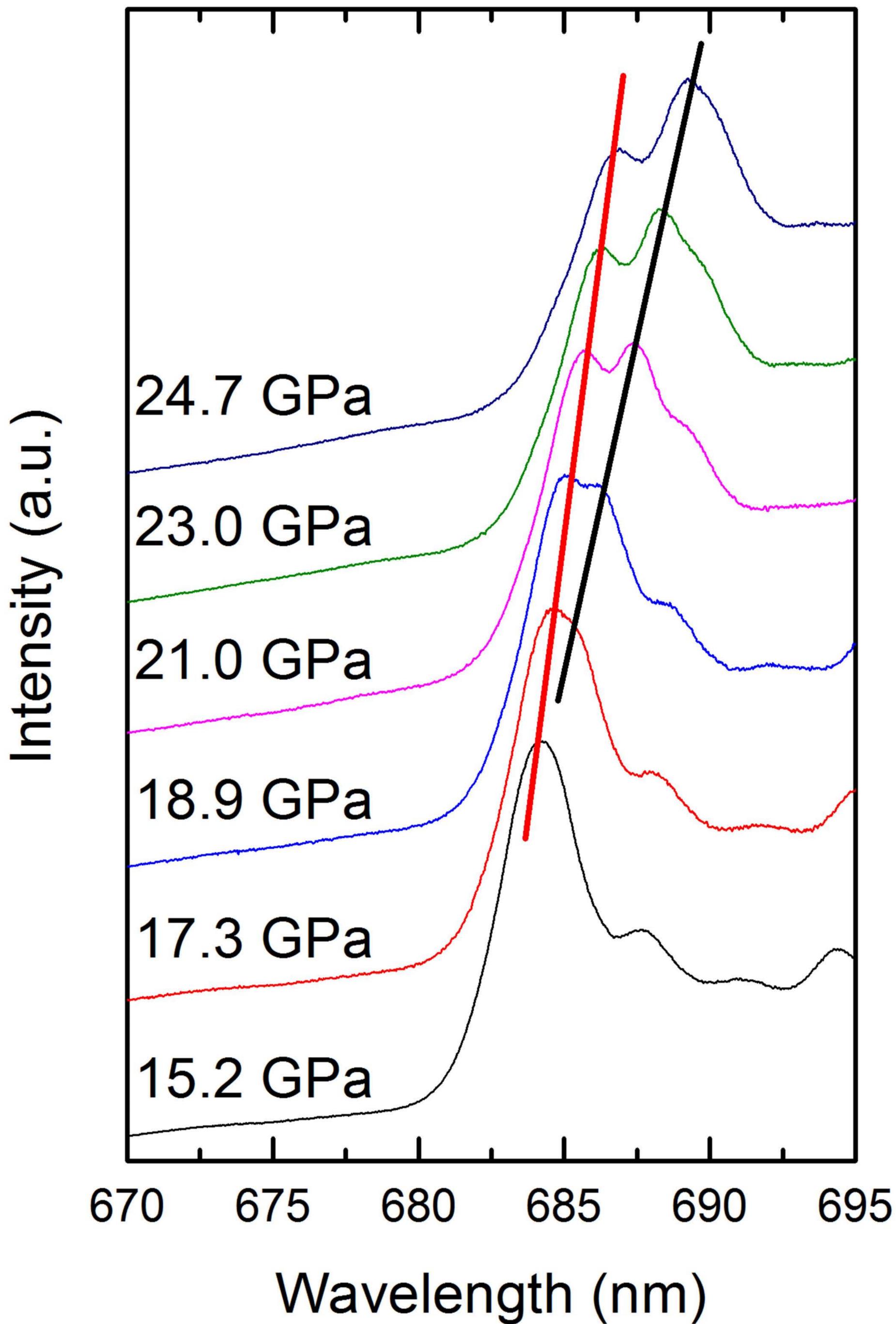


FIGURE 6

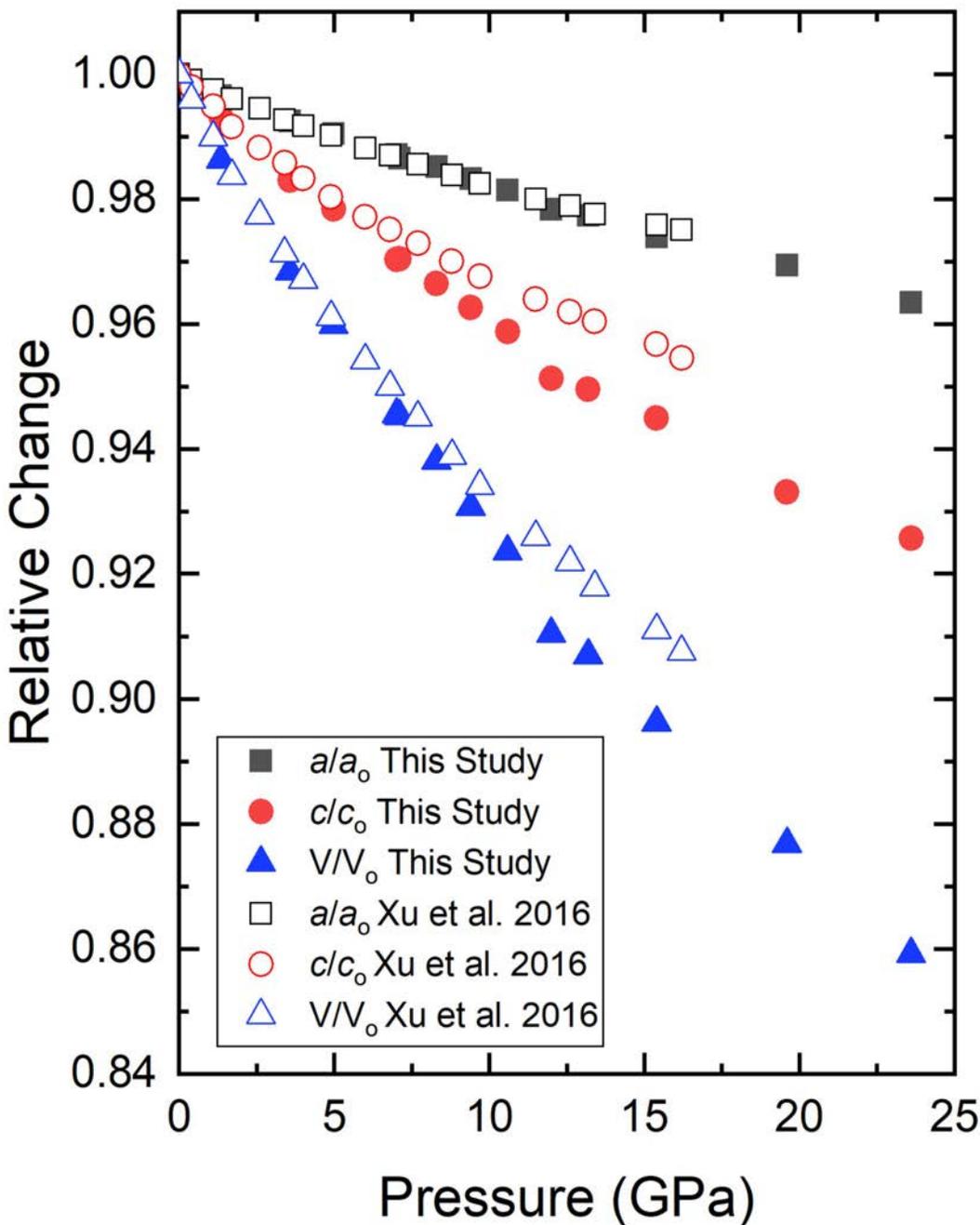


FIGURE 7

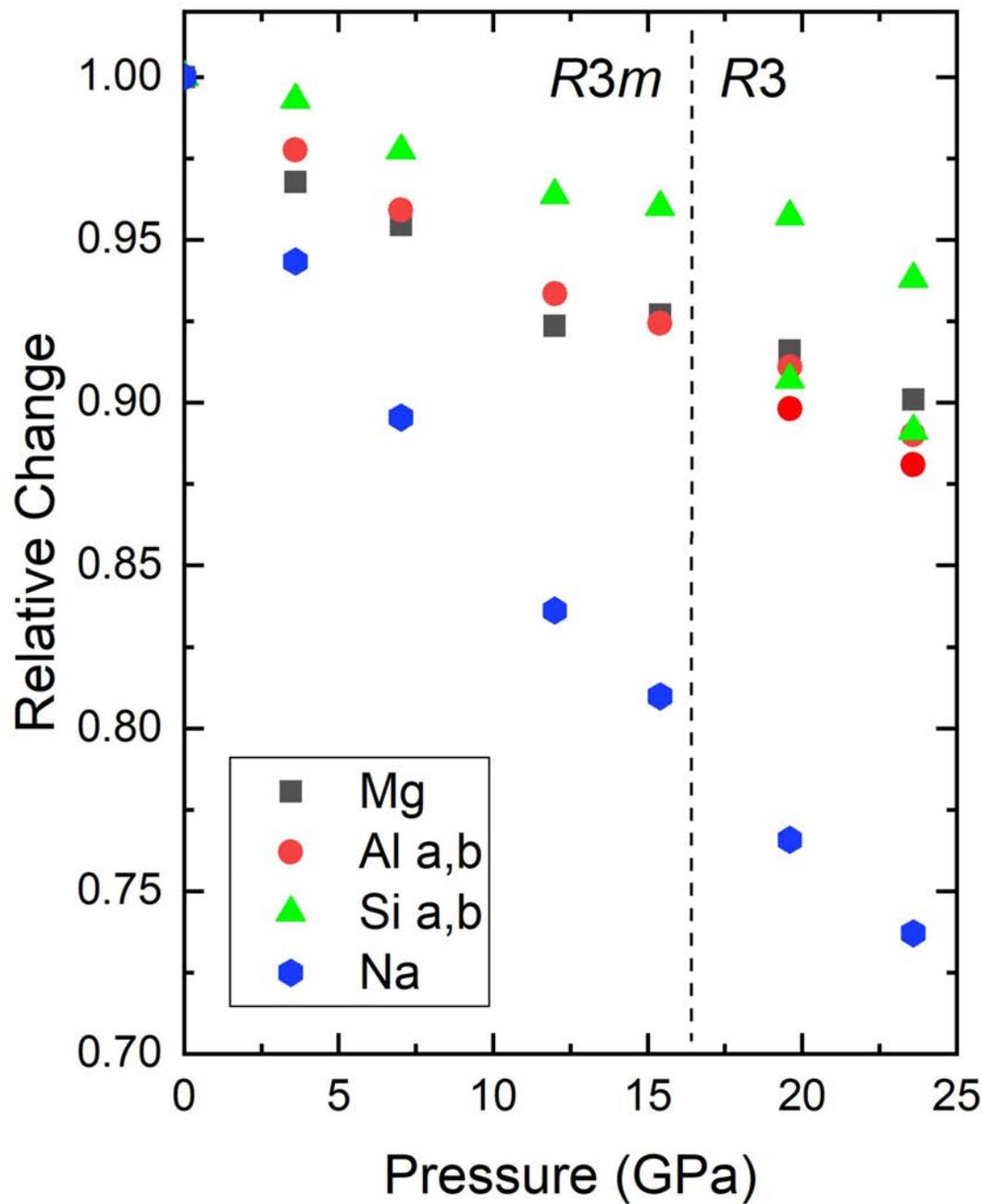


FIGURE 8

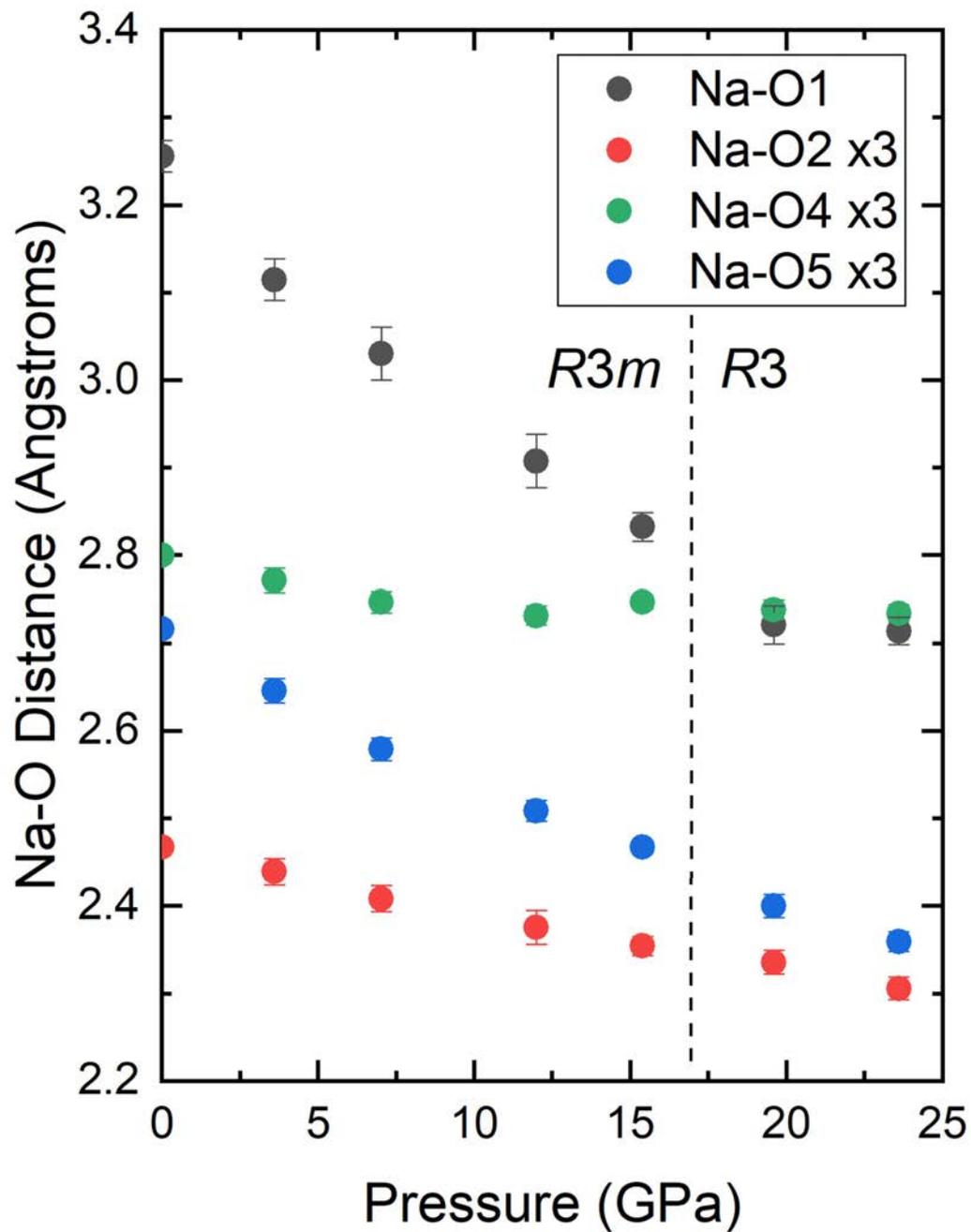


FIGURE 9a

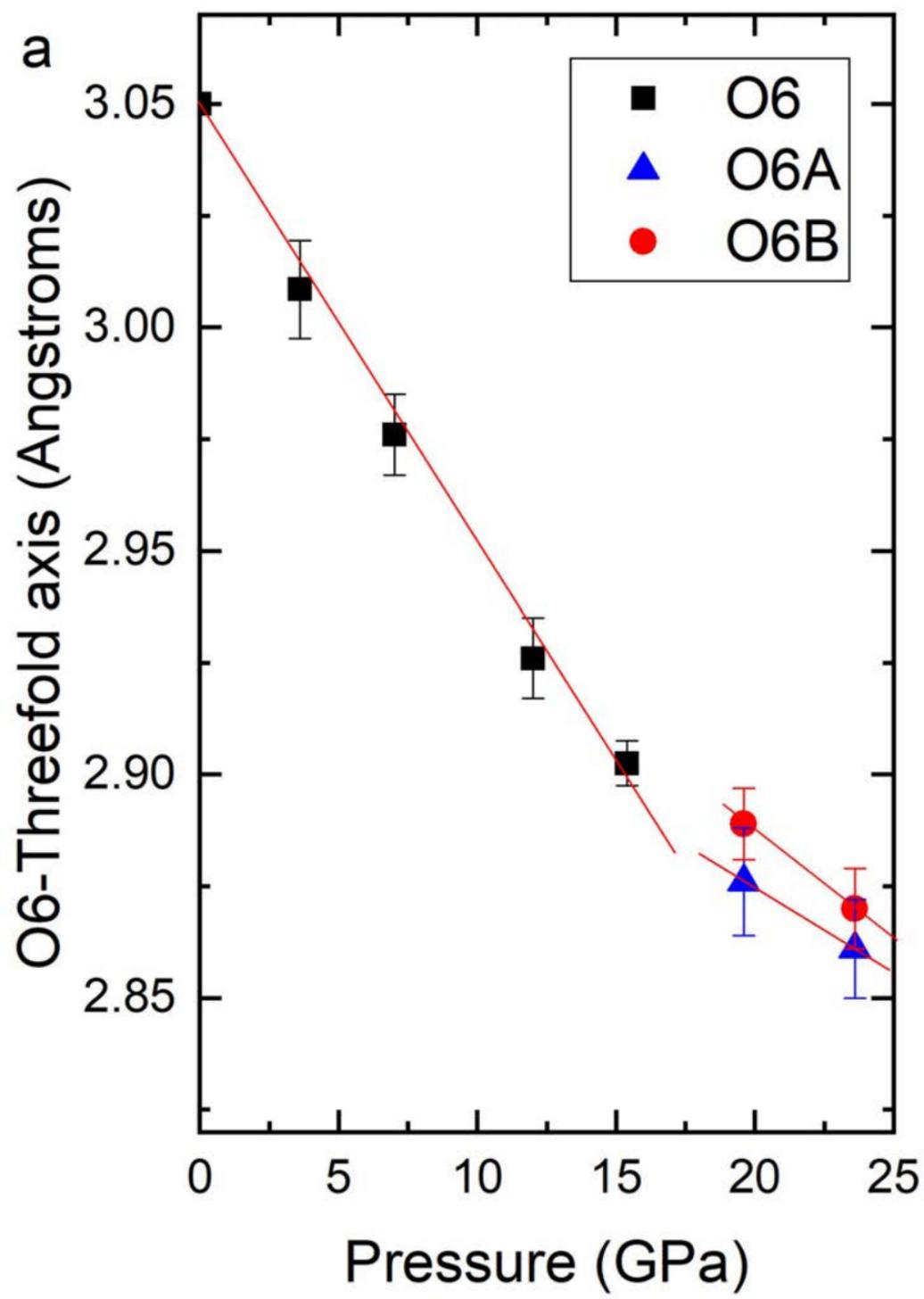
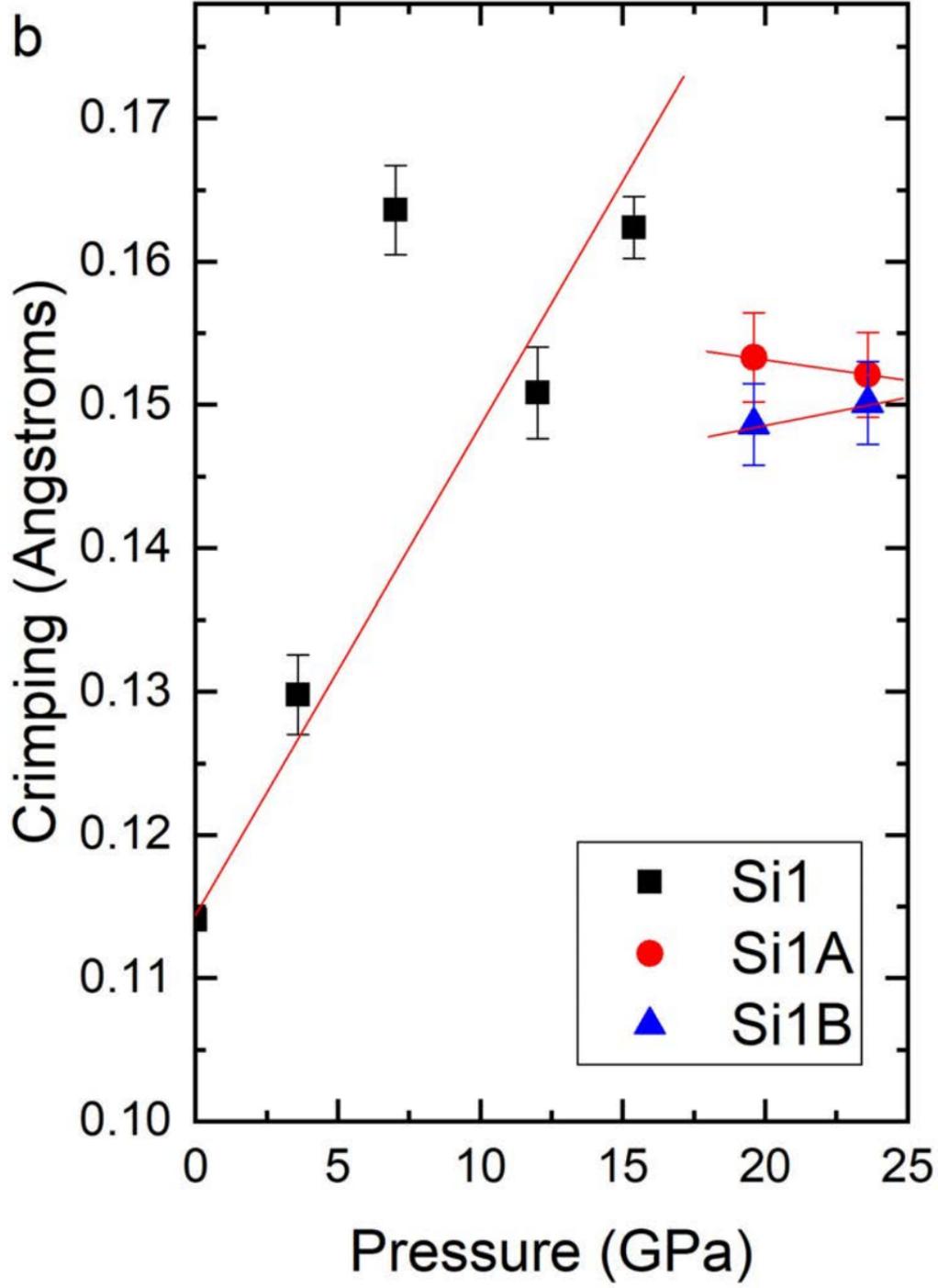


FIGURE 9b



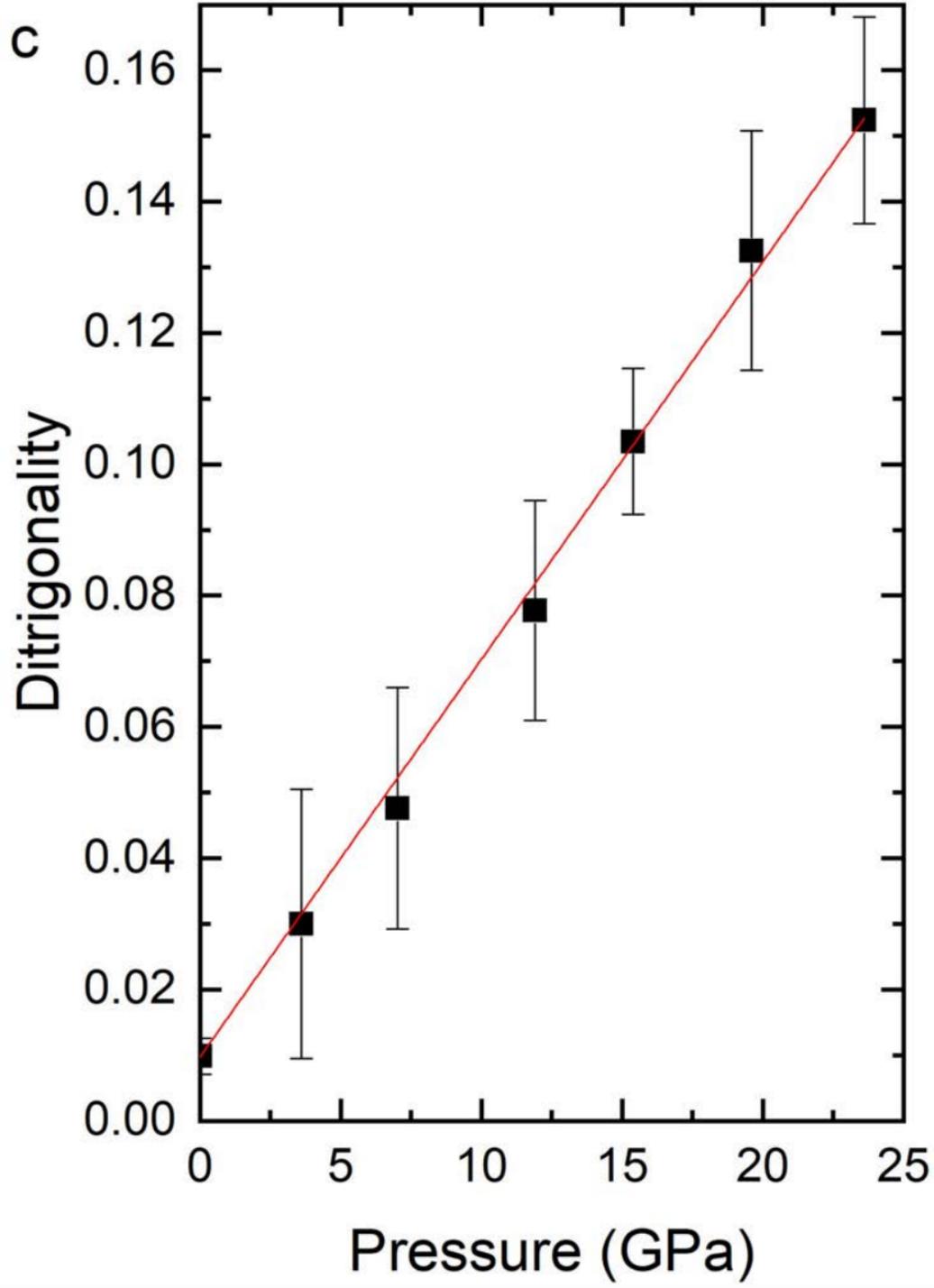
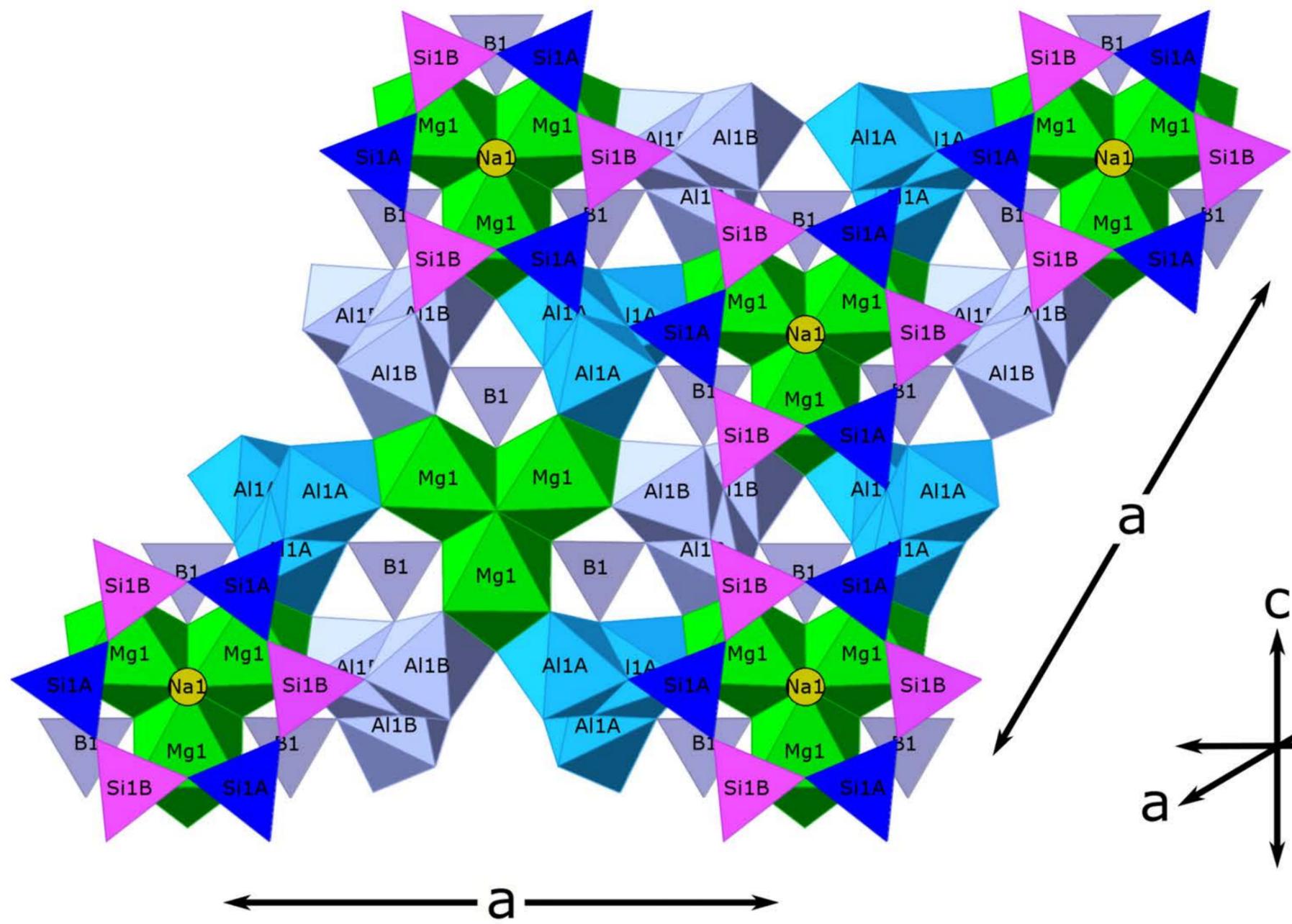


FIGURE 9

a



b

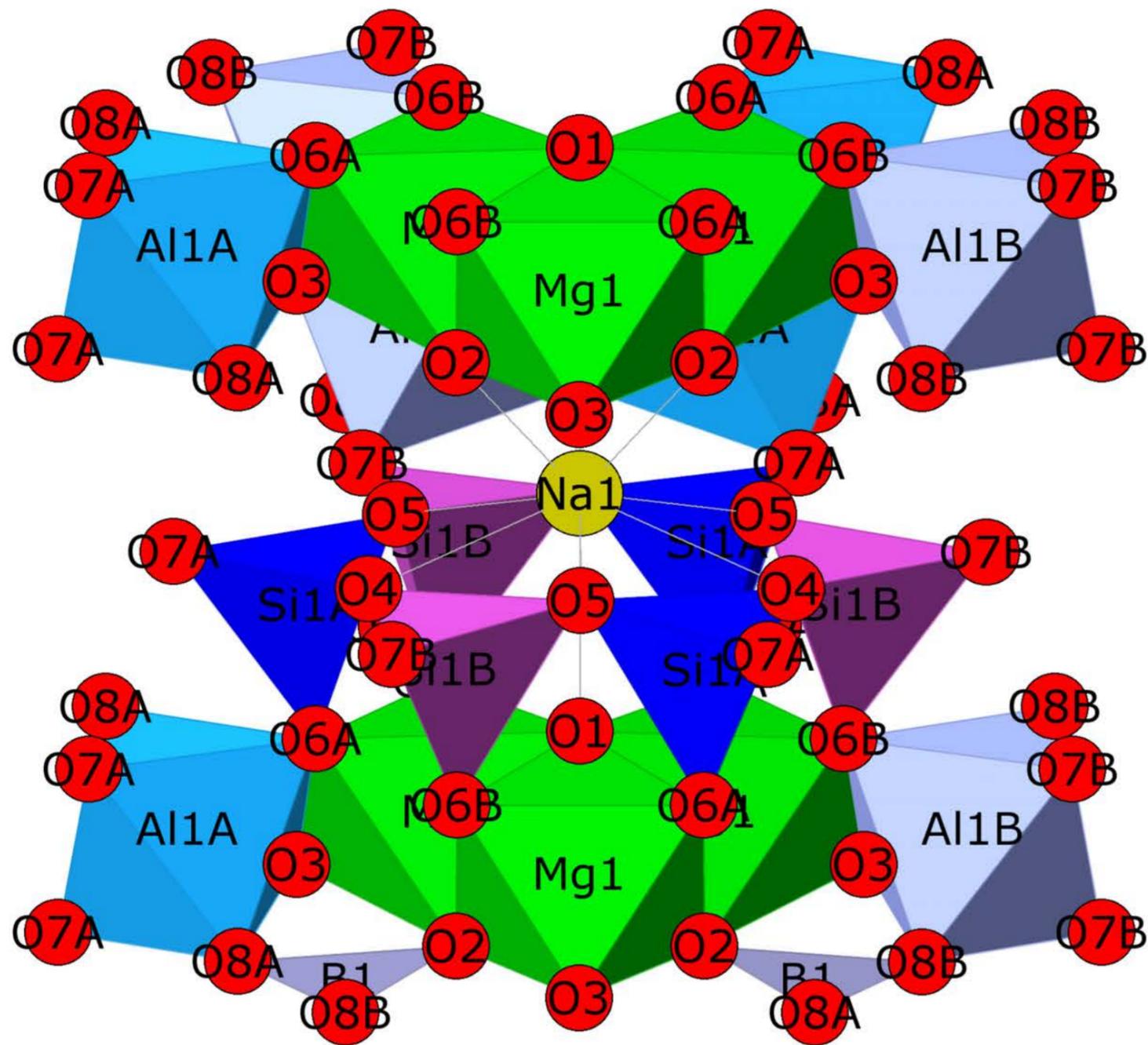


FIGURE 11a

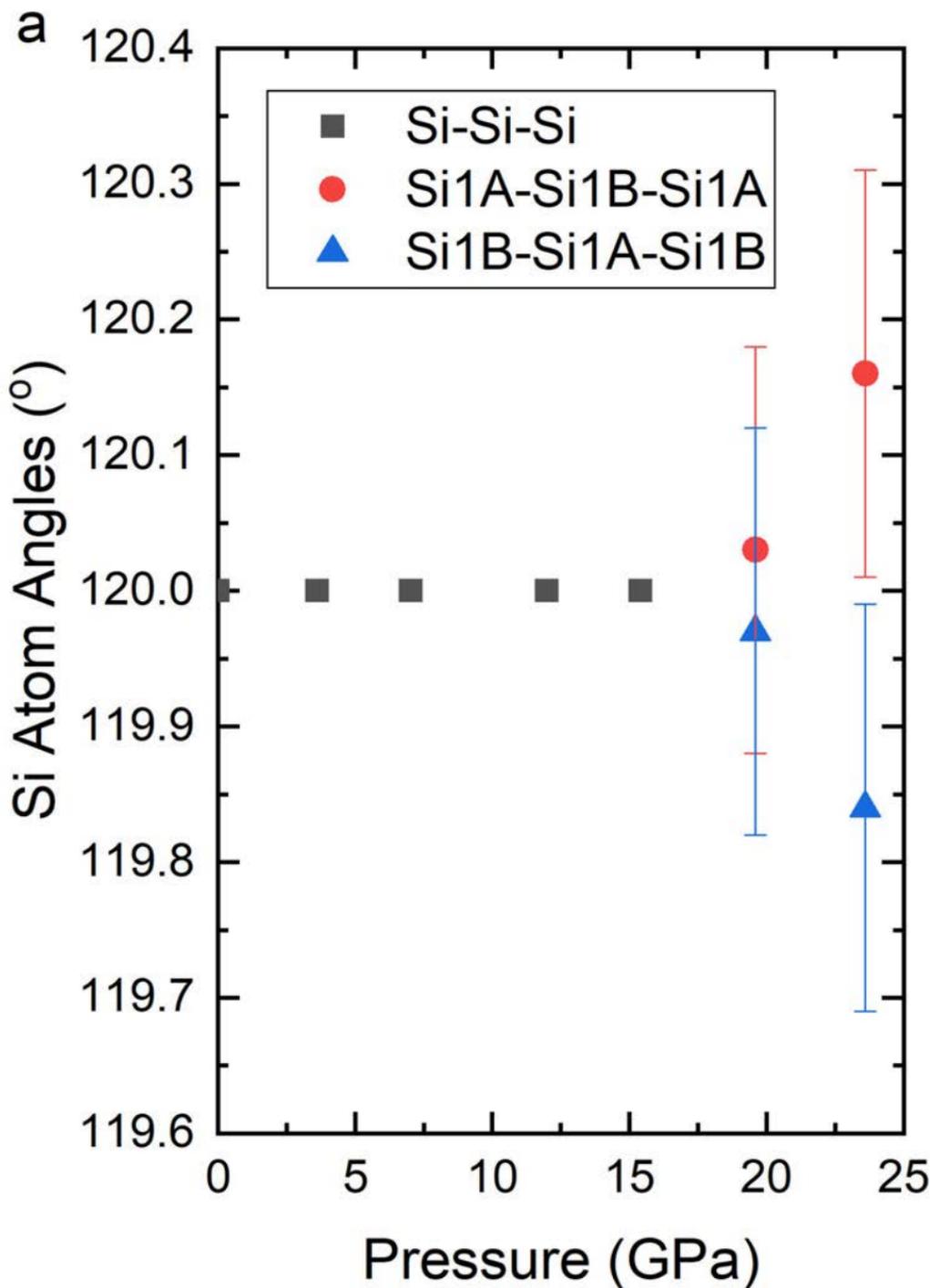


FIGURE 11b

b

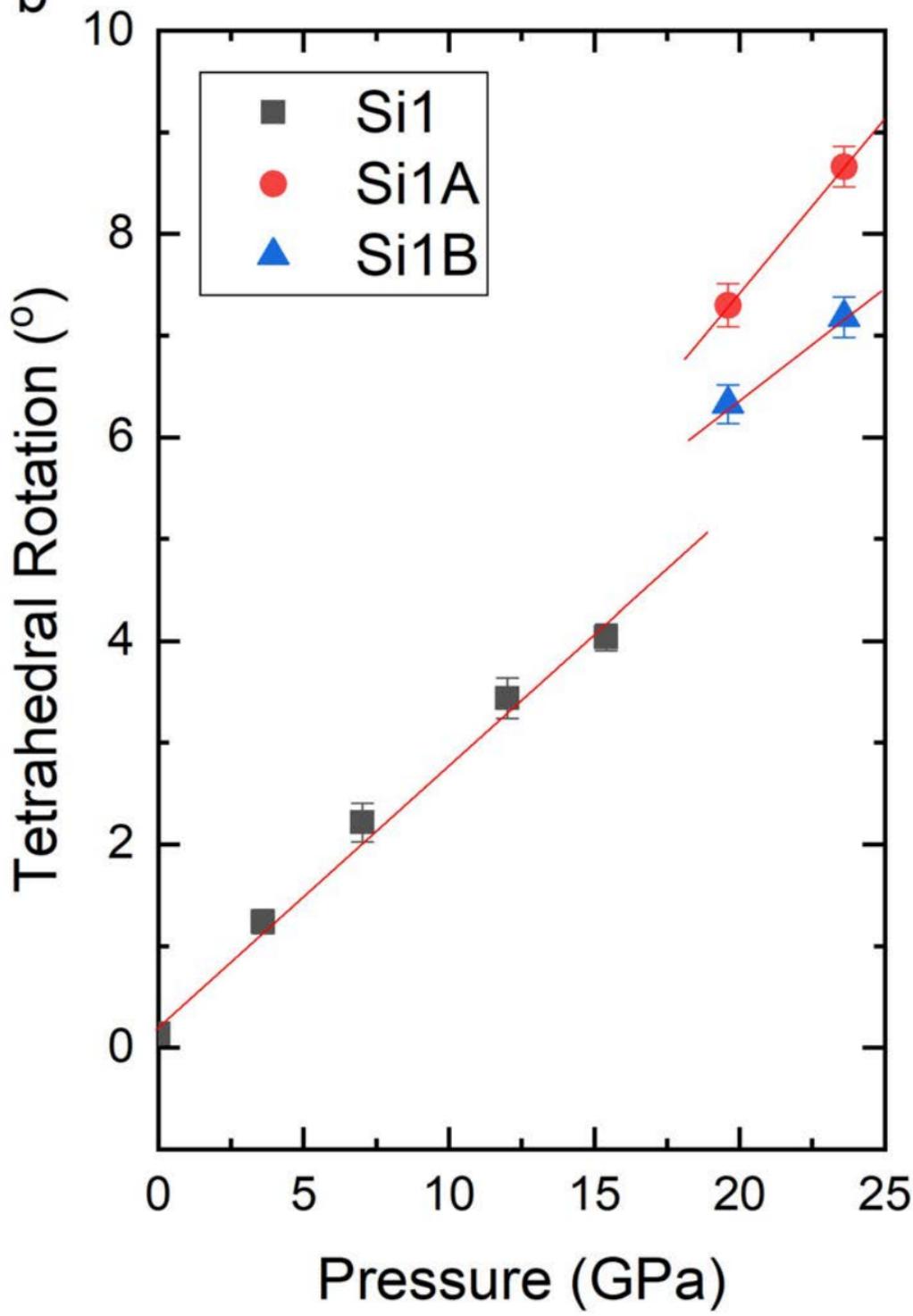
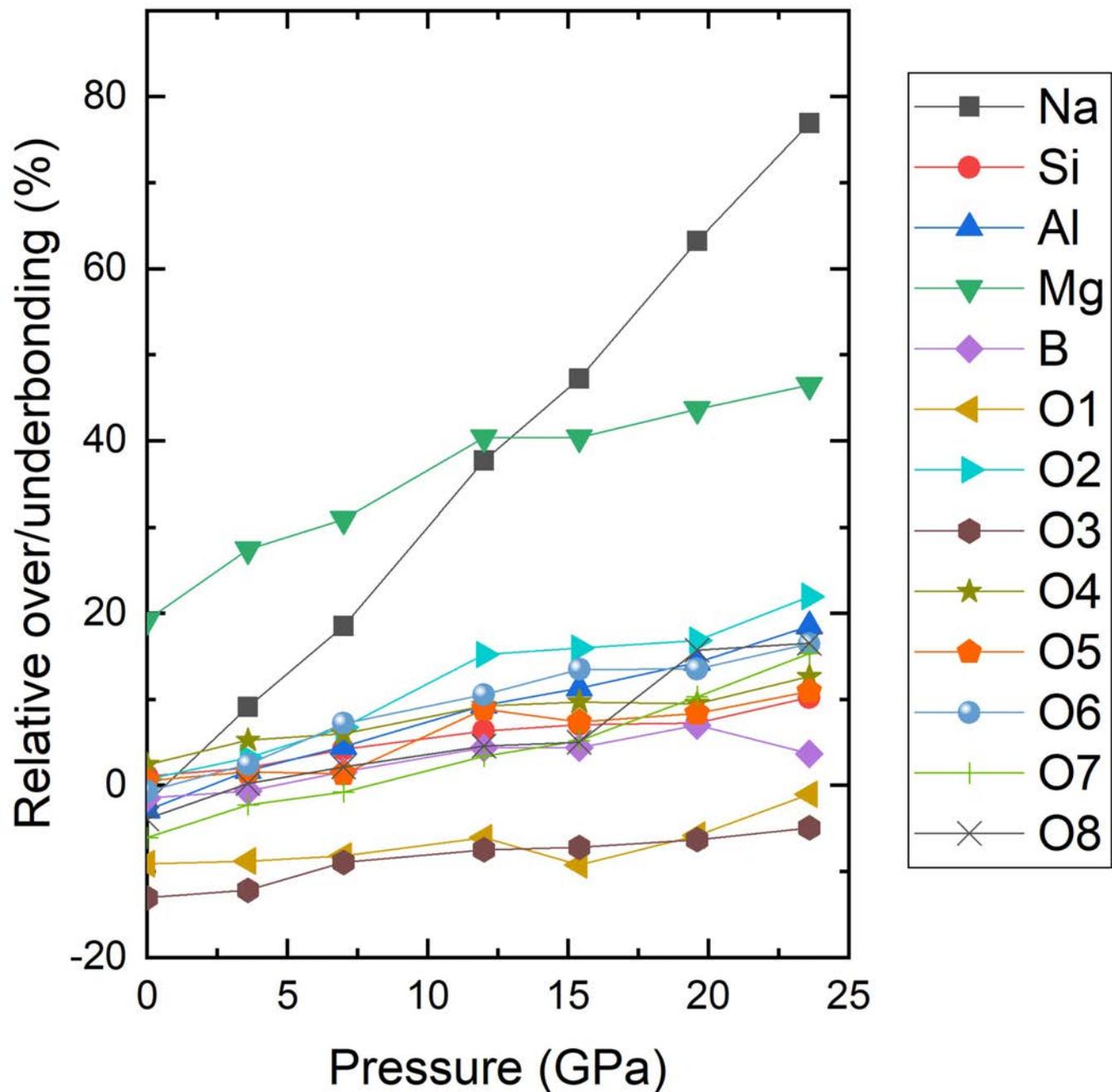


FIGURE 12



Pressure (GPa)	<i>a</i> -axis (Å)	<i>c</i> -axis (Å)	Volume (Å ³)
*0.0001	15.9370(6)	7.1749(3)	1578.19(14)
1.4(1)	15.8820(8)	7.1258(4)	1556.60(22)
*3.6(2)	15.8180(7)	7.0534(3)	1528.38(21)
5.0(1)	15.7860(7)	7.0198(3)	1514.95(20)
*7.0(1)	15.7323(7)	6.9617(4)	1492.21(16)
7.1(1)	15.7219(8)	6.9628(4)	1491.83(22)
8.3(2)	15.7015(8)	6.9344(4)	1480.55(23)
9.4(2)	15.6711(7)	6.9066(4)	1468.91(21)
10.6(2)	15.6418(10)	6.8794(5)	1457.66(29)
*12.0(2)	15.5915(20)	6.8253(11)	1436.90(58)
13.2(3)	15.5765(14)	6.8127(7)	1431.53(40)
*15.4(4)	15.5214(21)	6.7796(11)	1414.48(58)
*19.6(4)	15.4491(13)	6.6948(6)	1383.80(33)
*23.6(5)	15.3537(15)	6.6419(6)	1355.97(37)

Sample	V_0 (\AA^3)	K_0 (GPa)	K'_0 (GPa)	Method	Reference
Schorl	1595.52(1.98)	183.5(4.2)	4(fixed)	Powder XRD	Li et al. (2004)
Uvite	1537.1(11)	96.6(9)	12.5(4)	Powder XRD	Xu et al. (2016)
		120(2)	4(fixed)	Powder XRD	Xu et al. (2016)
Dravite	1578.18(20)	109.6(3.2)	4.6(8)	Single-crystal	This study
		112.0(1.0)	4(fixed)	Single-crystal	This study

Pressure (GPa)	0.0001	3.6(2)	7.0(1)	12.0(2)	15.4(4)
Temperature (K)	298(2)	298(2)	298(2)	298(2)	298(2)
Crystal System, space group	rhombohedral <i>R3m</i>	rhombohedral <i>R3m</i>	rhombohedral <i>R3m</i>	rhombohedral <i>R3m</i>	rhombohedral <i>R3m</i>
R ₁ (%)	2.99	6.57	4.89	4.05	3.06
wR ₂ (%)	7.35	16.21	11.86	9.79	8.29
Goodness-of-fit	1.080	1.155	1.238	1.261	1.187
No. of reflections	2167	634	512	354	484
No. refined parameters	47	44	44	44	44

19.6(4)	23.6(5)
298(2)	298(2)
rhombohedral	rhombohedral
<i>R3</i>	<i>R3</i>
6.95	7.35
17.60	18.50
1.050	1.051
874	844
69	69