Revision 2 1 2 Raman and IR studies of the effect of Fe substitution in hydroxyapatites and 3 deuterated hydroxyapatite Anastasios Antonakos¹, Efthymios Liarokapis¹, Andreas Kyriacou², and Theodora 4 Leventouri² 5 ¹Department of Physics, National Technical University of Athens, 15780 Athens, 6 7 Greece ²Physics Department, Florida Atlantic University, Boca Raton, FL 33431, USA 8 9 Abstract We have studied synthetic Fe-substituted hydroxyapatite $Ca_{5-x}Fe_x(PO_4)_3OH$ and the 10 corresponding deuterated samples with varying Fe concentrations x ($0 \le x \le 0.3$) by Raman 11 12 and IR spectroscopy at room temperature. In the IR spectra, substitution of deuterons for protons affects the OH internal mode in a way consistent with the mass difference of the 13 14 substituting ions, as well as a mode attributed to vibrations of the Ca₃-(OH) unit. In the

Raman spectra, the frequency of all modes is not noticeably affected by the Fe substitution. Raman bands show increased width and substantial reduction in intensity with increasing amount of Fe, presumably related to disorder introduced by the substitution. We find that the disorder is smaller in the hydroxyapatites compared to the deuterated ones.

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21 Keywords: hydroxyapatites, Fe-substitution, Raman spectroscopy, FTIR

22 Introduction

23 Iron is a biologically important trace element substituting for Ca in the hydroxyapatite (HAp) structure (0.003 wt% in enamel, 0.01 - 0.1 wt% in bone). It has been 24 25 demonstrated that Fe-deficient rat bones exhibit decreased mechanical strength, decreased bone mass density, and increased fragility (Medeiros et al. 2002), whereas Fe 26 27 overload leads to decreased osteoblast cell number and activity (De Vernejoul et al. 1984). Disagreements on the structural modifications and related properties that the Fe 28 29 for Ca substitution causes in the apatite lattice are reported in the literature (Wu et al. 30 2007; Low et al. 2008; Khudolozhkin et al. 1974; Morissay et al. 2005; Jiang et al. 2002; Li et al. 2012; Salviulo et al. 2011). In a recent publication we have shown that Fe 31 32 substitutes at both Ca(1) and Ca(2) crystallographic sites with a preference at the Ca(2)33 site by applying simultaneous Rietveld refinement of X-ray diffraction (XRD) and 34 neutron powder diffraction (NPD) powder patterns of Fe-substituted, deuterated (FeDAp) 35 powder samples (Kyriakou et al. 2013).

36 Here we present Raman and FTIR studies of the same series of FeHAp and FeDAp 37 powder samples. We have studied the effect of the deuteron substitution for H on certain 38 modes and compared experimental results with first principles calculations (Corno et al. 39 2006; Pedone et al. 2007; Calderin et al. 2005; Ulian et al. 2013). Our results indicate that Fe substitution strongly affects the intensity of the internal modes of the PO₄ tetrahedra, 40 41 due to the lattice distortion introduced by Fe at the Ca sites. Furthermore, it is found that 42 the disorder in the lattice of deuterated hydroxyapatites is more sensitive to the Fe substitution. 43

44 **Experimental**

The powder compounds studied have the chemical form $Ca_{5-x}Fe_x(PO_4)_3OH$ (FeHAp) with Fe atomic concentration $0 \le x \le 0.3$). A similar set had been deuterated in flowing oxygen-enriched D₂O at 600° C (FeDAp). Table 1 lists the composition of the investigated samples. A 0.2 M solution of Ca(NO₃)₂.4H₂O (99.98% ALPHA AESAR) was added drop wise to a 0.12 M solution of (NH₄)₂PO₄.4H₂O (99.99% ALDRICH) at ambient conditions. The pH was maintained at 9.5 ± 0.1 by adding NH₄OH. Hydroxyapatite was formed according to the reaction:

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$$5Ca(NO_3)_2 + 3(NH_4)_2HPO_4 \xrightarrow{NH_4OH} Ca_5(PO_4)_3OH$$

53 Then, FeCl₂.4H₂O solution, heated to 85 °C, was added drop wise to the HAp solution.

54 Fe-substituted HAp formed according to the reaction:

55
$$Ca_5(PO_4)_3OH + xFeCl_2 \longrightarrow Ca_{5-x}Fe_x(PO_4)_3OH$$

The final precipitate was aged for 20 h, washed, dried and heat-treated in a tube furnace at 650 °C for 15 h under flowing N_2 to prevent oxidation of the Fe ions.

58 Detailed information on the preparation conditions and characterization from 59 simultaneous Rietveld refinements of X-ray and neutron powder diffraction, x-ray 60 diffraction (XRD), transmitted electron microscopy (TEM), and magnetometry methods 61 has been published elsewhere (Kyriakou et al. 2013). A mixture of irregularly shaped 62 rods and spherical particles with dimensions between 15 nm and 65 nm was observed in 63 the TEM image of FeHAp0, while the size range is limited to 20 to 50 nm in the FeDAp0

sample. When Fe substitutes for Ca, the size range and shapes of the samples remainunaltered regardless of the Fe concentration *x*.

66 Micro-Raman spectra were obtained at room temperature using a T64000 Jobin-Yvon triple spectrometer, equipped with a liquid nitrogen cooled Charge Coupled Device 67 (CCD) and a microscope. The 514.5 nm wavelength of an Ar⁺ laser was used for 68 69 excitation in a backscattering geometry of the sample under the microscope. The 70 spectrum accumulation time was 3 hours and the laser power at the sample was 0.2 mW. 71 The spectrometer was calibrated before and after each measurement using a silicon wafer 72 or a reference Kr lamp. Because of the random orientation of the microcrystals in the 73 powder samples, no exact selection rules could be applied. The selection rules could give 74 only limited information about the properties of the Raman tensors because of the random 75 orientation of the microcrystallites in the powder samples. The differentiation of orientation was studied by turning the sample stage between 0° to 90° while no 76 dependence was found by changing the polarization of the laser. Therefore, the spectra of 77 78 all samples were recorded with random polarization. The $\times 100$ (N.A. 0.95) and $\times 50$ 79 (N.A. 0.80) magnification lenses were employed for the Raman measurements with the 80 laser spot focussed to $\sim 1 \mu m$ and $\sim 2 \mu m$, respectively. Theoretically one would expect a 81 20% larger spot for the $\times 50$ magnification and roughly two times larger confocal volume. The confocal configuration of the spectrometer, together with the micrometric probe size, 82 83 reduces the fluorescence problems encountered in the conventional Raman spectroscopy 84 of such materials (Penel et al. 1998; Okazaki and Takahashi 1997; Wopenka and Pasteris 85 2005).

The Fourier transform infrared (FTIR) spectra were recorded using a Bruker Optic 86 IFS66v/S interferometer equipped with a MIRacle[™] Single Reflection ATR (Attenuated 87 Total Reflectance) unit with a diamond/ZnSe crystal plate. The range of frequencies was 88 525 to 4000 cm⁻¹, and the spectra were recorded at ambient conditions with a resolution 89 of 2 cm⁻¹. In order to obtain a good signal-to-noise ratio, 1200 scans were collected and 90 averaged. A KBr beam splitter was used for the M-IR source. The ATR unit permits 91 spectra collection without any specific sample preparation, but only for the region above 92 525 cm⁻¹. A high-pressure lamp has been used in order to achieve high spectral quality 93 94 and to avoid low pressure effects. Since the pressure effect of the ATR accessory could not be measured, an empirical method was used whereby the variations of intensities and 95 the ratio of the intensities were considered as a function of pressure; they increase to a 96 97 maximum and then remain fairly constant as the applied force is increased further.

98 **Results and discussion**

Figure 1a presents the ATR FTIR spectra of all samples in the high frequency region. 99 The continuous line spectra correspond to the as prepared FeHAp samples, and the dotted 100 ones to the deuterated FeDAp samples. The sharp band at \sim 3572 cm⁻¹ in the FeHAp 101 102 spectra for all Fe concentrations is assigned to the OH stretching mode (Elliott 1994). The 103 mode frequency is mostly related to the mass of the H and therefore, D substitution for H 104 should modify its frequency mainly according to the mass ratio of the two H isotopes. The absence of this mode from the FeDAps and the appearance of a sharp band at ~2633 105 cm⁻¹ confirms the deuteration of the samples. In particular, the frequency of this band 106 107 follows approximately the mass rule

$$\omega_{\rm D-O} = \omega_{\rm H-O} \left(M_{\rm D-O} / M_{\rm H-O} \right)^{1/2}$$

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Figure 1b shows the lower frequency part of the ATR FTIR spectra of all samples. 109 Several peaks appear, located at \sim 566, 600, 631, 963, 1026, and \sim 1088 cm⁻¹. Of these 110 modes, only that at ~631 is affected by the deuteron substitution, while the remainder are 111 not noticeably modified. Some small changes in frequency of the ~ 1026 cm⁻¹ peak are 112 113 most likely due to intensity modifications of those modes that contribute to this peak. The first two modes at ~566 and ~600 cm⁻¹ have been attributed to the v_4 doubly degenerate 114 asymmetric O-P-O bending mode of the PO₄ tetrahedron (Antonakos et al. 2007). The 115 higher frequency peaks at 1026, and ~1088 cm⁻¹ are related to the v_3 triply degenerate 116 asymmetric P-O stretching mode of the PO₄ tetrahedron (Antonakos et al. 2007). The 117 weak but sharp band at 963 cm^{-1} can be assigned to the v₁ vibration (symmetric P-O 118 stretching mode) of the PO₄ tetrahedron (Penel et al. 1998; Antonakos et al. 2007, Park et 119 al. 2002). Finally, the weak mode at ~ 631 cm⁻¹ has been attributed to the OH librational 120 121 mode of the Ca3-OH group, i.e. the OH ion straddles across the Ca triangles that are 122 situated on the mirror plane of the apatite structure (Cant et al. 1971, Elliott 1994; Antonakos et al. 2007). Some *ab initio* calculations assign this mode to the internal v_4 123 124 vibrations (Corno et al. 2006; Pedone et al. 2007). This does not appear to be correct, since this mode disappears almost completely on deuteration, and is neither affected in 125 126 intensity nor width by the Fe substitution. As we will see below in the Raman spectra, Fe substitution reduces substantially the intensity and increases the width of the internal 127 vibration modes of the PO₄ tetrahedron. Therefore, the mode must be related to the OH 128 librational vibration of the Ca₃-(OH) group. Unfortunately, ATR did not allow lower 129 frequency measurements to follow the mode in the expected frequency from the mass 130

difference H-D. Table 2 presents the detected IR and Raman modes and their assignment

132 (Elliott 1994; Antonakos et al. 2007).

Figure 2 presents data with $\times 100$ (dotted line) and $\times 50$ (continuous line) 133 magnifications for the low (Fig. 2a) and high (Fig. 2b) frequency regions. Although in 134 135 our previous investigations of apatite higher magnification gave stronger peaks 136 (Antonakos et al. 2007), this did not occur in this study, presumably due to the diffused 137 light by the nanostructured samples that favors the longer focusing. Several peaks can be discriminated mainly with the $\times 50$ magnification. In the low frequency part (Fig. 2a) 138 these are located at $\sim 330 \text{ cm}^{-1}$ (very weak) and $\sim 431, 448, 581, 593, 610, \text{ and } \sim 617 \text{ cm}^{-1}$ 139 (strong ones). The two strong ones at ~431 and ~448 cm⁻¹ have been assigned to v_2 140 vibrations, while the other four modes at higher frequency belong to the v_4 mode of the 141 PO₄ group. The origin of the weak mode at \sim 330 cm⁻¹ will be discussed below. In the 142 high frequency region (Fig. 2b) the peaks are located at ~839, 894, 916, 925, 962, 1027, 143 1040, 1046, 1075, 1122, 1252, and 1297 cm⁻¹. Most of them can be only observed with 144 the $\times 50$ magnification. From those peaks, the ones at ~ 1027 , 1040, 1046, and ~ 1075 cm⁻¹ 145 have been assigned by several authors to the v_3 vibrations, and the one at ~962 cm⁻¹ to v_1 146 of the PO₄ unit (Antonakos et al. 2007). The lower frequency modes $(843 - 925 \text{ cm}^{-1})$ are 147 148 related neither to Fe introduced to the sample nor to an Fe oxide impurity since they are not modified in intensity by increasing the amount of Fe. Modes at relative frequencies 149 150 have been proposed in shell model lattice dynamic calculations (Calderin et al. 2005), but other more recent theoretical calculations did not find any modes related to the undoped 151 152 HAp in this frequency region (Corno et al. 2006; Pedone et al. 2007; Ulian et al. 2013). Most likely the bands at 843 and 894 cm⁻¹ are related to atmospheric CO_3^{-2} ions entering 153

the HAp structure during the sample preparation. Similar results were obtained in other studies (Leventouri at al. 2003; Antonakos et al. 2007). Then the two broad bands at ~1252 and ~1297 cm⁻¹ are related to CO_3^{-2} /amide and/or HPO₄⁻² impurities (Vignoles-Montrejaud 1984; Penel et al. 1998; Antonakos et al. 2007). Finally, the very narrow band at ~1122 cm⁻¹ is a signal from the fluorescence lamp.

Figure 3 shows the effect of the 0.30 atomic nominal Fe substitution for Ca 159 160 corresponding to the two magnifications. In the low frequency region (Fig. 3a) all strong 161 modes that appear in the pure HAp spectrum completely disappear, while new peaks appear located at ~224, 244, 291, 409, and ~498 cm⁻¹. All of them belong to hematite (de 162 Faria et al. 1997). A peak at $\sim 612 \text{ cm}^{-1}$ is mainly from the strong hematite band, which 163 164 masks the narrow weaker v_4 PO₄ peaks of the HAp (seen in Fig.2a). The lower frequency part of Figure 3b (843 - 925 cm⁻¹) and the v₁ PO₄ mode remain practically unchanged in 165 166 intensity. The $v_3 PO_4$ modes are damped and broaden due to disorder introduced by the Fe 167 substitution. The $v_3 PO_4$ region was found in previous studies to be very sensitive upon substitution due to the different HAp environments present (Elliott 1994, Penel et al. 168 1998, Antonakos et al. 2007). At higher frequency, a new strong broad band develops 169 upon Fe substitution at 1320 cm⁻¹. This band is expected to appear strong when excited 170 171 with the 514 nm laser and is related to α -Fe₂O₃ (hematite) (Wang et al. 2005). Although this band was observed previously, the explanation for the wavelength response behavior 172 173 and for its origin is still unclear, and may be caused by a two magnon scattering process 174 (Martin et al. 1977) or a resonance-enhanced two-phonon scattering process (McCarty et 175 al. 1988; Massey et al. 1990; Shim and Duffy 2001). The fundamental mode could be the 660 cm⁻¹ IR active Eu band of hematite and therefore an overtone. 176

177 The evolution of the Raman spectra with increasing Fe substitution is presented in 178 Figure 4a for the low frequency part of FeDAps. It is observed that the intensity of the modes assigned to hematite (marked by straight lines) increase with the amount of Fe 179 substitution, as presented in Figure 4b. On the one hand, XRD results have detected 180 181 traces of increasing amounts of hematite starting from the x = 0.20 nominal atomic substitution (Kyriakou et al. 2013). On the other hand, Raman spectroscopy that is more 182 183 sensitive to detect impurities indicates that the hematite phase is present in all Fe 184 substituted compounds, even for x = 0.05 (Fig.4b). The intensity of the modes assigned to 185 internal vibrations of the PO₄ tetrahedra (marked by arrows) continuously decreases with 186 increasing amounts of Fe, and almost completely disappears for the 0.02 atomic nominal Fe substitution for Ca. The comparison of Figure 2a (x = 0) with Figure 3a (x = 0.3) 187 188 shows that the effect is similar for the as-prepared set of samples (FeHAp). In the low frequency region of the Raman spectra, a mode of E1 symmetry related to Ca-OH is 189 expected at 330 cm⁻¹ (Pedone et al. 2007). A mode at close frequency (332 cm⁻¹) is 190 191 observed in pure DAp, which gradually disappears with Fe substitution (Fig.4a). This 192 mode involves vibrations of OH. Therefore, the D substitution would affect (decrease) its 193 frequency. In Figure 2a (HAp), this mode appears unshifted at the same frequency, hence, 194 questioning its assignment to vibrations that involve the OH units (Pedone et al. 2007).

The Fe substitution for Ca could introduce disorder by the breaking of long-range symmetry and shifting to lower frequencies of the modes involving the Ca atoms owing to their mass difference. The internal modes of the PO_4 units should not be affected significantly. Except from a substantial reduction in intensity (Fig.4a), the internal modes

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are not modified in frequency, probably because of the only small amounts of Fe (up to0.3 atomic, nominal content).

201 Depending on the mass difference between the substituting (Fe) and the original 202 (Ca) atom, the compound could follow one or two mode behaviors (Srivastava 1990). In 203 the former case, there will be a gradual shift in frequency towards the limiting value of a 204 hypothetical system with 100% Fe at the Ca site. In the latter, there will be a gradual 205 development of another mode that will correspond to the frequency of the hypothetical 206 end member. Due to the different coordination of the substituting Fe, it is very unlikely to 207 assume one-mode behavior, since the internal modes of PO₄ will be affected only indirectly by the breaking of the external bonds to the Ca sites. Concerning the effect of 208 209 the mass difference between Fe and Ca on the vibrational modes, *ab initio* calculations have investigated ⁴²Ca isotopic substitution for ⁴⁰Ca (5% mass difference at the Ca sites) 210 and deuteration (Corno et al. 2006). It has been found (Corno et al. 2006), that the 211 expected frequency shifts for modes above $\sim 600 \text{ cm}^{-1}$ will not exceed $\sim 1 \text{ cm}^{-1}$. Only 212 Raman bands of frequency less than $\sim 400 \text{ cm}^{-1}$ are expected to be shifted by a few cm⁻¹ 213 (Corno et al. 2006). In our case, even for x = 0.30 (6% atomic Fe substitution for Ca) 214 215 there will only be a 2.5% mass difference with the partial substitution of Fe for Ca. 216 Therefore, the mass effect on the vibration frequencies will be even smaller. Concerning deuteration, only peaks in the range $\sim 600 \text{ cm}^{-1}$ are expected to be shifter substantially 217 218 (Corno et al. 2006). From the observed modes in both IR and Raman spectra, only the above-mentioned mode at $\sim 631 \text{ cm}^{-1}$ is affected, but, as previously discussed, should be 219 220 correlated to an OH vibrational mode and not a PO_4 one (Fig.1b). The rest of the modes 221 appear at the same frequency in both HAp and DAp (Figs.1b, 2, & 4a).

222 What is impressive is the strong reduction in intensity of the majority of the modes in the Raman spectra (Figs. 4a and 5) even with such small amounts of Fe 223 substitution (maximum nominal atomic amount 6%). As found by Kyriakou et al. (2013), 224 Fe^{3+} appears in a fourfold coordination breaking some bonds with the O(2) and O(3) 225 nearest neighbors, while Fe^{+2} is in a sixfold coordination breaking the bond with O(1). 226 227 The difference in mass between the two elements (Fe and Ca) cannot justify such disorder, as similar differences in mass without any chemical variation usually modify the 228 phonon characteristics, without inducing disorder. We can recognize disorder from the 229 increase in the modes width and the reduction in its intensity. The variation of disorder 230 with doping is depicted in Figure 6 for the stronger v_1 PO₄ mode at ~962 cm⁻¹ and both 231 isotopic compounds. It can be observed that in FeHAp there is a continuous decrease in 232 233 intensity and, at the same time, substantial increase in width with a characteristic change in behavior at $x \sim 0.2$ (Fig.6a). On the other hand, FeDAp appears to become disordered 234 faster, reaching the same asymptotic tendency at $x \sim 0.1$ (Fig. 6b). This behavior 235 characterizes the other internal modes of the PO₄ tetrahedron, as shown in Figure 7. 236 237 Again, in the FeHAp samples there is a slower continuous decrease in intensity than in FeDAp. Refinements of the XRD patterns for FeHAp and FeDAp revealed the same Fe 238 239 content for same nominal x and therefore the differences in the disorder could not be due 240 to differing amounts of Fe concentration in the two sample sets. The observed increased disorder is in agreement with the increased distortion index of the phosphate tetrahedron 241 in the Fe doped samples compared to the pure DAp. The distortion index was calculated 242 from the P-O bond lengths found from the simultaneous Rietveld refinement of the X-ray 243 244 and neutron diffraction patterns of the FeDAp sample series (Kyriakou et al. 2013). The

obvious modification of the induced disorder by the deuteron substitution that affects the long-range order of the internal modes could be due to the additional processing (heat treatment at 600 °C during ion exchange of H with D) that would increase the disorder. In any case it is worthwhile to conduct further investigation since disorder may also mark an anharmonic phonon behavior.

In conclusion, we have studied by Raman and IR spectroscopy two sets of 250 251 hydroxyapatite compounds with varying degrees of atomic Fe substitution for Ca from 0 252 to 6%; one set of samples with H and the other deuterated. Only two modes were found to be affected by the deuteration, related to the OH group at \sim 3572 cm⁻¹ (internal 253 stretching mode of OH) and ~631 cm⁻¹ (vibrational mode of Ca₃-OH). Iron substitution 254 for Ca did not significantly affect the frequency of the modes, but damped the intensity of 255 the internal modes substantially, whereby increasing their width. This is a clear indication 256 257 of increased lattice disorder with increasing degree of Fe substitution. It was also found that HAp is less sensitive to disorder than the deuterated samples. 258

259 Implications

Although Fe is a minor substituent for Ca in the structure of physiological hydroxyapatite, it affects their properties decidedly (Medeiros et al. 2002, De Vernejoul et al. 1984). Motivated by the disagreement on the lattice modifications induced by the Fe substitution, and a recent relevant study by combined XRD and neutron scattering on Feand deuteron-substituted hydroxyapatite (Kyriakou et al. 2013), we have carried out optical measurements on the same series of samples.

In this work, hematite has been detected even in samples with the smallest Fe concentration, which was below the detection limits of the XRD measurements,

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268 demonstrating the advantageous discrimination level of the Raman technique (Fig. 4). We 269 have found that Fe substitution does not affect significantly the frequency of the strong 270 modes that are related to internal vibrations of the PO_4 tetrahedra. On the contrary, the 271 width and the intensity of those modes were found to be very sensitive to the structural 272 disorder introduced by the Fe substitution. The disorder is related to the breaking of 273 external bonds of the PO_4 tetrahedra, as previously found by XRD and neutron scattering 274 (Kyriakou et al. 2013) and it is smaller in the hydroapatites compared to the deuterated 275 ones The FTIR measurements have revealed modes related to OH that challenge the 276 results of *ab initio* calculations. Our data provide substantial experimental evidence to 277 test relative theories on the lattice effects of the Fe substitution.

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367 Figure captions

368	Figure 1. IR spectra of iron-doped hydroxyapatites (continuous lines) and deuterated
369	oxypatites (dotted lines) for all samples in the high (a) and low (b) frequency spectral
370	region. The arrow in (b) indicates the mode that disappears with deuteration.
371	Figure 2. Raman spectra of the low (a) and high (b) frequency regions of the $x = 0.0$
372	hydroxyapatite sample with $\times 100$ (dotted line) and $\times 50$ (continuous line) magnification.
373	Figure 3. Raman spectra of the low (a) and high (b) frequency regions of the Fe-doped (x
374	= 0.3) hydroxyapatite sample with $\times 100$ (dotted line) and $\times 50$ (continuous line)
375	magnification.

- Figure 4. (a) Raman spectra of the FeDAp samples (low frequency region). Straight lines
- indicate the peaks due to the α -Fe₂O₃ phase and arrows the Raman modes of the apatite.
- 378 (b) The dependence on the Fe nominal atomic substitution of the new modes attributed to
- 379 the hematite phase.
- Figure 5. Raman spectra of FeHAp (continuous lines) and FeDAp (dotted lines) for all Fe
- 381 concentrations. All spectra have the same scale presented on the vertical axis.
- Figure 6. The dependence on Fe concentration of the intensity (squares, left axis) and
 width (crosses, right axis) of the v₁ mode of FeHAp (a) and FeDAp (b).
- Figure 7. The dependence on Fe concentration of the intensity for two of the strong v_3
- modes, v3-1 (squares) at ~1027 cm⁻¹ and v3-3 (crosses) at ~1046 cm⁻¹ of FeHAp (a) and
- 386 FeDAp (b).
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- 388 Table 1. Samples studied, corresponding phase identification, and identified phases (from
- 389 Kyriakou et el. 2013).

File names of	Chemical formula	File names of	Nominal Fe	Identified
hydroxyapati		deuterated	content <i>x</i>	impurity
te		apatites		phases (from
				Kyriakou et el.
				2013)
FeHAp0	Ca ₅ (PO ₄) ₃ OH	FeDAp0	0.00	
FeHAp005	Ca _{4.95} Fe _{0.05} (PO ₄) ₃ OH	FeDAp005	0.05	
FeHAp01	Ca _{4.90} Fe _{0.10} (PO ₄) ₃ OH	FeDAp01	0.10	
FeHAp02	Ca _{4.80} Fe _{0.20} (PO ₄) ₃ OH	FeDAp02	0.20	α-Fe ₂ O ₃
FeHAp03	Ca _{4.70} Fe _{0.30} (PO ₄) ₃ OH	FeDAp03	0.30	α-Fe ₂ O ₃

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Table 2. Raman and IR modes and their assignments.

Mode frequency (cm ⁻¹)	Detected in the	Spectroscopic	Assignment (Elliott
	compounds	method	1994; Antonakos et
			al. 2007)
223, 244, 291, 409, 612,	all with Fe,	Raman	α -Fe ₂ O ₃ (Hematite)
1320	increase with Fe		
839, 894, 916, 925	all	Raman	v ₄ CO ₃ and/or
			HPO ₄ ²⁻ impurities
566, 600	all	IR	v ₄ PO ₄
631	FeHAp	IR	Са ₃ -(ОН)

			librational OH
			mode
963		IR	v ₁ PO ₄
1026, 1088	all	IR	v ₃ PO ₄
2633	FeDAp	IR	ОН
3572	FeHAp	IR	ОН
330 (very weak)	Dap, HAp	Raman	Ca ₃ -(OH)?
431, 448	all, decrease with Fe content	Raman	v ₂ PO ₄
581, 593, 610, 617	all, decrease with Fe content	Raman	v ₄ PO ₄
962	all, decrease with Fe content	Raman	v ₁ PO ₄
1027, 1040, 1046, 1075	all, decrease with Fe content	Raman	v ₃ PO ₄
1252, 1297	all	Raman	CO ₃ , amide and/or HPO ₄ ²⁻ impurities

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