

1 **Revision 2**

2 **Title:** High-pressure spectroscopic study of siderite (FeCO₃) with focus
3 on spin crossover

4

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15

16 **Abstract:**

17 Fe-bearing carbonates have been proposed as possible candidate host minerals
18 for carbon inside the Earth's interior and hence their spectroscopic properties can
19 provide constraints on the deep carbon cycle. Here we investigate high-pressure spin
20 crossover in synthetic FeCO₃ (siderite) using a combination of Mössbauer, Raman
21 and X-Ray Absorption Near Edge Structure spectroscopy in diamond anvil cells.
22 These techniques sensitive to the short-range atomic environment show that at room
23 temperature and under quasi-hydrostatic conditions, spin crossover in siderite takes
24 place over a broad pressure range, between 40 and 47 GPa, in contrast to previous X-
25 Ray diffraction data that described the transition as a sharp volume collapse at
26 approximately 43 GPa. Based on these observations we consider electron spin pairing
27 in siderite to be a dynamic process, where Fe atoms can be either high spin or low
28 spin in the crossover region. Mode Grüneisen parameters extracted from Raman
29 spectra collected at pressures below and above spin crossover show a drastic change
30 in stiffness of the Fe-O octahedra after the transition, where they become more
31 compact and hence less compressible. Mössbauer experiments performed on siderite
32 single crystals as well as powder samples demonstrate the effect of differential stress
33 on the local structure of siderite Fe atoms in a diamond anvil cell. Differences in

34 quadrupole splitting values between powder and single crystals show that local
35 distortions of the Fe site in powder samples cause spin crossover to start at higher
36 pressure and broaden the spin crossover pressure range.

37

38 **Keywords:** siderite, carbonate, deep carbon cycle, spin transition, diamond anvil cell,
39 Mössbauer spectroscopy, Raman spectroscopy, XANES, differential stress

40

41 **Body of the paper**

42 **Introduction**

43 Chemical and physical interactions between atmosphere, biosphere and geosphere are
44 well known and established processes on our planet. However, the mechanisms
45 governing the dynamics and the stability of materials in the deep Earth's interior are
46 still the subject of ongoing debate. The possibility of carbon cycling through the deep
47 Earth (e.g., Dasgupta and Hirschmann, 2010) is demonstrated by observations such as
48 the occurrence of diamonds from the lower mantle, carbonate inclusions in mantle
49 xenoliths and diamonds, and the presence of CO₂ in gases from volcanic eruptions
50 (e.g., Brenker et al., 2007; Harte, 2011; Walter et al., 2011). However, the size of the
51 carbon reservoir inside the Earth is still not well constrained, and is closely linked to
52 the nature of the dominant carrier(s) of carbon down to the core-mantle boundary
53 (e.g., Biellmann et al., 1993; Lavina et al., 2009; Stagno et al., 2011; Boulard et al.,
54 2011).

55 Deep carbon is predominantly stored in accessory phases as a consequence of
56 its low solubility in dominant mantle minerals (e.g., Keppler et al., 2003), where these
57 accessory phases include carbonates, diamonds/graphite, methane and carbides,
58 depending on pressure, temperature, and oxygen fugacity. In highly reducing
59 environments (i.e., low oxygen fugacity), the crystalline form of carbon is graphite or
60 diamond. At more oxidizing conditions, carbonates are favored due to the reaction
61 between elemental carbon and oxygen to form (CO₃)²⁻ groups that bond to other
62 cations such as Ca²⁺, Mg²⁺, Fe²⁺, Ni²⁺ and Na⁺ depending on the composition of the
63 original bulk assemblage.

64 The three major carbonate components in the crust and upper mantle are CaCO₃
65 (calcite), MgCO₃ (magnesite), and FeCO₃ (siderite). The presence of carbonates in the
66 Earth's interior is related to the subduction process, one of the first steps in cycling
67 carbon through the Earth. The CaCO₃-MgCO₃-FeCO₃ system has been

68 experimentally investigated at relatively low pressures and moderate temperatures (P
69 < 3.5 GPa, T < 1100° C) since many decades (e.g., Goldsmith et al., 1962; Rosenberg,
70 1967; Merrill and Bassett, 1975; Franzolin et al., 2011). Low-pressure ternary phase
71 diagrams indicate the formation of extensive solid solutions, with compositions
72 depending on pressure and temperature. Observations from geophysics (e.g., Fukao et
73 al., 2009) and natural samples (e.g., Brenker et al., 2007; Walter et al., 2011) suggest
74 that some subducted slabs penetrate the lower mantle. In this case, carbonates could
75 be stable phases due to the proposed higher oxygen fugacity in the region of the slab.
76 The properties of carbonates are, however, not well understood, and many questions
77 remain open regarding their stability, composition and geophysical properties.

78 Previous high-pressure studies on the carbonate endmembers CaCO₃, MgCO₃,
79 and FeCO₃ revealed high-pressure phase transitions in all phases (e.g., Ono et al.,
80 2005; Merlini et al., 2012; Biellmann et al., 1993; Isshki et al., 2004). Magnesite
81 shows a structural change at 115 GPa and 2100-2200 K (Isshki et al., 2004), whereas
82 magnesiosiderite experiences a volume collapse of almost 10 % between 40 and 50
83 GPa at ambient temperature, caused by a high-spin to low-spin (HS-LS) transition of
84 iron (Lavina et al., 2009). Shi et al. (2008) predicted the spin transition of Fe in pure
85 siderite by *ab initio* simulations to be between 15 to 28 GPa, while the first
86 experimental observation was reported by Mattila et al. (2007) to occur at roughly 50
87 GPa from natural siderite powder using X-ray emission spectroscopy. Lavina et al.
88 (2009) described a sharp volume collapse at 43 GPa from X-ray single crystal
89 diffraction, whereas Farfan et al. (2012) observed spin crossover in a (Mn,Mg)-
90 siderite at 46 GPa using Raman spectroscopy and X-ray single crystal diffraction. Lin
91 et al. (2012) studied Mg_{0.35}Fe_{0.65}CO₃ up to 80 GPa at ambient temperature and
92 observed the spin transition at 45 GPa, where their data suggested that the low-spin
93 state has different vibrational and elastic properties compared to the high-spin state.
94 Temperature extends the spin crossover region of magnesiosiderite (Liu et al., 2014).
95 Merlini et al. (2012) postulated that the low-spin state could contribute to the
96 stabilization of Fe-bearing Ca,Mg-carbonate (so-called dolomite-III) above 35 GPa
97 and 2000 K, and Liu et al. (2015) reported an orthorhombic phase of magnesiosiderite
98 above 50 GPa and 1400 K that also appeared to be stabilized by spin crossover. It is
99 therefore clear that iron can play a fundamental role in the behavior of carbonates at
100 high pressure and high temperature.

101 To elucidate the effect of iron on the behavior of carbonates at high pressure, in
102 particular the nature of spin crossover, we undertook a combined spectroscopic study
103 of the siderite endmember. Nuclear resonance techniques (including Mössbauer
104 spectroscopy) provide a sensitive method to detect the response of Fe atoms to
105 physical changes like spin crossover, and X-ray Absorption Near Edge Structure
106 (XANES) and Raman spectroscopies provide complementary information.

107

108 **Methodology**

109 **Synthesis**

110 Iron oxalate (FeC_2O_4) was used as a starting material to synthesize FeCO_3 .
111 Commercial FeC_2O_4 from Alfa Aesar was used for ^{57}Fe -unenriched samples, while
112 for ^{57}Fe -enriched samples, iron oxalate was synthesized by chemical reaction of ^{57}Fe
113 metal with acetic acid, forming ^{57}Fe -diacetate precipitate, $^{57}\text{Fe}(\text{C}_2\text{H}_3\text{O}_2)_2$. The
114 precipitate was then reacted in the presence of oxalic acid to produce ^{57}Fe -enriched
115 iron oxalate. All chemical reactions were run under argon to prevent oxidation of iron
116 to Fe^{3+} at ambient conditions. FeC_2O_4 powder was then sealed in gold capsules of 2.3
117 mm outer diameter and loaded into an externally heated cold seal vessel and run at
118 kbars and 360 °C for seven days, following French (1971). The obtained powder was
119 used to grow FeCO_3 single crystals using a multi-anvil press at Bayerisches
120 Geoinstitut. The powder was pressed inside Re capsules to avoid loss of Fe and
121 placed inside assemblies with LaCrO_3 heaters where temperature was measured using
122 $\text{W}_{75}\text{Re}_{25}/\text{W}_{97}\text{Re}_3$ thermocouples. The samples were annealed for 10 minutes at 18
123 GPa and 1600 °C, which yielded single crystals of approximately 5 to 50 μm
124 diameter. The same procedure was used to synthesize both ^{57}Fe -enriched (96% ^{57}Fe)
125 and unenriched (2% ^{57}Fe) FeCO_3 single crystals.

126

127 **High pressure experiments**

128 Diamond anvil cells (DACs), type BX-90 (Kantor et al., 2012), were mounted with
129 diamonds with culet sizes of 250 μm and rhenium gaskets with 120 μm starting
130 diameter hole were employed in all experiments. We used neon gas loaded under
131 pressure (1.4 kbar) (Kurnosov et al., 2008) as a pressure-transmitting medium to
132 obtain hydrostatic conditions, and we measured the wavelength of the ruby
133 fluorescence before and after each experimental run to calculate pressure (Mao et al.,
134 1986). The difference in values contributed to the pressure uncertainty, which was

135 estimated to be 1 and 3 GPa for single crystal and powder experiments, respectively.
136 All DACs were prepared by loading a ~ 15 - 20 μm diameter isomeric FeCO_3 single-
137 crystal chip and a smaller ruby crystal (Fig. 1a), with the exception of Mössbauer
138 measurements using a radioactive source where more than ten ^{57}Fe -enriched crystals
139 were used in the pressure chamber to optimize signal acquisition time and signal to
140 noise ratio (Fig. 1b). A similar series of Mössbauer measurements was also performed
141 using FeCO_3 powder, which was compressed to a pellet before loading in the DAC in
142 order to reduce porosity between the grains. All single crystals survived intact
143 throughout each experimental run without breaking or cracking. To test the degree of
144 hydrostaticity, in some experiments we placed two or three ruby crystals in different
145 positions inside the pressure chamber. In all cases we measured the pressure from
146 each ruby within the same experiment to be within ± 1 GPa of each other.

147

148 ***In situ* analytical techniques**

149 Mössbauer spectroscopy

150 ^{57}Fe Mössbauer spectra were recorded at room temperature in transmission mode on a
151 constant acceleration Mössbauer spectrometer with a nominal 370 MBq ^{57}Co high
152 specific activity source in a 12 μm -thick Rh matrix (referred to below as “in house”
153 spectra). Spectra collection took 2 to 4 days for each pressure point. A set of
154 Mössbauer spectra was also collected during compression using the Synchrotron
155 Mössbauer Source (SMS) at the European Synchrotron Radiation Facility (ESRF),
156 Grenoble, France on the Nuclear Resonance beamline (Rüffer et al., 1996) ID18. The
157 narrow (~6 neV) energy component of X-rays at the Mössbauer energy of 14.4 keV
158 was extracted from a wide spectrum of synchrotron radiation using a $^{57}\text{FeBO}_3$ single
159 crystal monochromator and focused to a beam width of size of $10 \times 15 \mu\text{m}^2$ using
160 Kirkpatrick-Baez mirrors (Potapkin et al., 2012). Each SMS spectrum took
161 approximately thirty minutes to collect. The velocity scales of all Mössbauer spectra
162 were calibrated relative to 25 μm thick $\alpha\text{-Fe}$ foil, and all spectra were fitted using the
163 software package MossA (Prescher et al., 2012a). All Mössbauer measurements were
164 performed on ^{57}Fe -enriched FeCO_3 .

165

166 Raman spectroscopy

167 Raman measurements were performed on a DILOR XY triple spectrometer using the
168 514 nm line of Ar⁺ laser with 2 cm⁻¹ spectral resolution and 2 μm spatial resolution at
169 300 mW of power. Data acquisition was performed with a frequency range extending
170 from 150 to 1300 cm⁻¹. In order to maximize the signal to noise ratio, each spectrum
171 was collected in three steps over three smaller regions of frequency, where each
172 region was measured five times using an exposure time of 60 seconds, and then
173 merged together at the end of the acquisition. Orientation effects of the single crystals
174 and hence any polarization effects on the spectra are considered to be negligible since
175 we used an unpolarized green Raman source and all crystals were maintained in the
176 same orientation during the experiment. All Raman data presented in the paper were
177 collected on the same single crystal up to the target pressure. Raman measurements
178 were performed at Bayerisches Geoinstitut, Bayreuth, Germany, and spectral fitting
179 was carried out using the software package PeakFit (Systat Software).

180

181 XANES spectroscopy

182 Fe K-edge XANES measurements were performed at ESRF at the energy dispersive
183 X-ray absorption spectroscopy (XAS) beamline ID24 (e.g., Pascarelli et al., 2006).
184 The beam was focused horizontally using a curved polychromator Si 111 crystal in
185 Bragg geometry and vertically with a bent Si mirror. The obtained cross-section is
186 about 3x5 (HxV) μm². The measured XANES spectra were normalized using the
187 Athena software package (Ravel et al., 2005), and the second-order polynomial for
188 the pixel to energy conversion parameters was calibrated using a reference α-Fe foil
189 spectrum.

190

191 **Results**

192 **Mössbauer spectroscopy**

193 Mössbauer spectroscopy provides information about the electronic, magnetic and
194 structural properties of specific elements within a material. The extremely narrow line
195 width of gamma rays (4.5 x 10⁻⁹ eV in the case of ⁵⁷Fe) provides Mössbauer
196 spectroscopy with the sensitivity to detect spin transitions in compounds with
197 Mössbauer-active nuclei.

198 The evolution of FeCO₃ Mössbauer spectra at 298 K and increasing pressure is
199 similar to spectra already reported for other non-magnetic compounds containing Fe²⁺
200 in an octahedral environment, for example ferropericlase (Fe,MgO) (Kantor et al.,

201 2006b). At pressures below 40 GPa, Fe^{2+} in octahedral coordination is represented by
202 a doublet, while at higher pressures the doublet is progressively replaced by a singlet,
203 which at pressures higher than 45 GPa is the only component present in the spectra of
204 single-crystal FeCO_3 (Fig. 2a, 2b). Mössbauer spectra of powder FeCO_3 show a
205 similar evolution with increasing pressure, but over a larger pressure interval (Fig.
206 2c).

207 The centre shift (CS) is sensitive to the s-electron density at the nucleus, while
208 the quadrupole splitting (QS) measures the electric field gradient (EFG) caused by a
209 non-symmetrical charge distribution around the nucleus. The evolution of hyperfine
210 parameters with increasing pressure for both single crystals and powder samples is
211 shown for CS and QS in Figs. 3a and 3b, respectively. The general trend of the CS of
212 is to decrease with increasing pressure, where the CS of the singlet is approximately
213 0.3 mm/s lower than the CS for the doublet (Fig. 3a). The behavior of the CS is
214 similar for single crystals and powder; however for the QS the behavior differs, which
215 will be discussed in more detail below. Here we just mention that the QS of the
216 doublet for single crystals remains essentially constant with increasing pressure with a
217 slight decrease just before the singlet appears at 45 GPa.

218 Mössbauer spectra of FeCO_3 single crystals collected in house and using the
219 SMS are slightly different. Each SMS spectrum shows significantly higher resonant
220 absorption and unequal areas of doublet components compared to spectra collected
221 using a radioactive source (Fig. 4). Both of these features are due to properties of the
222 synchrotron X-ray source, namely the low amount of non-resonant radiation and the
223 polarization of the source, which leads to selective excitations of nuclear levels.
224 However all hyperfine parameters of single crystal spectra are the same within
225 experimental uncertainty (Figs. 2, 3, 5), hence SMS and in house spectra can be used
226 interchangeably for interpreting hyperfine parameter variations. In contrast there are
227 significant differences between in house spectra of single crystal and powder FeCO_3 ,
228 including the linewidths (Fig. 5), which will be discussed below in the context of the
229 effect of stress in DACs.

230

231 **Raman Spectroscopy**

232 Raman spectroscopy provides direct insight into the vibrational behavior of ions or
233 groups of ions bonded together, such as $(\text{CO}_3)^{2-}$, and their variation as a response to
234 changes of external parameters such as pressure and temperature. In FeCO_3 , $(\text{CO}_3)^{2-}$

235 groups form planes perpendicular to the c axis with Fe occupying the interstitial space
236 between them to form octahedral sites coordinated by oxygen atoms. This structural
237 arrangement is typical for carbonates and is known as the calcite or magnesite-type
238 structure. For symmetry reasons (e.g., Rutt and Nicola 1974) the cations give rise to
239 modes that are Raman inactive. Hence, typical Raman spectra of calcite-type
240 carbonates show mainly external and internal vibrations due to the motion of $(\text{CO}_3)^{2-}$
241 ions.

242 At ambient conditions carbonate spectra show six Raman bands in the region
243 from 200 to 1800 cm^{-1} . However, above 1300 cm^{-1} only two bands are detected and
244 correspond to weak asymmetric stretching vibrations of $(\text{CO}_3)^{2-}$. We neglect these
245 bands to avoid the signal from the diamonds and focus on frequencies below 1300
246 cm^{-1} . In this region the FeCO_3 Raman spectrum has four vibrational modes (176, 274,
247 728 and 1077 cm^{-1}) (Fig. 6). Based on previous studies (Popkov et al., 1973, Rutt and
248 Nicola, 1974) we assign the two lower frequency lines to external E_g modes that are
249 translations between cation and anion groups and the two higher frequency lines to
250 internal modes, E_g and A_{1g} , that are caused by symmetric bending and stretching of
251 $(\text{CO}_3)^{2-}$, respectively. A broad low intensity peak was also observed in the region
252 around 500 cm^{-1} , which is attributed to electronic Raman scattering from the Fe^{2+} ion
253 (Rutt and Nicola, 1974). However, the peak is only visible at ambient conditions and
254 is too weak to be observed at higher pressures.

255 All bands shift to higher frequencies with increasing pressure up to 40 GPa, but
256 at slightly higher pressure there is a clear change in the behavior of Raman modes
257 (Fig. 6 and 7). The A_{1g} mode splits with the growth of a second peak at slightly lower
258 frequencies. With increasing pressure the new Raman band becomes progressively
259 more intense, growing at the expense of the higher frequency peak until the latter
260 completely disappears between 47 and 50 GPa. The internal E_g line jumps
261 discontinuously to higher wavenumbers (from 782 to 840 cm^{-1}) at roughly 43 GPa,
262 and a broad hump emerges at 755 cm^{-1} , which might also be attributed to the
263 electronic Raman scattering of the Fe^{2+} ions at high pressures. The external E_g
264 vibrations are also affected by the transition. The E_g vibrational mode at 289 cm^{-1}
265 completely disappears after 45 GPa, while the higher frequency external E_g vibration
266 shifts from 449 cm^{-1} to 543 cm^{-1} between 43 and 45 GPa (Figs. 6 and 7). Above 47
267 GPa and up to 58 GPa, the highest pressure reached by Raman spectroscopy, the new

268 HP Raman modes continue to shift to higher frequencies, although with lower slopes
269 than observed at pressures below the transition.

270 Normalized spectral intensities highlight the progressive decrease of intensity of
271 lower frequency bands between 0 and 40 GPa (Fig. 8). At higher pressures (> 45 GPa)
272 the high frequency external and internal E_g modes, which are weak and difficult to
273 distinguish in spectra near the transition pressure, gradually become more intense.

274

275 **XANES spectroscopy**

276 X-ray Absorption Fine Structure (XAFS) spectroscopy is an element-specific method
277 to study the local atomic structure within a material. XAFS spectra can be divided
278 into different regions based on the energy range from the absorption edge in the
279 spectrum. Our study focuses on the XANES region, which is generally taken to
280 extend 50 eV beyond the edge itself.

281 Fig. 9 shows normalized XANES spectra of FeCO_3 collected during
282 compression. The 1 bar spectrum is consistent with the data of Wilke et al. (2001),
283 although the pre-edge peak is less pronounced in our study due to differences in
284 energy resolution. Each spectrum is characterized by two peaks (identified in Fig. 9 as
285 “1” and “2”) below 7150 eV. Their relative intensities change at increasing pressure
286 and their positions shift to higher energies (Fig. 10). Peak shifts of the main-edge
287 energy on compression was already observed in previous studies on ferroperricite
288 (Kantor et al., 2006a; Narygina et al., 2009) and bridgmanite (Narygina et al., 2009)
289 and is related to changes in the electronic structure due to shortening of Fe-O
290 distances. This shortening is directly reflected in the shift of the first EXAFS
291 maximum between 7160 and 7180 eV to higher energy, which may be described by
292 the relation $\Delta E \times R^2 = \text{const.}$ (e.g., Bianconi et al., 1983, Wilke et al., 2007), where ΔE
293 is the energy difference between the onset of the edge and the first EXAFS maximum
294 and R is the Fe-O distance.

295 All peaks change strongly between 40 and 44 GPa, where the pre-edge region
296 shows the largest variation (Fig. 9a and 10). The pre-edge peak at ~ 7112 eV, which is
297 not well resolved in this dataset, becomes more intense and sharper above 44 GPa
298 (Fig. 9b). At ~ 7117 eV, ~ 5 eV higher than the pre-edge, a shoulder present at lower
299 pressure evolves into a better-resolved feature, so that it appears like an additional
300 pre-edge. The grey dashed area in Fig. 10 marks the energy where the appearance of
301 this feature is observed, matching the discontinuity in the trends of the main and pre-

302 edge peaks. Similarly, the main-edge region (7120-7140 eV) shows considerable
303 changes between 40 and 44 GPa. The intensities of the main peaks are reversed for
304 spectra above 40 GPa, where the second peak becomes more intense than the first
305 (Figs. 9 and 12). Additionally, all spectra above 44 GPa show the presence of a new
306 pronounced feature, identified as “3” in Fig. 9.

307 The derivative $\delta\mu/\delta E$ provides a more sensitive probe of changes in the spectra
308 (Fig. 11). The grey arrows together with the dashed lines highlight three features that
309 are related to the changes observed in the spectra: 1) the derivative maximum at ~
310 7115 eV that disappears below 44 GPa and is substituted by another maximum at ~
311 7117 eV; 2) a shoulder at ~ 7128 eV that appears above 44 GPa and sharpens on
312 further increase of pressure to 51.5 GPa; 3) the derivative minimum at ~ 7150 eV that
313 appears above 44 GPa. These changes in the pre-edge, XANES and low-energy
314 EXAFS regions imply modification of the local structure around Fe atoms and/or
315 chemical bonding between 40 and 44 GPa.

316 The intensity ratio of the two peaks at the main edge, I_1/I_2 , provides another
317 qualitative probe of changes in the local structure. The intensity ratio decreases
318 monotonously until 40 GPa, where it then sharply decreases by 18.5 % in one step to
319 44 GPa (Fig. 12). Above 44 GPa the intensity ratio does not vary significantly with
320 pressure up to 51.5 GPa, the highest pressure reached.

321

322 Discussion

323 Spin transition

324 Changes observed in the Mössbauer, Raman and XANES spectra of FeCO₃ that take
325 place between ~ 40 and 45 GPa (or 47 GPa according to Raman data) result from
326 electronic spin crossover of 3d electrons of Fe²⁺ atoms. Mössbauer spectra of single
327 crystal FeCO₃ show decreasing values of CS with increasing pressure up to 40 GPa
328 that can be explained by increasing s electron density at the Fe nucleus due to the
329 progressive reduction of interatomic distances. The new component that appears at 40
330 GPa has zero QS and a CS that is much smaller than the HS component, similar to LS
331 Fe²⁺ in ferropierclase and bridgmanite (e.g., Kantor, 2007, McCammon et al., 2010).
332 LS Fe²⁺ in octahedral coordination has no unpaired electrons with nearly spherical
333 charge distribution; hence the EFG is essentially zero because all 3d energy levels are
334 nearly equally occupied. Spin pairing changes abruptly the degree to which s-electron
335 density at the Fe nuclei is shielded by 3d electrons; hence the CS drops considerably.

336 The decrease in CS at the transition is roughly 0.3 mm/s, which is higher than the
337 value observed for ferropicrinite (~ 0.17 mm/s, Kantor 2007), but lower than for iron-
338 organic complexes (~0.5 mm/s, Gütlich et al., 1978) at ambient or low (< 5 GPa)
339 pressure.

340 Raman spectroscopy also provides evidence of the spin transition in FeCO₃, but
341 through the interatomic vibrations in the sample that are probed by inelastic scattering
342 of light. Unit cell compression between 1 bar and 40 GPa causes a shift of the Raman
343 bands to higher frequencies, to a first approximation following eq. (1):

344

$$345 \quad v = \frac{1}{2\pi} \sqrt{\frac{F_r}{\mu}} \quad (1)$$

346

347 where v is the frequency, F_r is the restoring force and μ is the weighted mass of the
348 molecule or group of atoms.

349 Between 40 and 47 GPa the A_{1g} mode shifts abruptly to lower frequency,
350 comparable to observations by Farfan et al. (2012) and Lin et al. (2012) for natural
351 siderite enriched in manganese and magnesium, respectively. The A_{1g} mode is
352 attributed to the internal stretching vibration of the carbonate groups. It is therefore
353 strongly dependent on the distance between C and O, and its frequency will decrease
354 with increasing distance. The shift to lower frequency is thus a clear sign of an
355 increase of the C-O distance. This interpretation is consistent with observations by
356 Lavina et al. (2010a), who observed an increase in the C-O bond length in natural
357 siderite at 43 GPa and attributed it to structural re-organization caused by spin pairing
358 of Fe²⁺ atoms, where shared oxygen atoms that are shared by CO₃-groups and Fe²⁺ are
359 displaced due to the volume collapse of the octahedral site. In contrast, we observed a
360 frequency increase of the external and internal E_g bands at the spin transition. The
361 external modes, which in FeCO₃ are translational modes between the anion (CO₃)²⁻
362 and cation (octahedral Fe²⁺ coordinated by oxygen) sites, are clearly influenced by the
363 volume collapse of Fe-O octahedra at the transition. The distances between carbonate
364 groups and cations become shorter; hence the frequency increases. The internal E_g
365 mode, which is caused by in-plane symmetric bending of the carbonate molecule,
366 shifts to higher frequency due to the systematic decrease in the O-O bond length after
367 the transition (Lavina et al., 2010a), which induces higher vibrational frequencies due
368 to the decrease of the atomic distance between neighboring oxygen atoms.

369 Both Mössbauer and Raman data highlight an important point regarding spin
370 pairing in FeCO₃: under quasi-hydrostatic conditions the process occurs over a
371 pressure interval of at least 5 GPa (Figs. 2, 6, 7 and 9). The integrated area of the two
372 components in the Mössbauer spectra and the two A_{1g} bands in Raman spectra allow
373 the percentage of iron atoms in the HS and LS states to be estimated. From
374 Mössbauer spectra at 41 GPa, 12% of Fe²⁺ atoms are in the LS state. This observation
375 suggests that the proportion of Fe atoms changing to the LS state does affect the local
376 symmetry and hence the position of the surrounding atoms, inducing short-range (on
377 the Ångstrom scale) local “structural relaxation” that broadens the transition pressure
378 range. Indeed, since there is a mixture of Fe atoms in HS and LS in the transition
379 region that progressively change to LS, a local structural relaxation that results in
380 stretched bonds between Fe and O will enable some Fe atoms to persist in the HS
381 state over a broader pressure range, being less compressed by the surrounding oxygen
382 atoms.

383 While a detailed quantitative analysis of XANES spectra incorporating *ab initio*
384 simulation is beyond the scope of this study, our XANES data can be used to
385 investigate the degree to which spin crossover in FeCO₃ can be detected qualitatively,
386 and how well changes in the local structure can be derived from empirical analysis of
387 the spectra.

388 The most striking change in the pre-edge region is the abrupt appearance of an
389 additional peak at 7117 eV at 44 GPa. This feature shifted from higher energy, where
390 it was present only as a shoulder in the onset of the main edge. This feature is related
391 to excitations of 1s to 4p electron states of the valence band (e.g., Caliebe et al.,
392 1997), which shift to lower energy with decreasing Fe-O distance. In the actual pre-
393 edge region at ~ 7112 eV which is related to 1s → 3d transitions, a change from a
394 doublet feature for HS to a singlet in LS is expected as shown by Westre et al. (1997)
395 by experiment and multiplet analysis on octahedral high and low spin complexes. The
396 quality of the data shown here is insufficient for a detailed analysis of the pre-edge
397 due to limited energy resolution and a high level of noise. Nevertheless, a detectable
398 “sharpening” of the pre-edge peak between 40 and 48 GPa can be observed as seen in
399 Fig. 9b. Westre et al. (1997) explains the change in the pre-edge region as a change in
400 the accessible electronic states that contribute to the pre-edge after the change in spin
401 state of 3d electrons. In the HS state the octahedral crystal field splits the 3d electron
402 levels into e_g and t_{2g} states, giving rise to three possible transitions from the 1s level.

403 Two of the levels are close in energy; hence an apparent doublet is observed for
404 octahedral site symmetry (see also Westre et al., 1997, Wilke et al., 2001). In the LS
405 state only the e_g level is unoccupied, so only one transition is possible.

406 In the energy region of the main edge (7120-7140 eV) the peaks in the spectrum
407 of FeCO₃ correspond to a combination of $1s \rightarrow 4p$ electronic transitions and/or
408 multiple scattering of the photoelectron by neighboring atoms (Fig. 9a, peaks 1 and 2)
409 (Caliebe et al., 1997, Mottana, 2004). The positions of these peaks shift to higher
410 energy from 1 bar to 40 GPa (Fig. 10) due to unit cell compression and the resulting
411 decrease in Fe-O bond length, that is also responsible for the increase in the intensity
412 of peak 2 with increasing pressure. The decrease in intensity is likely related to a
413 decrease in the density of p -like states in that energy region as a consequence of
414 changes in bond length. Between 40 and 44 GPa where spin crossover occurs, the
415 spectra show an abrupt change in the relative amplitudes of peaks 1 and 2, the
416 appearance of peak 3 and the sudden shift to higher energies of the first EXAFS
417 oscillation (roughly from 7170 to 7180 eV, see Figs. 9 and 11). Again, these changes
418 are directly related to the collapse of the Fe coordination polyhedron, i.e., the change
419 in Fe-O distance (Wilke et al., 2007, Bianconi et al., 1983). In addition, the
420 appearance of peak 3 above 44 GPa could arise from multiple scattering of the
421 photoelectron from the neighboring O atoms, which move closer to the Fe atoms after
422 the transition.

423 While a comprehensive quantitative analysis of the spectra in terms of structural
424 changes would require *ab initio* calculations that are beyond the scope of this paper,
425 we may use the relation $\Delta E \times R^2 = \text{const.}$ to estimate the change in the Fe-O distance
426 with pressure from the XANES spectra. To perform this analysis, we use the spectral
427 feature that is located at 7172.5 eV at 0 GPa and define the onset of the edge at the
428 position of the pre-edge (7112 eV) (see also Wilke et al., 2007). This analysis yields
429 $\Delta E_0 = 60.5$ eV at 0 GPa, $\Delta E_{40} = 68.9$ eV at 40 GPa and $\Delta E_{48} = 75$ eV at 48 GPa.
430 Assuming that the constant in the relation given above is independent of pressure, we
431 can write for 0 and 40 GPa:

$$432 \quad \Delta E_0 \times R_0^2 = \Delta E_{40} \times R_{40}^2, \quad (2)$$

433 which can be rearranged to

$$434 \quad (\Delta E_0 / \Delta E_{40})^{0.5} = R_{40} / R_0. \quad (3)$$

435 From the two values obtained at 0 GPa and 40 GPa, a shortening of the Fe-O distance
436 by a factor of 0.94 is estimated, while across the spin transition from 40 and 48 GPa,

437 we obtain a value of 0.95. This is comparable to the value of 0.92 obtained by Lavina
438 et al. (2010) based on single crystal diffraction data (their Fig. 7). The consistency of
439 results is striking, particularly considering the large simplifications made in our
440 analysis, and highlights the sensitivity of XANES to the local environment.

441 Our results clearly demonstrate the capability of XANES spectra to detect spin
442 crossover in FeCO_3 . Indeed, the sensitivity has improved markedly since previous
443 studies of other geophysically relevant phases such as ferropericlaite (Kantor et al.,
444 2006a) and bridgmanite (Narygina et al., 2009) due to the newly upgraded
445 experimental setup for high-pressure XAFS studies on beamline ID24 at ESRF,
446 together with the implementation of more hydrostatic conditions in the DAC, for
447 example through the use of neon as a quasi-hydrostatic pressure medium. The higher
448 sensitivity could also be related in part to the specific structural arrangement in
449 carbonates (planes of $(\text{CO}_3)^{2-}$ alternated by octahedrally coordinated cations).
450 Simulations to quantitatively evaluate the effect of spin crossover on the XAFS
451 spectra are ongoing and will be reported elsewhere.

452

453 *Dynamic spin state distribution*

454 Previous X-ray diffraction (XRD) experiments on siderite and ferromagnesite (e.g.,
455 Lavina et al., 2009, 2010a, Merlini et al., 2012, Lin et al., 2012) interpreted spin
456 crossover to be a relatively sharp discontinuity taking place at roughly 43 GPa and
457 causing a discontinuous volume reduction of the unit cell. Indeed, XRD is a technique
458 sensitive to long-range structures and cannot probe short-range order such as, for
459 example, HS and LS state distributions. Raman, Mössbauer and XANES spectroscopy
460 probe the local structure of neighboring atoms and groups of atoms and are therefore
461 sensitive to short-range order features that are not always evident from XRD. Our
462 results, i.e., a successive change of all spectra from HS to LS across a relatively wide
463 pressure range, support the idea of a dynamic nature of spin crossover, which involves
464 a non-static distribution of HS and LS states until the transition is complete. Indeed, a
465 static distribution of the two spin states, which could be considered as a long-range
466 order feature, should be easily resolved by XRD, but so far it has not been reported.
467 Hence, we propose that if a static distribution of the two spin states can be excluded
468 because it was not observed by XRD, the only way to explain the broad and transient
469 pressure range over which spin crossover takes place is a dynamic distribution of HS
470 and LS states in the transition region. On the basis of our result, we suggest that

471 previous XRD studies could be re-examined to investigate if certain features (i.e., the
472 peak splitting and diffuse scattering in Lavina et al., 2010a or the small deviation in
473 the low pressure equation of state before the volume collapse in Merlini et al., 2013),
474 could reflect a more complex transition.

475 A direct consequence of the broad spin crossover pressure range is highlighted
476 by the weakening of all Raman E_g bands as spin crossover is approached, followed by
477 an intensity increase once the final LS state is achieved (Fig. 8). All modes become
478 progressively weaker during compression due to the gradual pressure increase that
479 affects the interaction between the probing laser and the sample, hence reducing
480 signal quality. However, the laser-sample coupling does not explain the almost
481 complete disappearance and reappearance of the lower frequency modes at the
482 transition. Lavina et al. (2009) described a change in the color of the crystals, where
483 they become progressively darker during and after spin pairing. We observed a
484 similar behavior, where crystals changed from a light reddish color to a slightly
485 darker one. Since we used a green laser with 514 nm wavelength one might attribute
486 the decrease in intensity of the Raman bands to a change in phonon scattering in the
487 visible light region. However, we consider this not to be the case since this does not
488 account for the regain in intensity of the other Raman bands after the transition, and
489 also the internal symmetric stretching vibration A_{1g} is not affected. Hence, we
490 attribute these intensity minima (Fig. 6) to static disorder caused by spin pairing
491 through the decreased percentage of atoms scattering in the HS state, and
492 consequently the increased amount of scattering by LS atoms. Once the transition is
493 complete, all atoms/molecules in the LS state scatter again equally based on their
494 different vibrational symmetries and new and more intense modes reappear in the
495 spectra. One mode is completely lost after the transition (the external vibration that
496 occurs at 176 cm^{-1} at 1bar); however it is the lowest frequency E_g translatory
497 oscillation and might be simply too weak to be detected.

498

499 *Grüneisen parameters*

500 We calculated the mode Grüneisen parameters (γ_i) for each vibrational mode with
501 increase of pressure using the following equation:

502

$$503 \quad \gamma_i = \frac{K}{\omega} \frac{d\omega}{dP} \quad (4)$$

504

505 where the bulk modulus (K) is 117 ± 1 GPa (Zhang and Reeder, 1999) and 148 ± 12
506 GPa (Lavina et al., 2010) for the HS and LS states, respectively. We also calculated
507 the pressure shift ($d\omega/dP$) in order to verify that the differences in the Grüneisen
508 parameters before and after the spin transition were not caused simply by the use of a
509 different bulk modulus (Table 1).

510 The Grüneisen parameters of HS FeCO_3 discriminate the two different vibration
511 typologies, internal and external, characterized by smaller and larger values,
512 respectively. In both cases our results match the observations of Santillán and
513 Williams (2004) that were based on infrared measurements. Indeed, the Grüneisen
514 parameters reflect the effect that changing the volume of a crystal lattice has on its
515 vibrational properties, in this case due to the increase of pressure. The higher values
516 of the external translatory oscillations are consistent with values reported by Ross and
517 Reeder (1992), who described by means of single-crystal XRD the higher
518 compressibility of Fe-O octahedra with respect to the relative incompressibility of the
519 CO_3 group. Hence, the vibrations associated with the relative translation of the cation
520 with respect to the anionic group are expected to have higher Grüneisen parameters,
521 where the cation site is more affected by the pressure increase due to its higher
522 compressibility.

523 The Grüneisen parameters measured at spin crossover are also characteristic of
524 the vibrational typology: the internal modes show no variation (A_{1g}) or a moderate
525 increase (E_g) in their values; whereas the remaining external mode (E_g) drops
526 drastically from 1.16 to 0.72, similar to observations by Lin et al. (2012) for
527 ferromagnesite. Indeed at spin pairing the Fe-O octahedra shrink, pulling the oxygen
528 atoms towards the iron atoms and consequently stretching the C-O bonds in the
529 carbonate groups. The external vibration (translatory oscillation) is affected by the
530 shape and the size of the cation site, which is less distorted (Lavina et al., 2010a) and
531 has a smaller volume after the transition due to the more compact environment (lower
532 Grüneisen parameter). On the other hand, the stretching of the C-O bond does not
533 influence the carbonate groups; hence the Grüneisen values for the internal vibrations
534 remain constant for symmetric stretching (A_{1g}) or show a slight increase for
535 symmetric bending (E_g), where the difference is probably related to the different
536 vibrational geometry.

537 Compared to the study of Lin et al. (2012), we obtain lower values of calculated
538 Grüneisen parameters for both HS and LS states. This may be attributed to the

539 difference in composition and its effect on the bulk modulus, since the latter is the
540 only adjustable parameter used to calculate γ_i (Eq. 4). Indeed, the frequency variation
541 with pressure ($d\omega/dP$) of each mode displays similar values in both studies.

542 Our calculations relate to the mean mode Grüneisen parameter for all observed
543 vibrations in HS and LS states. Since we recognize that not all phonon modes are
544 taken into account, for example all Raman bands above 1300 cm^{-1} are excluded, it is
545 not appropriate to compare it with the thermodynamic Grüneisen parameter (e.g.,
546 Santillán and Williams, 2004, Lin et al., 2012, Liu et al., 2014). Instead our aim is to
547 provide an indication of the difference in compressibility of the FeCO_3 structure
548 before and after the transition based only on a limited number of vibrations. From our
549 calculations, the average Grüneisen parameters of HS and LS states are, respectively,
550 0.69 and 0.42. As stated by Lin et al. (2012), the LS state is expected to have a
551 thermodynamic Grüneisen parameter lower than ~ 1.1 , which is the value for the HS
552 state. Our available data suggests the decrease of the Grüneisen parameter to be \sim
553 30%, but this should be tested by further experiments.

554

555 **Effect of stress on Mössbauer spectra collected in a DAC**

556 Differences between powder and single crystal Mössbauer spectra of FeCO_3 provide
557 information on the effects of deviatoric and local stresses caused by non-hydrostatic
558 conditions. Any DAC, due to its intrinsic geometry, causes cylindrically symmetric
559 non-hydrostatic stress fields during compression that affect all materials with non-
560 zero yield strength (e.g., Meng et al., 1993; Dubrovinsky and Dubrovinskaia, 2004;
561 Kenichi T., 2000; Zhao et al., 2010). For any type of pressure-transmitting medium
562 there is a pressure limit at which hydrostatic behavior of the medium is lost and
563 consequently deviatoric stresses develop within the sample. These stresses are
564 macroscopic when they are caused by the action of the transmitting medium on the
565 sample and microscopic when they develop at the grain boundaries of polycrystalline
566 samples, i.e., powders (Weidner et al., 1994). Observables from Mössbauer
567 experiments provide qualitative insight into the effects of deviatoric stress on the
568 nuclear behavior and spin crossover of Fe atoms in siderite powder in DACs gas-
569 loaded with neon.

570 The observed differences between powder and single crystal Mössbauer spectra
571 arise from the action of microscopic local stresses present at grain boundaries. Indeed,
572 the powder sample was compressed with the intention of producing a pellet with

573 reduced interstitial porosity (20 μm size in diameter), so that Ne was acting more as a
574 lubricant between powder grains than as an effective pressure-transmitting medium.
575 This effect is apparent from the influence of pressure on the FWHM of Mössbauer
576 absorption peaks for powder and single crystals experiments (Fig. 5). With the
577 exception of the 1 bar powder spectrum, all peaks at higher pressures are at least
578 twice as broad for powder compared to single crystals, where the difference increases
579 at higher pressure. The broadening originates from pressure gradients in the gasket
580 hole, which in turn create pressure gradients within the sample. Due to the relatively
581 large cross section of the gamma-ray beam in our laboratory ($\sim 500 \mu\text{m}$) compared to
582 the size of the gasket hole, at each pressure point the resultant Mössbauer spectrum
583 that is measured represents the combination of several spectra collected at slightly
584 higher and lower pressures compared to the pressure measured by ruby fluorescence
585 due to the pressure gradient. In this way spectra characterized by higher and lower CS
586 are merged together and the components in the final spectrum appear broadened.
587 There is little difference in the measured CS of powder and crystals (Fig. 3a), but the
588 differences measured for QS are significant (Fig. 3b). Indeed, the QS values for
589 powder spectra are systematically (except at 1 bar) at least 15% higher than for the
590 single crystal measurements. The physical explanation is straightforward: powder
591 experiments involve grain-grain interactions that develop under compression and
592 cause the formation of deviatoric stress acting on the local structure around the Fe
593 atoms, causing the Fe-O octahedra to be more distorted than in “regular” quasi-
594 hydrostatic experiments. The distortion induces variations in the shape and size of the
595 charge distribution around the Fe nuclei and ultimately on the electric field gradient,
596 hence on the quadrupole splitting.

597 The macroscopic and microscopic stresses that develop inside the gasket hole
598 also influence spin crossover. Important differences between powder and single
599 crystal Mössbauer spectra are: 1) spin crossover starts at lower pressure for single
600 crystals ($\sim 40 \text{ GPa}$) compared to powder ($\sim 50 \text{ GPa}$); and 2) the pressure range over
601 which spin crossover occurs is smaller for single crystals ($\sim 5 \text{ GPa}$) than for powder
602 ($> 10 \text{ GPa}$). Moreover, the sudden increase ($\sim 15 \%$) of QS in powder spectra at the
603 start of spin crossover is caused by enhanced distortion of Fe octahedra in the crystal
604 structure. Indeed at the spin pairing transition, the dynamic distribution of HS and LS
605 states, with respectively larger and smaller octahedra, unbalances the surrounding

606 atomic distribution, stretching or shortening atomic bonds that favor the formation of
607 local stresses.

608

609 **Implications**

610 On the basis of this study, we found that spectroscopic techniques provide insight into
611 the short-range atomic order during FeCO₃ compression from 1 bar to 58 GPa. In
612 addition to XRD, which measures long-range atomic order in crystals, we observed
613 that the spin transition in synthetic FeCO₃ is a transient process that takes place over a
614 finite range of pressure under quasi-hydrostatic conditions. Moreover, our results
615 demonstrate that spin crossover is a dynamic process during which the Fe atoms in
616 HS and LS states coexist until the LS configuration is fully reached. A static
617 distribution of the two spin states can be excluded based on previous XRD studies
618 that reported only a sudden collapse of unit cell volume, but no other ordering
619 features.

620 The calculated Grüneisen parameters for all observed Raman modes agree with
621 the structural arrangement for FeCO₃ in HS and LS states reported previously in
622 single crystal XRD studies. While the CO₃ groups are relatively incompressible over
623 the entire pressure range investigated, Fe-O octahedra show a drastic change in their
624 stiffness after spin crossover, becoming more compact and hence harder to compress.

625 Deviatoric stress, both macroscopic and microscopic, causes remarkable
626 changes in the high-pressure behavior of FeCO₃. Local distortions of the Fe site cause
627 spin crossover to start at higher pressure and a broadening of the spin transition
628 pressure range.

629 Our results contribute to understanding the fate of iron-bearing carbonates
630 during subduction. Our experiments clearly show that deformation, which in the Earth
631 is caused by differential stress in the matrix due to processes such as convection,
632 increases the depth at which spin crossover starts and broadens its pressure range
633 beyond 5 GPa. This conclusion is valid both for our room temperature experiments as
634 well as for mantle conditions. Combined with the results of Liu et al. (2014) who
635 demonstrated that temperature extends the spin crossover region to higher pressure,
636 spin crossover in magnesiosiderite would start at much greater depths (> 150 km)
637 than postulated so far in previous work (e.g., Lavina et al., 2009), and would be
638 prolonged over a depth range of at least 300-350 km instead of being a sharp
639 discontinuity in the shallower part of the lower mantle. However, further experiments

640 are required in order to quantify more precisely the effect of temperature on natural
641 systems under the influence of deviatoric stress.

642

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879

880 **Figure captions**

881

882 **Figure 1.** Photographs of FeCO₃ samples in DACs at the indicated pressures with (a) one single
883 crystal and (b) multiple single crystals, each surrounded by Ne. Both loadings contain a small ruby
884 crystal as a pressure calibrant.

885

886 **Figure 2.** Room temperature Mössbauer spectra of FeCO₃ (a) single crystals and (c) powder collected
887 in house at the indicated pressures. (b) SMS spectra of one FeCO₃ single crystal collected at ESRF,
888 Grenoble (France). Solid circles: experimental data; black lines: full transmission integral fit; blue and
889 red lines: individual subspectra. Note that the subspectra do not add up to the total spectrum due to the
890 properties of the full transmission integral fit. Percentage bars indicate the relative absorption.

891

892 **Figure 3.** Room temperature centre shift (a) and quadrupole splitting (b) (relative to α -Fe) of FeCO₃ as
893 a function of pressure. Solid circles: in house single crystals experiments; open circles: in house
894 powder experiment; solid triangles: SMS single crystal experiment.

895

896 **Figure 4.** Comparison of room temperature FeCO₃ Mössbauer spectra: (a) SMS collected on one single
897 crystal at 19 GPa; (b) in house collected on multiple single crystals at 18 GPa; and (c) in house
898 collected on powder at 20.5 GPa. The area asymmetry of the doublet in (a) is due to polarization of the
899 synchrotron Mössbauer source. Percentage bars indicate the relative absorption.

900

901 **Figure 5.** Effect of pressure on the full-width at half-maximum (FWHM) of FeCO₃ spectra. Open
902 circles: in house powder; solid circles: in house single crystals; solid triangles: SMS single crystal.
903 Dashed lines indicate the pressure at which spin crossover starts for single crystal (41 GPa) and powder
904 (52 GPa) measurements.

905

906 **Figure 6.** Raman spectra of FeCO₃ collected at ambient temperature and the indicated pressures. The
907 two modes below 500 cm⁻¹ are external or lattice vibrations of CO₃²⁻ relative to Fe atoms, while modes
908 above 700 cm⁻¹ are internal vibrations of CO₃²⁻. The insets indicate the lower frequency Raman bands
909 for the spectra on a different scale.

910

911 **Figure 7.** Pressure variation of the peak shift of observed Raman modes of FeCO₃. Dashed grey area
912 indicates the spin crossover pressure range. The frequency uncertainty is smaller than the size of the
913 symbols.

914

915 **Figure 8.** Pressure variation of Raman mode intensity normalized to the most intense mode, A_{1g}, in
916 FeCO₃. Solid circles: lower frequency external E_g vibration; open circles: higher frequency external E_g
917 vibration; solid triangles: internal E_g vibration (see also Figs. 7 and 8). Note that the lowest frequency
918 external E_g mode disappears completely above 45 GPa.

919

920 **Figure 9.** (a) Normalized XAS spectra of FeCO₃ at room temperature collected on compression at the
921 indicated pressures. The inset at the bottom right highlights the pre-edge region of the 1 bar spectrum.
922 (b) Close up on the evolution of the pre-edge region with pressure. The spectra are shifted vertically for
923 clarity.

924

925 **Figure 10.** Pressure variation of maximum peak positions for FeCO₃ XANES spectra. The peaks are
926 labeled according to Fig. 9 and the grey dashed area indicates where the transition is observed in the
927 XANES spectra.

928

929 **Figure 11.** $\delta\mu/\delta E$ derivatives of room temperature XANES spectra of FeCO₃ at the indicated
930 pressures. The vertical unit is arbitrary, and the derivatives are taken from the normalized spectra (Fig.
931 10). Grey arrows and dashed lines indicate the energies at which changes are observed at 44 GPa.

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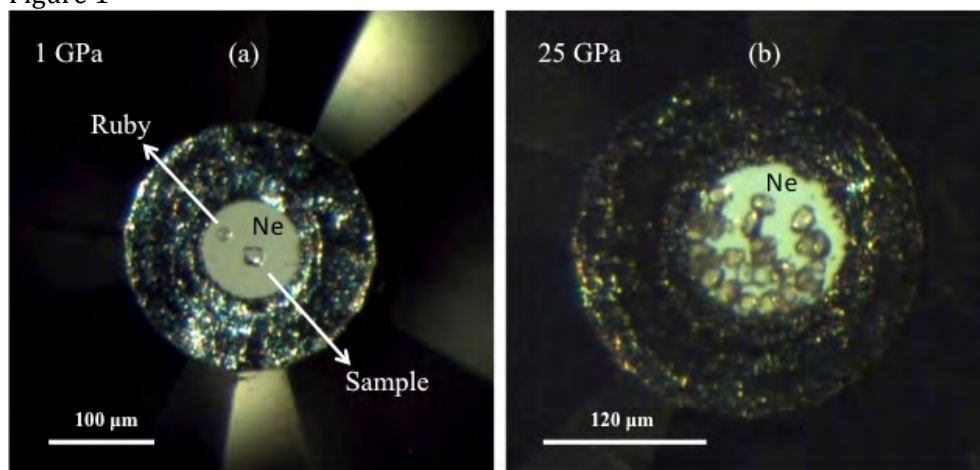
933 **Figure 12.** Pressure variation of the intensity ratio (I₁/I₂) of the main peaks (see Fig. 9) of FeCO₃
934 XANES spectra. The percentage change at the transition pressure is indicated.

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936 Figures

937

938 Figure 1

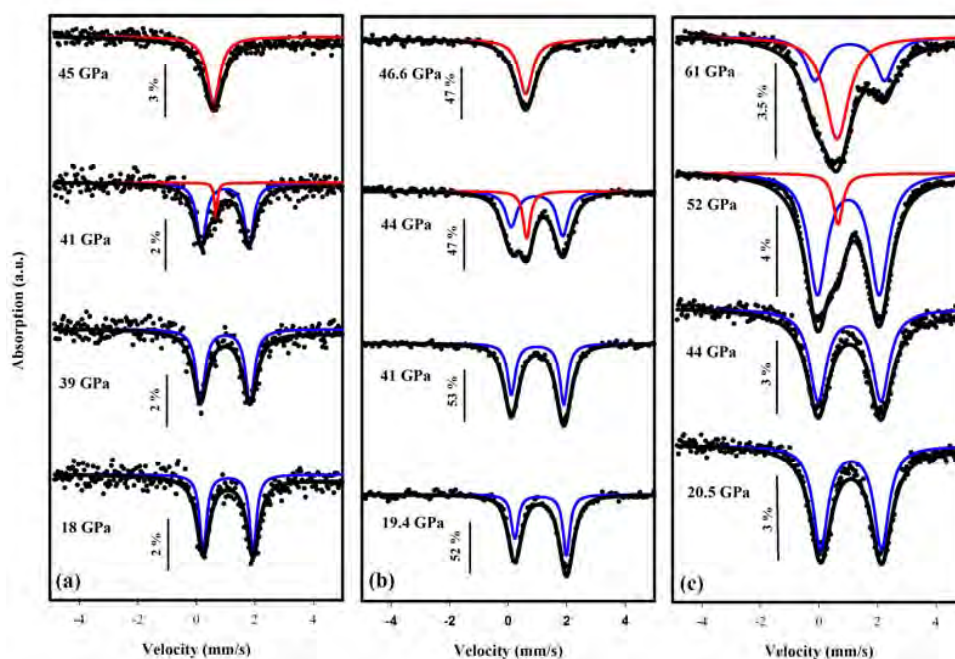


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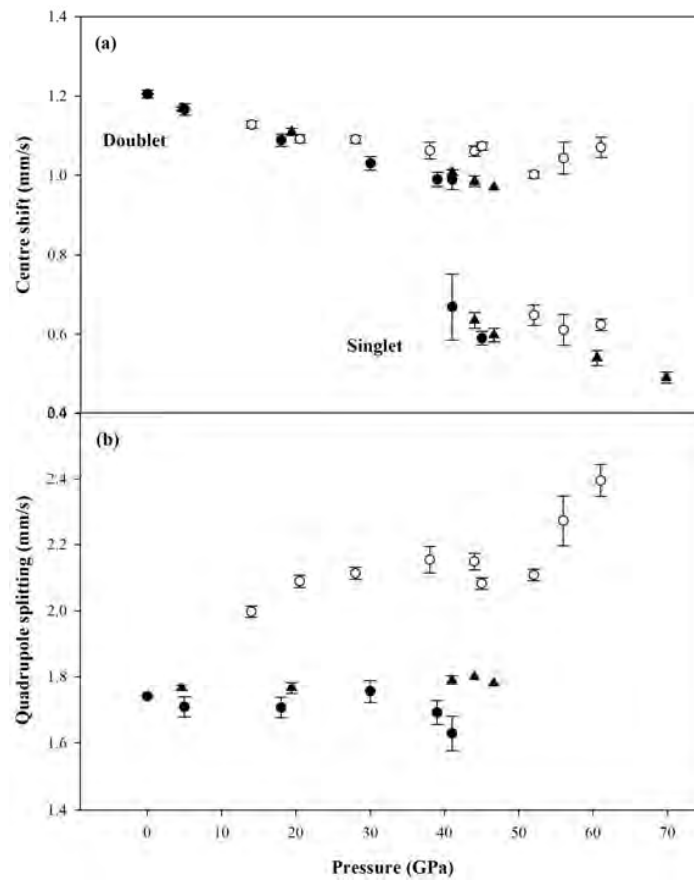
Figure 2



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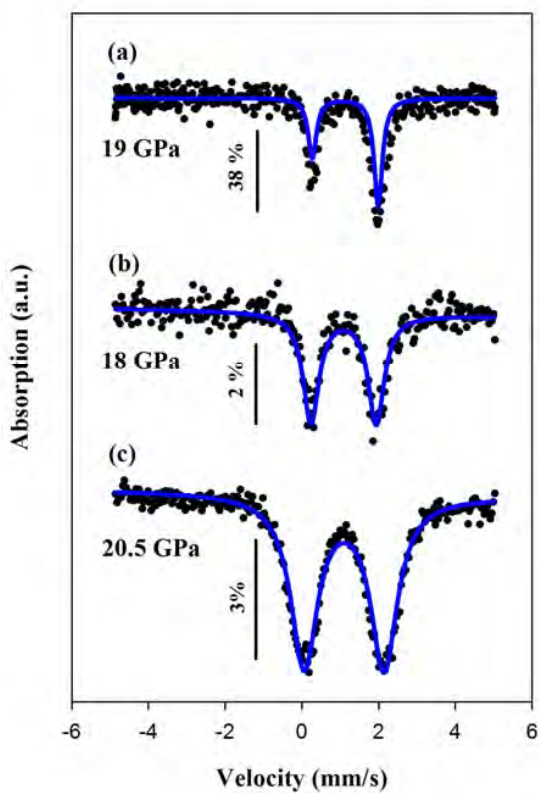
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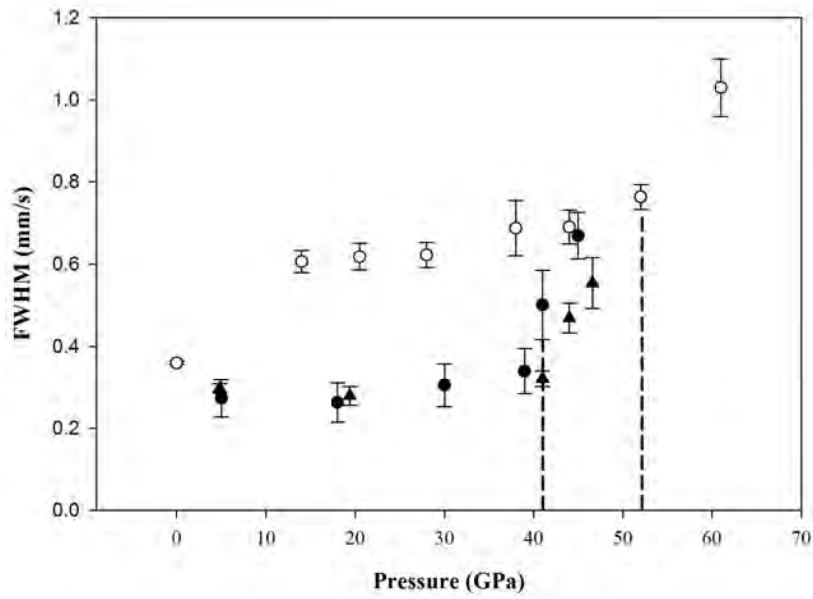
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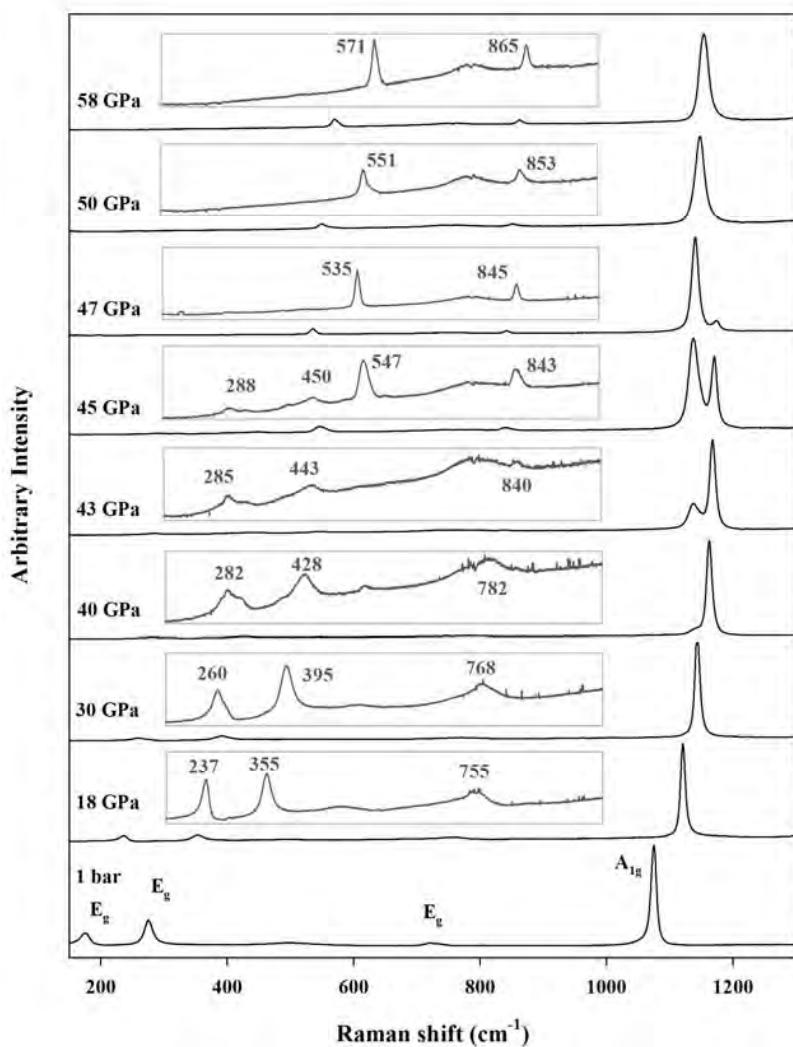
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Figure 5



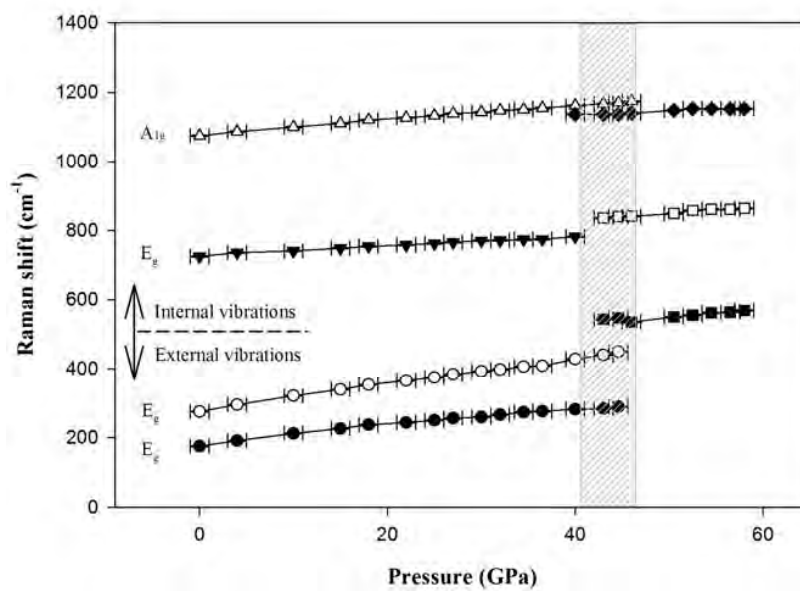
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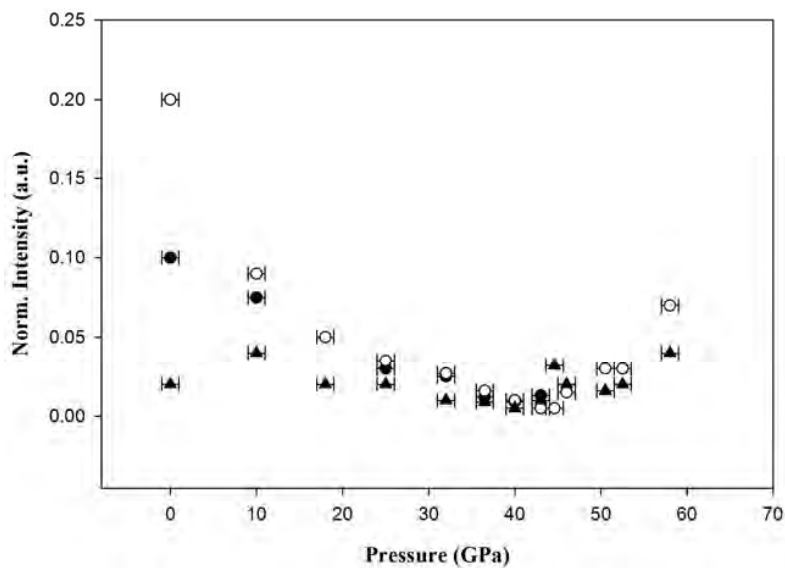


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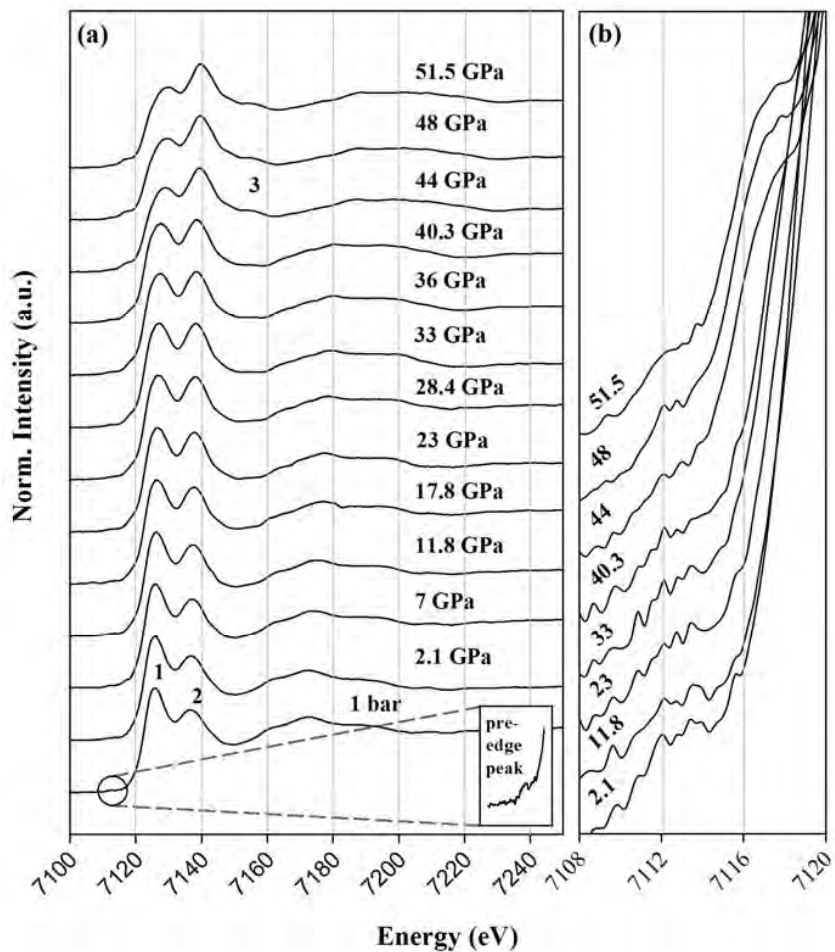


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1043 Figure 9



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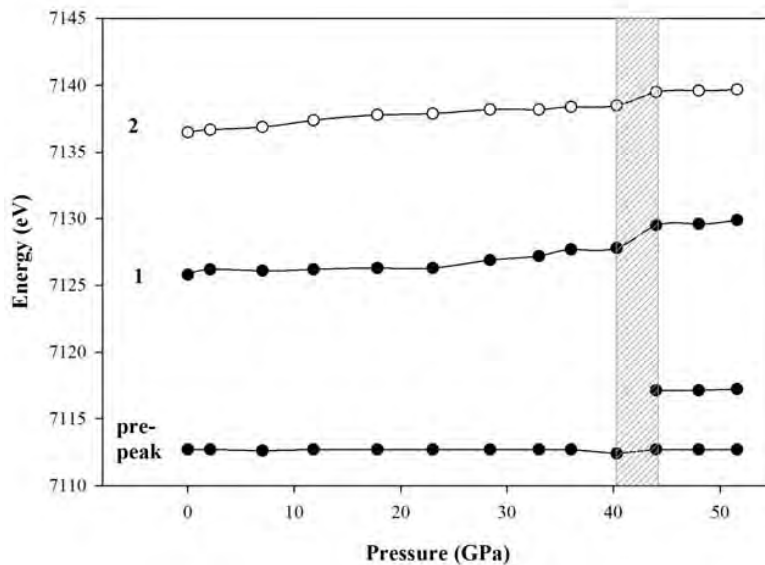
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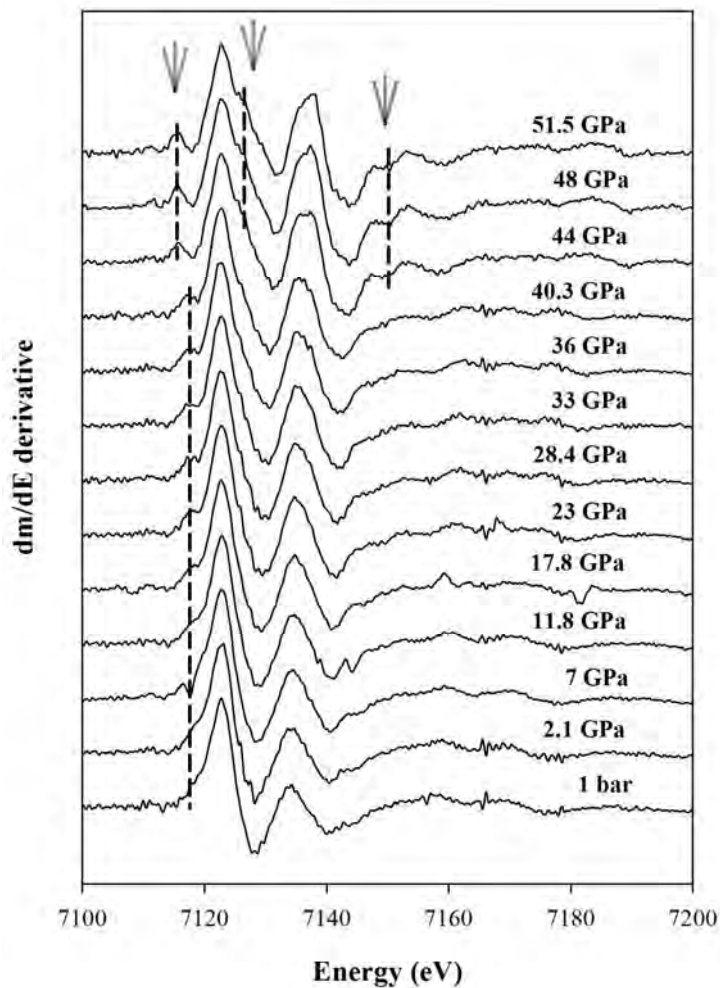
Figure 10



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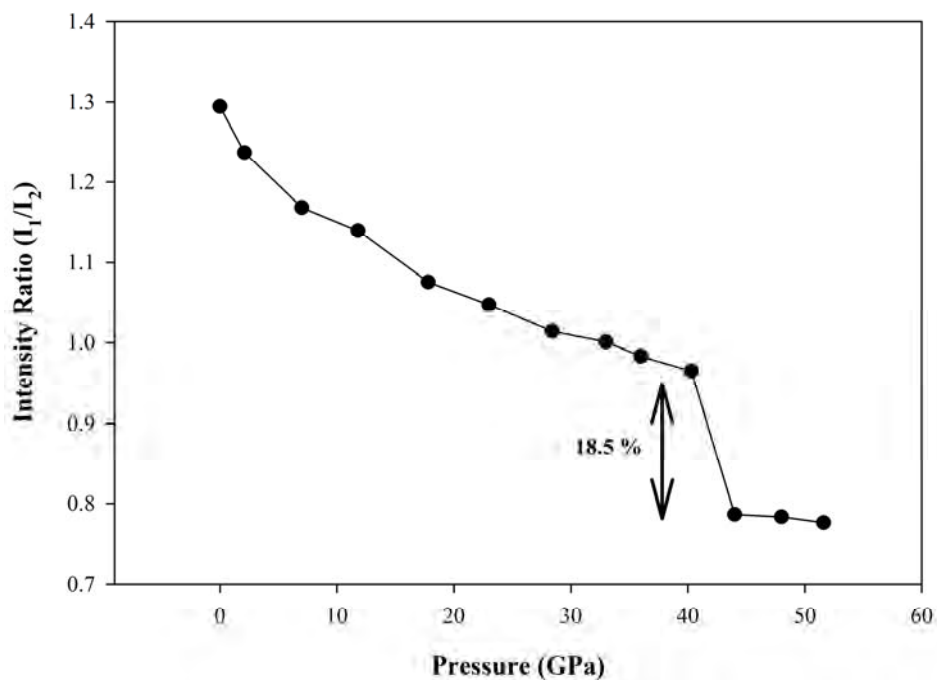
Figure 11



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Figure 12



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Tables

Table 1

Grüneisen parameter (\square_i) for Raman vibrational modes

Modes	Assignment	$d\omega/dP$ (cm ⁻¹ GPa ⁻¹)		Mean mode \square_i	
		HS	LS	HS	LS
A _{1g} (internal)	Symmetric (in plane) stretching	2.17	1.6	0.22	0.21
E _g (internal)	Symmetric (in plane) bending	1.37	1.86	0.21	0.32
E _g (external)	Translatory oscillations of the CO ₃ groups	3.82	2.68	1.16	0.72
E _g (external)		2.51		1.18	
Mean mode Grüneisen parameter for observed vibrational mode in HS state:				0.69	
Mean mode Grüneisen parameter for observed vibrational mode in LS state:				0.42	
Bulk moduli from Zhang and Reeder (1998) and Lavina et al. (2010) for HS and LS configuration respectively.					
Band assignments from Rutt and Nicola (1974).					

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