1	REVISION 3
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3	Trioctahedral Fe-rich micas: relationships between magnetic behavior
4	and crystal chemistry
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15	Running title: Structure and magnetic properties of trioctahedral micas.
16	
17	Abstract. Six Fe-bearing mica samples with different Fe ordering, $Fe^{2+}/(Fe^{2+}+Fe^{3+})$ ratio,
18	octahedral and tetrahedral composition were studied. Four micas belong to the phlogopite-annite
19	join (space group $C2/m$), two are Mg-rich annite and two are Fe-rich phlogopite, one is a tetra-
20	ferriphlogopite (space group $C2/m$) and one is Li-rich siderophyllite (space group $C2$). Thus these
21	samples had a different environment around the Fe cations and layer symmetry. These micas were
22	characterized by chemical analyses, single crystal X-ray diffraction, X-ray Absorption
23	Spectroscopy, and magnetic measurements. In samples with Fe mostly in octahedral coordination,
24	dominant magnetic interactions among Fe atoms are ferromagnetic, which reach a maximum at

25	higher $Fe^{2+}/(Fe^{2+}+Fe^{3+})$ ratios. Samples with higher $Fe^{2+}/(Fe^{2+}+Fe^{3+})$ ratio are also characterized by
26	higher values of the Curie-Weiss θ constant. Where $Fe^{2^+}/(Fe^{2^+}+Fe^{3^+})$ ratios decrease, θ values also
27	decrease. The Fe^{3+} -rich phlogopite shows predominant Fe^{3+} in tetrahedral coordination and shows
28	anti-ferromagnetic interactions with a negative value of the Curie-Weiss θ constant (i.e. θ = - 25 K).
29	Fe ordering in octahedral <i>trans</i> - and in one of the two <i>cis</i> -sites accounts for a greater θ value in Li-
30	rich siderophyllite when compared to other samples showing similar octahedral Fe content. Our
31	data suggest that Fe ³⁺ cations and other non-ferromagnetic cations hinder long range magnetic
32	ordering. This observation may produce for the different role of octahedral Fe magnetic
33	interactions that can in principle develop along long Fe-rich octahedral chains, when compared to
34	tetrahedral-octahedral interactions that are confined within the layer by the non-ferromagnetic
35	cations of the interlayer. Spin glass behavior is indicated by the dependency of the temperature to
36	produce maxima in the susceptibility curve. These maxima are related to the frequency of the
37	applied AC magnetic field.
38	

Key words: Fe-rich micas, X-ray absorption spectroscopy, chemical analysis, single crystal X-ray
structure determination, magnetic measurements.

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Introduction

43 Magnetic techniques are powerful methods to understand electronic and thermodynamic 44 properties of materials (Catalli et al. 2011; Hsu et al. 2010). When complex materials, such as 45 minerals, are involved, magnetic investigations become extremely challenging. Iron is 40 times 46 more abundant than any other magnetic element in the Earth's crust, which is predominantly 47 composed of silicates, thus suggesting why most magnetic studies on minerals are focused on iron 48 silicates such as pyroxenes (Redhammer et al. 2011; 2012). These minerals represent model 49 systems to study magnetic properties characterized by a competitive interaction within and between 50 the quasi-one-dimensional (1D) infinite chains of transition metal bearing octahedral sites. Micas 51 are typical examples of a layer structure, thus showing properties of quasi-two-dimensional (2D) 52 infinite systems. The characterization of magnetic properties of natural Fe-rich trioctahedral micas 53 was addressed by some authors (Anagnostopoulos et al. 1973; Marcelli et al. 2004; Dunlop et al. 54 2006; Pini et al. 2008, 2012), without however obtaining a full understanding of the role of local 55 topology of Fe atoms on magnetic properties. Even if limited in number, studies devoted to 56 magnetic properties of trioctahedral micas increased significantly in the few past years. These 57 studies, in part were motivated by the abundance of these minerals in nature, but also because of the 58 great variability in content, oxidation condition and local topology of Fe atoms present (Biedermann 59 et al. 2014). The correlation between magnetic properties of natural trioctahedral micas and their 60 crystal chemical and structural features have been explored previously to understand the origin of magnetic properties in minerals and to better characterize their structure, with particular reference 61 62 for Fe coordination.

In trioctahedral micas, the $Fe^{2+}/(Fe^{2+} + Fe^{3+})$ ratio can vary significantly, together with its distribution within the layer. Fe^{3+} was found to occupy both tetrahedral (T) and octahedral (M) positions, whereas Fe^{2+} was located at M positions only (Brigatti and Guggenheim 2002). Fe^{2+} for Mg²⁺ octahedral substitution defines the trioctahedral mica join from phlogopite [KMg²⁺₃AlSi₃O₁₀

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67	$(OH)_2$] to annite [KFe ²⁺ ₃ AlSi ₃ O ₁₀ (OH) ₂] (Bailey1984). Tetrahedral substitution of Fe ³⁺ for Si ⁴⁺
68	gives tetra-ferriphlogopite $[KMg^{2+}_{3}Fe^{3+}Si_{3}O_{10}(OH)_{2}]$ from phlogopite, and tetra-ferri-annite
69	$[KFe^{2+}{}_{3}Fe^{3+}Si_{3}O_{10}(OH)_{2}] \text{ from annite. In trioctahedral lithian micas the } ^{[iv]}Si^{4+}{}_{-1}{}^{[vi]}Li{}_{-1}{}^{[iv]}Al \\ ^{[vi]}Fe^{2+}{}_{-1}Si^{4+}{}_{-1}{}^{[vi]}Li{}_{-1}{}^{[vi]}Fe^{2+}{}_{-1}Si^{4+}{}_{-1}{}^{[vi]}Li{}_{-1}{}^{[vi]}Fe^{2+}{}_{-1}Si^{4+}{}_{-1}{}^{[vi]}Fe^{2+}{}_{-1}Si^{4+}{}_{-1}{}^{[vi]}Fe^{2+}{}_{-1}Si^{4+}{}_{-1}{}^{[vi]}Fe^{2+}{}_{-1}Si^{4+}{}_{-1}{}^{[vi]}Fe^{2+}{}_{-1}Si^{4+}{}_{-1}{}^{[vi]}Fe^{2+}{}_{-1}Si^{4+}{}_{-1}{}^{[vi]}Fe^{2+}{}_{-1}Si^{4+}{}_{-1}{}^{[vi]}Fe^{2+}{}^{[vi]}Fe^{2$
70	vector defines the polylithionite $[KLi_2AlSi_4O_{10}(F,OH)_2]$ - siderophyllite $[KFe^{2+}_2AlSi_4O_{10}(F,OH)_2]$
71	join (Rieder et al.1998).
72	Octahedrally-coordinated Fe cations are distributed over three crystallographic positions, i.e. in
73	the <i>trans</i> -oriented octahedron (M1) and in the two <i>cis</i> -oriented octahedra (M2 and M3). M2 and
74	M3 octahedra are symmetrically related in micas belonging to $C2/m$ space group. Fe is usually
75	disordered in M1, M2, M3 with a slight preference for the M1 site in trioctahedral micas of the
76	phlogopite-annite join, whereas it can be partially ordered in trioctahedral micas of the
77	polylithionite–siderophyllite join where the much larger Li^{1+} and Fe^{2+} cations occupy the <i>trans</i> -site
78	and one of the <i>cis</i> -oriented octahedral sites (commonly, M3) and Al^{3+} occupies one of the <i>cis</i> -sites
79	(commonly, M2) (Brigatti and Guggenheim 2002).
80	An antiferromagnetic behavior was observed for micas at T values close to 10K and in presence
81	of a magnetic field, following from magnetic moments generated by in-plane ferromagnetic
82	interactions involving Fe ²⁺ -Fe ²⁺ ions (Coey 1985, 1987; Coey and Ghose 1988; Coey et al. 1981,
83	1982, 1984). Possible magnetic ordering was suggested at T values of < 10 K and in a strong
84	magnetic field (Ballet and Coey 1982).
85	Beausoleil et al. (1983) demonstrated that magnetic properties for micas, and specifically for
86	trioctahedral micas close to annite in composition, are remarkably anisotropic, with magnetic
87	susceptibility changing in value if measured in-plane or out-of-plane, following their 2D
88	arrangement and crystal chemistry. Fe atoms can interact with neighboring Fe atoms in the
89	octahedral sheet, whereas interactions out of the sheet are commonly with non-magnetic cations.
90	Townsend and Longworth (1985) suggested a positive super-exchange (ferromagnetic)
91	interaction arising from the interaction of neighboring Fe^{3+} Fe^{3+} ions, thus in contrast to previous

92 studies that claimed these cations to be antiferromagnetically coupled (Coey et al. 1981; Ballet and

- 93 Coey 1982; Ballet et al. 1985).
- 94 Several studies tried to explain the occurrence of these different experimental results (Coey
- 95 1985, 1987; Ballet 1986; Townsend 1987; Townsend et al. 1987; Longworth et al. 1987; Marcelli et
- al. 2004), without however, reaching a final conclusion. Some authors (Christie et al. 1992;
- 97 Rancourt et al. 1994) observed an antiferromagnetic trend in the susceptibility-temperature curve
- 98 for annite crystals, and a magnetic ordering at temperature greater than 10K.
- 99 In this work, six Fe-bearing mica samples, significantly different in Fe composition and Fe
- 100 coordination, are selected to relate their magnetic behavior to their chemical composition and
- 101 cation ordering. All the samples studied belong to 1M polytype, but differ in their tetrahedral and
- 102 octahedral composition, $Fe^{2+}/(Fe^{2+}+Fe^{3+})$ ratio, and layer symmetry. Four samples are from the
- 103 phlogopite–annite join(space group C2/m), one is a tetra-ferriphlogopite (space group C2/m) and
- 104 one is a Li-rich siderophyllite (space group *C*2).
- 105
- 106

Samples and Experimental Methods

107 Samples

108 Sample Albano is a ferroan phlogopite from "Peperino Albano" ignimbrite (Albano Laziale,

109 Italy), originating from the top of the eruption cycle of holocrystalline ejecta of Albano Maar lake.

110 The Albano Maar is a multiple tuff ring consisting of five main explosive cycles (Brigatti et al.

111 2005). Holocrystalline ejecta show granular hypidiomorphic to porphyric structures, with

112 clinopyroxene, brownish mica, and leucite as main mineral phases (Laurora et al. 2009).

113 Samples Oquossoc and Bryant Pond are^[vi]Al-rich magnesian annites from the peraluminous-

- 114 granite zone, located in Oquossoc and Bryant Pond areas (northwest part of the State of Maine,
- 115 USA), respectively. The mineral assemblage ensures that the mica composition is Al, Si and Ti

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- 116 saturated. Petrological and mineralogical data of this area and of micas occurring therein can be
- 117 found in Guidotti et al. (1988).
- 118 Samples Tapira and Catalão are from Tapira and Catalão alkaline-carbonatite complexes,
- 119 respectively. These complexes are located into the Alto Paranaíba Igneous Province (APIP) that
- 120 occupies a northwest-elongated area between the São Francisco Craton and the northeast border of
- 121 the Paraná Basin (Brazil). Sample Tapira is from a Bebedeurite rock mainly composed of Fe^{3+} -rich
- 122 phlogopite, variable amounts of clinopyroxene (> 20%), and subordinate perovskite, apatite, calcite,
- 123 ilmenite, and titaniferous magnetite (Brigatti et al. 1996). Sample Catalão is from a clinopyroxenite
- 124 rock that contains clinopyroxene, variable amounts of Fe^{3+} -rich phlogopite, and minor amounts of
- allotriomorphic carbonate and amphibole (Brod et al. 2001).
- 126 Sample Eurajoki is a lithian siderophyllite from the Väkkärä granite outcropping in the central
- 127 part of the polyphasic Eurajoki stock, a rapakivi-granite complex in southwestern Finland. Lithian
- 128 siderophyllite is associated with topaz, microcline, albite and minor monazite, Nb-, Ta-rich
- 129 cassiterite, bastnaesite, xenotime, columbitethorite (Haapala 1977).
- 130

131 Chemical composition and calculation of the chemical formula

132 Chemical composition (Table 1) was determined on crystals used for single crystal experiments

133 with a wavelength dispersive ARL-SEMQ electron microprobe (operating conditions: 15 kV

accelerating voltage; 15 nA sample current; defocused electron beam with a spot of about 4 µm;

135 counting times of 10 s for both peak and background). Analyses and data processing were

- 136 performed using the Probe software package of Donovan (1995). The following standards were
- 137 used: microcline (K, Al), albite (Na), spessartine (Mn), ilmenite (Fe, Ti), clinopyroxene (Si, Ca),
- 138 olivine (Mg), chromite (Cr), fluorite (F), barite (Ba), sphalerite (Zn), and synthetic Rb-rich glass
- 139 (Rb). The F content was determined via the method reported by Foley (1989). No evidence of
- 140 volatilization of F was observed. Several spots (up to 6 spots on each fragment) were analyzed to

141 verify sample homogeneity and to obtain a statistically significant average. Rim-core 142 compositional variation was observed to be within 3% of the estimated standard error for each 143 element (3% relative), indicating a high degree of chemical homogeneity. Individual spot analyses 144 for each crystal fragment were averaged. FeO determination followed the procedure by Meyrowitz 145 (1970) with an estimated measured standard deviation of $\sigma < 4\%$. 146 The Li content was determined via laser ablation inductively coupled to plasma mass 147 spectroscopy (LA-ICP-MS, UP 213, New Wave Research), where the ablated material was 148 transported to the quadrupole ICP-MS XSeriesII (Thermo Fischer Scientific). Three line scans 149 traced along the entire crystal and taken at three points over a 60 μ m spot at a scan speed of 25 μ m/s were performed. The dwell time was set at 20 s. The ablation frequency was 10 Hz in all scans. 150 151 The ablated aerosol was atomized at 0.68 bar into the plasma of the ICP-MS unit. NIST 612 (a 152 synthetic glass trace-element reference standard) was selected as standard reference used to create the calibration curve. ²⁹Si was selected as the internal standard. The data obtained were then 153 154 normalized to the sample SiO_2 content, as measured via electron microprobe analysis. The (OH)⁻ 155 content was derived from thermo-gravimetric analysis in He gas flow using a Seiko SSC 5200 156 thermal analyzer (heating rate 10 °C/min and gas flow rate 100 μ l/min) equipped with a mass 157 spectrometer (GeneSys ESS, Quadstar 422). 158 The chemical formula was calculated by an iterative approach. As a first step, the charges of 159 cations in tetrahedral, octahedral and interlayer sites were set equal to 22. The determined 160 composition of the anion site, populated by OH, F and O, was a result of this step. Given X = 2-161 (OH+F), the chemical formula was thus calculated assuming 22+X positive charges for tetrahedral,

162 interlayer and octahedral sites, which leads to an assessment of anion composition and therefore to a

163 value for X. The iterative procedure was used until the difference in the X value from each step,

164 was $< 10^{-7}$ in magnitude.

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166 Single crystal X-ray diffraction

167 Small crystal fragments of each sample, optically homogeneous and inclusion-free, were hand-168 picked from the bulk sample and then analyzed using a Bruker X8 APEX four circle diffractometer 169 combined with an APEX 4K CCD detector, flat graphite monochromator and Mo K α -radiation (λ = 170 0.7107 Å) from a fine focus sealed tube. Data were processed using SAINT Plus (Bruker 2003). 171 The SADABS package (Sheldrick 1996) was used to perform a semi-empirical absorption 172 correction. Space group determination and data merging were performed using XPREP (Sheldrick, 173 1997). The crystal structure was refined by the SHELX-97 package of programs (Sheldrick, 1997). 174 Table 2 reports refined cell-parameters and other crystal data. Tables 3a and 3b list selected 175 interatomic distances and parameters obtained from the structure refinement for crystals showing 176 $C_{2/m}$ and C_{2} symmetry, respectively. The mean electron counts for the octahedral and interlayer 177 sites, determined following the crystal structure refinement and electron microprobe analyses, are 178 listed in Table 1. Atom coordinates, equivalent and anisotropic displacement factors are reported in Table 4^1 . 179

180

181 X-ray Absorption Spectroscopy (XAS)

182 Fe K-edge XAS spectra were collected at the ESRF European Synchrotron Radiation Facilities

183 (ESRF, Grenoble, France) at the beamline BM-25 (Spanish Beamline, SpLine, Branch A). The

184 storage ring conditions were 6 GeV working in the current range 180-200 mA. A Si(111) double-

- 185 crystal monochromator was used, providing an energy resolution of ~ 0.4 eV at the Fe K-edge,
- 186 which is much less than the Fe K natural line width (~ 1.15 eV; Krause and Oliver 1979). A
- 187 metallic Fe foil was used as reference for energy calibration (energy reproducibility: ± 0.05 eV).

¹For a copy of Table 4, deposit item AM-xx-yyy, contact the Business Office of the Mineralogical Society of America (see inside front cover of recent issue for price information). Deposit items may also be available on the American Mineralogist web at http://www.minsocam.org.

188	Data were collected on powder-pressed disks, made of an appropriate mixture of sample and
189	cellulose, in fluorescence mode at room conditions. Spectra were recorded over a range of 500 eV
190	across the Fe absorption <i>K</i> -edge with a 1 eV energy step in the edge region (7100-7150 eV) and a 3
191	eV energy steps in the Extended X-ray Absorption Fine Structure (EXAFS) region (7150-7500 eV).
192	Data from phlogopite, containing only a small amount of Fe ²⁺ in the M1 site, annite characterized
193	by a full occupancy of Fe^{2+} in all the octahedral sites, and tetra-ferriphlogopite characterized by
194	Fe ³⁺ for Si tetrahedral substitution were also collected as additional standards, and the results
195	compared to those of Giuli et al. (2001) and Tombolini et al. (2002). Standards analyzed were
196	chosen because of their similarity in topology to the samples under study and because of the
197	different iron oxidation and distribution within the layer.
198	
199	Site population
200	The site preferences and site populations were assigned on the basis of: 1) the refined site-scattering
201	values (Table 1); 2) the unit formulae derived from the chemical compositions (Table 1); 3) the
202	observed bond-lengths (Table 3a, 3b); 4) the constraint to all sites, except M1, to be fully occupied;
203	5) the ^[vi] Fe ³⁺ and ^[vi] Ti ⁴⁺ occupancies constrained to the M1 and M2 site, respectively, following

204 Cesare et al. (2003); and 6) the electronic neutrality of the resulting formulae. The results are

205 reported in Table 5.

206

207 Magnetic measurements

208 Measurements of magnetic susceptibility were made on the six powder samples considered in a

209 temperature range of 2.2 - 300 K. Experiments were performed using a Quantum Design PPMS-7T

- 210 cryomagnetic system. Magnetic measurements were performed using a disk (about 10 mm in
- diameter and 0.5 mm thick) prepared by a die-press of 36 mg of powdered mica crystals.

212	AC susceptibility (χ) was performed in the magnetic field of 10 Oe in field cooling mode. For
213	the Tapira sample, AC susceptibility was performed also in the magnetic field of 5000 Oe, both in
214	zero field cooling and field cooled modes. Isothermal magnetization curves and thermal scans in
215	constant field were measured, and the temperature dependence of the susceptibility deduced.
216	Data were obtained with the magnetic field applied along the plane of the sheets and thus
217	perpendicularly to c*. No direction was specified within the plane since the samples were generally
218	composed of several thin crystals with their planes approximately parallel to each other.
219	Diamagnetic contributions to the susceptibility from the sample holder and the atom cores were
220	approximately 0.4×10^{-6} emu. Demagnetization effects were negligible. Molar magnetic
221	susceptibility χ (χ = C/T) was expressed in emu mol ⁻¹ . In particular, C is the Curie constant
222	$\left(C = \frac{Ng^2 \mu_B^2 S(S+1)}{3k_B}\right)$, where: N = number of iron atoms per cm ³ ; μ_B = Bohr magneton; g =
223	gyromagnetic ratio; $S = \text{spin angular momentum } (S = 5/2 \text{ for Fe}^{3+} \text{ and } S = 2 \text{ for Fe}^{2+}); k_B =$
224	Boltzman constant. The number of iron atoms per cm ³ and the Curie constants are listed in Table 1.
225	
226	Results and Discussion
227	Crystal chemistry and XANES spectroscopy
228	Based on X-ray diffraction (XRD) and chemical data, iron is present in two different oxidation
229	states (Fe ²⁺ , Fe ³⁺) and coordination (iv and vi) in the 1 <i>M</i> trioctahedral micas under study. Four of
230	the samples (Albano, Oquossoc, Bryant Pond, and Tapira) with octahedral Fe ²⁺ content ranging
231	from 0.378 to 1.476 apfu (0.158> ^[vi] [Fe ²⁺ /(Fe ²⁺ +Mg)]<0.679) belong to the phlogopite-annite join,
232	two of these micas (Oquossoc and Bryant Pond) are Mg-rich annite, and two (Albano and Tapira)
233	are Fe-rich phlogopite (Rieder et al. 1998). Sample Catalão, with ^[iv] Fe ³⁺ content 0.898 apfu, is a

tetra-ferriphlogopite and sample Eurajoki is a Li-rich siderophyllite with octahedral Fe²⁺ and Li

236 $(^{[vi]}Fe^{3+}=0.279 \text{ apfu}) Fe^{3+}$ octahedral occupancy is also observed (Table 1).

- Figure 1 shows how samples are classified in terms of *mgli* vs. *feal* parameters, as defined by
- 238 Tischendorf et al. (1999), i.e. $mgli = (Mg Li); feal = (Fe_{total} + Mn + Ti [vi]Al)$. In crystals of the
- phlogopite-annite join, ^[vi]Al is zero in Tapira and 0.073 in Albano (Table 1), thus indicating that the
- siderophyllite end-member content is low. Mg-rich annites Oquossoc (^[vi]Al = 0.377) and Bryant
- Pond ($^{[vi]}Al = 0.485$) plot in the annite field near the siderophyllite boundary.
- 242 The main exchange mechanisms where Fe increases in layer content in addition to (i)
- 243 $^{[iv]}\text{Fe}^{3+[iv]}\text{Al}^{3+}$, which defines the phlogopite to tetra-ferriphlogopite join, include: (ii) $^{[vi]}\text{Fe}^{2+[vi]}\text{Mg}$.
- 244 1, which defines the phlogopite to annite join, and (iii) $^{[iv]}Si ^{[vi]}Li ^{[iv]}Al_1 ^{[vi]}Fe^{2+}$ which defines the
- polylithionite to siderophyllite join. The full heterovalent exchange vectors are: (i) $^{[vi]}Fe^{3+}_{2}$

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$$^{[vi]}Mg^{2+}_{-3}$$
, (ii) $^{[vi]}(Mg, Fe)^{2+}_{-1}$ $^{[iv]}Si^{4+}_{-1}$ $^{[vi]}Fe^{3+}[iv]Al^{3+}$, and (iii) $^{[vi]}Fe^{2+}_{-1}$ (OH) $_{-1}$ $^{[vi]}Fe^{3+}O_{-2}$. All

- trioctahedral micas under investigation are the 1Mpolytype. The crystal structure refinements
- indicate the space group C2/m for five samples (Albano, Oquossoc, Bryant Pond, Tapira, and
- 249 Catalão). Based on $(Fe^{2+}+Li)$ ordering in the *trans*-M1 site and in one of the two *cis*-sites (M3), the
- 250 layer symmetry is reduced to space group C2 in sample Eurajoki. Variability in chemical

251 composition, in particular when concerning the $Fe^{2+}/(Fe^{2+}+Fe^{3+})$ ratio, is reflected by several

structural parameters relating both to the tetrahedral (T) and octahedral (M) sheets.

253 The $^{[iv]}$ Fe³⁺for $^{[iv]}$ Si⁴⁺substitution notably affects tetrahedral features. Catalão tetra-

254 ferriphlogopite is characterized by the greatest tetrahedral volume ($T_{volume}=2.40 \text{ Å}^3$) and by the

- lowest tetrahedral flattening angle ($\tau = 109.94^{\circ}$), very close to the theoretical 109.47° value. These
- results indicate that ^[iv]Fe³⁺substitution produces a regular tetrahedral site with the tetrahedral cation
- 257 located at the center. The mean tetrahedral basal edge length is greater in tetra-ferriphlogopite ((O-
- 258 $O_{basal} = 2.725(3)$ Å) than in Fe-rich phlogopite ($\langle O-O_{basal} = 2.701(3)$ Å for Tapira and ($\langle O-O_$
- 259 2.700(3) Å for Albano) and Mg-rich annite samples ($\langle O-O \rangle_{basal} = 2.701(3)$ Å for Oquossoc and $\langle O-$

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- 260 $O_{basal} = 2.698(3)$ Å for Bryant Pond) (Table 3a). The mean tetrahedral basal edge is approximately
- related to the octahedral edge (i.e. average (O3-O3) distance) by the following relationship:

262
$$\cos \alpha = \frac{\sqrt{3}}{2} \times \frac{\langle O3 - O3 \rangle}{\langle O - O \rangle_{\text{basal}}}$$
 (Brigatti and Guggenheim 2002), where α is the tetrahedral rotation

- 263 angle. Both the $\langle O3-O3 \rangle$ octahedral mean edge distance and α increase following ^[iv]Fe³⁺ for ^[iv]Si
- substitution, thus indicating that $^{[iv]}Fe^{3+}$ also produces a significantly greater tetrahedral ring
- 265 distortion (α in tetra-ferriphlogopite = 10.59°). The tetrahedral ring distortion significantly affects
- 266 interlayer coordination and the separation between two adjacent layers. In particular, ECoN
- 267 (effective coordination number, Weiss et al. 1992) of tetra-ferriphlogopite is 8.87° and the
- 268 interlayer separation is 3.463Å, which represent the lowest and the greatest value, respectively in
- the considered dataset (Table 3a).
- 270 Octahedral mean bond distances (Table 3a) together with the scattering efficiencies (m.e.c.,
- Table 1) indicate disorder of octahedral cation for samples Tapira, Albano and Catalão with m.e.c.
- 272 M1 \approx m.e.c. M2 and mean bond distances (M1-O) \approx (M2-O). For increasing ^[vi]Fe_{tot}/^[vi](Fe_{tot}+Mg)
- 273 ratios, Mg-rich annites Oquossoc and Bryant Pond display meso-octahedral ordering with m.e.c.
- 274 M1> m.e.c. M2 and \langle M1-O \rangle > \langle M2-O \rangle .
- 275 The mean electron count determined for M1, M2, and M3 sites in the Eurajoki sample is M3 <
- 276 M1 < M2 (Table 1), whereas mean bond distances for two octahedral sites are similar (\langle M1-O \rangle \approx
- $\langle M3-O \rangle$) and greater than values observed for the remaining site ($\langle M2-O \rangle$) (Table 3b). Al
- 278 preferentially occupies the M2 site, whereas Fe and Li are disordered between M1 and M3 sites
- 279 with a slight preference of Fe for the M1 site.
- Figure 2 shows the XANES Fe *K*-edge spectra of all the trioctahedral micas considered.
- 281 Phlogopite shows Fe²⁺ cations highly diluted among Mg atoms and, most probably, located in the
- 282 M1 site only. Like phlogopite, tetra-ferriphlogopite shows a limited amount of Fe in octahedral

- 283 coordination and Fe^{3+} mostly in tetrahedral coordination. Annite has Fe^{2+} completely disordered
- 284 over the entire octahedral sheet in all octahedral sites.
- 285 The pre-edge feature of tetra-ferriphlogopite (feature A, Figure 2a), attributed to the Fe^{3+} in
- tetrahedral coordination (Dyar et al. 2001), is evident in sample Catalão at 7115.52 eV. Fe^{3+} is also
- 287 present, but less evident, in sample Tapira at 7116.72 eV. Feature A, is negligible in other samples,
- thus suggesting that Fe³⁺ for samples Albano, Oquossoc, and Eurajoki can be assigned to octahedral
- coordination only (Waychunas et al. 1983; Wilke et al. 2001).
- 290 The position of the threshold structure at 7127.2 eV in tetra-ferriphlogopite (feature B) is
- 291 consistent with Fe³⁺. This feature is also present in samples Catalão, Tapira, Albano, Oquossoc and
- 292 Eurajoki. As suggested by Tombolini et al. (2002) the absorption edge position appears to follow
- 293 the $Fe^{3+}/(Fe^{3+}+Fe^{2+})$ ratio almost independently from the total iron content (Figure 2b).
- 294 The Fe^{2+} distribution in the octahedral sites is derived from the comparison of C, D, and E
- 295 features in phlogopite and annite. The intensity ratio between C and D effects suggests that samples
- 296 Oquossoc and Bryant Pond have the highest Fe/(Fe+Mg) ratio, whereas samples Tapira and Albano
- 297 have the lowest. Feature E can be related to Fe^{2+} ordering in octahedral sites (Tombolini et al.
- 2002). This feature occurs at 7141.6 eV and at 7144.5 eV in phlogopite and annite, respectively.
- 299 Feature E is in the range between 7144.4 eV (sample Oquossoc) and 7145.3 eV (sample Eurajoki)
- 300 thus suggesting that Fe is distributed in all octahedral sites. For sample Catalão, the Fe *K*-edge
- 301 XANES spectrum is similar to the spectrum of the standard tetra-ferriphlogopite, and shows high
- 302 complexity of the edge region. This pattern consists of several feature up to 20 eV above the edge
- 303 inflection at 7131.57 eV, and by a less well defined series of broad multiple scattering contributions
- in the XANES region. This behavior is related to Fe structural disorder (i.e. Fe^{2+} and Fe^{3+} in the 2:1
- 305 layer in both tetrahedral and octahedral coordination), and results in a great number of Fe-O
- 306 distances to produce a XANES spectrum with broad features owing to the overlapping of multiple
- 307 scattering contributions (Tombolini et al. 2002).

308 Magnetic behavior

309 Fe is the main magnetic element in all samples under study (Table 1) and where present in 310 octahedral coordination, the magnetic interaction is super-exchange. This interaction occurs if Fe 311 atoms occupy two neighboring octahedral sites, and its sign depends on the angle defined by the 312 bonds of the two Fe cations with their shared oxygen atoms: weakly positive super-exchange 313 interaction is observed if the angle is near 90°, whereas negative super-exchange interaction is 314 observed if the angle is near 180° (Coey 1988; Coey and Ghose 1988). In samples under study, 315 relevant bond angles range from 91.5° to 121.4°. The lowest angle is observed in the Eurajoki Lirich siderophyllite, with a significant Fe^{2+} content present in two octahedral sites (i.e. M1 and M3, 316 with some Fe³⁺in M2). The highest value is observed in the Bryant Pond Mg-rich annite, with a 317 homogeneous Fe^{2+} distribution in the three octahedral sites. Angles defined by apical oxygen atoms 318 319 with adjacent tetrahedral and octahedral cations are close to or >120°. Similarly, the angle defined 320 by adjacent tetrahedral cation with their shared basal oxygen atom is near 120°. 321 Bryant Pond and Oquossoc Mg-rich annite samples show maxima in the low field magnetic

susceptibility (χ) at 5.5 and 6.3 K, respectively. Following literature results (Coey 1988; Rancourt et al. 1994; Pini et al. 2012), this result suggest an incipient antiferromagnetic ordering at low

temperature (Figures 3a, 3b). The Weiss θ constant, determined by extrapolation of the best fit

325 intercept of the $1/\chi$ curve with horizontal axis is 19 K and 13 K for Bryant Pond and Oquossoc

326 samples, respectively, thus suggesting a predominant ferromagnetic interaction (Figures 4a, 4b).

327 The positive value of the Weiss θ constant is consistent with the trends of χ ×T-vs-T curves because

328 they plot above the constant values calculated at $\theta = 0$ and define trends typical for $\theta > 0$ values.

329 The $\chi \times T$ product increases with decreasing temperature, down to the values where the magnetic

330 ordering occurs (Figures 5a, 5b).

These results agree with the model of Coey et al. (1981) who studied the magnetic structure of phyllosilicates. The model suggests that Fe atoms of the octahedral sheet are strongly coupled within the plane by super-exchange ferromagnetic interactions. The magnetic interactions arising from the plane with other octahedral sheets are weaker dipole-dipole interactions with dominant antiferromagnetic character.

336 Figure 6 displays real and imaginary component of susceptibility as a function of temperature 337 for different frequencies for Bryant Pond sample. The dependency of the temperature of the peak 338 from the frequency of the applied magnetic field is consistent with a spin glass behavior. Further 339 evidence of spin grass is derived from the imaginary component trend that shows the inflection 340 point, corresponding to freezing temperature, moving to lower temperature values when decreasing 341 frequency of the applied magnetic field (Petracic et al. 2006). Spin glass involves short range 342 magnetic ordering below a given freezing temperature, where spins are frozen at random 343 orientations. The resulting system is different from a disordered pattern, observed at higher 344 temperature values, where the material shows paramagnetic behavior and clearly shows a 345 cooperative phenomenon.

Figure 7 reports the magnetization curves as a function of an increasing magnetic field H at a constant temperature of 2.2K. For both Mg-rich annites, experimental values plot below their theoretical Brillouin curves, thus suggesting that saturation is not reached under the effect of significantly strong magnetic fields (7 x 10^4 Oe).

350 The low-field susceptibility of the Albano (Figures 3c, 4c, 5c) and Tapira (Figures 3d, 5d) Fe

351 rich-phlogopite samples were characterized at magnetic fields of 10 Oe by cooling from room

352 temperature to 2 K. High-field susceptibility was measured for sample Tapira, also at the

- 353 magnetic field of 50000e both in field cooling and field warming mode (Figure 8). Trends
- 354 obtained at H=5000 Oe for sample Tapira do not shows appreciable differences in behavior between

355	the two Fe-rich phlogopites (both at H=5000) and between sample Albano at H =10 and H=5000
356	Oe. The Weiss constant θ values are 4.4K and 5.2K for Albano and Tapira, respectively. These
357	values are lower than those previously calculated for Mg-rich annites. Different behavior is
358	observed between the two samples at a magnetic field of 10 Oe. Sample Tapira shows a peak at a T
359	≅120K (Figure 3d), i.e. a temperature relatively high if compared to magnetic ordering temperatures
360	commonly observed for silicates (<100K). An effect at a much lower temperature (i.e. 2.4 K,
361	Figure 3d) is also identified. This latter effect, unlike the one observed at 120K, can be observed if
362	increasing magnetic intensity to ≤5000 Oe. The lack of the effect at higher temperature, when
363	increasing magnetic intensity, is related to small Fe oxide inclusions in the Tapira sample, as
364	demonstrated by microprobe analysis and by calorimetric analysis.
365	Sample Tapira shows in the temperature range from 100 to 140K, a regular decrease in heat
366	capacity without displaying the variation in slope commonly observed for this and for other samples
367	at lower temperatures, in correspondence with their peak in magnetic susceptibility. This behavior
368	is consistent with a magnetic transition affecting a small portion of the crystal that does not affect
369	thermal capacity. Fe oxides, such as magnetite, undergoes a well-defined magnetic transition at T
370	≅120 K (Verwey magnetic transition, Gridin et al. 1996; Tarnawsky et al. 2004; Tabiśet al. 2007).
371	This mechanism is consistent with the oxidation process described by Rancourt et al. (2001) and
372	Dunlop et al. (2006) for micas originating at higher oxygen fugacity values. Thus the mechanism is
373	consistent with the petrogenetic processes accepted for the Tapira complex.
374	In tetra-ferriphlogopite (Catalão), χ variation versus temperature defines the trend commonly
375	observed for paramagnetic materials (Figure 3e). The curve $\chi \times T$ plots below the theoretical value
376	for the sample composition, at $\theta=0$, thus suggesting that the magnetic interaction among Fe atoms is
377	mostly anti-ferromagnetic (Figure 5e). This result also matches the trend of the $1/\chi$ curve (Figure
378	4d) that defines a negative value for Weiss θ constant (\approx -25 K). This behavior is consistent with

379	super-exchange that describes Fe ³⁺ -O-Fe ³⁺ as negative (i.e. antiferromagnetic) and with literature
380	data (e.g., Coey, 1981). The magnetization curve for the Catalão sample shows a lower increase in
381	magnetization at increasing magnetic field, when compared to other samples and remarkably lower
382	values at high values of an applied magnetic field (Figure 7). This behavior is related to the
383	interaction of Fe cations in octahedral coordination, only in M1, with Fe cations in tetrahedral
384	coordination, and to interactions among tetrahedral Fe cations. The different behavior of this
385	sample, is related to the different oxidation state of the Fe atoms involved in the interaction (i.e.
386	Fe^{3+} vs. Fe^{2+}) and to their coordination (tetrahedral and octahedral). The bond angles defined by
387	two adjacent Fe cations with their shared octahedral oxygen atoms is near 120°. Similarly two
388	adjacent tetrahedral cations define with their shared basal oxygen atoms angles near 120° . The
389	interaction inside the tetrahedral sheet is however extremely unlikely to occur, if considering charge
390	balance of the bridging basal oxygen atoms. In contrast, the interaction among tetrahedral and
391	octahedral sites are confined to the atoms inside the layer, because two adjacent layers are separated
392	by the non-magnetic interlayer cations.

Li-rich siderophyllite Eurajoki sample shows a peak in χ at higher temperatures than observed for other samples (i.e. 7.6 K), possibly as a consequence of uniform Fe distribution among the octahedral sites with Fe²⁺ mostly located in M1 and M3 sites and Fe³⁺ in M2 site (Figure 3f). This sample shows magnetic interactions that are predominantly ferromagnetic within the layers. The inverse of χ approximates the Curie law and defines a θ constant of 17K (Figure 4e). Similarly to other samples, the χ ×T curve plots above the theoretical value characterizing θ =0 for sample composition (Figure 5f).

400

401 **Concluding remarks and implications.**

402 Reported data detail magnetic properties for micas with different Fe content, different Fe
 403 coordination, and different Fe oxidation state. Total Fe content is not able to account for the 17

404 observed magnetic properties of the studied micas. Some samples (e.g., Oquossoc, Bryant Pond) 405 show for T < 10 K similar magnetic effects: i.e. maxima in magnetic susceptibility and similar 406 variation in $\gamma \times T$ -vs-T. In contrast, the Albano sample does not show any maximum in magnetic 407 susceptibility, despite intermediate Fe content, at low temperature, which is the trend commonly 408 observed for paramagnetic materials. Different magnetic behavior is thus better explained by 409 considering the Fe crystal chemistry, such as distribution of Fe among octahedral and tetrahedral sites and the $Fe^{2+}/(Fe^{2+}+Fe^{3+})$ ratio. Where Fe is mostly present in octahedral coordination, 410 411 dominant magnetic interactions among Fe atoms are ferromagnetic, which becomes stronger at high $Fe^{2+}/(Fe^{2+}+Fe^{3+})$ ratio. Samples with high $Fe^{2+}/(Fe^{2+}+Fe^{3+})$ ratio are also characterized by high 412 values of Curie-Weiss θ (Bryant Pond $\theta = 19$ K and Oquossoc $\theta = 13$ K). Where Fe²⁺/(Fe³⁺+ Fe²⁺) 413 ratios decrease, the θ value also decreases. For sample Catalão showing predominant Fe³⁺ in 414 415 tetrahedral coordination, the sample shows predominant anti-ferromagnetic interactions with a 416 negative value of the Curie-Weiss θ constant (i.e. θ = - 25 K).

417 The data for micas suggest a spin-glass behavior, as indicated by the dependency of the 418 maxima in magnetic susceptibility from the frequency of the applied magnetic field. This result 419 differs from the long range magnetic ordering at low T reported by Coey et al. (1981), Beausoleil et al. (1983) and Rancourt et al. (1994). This difference is related to Fe^{3+} content that, as suggested by 420 421 Ballet et al. (1985), can locally randomize the ferromagnetic ordering inside the plane, and to non-422 ferromagnetic elements. This result is consistent with the lower Fe content, which is characterized 423 by the studied samples, when compared to those reported in the literature. 424 Figure 9 relates magnetic ordering, either antiferromagnetic planar long range order or spin

glass short range order, as a function of p33 (i.e. the probability that two Fe^{3+} cations are placed in adjacent octahedra) and the total number of octahedral Fe cations per unit formula (Z). This plot demonstrates that spin glass behavior is favored at low Z and high p33 values.

5/6

428	Both θ and Tm (i.e, the maximum susceptibility temperature) seem to linearly depend on the
429	number of Fe atoms in octahedral coordination per unit cell, Z (Figure 10, Table 6). Greater θ
430	values are however observed in a sample showing intermediate Z, thus suggesting an influence of a
431	lower than ideal symmetry and to the presence of Fe^{2+} ordering on M1 and M3 sites. This
432	distribution increases the probability of ferromagnetic interaction among two adjacent Fe ²⁺ atoms.
433	
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588	List of figure captions
589	Figure 1. Micas on the classification diagram after Tischendorf et al. (1999). mgli = (Mg - Li); feal
590	= (Fe _{Tot} + Mn + Ti - $^{[vi]}$ Al). Symbols: circle = Bryant Pond; triangle down = Oquossoc;
591	hexagon = Tapira; diamond = Albano; triangle up: Catalão; square = Eurajoki.
592	Figure2.a) Fe K-edge XANES spectra of studied micas compared with phlogopite, tetra-
593	ferriphlogopite and annite. b) Comparison between the energy position of the threshold
594	structure of micas containing both Fe^{2+} and Fe^{3+} mainly in octahedral position.
595	Figure 3. Trend of the real (in-phase) component of the magnetic susceptibility (χ) vs. temperature,
596	measured at $H = 10$ Oe in Field Cooling. a) Bryant Pond; b) Oquossoc; c) Albano; d) Tapira;
597	e) Catalão; f) Eurajoki.
598	Figure 4. Inverse of the real component of the magnetic susceptibility $(1/\chi)$ vs. temperature
599	measured at $H = 10$ Oe in Field Cooling. The solid line represents the linear fit of the curve in
600	the region of temperature between 80 and 200 K, together with r^2 values. a) Bryant Pond; b)
601	Oquossoc; c) Albano; d) Catalão; e) Eurajoki.
602	Figure 5. Temperature dependence of the product ($\chi \times T$). The solid line represents the theoretical
603	product $\chi \times T \approx \frac{1}{8}g^2 S(S+1)$ where g is the gyromagnetic ratio and S the spin angular momentum
604	$(S = 5/2 \text{ for Fe}^{3+} \text{ and } S = 2 \text{ for Fe}^{2+})$. a) Bryant Pond; b) Oquossoc; c) Albano; d) Tapira; e)
605	Catalão; f) Eurajoki.
606	Figure 6. Real χ (a) and the imaginary part χ' (b) of the AC susceptibility as a function of
607	temperature and frequency.
608	Figure 7. Sample magnetization vs. magnetic field (H). Symbols: filled circles = Bryant Pond;
609	filled diamonds = Oquossoc; filled triangles up = Albano; filled squares = Catalão; filled
610	triangles down = Eurajoki; open triangles = Brillouin curve for $Fe^{2+}(J=2)$; open squares =
611	Brillouin curve for Fe ³⁺ (J=5/2). The Brillouin curve was defined as $M = M_S B_J(y)$ where M_S

612 represent the magnetization at the saturation point (referred to one mole) and calculated as

613
$$M_s = N_A g \mu_B J$$
. N_A = Avogadro constant, g = gyromagnetic ratio, μ_B = Bohr magneton; J

614 = magnetic total angular momentum; $B_J(y)$ (i.e. the Brillouin function) =

615
$$B_J(y) = \frac{2J+1}{2J} \coth(\frac{2J+1}{2J}y) - \frac{1}{2J} \coth\frac{y}{2J} \text{ where } y = g\mu_B J \frac{B}{k_B T}.$$

616 Figure 8. Sample Tapira: a) In-phase susceptibility as a function of temperature with an applied

- 617 filed of 5000 Oe. Open symbol show susceptibility when cooling from room temperature
- and crosses when warming from low temperature. b) $\chi \times T$ vs. temperature; the solid line
- 619 represents the theoretical product (for details see the footnote of Figure 5; c) Inverse of the in-
- 620 phase magnetic susceptibility $(1/\chi)$ vs. temperature; the solid line represents the linear fit of
- the curve in the region of temperature between 80 and 200K.
- 622 Figure 9. Magnetic behavior (spin-glass vs. long range magnetic ordering), as a function of Z
- 623 (octahedral Fe content) and *p33* representing the probability that two adjacent octahedral sites
- 624 are occupied by Fe³⁺ cations. Samples plotting to the left of the dashed line all present spin-
- 625 glass behavior, unlike samples plotting to the right that present long range magnetic ordering.
- 626 Filled symbols: this study (circle = Tapira, triangle up =Albano; triangle down = Oquossoc;
- 627 square: Bryant Pond; diamond = Catalão; hexagon = Eurajoki). Open symbols: samples from
- 628 literature (Ballet et al. 1985, Ballet and Coey 1982, Rancourt et al. 1994).
- 629 Figure 10. a) Curie-Weiss constant vs. Z (Fe cations in octahedral coordination); b) temperature of
- 630 maximum in magnetic susceptibility vs. the tetrahedral rotation angle α . Filled symbols: this
- 631 study (circle = Tapira, triangle up =Albano; triangle down = Oquossoc; square: Bryant Pond;
- 632 diamond = Catalão; hexagon = Eurajoki). Open symbols: samples from literature (Ballet et
- 633 al. 1985, Ballet and Coey 1982, Rancourt et al. 1994).



Figure 1



Figure 2a



Figure 2b

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Figure 6

Figure 7

Figure 8

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Figure 9

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Sample	Tapira	Albano	Oquossoc	Bryant Pond	Catalão	Eurajoki
a.o		Chen	nical compos	ition (oxide w	vt%)	
SiO ₂	38.30	36.40	36.96	34.22	41.52	42.79
T_1O_2	2.32	2.10	2.11	2.64	0.23	0.12
Al_2O_3	13.42	16.40	16.81	20.00	0.00	18.71
Fe_2O_3	6.30	5.07	3.05	0.00	17.20	2.28
Cr_2O_3		0.01				
FeO	6.60	6.18	19.80	22.75	2.48	16.73
MnO	0.18	0.20	1.01	0.64	0.03	0.15
MgO	18.15	18.50	6.55	6.03	24.03	0.20
BaO		0.25	0.06			2.52
Li_2O	0.20	0.20	0.02	0.00	0.02	3.53
Na_2O	0.20	0.30	0.02	0.09	0.03	0.18
$K_2 O$	10.50	10.25	10.12	9.98	10.40	10.40
Kb ₂ O						0.12
F	0.55	2.42	0.03		1.10	5.96
Cl	2 70	2.05	2.50	0.22	2.45	1.05
H ₂ O	3.70	2.95	3.50	3.50	3.45	1.25
F,Cl=O	-0.23	-1.02	-0.01	-0.05	-0.46	-2.51
Sum	99.99	100.01	100.01	100.02	100.01	99.91
			Init-cell cont	tent (2 n f u)		
[iv] _{Si}	2 807	2 660	2 850	2 656	3 102	3 1/1
^[iv] A 1	2.807	2.000	2.850	2.050	5.102	0.850
$[iv]_{\mathbf{Fe}^{3+}}$	0.03/	1.540	1.150	1.344	0 808	0.859
	4 000	4 000	4 000	4 000	4 000	4 000
L	4.000	4.000	4.000	4.000	4.000	4.000
^[vi] Al		0.073	0.377	0.485		0.760
^[vi] Mg	1.982	2.015	0.753	0.697	2.676	0.022
^[vi] Fe ³⁺	0.313	0.279	0.177		0.069	0.126
$^{[vi]}$ Fe ²⁺	0.404	0.378	1.277	1.476	0.155	1.027
^[vi] Cr		0.001				
^[vi] Mn	0.011	0.012	0.066	0.042	0.002	0.009
^[vi] Ti	0.128	0.115	0.122	0.154	0.013	0.007
^[vi] Li						1.042
$^{[vi]}\Sigma$	2.838	2.873	2.772	2.855	2.915	2.993
[xii]Na	0.028	0.043	0.003	0.013	0.004	0.026
[xii] _K	0.028	0.045	0.003	0.015	0.004	0.020
^[xii] Bb	0.981	0.950	0.995	0.988	0.991	0.974
		0.007	0.002			0.000
[xii]	1 009	1.005	1 000	1 001	0 995	1 005
	1.007	1.005	1.000	1.001	0.775	1.005
F	0.127	0.559	0.007		0.260	1.384
Cl				0.029		
OH	1.809	1.438	1.800	1.812	1.719	0.612
0	10.064	10.003	10.193	10.159	10.021	10.004
Σ	12.000	12.000	12.000	12.000	12.000	12.000
		Number	firon atoms ((N) per cm^3	ofminoral	
$N = 10^{21} (molos \text{ cm}^{-1})$	1.50			(N) per chi o		2 20
W X TO (moles cm)	1.50	1.32	2.92	2.90	2.22	2.39
		Curie con	stant (C) ext	pressed in cos	system	
$C \times 10^{-3}$ (emuK/gOe)	6 1 3 9	5 497	14 790	14 701	7 458	11 300
CXIO (Cillux/gOC)	0.139	5.497	14.790	14./91	7.450	11.390
		Octahedra	al sites mean	electron coun	t (m.e.c.)	
$M1_{Xref}$	15.4	14.8	19.3	19.8	12.1	14.5
$M2_{Xref}$	15.1	15.1	18.3	18.7	12.6	15.9
$M3_{Xref}$	15.1	15.1	18.3	18.7	12.6	13.0
M1+M2+M3 $(Xref)$	45.6	45.0	55.9	57.2	37.3	43.4
M1+M2+M3 $_{(Epma)}$	45.5	45.1	56.0	57.5	37.6	43.6
		T , 1	•,	1 .		
٨	10.0	Interlaye	$\frac{1800}{1800}$	lectron count	(m.e.c.)	10.0
A (Xref)	19.0	19.0	10.9	19.0	18.9	19.0
A (Epma)	19.0	19.0	19.0	18.9	18.9	19.0

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	Ц	4	4	4	4-	Ц	4	4.	4	4	4	4	4	1	L.,	4	4	4	1	4-	4	4	4	11	4	4.	Ц	4	4.	4	4	4	4	4	4	4	44	4	4	44	4	+	4-	4	4	4	4	14	4	4	4	4.	11	4	4	11	+	44	4.	4	4	4	4	4	4	4	4-	4	4	4.	4	4	Ļ	
1	H	-+	$^{+}$	H	+ŧ	Ŧ	+	t	H	-+	+	t	t	Н	-1	+	+	t	Н	++	+	+	t_1	H	+	t 1	H	·+	+:	H	-†	+	H	-+	+	H	-+	+	H	-+	+	Ħ	++	H	•+	+	H	-+	+	H	-+	+1	H	-+	+:	H	Ħ	++	+	H	-+	+	H	-1	+	t	+1	+	$^{+}$	+1	4	Ħ	F	

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Table 2. Information related to X-ray data collection, structure refinement and unit cell parameters.

Sample	N _{total}	N_{unique}	R_{int} (%)	$\theta_{(min)}$	$\theta_{(max)}$	Miller index limits	<i>a</i> (Å)	$b(\text{\AA})$	<i>c</i> (Å)	β(°)	$V(Å^3)$	R(%)
Tapira	2033	830	2.53	2.01	32.14	$-7 \le h \le 6, -13 \le k \le 13, -15 \le l \le 15$	5.3366(4)	9.2421(6)	10.2754(6)	100.012(4)	499.08(6)	3.68
Albano	1999	573	1.78	4.04	27.08	$-6 \le h \le 6, -9 \le k \le 11, -10 \le l \le 13$	5.3303(3)	9.2281(4)	10.2342(5)	99.989(2)	495.77(4)	2.00
Oquossoc	2826	929	3.35	4.04	34.28	$-7 \le h \le 7, -13 \le k \le 12, -15 \le l \le 16$	5.3438(5)	9.2550(7)	10.2390(9)	100.222(4)	498.35(7)	3.78
Bryant Pon	3304	1133	2.06	4.05	35.30	$-8 \le h \le 8, -10 \le k \le 14, -16 \le l \le 16$	5.3473(2)	9.2609(3)	10.2382(3)	100.293(2)	498.84(3)	2.70
Catalão	2928	1107	3.43	4.01	37.91	$-9 \le h \le 8, -15 \le k \le 10, -16 \le l \le 17$	5.380(1)	9.241(2)	10.315(3)	100.08(2)	505.0(2)	3.75
Eurajoki	2094	1069	2.88	4.10	30.80	$-6 \le h \le 7, -13 \le k \le 10, -14 \le l \le 12$	5.3060(5)	9.1636(9)	10.116(1)	100.786(4)	483.17(8)	3.61

Notes: $R_{\text{int}} = \left(\sum_{n=1}^{\infty} |F_0^2 - F_0^2(mean)| / \sum_{n=1}^{\infty} |F_0^2| + |F_0| + |F_0| \right) = \frac{1}{2} |F_0| + \frac{1}{2}$

Table 3a. Relevant cation-anion bond lengths of refined mica crystals, space group C2/m, and selected geometrical parameters obtained from structure refinements.

Sample	Tapira	Albano	Oquossoc	Bryan Pond	Catalão			
Tetrahedron								
T-O1	1.661(2)	1.660(1)	1.660(2)	1.660(1)	1.679(2)			
T-O1'	1.663(2)	1.663(1)	1.660(2)	1.660(1)	1.674(2)			
T-O2	1.663(1)	1.662(1)	1.659(1)	1.660(1)	1.669(1)			
T-O3	1.664(2)	1.666(1)	1.679(2)	1.668(1)	1.672(2)			
$\langle T-O \rangle$	1.663	1.663	1.665	1.662	1.674			
	Tetrah	edral param	neers					
α (°)	8.95	9.26	8.38	7.78	10.59			
basal area (A ²)	18.34	18.28	18.41	18.45	18.43			
Δz (Å)	0.003	0.004	0.018	0.024	0.006			
Volume (A^3)	2.36	2.36	2.37	2.35	2.40			
τ (°)	110.26	110.26	110.03	110.2	109.94			
$\langle O-O \rangle_{basal}$	2.701	2.700	2.701	2.698	2.725			
	Octa	ahedron (M	1)					
M1-O3 (×4)	2.096(1)	2.090(1)	2.103(2)	2.1168(9)	2.102(1)			
M1-O4 (×2)	2.060(2)	2.050(1)	2.070(1)	2.086(1)	2.061(2)			
$\langle M1-O \rangle$	2.084	2.077	2.092	2.107	2.088			
	Octa	ahedron (M	2)					
M2-O3' (×2)	2.092(2)	2.088(1)	2.078(2)	2.076(1)	2.105(1)			
M2-O3 (×2)	2.087(2)	2.085(1)	2.072(2)	2.072(1)	2.090(1)			
M2-O4 (×2)	2.063(2)	2.040(1)	2.046(2)	2.044(1)	2.066(1)			
$\langle M2-O \rangle$	2.081	2.071	2.065	2.064	2.087			
		Octahedral	parameters					
M1-O4-M2 (°)	96.75(5)	97.41(5)	97.08(6)	96.75(7)	97.42(5)			
M1-O4-M2' (°)	96.75(6)	97.41(4)	97.08(6)	96.75(5)	97.42(6)			
M2-O4-M2 (°)	96.76(7)	97.18(5)	97.79(5)	98.05(6)	96.63(5)			
M1-O3-M2 (°)	109.85(6)	109.62(4)	109.67(5)	109.28(5)	109.96(7)			
M1-O3-M2' (°)	120.39(6)	120.91(5)	121.27(6)	121.42(6)	120.76(6)			
ψ (M1)°	58.76	59.08	59.47	59.53	58.85			
ψ (M2)°	58.70	58.98	59.04	58.84	58.83			
(03-03)	3.081	3.077	3.085	3.087	3.093			

	Inte	rlayer catio	n		
A-O1 (×4)	2.975(2)	2.956(1)	2.983(1)	2.997(1)	2.949(2)
A-O2 (× 2)	2.968(3)	2.958(2)	2.982(2)	2.997(2)	2.957(2)
A-O1' (×4)	3.378(2)	3.377(1)	3.358(1)	3.343(1)	3.431(2)
A-O2' (× 2)	3.383(2)	3.380(2)	3.379(2)	3.370(2)	3.446(2)
$\langle A-O \rangle_{inner}$	2.973	2.957	2.983	2.997	2.952
$\langle A-O \rangle_{outer}$	3.380	3.378	3.365	3.352	3.436
	Interla	yer parame	eters		
ECoN	9.38	9.27	9.58	9.70	8.87
A - O4 (Å)	4.018(3)	4.022(2)	4.015(2)	4.009(2)	4.045(2)
	She	et thicknes	S		
Tetrahedral (Å)	2.260	2.266	2.267	2.260	2.266
Octahedral (Å)	2.162	2.134	2.125	2.136	2.160
Interlayer separation (Å)	3.436	3.413	3.418	3.417	3.463

	Tet	rahedra		Interlayer cati	on	
T1-O1	1.653(8)	T2-O1	1.637(8)	A-O1 (×2)	3.054(3)	
T1-O2	1.640(6)	T2-O2	1.659(6)	A-O1' (×2)	3.244(3)	
T1-O22	1.627(6)	T2-O22	1.651(5)	A-O2 (×2)	3.024(6)	
T1-O3	1.642(6)	T2-O33	1.645(6)	A-O2' (×2)	3.263(7)	
(T1-O)	1.641	$\langle T2-O \rangle$	1.648	A-O22 (×2)	3.063(6)	
				A-022' (x2)	3.131(7)	
	Tetrahedr	al parameters		<a-o>_{inner}</a-o>	3.047	
α (°)	3.67	τ (°) (T1)	111.14	<a-o>_{outer}</a-o>	3.213	
basal area (A^2)	18.21	τ (°) (Τ2)	111.14			
V(T1) (A ³)	2.26	Δz (Å)	0.087	Iterlayer parameters		
V(T2) (A ³)	2.29	Sheet tickness (Å)	2.252	ECoN	11.66	
				A - O4 (Å)	3.967(3)	
				Interlayer separation	3.332	
		Octahe	edra			
Octahedro	n M1	Octahedron	M2	Octahedron M3		
M1-O3 (×2)	2.129(8)	M2-O3 (×2)	1.938(7)	M3-O3 (×2)	2.124(5)	
M1-O33 (×2)	2.146(7)	M2-O33 (×2)	1.923(5)	M3-O33 (×2)	2.116(7)	
M1-O4 (×2)	2134(4)	M2-O4 (×2)	1 872(5)	M3-O4 (×2)	2 157(5)	

Table 3b. Relevant cation-anion bond lengths of sample EP, space group C2, and selected geometrical parameters obtained from structure refinements.

Octahedro	on M1	Octahedron	M2	Octahedron M3		
M1-O3 (×2)	2.129(8)	M2-O3 (×2)	1.938(7)	M3-O3 (×2)	2.124(5)	
M1-O33 (×2)	2.146(7)	M2-O33 (×2)	1.923(5)	M3-O33 (×2)	2.116(7)	
M1-O4 (×2)	2.134(4)	M2-O4 (×2)	1.872(5)	M3-O4 (×2)	2.157(5)	
(M1-O) 2.136		(M2-O) 1.911		$\langle M3-O \rangle$	2.132	
		Octahedral p	arameters			
M1-O4-M3	91.50(6)	M1-O3-M3	92.57(5)	ψ (M1)°	60.53	
M1-O4-M2	99.93(4)	M1-O33-M2	97.85(6)	ψ (M2)°	56.63	
M2-O4-M3	100.44(7)	Sheet tickness (Å)	2.102	ψ (M3)°	60.47	

Sample	Tapira	Albano	Oquossoc	Bryant Pond	Catalão	Eurajoki
			M1 s	ite		
^[vi] Mg	0.455	0.580	0.046	0.169	0.828	0.004
^[vi] Fe ³⁺	0.314	0.278	0.176		0.069	
^[vi] Fe ²⁺	0.058	0.012	0.484	0.643	0.016	0.500
^[vi] Mn	0.011	0.012	0.066	0.042	0.002	
^[vi] Li						0.489
vacancies	0.162	0.118	0.228	0.146	0.085	0.007
[vi]	1.000	1.000	1.000	1.000	1.000	1.000
			M2 s	ite		
^[vi] Al		0.037	0.189	0.243		0.760
^[vi] Mg	0.763	0.718	0.354	0.264	0.924	0.013
^[vi] Fe ³⁺						0.126
[vi]Fe ²⁺	0.173	0.183	0.396	0.416	0.069	0.085
		0.005				
^[v1] Mn						0.009
	0.064	0.057	0.061	0.077	0.007	0.007
$[v_1]$	1.000	1.000	1.000	1.000	1.000	1.000
			M3 s	ite		
^[vi] Al		0.037	0.189	0.243		
^[vi] Mg	0.763	0.718	0.354	0.264	0.924	0.005
^[vi] Fe ³⁺						
^[vi] Fe ²⁺	0.173	0.183	0.396	0.416	0.069	0.442
^[vi] Cr		0.005				
^[vi] Mn						
^[vi] Ti	0.064	0.057	0.061	0.077	0.007	
^[vi] Li						0.553
$^{[vi]}\Sigma$	1.000	1.000	1.000	1.000	1.000	1.000

Table 5. Octahedral site population obtained by combining electron microprobe and spectroscopic data and taking into account the results of structure refinement.

	the litera	ture.						
Sample	S	layer type	Ζ	θ	Tm	θ/Tm		Reference
Chamosite	7.1	1:1	3	12	8	1.5	SG	Ballet et al. (1985)
Minnesotaite	9.6	2:1	4.5	38	19	2	AF	Ballet et al. (1985)
Biotite	10	2:1	5.3	43	7	6.1	SG	Ballet and Coey (1982)
Thuringite	7.1	2:1:1	3.9	17	5	3.4	SG	Ballet et al. (1985)
Annite	10	2:1	5.1	60	58	1	AF	Rancourt et al. (1994)
Tapira	10.24	2:1	1.43	5.2	2.5	2.1	SG	
Albano	10.24	2:1	1.31	4.4	2.3	1.9	SG	
Oquossoc	10.23	2:1	2.91	13	6.3	2.1	SG	
Bryant Pond	10.28	2:1	2.95	19	5.5	3.5	SG	
Catalão	10.32	2:1	0.44	-25	2.5	-10.0	PM	
Eurajoki	10.12	2:1	2.31	17	7.6	2.2	SG	

Table 5.Crystal chemical and magnetic parameters for studied samples and for some phyllosilicates reported in
the literature

S = distance between two magnetic sheets. Z = octahedral Fe atoms per unit cell; θ = Curie-Weiss constant; Tm = temperature of magnetic suscriptivity maximum; AF = planar antiferromagnetic behavior; SG = spin glass behavior; PM = paramagnetic behavior.