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# 1 Version 8

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3	Running title: New structure of high-pressure body-centered orthorhombic Fe <sub>2</sub> SiO <sub>4</sub>
4	
5	Takamitsu Yamanaka <sup>1</sup> Atsushi Kyono <sup>1,3</sup> Yuki Nakamoto <sup>1,4</sup> Svetlana Kharlamova <sup>1</sup> Viktor V Struzhkin <sup>1</sup>
6	Stenhen A. Gramsch <sup>1</sup> Ho-kwang Mao <sup>1,2</sup> and Russell I. Hemley <sup>1</sup>
7	Supren A. Oranisen, Tro-Kwang Wao and Russen J. Henney
8	Washington, D.C. 20015 USA
9	<sup>2</sup> High Pressure Collaborative Access Team, Geophysical Laboratory, Carnegie Institution of Washington,
10	Argonne, Illinois, 60439, USA
11	<sup>3</sup> Division of Earth Evolution Sciences, Graduate School of Life and Environment Sciences, University of
12	Tsukuba, Tsukuba Ibaraki 305-8572 Japan
13	<sup>4</sup> Center for Quantum Science and Technology Under Extreme Conditions, Osaka University,
14	Toyonaka Osaka, 560-8531 Japan
15	
16	Corresponding Author: Takamitsu Yamanaka
17	tyamanaka@ciw.edu
18	
19	Abstract
20	A structural change in Fe <sub>2</sub> SiO <sub>4</sub> spinel (ringwoodite) has been found by synchrotron powder diffraction
21	study and the structure of a new high-pressure phase was determined by Mote-Carlo simulation method and
22	Rietveld profile fitting of x-ray diffraction data up to 64 GPa at ambient temperature. A transition from the
23	cubic spinel structure to a body centered orthorhombic phase (I-Fe <sub>2</sub> SiO <sub>4</sub> ) with space group Imma and Z=4
24	was observed at approximately 34 GPa. The structure of I-Fe <sub>2</sub> SiO <sub>4</sub> has two crystallographically independent
25	FeO <sub>6</sub> octahedra. Iron resides in two different sites of six-fold coordination: Fe1 and are arranged in layers
26	parallel to (101) and (011) and are very similar to the layers of FeO <sub>6</sub> octahedra in the spinel structure. Silicon
27	is located in the six-fold coordination in I-Fe <sub>2</sub> SiO <sub>4</sub> . The transformation to the new high-pressure phase is

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28	reversible under decompression at ambient temperature. A martensitic transformation of each slab of the
29	spinel structure with translation vector $\langle \vec{1/8}   \vec{1/8}   \vec{1/8} \rangle$ generates the <i>I</i> -Fe <sub>2</sub> SiO <sub>4</sub> structure. Laser heating of
30	I-Fe <sub>2</sub> SiO <sub>4</sub> at 1500 K results in a decomposition of the material to rhombohedral FeO and SiO <sub>2</sub> stishovite.
31	Fe $K\beta$ x-ray emission measurements at high pressure up to 65GPa show that the transition from a high spin
32	(HS) to an intermediate spin (IS) state begins at 17 GPa in the spinel phase. The IS electron spin state is
33	gradually enhanced with pressure. The $Fe^{2+}$ ion at the octahedral site changes the ion radius under compression
34	at the low spin, which results in the changes of the lattice parameter and the deformation of the octahedra of the
35	spinel structure. The compression curve of the lattice parameter of the spinel is discontinuous at approximately
36	20 GPa. The spin transition induces an isostructural change.
37	
38	Keywords
39	New high-pressure structure, Fe <sub>2</sub> SiO <sub>4</sub> ringwoodite, x-ray emission spectra, spin transition, martensitic
40	transition.
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#### 44 Introduction

45 A great deal of attention has been paid to the high-pressure structural transitions of the many spinel phases 46 present in the Earth's crust due to their geophysical importance (Akimoto and Fujisawa, 1967; Basset and Ming, 47 1972; Ito and Takahashi, 1987; Irifune et al. 1998). One of the major minerals in the crust, (Mg,Fe)<sub>2</sub>SiO<sub>4</sub> olivine ( $\alpha$ -phase), transforms to wadslevite ( $\beta$ -phase, modified spinel) and further to ringwoodite ( $\gamma$ -phase, spinel). 48 49 These transitions were proposed for the origin of the seismic discontinuity of the transition zone from 410 km to 50 660 km depth (Ringwood and Irifune, 1988). These high-pressure transformations have been studied from 51 various viewpoints, including the electronic and elastic properties of participating phases (Kiefer et al. 1997; Li 52 et al. 2007) and continue to provide significant information for seismic interpretation (Leven et al. 1981; 53 Burnley et al. 1991; Shim et al. 2001).

54 Recently Fe<sub>2</sub>SiO<sub>4</sub> phase with spinel structure was found in meteorite and Silicate spinels of transition 55 elements are investigated for their influence on the magnetic and electrical properties of the Earth's crust and 56 mantle. Their phase stabilities and structures under high-pressure and high-temperature conditions have been This is a preprint, the final version is subject to change, of the American Mineralogist (MSA) Cite as Authors (Year) Title. American Mineralogist, in press.

(DOI will not work until issue is live.) DOI: http://dx.doi.org/10.2138/am-2015-4744 2/18 57 intensively studied. Phase relations in the Mg<sub>2</sub>SiO<sub>4</sub>-Fe<sub>2</sub>SiO<sub>4</sub> ringwoodite solid solution system have been 58 investigated in numerous high-pressure studies (Katsura et al., 1989; Ito and Takahashi. 1989; Fei et al., 1999; 59 Matsuzaka et al. 2000). Mechanisms for the polymorphic transformations of the  $\alpha$ ,  $\beta$  and  $\gamma$  phases of (Mg, have 60 been discussed in terms of the observation of dislocations by TEM (Price et al. 1982; Madon and Poirier 1983). 61 Structures of silicate spinels M<sub>2</sub>SiO<sub>4</sub> (M= Mg, Fe, Co, Ni) have been refined from x-ray diffraction data at 62 ambient conditions (Yagi et al. 1974; Morimoto et al. 1970, 1974; Ma, 1975; Marumo et al. 1974, 1977). (1981). End member of ringwoodite solid solution, Fe<sub>2</sub>SiO<sub>4</sub>, undergoes transitions at 6.4 GPa at 1700°C (Yagi 63 64 et al., 1987). Further experimental studies have been undertaken by Fei et al., (1991) and Matsuzaka a (2000). 65 Significant work is also focused on the physical properties of iron-bearing spinels in order to understand 66 their strong electronic correlations as manifested in charge transfer, electron hopping and the Jahn-Teller 67 effect, in particular through magnetic and electrical conductivity measurements and Raman spectroscopy l. 68 1998; Woodland and Angel, 2000; Woodland et al., 2010; Yamanaka et al., 2001a, 2001b; et al., 2004). 69 Recently, at pressures above 30 GPa and ambient temperature, a pseudo-rhombohedral phase of  $Fe_2SiO_4$ 70 was proposed as high-pressure phase by powder diffraction and Mössbauer spectroscopy measurements 71 (Greenberg et al., 2011). It has been reported that the spinel phase of Fe<sub>2</sub>SiO<sub>4</sub> decomposes to FeO (wüstite) and 72 SiO<sub>2</sub> (stishovite) at higher pressures (Katsura and Ito, 1989; Ito and Takahashi, 1989), because the Si atom is too 73 small make any of the post-spinel phases stable at high pressure (Ito and Takahashi. 1989). 74 Single crystal structure analyses of Fe<sub>2</sub>SiO<sub>4</sub> spinel have been carried out using a diamond anvil cell up to 8.9 75 GPa by Hazen (1993) and up to 10.2 GPa (2011). Lattice dynamics and thermodynamic properties of 76 antiferromagnetic Fe<sub>2</sub>SiO<sub>4</sub> spinel were determined using phonon dispersion curves from density functional

77 theory at pressures up to 20 GPa (et a et al., 2011).

78 Electronic spin transitions attributed to change from the high-spin (HS) to low-spin (LS) states in 79 transition metals, particularly iron, have been increasingly used to understand anomalies in the properties of 80 oxides and silicates at high pressure. X-ray emission spectroscopy (XES), carried out using a synchrotron 81 source and DAC techniques, is a method for probing the spin state of iron at high pressures et al., 2007; Li et al., 2004, Lin et al., 2005). The spin state change of  $Fe^{2+}$  in the octahedral site is accelerated at higher pressures. 82

83 In the present experiment, the details of the structural transformation of  $Fe_2SiO_4$  ringwoodite due to an 84 electronic spin transition were elucidated by x-ray diffraction and x-ray emission measurements up to 64 GPa at

ambient temperature. A new high-pressure phase was determined to have a body-centered orthorhombic structure in space group *Imma* (z = 4) rather than rhombohedral  $R\bar{3}m$ , which was previously proposed by Greenberg et al., (2011). The correlation between the structural transition and the spin state is also investigated by XES experiment.

89

# 90 Experimental Methods

91 Synchrotron x-ray Diffraction and structure analysis

92 Fe<sub>2</sub>SiO<sub>4</sub> spinel was prepared using a multianvil high-pressure apparatus<sub>0</sub> A mixture of Fe<sub>2</sub>O<sub>3</sub> and SiO<sub>2</sub> was 93 heated at 1500 °C for five hours under an atmosphere of CO<sub>2</sub> and H<sub>2</sub> with a mixing ratio of 1/1. The quenched 94 sample was confirmed to be favalite Fe<sub>2</sub>SiO<sub>4</sub>, which then served as the starting material for a high-pressure 95 synthesis of the spinel phase. The powdered favalite was placed directly into a cylindrical graphite heater that 96 was set at the center of an octahedral pressure medium. The synthesis conditions were 8 GPa and 1400  $^{\circ}$ C for 97 1 hour, employing the Kawai-type multianvil apparatus installed at ISEI, Okayama University. The graphite 98 heater prevented oxidation to ferric iron. The product was confirmed to be single-phase spinel by electron micro 99 probe analyzer and powder x-ray diffraction.

100 Angle-dispersive powder x-ray diffraction was carried out using synchrotron radiation with wavelength 101 0.4262 Å at beam line 16-BM-D at the Advanced Photon Source (APS), Argonne National Laboratory. A 102 highly focused x-ray beam with 2 mm in diameter was aligned with the center of the sample chamber in the 103 diamond-anvil cell (DAC). The diffraction patterns of the samples were recorded with an imaging plate 104 (MAR-345) and processed with FIT2D software (http://www.esrf.fr/computing/scientific/FIT2D/). The 105 detector tilting and the distance between the sample and detector were calibrated against the known lattice 106 parameters of CeO<sub>2</sub>. The lattice parameters of the samples were determined by fitting the observed diffraction 107 peaks. A symmetric diamond anvil cell (DAC) was prepared with diamonds having culet and 450 in diameter. 108 The sample chamber consisted of a hole drilled in a rhenium gasket with initial thickness of 200 µm and to 60 109 μm. The fine powder spinel sample, together with a ruby chip for a pressure marker, was loaded in the sample 110 chamber. Ne served as the pressure medium and was loaded into the cell using a gas loading apparatus. Pressure 111 was determined by the ruby luminescence method (et al., 1975; Mao et al., 1986). An off-line laser heating 112 system in the double-sided configuration, and consisting of a laser along with associated optics, was used to heat

the sample. The temperature during laser heating was approximately 1500 K, which was determined using spectral radiometry based on the Planck radiation function.

115 For the diffraction patterns of two-phase mixture over 39 GPa, the lattice parameters were obtained using 116 the DICVOL, indexing routine (Boultif and Louer, 2004), with about 12 reasonable and strong reflections in 117 range 2 =  $4^{\circ}$  -  $20^{\circ}$ . Subsequently, a Monte-Carlo method was applied to find candidates for the high-pressure 118 structure using the diffraction intensities with fixed lattice parameters as determined by DICVOL. Rietveld 119 profile fitting was then performed with the program RIETAN-2000 (Izumi and Ikeda, 2000) using the initial 120 model observed from the Mote-Carlo simulation. To fit the data, the background intensity distribution was first 121 adjusted for the refinement. Lattice parameters, atomic positional coordinates and temperature factors were 122 treated as variable parameters. Subsequently, profile parameters (full width at half maximum, asymmetry 123 parameters and angular decay parameter) were varied in the refinement. A preferred orientation correction was 124 made considering the spinel cleavage habit. Finally a full matrix, least-squares refinement was conducted.

125

126 X-ray Emission Spectroscopy (XES)

127 X-ray emission spectra (XES) measurements on Fe<sub>2</sub>SiO<sub>4</sub> were carried out up to 65 GPa at ambient 128 temperature at beam line 16-ID-D APS. The spin state of ferrous iron is characterized by the appearance of a 129 satellite emission peak located in the lower energy region of the main emission peak which is a result of the 130 3p-3d core-hole exchange. The experimental setup and specifications of the apparatus are described in a 131 previous report (Yamanaka et al., 2013). A panoramic DAC, with a configuration similar to that used in the 132 synchrotron diffraction experiments was prepared for the XES measurements, except that a Be gasket was used 133 instead of an Re gasket.

We applied the variation of the spin state through the integrated absolute values of the difference spectra (IAD et al., 2006). The high-spin (HS) and low-spin (LS) spectral functions are normalized to unit area at integration. The IAD value for the complete spin transition can be given as  $IAD_{HL} = \int |h(E) - l(E)| A$  spectrum in the transition region is a superposition of those of the two spin states, thus it can be expressed as  $s = \gamma + (1 - \gamma_{HS})$  is the high-spin fraction. Its difference from the low-spin reference l is  $s - l = \gamma_{HS}(h - l)$ . The integral of its absolute value is

140  $\operatorname{IAD}(s) = \int |s(E) - l(E)| dE = \gamma_{\text{HS}} \operatorname{IAD}_{\text{HL}}$ 

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141 The IAD is proportional to a fraction of the high spin and is a good indicator of the amount of the transition.

142

# 143 **Results**

144 Compression of the lattice parameter Fe<sub>2</sub>SiO<sub>4</sub> spinel and structure transition

Selected powder diffraction patterns obtained at increasing pressures up to 54.6 GPa at ambient temperature are shown in Fig. 1. Up to 34.8 GPa, the patterns show a single phase with the spinel structure. A new pattern, different from that of the spinel phase, was observed above 38.8 GPa, although the spinel pattern was detected even at 54.6 GPa as a residual phase.

149 In Fe<sub>2</sub>SiO<sub>4</sub> spinel, Si is located in the tetrahedral site and Fe on octahedral sites, resulting in a normal

spinel with space group 3m. Only two parameters, the oxygen positional parameter u and the lattice

151 parameter *a*, are variables in the structure refinement. The lattice parameter and density as determined from

152 the Rietveld refinement are presented in Table 1 in which those data of the residual spinels over 38.8 GPa are

also presented. The variation in the lattice parameters of spinel phase with pressure is illustrated in Fig. 2. The

154 parameter shows two distinct regimes of compression behavior, one in a lower pressure region between

ambient pressure and 20 GPa, and another above the pressure. The Si-O and Fe-O distances are presented

156 in Table 2. The Si-O and Fe-O distances and the volume of tetrahedron (SiO<sub>4</sub>) and octahedron (FeO<sub>6</sub>) are

157 calculated from the Rietveld analysis. The variations of the volume ratios of these sites  $SiO_4$ ) and R(FeO<sub>6</sub>)

158 with pressure are shown in Fig. 3. The bulk modulus was calculated using the third-order (BM) equation of

state (Birch, 1947). In the lower pressure region between ambient pressure and 20 GPa, the bulk modulus is

160  $K_0 = 177 (10)$  GPa and  $K_0' = 8.8 (1.9)$ , while in the high-pressure region between 20 GPa and 54.6 GPa,  $K_0 =$ 

161 209 (8) GPa and  $K_0$ ' = 3.8 (0.5). Overall data all through pressures are  $K_0$  = 200(9) GPa and  $K_0$ ' = 4.3(7).

162 These data are similar to the previous data:  $K_0 = 201(8)$  GPa,  $K_0' = 3.7(7)$  (Greenberg et al, 2011) and  $K_0 =$ 

163 204.5(7) GPa,  $K_0$ '=4.3(3) (Liu et al., 2008), and  $K_0$ =202(4) GPa,  $K_0$ '=4

Diffraction patterns obtained under decompression from 54.6 GPa down to 31.0 GPa are shown in Fig. 4, and reveal the back-transformation from the high-pressure phase to the spinel phase, indicating a reversible transition. After the back-transformation, the spinel phase was heated to 1500 K, resulting in a decrease in pressure from 33 GPa to 31 GPa. The diffraction pattern taken at 31 GPa shows partial decomposition of the spinel to rhombohedral FeO and SiO<sub>2</sub> stishovite together with residual spinel. It is probable that a longer heating

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period would enhance this decomposition. The recovered sample was again compressed to 55 GPa at ambient temperature, and the residual spinel transformed again to the high-pressure phase. The decomposition process confirmed the disproportionation of  $Fe_2SiO_4$  to  $2FeO + SiO_2$ , reported by Bassett and Ming (1972).

172

173 Orthorhombic high-pressure phase of Fe<sub>2</sub>SiO<sub>4</sub>

174 Rietveld profile fitting analysis of the diffraction patterns taken at 44.6 GPa and 54.6 GPa at ambient 175 temperature indicate a two-phase mixture consisting of the new high-pressure phase and the residual spinel 176 phase. The pattern at 64 GPa confirms only high-pressure single phase without spinel phase. Greenberg et al. 177 (2011) proposed a pseudo-rhombohedral phase at pressures above 30 GPa at ambient temperature from x-ray 178 diffraction and Mössbauer data. In their structure analysis, the residual spinel phase was not considered in the 179 Rietveld refinement and provided a fit to the rhombohedral R 3m structure. Several peaks in the pattern, 180 however, were not indexed or poorly fitted, as shown in Fig. 5. A fit of the data obtained at 44.6 GPa to the 181 rhombohedral structure does not result in a satisfactory refinement. On the other hand, a mixture of the 182 orthorhombic high-pressure phase and spinel fits the 54.6 GPa data quite well (Fig. 6). The diffraction pattern 183 corresponding only to the high-pressure phase was observed at 64 GPa (Fig. 7 and CIF in the supplement 184 file).

185 Rietveld analysis of the high-pressure diffraction data confirms that the new phase has a body-centered 186 orthorhombic structure with space group symmetry of *Imma* and Z=4. Hereafter the new phase is referred to 187 as I-Fe<sub>2</sub>SiO<sub>4</sub>. Iron resides in two different sites Fe1 with site symmetry of 2/create layers parallel to (101) and (011) planes, and they are very similar to the octahedral layers that form (111) and (111) planes of the spinel 188 189 structure. The Fe octahedra in the spinel structure split two different F1 and Fe2 octahedra in the I-Fe<sub>2</sub>SiO<sub>4</sub> 190 structure. And the Si tetrahedra subsidiary changes the coordination to the six-fold octahedron. The 191 Mössbauer spectra of Fe<sub>2</sub>SiO<sub>4</sub> taken at 61 GP by Greenberg et al. (2011) revealed two doublets indicating 192 two independent ferrous sites. The spectra are more consistent with I-Fe<sub>2</sub>SiO<sub>4</sub> structure than their proposed 193 rhombohedral structure with only one Fe site.

The Si atom changes coordination from four-fold in the spinel phase to six-fold in *I*-Fe<sub>2</sub>SiO<sub>4</sub> with site symmetry 2/m (4*a*). This change in coordination results in a chain of octahedra parallel to the <100> direction in *I*-Fe<sub>2</sub>SiO<sub>4</sub>. The octahedral arrays of SiO<sub>6</sub> and FeO<sub>6</sub> octahedra are shown in Fig. 8, and the result of the

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197 structural refinement of the *I*-Fe<sub>2</sub>SiO<sub>4</sub> phase is shown in Table 3. The transition is induced by atomic 198 displacements in the spinel structure, which generates the orthorhombic distortion in the *I*-Fe<sub>2</sub>SiO<sub>4</sub> 199 arrangement. The lattice parameters of spinel and *I*-Fe<sub>2</sub>SiO<sub>4</sub> is characterized by the following the axial 200 relation:

- 201 This is equivalent to a martensitic transformation with translation vector of <1/8 1/8 202  $\vec{1/8}>$  on each slab in the spinel structure, as illustrated in Fig. 9. The density of *I*-Fe<sub>2</sub>SiO<sub>4</sub> at 54.6 GPa is 5.620
- 203 g/cm<sup>3</sup>, about 1 % larger than that of the residual spinel phase 5.572 g/cm<sup>3</sup>, as shown in Table 1.
- 204

205 Photoemission spectroscopy shows that intra-atomic interactions dominate the  $K\beta$  spectral line shape. 206  $K\beta'$  and  $K\beta_{13}$  lines shift towards each other with decreasing valence spin and (3p, 3d) exchange interaction 207 (Lin et al., 2005, 2007; Li et al., 2006). In previous work, we found a high-spin to intermediate-spin transition 208 in Fe<sub>2</sub>TiO<sub>4</sub> at about 19 GPa (Yamanaka et al., 2013). The result and the transition pressure of Fe<sub>3</sub>O<sub>4</sub> at 15.8 209 GPa (Ding et al., 2008) and) are much lower spin transition pressures as compared with magnetio-wüstite 210 and other earth materials (et al., 1999; Lin et al., 2005, 2010). X-ray emission (XES) spectra in the Fe  $K\beta$ 211 region for the spinel phase of  $Fe_2SiO_4$  are presented in Fig. 10, and indicate an intermediate spin state, 212 resulting from the relative integrated intensities due to the energy shift of. The integrated absolute difference 213 (IAD) is proportional to the Fe fraction in high spin state (Venko et al., 2006). The observed electronic spin 214 transition pressure of I-Fe<sub>2</sub>SiO<sub>4</sub> starts at about 17 GPa. This is a little lower pressure than the structural 215 transition pressure of 20 GPa observed by XRD. From the high spin (HS) to intermediate spin (IS) transition, 216 the  $K\beta$  peak intensity decreases gradually. However, an ideal low-spin state (LS) was not generated even at 217 65 GPa of the highest pressure achieved in this work. The spin transition starts at a little lower pressure than 218 the structural transition pressure of many iron-bearing oxides and silicates, as observed by x-ray diffraction. 219 For example, XES measurements on Fe<sub>3</sub>O<sub>4</sub> by Ding et al. (2008) indicate that the spin transition takes place 220 at a lower pressure (15.8 GPa) than the structural change to the post-spinel phase (23 GPa). Fe<sub>2</sub>TiO<sub>4</sub> also 221 undergoes a transition to an intermediate-spin state beginning at 14 GPa (Yamanaka et al., 2013). 222

# 223 Discussion

224 Pressure-induced electronic spin transitions of ferrous ion are a crucial factor in the compression behavior

225 of spinels, magnesiowüstite, perovskites and post-perovskites. A high-spin to intermediate-spin transition of 226 Fe<sub>3</sub>O<sub>4</sub> occurs at 15.8 GPa (Ding et al., 2008) and Fe<sub>2</sub>TiO<sub>4</sub> at about 19 GPa (Yamanaka et al., 2013). Those 227 pressures are a little lower pressure than structure transition pressure. The change in the spin state of  $Fe_2SiO_4$ 228 observed at approximately 17 GPa should also have a significant effect on the effective ionic radii. In the high-spin state  $\mathrm{Fe}^{2+}$  (3d<sup>6</sup>) the octahedral site at ambient conditions, two electrons reside in the doubly  $e_g$ 229 230 orbital in  $t_{2g}$  orbital. In low-spin state under high-pressure conditions, one or two  $e_g$  electrons move down in energy to the  $t_{2g}$  orbital additional possibilities for d-p- $\pi$  bonding (Fe  $t_{2g}$ -O 2p), but less possibilities for 231 232  $\sigma$ -type (Fe - O 2p) bonding. These ligand configuration effects combine to give a smaller effective ionic radius for the low spin state of Fe<sup>2+</sup>. In the case of Fe<sub>3</sub>O<sub>4</sub>, the XES result can be interpreted as the spin 233 transition at  $Fe^{2+}$  in the octahedral site, while the two  $Fe^{3+}$  remain in the high spin state. 234 The ionic radii reported by Shannon (1976) gives the radii for  $Fe^{2+}$  as 0.780 Å at HS and 0.61 Å at LS. 235

The spin transition therefore reduces the ferrous ion radius by about 20 %. The Fe<sup>2+</sup>-O bond distance is 236 237 likewise reduced from A change in the effective ionic radius brings about a polyhedral distortion. While the 238 Si-O bond is not easily compressed, the Fe-O bond length is reduced dramatically under compression. The 239 observed decrease in the lattice parameter leads to the corresponding change in octahedral volume. The 240 volume ratio of the octahedron shows a distinct change due to the compression behavior of Fe-O bond 241 distance at 20 GPa. These discontinuous changes are induces by the spin transition, starting at 17 GPa. 242 However, the volume of the tetrahedron does not show a noticeable change in contrast to the octahedra. Fe2 243 ion changes the spin state to an intermediate spin, resulting in its greater distortion of  $FeO_6$  and smaller bond 244 lengths, but Fe1 ion located at a larger and less distorted site probably remains in the HS state even at 64 GPa, 245 (Table 4).

246

# 247 Implication of this paper

Numerous investigations of the structure transitions and decompositions of spinels have been executed under extreme conditions. Silicate spinels with transition elements or mixed-charge cations have been intensively studied from various viewpoints such as their magnetic susceptibility, electric conductivity, or elastic property.

252 The olivine-spinel transformation has been strongly studied for significance at the transition zone of the

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earth's mantle. Fe<sub>2</sub>SiO<sub>4</sub> has an olivine structure ( $\alpha$ -Fe<sub>2</sub>SiO<sub>4</sub>, favalite) at ambient conditions and transforms

254 directly into spinel (γ-Fe<sub>2</sub>SiO<sub>4</sub>, ahrensite) under high pressure at ambient temperature. Silicate spinels have

255 been known as an essentially metastable phase at ambient conditions.

There are several passes of spinel structure transformations. Some oxide spinels with transition elements have their high-pressure polymorphs due to the transition without decomposition. A pseudo-rhombohedral phase at pressures above 30 GPa and ambient temperature was reported on the basis of powder diffraction data and Mössbauer data (Greenberg et al., 2011). However, structure parameters including atomic positional parameters of the high-pressure phase are not reported in their paper and their Rietveld analysis at 48(2) GPa did not consider the residual spinel phase in the refinement. Hence, several peaks are not clearly fitted.

Present paper shows a structural change in  $Fe_2SiO_4$  spinel under high pressure up to 64 GPa. A new high-pressure structure of *I*-Fe<sub>2</sub>SiO<sub>4</sub> is determined by Rietveld profile fitting of synchrotron x-ray diffraction data at ambient temperature. A transition from the cubic spinel structure to a body centered orthorhombic phase (*I*-Fe<sub>2</sub>SiO<sub>4</sub>) with space group *Imma* and Z=4 was first observed at approximately 39 GPa. The

- structure of *I*-Fe<sub>2</sub>SiO<sub>4</sub> has two crystallographically distinct FeO<sub>6</sub> octahedral sites and Si atom changes its
   configuration.
- In addition to the new structure, we first found two different compression curves of the lattice parameter in the spinel phase and the discontinuity at approximately 20 GPa. Fe- $K\beta$  x-ray emission measurements at elevating pressure show that the transition from high-spin (HS) state to intermediate-spin (IS) state begins at 17 GPa in the spinel phase. The IS electronic state is gradually enhanced with pressure, which generates an isostructural change in the lattice parameter at 20 GPa. The spin transition induces the compression of the bond length, resulting in the structure transition at 39 GPa. The spin transition can be emphasized for the trigger of the structure transition.
- 275

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- auspices of KEK proposals No. 2004G229 for powder diffraction at BL-13A and BL-18C of the Photon 281
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326 **Table captions** 

- 327 **Table 1.**
- 328 Lattice parameters were determined by Rietveld profile fitting analysis. Spinel phase is found at pressures above
- 329 the transition pressure of 38.8 GPa and coexists with the high-pressure phase of *I*-Fe<sub>2</sub>SiO<sub>4</sub>.

330

331 Table 2.

332

- **Table 3.**
- 334 Reliability factors for the least-squares calculation are

335 
$$wR_{p} = \left[\frac{\sum_{i} w_{i} |y_{i} - f_{i}(x)|}{\sum_{i} w_{i} y_{i}}\right]^{1/2} R_{p} = \left[\frac{\sum_{i} |y_{i} - f_{i}(x)|}{\sum_{i} y_{i}}\right]$$

336 
$$R_{F} = \left[\frac{\sum_{k} |F_{k obs}| - |F_{k cal}|}{\sum_{k} F_{k obs}}\right] \qquad s = \left\lfloor\frac{\sum_{i} w_{i} |y_{i} - f_{i}(x)|}{N - P}\right\rfloor$$

337

338

**Table 4.** 

340

# **Figure captions**

- 342 Figure 1. Selected x-ray diffraction patterns of Fe<sub>2</sub>SiO<sub>4</sub> taken with increasing pressure at ambient
- 343 temperature.
- 344 The high-pressure phase was found above 39.2 GPa and observed up to 56.6 GPa.
- 345 Figure 2.
- 346 There are two distinct compression regimes indicating the boundary at about 20 GPa. This behavior is a
- 347 little higher pressure than the HS-to-IS transition of 17 GPa observed by XES. The data of Greenberg et al.
- 348 (2011) are also presented. The dotted curve represents the lattice parameter of the residual spinel coexist with
- 349 *I*-Fe<sub>2</sub>SiO<sub>4</sub>. The observed error is smaller than the data point.
- **Figure 3.** Average bond length ratio of tetrahedral and octahedral sites of Fe<sub>2</sub>SiO<sub>4</sub> spinel.
- Bond lengths were obtained from the Rietveld profile fitting method. Error is smaller than the symbol mark.
- 352 The Fe-O bond length in the octahedral site confirms the two different compression regimes at 20 GPa.
- 353 Figure 4. The decompression experiment confirms a reversible transition between cubic and orthorhombic
- 354 phases. Subsequent laser heating at 1500 K and 31 GPa reveals decomposition from spinel to rhombohedral
- 355 FeO and SiO<sub>2</sub> stishovite. Reflections with symbols of I and in the figure represent *I*-Fe<sub>2</sub>SiO<sub>4</sub> and spinel phase,
- 356 respectively.
- **Figure 5.** Rietveld profile fitting of the initial structure model of rhombohedral Fe<sub>2</sub>SiO<sub>4</sub> to the data obtained
- 358 at 54.6 GPa. The rhombohedral structure model (space group R 3m) is simply derived from a distortion
- along the <111 > direction of the spinel structure. Peaks indicated by the arrows are not indexed.
- 360 Figure 6. Rietveld profile fitting of diffraction data obtained at 54.6 GPa, assuming a two-phase mixture of
- orthorhombic Fe<sub>2</sub>SiO<sub>4</sub> and spinel. Upper and lower vertical bars indicate peak positions for orthorhombic
   Fe<sub>2</sub>SiO<sub>4</sub> and spinel, respectively.
- Figure 7. Rietveld profile fitting of diffraction data obtained at 64 GPa. Rietveld profile fitting was carried out in consideration of two-phase mixture of I-Fe<sub>2</sub>SiO<sub>4</sub>. Vertical bars indicate peak positions of the I-Fe<sub>2</sub>SiO<sub>4</sub>.
- 366 **Figure 8.** There are two distinct
- Figure 8. There are two distinct octahedral Fe sites, Fe1 and Fe2. Si also is located at the site of octahedral coordination. Fe1 and Fe2 octahedra create layers parallel to (011) respectively. SiO<sub>6</sub> octahedra make an array in the direction of <100]>.

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369 **Figure 9.** 

- 370 Fe<sub>2</sub>SiO<sub>4</sub> spinel (ringwoodite ahrensite) and high-pressure phase *I*-Fe<sub>2</sub>SiO<sub>4</sub> are presented in the left and right
- figure, respectively. Arrow indicates a martensitic transformation with translation vector  $\langle 1/8 \ 1/8 \ 1/8 \rangle$ ,
- 372 which generates the *I*-Fe<sub>2</sub>SiO<sub>4</sub> structure from the spinel. The shadowed circles in the left figure represent the
- 373 atomic positions of I-Fe<sub>2</sub>SiO<sub>4</sub>.
- **Figure 10.** Fe  $K\beta$  x-ray emission spectra with increasing pressure up to 64.8 GPa. The upper left figure
- 375 shows the expanded  $K\beta$  spectra, indicating an intermediate spin transition. The spin transition occurs at
- approximately 17 GPa.
- 377
- 378



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P (GPa)	LC (Å)	P (GPa)	LC (Å)	
spinel sta	ble region	two-phase mixuture		
		metasatb	le spinel	
0.0001	8.2374 (4)	38.8	7.859(4)	
1.2	8.2156 (8)	39.2	7.855(4)	
3.7	8.1783 (9)	44.6	7.756(2)	
4.0	8.1759 (8)	54.6	7.730(2)	
7.8	8.1325 (8)			
9.1	8.1140 (8)			
11.5	8.0905 (8)			
13.3	8.0759(9)			
15.4	8.057(1)			
17.3	8.044 (1)			
18.8	8.028 (1)			
20.3	8.013 (1)			
22.3	7.996(1)			
23.1	7.990 (1)			
24.2	7.976 (2)			
27.2	7.951 (2)			
31.0	7.916 (2)			
32.9	7.895(2)			
34.8	7.889 (3)			

 $Table \ 1 \quad Lattice \ constant \ of \ Fe_2SiO_4 \ spinel$ 

Press GPa	Cell const Å	Cell vol Å <sup>3</sup>	u-parameter	d(Si-O) Å	d(Fe-O) Å Vol	(SiO4) Å <sup>3</sup> Vol	(FeO6) Å <sup>3</sup>
0.0001	8.2374(4)	558.95(6)	0.3659(2)	1.654(7)	2.137(7)	2.321(2)	12.914(2)
1.2	8.2156(8)	554.52(7)	0.3661(2)	1.652(6)	2.130(6)	2.314(4)	12.784(3)
4.0	8.1759(8)	546.46(8)	0.3666(2)	1.651(6)	2.115(6)	2.310(2)	12.530(2)
9.1	8.1142(8)	533.81(7)	0.3670(2)	1.644(7)	2.096(6)	2.282(3)	12.197(3)
13.3	8.0759(5)	526.71(5)	0.3673(8)	1.641(5)	2.083(5)	2.267(1)	11.986(1)
17.3	8.044(1)	520.5(2)	0.3678(9)	1.641(17)	2.071(17)	2.269(9)	11.779(9)
23.1	7.990(1)	510.0(2)	0.3692(11)	1.650(18)	2.045(18)	2.304(12)	11.368(13)
32.9	7.895(3)	492.1(1)	0.3700(11)	1.641(13)	2.014(13)	2.268(12)	10.868(13)
34.8	7.889(3)	490.9(1)	0.3702(13)	1.642(12)	2.011(12)	2.274(22)	10.818(23)
metastable spinel above the transition pressure							
38.8	7.879(4)	489.1(1)	0.3705(25)	1.644(17)	2.006(17)	2.272(23)	10.74(23)
39.2	7.855(4)	484.7(1)	0.3711(22)	1.648(16)	1.995(16)	2.295(23)	10.57(23)
44.6	7.756(2)	466.6(3)	0.3707(15)	1.622(12)	1.973(12)	2.188(16)	10.22(16)
54.6	7.730(2)	461.9(2)	0.3708(11)	1.617(10)	1.966(10)	2.171(14)	10.11(14)

 Table 2
 Bond distance and site volume of spinel phase under pressure

Pressure			44.6GPa*	54.6GPa*	64.8GPa
Space group			Imma	Imma	Imma
Unit molecule			4	4	4
a (Å)		5.551(3)	5.543(1)	5.522(3)	
b	(Å)		6.030(4)	6.032(2)	6.025(4)
с	(Å)		7.241(5)	7,201(4)	7.185(5)
volum	e (Å3)		242.4(4)	240.8(2)	239.0(2)
Rwp	)		2.996	2.531	2.369
RB			1.993	1.354	7.573
$R_{\rm F}$			0.985	1.042	4.088
s			0.4047	0.3436	1.8538
Si	4a	X	0.0	0.0	0.0
		y	0.0	0.0	0.0
		Z	0.0	0.0	0.0
		Biso	4.83(2)	4.87(2)	2.05(2)
Fe1	4b	X	0.0	0.0	0.0
		У	0.0	0.0	0.0
		Z	0.5	0.5	0.5
		Biso	6.74(1)	3.54(1)	2.90(1)
Fe2	4c	X	0.25	0.25	0.25
		У	0.25	0.25	0.25
		Z	0.25	0.25	0.25
		Biso	4.92(1)	3.88(3)	2.92(5)
01	8h	X	0.0	0.0	0.0
		У	0.5101(9)	0.5073(5)	0.5068(9)
		Z	0.7652(9)	0.7608(8)	0.7610(8)
		Biso	3.6531)	4.00(1)	3.72(2)
O2	8i	X	0.2371(9)	0.2342(8)	0.2333(2)
		У	0.25	0.25	0.25
		Z	0.0022(9)	0.0014(9)	0.0012(6)
		Biso	5.91(2)	6.24(5)	5.54(5)

Table 3 Structure parameters of high-pressure phase of *I*-Fe<sub>2</sub>SiO<sub>4</sub>

Pressure	44.6GPa	54.6GPa	64.8GPa
Si $O1 \times 2$ (Å)	1.701(4)	1.723(4)	1.718(4)
O1   x 2   (h) O2   x 4   (Å)	2.001(4)	1.990(4)	1.982(4)
mean (Å)	1.901(4)	1.901(4)	1.894(4)
volume (ų)	8.99(6)	8.99(6)	8.89(4)
Fe1 O1 x2 (Å)	1.921(5)	1.879(3)	1.876(3)
O2 x4 (Å)	2.098(5)	2.108(3)	2.107(3)
mean (Å)	2.039(5)	2.032(3)	2.030(3)
volume (ų)	11.26(6)	11.13(5)	11.09(5)
Fe2 O1 x4 (Å)	2.008(6)	2.017(6)	2.015(4)
$O2 \times 2 (\text{\AA})$	1.796(6)	1.792(6)	1.790(4)
mean (Å)	1.937(6)	1.942(8)	1.940(4)
volume (ų)	9.58(4)	9.66(2)	9.61(2)

Table 4 Deformation of Si, Fe1 and Fe2 octahedra of *I*-Fe<sub>2</sub>SiO<sub>4</sub>