Abstract

A structural change in Fe$_2$SiO$_4$ spinel (ringwoodite) has been found by synchrotron powder diffraction study and the structure of a new high-pressure phase was determined by Mote-Carlo simulation method and Rietveld profile fitting of x-ray diffraction data up to 64 GPa at ambient temperature. A transition from the cubic spinel structure to a body centered orthorhombic phase ($I$-Fe$_2$SiO$_4$) with space group $Imma$ and $Z=4$ was observed at approximately 34 GPa. The structure of $I$-Fe$_2$SiO$_4$ has two crystallographically independent FeO$_6$ octahedra. Iron resides in two different sites of six-fold coordination: Fe1 and are arranged in layers parallel to (101) and (011) and are very similar to the layers of FeO$_6$ octahedra in the spinel structure. Silicon is located in the six-fold coordination in $I$-Fe$_2$SiO$_4$. The transformation to the new high-pressure phase is
reversible under decompression at ambient temperature. A martensitic transformation of each slab of the spinel structure with translation vector $<1/8, 1/8, 1/8>$ generates the $I$-$Fe_2SiO_4$ structure. Laser heating of $I$-$Fe_2SiO_4$ at 1500 K results in a decomposition of the material to rhombohedral FeO and SiO$_2$ stishovite. Fe $K\beta$ x-ray emission measurements at high pressure up to 65GPa show that the transition from a high spin (HS) to an intermediate spin (IS) state begins at 17 GPa in the spinel phase. The IS electron spin state is gradually enhanced with pressure. The Fe$^{2+}$ ion at the octahedral site changes the ion radius under compression at the low spin, which results in the changes of the lattice parameter and the deformation of the octahedra of the spinel structure. The compression curve of the lattice parameter of the spinel is discontinuous at approximately 20 GPa. The spin transition induces an isostructural change.

Keywords
New high-pressure structure, Fe$_2$SiO$_4$ ringwoodite, x-ray emission spectra, spin transition, martensitic transition.

Introduction
A great deal of attention has been paid to the high-pressure structural transitions of the many spinel phases present in the Earth's crust due to their geophysical importance (Akimoto and Fujisawa, 1967; Basset and Ming, 1972; Ito and Takahashi, 1987; Irifune et al. 1998). One of the major minerals in the crust, (Mg,Fe)$_2$SiO$_4$ olivine ($\alpha$-phase), transforms to wadsleyite ($\beta$-phase, modified spinel) and further to ringwoodite ($\gamma$-phase, spinel). These transitions were proposed for the origin of the seismic discontinuity of the transition zone from 410 km to 660 km depth (Ringwood and Irifune, 1988). These high-pressure transformations have been studied from various viewpoints, including the electronic and elastic properties of participating phases (Kiefer et al. 1997; Li et al. 2007) and continue to provide significant information for seismic interpretation (Leven et al. 1981; Burnley et al. 1991; Shim et al. 2001).

Recently Fe$_2$SiO$_4$ phase with spinel structure was found in meteorite and Silicate spinels of transition elements are investigated for their influence on the magnetic and electrical properties of the Earth's crust and mantle. Their phase stabilities and structures under high-pressure and high-temperature conditions have been
intensively studied. Phase relations in the Mg$_2$SiO$_4$-Fe$_2$SiO$_4$ ringwoodite solid solution system have been investigated in numerous high-pressure studies (Katsura et al., 1989; Ito and Takahashi, 1989; Fei et al., 1999; Matsuzaka et al., 2000). Mechanisms for the polymorphic transformations of the α, β and γ phases of (Mg, have been discussed in terms of the observation of dislocations by TEM (Price et al. 1982; Madon and Poirier 1983).

Structures of silicate spinels M$_2$SiO$_4$ (M= Mg, Fe, Co, Ni) have been refined from x-ray diffraction data at ambient conditions (Yagi et al. 1974; Morimoto et al. 1970, 1974; Ma, 1975; Marumo et al. 1974, 1977). (1981). End member of ringwoodite solid solution, Fe$_2$SiO$_4$, undergoes transitions at 6.4 GPa at 1700℃ (Yagi et al., 1987). Further experimental studies have been undertaken by Fei et al., (1991) and Matsuzaka a (2000).

Significant work is also focused on the physical properties of iron-bearing spinels in order to understand their strong electronic correlations as manifested in charge transfer, electron hopping and the Jahn-Teller effect, in particular through magnetic and electrical conductivity measurements and Raman spectroscopy l. (1998; Woodland and Angel, 2000; Woodland et al., 2010; Yamanaka et al., 2001a, 2001b; et al., 2004).

Recently, at pressures above 30 GPa and ambient temperature, a pseudo-rhombohedral phase of Fe$_2$SiO$_4$ was proposed as high-pressure phase by powder diffraction and Mössbauer spectroscopy measurements (Greenberg et al., 2011). It has been reported that the spinel phase of Fe$_2$SiO$_4$ decomposes to FeO (wüstite) and SiO$_2$ (stishovite) at higher pressures (Katsura and Ito, 1989; Ito and Takahashi, 1989), because the Si atom is too small make any of the post-spinel phases stable at high pressure (Ito and Takahashi. 1989).

Single crystal structure analyses of Fe$_2$SiO$_4$ spinel have been carried out using a diamond anvil cell up to 8.9 GPa by Hazen (1993) and up to 10.2 GPa (2011). Lattice dynamics and thermodynamic properties of antiferromagnetic Fe$_2$SiO$_4$ spinel were determined using phonon dispersion curves from density functional theory at pressures up to 20 GPa (et a et al., 2011).

Electronic spin transitions attributed to change from the high-spin (HS) to low-spin (LS) states in transition metals, particularly iron, have been increasingly used to understand anomalies in the properties of oxides and silicates at high pressure. X-ray emission spectroscopy (XES), carried out using a synchrotron source and DAC techniques, is a method for probing the spin state of iron at high pressures et al., 2007; Li et al., 2004, Lin et al., 2005). The spin state change of Fe$^{2+}$ in the octahedral site is accelerated at higher pressures.

In the present experiment, the details of the structural transformation of Fe$_2$SiO$_4$ ringwoodite due to an electronic spin transition were elucidated by x-ray diffraction and x-ray emission measurements up to 64 GPa at
ambient temperature. A new high-pressure phase was determined to have a body-centered orthorhombic
structure in space group $Imma$ ($z = 4$) rather than rhombohedral $R\overline{3}m$, which was previously proposed by
Greenberg et al., (2011). The correlation between the structural transition and the spin state is also investigated
by XES experiment.

**Experimental Methods**

Synchrotron x-ray Diffraction and structure analysis

Fe$_2$SiO$_4$ spinel was prepared using a multianvil high-pressure apparatus. A mixture of Fe$_2$O$_3$ and SiO$_2$ was
heated at 1500 °C for five hours under an atmosphere of CO$_2$ and H$_2$ with a mixing ratio of 1/1. The quenched
sample was confirmed to be fayalite Fe$_2$SiO$_4$, which then served as the starting material for a high-pressure
synthesis of the spinel phase. The powdered fayalite was placed directly into a cylindrical graphite heater that
was set at the center of an octahedral pressure medium. The synthesis conditions were 8 GPa and 1400 °C for
1 hour, employing the Kawai-type multianvil apparatus installed at ISEI, Okayama University. The graphite
heater prevented oxidation to ferric iron. The product was confirmed to be single-phase spinel by electron micro
probe analyzer and powder x-ray diffraction.

Angle-dispersive powder x-ray diffraction was carried out using synchrotron radiation with wavelength
0.4262 Å at beam line 16-BM-D at the Advanced Photon Source (APS), Argonne National Laboratory. A
highly focused x-ray beam with 2 mm in diameter was aligned with the center of the sample chamber in the
diamond-anvil cell (DAC). The diffraction patterns of the samples were recorded with an imaging plate
(MAR-345) and processed with FIT2D software (http://www.esrf.fr/computing/scientific/FIT2D/). The
detector tilting and the distance between the sample and detector were calibrated against the known lattice
parameters of CeO$_2$. The lattice parameters of the samples were determined by fitting the observed diffraction
peaks. A symmetric diamond anvil cell (DAC) was prepared with diamonds having culet and 450 in diameter.
The sample chamber consisted of a hole drilled in a rhenium gasket with initial thickness of 200 μm and to 60
μm. The fine powder spinel sample, together with a ruby chip for a pressure marker, was loaded in the sample
chamber. Ne served as the pressure medium and was loaded into the cell using a gas loading apparatus. Pressure
was determined by the ruby luminescence method (et al., 1975; Mao et al., 1986). An off-line laser heating
system in the double-sided configuration, and consisting of a laser along with associated optics, was used to heat
the sample. The temperature during laser heating was approximately 1500 K, which was determined using spectral radiometry based on the Planck radiation function.

For the diffraction patterns of two-phase mixture over 39 GPa, the lattice parameters were obtained using the DICVOL, indexing routine (Boultif and Louer, 2004), with about 12 reasonable and strong reflections in range 2θ = 4° - 20°. Subsequently, a Monte-Carlo method was applied to find candidates for the high-pressure structure using the diffraction intensities with fixed lattice parameters as determined by DICVOL. Rietveld profile fitting was then performed with the program RIETAN-2000 (Izumi and Ikeda, 2000) using the initial model observed from the Monte-Carlo simulation. To fit the data, the background intensity distribution was first adjusted for the refinement. Lattice parameters, atomic positional coordinates and temperature factors were treated as variable parameters. Subsequently, profile parameters (full width at half maximum, asymmetry parameters and angular decay parameter) were varied in the refinement. A preferred orientation correction was made considering the spinel cleavage habit. Finally a full matrix, least-squares refinement was conducted.

X-ray Emission Spectroscopy (XES)

X-ray emission spectra (XES) measurements on Fe$_2$SiO$_4$ were carried out up to 65 GPa at ambient temperature at beam line 16-ID-D APS. The spin state of ferrous iron is characterized by the appearance of a satellite emission peak located in the lower energy region of the main emission peak which is a result of the 3p–3d core–hole exchange. The experimental setup and specifications of the apparatus are described in a previous report (Yamanaka et al., 2013). A panoramic DAC, with a configuration similar to that used in the synchrotron diffraction experiments was prepared for the XES measurements, except that a Be gasket was used instead of an Re gasket.

We applied the variation of the spin state through the integrated absolute values of the difference spectra (IAD et al., 2006). The high-spin (HS) and low-spin (LS) spectral functions are normalized to unit area at integration. The IAD value for the complete spin transition can be given as $IAD_{HL} = \int |h(E) - l(E)| dE$. A spectrum in the transition region is a superposition of those of the two spin states, thus it can be expressed as $s = \gamma + (1 - \gamma)\text{HS}$. Its difference from the low-spin reference $l$ is $s - l = \gamma \text{HS}(h - l)$. The integral of its absolute value is

$$IAD(s) = \int |s(E) - l(E)| dE = \gamma \text{HS} IAD_{HL}.$$
The IAD is proportional to a fraction of the high spin and is a good indicator of the amount of the transition.

Results

Compression of the lattice parameter $\text{Fe}_2\text{SiO}_4$ spinel and structure transition

Selected powder diffraction patterns obtained at increasing pressures up to 54.6 GPa at ambient temperature are shown in Fig. 1. Up to 34.8 GPa, the patterns show a single phase with the spinel structure. A new pattern, different from that of the spinel phase, was observed above 38.8 GPa, although the spinel pattern was detected even at 54.6 GPa as a residual phase.

In $\text{Fe}_2\text{SiO}_4$ spinel, Si is located in the tetrahedral site and Fe on octahedral sites, resulting in a normal spinel with space group $\overline{3}m$. Only two parameters, the oxygen positional parameter $u$ and the lattice parameter $a$, are variables in the structure refinement. The lattice parameter and density as determined from the Rietveld refinement are presented in Table 1 in which those data of the residual spinels over 38.8 GPa are also presented. The variation in the lattice parameters of spinel phase with pressure is illustrated in Fig. 2. The parameter shows two distinct regimes of compression behavior, one in a lower pressure region between ambient pressure and 20 GPa, and another above the pressure. The Si-O and Fe-O distances are presented in Table 2. The Si-O and Fe-O distances and the volume of tetrahedron ($\text{SiO}_4$) and octahedron ($\text{FeO}_6$) are calculated from the Rietveld analysis. The variations of the volume ratios of these sites SiO$_4$) and R(FeO$_6$) with pressure are shown in Fig. 3. The bulk modulus was calculated using the third-order (BM) equation of state (Birch, 1947). In the lower pressure region between ambient pressure and 20 GPa, the bulk modulus is $K_0 = 177 (10)$ GPa and $K'_0 = 8.8 (1.9)$, while in the high-pressure region between 20 GPa and 54.6 GPa, $K_0 = 209 (8)$ GPa and $K'_0 = 3.8 (0.5)$. Overall data all through pressures are $K_0 = 200(9)$ GPa and $K'_0 = 4.3(7)$.

These data are similar to the previous data: $K_0 = 201(8)$ GPa, $K'_0 = 3.7(7)$ (Greenberg et al., 2011) and $K_0 = 204.5(7)$ GPa, $K'_0 = 4.3(3)$ (Liu et al., 2008), and $K_0 = 202(4)$ GPa, $K'_0 = 4$.

Diffraction patterns obtained under decompression from 54.6 GPa down to 31.0 GPa are shown in Fig. 4, and reveal the back-transformation from the high-pressure phase to the spinel phase, indicating a reversible transition. After the back-transformation, the spinel phase was heated to 1500 K, resulting in a decrease in pressure from 33 GPa to 31 GPa. The diffraction pattern taken at 31 GPa shows partial decomposition of the spinel to rhombohedral FeO and SiO$_2$ stishovite together with residual spinel. It is probable that a longer heating
period would enhance this decomposition. The recovered sample was again compressed to 55 GPa at ambient
temperature, and the residual spinel transformed again to the high-pressure phase. The decomposition process
confirmed the disproportionation of Fe$_2$SiO$_4$ to 2FeO +SiO$_2$, reported by Bassett and Ming (1972).

Orthorhombic high-pressure phase of Fe$_2$SiO$_4$

Rietveld profile fitting analysis of the diffraction patterns taken at 44.6 GPa and 54.6 GPa at ambient
temperature indicate a two-phase mixture consisting of the new high-pressure phase and the residual spinel
phase. The pattern at 64 GPa confirms only high-pressure single phase without spinel phase. Greenberg et al.
(2011) proposed a pseudo-rhombohedral phase at pressures above 30 GPa at ambient temperature from x-ray
diffraction and Mössbauer data. In their structure analysis, the residual spinel phase was not considered in the
Rietveld refinement and provided a fit to the rhombohedral $R\bar{3}m$ structure. Several peaks in the pattern,
however, were not indexed or poorly fitted, as shown in Fig. 5. A fit of the data obtained at 44.6 GPa to the
rhombohedral structure does not result in a satisfactory refinement. On the other hand, a mixture of the
orthorhombic high-pressure phase and spinel fits the 54.6 GPa data quite well (Fig. 6). The diffraction pattern
corresponding only to the high-pressure phase was observed at 64 GPa (Fig. 7 and CIF in the supplement
file).

Rietveld analysis of the high-pressure diffraction data confirms that the new phase has a body-centered
orthorhombic structure with space group symmetry of $Imma$ and $Z=4$. Hereafter the new phase is referred to
as $I$-Fe$_2$SiO$_4$. Iron resides in two different sites Fe1 with site symmetry of $2/m$ create layers parallel to (101) and
(011) planes, and they are very similar to the octahedral layers that form (111) and (1 1 1) planes of the spinel
structure. The Fe octahedra in the spinel structure split two different F1 and Fe2 octahedra in the $I$-Fe$_2$SiO$_4$
structure. And the Si tetrahedra subsidiary changes the coordination to the six-fold octahedron. The
Mössbauer spectra of Fe$_2$SiO$_4$ taken at 61 GP by Greenberg et al. (2011) revealed two doublets indicating
two independent ferrous sites. The spectra are more consistent with $I$-Fe$_2$SiO$_4$ structure than their proposed
rhombohedral structure with only one Fe site.

The Si atom changes coordination from four-fold in the spinel phase to six-fold in $I$-Fe$_2$SiO$_4$ with site
symmetry $2/m$ ($4a$). This change in coordination results in a chain of octahedra parallel to the $<100>$ direction
in $I$-Fe$_2$SiO$_4$. The octahedral arrays of SiO$_6$ and FeO$_6$ octahedra are shown in Fig. 8, and the result of the
structural refinement of the $I$-Fe$_2$SiO$_4$ phase is shown in Table 3. The transition is induced by atomic displacements in the spinel structure, which generates the orthorhombic distortion in the $I$-Fe$_2$SiO$_4$ arrangement. The lattice parameters of spinel and $I$-Fe$_2$SiO$_4$ is characterized by the following the axial relation:

This is equivalent to a martensitic transformation with translation vector of $<1/8 \ 1/8>$ on each slab in the spinel structure, as illustrated in Fig. 9. The density of $I$-Fe$_2$SiO$_4$ at 54.6 GPa is 5.620 g/cm$^3$, about 1 % larger than that of the residual spinel phase 5.572 g/cm$^3$, as shown in Table 1.

Photoemission spectroscopy shows that intra-atomic interactions dominate the K$\beta$ spectral line shape. K$\beta'$ and K$\beta_{1,3}$ lines shift towards each other with decreasing valence spin and (3$p$, 3$d$) exchange interaction (Lin et al., 2005, 2007; Li et al., 2006). In previous work, we found a high-spin to intermediate-spin transition in Fe$_2$TiO$_4$ at about 19 GPa (Yamanaka et al., 2013). The result and the transition pressure of Fe$_3$O$_4$ at 15.8 GPa (Ding et al., 2008) and) are much lower spin transition pressures as compared with magnetio-wüstite and other earth materials (et al., 1999; Lin et al., 2005, 2010). X-ray emission (XES) spectra in the Fe K$\beta$ region for the spinel phase of Fe$_2$SiO$_4$ are presented in Fig. 10, and indicate an intermediate spin state, resulting from the relative integrated intensities due to the energy shift of. The integrated absolute difference (IAD) is proportional to the Fe fraction in high spin state (Venko et al., 2006). The observed electronic spin transition pressure of $I$-Fe$_2$SiO$_4$ starts at about 17 GPa. This is a little lower pressure than the structural transition pressure of 20 GPa observed by XRD. From the high spin (HS) to intermediate spin (IS) transition, the K$\beta'$ peak intensity decreases gradually. However, an ideal low-spin state (LS) was not generated even at 65 GPa of the highest pressure achieved in this work. The spin transition starts at a little lower pressure than the structural transition pressure of many iron-bearing oxides and silicates, as observed by x-ray diffraction. For example, XES measurements on Fe$_3$O$_4$ by Ding et al. (2008) indicate that the spin transition takes place at a lower pressure (15.8 GPa) than the structural change to the post-spinel phase (23 GPa). Fe$_2$TiO$_4$ also undergoes a transition to an intermediate-spin state beginning at 14 GPa (Yamanaka et al., 2013).

**Discussion**

Pressure-induced electronic spin transitions of ferrous ion are a crucial factor in the compression behavior
of spinels, magnesiowüstite, perovskites and post-perovskites. A high-spin to intermediate-spin transition of FeO occurs at 15.8 GPa (Ding et al., 2008) and FeTiO at about 19 GPa (Yamanaka et al., 2013). Those pressures are a little lower pressure than structure transition pressure. The change in the spin state of FeSiO observed at approximately 17 GPa should also have a significant effect on the effective ionic radii. In the high-spin state Fe²⁺ (3d⁶) the octahedral site at ambient conditions, two electrons reside in the doubly $e_g$ orbital in $t_{2g}$ orbital. In low-spin state under high-pressure conditions, one or two $e_g$ electrons move down in energy to the $t_{2g}$ orbital additional possibilities for $d-p-\pi$ bonding (Fe $t_{2g}$-O 2p), but less possibilities for $\sigma$-type (Fe - O 2p) bonding. These ligand configuration effects combine to give a smaller effective ionic radius for the low spin state of Fe²⁺. In the case of FeO, the XES result can be interpreted as the spin transition at Fe²⁺ in the octahedral site, while the two Fe³⁺ remain in the high spin state.

The ionic radii reported by Shannon (1976) gives the radii for Fe²⁺ as 0.780 Å at HS and 0.61 Å at LS. The spin transition therefore reduces the ferrous ion radius by about 20 %. The Fe²⁺-O bond distance is likewise reduced from A change in the effective ionic radius brings about a polyhedral distortion. While the Si-O bond is not easily compressed, the Fe-O bond length is reduced dramatically under compression. The observed decrease in the lattice parameter leads to the corresponding change in octahedral volume. The volume ratio of the octahedron shows a distinct change due to the compression behavior of Fe-O bond distance at 20 GPa. These discontinuous changes are induces by the spin transition, starting at 17 GPa. However, the volume of the tetrahedron does not show a noticeable change in contrast to the octahedra. Fe² ion changes the spin state to an intermediate spin, resulting in its greater distortion of FeO₆ and smaller bond lengths, but Fe³ ion located at a larger and less distorted site probably remains in the HS state even at 64 GPa, (Table 4).

**Implication of this paper**

Numerous investigations of the structure transitions and decompositions of spinels have been executed under extreme conditions. Silicate spinels with transition elements or mixed-charge cations have been intensively studied from various viewpoints such as their magnetic susceptibility, electric conductivity, or elastic property.

The olivine-spinel transformation has been strongly studied for significance at the transition zone of the
earth’s mantle. Fe$_2$SiO$_4$ has an olivine structure (α-Fe$_2$SiO$_4$, fayalite) at ambient conditions and transforms directly into spinel (γ-Fe$_2$SiO$_4$, ahrensite) under high pressure at ambient temperature. Silicate spinels have been known as an essentially metastable phase at ambient conditions.

There are several passes of spinel structure transformations. Some oxide spinels with transition elements have their high-pressure polymorphs due to the transition without decomposition. A pseudo-rhombohedral phase at pressures above 30 GPa and ambient temperature was reported on the basis of powder diffraction data and Mössbauer data (Greenberg et al., 2011). However, structure parameters including atomic positional parameters of the high-pressure phase are not reported in their paper and their Rietveld analysis at 48(2) GPa did not consider the residual spinel phase in the refinement. Hence, several peaks are not clearly fitted.

Present paper shows a structural change in Fe$_2$SiO$_4$ spinel under high pressure up to 64 GPa. A new high-pressure structure of I-Fe$_2$SiO$_4$ is determined by Rietveld profile fitting of synchrotron x-ray diffraction data at ambient temperature. A transition from the cubic spinel structure to a body centered orthorhombic phase (I-Fe$_2$SiO$_4$) with space group *Imma* and Z=4 was first observed at approximately 39 GPa. The structure of I-Fe$_2$SiO$_4$ has two crystallographically distinct FeO$_6$ octahedral sites and Si atom changes its configuration.

In addition to the new structure, we first found two different compression curves of the lattice parameter in the spinel phase and the discontinuity at approximately 20 GPa. Fe-$K\beta$ x-ray emission measurements at elevating pressure show that the transition from high-spin (HS) state to intermediate-spin (IS) state begins at 17 GPa in the spinel phase. The IS electronic state is gradually enhanced with pressure, which generates an isostructural change in the lattice parameter at 20 GPa. The spin transition induces the compression of the bond length, resulting in the structure transition at 39 GPa. The spin transition can be emphasized for the trigger of the structure transition.

**Acknowledgments**

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References


Journal Geophysical Research, B96, 2157–2169.

Pressure-induced structural phase transition of the iron end-member of ringwoodite (γ-Fe₂SiO₄) investigated by x-ray diffraction and Mössbauer spectroscopy. Science, 279, 1698–1700.


Li, L., Cares, P., and Weidner, D.J. (2007) Effect of cation ordering and pressure on spinel elasticity by ab initio simulation.

Lin, J.F., Struzhkin, V.V., Jacobsen, S.D., Shen, G., Spin transition of iron in magnesiowüstite in the Earth’s lower mantle.


Table captions

Table 1.

Lattice parameters were determined by Rietveld profile fitting analysis. Spinel phase is found at pressures above the transition pressure of 38.8 GPa and coexists with the high-pressure phase of I-Fe₂SiO₄.

Table 2.

Table 3.

Reliability factors for the least-squares calculation are

\[
\begin{align*}
\text{wR}_p &= \left[ \frac{\sum_i w_i |y_i - f_i(x)|}{\sum_i w_i y_i} \right]^{1/2}, \\
\text{R}_p &= \left[ \frac{\sum_i |y_i - f_i(x)|}{\sum_i y_i} \right], \\
\text{R}_F &= \left[ \frac{\sum_k |F_{k, \text{obs}} - F_{k, \text{calc}}|}{\sum_k F_{k, \text{obs}}} \right], \\
\text{S} &= \left[ \frac{\sum_i w_i |y_i - f_i(x)|}{N - P} \right]^{1/2}
\end{align*}
\]

Table 4.
Figure captions

Figure 1. Selected x-ray diffraction patterns of Fe$_2$SiO$_4$ taken with increasing pressure at ambient temperature.

The high-pressure phase was found above 39.2 GPa and observed up to 56.6 GPa.

Figure 2. There are two distinct compression regimes indicating the boundary at about 20 GPa. This behavior is a little higher pressure than the HS-to-IS transition of 17 GPa observed by XES. The data of Greenberg et al. (2011) are also presented. The dotted curve represents the lattice parameter of the residual spinel coexist with I-Fe$_2$SiO$_4$. The observed error is smaller than the data point.

Figure 3. Average bond length ratio of tetrahedral and octahedral sites of Fe$_2$SiO$_4$ spinel. Bond lengths were obtained from the Rietveld profile fitting method. Error is smaller than the symbol mark.

The Fe-O bond length in the octahedral site confirms the two different compression regimes at 20 GPa.

Figure 4. The decompression experiment confirms a reversible transition between cubic and orthorhombic phases. Subsequent laser heating at 1500 K and 31 GPa reveals decomposition from spinel to rhombohedral FeO and SiO$_2$ stishovite. Reflections with symbols of I and in the figure represent I-Fe$_2$SiO$_4$ and spinel phase, respectively.

Figure 5. Rietveld profile fitting of the initial structure model of rhombohedral Fe$_2$SiO$_4$ to the data obtained at 54.6 GPa. The rhombohedral structure model (space group $R\bar{3}m$) is simply derived from a distortion along the <111> direction of the spinel structure. Peaks indicated by the arrows are not indexed.

Figure 6. Rietveld profile fitting of diffraction data obtained at 54.6 GPa, assuming a two-phase mixture of orthorhombic Fe$_2$SiO$_4$ and spinel. Upper and lower vertical bars indicate peak positions for orthorhombic Fe$_2$SiO$_4$ and spinel, respectively.

Figure 7. Rietveld profile fitting of diffraction data obtained at 64 GPa. Rietveld profile fitting was carried out in consideration of two-phase mixture of I-Fe$_2$SiO$_4$. Vertical bars indicate peak positions of the I-Fe$_2$SiO$_4$.

Figure 8. There are two distinct octahedral Fe sites, Fe1 and Fe2. Si also is located at the site of octahedral coordination. Fe1 and Fe2 octahedra create layers parallel to (011) respectively. SiO$_6$ octahedra make an array in the direction of <100>. 
Figure 9. Fe$_2$SiO$_4$ spinel (ringwoodite ahrensite) and high-pressure phase $I$-Fe$_2$SiO$_4$ are presented in the left and right figure, respectively. Arrow indicates a martensitic transformation with translation vector $<1/8 \ 1/8 \ 1/8>$, which generates the $I$-Fe$_2$SiO$_4$ structure from the spinel. The shadowed circles in the left figure represent the atomic positions of $I$-Fe$_2$SiO$_4$.

Figure 10. Fe $K\beta$ x-ray emission spectra with increasing pressure up to 64.8 GPa. The upper left figure shows the expanded $K\beta$ spectra, indicating an intermediate spin transition. The spin transition occurs at approximately 17 GPa.
Fig. 2
54.6 GPa
$R\bar{3}m$ model fitting

Intensity

2000
1500
1000
500
0

2θ  ($\lambda = 0.4262$ Å)

Fig. 5
Fig. 6

Intensity

54.6 GPa
$I$-$Fe_2SiO_4 + Spinel$

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2θ (λ = 0.4262 Å)
Fig. 7

64 GPa
$I$-Fe$_2$SiO$_4$

Intensity

2\(\theta\) (\(\lambda = 0.4262\) Å)

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Always consult and cite the final, published document. See http://www.minsocam.org or GeoscienceWorld
Fe$_2$SiO$_4$ spinel

$Fd3m$  $z = 8$

High-pressure $I$-Fe$_2$SiO$_4$

$Imma$  $z = 4$

Fe  Si

translation vector is

$1/8$  $1/8$  $1/8$

Fig. 9
Fig. 10
Table 1  Lattice constant of Fe$_2$SiO$_4$ spinel

<table>
<thead>
<tr>
<th>P (GPa)</th>
<th>LC (Å)</th>
<th>P (GPa)</th>
<th>LC (Å)</th>
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<td></td>
<td>two-phase mixture metastable spinel</td>
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<td>3.7</td>
<td>8.1783 (9)</td>
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<td>4.0</td>
<td>8.1759 (8)</td>
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<td>7.8</td>
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<td>9.1</td>
<td>8.1140 (8)</td>
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<tr>
<td>11.5</td>
<td>8.0905 (8)</td>
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<tr>
<td>13.3</td>
<td>8.0759 (9)</td>
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<td>15.4</td>
<td>8.057 (1)</td>
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<tr>
<td>17.3</td>
<td>8.044 (1)</td>
<td></td>
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<tr>
<td>18.8</td>
<td>8.028 (1)</td>
<td></td>
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<tr>
<td>20.3</td>
<td>8.013 (1)</td>
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<td>22.3</td>
<td>7.996 (1)</td>
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<td>23.1</td>
<td>7.990 (1)</td>
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<td>24.2</td>
<td>7.976 (2)</td>
<td></td>
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<tr>
<td>27.2</td>
<td>7.951 (2)</td>
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<td>31.0</td>
<td>7.916 (2)</td>
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<tr>
<td>32.9</td>
<td>7.895 (2)</td>
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<tr>
<td>34.8</td>
<td>7.889 (3)</td>
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</table>
Table 2  Bond distance and site volume of spinel phase under pressure

<table>
<thead>
<tr>
<th>Press GPa</th>
<th>Cell const Å</th>
<th>Cell vol Å³</th>
<th>u-parameter</th>
<th>d(Si-O) Å</th>
<th>d(Fe-O) Å</th>
<th>Vol(SiO4) Å³</th>
<th>Vol(FeO6) Å³</th>
</tr>
</thead>
<tbody>
<tr>
<td>0.0001</td>
<td>8.2374(4)</td>
<td>558.95(6)</td>
<td>0.3659(2)</td>
<td>1.654(7)</td>
<td>2.137(7)</td>
<td>2.321(2)</td>
<td>12.914(2)</td>
</tr>
<tr>
<td>1.2</td>
<td>8.2156(8)</td>
<td>554.52(7)</td>
<td>0.3661(2)</td>
<td>1.652(6)</td>
<td>2.130(6)</td>
<td>2.314(4)</td>
<td>12.784(3)</td>
</tr>
<tr>
<td>4.0</td>
<td>8.1759(8)</td>
<td>546.46(8)</td>
<td>0.3666(2)</td>
<td>1.651(6)</td>
<td>2.115(6)</td>
<td>2.310(2)</td>
<td>12.530(2)</td>
</tr>
<tr>
<td>9.1</td>
<td>8.1142(8)</td>
<td>533.81(7)</td>
<td>0.3670(2)</td>
<td>1.644(7)</td>
<td>2.096(6)</td>
<td>2.282(3)</td>
<td>12.197(3)</td>
</tr>
<tr>
<td>13.3</td>
<td>8.0759(5)</td>
<td>526.71(5)</td>
<td>0.3673(8)</td>
<td>1.641(5)</td>
<td>2.083(5)</td>
<td>2.267(1)</td>
<td>11.986(1)</td>
</tr>
<tr>
<td>17.3</td>
<td>8.044(1)</td>
<td>520.5(2)</td>
<td>0.3678(9)</td>
<td>1.641(17)</td>
<td>2.071(17)</td>
<td>2.269(9)</td>
<td>11.779(9)</td>
</tr>
<tr>
<td>23.1</td>
<td>7.990(1)</td>
<td>510.0(2)</td>
<td>0.3692(11)</td>
<td>1.650(18)</td>
<td>2.045(18)</td>
<td>2.304(12)</td>
<td>11.368(13)</td>
</tr>
<tr>
<td>32.9</td>
<td>7.895(3)</td>
<td>492.1(1)</td>
<td>0.3700(11)</td>
<td>1.641(13)</td>
<td>2.014(13)</td>
<td>2.268(12)</td>
<td>10.868(13)</td>
</tr>
<tr>
<td>34.8</td>
<td>7.889(3)</td>
<td>490.9(1)</td>
<td>0.3702(13)</td>
<td>1.642(12)</td>
<td>2.011(12)</td>
<td>2.274(22)</td>
<td>10.818(23)</td>
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</tbody>
</table>

metastable spinel above the transition pressure

<table>
<thead>
<tr>
<th>Press GPa</th>
<th>Cell const Å</th>
<th>Cell vol Å³</th>
<th>u-parameter</th>
<th>d(Si-O) Å</th>
<th>d(Fe-O) Å</th>
<th>Vol(SiO4) Å³</th>
<th>Vol(FeO6) Å³</th>
</tr>
</thead>
<tbody>
<tr>
<td>38.8</td>
<td>7.879(4)</td>
<td>489.1(1)</td>
<td>0.3705(25)</td>
<td>1.644(17)</td>
<td>2.006(17)</td>
<td>2.272(23)</td>
<td>10.74(23)</td>
</tr>
<tr>
<td>39.2</td>
<td>7.855(4)</td>
<td>484.7(1)</td>
<td>0.3711(22)</td>
<td>1.648(16)</td>
<td>1.995(16)</td>
<td>2.295(23)</td>
<td>10.57(23)</td>
</tr>
<tr>
<td>44.6</td>
<td>7.756(2)</td>
<td>466.6(3)</td>
<td>0.3707(15)</td>
<td>1.622(12)</td>
<td>1.973(12)</td>
<td>2.188(16)</td>
<td>10.22(16)</td>
</tr>
<tr>
<td>54.6</td>
<td>7.730(2)</td>
<td>461.9(2)</td>
<td>0.3708(11)</td>
<td>1.617(10)</td>
<td>1.966(10)</td>
<td>2.171(14)</td>
<td>10.11(14)</td>
</tr>
</tbody>
</table>
Table 3  Structure parameters of high-pressure phase of $IFe_2SiO_4$

<table>
<thead>
<tr>
<th>Pressure</th>
<th>44.6GPa*</th>
<th>54.6GPa*</th>
<th>64.8GPa</th>
</tr>
</thead>
<tbody>
<tr>
<td>Space group</td>
<td>Imma</td>
<td>Imma</td>
<td>Imma</td>
</tr>
<tr>
<td>Unit molecule</td>
<td>4</td>
<td>4</td>
<td>4</td>
</tr>
<tr>
<td>a (Å)</td>
<td>5.551(3)</td>
<td>5.543(1)</td>
<td>5.522(3)</td>
</tr>
<tr>
<td>b (Å)</td>
<td>6.030(4)</td>
<td>6.032(2)</td>
<td>6.025(4)</td>
</tr>
<tr>
<td>c (Å)</td>
<td>7.241(5)</td>
<td>7.201(4)</td>
<td>7.185(5)</td>
</tr>
<tr>
<td>volume (Å³)</td>
<td>242.4(4)</td>
<td>240.8(2)</td>
<td>239.0(2)</td>
</tr>
<tr>
<td>Rwp</td>
<td>2.996</td>
<td>2.531</td>
<td>2.369</td>
</tr>
<tr>
<td>R_B</td>
<td>1.993</td>
<td>1.354</td>
<td>7.573</td>
</tr>
<tr>
<td>R_F</td>
<td>0.985</td>
<td>1.042</td>
<td>4.088</td>
</tr>
<tr>
<td>s</td>
<td>0.4047</td>
<td>0.3436</td>
<td>1.8538</td>
</tr>
</tbody>
</table>

Si  4a  x  0.0  0.0  0.0  
y  0.0  0.0  0.0  
z  0.0  0.0  0.0  
Biso  4.83(2)  4.87(2)  2.05(2)  
Fe1  4b  x  0.0  0.0  0.0  
y  0.0  0.0  0.0  
z  0.5  0.5  0.5  
Biso  6.74(1)  3.54(1)  2.90(1)  
Fe2  4c  x  0.25  0.25  0.25  
y  0.25  0.25  0.25  
z  0.25  0.25  0.25  
Biso  4.92(1)  3.88(3)  2.92(5)  
O1  8h  x  0.0  0.0  0.0  
y  0.5101(9)  0.5073(5)  0.5068(9)  
z  0.7652(9)  0.7608(8)  0.7610(8)  
Biso  3.6531(1)  4.00(1)  3.72(2)  
O2  8i  x  0.2371(9)  0.2342(8)  0.2333(2)  
y  0.25  0.25  0.25  
z  0.0022(9)  0.0014(9)  0.0012(6)  
Biso  5.91(2)  6.24(5)  5.54(5)  

Note: Values in parentheses represent standard deviations.
Table 4  Deformation of Si, Fe1 and Fe2 octahedra of $I\text{Fe}_2\text{SiO}_4$

<table>
<thead>
<tr>
<th>Pressure</th>
<th>44.6GPa</th>
<th>54.6GPa</th>
<th>64.8GPa</th>
</tr>
</thead>
<tbody>
<tr>
<td>Si</td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>O1 x2 (Å)</td>
<td>1.701(4)</td>
<td>1.723(4)</td>
<td>1.718(4)</td>
</tr>
<tr>
<td>O2 x4 (Å)</td>
<td>2.001(4)</td>
<td>1.990(4)</td>
<td>1.982(4)</td>
</tr>
<tr>
<td>mean (Å)</td>
<td>1.901(4)</td>
<td>1.901(4)</td>
<td>1.894(4)</td>
</tr>
<tr>
<td>volume (Å³)</td>
<td>8.99(6)</td>
<td>8.99(6)</td>
<td>8.89(4)</td>
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<tr>
<td>Fe1</td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>O1 x2 (Å)</td>
<td>1.921(5)</td>
<td>1.879(3)</td>
<td>1.876(3)</td>
</tr>
<tr>
<td>O2 x4 (Å)</td>
<td>2.098(5)</td>
<td>2.108(3)</td>
<td>2.107(3)</td>
</tr>
<tr>
<td>mean (Å)</td>
<td>2.039(5)</td>
<td>2.032(3)</td>
<td>2.030(3)</td>
</tr>
<tr>
<td>volume (Å³)</td>
<td>11.26(6)</td>
<td>11.13(5)</td>
<td>11.09(5)</td>
</tr>
<tr>
<td>Fe2</td>
<td></td>
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</tr>
<tr>
<td>O1 x4 (Å)</td>
<td>2.008(6)</td>
<td>2.017(6)</td>
<td>2.015(4)</td>
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<tr>
<td>O2 x2 (Å)</td>
<td>1.796(6)</td>
<td>1.792(6)</td>
<td>1.790(4)</td>
</tr>
<tr>
<td>mean (Å)</td>
<td>1.937(6)</td>
<td>1.942(8)</td>
<td>1.940(4)</td>
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<tr>
<td>volume (Å³)</td>
<td>9.58(4)</td>
<td>9.66(2)</td>
<td>9.61(2)</td>
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