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Revision 1

Effect of crystal defects on diamond morphology during dissolution in the mantle

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ABSTRACT

The influence of three-dimensional defects on the morphology of diamond dissolution in water-containing carbonate melts was studied at a pressure of 5.7 GPa and a temperature of 1300°C, using a BARS multi-anvil apparatus. Experiments on stage-by-stage dissolution were performed for four blocky synthetic diamond crystals. Initial crystals had polycentric structure of the faces, strong strains, block structure, numerous microtwins and microinclusions. It has been established that the main relief features of partly dissolved diamond crystals are shield-shaped laminae and negative trigons on remnants of {111} faces, deep etch channels, rectilinear steps on microtwins and hillocks on rounded surfaces. The produced dissolution forms have shagreen or block-type rounded surfaces. The main element of the relief are hillocks. Their shape is controlled by the orientation of the surface on which they are localized. For natural rounded diamonds it is found that the dissolution drop-like hillocks on the surfaces are also related to strong lattice strains occurring in the crystals. The established relation between the dissolution

27 hillocks and diamond deformation can be used for the reconstruction of the post-growth history
28 of natural diamonds as well as for the preliminary evaluation of their quality.

29 **Keywords:** Diamond dissolution, crystal morphology, defect of crystals, dissolution
30 hillocks.

31

32 INTRODUCTION

33

34 Natural rounded diamonds demonstrate a wide variety of surface structures. Some crystals
35 have macroscopically smooth, glossy surfaces. Others show on their surfaces specific features,
36 including serrate and shield-shaped laminae, drop-like hillocks, striations, ruts and many other
37 patterns (Kukhareenko 1955; Orlov 1977; Robinson 1978). There is no doubt that the rounded
38 form of natural diamond crystals is a result of dissolution in water-containing kimberlitic melt
39 (Moore and Lang 1974; Orlov 1977; Robinson 1978; Gurney et al. 2004; Kozai and Arima 2005;
40 Khokhryakov and Palyanov 1990, 2007; Fedortchouk et al. 2007). Therefore the details of the
41 micro-relief of natural diamond crystals reflect first of all the conditions of dissolution in natural
42 processes. At the same time, diamond characteristically contains various structural imperfections
43 ranging from point to three-dimensional defects. The dissolution at the defects outcrops proceeds
44 in a different way than for perfect surfaces (Heimann 1975). As a result, the micro-relief
45 elements may form, which to a greater extent reflect the features of the real structure of diamond
46 crystals rather than the dissolution conditions. This question has not been adequately explored so
47 far. Only for some micro-relief elements of natural rounded diamonds, their relation to the real
48 structure has been determined. It is established that trigons on {111} faces are located at the
49 dislocation outcrops (Lang 1964). It is shown that the intersecting striation is related to the bands
50 of plastic deformations (Robinson 1978) and rhombic patterns of the etching channels to the
51 surface cracks (Orlov 1977). The nature of many other widely abundant micro-relief elements
52 and surface sculptures has not been clarified yet. The scarcity of experimental data on this

53 subject is partly due to the fact that in most cases only high-quality diamond crystals were used
54 in the dissolution experiments (Kozai and Arima 2005; Khokhryakov and Palyanov 2007;
55 Fedortchouk et al. 2007). Furthermore, most of previous experimental works on diamond
56 dissolution were focused mainly on the effect of conditions and composition of the medium on
57 dissolution morphology.

58 In the present study we have explored the dissolution of synthetic diamonds of block
59 structure. The specific features of dissolution surfaces of mosaic crystals are established. It is
60 shown that the origin of the drop-like hillocks on diamond tetrahedra is related to the
61 occurrence of strong local strains in the diamond lattice.

62

63

EXPERIMENTAL

64

65 In the experiments we studied stage-by-stage dissolution of four blocky crystals of
66 synthetic diamond weighting from 1.2 to 5.92 mg (Fig. 1). The diamond crystals were grown by
67 the temperature gradient method in the Ni₇Fe₃-C system with nitrogen content from 0.25 to 0.3
68 at.% (Palyanov et al. 2010). A detailed investigation of these crystals was performed in
69 (Palyanov et al. 2010). It was shown that the crystals had a polycentric structure of the faces,
70 strong strains, block structure, numerous microtwins and microinclusions. Such characteristics
71 are typical for many natural diamond crystals. Crystal No. 1 (5.92 mg) has small {100} faces
72 (Fig. 1a). Approximately one third of the crystal has flat faces, and the rest surface is blocky with
73 numerous microtwins. Crystal No. 2 (2.47 mg) has only {111} faces (Fig. 1d). The surface
74 exhibits large blocks and twinning intergrowths. Crystals No. 3 (1.51 mg) and No. 4 (1.20 mg)
75 are of octahedral habit. These crystals have a rough relief with numerous trigonal blocks and
76 depressions (Fig. 1g,j). All crystals contained micro inclusions of metal catalyst. The amount of
77 micro inclusions in the crystals increases with the degree of their blockiness from crystal No. 1
78 to crystal No. 4. From the infrared absorption measurements it was found that synthetic diamond

79 crystals contained approximately 120-130 ppm of nitrogen impurity predominantly in the form
80 of single substitutional atoms, C-centers.

81 Experiments on diamond dissolution were carried out using a “split-sphere” type multi-
82 anvil apparatus at 5.7 GPa and 1300°C in water-containing carbonate melts. Details on the high-
83 pressure cell design, pressure and temperature measurements and calibration were presented in
84 our previous works (Khokhryakov and Palyanov 2007, 2010). Starting materials for each
85 experiment were 99.9% pure CaCO₃ (700 mg) with addition of H₂O (78 mg). Thus the water
86 content in all experiments was 10wt.%. Starting materials and four diamond crystals were placed
87 into sealed Pt capsules (6 mm in diameter, 5 mm long, and 0.2 mm thick walls). All four crystals
88 were used in two experiments with duration of 30 and 45 hours.

89 Several rounded natural diamonds from Udachnaya and Yubileinaya kimberlite pipes were
90 studied to reveal possible effects of lattice imperfections on the surface morphology. The
91 selected crystals were 1.0-1.5 mm in size and exhibited nonuniform surface structure. Cross
92 section plates 0.5-0.7 mm thick were prepared from three crystals by mechanical polishing. The
93 plates were oriented so that areas with different types of surface relief were exposed.

94 The crystals before and after each experiment were weighed (± 0.01 mg) and analyzed with
95 optical and scanning electron microscopes. The micromorphology of diamond crystals was
96 studied using a LEO420 scanning electron microscope. Internal structure of diamond plates in
97 polarized light was studied using a Zeiss AXIO Imager.Z2m” microscope. Infrared absorption
98 spectra were measured using a Bruker Vertex 70 Fourier Transform Infrared (FTIR)
99 spectrometer equipped with a Hyperion 2000 microscope.

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RESULTS AND DISCUSSION

102

103 The overall view of the crystals and their weights before and after experiments are
104 presented in Fig.1. The crystals dissolved to different degrees. The mass loss (LM) of crystal No.

105 1, which was the largest one, was only 19 %, whereas smaller and more defective crystals No. 3
106 and No. 4 lost 58-59 % of their initial mass.

107 The main relief features of partly dissolved diamond crystals are shield-shaped laminae and
108 negative trigons on the remnant {111} faces; deep etch channels, rectilinear steps on microtwins
109 and hillocks on rounded surfaces. The number and size of the negative trigons differ for the
110 studied diamond crystals. A great number of trigons are present on the crystals which initially
111 had numerous trigonal depressions (crystals No. 3 and 4). Trigons on these crystals did not
112 disappear even when the weight loss was 58-59%. This apparently is connected with a large
113 number of microinclusions in crystals No. 3 and 4.

114 Ditrigonal or shield-shaped dissolution laminae were present on the crystals only after the
115 first experiment (LM from 6.4 to 34%). These laminae had rough, wavy edges (Fig. 2a) in
116 contrast to the dissolution laminae on high-quality diamond crystals. As it was shown previously
117 (Khokhryakov and Palyanov 2007, 2010) the ditrigonal laminae formed on the (111) faces of
118 high-quality crystals had smooth rectilinear edges (Fig. 2b). The presence of microtwins and
119 large trigonal depressions and cracks in the mosaic crystals resulted in polycentric lamination.

120 Typical elements of blocky crystals after dissolutions are the etching channels. The number
121 and size of the channels increase with decreasing quality of the crystals from No. 1 to No. 4
122 (Fig.1). The etch channels form, most likely, along the cracks and boundaries between the crystal
123 blocks. Numerous microtwins appear after dissolution as single rectilinear steps or blocks of
124 polysynthetic twinning (Fig. 3a).

125 The produced dissolution forms have shagreen or block-type rounded surfaces. The main
126 elements of the relief are hillocks. Their shape is controlled by the orientation of the surface on
127 which they are localized (Fig. 3). On the surfaces close to the {100} faces they have isometric
128 irregular shape (Fig. 3b). Surfaces close in orientation to {111} are covered with flat hillocks
129 forming a regular structure similar to the sculpture of roof tiles (Fig. 3c). Hillocks formed on the
130 surfaces close to the {110} faces have an elongated drop-like shape (Fig. 3d).

131 It is known that hillocks are one of the widespread relief elements of tetrahexahedroids of
132 natural diamond. Depending on the size, shape and degree of elongation of hillocks, their
133 accumulations form various types of microrelief of rounded natural diamonds. These include
134 block-type, drop-like, serrate and shagreen features (Orlov 1977). The origin of hillocks is still
135 debatable. In earlier works considering the origin of rounded diamonds within the "growth"
136 hypothesis, these sculptures were treated as vicinal growth hillocks and their growth occurred by
137 the (110) layers (Evdokimova and Shemanin 1962, Shemanina and Shemanin 1964). Orlov
138 (Orlov 1977) suggested that hillocks formed as a result of partial dissolution of polyhedra
139 initially present on the surface of the diamond. In other studies it was supposed that the
140 formation of hillocks took place at high intensity and duration of the dissolution of diamonds
141 (McCandless et al. 1999) or when diamond resorption occurs in the CO₂ fluid (Fedortchouk et al.
142 2007).

143 The diamond crystals used in present study had strong stresses and possibly this was the
144 reason for the formation of dissolution hillocks as the main element of the dissolution
145 microrelief. For natural diamonds there is a number of observations indicating that hillocks are
146 more typical for crystals with high level of internal strains. Orlov (1977) noted that hillocks are
147 often located along the intersecting striation associated with the bands of plastic deformation in
148 diamond. They are also very characteristic of natural diamond crystals of type II, which often
149 exhibit anomalous birefringence of "tatami" type (Orlov 1977). Kaminsky et al. (2000) noted
150 that dissolution of natural diamond crystals with plastic deformation produces sculptures
151 consisting of hillocks appearing as shagreen, hackly, drop-like, grooved and block-type features.
152 Recently, Gainutdinov et al. (2013) also suggested possible association of the dissolution
153 hillocks in diamond with post-growth deformation events. In addition it has been demonstrated
154 experimentally (Heiman 1975, Sangwal 1990) that dissolution hillocks on rounded surfaces of
155 crystals form not only as a result of blocking of individual sections by impurities (for example,

156 MgO and KCl crystals), but also due to the presence of strong gradients of stresses in crystals
157 (for example, Ag crystals).

158 To the best of our knowledge, there are no available literature data correlating the location
159 of drop-like hillocks on the rounded surfaces of natural diamond crystals with the internal
160 structure of the crystals. In order to clarify whether such a correlation exists we examined several
161 natural diamond crystals with rounded surfaces exhibiting drop-like hillocks. The most
162 illustrative of these crystals is shown in Fig. 4. For this crystal the boundary between the areas
163 with different surface relief is the most sharp. Most of the crystal surface is smooth and glossy
164 (area B in Fig. 4) and small fragments of the surface are composed of accumulations of small
165 drop-like hillocks (area A in Fig. 4). The (110) cross section plates were cut from the crystals
166 through the surface areas with drop-like hillocks, as shown in Fig. 4a. Using polarized light
167 microscopy it was established that in the volume of the crystals there are distinct strongly
168 strained regions whose borders coincide with the borders of distribution of hillocks (Fig. 4c,d).
169 The birefringence patterns within these strained regions has a specific “tatami” form, which is
170 attributed to plastic deformation (Lang 1967; Orlov 1977). To examine if other defects such as
171 nitrogen impurity may be responsible for the observed surface morphology we measured infrared
172 absorption spectra from several local areas on the sample. Figure 5 shows IR spectra recorded
173 from the selected locations as indicated in the insert. It follows that the periphery area which
174 exhibit high strains and which is bound by surfaces with hillocks contain relatively low nitrogen
175 concentrations of approximately 100-120 ppm (spectrum 1 in Fig. 5). In this case, 90-95 ppm of
176 nitrogen is in the form nitrogen pairs, A-centers and 15-20 ppm in the form of aggregates of four
177 nitrogen atoms, B-centers. Moving from the periphery to the center the concentration of nitrogen
178 significantly increases to approximately 580-600 ppm (spectrum 3 in Fig. 5). From this we can
179 infer that nitrogen defects itself can not be responsible for the observed dissolution hillocks. On
180 the other hand our results are in agreement with previous observations that diamond crystals or

181 local areas within the crystals with low nitrogen concentrations deform plastically more easily
182 than crystals or areas with high nitrogen concentrations (e.g. Nailer et al., 2007).

183 The established correlation between the dissolution hillocks and defects due to lattice
184 strains in diamond is preliminary and needs further verification over a greater amount of crystals
185 with detailed investigations of surface micro-relief and internal structure with quantitative
186 measurements of lattice strains. Nevertheless, the results obtained in this study allow us to
187 suggest that the formation of dissolution hillocks occurs on strained natural diamonds.

188 The shape of hillocks is worth special consideration. In spite of the different external forms
189 of hillocks, they have the same crystallography. Goniometric studies of hillocks on natural
190 diamonds were performed in the middle of the last century (Fersman 1955; Kukharenko 1955).
191 These data allow an accurate description of the dissolution hillocks. Fersman (1955) found that
192 rounded surfaces with drop-like hillocks always gave reflection on the goniometer in the form of
193 arcuate bands. On the stereographic projection the bands of reflections from the rounded surface
194 of hillocks have the shape of an ellipse, designated by Fersman as band CE (Fig. 6). Such a
195 shape of light reflexes corresponds to a conical surface with an elliptic cross section. It is known
196 that conical surfaces are very characteristic forms of dissolution of crystals of various minerals
197 and diamond is not an exception (Shafranovsky 1961; Glazov 1981). Also goniometric study of
198 hillocks with flat faces was conducted by A.A. Kukharenko (Kukharenko 1955). He measured
199 coordinates of faces which bound relatively large (up to 60 μm) conical hillock. The most
200 prominent light reflexes from the surfaces of these hillocks corresponded to hexaoctahedron and
201 tetrahedron series of faces. As shown in Fig. 6, the reflections from the flat areas of hillocks
202 are located on bands CE. Thus, rounded and faceted dissolution hillocks have similar conical
203 shape. The center of ellipses (i.e. the spherical projection of the axes of cones) of hillocks is
204 calculated as the middle point between the reflections $i(304)$ and $e(102)$ located on the symmetry
205 line. The calculation showed that it corresponds to the normal to plane (508) $\rho=32^\circ 00'$. Thus, the
206 axis of the cone-shaped surface hillocks or hillocks vertexes are oriented along the $\langle 850 \rangle$

207 directions. As is known the vertexes of the dissolution forms correspond to directions of minimal
208 dissolution rate (eg Heiman 1975). Thus, the <850> direction is the direction of minimal
209 dissolution rate of rounded surfaces of natural diamond tetrahexahedroids.

210

211

IMPLICATIONS

212

213 The results obtained in this study allow us to conclude that the hillocks on the rounded
214 surfaces of diamond do not reflect any specific features of the dissolution conditions, but indicate
215 the occurrence of strong deformations in the crystals. The drop-like shape of hillocks
216 corresponds to the conical surface with an elliptical cross section and the axis along <850>
217 directions. Thus, our results demonstrate that the morphologic features of rounded natural
218 diamonds do not indicate exclusively the composition of the dissolution media and dissolution
219 conditions. Sculptures on the diamond surface can also be related to the specific features of the
220 real structure of the diamond crystals. The established relation between the dissolution hillocks
221 and diamond deformation can be used for the reconstruction of the post-growth history of natural
222 diamonds as well as for the preliminary evaluation of their quality.

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- 288

289

Figure captions

290

291 **Figure 1.** SEM images of initial and partly dissolved diamond crystals in water-bearing
292 carbonate melts after 30 and 75 h. LM = mass loss of crystals. Scaling bar corresponds to 200
293 μm .

294

295 **Figure 2.** A dissolution layer on the $\{111\}$ faces. (a) block diamond crystals of No. 3 after
296 dissolution for 30 h.; (b) high-quality diamond crystals. Figure 2f from the work (Khokhryakov
297 and Palyanov 2007).

298

299 **Figure 3.** SEM image of surface textures of partly dissolved diamond crystals. (a) Microtwin on
300 the crystal No 2 after 30 h of dissolution. Dissolution hillocks on the surfaces close to (b) $\{100\}$
301 (crystal No. 4, 30 h), (c) $\{111\}$ (crystal No. 1, 75 h) and (d) $\{110\}$ (crystal No 2, 75 h) faces.

302

303 **Figure 4.** A natural diamond crystal with drop-like hillocks. (a) General view of the crystal.
304 Black lines show the directions of cutting the crystal. (b) A fragment of the crystal surface. (c) A
305 plate cut from the crystal and (d) a fragment of the plate viewed with cross polarizers. **A** - crystal
306 surface without drop-like hillocks. **B** - crystal surface with drop-like hillocks.

307

308 **Figure 5.** Infrared absorption spectra recorded from a plate prepared from the natural diamond
309 crystal shown in Fig. 4. The insert shows a fragment of the plate. Squares mark the areas from
310 with spectra were taken and their sizes correspond to the sizes of the microscope aperture
311 (approximately $80 \times 80 \mu\text{m}$). Labels on the spectra corresponds to labels shown on the image.

312

313 **Figure 6.** A fragment of stereographic projection of hillocks on the surfaces of natural diamond
314 crystals. Continuous bands of reflexes (gray) are from (Fersman 1955). Single reflexes are given

315 according to (Kukharenko 1955): k - {702}, a - {301}, g - {502}, e - {201}, i - {403}, r - {961},
316 t - {951}, l - {742}, p - {532}, Ξ - {731}, z - {531}, x - {321}, y - {432}, w - {431}, Σ - {541},
317 Φ - {651}, S - {833}.

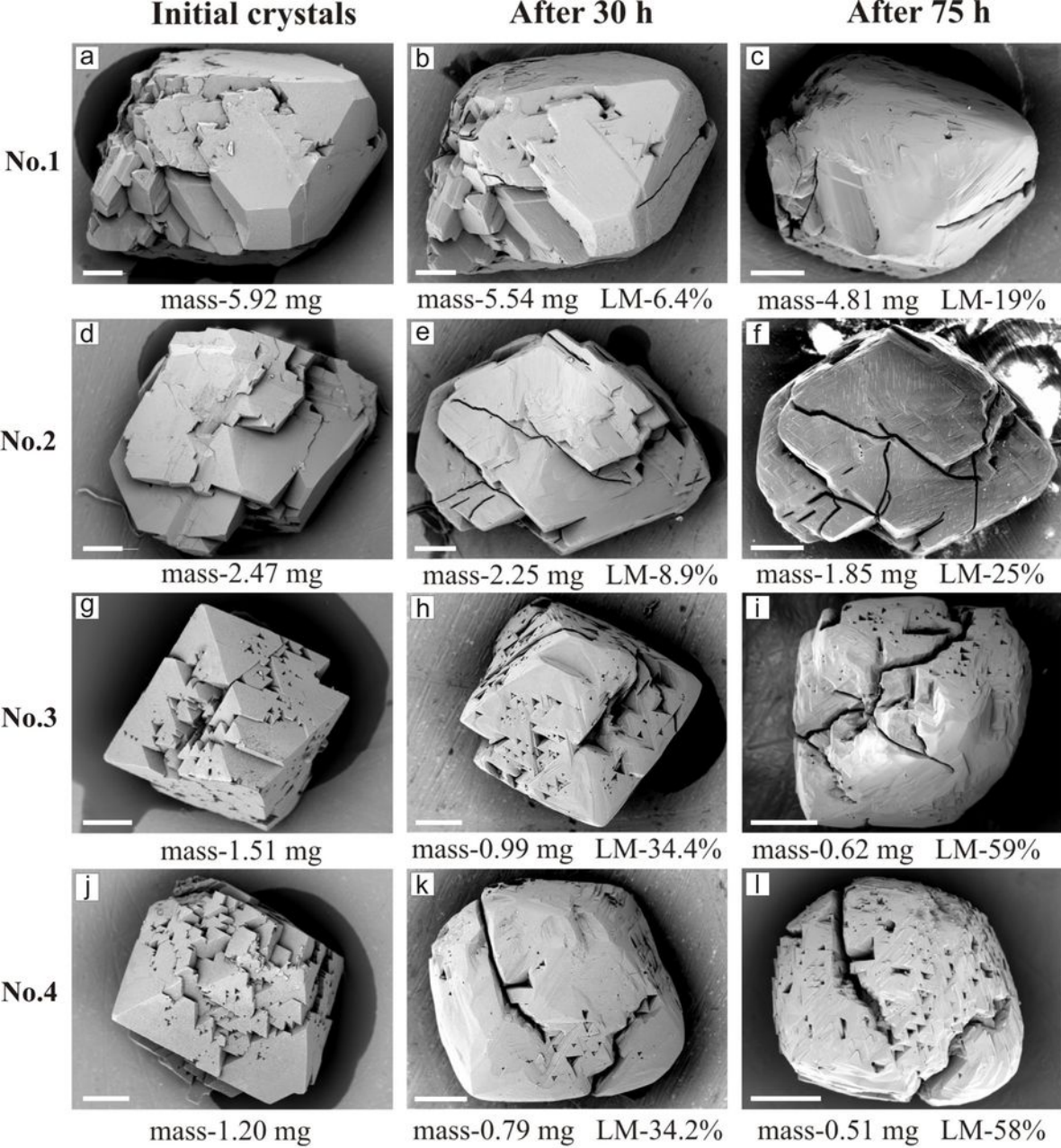


Figure 1.

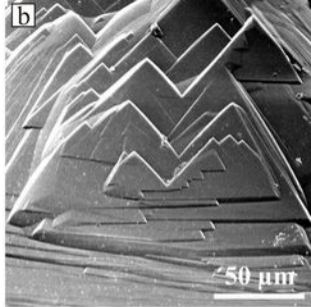
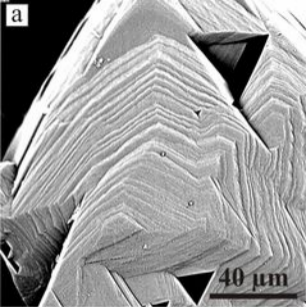


Figure 2.

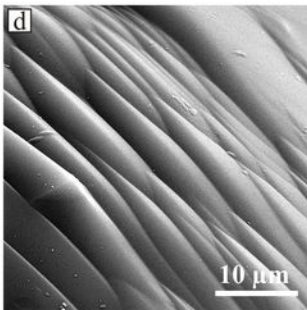
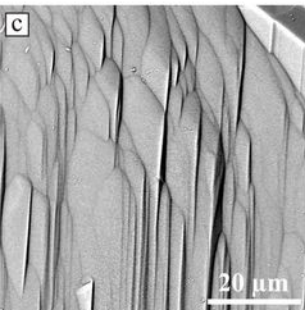
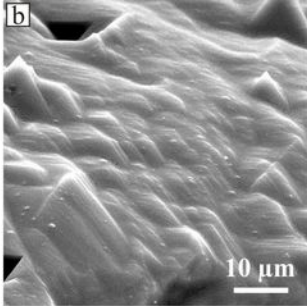
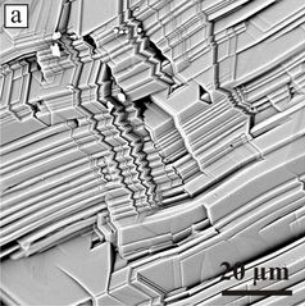


Figure 3.

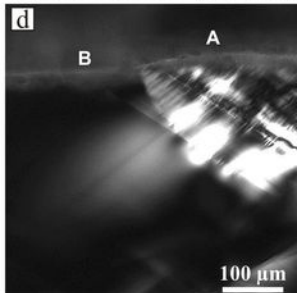
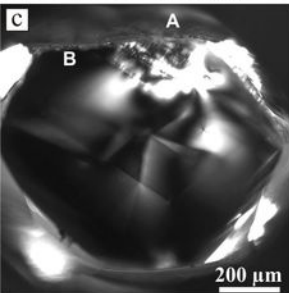
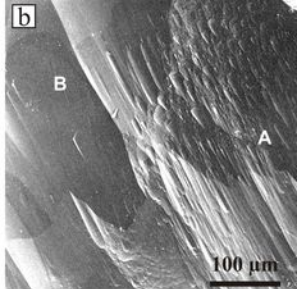
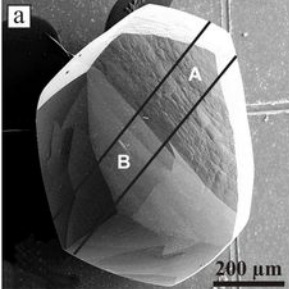


Figure 4.

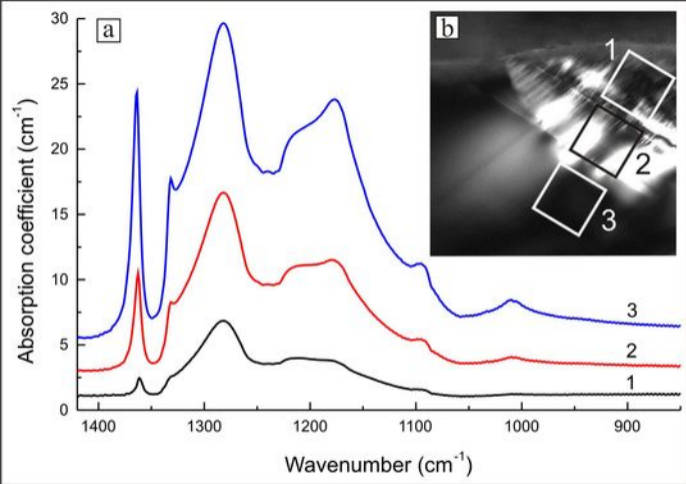


Figure 5.

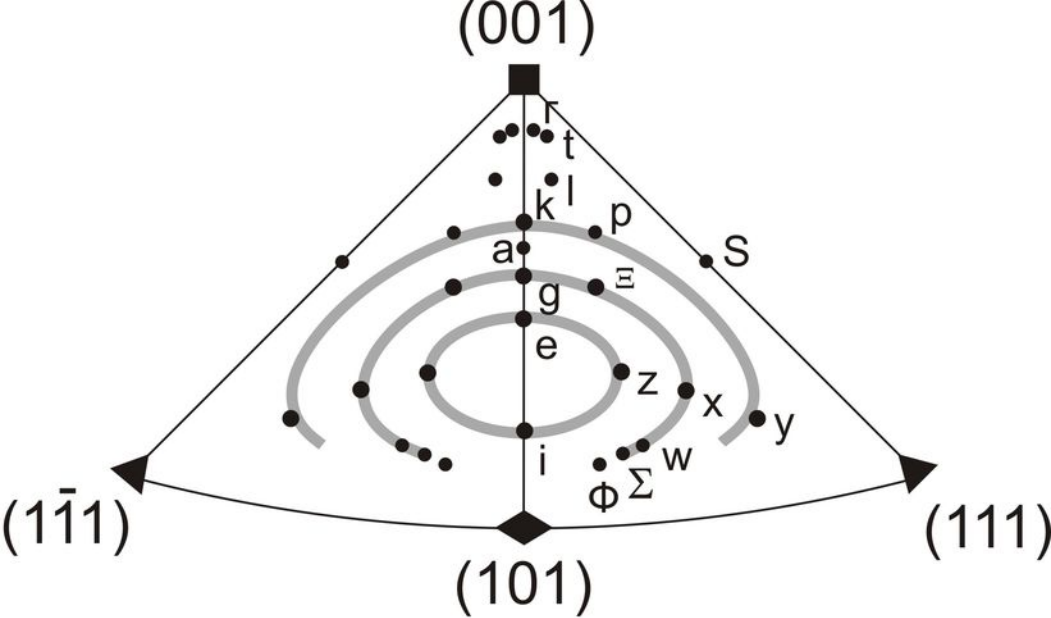


Figure 6.