Cite as Authors (Year) Title. American Mineralogist, in press. (DOI will not work until issue is live.) DOI: http://dx.doi.org/10.2138/am-2015-5114 **REVISION 1** Toward theoretical mineralogy: a bond-topological approach Frank C. Hawthorne<sup>1,\*</sup> <sup>1</sup> Department of Geological Sciences, University of Manitoba, Winnipeg, Manitoba R3T 2N2, Canada \* E-mail: frank hawthorne@umanitoba.ca 

This is a preprint, the final version is subject to change, of the American Mineralogist (MSA)

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#### ABSTRACT

20 The patterns of linkage of chemical bonds in space contain significant energetic information that may be used as the basis of a theoretical approach to the structure and chemical 21 composition of minerals. This approach combines aspects of graph theory, bond-valence theory, 22 23 and the moments approach to the electronic-energy density-of-states to interpret topological aspects of crystal structures, and allows consideration of many issues of crystal structure, mineral 24 composition and mineral behavior that are not addressed by common theoretical methods. The 25 chemical composition of a mineral is controlled by the weak interaction between the structural 26 unit and the interstitial complex. The principle of correspondence of Lewis acidity-basicity 27 28 asserts that stable structures will form when the Lewis-base strength of the structural unit closely matches the Lewis-acid strength of the interstitial complex. This principle allows analysis of the 29 factors that control the chemical compositions and aspects of the structural arrangements of 30 minerals, and provides a mechanism to understand the relations between structure, the speciation 31 of its constituents in aqueous solution, and its mechanism of crystallization. (H<sub>2</sub>O) groups in the 32 structural unit limit the polymerization of the structural unit in one or more directions, 33 controlling the polymerization of the structural unit. This is a major cause of structural diversity 34 in oxygen-based minerals, and accounts for the systematic distribution in mineral species from 35 the core to the surface of the Earth. 36 The moments approach to the electronic-energy density-of-states provides a bond-37

topological interpretation of the energetics of a structure. When comparing structures, the most important structural differences involve the first few disparate moments of the electronic-energy density-of-states. We may classify chemical reactions according to the lowest-order moment of the electronic-energy density-of-states that is conserved, which allows us to identify the principal structural changes that drive chemical change: (a) coordination number for discontinuous

reactions; (b) short-range order for continuous reactions. This relation between the bond
topology of a structure and its enthalpy of formation from constituent oxides is indicated by a
correlation between change in anion-coordination number and reduced enthalpy of formation for
the reactions <sup>[6]</sup>Mg<sub>m</sub><sup>[4]</sup>Si<sub>n</sub>O<sub>(m+2n)</sub> = mMgO + nSiO<sub>2</sub>. *Keywords:* Bond topology, graph theory, bond-valence theory, electronic-energy density-of-

- 49 states, polyhedron linkage, chemical composition, structural unit, interstitial complex.

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#### **INTRODUCTION**

51 Minerals are the stuff of the Earth; without them, there would be no Earth or any other rocky planet. As geologists sensu lato (i.e., scientists who study the Earth), we are all interested 52 in the properties of minerals and how minerals behave in Earth processes—how they respond to 53 54 changing temperature, pressure, etc., how they interact with each other, and especially how they interact with both natural and anthropogenic fluids. Moreover, minerals are our principal source 55 of economic materials and a major constituent of soils, and their chemical compositions and 56 surface properties are key in this regard. Thus Mineralogy has focused on describing minerals 57 and characterizing their chemical compositions and physical properties, providing this 58 59 information for use in Petrology, Geochemistry, Geophysics, Soil Science, etc. However, from the perspective of Mineralogy as a science, we also wish to understand why minerals have the 60 chemical compositions, atomic arrangements and physical and surface properties that they do. 61 62 Such understanding requires a theoretical framework within which we can consider the constitution and behavior of minerals, and it is such a framework that I will consider here. 63

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#### 65 Established theoretical methods

What kind of methods do we have to understand and interpret mineralogical information? 66 67 We use crystal chemistry to systematize mineral properties, classical thermodynamics to analyze processes involving minerals, and computational mineralogy to understand mineral properties 68 and to calculate properties of minerals the stabilities of which are beyond the range of current 69 70 experimental methods. Using thermodynamics, we can make calculations for mineral reactions while not knowing much about where the atoms are and what the atoms are doing. There is now 71 an enormous amount of information on atomic arrangements in minerals, and we would prefer to 72 73 have an atomic-scale understanding of the factors controlling atomic arrangements, chemical

74 compositions, mineral stability and chemical reactions. We have gained significant 75 understanding of minerals, mineral stability and mineral reactions though crystal chemistry, thermodynamics and computational mineralogy. However, these methods have tended to dictate 76 the questions that we ask about minerals—we ask questions to which these methods can give us 77 78 answers. There are many other questions of scientific interest which resist our current theoretical 79 approaches. Such questions tend to be ignored as they are seen as intractable, or even irrelevant to current issues in Petrology, Geochemistry, Geophysics, etc. Let us consider some of these 80 questions: (1) Why do minerals have the chemical formulae that they do? (2) Why do they have 81 82 their specific structural arrangements? (3) Why are minerals stable over specific ranges of pH, 83 Eh, temperature, pressure and activities of their various constituents? (4) What are the relations between crystal structure and both enthalpy and Gibbs free energy of formation? Many of these 84 questions are fundamental to our understanding of minerals and their behavior, and yet have 85 86 tended to be ignored in the past because they are not susceptible to established theoretical techniques in Physics and Chemistry. 87

We conventionally represent a crystal structure as a space group plus a set of unit-cell 88 89 dimensions plus a list of atom coordinates (and displacement parameters), and we use these parameters together with techniques in computational mineralogy to calculate various properties 90 of the crystal. This general approach is extremely successful in understanding the physical 91 92 properties of materials, and it is successful for a reason. The underlying theories deal with electron sharing between atoms fairly rigorously, and many physical properties of crystals are 93 94 dependent on the details of electron sharing between atoms. However, what the underlying theory does not do is explain the origin of translational (and quasi-crystal) symmetry in crystals. 95 Moreover, crystal-structure arrangements can be surprisingly insensitive to major variations in 96 97 electron delocalization and corresponding variations in physical properties. For example, the

NaCl structure is adopted both by the insulator LiF, with a band gap of 13.6 eV, and the
semiconductor PbS, with a band gap of 0.37 eV. Here, structure type is rather insensitive to bond
type (although other semiconductors with small band gaps do adopt the sphalerite and wurtzite
arrangements).

102 What is also interesting is that our perceptions of crystal structure are very insensitive to the structural parameters used for such calculations. We cannot generally appreciate the salient 103 104 features of a crystal structure from its space group, unit-cell dimensions and atom coordinates; we need a picture of that structure to recognize these features. This has been obvious since the 105 dawn of crystal-structure determination, and crystal structures have been classified and 106 107 interpreted according to the details of their atom connectivity as expressed visually (e.g., Barlow 1883, 1898; Bragg 1930; Belov 1961; Moore 1970, 1974; Burns 1999, 2005; Krivovichev 2004, 108 2008, 2009; Krivovichev et al. 1998; Hawthorne 1985, 1986, 1990, 2014; Grice et al. 1999; 109 Hawthorne and Huminicki 2002; Huminicki and Hawthorne 2002). This suggests (at least to me) 110 that if we wish to understand why crystal structures have the atom arrangements that they do, we 111 should not seek this understanding *via* calculations that involve the sharing of electrons between 112 113 atoms/ions; we need to understand the commonalities and differences in atom connectivity in crystal structures (cf. Bowen 1928). 114

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#### 116 Mineral chemistry and structure

117 Consider the hydrated magnesium-sulfate compounds,  $Mg(SO_4)(H_2O)_n$  where n = 0-7,

118 11: synthetic Mg(SO<sub>4</sub>), kieserite, sanderite, synthetic Mg(SO<sub>4</sub>)(H<sub>2</sub>O)<sub>3</sub>, starkeyite,

119  $Mg(SO_4)(H_2O)_4$ , cranswickite,  $Mg(SO_4)(H_2O)_4$ , pentahydrite, hexahydrite, epsomite and

meridianiite,  $Mg(SO_4)(H_2O)_{11}$ . The  $Mg(SO_4)$  part of the chemical formulae is fixed by the

121 electroneutrality principle, but other factors control the degree of hydration of each mineral.

122	With increasing (H <sub>2</sub> O) content, there is a gradual depolymerization of the (Mg $\Phi_6$ ) and (SO <sub>4</sub> )
123	polyhedra ( $\Phi = O, H_2O$ ) as the valence-sum rule (Brown 2002a) prevents linkage of these
124	polyhedra through (H <sub>2</sub> O) ligands (Hawthorne 1992; Hawthorne and Sokolova 2012). We know
125	that (H <sub>2</sub> O) tends to depolymerize the structures of minerals (and synthetic inorganic solids).
126	However, we have little idea of (1) how such depolymerization is quantitatively related to
127	increasing $(H_2O)$ content, and $(2)$ what is the effect of interstitial cations on the bond topologies
128	of the resulting structures.
129	Consider structurally and chemically complicated minerals such as botryogen,
130	$Mg_{2}(H_{2}O)_{12}[Fe^{3+}_{2}(SO_{4})_{4}(OH)_{2}](H_{2}O)_{2}, or metavoltine, K_{2}Na_{6}Fe^{2+}(H_{2}O)_{6}[Fe^{3+}_{3}O(SO)_{4})_{6}(H_{2}O)_{3}]_{2}$
131	(H <sub>2</sub> O) <sub>6</sub> . Their chemical formulae are constrained by the requirement of electroneutrality, but
132	what dictates the other details of their chemical formulae? Why does botryogen have Mg <sub>2</sub> as its
133	interstitial cation, and not Ca <sub>2</sub> or Na <sub>4</sub> ? Why does botryogen have 14 (H <sub>2</sub> O) groups in its formula?
134	Why doesn't it have (for example) 8 ( $H_2O$ ) groups? Why does botryogen contain any ( $H_2O$ )
135	groups at all and what are the roles of these (H <sub>2</sub> O) groups in the structure? How do these aspects
136	of chemistry and structure relate to the stability of botryogen as a function of Eh and pH?
137	Such questions as these commonly cannot be addressed by our established methods of
138	theoretical investigation, and for those that can, we often must be satisfied with explanations at
139	the macroscopic scale. Here I will examine (1) how we can address such questions for oxygen-
140	bearing minerals from a theoretical perspective, and (2) what are the advantages of trying to
141	incorporate process (e.g., crystallization, dissolution) into these considerations. The ideas given
142	here are based on bond topology, the arrangement of chemical bonds in space. I also emphasize
143	that these ideas are currently under development, and urge others (particularly students) to
144	develop new modifications and applications of this approach (and other approaches) to
145	answering the very basic questions raised here.

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#### **GRAPH THEORY**

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A graph is a mathematical structure that is used to examine pairwise relations between discrete objects. A chemical bond defines the pairwise relation between bonded atoms; similarly, linkage between structural fragments (e.g., coordination polyhedra) also defines such a pairwise relation. Thus a graph seems a natural representation of a bonded array of atoms, with the advantage that we may use graph theory to examine the properties of such bonded arrays.

153 A graph is defined as a nonempty set of elements, V(G), called vertices, and a nonempty set of unordered pairs of these vertices, E(G), called edges (Wilson 1979). We may label the 154 vertices, we may color the vertices, we may assign a direction to the edges, and we may assign 155 156 weights to the edges, resulting in a *weighted labelled polychromatic digraph*, shown pictorially 157 in Figure 1a. The square molecule shown in Figure 1b corresponds to the graph of Figure 1a as there is a one-to-one mapping of the atoms of the molecule (Fig. 1b) onto the vertex set of the 158 159 graph (Fig. 1a) and of the chemical bonds of the molecule (Fig. 1b) onto the edge set of the graph (Fig. 1a). We may represent this graph as a matrix (Fig. 2) in which each column and row 160 of the matrix is associated with a specific (colored labelled) vertex and the corresponding matrix 161 162 entries denote whether (positive) or not (zero) two vertices are adjacent (that is, joined by an edge). If the matrix elements are the weight functions of the edge set, then this matrix is called 163 the *adjacency matrix*, which is thus a numerical representation of the graph. The number of 164 edges involving a vertex is known as the *degree* of that vertex. In a digraph, the *indegree* of a 165 vertex is the number of edges incident at that vertex, and the *outdegree* of a vertex is the number 166 167 of edges exident at that vertex.

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#### 170 The handshaking lemma

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# The sum of the degrees of all the vertices of a graph is equal to twice the total

*number of its edges.* 172

Each edge in a graph contributes two degrees to the sum of the degrees of all the vertices of a 173 graph, and hence this sum must be even. This relation is known as the *handshaking lemma* 174 (Wilson 1979) for obvious reasons: each edge involves two vertices, just as a handshake involves 175 two hands, and hence the sum of the degrees of all vertices in a graph is even, as is the number of 176 hands shaken, independent of the number of vertices in the graph or the number of people 177 shaking hands. The handshaking lemma is extremely powerful in dealing with problems 178 179 involving coordination number and connectivity in structures, particularly where the vertex set is partitioned into two subsets, as is the case where there are significant differences in 180 electronegativity of the constituent atoms in a structure, and atoms of the same set do not bond to 181 182 one another. For example, in AB compounds, the coordination numbers of A and B must be equal (e.g., [6] in NaCl and [8] in CsCl), whereas in AB<sub>2</sub> compounds, the coordination number 183 of A must be twice that of B (e.g., [6] and [3] in  $TiO_2$  and [8] and [4] in CaF<sub>2</sub>). 184 185 Graphs with multiple edges 186

# The definition of a graph given above is that of a *simple graph*; that is, a graph where there cannot be more than one edge connecting two vertices. We may define a *general graph* (or more simply, a graph) as a nonempty set of elements, V(G), called vertices, and a nonempty *family* of unordered pairs of these vertices, E(G), called edges. A family is a collection of elements, some of which may occur several times in that collection. The existence of family (rather than a set) in the definition of a general graph allows multiple edges between a pair of

vertices, and this gives us much greater flexibility in applying graph theory to crystal structuresthan would otherwise be the case.

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# 196 Graphical representation of linkage between polyhedra

197 Above, we were using simple graphs to represent the linkage of individual atoms by chemical bonds. However, complicated crystal structures are widely considered as (and 198 199 represented by) arrangements of linked coordination polyhedra. We may assign different coordination polyhedra to different vertices of a (general) graph, and the capability of having 200 201 multiple edges allows us to efficiently denote the details of linkage between different 202 coordination polyhedra (Hawthorne 1983). Polyhedra may be represented by coloured vertices of 203 a labelled graph in which different colors represent different coordination and labels denote chemically and crystallographically distinct polyhedra. Linkage is indicated by an edge or edges 204 205 between vertices, and the number of edges between two vertices denotes the number of atoms common to both polyhedra (Fig. 3, M = octahedrally coordinated cation; T = tetrahedrally 206 207 coordinated cation;  $\varphi =$  unspecified ligand); round brackets and curly brackets denote a 208 polyhedron or a group, e.g.,  $(SO_4)$ ,  $(H_2O)$ ; square brackets denote linked polyhedra, e.g., 209  $[M(TO_4)_2\phi_4]$ . For two vertices, no edge denotes disconnected polyhedra (Fig. 3a), one edge 210 denotes corner-sharing between two polyhedra (Fig. 3b), and two edges denote edge-sharing between two polyhedra (Fig. 3c). Figure 3d shows the cluster  $[M_2(T\phi_4)_2\phi_8]$  and its graphical 211 representation. In a graphical representation, geometrical information is lost. This is illustrated in 212 213 Figure 3e which shows two different possible arrangements of the corner-linked cluster 214  $[M(T\phi_4)_2\phi_4]$ . Both these clusters are described by the same graph; such clusters are called 215 geometrical isomers (Hawthorne 1983). It is very useful to represent the FBB (Fundamental 216 Building Block) of a mineral in this graphical fashion as the hierarchical aspects of the

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222	<b>BOND-VALENCE THEORY</b>
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220	1995, 1999, 2005; Burns et al. 1995).
219	structures (e.g., Hawthorne 1983, 1994; Hawthorne et al. 2000a; Krivovichev 2008, 2009; Burns
218	of graphical representation is used quite commonly to consider the bond topology of complex
217	classification are immediately grasped from the arrangement of the constituent graphs. This type

# 223 Bond topology, bond-valence theory and bond-valence curves

Eighty-five years ago, Pauling (1929) introduced his "rules for ionic structures"

225 (discussed in detail by Hawthorne 2007a). These rules were extremely useful in helping to solve

crystal structures in the early days of structural crystallography. While some of the rules were

given justification via somewhat vague ionic arguments (Burdett and McLarnan 1984), they are

actually collective observations of the structural arrangements available at that time. Hence they

need no "theoretical justification" and their long-term utility in comparing structural

arrangements should not be surprising. For many years, Pauling's rules were identified with

231 "ionic materials", despite the fact that Pauling (1929) discussed bond angles and Pauling (1960)

states that quite covalent materials may obey rules similar to those applicable to ionic crystals.

233 Moreover, Bragg (1930) interpreted Pauling's second rule in terms of only nearest-neighbor

forces, this being the first covalent interpretation of Pauling's second rule (see Hawthorne 2007b

235 for details).

In the late 1960s, it became apparent from the large amount of crystal-structure data (made available by the development of automated X-ray diffractometers) that there are relations between the lengths of chemical bonds (for specific pairs of atoms) and the strengths of those bonds, and many schemes were put forward to relate these variables in a quantitative manner. Pauling (1929) introduced the term *bond strength* to represent the strength of a bond between a

241	cation and an anion as measured by the cation valence divided by the cation coordination
242	number. It became apparent in the 1970s that a new term was needed for the strength of a bond,
243	where this strength is a function of bond length, in order to distinguish it from the Pauling bond-
244	strength, and the term bond valence was introduced. Bond valence is defined as the strength of a
245	chemical bond where, for any pair of bonded atoms, that strength is inversely proportional to the
246	distance between those atoms. Brown and Shannon (1973) introduced their widely used bond-
247	valence parameters; these have been (and are continually being) refined (e.g., Brown 2002a,
248	2009, 2013) and are now almost universally used to check the validity of refined crystal
249	structures.

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# 251 The bond-valence model: background

I am interested primarily in oxide and oxysalt minerals which have significant differences 252 between the electronegativities of the bonded atoms; I will refer to these atoms as *cations* and 253 anions, with no implication as to the character of their chemical bonding. Let us define a crystal, 254 liquid or molecule as a network that consists of atoms connected by heteronuclear chemical 255 256 bonds. Cations and anions alternate along any bond path through this network, and the network must conform to the law of electroneutrality: the total valence of the cations is equal to the total 257 258 valence of the anions. Bond valence is defined as the strength of a chemical bond between any two ions. 259

There has been extensive work relating the form and numerical parameters of bondvalence curves to different models of chemical bonding (e.g., Burdett and Hawthorne 1993; Preiser et al. 1999; Gibbs et al. 2014), and it has become apparent that this approach is quite general in that it applies equally well to structures with predominantly ionic bonds and with predominantly covalent bonds. Brown (1981, 2002a, 2002b, 2009, 2013) has systematically

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265	developed this approach into a comprehensive model, the Bond-Valence Model, that addresses
266	many aspects of chemical bonding in inorganic crystals. The bond-valence model actually
267	consists of two distinct parts, Bond-Valence Theory and empirical bond-valence curves (Fig. 4).
268	Most scientists have tended to ignore bond-valence theory and focus solely on using empirical
269	bond-valence curves for (1) validating the stereochemical details resulting from crystal-structure
270	refinement, and (2) various crystal-chemical purposes. It does not seem to be generally realized
271	that bond-valence theory is formally independent of the analytical bond-valence curves used
272	extensively in crystallography and crystal chemistry.
273	
274	Bond-valence theory
275	Bond-valence theory is based on three principal axioms (Fig. 5): (1) the valence-sum
276	<i>rule</i> , (2) the <i>loop rule</i> , and (3) the <i>valence-matching principle</i> .
277	
278	The valence-sum rule. The sum of the bond valences at each atom is equal to the
279	magnitude of the atomic valence.
280	For any field, Gauss' law relates the flux of the field intensity through a closed surface to
281	the total net charge enclosed within that surface. The valence-sum rule is thus a corollary of
282	Gauss's theorem applied to the electrostatic potential field, and the fluxes linking atoms in this
283	model correlate very strongly with the bond valences assigned in the bond-valence method
284	(Preiser et al. 1999). Topological properties of the various fields associated with an array of
285	atoms are discussed by Brown (2002b). Long-range Coulombic interactions are inductively
286	transmitted through a crystal by the operation of Gauss' law on the Coulomb field at each atom
287	in the crystal (Preiser et al. 1999).
288	

289 The loop rule. The sum of the directed bond valences around any closed path (loop) of290 bonds in the structure is equal to zero.

291

The valence-matching principle. The *Lewis acid strength* of a cation may be defined as
its characteristic (bond) valence which is equal to its atomic (formal) valence / mean
coordination-number (Brown 1981).

The Lewis-base strength of an anion can be defined as the characteristic valence of the 295 bonds formed by the anion. If two ions form a bond, the magnitude of the strength of the bond 296 from the cation to the anion is controlled by the Lewis-acid strength of that cation, and the 297 298 magnitude of the strength of the bond from the anion to the cation is controlled by the Lewisbase strength of that anion. However, the bond from the cation to the anion is the same bond as 299 that from the anion to the cation, and hence the magnitudes of the Lewis acid strength and the 300 Lewis base strengths of the constituent ions must be approximately the same for that bond to 301 form (Fig. 6). This argument leads to a particular criterion for chemical bonding, the valence-302 matching principle (Brown 2002a, 2009): 303 Stable structures will form where the Lewis-acid strength of the cation closely 304

# 305 *matches the Lewis-base strength of the anion.*

As a chemical bond involves both a cation and an anion, the electron-attracting capacity of the cation must match the electron-donating capacity of the anion for a chemical bond to form.

- The above definition of Lewis basicity is often not very useful, as variations in bondvalence around anions are much greater than variations in bond-valence around cations, and any characteristic bond-valence that is assigned has too large a dispersion to be useful. For example, in dravite (Hawthorne et al. 1993), Na is [9]-coordinated and the O atoms to which it is bonded
- receive on average 0.11 v.u. from each Na-O bond. In CrO<sub>3</sub> (Stephens and Cruickshank 1970),

which consists of pyroxene-like chains of <sup>[4]</sup>CrO<sub>3</sub>, one O is bonded only to Cr<sup>6+</sup> and receives 2.00 313 314 v.u. from the Cr-O bond. With this amount of variation in bond valence, 0.11–2.00 v.u., it is not useful to define a Lewis-base strength for  $O^{2-}$ . Consider a complex oxyanion such as  $(SO_4)^{2-}$ 315 (Fig. 7): The central  $S^{6+}$  cation provides 1.5 v.u. to each coordinating O atom and these need an 316 additional 0.5 v.u. from other neighbouring cations. If the coordination number of  $O^{2-}$  is [n], then 317 the average valence of the bonds to  $O^{2-}$  (exclusive of the S-O bond) is 0.5 / (n-1) v.u.; where n 318 = 2, 3, 4 or 5, the mean bond-valences to  $O^{2-}$  are 0.50, 0.25, 0.17 or 0.11 v.u., respectively. The 319 average bond-valence received by the  $(SO_4)^{2-}$  group is the same as the average bond-valence 320 received by each individual  $O^{2-}$  anion, and allows us to define a Lewis basicity for the oxyanion 321 group. For the  $(SO_4)^{2-}$  oxyanion, the possible average bond-valences are quite tightly constrained 322 (0.50–0.11 v.u.) and we may calculate a useful Lewis basicity. Tables 1 and 2 list Lewis acidities 323 324 and Lewis basicities for geochemically common cations and oxyanions.

The valence-matching principle is the most important and powerful idea in bond-valence theory (Hawthorne 2012): it allows us not just to interpret known structures or compounds; we can test the stability of possible compounds (in terms of whether they can exist or not), which moves us from *a posteriore* to *a priori* analysis. I will consider three simple examples (taken from Hawthorne 1994) to illustrate this principle.

Consider the composition Na<sub>2</sub>SO<sub>4</sub>. The Lewis basicity of the (SO<sub>4</sub>) group is 0.17 v.u. (Table 2) and the Lewis acidity of Na is 0.17 v.u. (Table 1). The Lewis basicity of the anion matches the Lewis acidity of the cation, the valence-matching principle is satisfied, and thenardite, Na<sub>2</sub>SO<sub>4</sub>, is stable.

Consider the composition Na<sub>4</sub>SiO<sub>4</sub>. The Lewis basicity of the (SiO<sub>4</sub>) group is 0.33 v.u.
(Table 2) and the Lewis acidity of Na is 0.17 v.u. The Lewis basicity of the anion does not match

the Lewis acidity of the cation, the valence-matching principle is not satisfied, and Na<sub>4</sub>SiO<sub>4</sub> is
not a mineral (or stable structure).

338	Consider the composition Na[AlSiO <sub>4</sub> ]. The Lewis basicity of the [AlSiO <sub>4</sub> ] group is 0.13
339	v.u. and the Lewis acidity of Na is 0.17 v.u. The Lewis basicity of the anion matches
340	(approximately) the Lewis acidity of the cation, the valence-matching principle is satisfied, and
341	nepheline, Na AlSiO <sub>4</sub> is a stable structure. Moreover, nepheline shows incommensurate
342	behaviour (e.g., Angel et al. 2008), perhaps reflecting the slight mismatch between the Lewis
343	basicity and acidity of its constituents.
344	These arguments illustrate the power of the valence-matching principle. We may consider
345	the possible stability of specific chemical compositions of interest. It is important to recognize
346	that this is a priori analysis; we need no crystal-structure information to evaluate the potential
347	stability (i.e., existence) of any chemical formula. Moreover, this is a "back-of-the-envelope"
348	calculation that is not obscured by numerical complexity.
349	
350	A priori bond-valences. The valence-sum rule and the loop rule provide a series of
351	simultaneous equations (sometimes called network equations) relating bond valences to the
352	constraints of the valence-sum rule and the loop rule. We may designate these bond valences as a
353	priori bond-valences as they need no geometrical information (i.e., experimental bond lengths)
354	to be calculated: they are derived from the bond topology of the structural arrangement and the

charges of the ions at the vertices of the graph of this arrangement.

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# 357 **Bond-valence curves**

For any pair of bonded atoms, bond valence is inversely proportional to the length of the bond: large bond-valences are associated with short bonds, and small bond-valences are

360 associated with long bonds. To obtain numerical values for the bond valences, each bond is 361 assigned a bond valence such that the valence-sum rule is satisfied (Brown 2002a): The sum of the bond valences at each atom is equal to the magnitude of the atomic valence. Thus bond 362 valences are scaled to the formal valences of the cations and anions involved in the chemical 363 364 bonds. If this is done for a relatively large number of structures, one may derive numerical parameters, bond-valence parameters (or bond-valence curves), that may be used to calculate 365 bond valences from bond lengths. Such parameters are listed by Brown (2002a, 2009, 2013) 366 from a wide variety of sources, and are commonly used to validate experimentally derived 367 crystal structures and to examine various crystal-chemical aspects of their atomic arrangements. 368 369 Brown and Shannon (1973) discussed the differences between the bond-valence model and the ionic model. In the bond-valence model, a structure consists of atom cores held together 370 by valence electrons associated with the chemical bonds between the atoms, and they explicitly 371 372 state that the valence electrons may be associated with chemical bonds in a symmetric (covalent) or asymmetric (ionic) manner. Thus a priori knowledge of the electron distribution is not 373 required to use this approach. Burdett and Hawthorne (1993) showed that the form of the bond-374 375 valence curves may be derived algebraically from a molecular-orbital description of a solid in which there is a significant energy gap between the interacting orbitals on adjacent atoms, 376 whereas Preiser et al. (1999) gave an ionic justification of the bond-valence model. One may 377 conclude that the bond-valence model is not a theory of "ionic" bonds or "covalent" bonds. It is a 378 simple yet quantitative method that allows us to examine and analyze the stereochemistry and 379 380 physical properties of both simple and complex solids; it is used primarily for crystals, but also can be used for surfaces (Schindler et al. 2004a, 2004b; Bickmore et al. 2004, 2006), glasses and 381 liquids. Although the idea of bond valence grew out of Pauling's second rule, the wide variety of 382 383 its application and subsequent examination of its theoretical underpinnings show that it is a

384	theory of atomic arrangements in its own right, without any reference to specific models of the
385	chemical bond. Its power lies in the fact that it is a back-of-the-envelope method in which the
386	physical details are not obscured by complexities of computation. Each year sees new
387	applications to an increasing array of problems as the bond-valence model takes a central role in
388	our understanding of complex materials.
389	
390	BOND-TOPOLOGICAL CONTROLS ON THE STRUCTURE AND
391	CHEMICAL COMPOSITION OF OXYSALT MINERALS
392	The valence-matching principle is a powerful method of assessing the stability (i.e.,
393	existence or otherwise) of potential chemical compounds, and we saw above how we can a priori
394	predict the existence of Na <sub>2</sub> SO <sub>4</sub> (thenardite), the non-existence of Na <sub>4</sub> SiO <sub>4</sub> , and the existence of
395	NaAlSiO <sub>4</sub> (nepheline). For such simple structures, this approach is straightforward. However, for
396	more complicated minerals, e.g., botryogen, $Mg_2(H_2O)_{10}[Fe^{3+}_2(SO_4)_4(OH)_2](H_2O)_2$ and
397	metavoltine, $K_2Na_6Fe^{2+}(H_2O)_6[Fe^{3+}_{3}O(SO_4)_6(H_2O)_3]_2(H_2O)_6$ , the approach is less transparent.
398	Yet these complex minerals raise some very fundamental questions pertaining to the details of
399	their chemical composition. Again, (1) why does botryogen have Mg rather than Ca or Ba as its
400	divalent interstitial cation; (2) why does it have divalent interstitial cations, Mg <sub>2</sub> , rather than
401	monovalent interstitial cations, $Na_4$ or $K_4$ ; (3) why does it have 14 (H <sub>2</sub> O) groups in its formula;
402	why doesn't it have (for example) 12 ( $H_2O$ ) groups; (4) why does it have any ( $H_2O$ ) groups at all
403	in its formula; what is the role of these $(H_2O)$ groups in the structure? How do the chemical
404	formula and structural arrangement of botryogen relate to its stability as a function of Eh and
405	pH? We cannot say that we understand minerals until we understand what controls their
406	chemical compositions at this level of detail.

407 In order to address this issue for complex structures, Hawthorne (1983, 1985) divided a 408 structure into two components: the structural unit, the strongly bonded part of the structure, consisting of oxyanions and low-coordination-number cations; and the *interstitial complex*, the 409 410 weakly bonded part of a structure, usually consisting of monovalent cations, large divalent 411 cations and (H<sub>2</sub>O) groups. To do this, we must define what we mean by "strongly bonded" and "weakly bonded". Many minerals contain octahedrally coordinated divalent cations, and many of 412 these minerals, e.g., forsterite, enstatite, are stable at high temperature. Many minerals contain 413 monovalent and divalent cations in higher coordination numbers, e.g., thenardite, gypsum, and 414 many minerals are held together by hydrogen bonds; these minerals commonly crystallize from 415 416 aqueous solution at ambient (or close to ambient) conditions. An appropriate boundary between "strongly bonded" and "weakly bonded" will be between 0.33 v.u. (for <sup>[6]</sup>Mg) and 0.20 v.u. (for 417 common hydrogen bonds, Ba,  $Pb^{2+}$ ) or 0.23 v.u. (for Sr), and I will take it as 0.30 v.u., although 418 419 the exact value will change depending on other bond-valences in a structure.

If we can define Lewis acidities and Lewis basicities for the structural unit and the 420 421 interstitial complex, we may examine their interaction using a mean-field version of the valence-422 matching principle. This *binary representation* of a complex structure is illustrated in Figure 8 for botryogen,  $\{Mg_2(H_2O)_{10}\}$  [Fe<sup>3+</sup><sub>2</sub>(SO)<sub>4</sub>(OH)<sub>2</sub>](H<sub>2</sub>O)<sub>2</sub>. We have partitioned the structure into a 423 structural unit:  $[Fe^{3+}_{2}(SO)_{4}(H_{2}O)_{2}]$ , a cluster of  $Fe^{3+}$  octahedra and sulfate tetrahedra (shown by 424 425 cation-centered polyhedra in Fig. 8), and an interstitial complex:  $\{Mg_2(H_2O)_{12}\}$ , Mg cations 426 together with their associated  $(H_2O)$  groups. We may calculate a Lewis basicity for the structural unit and a Lewis acidity for the interstitial complex as aggregate properties of the constituents of 427 these two units (see Hawthorne and Schindler 2008), and their interaction may be examined in a 428 manner similar to the application of the valence-matching principle to simple chemical 429

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430 compositions using the *principle of correspondence of Lewis acidity-basicity* (Hawthorne and

431 Schindler 2008):

432 Stable structures will form where the Lewis-acid strength of the interstitial
433 complex closely matches the Lewis-base strength of the structural unit.

434 As noted above, the *principle of correspondence of Lewis acidity-basicity* is thus the mean-field

equivalent of the valence-matching principle. We may now use this principle in conjunction with

the binary representation of complex structures to examine the reasons why minerals have the

437 chemical compositions that they do, and to predict the possible chemical compositions of

438 potential minerals.

439

# 440 The role of H<sub>2</sub>O in crystal structures

There are several different major roles for hydrogen (H) in crystal structures (Hawthorne 441 442 1992; Hawthorne and Baur 1994). The (OH) and (H<sub>2</sub>O) groups are very important because of their polar nature: on the O side, each group acts as an anion, whereas on the H side, the group 443 acts as a cation. The hydrogen-bond interaction is extremely important; it moderates many 444 biological interactions essential to life, and it imparts great diversity both to atomic arrangements 445 446 in minerals and to atom interactions in minerals. An  $(H_2O)$  group may (1) moderate Lewis 447 acidity and Lewis basicity, and (2) control the dimensional polymerization of structural units. First, I will consider how (H<sub>2</sub>O) can act as a moderator of bond valence. 448

449

(H<sub>2</sub>O) **bonded to one cation.** Consider the atomic arrangements in Figures 9a and 9b: A cation, M, bonds to an anion S with a bond valence of v v.u., and a cation, M, bonds to an (H<sub>2</sub>O) group, and the (H<sub>2</sub>O) group bonds to an anion, S. In Figure 9a, the anion receives one bond of bond valence v v.u.from the cation M. In Figure 9b, the O atom of the (H<sub>2</sub>O) group receives a

454	bond valence of v v.u. from the cation; the bond-valence requirements of the central O atom are
455	satisfied by two short O-H bonds of strength $(1 - v/2)$ v.u. Each H forms a hydrogen bond with
456	the S anion in order to satisfy its own bond-valence requirements, and the S anion thus receives a
457	bond valence one half (Fig. 9b) of what it received where it was bonded directly to the M cation
458	(Fig. 9a). The (H <sub>2</sub> O) group is functioning as a <i>bond-strength transformer</i> , dividing one bond
459	(bond strength = v v.u.) into two bonds of half the strength (bond valence = $v/2$ v.u.); this type of
460	(H <sub>2</sub> O) group is called a <i>transformer</i> (H <sub>2</sub> O) group (Hawthorne and Schindler 2008).

461

(H<sub>2</sub>O) bonded to two cations. Consider the atomic arrangement in Figure 9c: two cations bond to an (H<sub>2</sub>O) group which bonds to two anions. The O atom receives a bond valence of 2v v.u. from the cations, and the valence-sum rule at this O anion is satisfied by two short O-H bonds of strength (1 - v) v.u. Each H forms a hydrogen bond with a neighbouring anion which receives the same bond-valence (v v.u., Fig. 9c) as where it is bonded directly to one *M* cation (Fig. 9a). The (H<sub>2</sub>O) group does not act as a bond-valence transformer, is a *non-transformer* (H<sub>2</sub>O) group.

469

 $(H_2O)$  not bonded to any cation. Consider the atomic arrangement in Figure 9d:  $(H_2O)$ 470 471 is involved only in a hydrogen-bond network. In such an environment, the O atom is usually [4]coordinated, and the (H<sub>2</sub>O) group participates in two O-H (donor-hydrogen) bonds and two 472 H...O hydrogen bonds. Two hydrogen bonds of strength v v.u. are incident at the O atom of the 473 474 (H<sub>2</sub>O) group, the bond-valence requirements of the central O atom are satisfied by two O-H bonds of strength (1 - v) v.u., and each H atom forms a hydrogen bond of strength v v.u. to 475 476 another anion (Fig. 9d). Hence an  $(H_2O)$  group not bonded to any cation does not modify the 477 strengths of its exident chemical bonds, it merely propagates them to more distant anions, as is

478 the case where the  $(H_2O)$  group is bonded to two cations (Fig. 9c); this type of  $(H_2O)$  is

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479 designated non-transformer (H<sub>2</sub>O).
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481	$(H_2O)$ as a component of the interstitial complex. As a component of an interstitial
482	complex, (H <sub>2</sub> O) groups have two principal functions: (1) to satisfy the bond-valence
483	requirements of an interstitial cation where there are not enough anions to do so from adjacent
484	structural units; (2) to function as a bond-valence transformer between interstitial cations and the
485	structural unit. The first case involves only propagating bond valence through space, and hence
486	may involve non-transformer (H <sub>2</sub> O) groups. The second case involves transformer (H <sub>2</sub> O) groups,
487	and these will moderate the Lewis acidity of the interstitial complex. Hence the transformer
488	(H <sub>2</sub> O) groups of the interstitial complex affect the stability of a chemical composition through
489	the operation of the principle of correspondence of Lewis acidity-basicity. Below I will show
490	how this approach to understanding the role of (H <sub>2</sub> O) groups can give us a quantitative
491	understanding of the chemical compositions of interstitial complexes.
492	
493	CONTROLS ON THE AMOUNT OF $(H_2O)$ in minerals:
494	THE PRINCIPLE OF CORRESPONDENCE OF LEWIS ACIDITY-BASICITY
495	Calculation of Lewis basicity
496	The Lewis basicity of a structural unit is the average bond-valence of bonds to that
497	structural unit from adjacent interstitial complexes and structural units (Hawthorne and Schindler
498	2008). The bonds received by the structural unit must balance the charge of the structural unit,
499	and hence we may define the Lewis basicity of the structural unit as the charge on the structural
500	unit divided by the number of bonds to the structural unit. So we need to know (1) the effective

charge on the structural unit, and (2) the number of bonds (from the interstitial complex and
adjacent structural units) needed by the structural unit.

What is the effective charge of the structural unit? The formal charge is not necessarily 503 appropriate to use in this context, particularly for structures with formally neutral structural units 504 505 as then there is no mechanism for the structure to link together. Consider lizardite, Mg<sub>3</sub>Si<sub>2</sub>O<sub>5</sub>(OH)<sub>4</sub> (Fig. 10).  $[Mg_3Si_2O_5(OH)_4]^0$  sheets link to each other via hydrogen bonds from 506 the (OH) groups in the layer of octahedra of one sheet to the bridging O atoms in the layer of 507 tetrahedra of the adjacent sheet. The hydrogen bonds transfer charge from one sheet to the next, 508 and impart a polar character to the sheet; the sheet has cation character on the (OH) side and 509 510 anion character on the silicate side (shown by + and - signs in Fig. 10). In order to correctly describe the interaction between adjacent structural units, we must factor this transfer of charge 511 into our calculation of the charge of the structural unit. The effective charge of lizardite is 0 (the 512 513 formal charge of the structural unit) +  $4 \times 0.20$  (the charge transferred by hydrogen bonding, assuming a hydrogen bond-valence of 0.20 v.u., Brown 1981) =  $0.80^{-1}$ . Note that such a transfer 514 of charge can only involve cations with very asymmetric coordinations (commonly H<sup>+</sup>, less 515 commonly stereoactive-lone-pair cations such as  $Pb^{2+}$  or  $Bi^{3+}$ ). Note that minerals with formally 516 charged structural units may still be polar, and this transfer of charge must be built in to the 517 calculation of Lewis basicity. For metavoltine,  $K_2Na_6Fe^{2+}(H_2O)_6[Fe^{3+}_3O(SO_4)_6(H_2O)_3]_2(H_2O)_6$ , 518 the effective charge of the structural unit (in square brackets) is 10 (the formal charge) + 12 x 519 0.20 (the charge transferred by twelve hydrogen bonds) =  $12.4^+$ . We define the *effective charge* 520 of the structural unit as the formal charge as modified by charge transferred by hydrogen 521 522 bonding from donor anions within the structural unit. 523

524

#### 525 What is the number of bonds needed by the structural unit?

526 First, I will show how we can calculate this quantity if everything is known about the crystal structure. The total number of chemical bonds in a structure is the sum of the products of 527 the cation-coordination numbers and the numbers of those cations in the formula unit. We may 528 similarly calculate the number of bonds in the structural unit. The difference between these two 529 values is the number of bonds needed by the structural unit. This calculation is trivial if the 530 details of the crystal structure are known. However, we wish to *predict* information about crystal 531 structures, and we do not know such stereochemical details. We must be able to predict this 532 information if we want a priori analysis of crystal structures; how to do this is covered in the 533 534 next few sections.

535

#### 536 The charge deficiency per anion: CDA

537 Schindler et al. (2000b) defined average basicity as the average bond-valence per Oatom contributed by the interstitial species and adjacent structural units. Average basicity 538 correlates with the average O-coordination number of the structural unit, and this correlation 539 plays a critical role in stereochemical prediction. As indicated by its definition, this quantity is 540 the additional average incident bond-valence required from the interstitial complex by each O 541 atom of the structural unit to satisfy the principle of correspondence of Lewis acidity-basicity, 542 and Schindler et al. (2006) renamed this quantity the charge deficiency per anion, or CDA. 543 Below we will see that the CDA of a structural unit correlates strongly with the numbers of 544 545 bonds to those structural units from the interstitial complex and neighboring structural units. It is these correlations that play a major role in *a priori* prediction of structural features. 546 The CDA of a structural unit is the effective charge of the structural unit divided by the 547

number of O atoms in the structural unit. For bloedite,  $Na_2[Mg(SO_4)_2(H_2O)_4]$ , the effective

charge of the structural unit is  $2 + 0.2 \times 8 = 3.6^{-1}$  and the number of O atoms in the structural unit

550 is 12; thus the CDA = 3.6 / 12 = 0.30 v.u.

551

### 552 The number of bonds required by the structural unit

The CDA is a measure of the bond valence required by each O atom of the structural unit 553 from the interstitial complex and adjacent structural units. Schindler et al. (2006) showed that 554 555 there is a positive correlation between the CDA of the structural unit and the average number of 556 bonds received by O atoms of the structural unit from the interstitial complex and adjacent structural units,  $\langle NB \rangle_{in}$ . This relation, shown for sulphate minerals in Figure 11, is very 557 558 important as it allows us to predict a range for the number of bonds from the interstitial complex and adjacent structural units to a specific structural unit. In turn, we may then calculate the range 559 560 in Lewis basicity for that structural unit.

For bloedite,  $Na_2[Mg(SO_4)_2(H_2O)_4]$ , the CDA = 0.30 v.u. (see above). Using Figure 11,

we may read off the range for the number of bonds to anions of the structural unit: 1.55 to 2.44.

563 The corresponding range in the total number of bonds to the structural unit is (1.55 to 2.44) x 12

= 18.6 to 29.3, and the resulting range in Lewis basicity of the  $[Mg(SO_4)_2(H_2O)_4]^{2-}$  structural

unit is the effective charge divided by the range in the number of bonds to the structural unit:

566 3.6/(18.6 to 29.3) = 0.12 to 0.19 v.u.

The parameter  $\langle NB \rangle_{in}$  is required to establish a relation between O-coordination number and CDA. To have predictive power, we need to be able to derive the number of bonds required by O atoms *a priori*, without reference to an atomic arrangement, and the type of relation in Figure 11 allows such a prediction. There is another important issue: the data in Figure 11 form a band rather than a linear trend, indicating that the structural units can accommodate a range in the number of bonds from the interstitial complex. It seems apparent that *structural units*  573 maintain their stability as the pH of the environment changes by varying the number of bonds 574 they accept from the interstitial complex and adjacent structural units. Thus the range in 575 numbers of bonds from the interstitial complex and adjacent structural units to the structural unit 576 reflects the range in pH over which the mineral is stable (Hawthorne and Schindler 2008). As 577 shown above, Figure 11 allows calculation of the range of possible Lewis-base strength for a 578 specific structural unit (see example for bloedite given above).

579

580 FACTORS AFFECTING THE COMPOSITION OF THE INTERSTITIAL COMPLEX

It is useful to represent the variation in Lewis-acid strength of an interstitial complex as a function of chemical composition and structure in a graphical fashion, as this facilitates use of the principle of correspondence of Lewis acidity-basicity to examine the interaction between the structural unit and interstitial complex as a function of varying chemical composition of each component of a structure. The chemical formula of a generalized interstitial complex may be written as

587 
$$\{ {}^{[m]}M^{+}{}_{a}{}^{[n]}M^{2+}{}_{b}{}^{[l]}M^{3+}{}_{c}(\mathrm{H}_{2}\mathrm{O})_{d}(\mathrm{H}_{2}\mathrm{O})_{e}({}^{[q]}\mathrm{O}\mathrm{H})_{f}(\mathrm{H}_{2}\mathrm{O})_{g} \}^{Z+}$$

where M are interstitial cations of different coordination number [m], [n] and [l], and valence; d 588 is the amount of transformer ( $H_2O$ ); *e* is the amount of non-transformer ( $H_2O$ ); and *g* is the 589 590 amount of  $(H_2O)$  not bonded to any interstitial cation (Schindler and Hawthorne 2001a). The Lewis acidity of the interstitial complex may be represented graphically as a function of the 591 variables a to g, l to n, q and Z in the above expression (Fig. 12): the ordinate is the Lewis acidity 592 593 of the interstitial complex, the abscissa is the number of transformer (H<sub>2</sub>O) groups per cation, and the curved lines show the variation in Lewis acidity as a function of the number of 594 transformer (H<sub>2</sub>O) groups per cation for interstitial cations of different coordination number and 595 596 formal charge (the corresponding cation charges and coordinations are shown to the left of the

597 curves). Monovalent anions (OH, Cl) may also be incorporated into this procedure (see

Hawthorne and Schindler 2008 for details).

Figure 13 shows the operation of the principle of correspondence of Lewis acidity-599 basicity. The range in Lewis basicity of the structural unit is plotted on the graph of the Lewis-600 601 acidity function (Fig. 12). Where the functions representing the properties of the interstitial complexes and the structural unit do not intersect (i.e., outside the yellow band in Fig. 13a), 602 structures of those compositions are not stable as the Lewis acidities of these interstitial 603 complexes are not within the Lewis-basicity range of the interstitial complex represented on the 604 graph; the principle of correspondence of Lewis acidity-basicity is not satisfied, and structures of 605 606 these compositions will not form. Where the functions representing the properties of the interstitial complexes and the structural unit do intersect (i.e., within the yellow band in Fig. 607 13a), structures of those compositions are potentially stable as the Lewis acidities of these 608 609 interstitial complexes are within the Lewis-basicity range of the interstitial complex represented on the graph; the principle of correspondence of Lewis acidity-basicity is satisfied, and structures 610 611 of these compositions may form. Let us look at what we can do with this approach for a subset of 612 the sulfate minerals.

613

#### 614 Hydroxy-hydrated sulfate minerals

The structural hierarchy developed for sulfate minerals by Hawthorne et al. (2000a) forms a general framework for the examination of sulfate structures from a bond-topologic perspective. Schindler et al. (2006) examined sulfate structures in this way and showed that many crystal-chemical features of sulfate minerals may be understood in terms of the principle of correspondence of Lewis acidity-basicity, in parallel with similar work on borate minerals (Hawthorne et al. 1996a; Schindler and Hawthorne 2001a, 2001b, 2001c), vanadate minerals

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(Schindler et al. 2000b) and uranyl minerals (Burns 2005, Schindler and Hawthorne 2004, 2008).
We will look at two structural units in this group and use the principle of correspondence of
Lewis acidity-basicity to derive possible interstitial complexes and compare them with what is
observed in minerals.

625

626 
$$[M^{2+}(SO_4)_2(H_2O)_4]^{2-}$$
 (M = Mg, Ni, Zn, Fe<sup>2+</sup>). The structural unit  $[M^{2+}(SO_4)_2(H_2O)_4]^{2-}$ 

627 occurs in bloedite,  $\{Na_2\}[Mg(SO_4)_2(H_2O)_4]$ , nickelbloedite,  $\{Na_2\}[Ni(SO_4)_2(H_2O)_4]$ , leonite,

and roemerite,  $\{Fe^{2+}(H_2O)_6\}[Fe^{2+}(SO_4)_2(H_2O)_4]$  (Hawthorne et al. 2000a). Above, we calculated

a range in Lewis basicity for this structural unit (in bloedite): 0.12–0.19 v.u.

631 Values of Lewis acidity for interstitial monovalent cations with coordination numbers [6] 632 to [8] intersect the range in Lewis basicity for 0-2, 0-1 and 0 transformer (H<sub>2</sub>O) groups per

633 cation, respectively. With regard to divalent interstitial cations,  ${}^{[6]}M^{2+}$  can occur with 4–6

transformer (H<sub>2</sub>O) groups and  ${}^{[8]}M^{2+}$  can occur with 2–8 transformer (H<sub>2</sub>O) groups. With regard

to trivalent interstitial cations,  ${}^{[8]}M^{3+}$  can occur only with 8 transformer (H<sub>2</sub>O) groups,  ${}^{[7]}M^{3+}$  and

 $[6]M^{3+}$  cannot occur at all. All minerals of this group conform to these predictions: bloedite,

637 nickelbloedite and changoite have an interstitial complex  $\{^{[6]}Na_2(H_2O)_0....\}$ , leonite and

638 mereiterite have an interstitial complex  $\{^{[6]}K_2(H_2O)_0....\}$ , and roemerite has an interstitial

639 complex  $\{ {}^{[6]}Fe^{2+}(H_2O)_6.... \}$ .

640

641  $[Fe^{3+}(OH)(SO_4)_2]^{2-}$ . The structural unit  $[Fe^{3+}(OH)(SO_4)_2]^{2-}$  occurs in sideronatrite, 642  $\{Na_2(H_2O)_3\}[Fe^{3+}(SO_4)_2(OH)],$  metasideronatrite,  $\{Na_4(H_2O)_3\}[Fe^{3+}(SO_4)_2(OH)]_2(H_2O)_3,$ 643 chaidamuite,  $\{^{[6]}Zn(H_2O)_4\}[Fe^{3+}(SO_4)_2(OH)],$  and guildite,  $\{^{[4+2]}Cu^{2+}(H_2O)_4\}[Fe^{3+}(SO_4)_2(OH)].$ 644 The effective charge of this structural unit is  $(2 + 0.2 \times 1)^- = 2.2^-$ , the number of O atoms in the

645	structural unit is 9, and the CDA of the structural unit is $2.2 / 9 = 0.24$ v.u. With this value, we
646	may derive the lower and upper bounds for $\langle NB \rangle_{in}$ using Figure 11: 1.10–1.75. The resultant
647	range in the number of bonds required by the structural unit is $1.14-1.97 \ge 9 = 10.3-17.7$ .
648	Dividing the effective charge by the number of bonds required, $2.2 / (10.3-17.7)$ , gives the range
649	in Lewis basicity: 0.12–0.22 v.u. As before, we plot this range in Lewis basicity on the graph of
650	the Lewis-acidity function, Fig. 13b, and can predict the range in chemical composition for
651	possible interstitial complexes.
652	Values of Lewis acidity for interstitial monovalent cations with coordination numbers [5]
653	to [7] intersect the range in Lewis basicity for $0-2$ , $0-1$ and 0 transformer (H <sub>2</sub> O) groups per
654	cation, respectively. With regard to divalent interstitial cations, ${}^{[6]}M^{2+}$ can occur with 3–6
655	transformer (H <sub>2</sub> O) groups, ${}^{[7]}M^{2+}$ can occur with 2–7 transformer (H <sub>2</sub> O) groups, ${}^{[8]}M^{2+}$ can occur
656	with 1–8 transformer (H <sub>2</sub> O) groups, and ${}^{[8]}M^{3+}$ can occur with 5–8 transformer (H <sub>2</sub> O) groups.
657	All minerals of this group conform to these predictions: sideronatrite and metasideronatrite have
658	an interstitial complex $\{^{[6]}Na_2(H_2O)_0\}$ , guildite has an interstitial complex $\{^{[6]}Cu^{2+}(H_2O)_4\}$ ,
659	and chaidamuite has an interstitial complex ${^{[6]}Zn(H_2O)_4}$ (Schindler et al. 2006).
660	The approach described above provides significant understanding of what factors affect
661	the chemical compositions of minerals, and some prediction of the details of interstitial cations
662	and anions in minerals. For some structural units, the predicted interstitial complexes vary over a
663	wide range of cations or transformer (H <sub>2</sub> O) groups, which in terms of prediction, is not
664	satisfactory. This indicates the need for further development along these lines. It seems likely
665	that the compositions of interstitial complexes in these circumstances are also affected by the pH
666	of their environment during crystallization. Some very interesting questions now emerge
667	concerning the nature of the crystallization process. Does the pH of the environment have a
668	strong effect on the form of the structural unit or the amount of (H <sub>2</sub> O) incorporated into the

structure? Does the form of the structural unit dictate the identity of the interstitial cations, or does the availability of a particular interstitial cation dictate the form of the structural unit? Are there synergetic interactions between these factors? We can begin to investigate some of these questions using this bond topology approach.

673

#### 674 **Other applications**

This approach has also been used to examine the structure, chemical composition and

stability of vanadate (Schindler et al. 2000a, 2000b), borate (Schindler and Hawthorne 2001a,

677 2001b, 2001c) and uranyl-oxysalt minerals (Schindler and Hawthorne 2004, 2008), and has the

potential to be applied to other low-temperature oxysalt minerals. It has also been used to

679 consider crystal morphology and surface features (Schindler et al. 2004a, 2004b) and

680 crystallization-dissolution of minerals in aqueous solutions (Hawthorne and Schindler 2014). An

681 important aspect of this approach is that it relates bond topology and bond valence to processes

involved in crystallization, and it may also be applicable to chemical reactions. The valence-sum

rule is used for atoms in crystals, glasses and aqueous fluids. It seems reasonable that atoms in

transition between these various states of matter also tend to obey the valence-sum rule. This led

Hawthorne (2012) to propose the *Reaction Principle*:

686 During a chemical reaction, atoms move relative to each other such that they

687 *continually minimize local deviations from the valence-sum rule. Thus as the* 

688 atomic arrangements pass through their excited states, the atoms follow

- 689 *trajectories that are both consistent with those excited states and minimize the*
- 690 *local deviations from the valence-sum rule at all stages of the reaction.*

692 It also suggests that the arrangements of atoms in the reactants may significantly affect the 693 arrangements of atoms in the products, as many mineral reactions will tend to occur by breaking the weaker chemical bonds in the reactants and maintaining the stronger chemical bonds, thus 694 giving us a possible mechanism for explaining Ostwald's Step Rule<sup>1</sup> (see Morse and Casey 1988) 695 696 for an excellent description of this rule in geochemical reactions). An example of this mechanism was given by Gaskell et al. (1991) who showed that a CaSiO<sub>3</sub> glass has short- and medium-range 697 698 structure very similar to that of wollastonite. This suggests that in a CaSiO<sub>3</sub> melt close to the liquidus, the product in the crystallization of wollastonite is already templated in the reactant, 699 and the atoms in the system obey the Reaction Principle, and crystallize as wollastonite. It is 700 701 unfortunate that the structures of magmas are not well-characterized at medium range, but the possibility that such templating of minerals occurs in magmas provides additional incentive to 702 learn more about the structures of magma and the details of crystallization processes at the 703 atomic scale in magmatic systems. 704 705 706 LEWIS BASICITY OF THE STRUCTURAL UNIT, AND THE FORMATION OF ROCKS

#### 707 A major constraint on the chemistry and atomic arrangements of structural units

Above, we defined the boundary between the bonds of the structural unit and the bonds of the interstitial complex as 0.30 v.u. The strength of the bonds involving the interstitial complex is thus less than 0.30 v.u., and hence its Lewis acidity is less than 0.30 v.u. As the principle of correspondence of Lewis acidity-basicity requires that the Lewis basicity of the structural unit match the Lewis acidity of the interstitial complex, *the Lewis basicity of the structural unit must also be less than 0.30* v.u. This is an extremely important statement as it

<sup>&</sup>lt;sup>1</sup> There are many versions of Ostwald's step rule. Perhaps the most general states that there is a tendency for the least-stable product of a chemical reaction to crystallize first, and this phase subsequently reacts over time to form a sequence of progressively more stable phases.

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must exert stringent controls on the possible chemical compositions and atomic arrangements ofstructural units in minerals.

Let us examine this issue for structural units consisting of octahedrally and tetrahedrally 716 coordinated cations (e.g., Mg, Al and transition-metal phosphates, sulfates, silicates). The Lewis 717 718 basicity of a structural unit is affected by its chemical composition, aggregate formal charge, and the coordination numbers of its cations and anions. We may approximately calculate the Lewis 719 basicity of a structural unit by proposing an average anion-coordination number (a more accurate 720 721 but more complicated method is available, see Hawthorne and Schindler 2008) and calculating 722 the number of bonds required from the interstitial complex to produce this number; dividing the charge of the structural unit by this number gives the Lewis basicity of that structural unit. As 723 724 will be apparent later, small inaccuracies in the calculated Lewis basicities do not affect the resulting arguments and understanding. 725 726 First, let us consider divalent-metal phosphates with structural units of the form  $M^{2+}_{N}(PO_4)(OH)_m$ . The variation in Lewis basicity as a function of N, the number of octahedrally 727 coordinated M<sup>2+</sup> cations, and m, the number of (OH) groups, is shown in Figure 14a for N = 2 to 728 4 and m = 0 to 25 (the method of calculation is explained in the Appendix). For 729  $M^{2+}_{2}(PO_{4})(OH)_{0-15}$ , the Lewis basicity increases with increasing values of m, the amount of 730 (OH), but levels off at m > 7. For  $M^{2+}_{3}(PO_4)(OH)_{0-25}$ , the Lewis basicity is constant at 0.50 v.u. 731 for all values of m. For  $M^{2+}_{4}(PO_4)(OH)_{0-25}$ , the Lewis basicity is somewhat above 0.50 v.u. at 732 high values of m, and increases dramatically with decreasing values of m. Above, we showed 733 734 that the Lewis basicity of the structural unit must be less than 0.30 v.u. if it is to satisfy the principle of correspondence of Lewis acidity-basicity. If we mark this boundary on Figure 14a, 735 we see that most compositions of the general form  $M^{2+}_{N}(PO_4)(OH)_m$  lie to the higher side of the 736 737 0.30 v.u. boundary, and hence cannot occur as structural units in minerals. Only for N = 2 and m

 $\leq 2$  do we have Lewis basicities less that 0.30 v.u.:  $[M^{2+}_{2}(PO_{4})(OH)_{2}]^{-}$  and  $[M^{2+}_{2}(PO_{4})(OH)]^{0}$ . If 738 739 we look at minerals (Table 3), we see selected minerals of this form: farringtonite, sarcopside and zavalíaite (N = 1.5, m = 0), althausite (N = 2, m = 1), holtedahlite (N = 2, m = 1) and 740 wagnerite (N = 2, m = 1). Moreover, there are no minerals of the form  $M^{2+}_{N}(PO_4)(OH)_m$  with 741  $Mg^{2+}$  or  $OH^{-}$  greater than two ions per phosphate group. 742 743 Next, let us consider divalent-metal sulfates with structural units of the form  $M^{2+}N(SO_4)(OH)_m$ . The variation in Lewis basicity as a function of N and is shown in Figure 14b 744 for N = 1 to 5 and m = 0 to 18. For  $M^{2+}(SO_4)(OH)_{0-4}$ , the Lewis basicity increases rapidly from 745 m = 4 to 6, but then levels off at higher values of m, the amount of (OH), and overlaps with the 746 curve for N = 3 for larger values of m. For  $M^{2+}_{3}(SO_4)(OH)_{4-18}$ , the Lewis basicity increases 747 rapidly from m = 4 to 6, but then gradually levels off with increasing values of m. For 748  $M^{2+}_{4}(SO_{4})(OH)_{0-18}$ , the Lewis basicity is constant at 0.50 v.u. for all values of m. For 749  $M^{2+}{}_{5}(SO_{4})(OH)_{0-18}$ , the Lewis basicity increases with decreasing values of m at large values of m 750 (>12). Only for N = 1, m  $\leq$  4 and N = 3, m  $\leq$  6 are the Lewis basicity values below the cut-off 751 value of 0.30 v.u. Selected minerals of this form are listed in Table 3: zincosite (N = 1, m = 0), 752 linarite and chlorothionite (N = 1, m = 2), antlerite (N = 3, m = 4) and christelite (N = 2, m = 3) 753 for  $(SO_4)_1$ , plus the synthetic Mg<sub>3</sub> $(SO_4)_2$  $(OH)_2$  (N = 1.5, m = 1 per  $(SO_4)$  group). 754 755

### 756 The distribution of mineral stoichiometries and the existence of rocks

The above calculations and Figure 14 suggest that many stoichiometries cannot exist as structures as there are strong bond-topological controls on their possible compositions and structures. Indeed, Figure 14 suggests that stoichiometries of structural units cannot exceed a value of N  $\approx$  4 as the resulting Lewis basicity of the structural unit is too high to form a stable structure. Let us examine this point using the stoichiometries of oxysalt minerals. Figure 15

762	shows a frequency diagram for minerals whose formulae involve octahedrally coordinated (M)
763	and tetrahedrally coordinated (T) cations. The number of minerals is a maximum at an M:T ratio
764	of 1:1, and falls off to close to zero beyond the range $4:1 \le M:T \le 1:4$ except for $M:T = \infty:1$ and
765	M:T = 1: $\infty$ . All the oxysalt minerals occur in the central region; oxides occur at M:T = $\infty$ :1 and
766	tetrahedron-framework structures (e.g., quartz, feldspars) occur at $M:T = 1:\infty$ .
767	Thus the stoichiometry of minerals is extremely restricted. What "happens" to all those
768	other chemical compositions that cannot occur as single minerals (the yellow regions in Fig. 15)?
769	Let us consider a simple example. The composition $[Mg_5(SO_4)(OH)_{12}]^{4-}$ has a Lewis basicity of
770	0.67 v.u. (Fig. 14b) and hence cannot form a structural unit. So what will happen to such a
771	composition?
772	$Mg_5(SO_4)(OH)_{12} \rightarrow Mg(SO_4)(H_2O) + 4Mg(OH)_2 + H_2O + O_2$
773	Kieserite + Brucite
774	It will crystallize as two different minerals, which in this example have Lewis basicities of 0.0
775	v.u. as there are no available large low-valence cations to form interstitial complexes. In the
776	presence of potential interstitial cations (e.g., Na, K), other minerals of appropriate stoichiometry
777	will form. Thus such stoichiometries as $M^{2+}_{N>2}(PO_4)(OH)_{m>2}$ will crystallize as mixtures of
778	minerals, i.e., as rocks. For silicates, those compositions with M:T < 1:4 will form rocks
779	containing significant amounts of framework silicates (e.g., granite, syenite), those compositions
780	with $4:1 \le M:T \le 1:4$ will form rocks dominated by ferromagnesian silicates (e.g., peridotite),
781	and those compositions with $M:T > 4:1$ will contain major amounts of oxides and
782	ferromagnesian silicates (e.g., iron formations). This is the principal reason why most chemical
783	compositions do not crystallize as single minerals, but form rocks.
784	

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# 786 HYDROGEN, POLYMERIZATION OF THE STRUCTURAL UNIT, AND THE DISTRIBUTION OF STRUCTURAL COMPLEXITY OF MINERALS WITHIN THE EARTH 787 Above, we saw that both $(H_2O)$ and (OH) groups are extremely polar: on the O side, each 788 789 functions as an anion, whereas on the H side, each functions as a cation. The metal(M)-O bonds 790 are commonly relatively strong [ $\sim 0.40$ v.u. for (H<sub>2</sub>O); 0.80 v.u. for (OH)], whereas the H...O (hydrogen) bonds are much weaker [~0.20 v.u. for both (H<sub>2</sub>O) and (OH)]. Hence the M-O bonds 791 are commonly part of the structural unit, whereas the hydrogen bonds are *not* part of the 792 structural unit (Hawthorne 1985). The net result of this asymmetric arrangement of bond 793 valences is commonly to terminate the structural unit at the (H<sub>2</sub>O) and (OH) groups. 794 795 Consider the structure of newbervite (Sutor 1967), Mg<sub>3</sub>(PO<sub>3</sub>OH)(H<sub>2</sub>O)<sub>3</sub> (Fig. 16), in which (OH) and (H<sub>2</sub>O) play significant roles in limiting polymerization of polyhedra in the 796 structure. Newberyite contains an acid-phosphate group, (PO<sub>3</sub>OH). Each tetrahedron links to 797 798 three (Mg $\phi_6$ ) octahedra, forming a sheet in the **ac** plane (Fig. 16), and the fourth vertex of the tetrahedron points in the $\pm \mathbf{b}$ direction. In a (PO<sub>4</sub>) group, the fourth vertex of the tetrahedron 799 800 would link to another polyhedron of the structural unit in order to satisfy the valence-sum rule at that anion. However, in newberyite, H is attached to the O anion at this vertex, and the valence-801 sum rule prevents linkage to another tetrahedron or octahedron, preventing polymerization of the 802 structural unit in the **b** direction through the phosphate group. Newberyite also contains 803 $\{MgO_3(H_2O)_3\}$ octahedra that are linked by the tetrahedra into a sheet (Fig. 16) by each 804 tetrahedron sharing three vertices with adjacent tetrahedra. This linkage leaves three vertices of 805 806 the octahedron that can potentially link in the third dimension to form a framework structure. However, each of the O anions occupying these three vertices also link to two H atoms, forming 807 (H<sub>2</sub>O) groups; the H atoms satisfy the bond-valence requirements of the anions at these three 808 809 vertices, and prevent linkage in the **b** direction.

810	Although the presence of H prevents all intra-unit linkage at the (OH) and (H <sub>2</sub> O) groups
811	in newberyite, this is not necessarily the case in all H-bearing minerals: both (OH) and (H <sub>2</sub> O) can
812	allow linkage of a structural unit in some directions and prevent such linkage in other directions.
813	The structural unit in artinite, [Mg <sub>2</sub> (CO <sub>3</sub> )(OH) <sub>2</sub> (H <sub>2</sub> O) <sub>3</sub> ] (Akao and Iwai 1977), consists of a
814	ribbon of edge-sharing (MgO <sub>6</sub> ) octahedra, flanked by (CO <sub>3</sub> ) triangles (Fig. 17). In the centre of
815	the ribbon, the anions bond to three Mg cations, receiving $0.36 \times 3 = 1.08 \text{ v.u.}$ from Mg. The
816	additional bond valence required by these anions is provided by their associated H atoms which
817	weakly hydrogen-bond (bond-valence approximately 0.08 v.u.) to an adjacent ribbon. The (OH)
818	group prevents linkage of the structural unit in the Z- direction but allows linkage in the X- and
819	Y-directions. The anions along the edge of the ribbon bond to either one Mg, two Mg, or one Mg
820	and one C, with incident bond-valence values of $\sim 0.3$ , 0.6 and 1.7 v.u. The first two anions must
821	be (H <sub>2</sub> O) groups and cannot propagate linkage of the structural unit. The (H <sub>2</sub> O) group bonded to
822	one Mg prevents further polymerization of the structural unit in all three directions, whereas the
823	(H <sub>2</sub> O) group bonded to two Mg atoms allows polymerization of the structural unit in the Y-
824	direction but prevents polymerization in the X- and Z-directions. Thus in artinite, the (OH)
825	groups allow polymerization of the structural unit in two directions, the two types of $(H_2O)$
826	group allow polymerization in one and no directions, respectively, and all linkage between
827	structural units is through hydrogen bonding via the (OH) and (H <sub>2</sub> O) groups of the structural
828	unit.
829	In summary, H as (OH) and (H <sub>2</sub> O) can control <i>the dimensional polymerization of a</i>
830	structural unit, limiting it in one or more directions. This is the principal single chemical feature

that leads to the amazing structural diversity in oxygen-based minerals. Moreover, the

distribution of H throughout the Earth, together with the anharmonic nature of the hydrogen
bond, is a major factor in accounting for the systematic distribution of mineral species from thecore to the surface of the Earth.

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#### THE METHOD OF MOMENTS

There is little intuitive connection between the essential features of a crystal structure, the relative positions of the atoms and the disposition of the chemical bonds, and the usual methods for deriving the electronic energy density-of-states (Hawthorne 2012). However, the electronic energy density-of-states may be derived from the bond-topological aspects of a structure using the method of moments (Burdett et al. 1984). I will give a brief outline of the method; the reader should consult their paper for mathematical details.

A simple way to consider the electronic structure of a molecule is to construct the molecular-orbital wavefunction as a linear combination of atomic orbitals. These wavefunctions are eigenstates of an effective one-electron Hamiltonian,  $H^{\text{eff}}$ , that may be written as  $H^{\text{eff}}\psi = E\psi$ where *E* is the energy associated with  $\psi$ . The total electron energy of the state described by the wavefunction is

848 
$$E = (\int \psi^* H^{\text{eff}} \psi \, d\tau) / (\int \psi^* \psi \, d\tau) = (\langle \psi^* H^{\text{eff}} \psi \, d\tau \rangle) / (\langle \psi^* \psi \rangle)$$
(1)

849 where the integration is over all space,  $H^{\text{eff}}$  is an effective one-electron Hamiltonian that may be 850 written as  $H^{\text{eff}}\psi = E\psi$  where *E* is the energy associated with  $\psi$ , and the molecular-orbital 851 wavefunction is written as  $\psi = \sum_i c_i \varphi_i$  where  $\{\varphi_i\}$  are the valence orbitals of the atoms and  $c_i$  is the 852 contribution of a specific atomic orbital to a specific molecular orbital (e.g., Gibbs 1980).

853 Substitution for  $\psi(=\Sigma c_i \varphi_i)$  gives

854 
$$\mathbf{E} = (\sum_i \sum_j c_i c_j (\langle \varphi_i | H^{eff} | \varphi_j \rangle) / (\sum_i \sum_j c_i c_j \langle \varphi_i | \varphi_j \rangle)$$
(2).

Equation (2) may be simplified thus: (a)  $\langle \varphi_i | \varphi_j \rangle$  is the overlap integral between atomic orbitals on different atoms, and is written as  $S_{ij}$ , which is always  $\leq 1$ ; where i = j,  $\langle \varphi_i | \varphi_j \rangle = 1$  for a

normalized (atomic) basis set of orbitals; (b)  $\langle \varphi_i | H^{eff} | \varphi_j \rangle = H_{ii}$ ; this represents the energy of an electron in orbital  $\varphi_i$  and can be approximated by the orbital ionization potential; (c)  $\langle \varphi_i | H^{eff} | \varphi_j \rangle$  $= H_{ij}$ ; this is the resonance integral. Minimizing the energy with respect to the coefficients  $c_i$ , equation (2) gives the molecular-orbital energies. The eigenvalues of the following *secular determinant equation* give the molecular-orbital energy levels:

862 
$$|H_{ij} - S_{ij}E| = 0$$
 (3).

The Hückel approximation (Trinajstic 1983) best shows the topological content of this approach: For the  $p\pi$  orbitals, all  $H_{ii}$  values are set equal to  $\alpha$ , all  $H_{ij}$  are set equal to  $\beta$ , and all  $S_{jj}$ ( $i \neq j$ ) are set equal to zero. The expanded secular determinant equation for the square molecule of Figure 1b is as follows:

$$\begin{vmatrix} \alpha - E & \beta & 0 & \beta \\ \beta & \alpha - E & \beta & 0 \\ 0 & \beta & \alpha - E & \beta \\ \beta & 0 & \beta & \alpha - E \end{vmatrix} = 0$$

$$(4).$$

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Compare the structure of Figure 1b with the matrix entries in equation (4). Where atoms are bonded together (i.e., atoms 1 and 2 in Fig. 1b), there is a nonzero value at the corresponding (1,2) entry in the secular determinant; where atoms are not bonded together (i.e., atoms 1 and 3 in Fig. 1b), the corresponding entry in the secular determinant (1,3) is zero. Note also the correspondence of the zero off-diagonal matrix entries in the adjacency matrix of the graph of this molecule (Fig. 2) with the zero off-diagonal matrix entries in equation (4).

We cannot use this sort of calculation to deal with a crystal containing approximately Avogadro's number of atoms. Instead, we use Bloch orbitals (Ziman 1965) which assume a unit cell and constrain the orbital content of the unit cell to the translational periodicity of the crystal. Using the special-points method, the secular determinant is solved at a representative set of points within the Brillouin zone, giving a representative sampling of the orbital energy levels that

879 may be smoothed to give the electronic-energy density-of-states. The total orbital energy is 880 obtained by integrating the electronic energy density-of-states up to the Fermi level. To solve equation (4), we diagonalize the Hamiltonian matrix. The trace of this matrix may be 881 written as follows: 882

$$Tr(H^{n}) = \sum_{i} \sum_{j,k\dots n} H_{ij}H_{jk}\dots H_{ni}$$
(5).

883

A topological interpretation of one term in this sum is shown in Figure 18.  $H_{ij}$  is the interaction 884 integral between orbitals *i* and *j*; we may simplify the situation without loss of topological 885 content by adopting the Huckel approximation:  $H_{ij} = \beta$  where the atoms are bonded,  $H_{ij} = 0$ 886 where the atoms are not bonded, and  $\alpha = 0$  where i = j. In equation (5), as each single term  $\{H_{ij}\}$ 887  $H_{ik} \dots H_{ni}$  is a product, the term is nonzero only if all individual  $H_{ii}$  values in the term, e.g., 888 889  $\{H_{12}H_{23}H_{34}H_{41}\}$ , are nonzero. The last  $H_{ii}$  in each product is  $H_{ni}$ , the interaction between the nth orbital and the first orbital, and hence the product  $\{H_{ij}, H_{jk}, \dots, H_{ni}\}$  represents a closed path in the 890 graph of the orbitals (molecule). If one (or more) of the terms in the product is zero (e.g.,  $H_{31}$  in 891 Fig. 18), that product is zero, i.e.,  $\{H_{12}H_{23}H_{31}\} = 0$ , and does not contribute to the trace of the 892 Hamiltonian matrix. Hence the double-summation in equation (5) contains all closed paths 893 through the graph of (the orbital structure of) the array of atoms. 894 The trace of a matrix remains invariant under diagonalization, and thus

895

$$Tr(H^n) = Tr(E^n) = \mu_n \tag{6}$$

896

where E is the diagonal matrix of eigenvalues (energy levels) and  $\mu_n$  is the nth moment of E 897 (Burdett et al. 1984), denoted by 898

$$\mu_n = \sum_i E_i^n \tag{7}$$

The density-of-states may be obtained by inverting the collection of moments  $\{\mu_n\}$  (Burdett et al. 1984). The result is that we can evaluate  $Tr(H^n)$  directly from the bond topology, and, in so doing, derive the electronic energy density-of-states.

- 903 This method generalizes to infinite systems (i.e., crystals) in a straightforward manner.
- We may define the nth moment of E as

$$\mu_n = \int E^n \rho(E) dE$$

906 where  $\rho(E)$  is the density-of-states of the crystal. In this case, the moments may be evaluated in 907 principle as above and inverted to give the electronic energy density-of-states.

Burdett (1986) introduced an extremely important idea: The energy difference between two structures may be expressed in terms of the first few disparate moments of their electronicenergy density-of-states. This means that the most important energetic differences between two structures involve the most local bond-topological differences between those structures. Also, in structures with bonds of different strength, each edge is weighted according to the strength of the analogous bond. Thus, closed paths of strongly bonded atoms will contribute more to the electronic energy density-of-states than closed paths of weakly bonded atoms.

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#### 916 Low-order moments and crystal chemistry

The number of edges in a path through the bonded atoms in a structure is the *moment* of that path, and each path corresponds to a crystal-chemical feature of the structure. We will now consider the structural features corresponding to the lower-order moments that are the most energetically important. A *zero-moment path* has no steps and corresponds to remaining still (called a "walk in place"); as such, it specifies the identity of the atom at that vertex of the graph of the structure. Thus the complete set of zero-moment paths defines the chemical composition of the structure. A *second-moment path* is a walk from one vertex to an adjacent vertex and back

(8)

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Cite as Authors (Year) Title. American Mineralogist, in press. (DOI will not work until issue is live.) DOI: http://dx.doi.org/10.2138/am-2015-5114	9/24
again, and the set of second-moment paths from a single vertex defines the coordination nur	nber
of the atom corresponding to that vertex. A <i>fourth-moment path</i> is a walk from an atom (e.g.	., a
cation) to an anion to another cation to another anion and back to the first cation, and specifi	ies

- the linkage of two coordination polyhedra. Higher-moment paths describe more complicated 927
- 928 linkages of polyhedra, but these are less important from an energetic perspective than the low-
- moment linkages. Here is our energetic rationale for traditional crystal chemistry: we focus on 929
- 930 chemical composition (zero moment), coordination number (second moment), and local linkage
- 931 between coordination polyhedra (fourth moment) as the most important differences between
- 932 structures, as differences in low-order moments are the most energetically important differences
- 933 between structures.
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#### **Mineral reactions** 935

The moments approach tells us that the important energetic differences between two 936 structures are the most local topological differences between the structures. What does this mean 937 in terms of mineral reactions? 938

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**Zero-moment changes.** Zero-moment changes involve changes in the chemical 940 941 composition of the system, i.e., a reaction in which the constituents are not conserved. This can 942 be a *metasomatic reaction* and involve open-system behavior.

943

944 Second-moment changes. Second-moment changes involve changes in coordination number in the constituent phases. A change in coordination number usually involves a 945 946 discontinuous reaction. Consider:

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948 Forsterite = Periclase + Quartz

 ${}^{[6]}Mg_{2}{}^{[4]}Gi^{[4]}O_{4} = 2{}^{[6]}Mg^{[6]}O + {}^{[4]}Gi^{[2]}O_{2}$ 

in which I include the coordination numbers of all the constituent atoms. Whereas the coordination numbers of <sup>[6]</sup>Mg and <sup>[4]</sup>Si are conserved in the reaction, the coordination numbers of O are not conserved. The lowest-moment changes in this reaction involve the changes in coordination number of O, and these changes are the major driver of this reaction, as these are the lowest-moment differences involved in the reaction. We may also express the driving force of this reaction in terms of the enthalpy of reaction,  $\Delta H$ , suggesting a correlation between the changes in coordination number and the enthalpy of reaction. Consider the general reaction

957 
$${}^{[6]}Mg_{m}{}^{[4]}Si_{n}O_{(m+2n)} = mMgO + nSiO_{2}$$

for m,n = 2,1; 3,2; 1,1; 1,2; 2,5; 1,3.  $\Delta$ H of reaction may be calculated with the model of Aja et 958 al. (1992), using fictive enthalpies of formation. However, (change in) coordination number is an 959 960 intensive variable whereas enthalpy of formation is an extensive variable. We must transform the enthalpy of formation into an intensive variable, and I do this by dividing the enthalpy of 961 962 formation by the molecular weight of the reactant to produce the intensive variable  $\Delta H/MW$ 963 which I will call the *reduced enthalpy* of formation. There is a strong correlation between the reduced enthalpy of formation and the change in anion-coordination number (details of this 964 calculation will be given in a later paper) through the reaction (Fig. 19) in accord with the 965 influence of coordination number on the energetics of structures indicated by the moment 966 arguments given above. A similar relation for the hydrated magnesium sulfates  $Mg(SO_4)(H_2O)_n$ 967 968 (where n = 0-7, 11) was shown by Hawthorne and Sokolova (2012).

These second-moment changes throw considerable light on why the additive-fictive approach to predicting enthalpies of formation from oxides works so well. The relation  $[^{6]}Mg_m^{[4]}Si_nO_{(m+2n)} = mMgO + nSiO_2$  has no experimentally determined quantities; the

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972	coordination numbers are assumed (and hence have no experimental uncertainty attached to
973	them) and the relation is exact. Consider the relation $\Delta H(^{[6]}Mg_m^{[4]}Si_nO_{(m+2n)}) = m\Delta H(MgO) + M(MgO) + M(Mg$
974	$n\Delta H(SiO_2)$ where $\Delta H(MgO)$ and $\Delta H(SiO_2)$ are the fictive enthalpies for MgO and SiO <sub>2</sub> ,
975	respectively. This relation is also exact; $\Delta H(^{[6]}Mg_m^{[4]}Si_nO_{(m+2n)})$ is calculated from $\Delta H(MgO)$ and
976	$n\Delta H(SiO_2)$ . Hence $\Delta H(^{[6]}Mg_m^{[4]}Si_nO_{(m+2n)})$ must correlate with change in anion-coordination
977	number; this is an algebraic requirement. However, there is no algebraic requirement that change
978	in anion-coordination number must correlate with experimental enthalpies of formation from the
979	oxides. We know that the enthalpies calculated from the fictive enthalpies of the oxides correlate
980	with their experimental analogues; this is the whole point of using fictive enthalpies. Therefore
981	we may conclude that changes in anion-coordination number correlate with experimental
982	enthalpies of formation from the oxides, in accord with our prediction from the moments
983	approach to the electronic-energy density-of-states. In this regard, I should also emphasize that
984	using enthalpies of formation calculated from fictive enthalpies does not replace the
985	measurement of enthalpies of formation. One expects subtle differences in energetics with
986	higher-moment changes in bond topology, and this will not be reflected in enthalpies of
987	formation calculated using the fictive approach; they will only be apparent in measured
988	enthalpies of formation.

989

Fourth-moment changes. Fourth-moment changes involve maintaining chemical
composition and both cation- and anion-coordination numbers while changing the identities of
next-nearest-neighbor atoms. Such changes hence involve the nature of local (short-range)
clusters of ions. Such changes are common in amphiboles (e.g., Hawthorne et al. 1996b, 1996c,
1997, 2000b; Della Ventura et al. 1999; Hawthorne and Della Ventura 2007), and the short-range

995	versio	on of the valence-sum rule (Hawthorne 1997, see above) suggests that such short-range		
996	order	should be common in all solid solutions involving polyvalent substitutions.		
997	Major chemical variations in amphiboles in metabasic rocks involve the change from			
998	tremo	lite, $\Box Ca_2Mg_5Si_8O_{22}(OH)_2$ , to sadanagaite, NaCa <sub>2</sub> (Mg <sub>3</sub> Al <sub>2</sub> )(Si <sub>5</sub> Al <sub>3</sub> )O <sub>22</sub> (OH) <sub>2</sub> , with		
999	increa	using grade of metamorphism. In this reaction, the bond topology of the amphibole is		
1000	conserved, and any energetic differences with regard to the amphiboles involve atom identities			
1001	and their relative locations, i.e., short-range order-disorder. End-member tremolite is completely			
1002	ordered whereas end-member sadanagaite must show extensive short-range order/disorder. Such			
1003	short-range order/disorder must have a major effect on the energetics of the resulting minerals			
1004	and th	neir reactions with other phases.		
1005				
1006		SUMMARY		
1007		The theoretical approach outlined above examines the structure and chemical		
1008	comp	osition of minerals based on their bond topology, aspects of graph theory and bond-valence		
1009	theory	y, and the moments approach to the electronic-energy density of states. Below, I identify		
1010	the pr	incipal features of this approach, and some of its uses:		
1011	[1]	An arrangement of atoms and chemical bonds may be represented by a weighted		
1012		polychromatic digraph, and the handshaking principle may be used to examine many		
1013		aspects of atom coordination and the linkage of coordination polyhedra.		
1014	[2]	The moments approach to the electronic-energy density-of-states provides a bond-		
1015		topological interpretation of the energetics of a structure.		
1016	[3]	When comparing structures, the most important structural differences involve the first		
1017		few disparate moments of the electronic-energy density-of-states.		

...

1018	[4]	We may classify chemical reactions according to the lowest-order moment of the
1019		electronic-energy density-of-states that is conserved, which allows us to identify the
1020		principal structural changes that drive chemical change: (a) coordination number for
1021		discontinuous reactions, and (b) short-range order for continuous reactions.

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- [5] It may be shown that the quantitative aspects of bond-valence theory arise from the topological (or graphical) characteristics of structures as arrangements of atoms and 1023 chemical bonds. 1024
- [6] The principle of correspondence of Lewis acidity-basicity states that stable structures will 1025

form when the Lewis-acid strength of the interstitial complex closely matches the Lewis-1026

1027 base strength of the structural unit, and allows us to examine the factors that control the

chemical composition and aspects of the structural arrangement of minerals. 1028

- 1029 [7]  $(H_2O)$  groups in the structural unit limit the polymerization of the structural unit in one or
- more directions, controlling the polymerization of the structural unit. This is a major 1030
- factor affecting structural diversity in oxygen-based minerals and the systematic 1031
- distribution and relative complexity of mineral species from the core to the surface of the 1032
- Earth. 1033
- [8] Interstitial ( $H_2O$ ) groups may (1) satisfy the bond-valence requirements around an 1034
- 1035 interstitial cation where there are insufficient adjacent anions to do so from neighboring
- structural units, or (2) moderate the Lewis acidity of the interstitial complex and affect 1036
- the stability of a chemical composition through the operation of the principle of 1037
- 1038 correspondence of Lewis acidity-basicity.
- 1039

- 1040
- 1041

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1042	ACKNOWLEDGEMENTS
1043	I thank Drs. Elena Sokolova and Michael Schindler for their long-term influence on my
1044	ideas, and David Brown and an anonymous reviewer for their comments that clarified the text in
1045	many places. It gives me pleasure to acknowledge support by a Canada Research Chair in
1046	Crystallography and Mineralogy and a Discovery grant from the Natural Sciences and
1047	Engineering Research Council of Canada, and by grants from the Canada Foundation for
1048	Innovation.
1049	

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### 1050

#### APPENDIX

- 1051 Consider the structural unit  $[Mg_N(PO_4)(OH)_m]$ .
- 1052 The charge on the structural unit is 2N 3 m.
- 1053 The number of bonds involving the structural unit is 6N (for  ${}^{[6]}Mg_N$ ) + 4 (for the (PO<sub>4</sub>) group) +
- 1054 2m (for H, assuming a coordination number of [2]) = 6N + 4 + 2m.
- 1055 If we assume an ideal coordination number of [4] for oxygen, the number of bonds needed to
- 1056 produce such a coordination = 4(4 + m).
- 1057 The number of bonds needed from the interstitial complex is the difference of these two values: 1058 4(4 + m) - (6N + 4 + 2m) = 12 + 2m - 6N.
- 1059 The Lewis basicity of the structural unit is the charge divided by the number of bonds needed 1060 from the interstitial complex: (2N - 3 - m)/(12 + 2m - 6N).
- 1061 For N = 2, this expression reduces to (1 m)/2m, and for m = 1, 2, 4, 8, the Lewis basicities =
- 1062 0.00, 0.25, 0.38, 0.44 *vu*, respectively.
- 1063 For N = 3, this expression reduces to (3 m)/(2m 6) = 0.50 vu independent of the value of m.
- 1064 For N = 4, this expression reduces to (5 m)/(2m 12), and for m = 8, 12, 16, 20, the Lewis
- 1065 basicities = 0.75, 0.58, 0.55, 0.53 vu, respectively.
- 1066 The calculations for the structural unit  $[Mg_N(SO_4)(OH)_m]$  are similar, except that we assume an
- ideal coordination number of [3] for oxygen because of the higher bond-valence of the S-O bond.
- 1068

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1261	FIGURE CAPTIONS
1262	<b>Figure 1.</b> (a) A weighted polychromatic digraph with the coloured vertex set {1, 2, 3, 4}
1263	and the directed weighted edge set {12, 32, 34, 14}; (b) a simple idealized square molecule
1264	consisting of four atoms labelled 1-4; different chemical types of atoms are indicated by
1265	different colours.
1266	Figure 2. The adjacency matrix corresponding to the graph in Fig. 1a.
1267	Figure 3. Graphical representation of polyhedron clusters; octahedra are shown in
1268	yellow, tetrahedra are shown in orange. Each cluster of polyhedra is represented by a graph in
1269	which the yellow vertices represent octahedra, the orange vertices represent tetrahedra, and the
1270	edges represent the number of vertices common to pairs of polyhedra: (a) $(M\phi_6)_2$ ; (b) $[M_2\phi_{11}]$ ;
1271	(c) $[M_2\phi_{10}]$ ; (d) $[M_2(TO_4)_2\phi_8]$ ; (e) graphical isomers of $[M(TO_4)_2\phi_4]$ .
1272	Figure 4. The two components of the bond-valence model: Bond-Valence Theory (left)
1273	and Bond-Valence Curves (right).
1274	Figure 5. The three axioms of Bond-Valence Theory.
1275	Figure 6. The valence-matching principle.
1276	<b>Figure 7.</b> The bond-valence structure of the $(SO_4)^{2-}$ oxyanion in thenardite, with the
1277	individual bond valences shown in valence units (after Hawthorne 1994).
1278	Figure 8. Partitioning of the crystal structure of botryogen,
1279	$Mg_2(H_2O)_{12}[Fe^{3+}_2(SO_4)_4(OH)_2]$ (H <sub>2</sub> O) <sub>2</sub> , into two units, the strongly bonded structural unit
1280	(shown as coloured polyhedra) and the weakly bonded interstitial complex (shown as individual
1281	atoms and chemical bonds). Pink tetrahedra: (SO <sub>4</sub> ) groups; yellow octahedra: (Fe <sup><math>3+</math></sup> O <sub>6</sub> )
1282	octahedra; large orange circles: O atoms; small blue circles: Mg atoms; black lines: Mg-O bonds.
1283	Figure 9. The bond-valence structure around (H <sub>2</sub> O) as a function of local bond-topology;
1284	(a) a cation, C (green) bonded to an anion, S (yellow) with bond valence v v.u.; (b) a cation

1285	bonded to an $(H_2O)$ group (O: orange; H: black) with bond valence v v.u.; the H atoms
1286	hydrogen-bond to the anions S with bond valence v/2 v.u. per bond; (c) two cations bonded to an
1287	(H <sub>2</sub> O) group with bond valence v v.u. per bond; the H atoms hydrogen-bond to the anions S with
1288	bond valence v v.u. per bond; (d) two H atoms hydrogen-bonded to an (H <sub>2</sub> O) group with bond
1289	valence v v.u. per bond; the H atoms of the $(H_2O)$ group hydrogen-bond to the anions S with
1290	bond valence v v.u. per bond.
1291	Figure 10. Representation of the crystal structure of lizardite, showing the polar nature of
1292	the structural unit; yellow: Mg octahedra; lilac: Si tetrahedra; red circles: H atoms; thick black
1293	lines: O <sub>donor</sub> -H bonds; broken lines: hydrogen bonds. The acidic (+) and basic (-) parts of the
1294	structural unit are indicated.
1295	Figure 11. Correlation between the CDA of structural units and the average number of
1296	bonds from the interstitial complex and adjacent structural units, $\langle NB \rangle_{in}$ , to O-atoms in the
1297	corresponding structural units of sulfate minerals. The upper and lower bounds of the distribution
1298	are used to define the characteristic range in the number of bonds accepted by a specific
1299	structural unit.
1300	Figure 12. Variation in Lewis acidity of a general interstitial complex as a function of the
1301	number of transformer (H <sub>2</sub> O) groups per cation. The lines shown are for interstitial cations with
1302	formal charges and coordination numbers shown to the left of the plot. From Hawthorne and
1303	Schindler (2008).
1304	Figure 13. Variation in Lewis acidity with the number of transformer (H <sub>2</sub> O) groups per
1305	cation for different interstitial-cation charges and coordination numbers for a general interstitial
1306	complex; the range in Lewis basicity of the structural units for selected sulfate minerals are
1307	shown by the yellow fields: (a) $[M^{2+}(SO_4)_2(H_2O)_4]^{2-}$ ; (b) $[Fe^{3+}(OH)(SO_4)_2]^{2-}$ .

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Figure 14. (a) Lewis basicity of structural units of the form  $Mg_N(T^{5+}O_4)(OH)_n$  ( $T^{5+} = P$ , As, V) as a function of stoichiometry for N = 2, 3, 4; m = 1–24. (b) Lewis basicity of structural units of the form  $Mg_N(T^{6+}O_4)(OH)_n$  ( $T^{6+} = S$ , Cr) as a function of stoichiometry for N = 1, 3, 4, 5; m = 1–18.

1312 **Figure 15.** The distribution of mineral stoichiometries with regard to the ratio of

1313 octahedrally coordinated cations (M) and tetrahedrally and triangularly coordinated cations (T).

1314 The bars in pink show the numbers of minerals with M:T ratios approximately equal to 4:1,

1315 3:1...1:3, 1:4. The yellow areas denote compositions not corresponding to single minerals.

1316 **Figure 16.** The crystal structure of newberyite, Mg<sub>3</sub>(PO<sub>3</sub>OH)(H<sub>2</sub>O)<sub>3</sub>, projected onto

1317 (010); Mg octahedra are shown in yellow, P tetrahedra are shown in lilac, H atoms are shown as

1318 red circles, O<sub>donor</sub>-H bonds are shown as thick black lines.

**Figure 17.** The crystal structure of artinite,  $[Mg_2(CO_3)(OH)_2(H_2O)_3]$ , projected onto

1320 (001); Mg octahedra are shown in yellow, C triangles are shown in lilac, H atoms are shown as

1321 red circles, O<sub>donor</sub>-H bonds are shown as thick black lines.

1322 Figure 18. Interpretation of paths through the molecule shown in Figure 1b; the path  $1 \rightarrow$ 

1323  $2 \rightarrow 3 \rightarrow 4$  contains only non-zero  $H_{ij}$  terms and contributes to the trace of the matrix, whereas

the path  $1 \rightarrow 2 \rightarrow 4$  contains a zero  $H_{ij}$  term  $(H_{31})$  and does not contribute to the trace of the matrix.

Figure 19. Variation in reduced enthalpy of formation (from the oxides) *versus* change in anion-coordination number through the reaction  ${}^{[6]}Mg_m{}^{[4]}Si_nO_{(m+2n)} = m{}^{[6]}MgO + n{}^{[4]}SiO_2$ . Units on the ordinate are kJ/mol/Dalton.

Li	0.21	Sc	0.49	Cu <sup>2+</sup>	0.45
Be	0.50	Ti <sup>3+</sup>	0.50	Zn	0.35
В	0.87	Ti <sup>4+</sup>	0.67	Ga	0.65
С	1.35	V <sup>3+</sup>	0.50	Ge	0.89
N <sup>5+</sup>	1.67	V <sup>5+</sup>	1.20	As <sup>5+</sup>	1.13
Na	0.16	Cr <sup>3+</sup>	0.50	Se <sup>6+</sup>	1.50
Mg	0.33	Cr <sup>6+</sup>	1.50	Rb	0.12
Al	0.57	Mn <sup>2+</sup>	0.34	Sr	0.23
Si	1.00	Mn <sup>3+</sup>	0.52	Sn4+	0.68
Р	1.25	Mn <sup>4+</sup>	0.67	Sb <sup>5+</sup>	0.83
S	1.50	Fe <sup>2+</sup>	0.34	Te <sup>6+</sup>	1.00
Cl <sup>7+</sup>	1.75	Fe <sup>3+</sup>	0.50	Cs	0.11
Κ	0.13	Co <sup>2+</sup>	0.35	Ва	0.20
Са	0.27	Ni <sup>2+</sup>	0.34	Pb <sup>2+</sup>	0.20

 TABLE 1.
 Lewis acid strengths (vu) for cations

Values taken from Brown (2002), except V<sup>5+</sup> (Schindler et al., 2000) and Pb<sup>2+</sup> which was estimated from several oxysalt mineral structures.

TABLE 2.	Lewis basicities ( <i>vu</i> ) for selected oxyanions		
(BO <sub>3</sub> ) <sup>3-</sup>	0.33	(CO <sub>3</sub> ) <sup>2-</sup>	0.22
(SiO <sub>4</sub> ) <sup>4-</sup>	0.33	(NO <sub>3</sub> ) <sup>3-</sup>	0.11
(AIO <sub>4</sub> ) <sup>3-</sup>	0.42	(VO <sub>4</sub> ) <sup>3-</sup>	0.25
(PO <sub>4</sub> ) <sup>3-</sup>	0.25	(SO <sub>4</sub> ) <sup>2-</sup>	0.17
(AsO <sub>4</sub> ) <sup>3-</sup>	0.25	(CrO <sub>4</sub> ) <sup>2-</sup>	0.17

TABLE 3.	Selected minerals with structural units of the form $M^{2+}{}_{N}(TO_{4})(OH)_{m}$ , T = P, S		
Farringtonite	e Mg <sub>3</sub> (PO <sub>4</sub> ) <sub>2</sub>		
Sarcopside	$Mn_3(PO_4)_2$		
Zavalíaite	$Mn_3(PO_4)_2$		
Althausite	$Mg_2(PO_4)(OH)$		
Holtedahlite	Mg <sub>2</sub> (PO <sub>4</sub> )(OH)		
Wagnerite	Mn <sub>2</sub> (PO <sub>4</sub> )F		
Zincosite	Zn(SO <sub>4</sub> )		
Linarite	Pb[Cu(SO <sub>4</sub> )(OH) <sub>2</sub> ]		
Chlorothion	ite $K_2[Cu(SO_4)CI_2]$		
Antlerite	[Cu <sub>3</sub> (SO <sub>4</sub> )(OH) <sub>4</sub> ]		
Christelite	$Zn(H_2O)_4[Zn_2Cu_2(SO_4)_2(OH)_6]$		









THE VALENCE-SUM RULE

THE SUM OF THE BOND-VALENCES INCIDENT AT ANY ION IS EQUAL TO THE ION VALENCE

THE LOOP RULE

THE SUM OF THE DIRECTED BOND-VALENCES AROUND ANY CLOSED LOOP OF BONDS IN THE STRUCTURE IS ZERO

THE VALENCE-MATCHING PRINCIPLE

FOR A CHEMICAL BOND TO FORM, THE LEWIS ACIDITY AND LEWIS BASICITY OF THE CONSTITUENT IONS MUST MATCH



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FIGURE 14

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