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Revision 2

**Time-Resolved Synchrotron X-ray Diffraction Study of the Dehydration Behavior
of Chalcophanite**

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19 **ABSTRACT**

20 Time-resolved synchrotron X-ray powder diffraction data were used to investigate the
21 dehydration behavior of the chalcophanite ($\text{ZnMn}_3\text{O}_7 \cdot 3\text{H}_2\text{O}$) structure from 300 to 1060 K.
22 Rietveld refinements revealed two obvious phase changes, at ~ 450 K and ~ 950 K, corresponding
23 to the dehydration of chalcophanite followed by transformation to a spinel structure – Mn-
24 hetaerolite. Only small changes were observed in the chalcophanite unit-cell from RT to ~ 438 K;
25 the volume increased by $\sim 0.8\%$, mostly caused by thermal expansion of $\sim 0.5\%$ along c . Above
26 ~ 427 K, the interlayer water molecules were lost, resulting in a collapse of the interlayer spacing
27 from ~ 7 Å to ~ 4.8 Å. The newly formed anhydrous phase (ZnMn_3O_7) retained chalcophanite's $R\bar{3}$
28 3 space group and a dimension, but the c dimension decreased from ~ 21 Å to 14.3 Å, and the Zn
29 coordination changed from octahedral to tetrahedral. Above ~ 775 K the anhydrous chalcophanite
30 began to transform to a spinel structure, corresponding to a Mn-rich hetaerolite
31 $[(\text{Zn}_{0.75}\text{Mn}^{2+}_{0.25})\text{Mn}^{3+}_2\text{O}_4]$. By ~ 973 K the transformation was complete. The diffraction patterns
32 did not show a significant increase in background during the transformation, indicating that the
33 reaction did not involve transient amorphization. The phase change was likely triggered by loss of
34 1.25 of seven O atoms in the original anhydrous chalcophanite structure with a corresponding
35 reduction of Mn^{4+} to Mn^{3+} and Mn^{2+} .

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INTRODUCTION

38 Chalcophanite is a hydrated zinc-manganese oxide that is one of a family of
39 predominantly tetravalent Mn oxide minerals with layer structures (phylломanganates) that also
40 includes birnessite, ranciéite, takanelite, and lithiophorite. It is constructed of sheets of Mn⁴⁺-O
41 octahedra with the Zn cations and water molecules located between the layers. It is found in
42 oxidized Mn deposits that also contain Zn, as exemplified by its type locality – Sterling Hill in
43 the Franklin Mining District, New Jersey. Recently, Mg- and Ni-rich structural analogs to
44 chalcophanite have been reported – jianshuiite (Guiyin et al. 1992) and ernienickelite (Grice et
45 al. 1994), respectively.

46 Recent interest in the behavior of chalcophanite is motivated by its structural similarity to
47 members of the birnessite family. Birnessite-like layer minerals are common Mn oxides in soils,
48 stream deposits, and ocean-floor ferromanganese crusts and nodules (Taylor et al. 1964;
49 McKenzie 1976; Burns and Burns 1977; Potter and Rossman 1979; Post 1999). Birnessite-like
50 layer structures are also used as cathodic materials in rechargeable Li batteries and are being
51 developed as ion-exchange compounds and catalysts for industrial use (Golden et al. 1986; Bach
52 et al. 1995; Cai et al. 2002). Because cations are readily adsorbed on or incorporated into the
53 birnessite structure (McKenzie 1976), birnessites play an important role in the sequestration and
54 release of nutrients and toxic elements in surficial environments. For example, Toner et al.
55 (2006) have demonstrated that the biogeochemical cycling of Zn is controlled by sorption onto
56 birnessite minerals in soils and streams.

57 Despite their importance as battery and exchange materials and their active role in
58 geological systems, birnessite-like phases are challenging to investigate because they tend to
59 occur as fine-grained, poorly crystalline masses. Chalcophanite is unusual in the birnessite
60 family in that it typically is well-crystallized, commonly forming crystals up to several mm in
61 diameter. Consequently it serves as an excellent proxy for better understanding the structures and
62 behaviors of the broader group of birnessite-like phases.

63 The chalcophanite structure was originally described by Wadsley (1955) using a triclinic
64 unit cell, but a later study by Post and Appleman (1988) revealed that, in fact, the correct
65 symmetry is $R\bar{3}$. The structure consists of sheets of edge-sharing $\text{Mn}^{4+}\text{-O}_6$ octahedra
66 alternating with layers of Zn cations and water molecules in the stacking sequence: $\cdots \text{Mn-O-Zn-}$
67 $\text{H}_2\text{O-Zn-Mn} \cdots$ (Fig. 1). One out of every seven octahedral sites in the Mn-O sheet is vacant, and
68 the Zn cations are located above and below the vacancies. The vacant Mn sites are fully ordered.
69 The $\sim 7 \text{ \AA}$ spacing between the Mn-O sheets is one of the defining characteristics of birnessite-
70 like phases. (There is an analogous group of birnessite-like phases within the “buserite” family
71 with $\sim 10 \text{ \AA}$ layer spacings due to an additional water layer.) Synthetic birnessite analogues
72 crystallize in both triclinic and hexagonal forms, depending upon pH and composition. In the
73 triclinic phases, typically Mn^{3+} cations, rather than vacancies, offset the charges of the interlayer
74 cations (Post et al. 2002; Lanson et al. 2002). Synthetic hexagonal birnessites, on the other hand,
75 do not contain Mn^{3+} , but in contrast with chalcophanite, the Mn^{4+} vacancies are disordered in the
76 Mn-O octahedral sheets (Drits et al. 1997; Post et al, 2008). The mineral birnessite,
77 $(\text{Na,Ca,Mn}^{2+})\text{Mn}_7\text{O}_{14}\cdot 2.8\text{H}_2\text{O}$, was first described by Jones and Milne (1956). Natural samples
78 were also found to contain trace amounts of a variety of cations, including Co, Ni, and Pb
79 (McKenzie 1977). Various synthetic birnessite-like structures containing almost every alkali and

80 alkaline earth element, as well as many of the transition metals, have been produced in the
81 laboratory, and most readily undergo cation exchange (e.g., McKenzie 1971; Golden et al. 1986,
82 Lopano et al. 2007). Interestingly, analyses of chalcophanite samples from various localities
83 show only minor amounts of cations other than Zn. This suggests that the Zn cation is
84 particularly well suited for growing a well-crystallized birnessite-like phase at the proper
85 environmental conditions.

86 Ranciéite is a Ca-rich birnessite that is commonly associated with biologically
87 precipitated Mn oxides, and it also occurs in oxidized zones of Mn deposits and in low-
88 temperature hydrothermal veins. Recent studies (Post et al. 2008) have shown that it has a
89 hexagonal structure similar to that of chalcophanite, but apparently with disordered vacancies in
90 the Mn-O octahedral sheets.

91 All birnessite-like phases contain water molecules in addition to cations in the interlayer
92 regions. In situ infra-red spectroscopy studies of the dehydration behaviors of various cation-
93 exchanged birnessite phases by Johnson and Post (2006) suggest that cation hydration energies,
94 and consequently, cation-water interactions are important factors in the relative stabilities and in
95 the exchange behaviors of birnessite phases. Experimentally based descriptions of the interlayer
96 region, however, and particularly of the roles played by the water molecules, are severely limited
97 by the poor crystallinity of most birnessite-like materials. Density functional theory calculations
98 by Kwon et al. (2009) indicate that the hydration state of Zn in a chalcophanite-like phase
99 significantly affects the structural stability. As water loss occurs when birnessite phases are
100 heated even to relatively low temperatures (<200 °C), structural changes related to dehydration
101 will directly impact the material behavior in near-surface environments.

102 In the current study we used time-resolved synchrotron X-ray powder diffraction and
103 Rietveld refinements to better explore the interactions among the water molecules, interlayer
104 cations, and the MnO octahedral sheets in chalcophanite. Here we present the results of a series
105 of refinements of the chalcophanite structure over the temperature range 300 to ~1060 K. These
106 dehydration data provide, for the first time, a direct look at changes in the structure with
107 increasing temperature as it undergoes multiple phase changes, including formation of a stable
108 intermediate dehydrated layer structure.

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EXPERIMENTAL DETAILS

111 The chalcophanite sample used for this study, NMNH C1814 from Sterling Hill NJ, is the
112 same as that used for the single-crystal X-ray diffraction study by Post and Appleman (1988).
113 They reported that the composition was very close to $\text{ZnMn}_3\text{O}_7 \cdot 3\text{H}_2\text{O}$. The sample used for X-
114 ray diffraction was hand-ground under acetone in an agate mortar and passed through a 325-
115 mesh sieve and loaded into a 0.5 mm quartz-glass capillary for the synchrotron XRD study.
116 XRD data were collected at beam line X7B of the National Synchrotron Light Source (NSLS),
117 Brookhaven National Laboratory (BNL), using a wavelength of 0.9273 Å and a MAR345 full
118 image plate detector.

119 The heating experiment was performed in air using a Blake Instruments furnace with a
120 Pt-13%Rh coiled wire yoke encased in ZrO_2 cement (Brown et al. 1973). The temperature was
121 varied with an Omega controller and monitored with a Chromel-Alumel thermocouple located
122 ~2 mm from the specimen. The actual sample temperature was determined for the range 298 to
123 1273 K by a variety of phase and melting transitions and by the placement of an additional
124 thermocouple in the sample position. The highly linear relationship between the observed and

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125 actual temperatures ($r^2 = 0.983$) allowed us to calculate a calibration curve with an estimated
126 error of ± 5 K for a given temperature. Temperature-resolved data from 300 to 1063 K were
127 collected as a series of 120 s exposures. The temperature was increased continuously at 8.5
128 K/min and measurements were obtained every ~ 28 K, owing to down time for repositioning of
129 the sample and reading the imaging plate; thus, each exposure encompassed a temperature range
130 of ~ 17 K. During each exposure the sample was rotated through a 120° angle. Preferred
131 orientation of the powder was eliminated through a combination of the specimen rotation, use of
132 a capillary sample holder, and full intensity integration of the diffraction rings, as obtained using
133 the program Fit2D (Hammersley et al. 1996) with a polarization factor of 0.93.

134 Rietveld refinements were performed using the general structure analysis system (GSAS)
135 of Larson and Von Dreele (2006) and EXPGUI interface by Toby (2001). The RT starting
136 chalcophanite structural parameters were taken from Post and Appleman (1988). The model for
137 the “anhydrous” chalcophanite was derived with assistance from the program “Crystal Cracker”
138 (Build 189, K Leinenweber, unpublished), which allowed us to determine starting unit-cell
139 parameters. Difference-Fourier maps were calculated using the MnO octahedral layers in order
140 to locate the Zn atoms in the interlayer region. The diffraction pattern backgrounds were fit using
141 a linear interpolation function. Peak profiles were modeled by a pseudo-Voigt profile function as
142 parameterized by Thompson et al. (1987) with asymmetry corrections by Finger et al. (1994) and
143 microstrain anisotropic broadening terms by Stephens (1999). Displacement factors for
144 chalcophanite were fixed to values reported by Post and Appleman (1988), and subsequently
145 refined for the anhydrous phases.

146 During the initial cycles of refinement, only the background, scale, peak profile, and unit-
147 cell parameters were allowed to vary. After convergence, all atom positions, displacement

148 factors (for anhydrous phases), and appropriate occupancy factors, e.g. of the O atoms of the
149 H₂O molecules and Zn site in the anhydrous phases, were refined.

150 The final refinement parameters for 300 K chalcophanite, the intermediate anhydrous
151 phase (540 K), and Mn-hetaerolite (994 K) are listed in Table 1. The refined chalcophanite atom
152 positions did not differ significantly from those reported by Post and Appleman (1988) and
153 therefore are not listed here. Atom positions for the anhydrous chalcophanite were similar at all
154 temperatures; therefore, only the values determined at 540 K are reported in Table 2, along with
155 those for Mn-hetaerolite at 994 K. Selected bond distances for anhydrous chalcophanite and Mn-
156 hetaerolite are reported in Table 3. The final observed, calculated, and difference patterns for
157 chalcophanite (300 K), anhydrous chalcophanite (540 K), and Mn-hetaerolite (994 K) are plotted
158 in Figure 2. The standard deviations calculated by GSAS for the lattice parameters are likely
159 lower than the true errors (Post and Bish 1989).

160 RESULTS AND DISCUSSION

161 The synchrotron powder X-ray diffraction patterns collected over the range 300 K to
162 1060 K are plotted in Figure 3. The plot reveals two obvious phase changes, at ~450 K and ~950
163 K, corresponding to the dehydration of chalcophanite followed by transformation to a spinel
164 structure – Mn-hetaerolite, as discussed below.

165

166 Chalcophanite (ZnMn₃O₇ · 3H₂O) dehydration

167 The Rietveld refined structure for chalcophanite at room temperature is not significantly
168 different than that reported from the single-crystal diffraction study of Post and Appleman
169 (1988). The Mn-O distances correspond to within 0.02 Å, giving an indication of the general

170 accuracy of the Rietveld structures determined in this study. Some noteworthy aspects of the
171 structure that were also discussed by Post and Appleman (1988) are: 1) the Mn-O octahedra are
172 distorted, with bond distances ranging from 1.86 to 1.97 Å, caused by displacement of the Mn
173 toward the vacancy in the octahedral sheet; 2) the mean Mn-O distance is 1.91 Å, indicating that
174 all Mn is Mn⁴⁺; and 3) the Zn cations occupy sites above and below the vacancies in the
175 octahedral sheets and are coordinated to three O atoms and three water oxygen atoms.

176 The results of the heating study show only small changes in the chalcophanite unit-cell
177 from RT to ~438 K (Figure 4). The volume increased by ~0.8%, mostly caused by thermal
178 expansion of ~0.5% along *c*; the value of *a* increased only ~0.1%. Between ~438 and 473 K, the
179 volume increased more rapidly by additional 0.8%, primarily caused by a ~1.0% increase along
180 *c*. This latter change in volume corresponds to the period of maximum water evolution, as
181 discussed below. Our structure refinements revealed a gradual decrease in the occupancy factor
182 for the water oxygen atom, starting above ~416 K, giving values of 0.85 at 416 K, 0.75 at 427 K,
183 0.69 at 438 K, and 0 above ~450 K. The changes in occupancy factor correlated with drops in
184 the intensity of the 001 diffraction peak in Figure 3, with the first apparent change at 416 K. The
185 001 peak disappeared by ~473 K. This transition is consistent with the first appearance of
186 anhydrous chalcophanite at 427 K from the Rietveld refinements, and we infer that chalcophanite
187 had completely transformed by 473 K. The structure refinements did not show any significant
188 changes in chalcophanite Mn-O distances or occupancy factors other than for the water oxygen
189 atom during dehydration.

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192 **Anhydrous chalcophanite (ZnMn_3O_7)**

193 As chalcophanite was heated above ~ 427 K, the interlayer water molecules were lost,
194 resulting in a collapse of the interlayer spacing from ~ 7 to ~ 4.8 Å. The newly formed anhydrous
195 phase retains chalcophanite's $R\bar{3}$ space group and a dimension, but has a c dimension of 14.3 Å
196 instead of ~ 21 Å. The fraction of anhydrous chalcophanite increased as the sample was heated
197 above 427 K until at ~ 473 K no chalcophanite remained. The anhydrous chalcophanite remained
198 the stable phase until ~ 775 K, above which, as is discussed below, it transformed to a Mn-rich
199 hetaerolite, which has the spinel structure.

200 The existence of anhydrous chalcophanite was first noted by Dasgupta (1974) based on
201 results of heating chalcophanite single crystals. His diffraction and differential
202 thermogravimetric experiments indicated a transformation of chalcophanite to the anhydrous
203 phase at ~ 473 K, which agrees well with the results of the present study. Although he reported
204 unit-cell parameters that are close to the values determined here, he was not able to determine a
205 structure for the anhydrous phase.

206 The structure determined from our Rietveld study for anhydrous chalcophanite is shown
207 in Figure 1. The obvious difference from the chalcophanite structure is the loss of the water
208 molecules from the interlayer region and consequent collapse of the spacing between the
209 octahedral sheets. The Mn-O distances calculated from the refined structure for the anhydrous
210 phase are not significantly different from those in chalcophanite, ranging from 1.86 Å to 1.98 Å,
211 with a mean value of 1.91 Å (Table 4), indicating that the Mn oxidation state is still tetravalent,
212 and that the MnO octahedral sheets are essentially the same in both phases. The calculated bond
213 valence sum for Mn (Brown and Altermatt 1985), using bond distances in Table 3, is 3.93 . The

214 Zn atoms remain in the interlayer region, positioned above and below the vacancies in the MnO
215 octahedral sheets, but they are tetrahedrally coordinated to O atoms in the sheets, bonded to three
216 O(2) above and one O(3) below. This coordination differs from the octahedral coordination of
217 Zn in chalcophanite, to three O atoms in the octahedral sheet and three water oxygen atoms in
218 the interlayer. The Zn-O distances range from 1.98 Å (x3) to 2.00 Å, and they correspond well
219 with the predicted tetrahedral bond distance of 1.98 Å, using the ionic radii of Shannon (1976).
220 The bond valence sum for Zn is 1.87. The change in Zn coordination from octahedral to
221 tetrahedral during dehydration is consistent with modeling calculations by Kwon et al. (2009) for
222 a hypothetical monohydrate chalcophanite that shows Zn only in tetrahedral coordination. The
223 particular flexibility of Zn cations to adopt octahedral or tetrahedral coordination might explain
224 why chalcophanite forms a stable anhydrous phase, whereas birnessite-like materials having
225 other interlayer cations, e.g. Na, Ba, Cs, Ca, etc. generally dehydrate to disordered/amorphous
226 intermediate phases before transforming to spinel-like compounds (unpublished research).

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228 **Mn-rich hetaerolite [(Zn_{0.75}Mn²⁺_{0.25})Mn³⁺₂O₄]**

229 Above ~775 K the anhydrous chalcophanite began to transform to a spinel structure,
230 corresponding to a Mn-rich hetaerolite, and by ~973 K the transformation was complete. The
231 diffraction patterns did not show a significant increase in background during the transformation,
232 indicating that the reaction did not involve transient amorphization. The phase change is
233 associated with the reduction of Mn⁴⁺ to Mn³⁺ and Mn²⁺, with a corresponding loss of 1.25 of
234 seven O atoms from the original anhydrous chalcophanite structure. The Rietveld refinements of
235 the O atom occupancy factors for anhydrous chalcophanite as it was heated above ~700 K

236 showed a gradual decrease for the O(3) value from 0.94 to 0.76 at 845 K, while the values for
237 O(1) and O(2) stayed close to 1.0, suggesting that most of the O loss during the transformation
238 was O(3). We note that O(3) is structurally distinct from O(1) and O(2) in anhydrous
239 chalcophanite. Each O(1) anion coordinates only to three Mn octahedral sites, and the O(2)
240 anions serve as the basal oxygens of the Zn tetrahedra that are shared with the Mn octahedral
241 sheet. Therefore, the O(2) anions each coordinate to 2 Mn cations and 1 Zn cation. The O(3)
242 anions are the apical oxygens of the Zn tetrahedra, and thus each O(3) anion is bonded to 3 Mn
243 cations and 1 Zn cation. We presume that increasing thermal vibrations rupture Mn-O bonds,
244 resulting in the selective loss of the O(3) anion as O₂ and the consequent reduction of Mn⁴⁺ by
245 electrons released as O²⁻ is oxidized to O₂. The Mn²⁺ and Mn³⁺ cations migrate to the interlayer
246 as the structure transforms to Mn-rich hetaerolite. When the structures of hetaerolite and
247 anhydrous chalcophanite are viewed normal to the MnO octahedral sheets, their structural
248 similarity is revealed (Fig. 5). It is interesting to note that this migration of reduced Mn cations
249 from the octahedral sheets to the interlayer parallels the similar behavior during the room
250 temperature transformation of triclinic birnessite to hexagonal birnessite at pH below ~5.

251 The mineral hetaerolite has the nominal formula ZnMn₂O₄, but as the spinel phase in our
252 experiments was formed directly from anhydrous chalcophanite (and the diffraction patterns do
253 not show other phases), the product of our heating experiment apparently retained the Zn/Mn
254 ratio of the precursor phase, yielding the formula: (Zn_{0.75}Mn²⁺_{0.25})Mn³⁺₂O₄. Thus, the
255 dehydration of chalcophanite generated a compound whose chemistry lies one-quarter of the way
256 along the compositional series between hetaerolite and hausmannite (Mn²⁺Mn³⁺₂O₄). The
257 refined occupancy factor for the spinel Zn site was 0.96, which is not significantly different from
258 the value of 0.94, as would be expected for a tetrahedral site with 75% Zn and 25% Mn. The

259 refined Mn³⁺-O distances of 1.935 Å (x4) and 2.285 Å (x2) (Table 3) are the same as values
260 reported for hausmannite (Baron et al. 1998) and reflect the Jahn-Teller distortion associated
261 with Mn³⁺. The slightly larger tetrahedral (Zn,Mn)-O distance of 1.99 Å compared to the
262 predicted value of 1.98 Å for Zn-O given above reflects the presence of the larger Mn²⁺ cation
263 (Mn²⁺-O = 2.03; Shannon, 1976). The bond valence sum (Brown and Altermatt 1985) calculated
264 for the octahedral Mn³⁺, using bond distances from Table 3, is 2.97, and for the tetrahedral
265 Zn/Mn (weighting bond strengths proportional to Zn/Mn content) is 1.95.

266

Implications

267 Time-resolved synchrotron powder diffraction studies of the dehydration of
268 chalcophanite provides insights into the dehydration behavior of birnessite-like Mn oxide phases.
269 The ability of Zn to accommodate tetrahedral and octahedral coordination is likely the key to the
270 formation of a stable anhydrous chalcophanite phase. The transformation from the layer
271 structure to a spinel phase appears to be triggered by loss of O atoms during heating, and
272 consequent reduction of Mn. This work should guide selection of interlayer cations for
273 designing anhydrous phylломanganate battery materials, and provide a better understanding of
274 the stabilities and compositions of natural phylломanganates.

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FIGURES

373 **Figure 1.** Structure drawings of: a) chalcophanite, b) anhydrous chalcophanite, and c) Mn-rich
374 hetaerolite. Mn-O octahedra are colored pink, and Zn-O polyhedra are yellow. The O atoms are
375 indicated by red and blue [O(3)] spheres. For all three structures the *c* axis is vertical.

376 **Figure 2.** Final observed (red crosses), calculated (green solid line), and difference (purple
377 below) powder X-ray diffraction patterns from the Rietveld refinement for: a) chalcophanite, b)
378 anhydrous chalcophanite, and c) Mn-rich hetaerolite. The Bragg reflections are marked by the
379 set of small vertical lines

380 **Figure 3.** Synchrotron powder X-ray diffraction patterns vs. temperature from RT (front) to
381 1060 K

382 **Figure 4.** Plot of: a) unit-cell volume and b) *c* for chalcophanite for the temperature range 300
383 K to 438 K. Calculated esd's fall within the areas of the plotting symbols.

384

385 **Figure 5.** Polyhedral structure drawings for: a) anhydrous chalcophanite and b) Mn-rich
386 hetaerolite projected down *c*. Mn-O octahedra are colored pink, and Zn-O polyhedra are yellow.
387 The O atoms are indicated by red and blue [O(3)] spheres.

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392 Table 1. Final Rietveld refinement parameters for chalcophanite, anhydrous chalcophanite and

393 Mn-hetaerolite

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	Chalcophanite (300 K)	Anhydrous Chalcophanite (540 K)	Mn-hetaerolite (994 K)
Space Group	<i>R</i> -3	<i>R</i> -3	<i>I</i> 4 ₁ / <i>amd</i>
Unit Cell			
<i>a</i> (Å)	7.5437(1)	7.5482(4)	5.75337(26)
<i>b</i> (Å)	7.5437(1)	7.5482(4)	5.75337(26)
<i>c</i> (Å)	20.8182(7)	14.2955(13)	9.3162(6)
<i>V</i> (Å ³)	1026.0(1)	705.37(1)	308.38(4)
Refinement			
No. of data points	1687	1687	2050
No. of reflections	210	124	45
Diffraction range (2θ °)	13.5 – 47.3	13.5 – 47.3	8 - 49
No. of variables	42	46	37
R(F ²)	0.042	0.056	0.025
R _{wp}	0.014	0.017	0.013
χ ²	0.94	1.12	1.94

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398 Table 2. Atomic coordinates and isotropic displacement factors for Anhydrous chalcophanite and

399 Mn-hetaerolite

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Atom	<i>x</i>	<i>y</i>	<i>z</i>	Site occupancy Factor	<i>U</i> _{iso}
<u>Anhydrous chalcophanite (540 K)</u>					
Mn	0.7193(6)	0.5799(6)	0.00079(31)	1.0	0.0142(7)
Zn	0	0	0.1308(5)	1.0	0.0142(7)
O(1)	0.5233(23)	0.6291(29)	0.0678(10)	1.0	0.0318(18)
O(2)	0.2648(24)	0.2104(24)	0.0779(1)	1.0	0.0318(18)
O(3)	0	0	0.7292(15)	1.0	0.0318(18)
<u>Mn-hetaerolite (994 K)</u>					
Mn	0	0.5	0.5	1.0	0.0152(7)
Zn	0	0.25	0.875	0.941(2) ^a	0.0179(6)
O	0	0.4749(4)	0.2552(2)	1.0	0.0300(9)

402 Note: *U*_{iso} for O atoms in anhydrous chalcophanite were constrained to be the same; coordinates
 403 are for origin choice at 2/m [add (0,1/4,-1/8) to shift to origin at -4m2]

404 ^a Site also contains Mn but was refined as Zn

405

406 Table 3. Selected bond lengths for anhydrous chalcophanite and Mn-hetaerolite structures (Å)

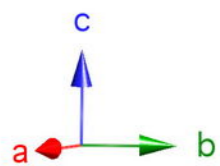
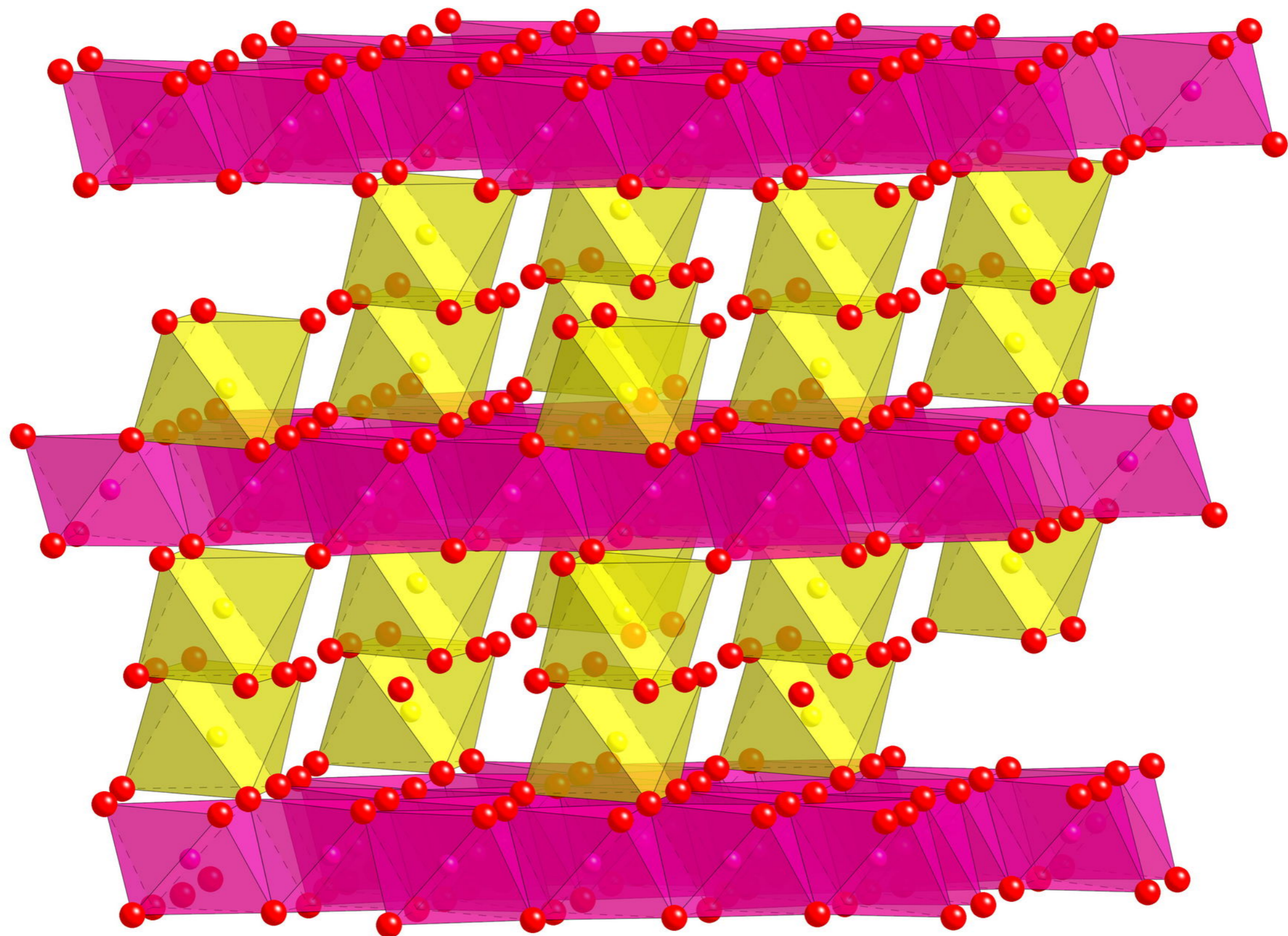
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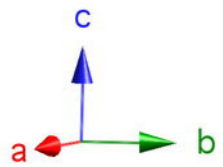
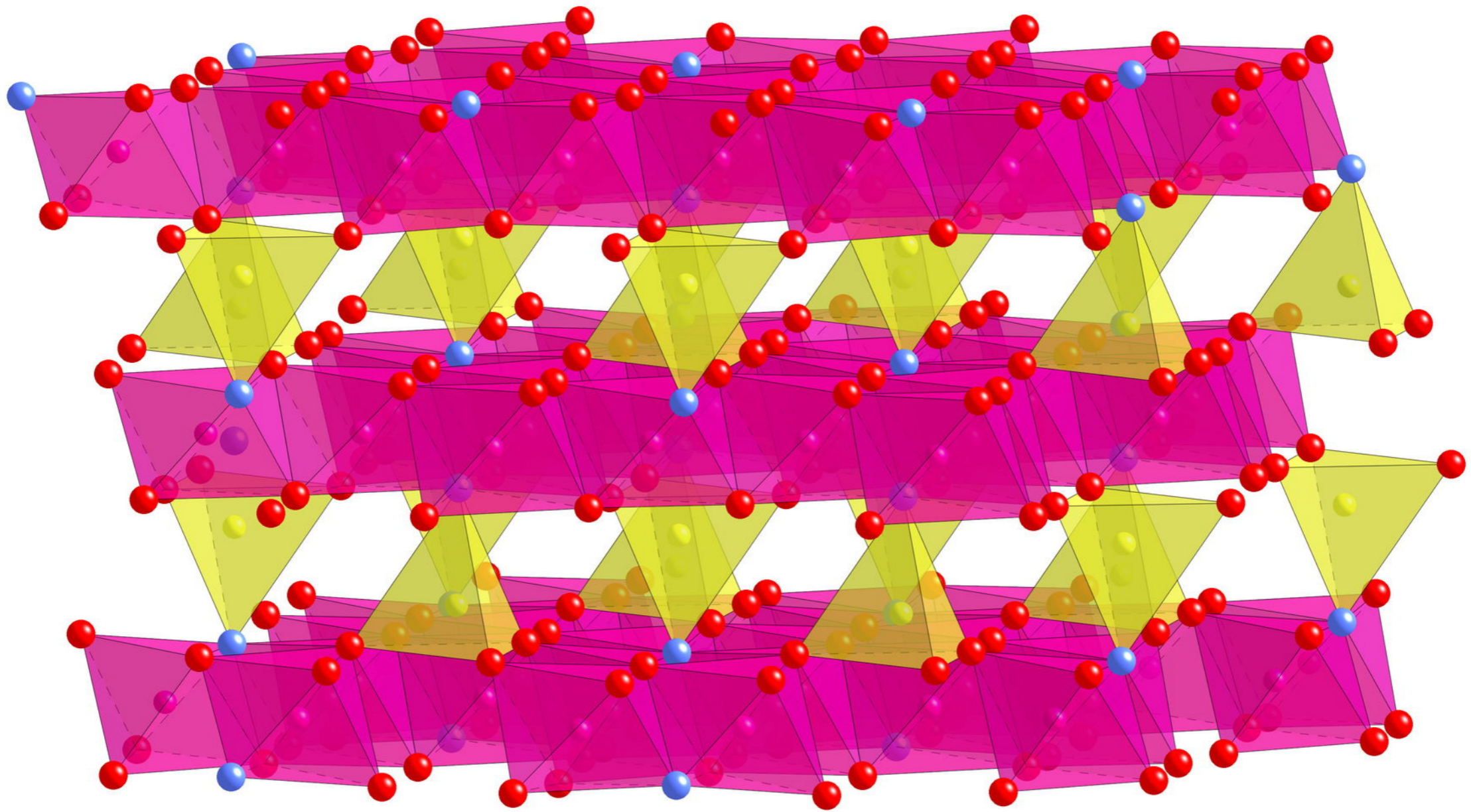
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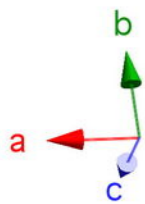
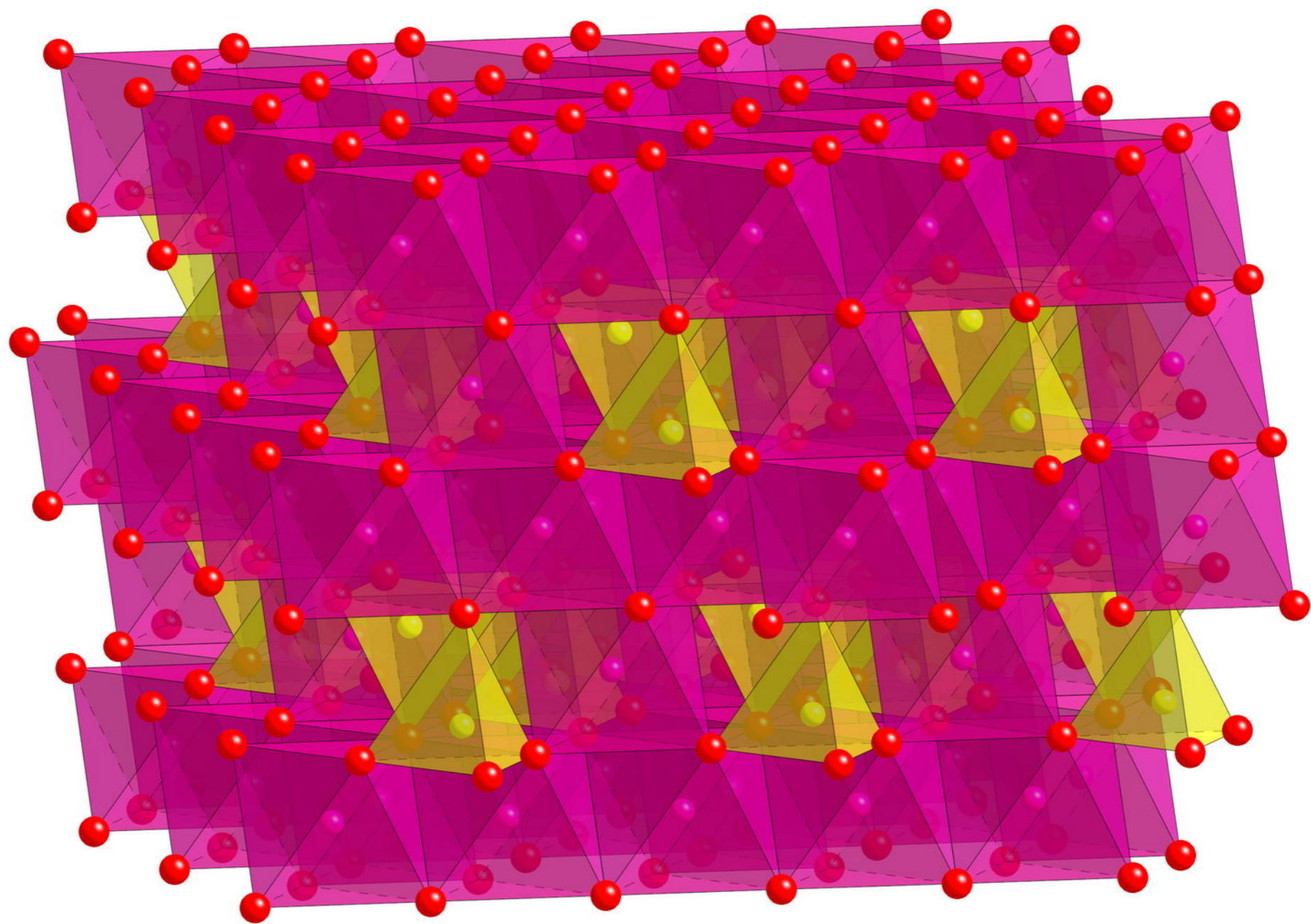
	Anhydrous chalcophanite (540 K)	Mn-hetaerolite (994 K)
Mn-O(1)	1.858(17)	1.93515(9) x 4
-O(1)	1.948(14)	2.28510(14) x 2
-O(1)	1.978(17)	
-O(2)	1.872(14)	
-O(2)	1.897(14)	
-O(3)	1.914(10)	
<Mn-O>	1.91	2.05
Zn-O(2)	1.978(15) x 3	1.99434(7) x 4 ^a
-O(3)	2.001(18)	
<Zn-O>	1.99	1.994

409

410 ^a Site contains ~ 0.75 Zn and 0.25 Mn







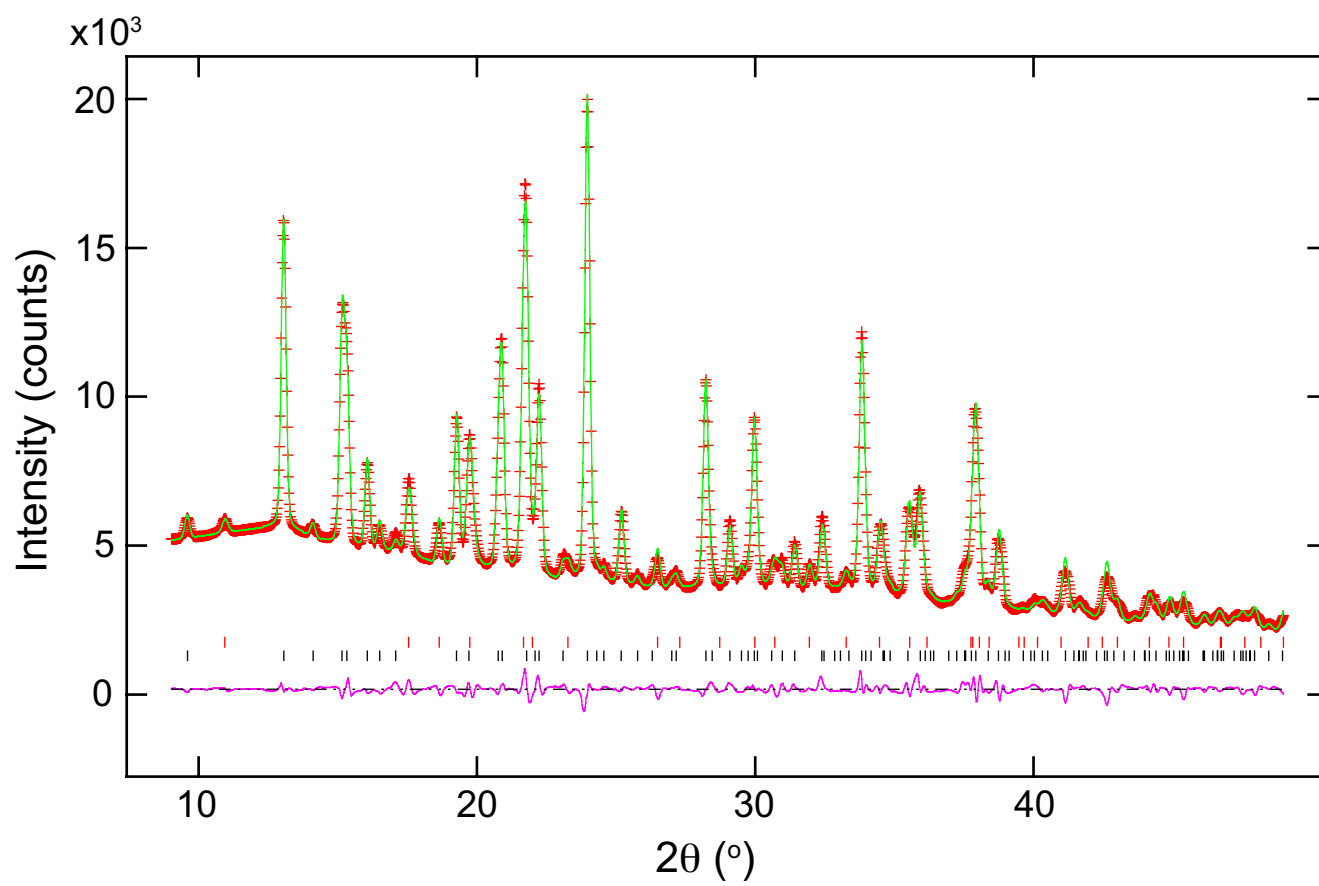


Figure 2a

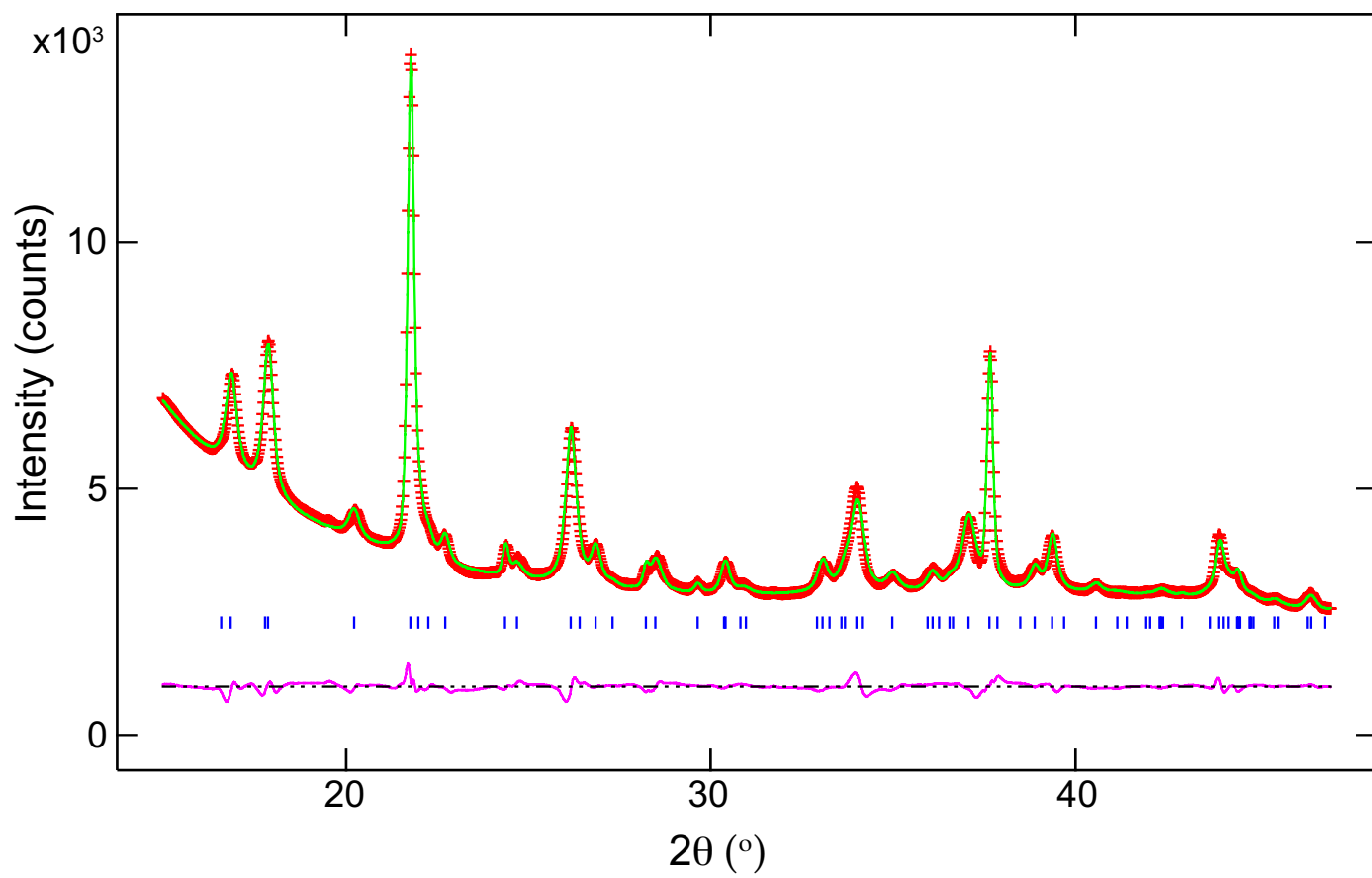


Figure 2b

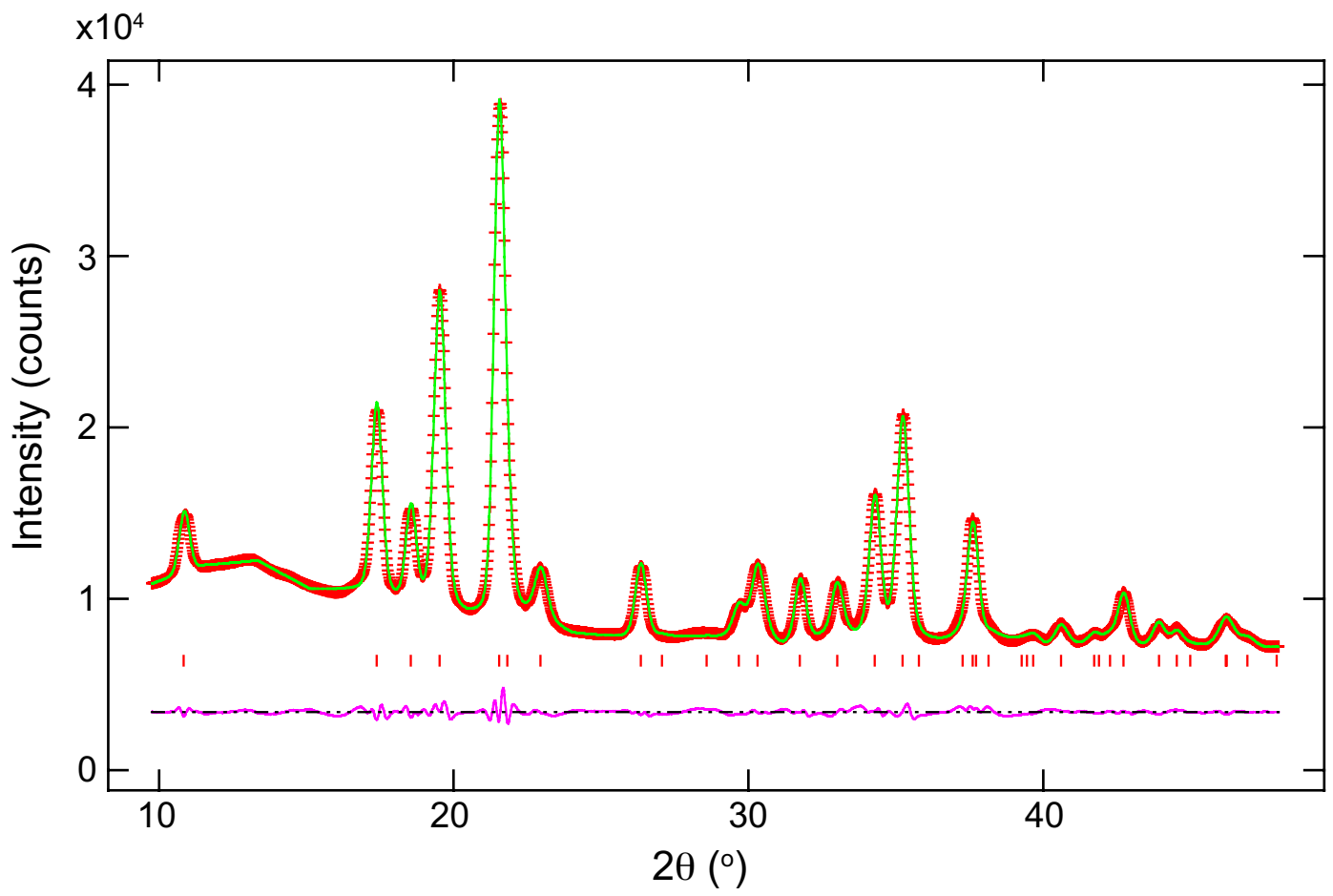
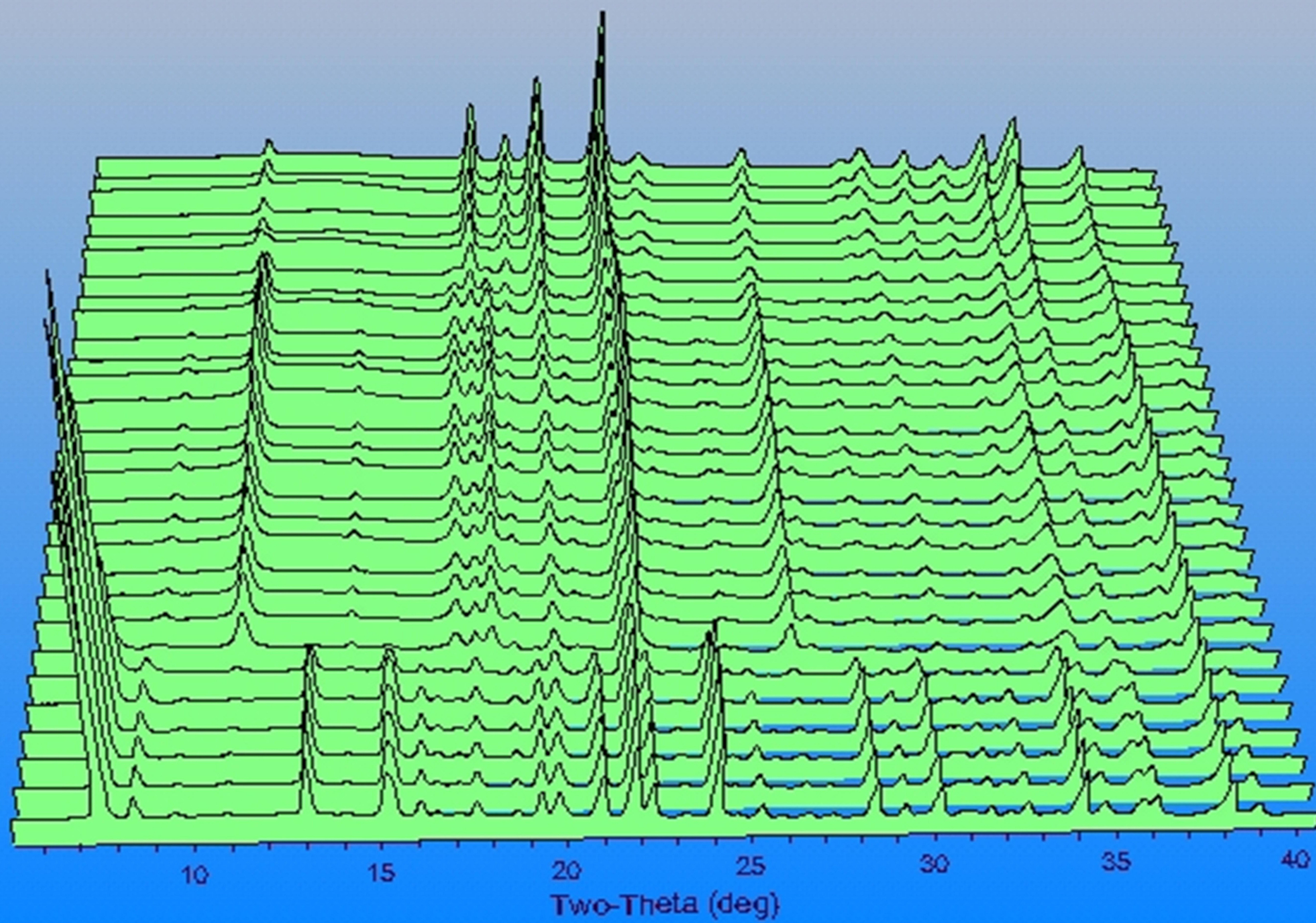


Figure 2c



Unit Cell Volume vs. T

