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2 **High-pressure elasticity of sodium majorite garnet, Na<sub>2</sub>MgSi<sub>5</sub>O<sub>12</sub>**

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6 **Abstract**

7 Garnet is the second most abundant mineral phase in the upper mantle and  
8 transition zone settings. The crystal structure of garnet is quite flexible and hence it is  
9 able to accommodate various cations, including large incompatible cation such as  
10 sodium. We used *first principles* simulation based on density functional theory and  
11 two widely used approximations- local density approximation (LDA) and generalized  
12 gradient approximation (GGA)- to explore the crystal structure, equation of state, and  
13 elasticity of sodium bearing majorite garnet with Na<sub>2</sub>MgSi<sub>5</sub>O<sub>12</sub> stoichiometry at  
14 pressures relevant to the upper mantle and transition zone. We find that the pressure-  
15 volume results based on LDA can be explained by a Birch Murnaghan finite strain  
16 equation of state with  $V_0 = 1447.6 (\pm 0.1) \text{ \AA}^3$ ,  $K_0 = 177.4 (\pm 0.4) \text{ GPa}$ , and  $K'_0 = 3.93$   
17  $(\pm 0.02)$ . The results based on GGA can be explained by a Birch Murnaghan finite  
18 strain equation of state with  $V_0 = 1525.8 (\pm 0.2) \text{ \AA}^3$ ,  $K_0 = 160.2 (\pm 0.4) \text{ GPa}$ , and  $K'_0 =$   
19  $3.96 (\pm 0.02)$ . The full elastic moduli tensor for Na-majorite with tetragonal symmetry  
20 exhibits slight deviation from the cubic symmetry with  $C_{11} < C_{33}$ ,  $C_{12} \sim C_{13}$ ,  
21 and  $C_{44} \sim C_{66}$ . The magnitude of the tetragonal strain also captures the slight deviation  
22 from the cubic symmetry. At pressures corresponding to the upper mantle and mantle  
23 transition zone, the compressional wave velocity,  $V_p$ , and shear wave velocity,  $V_s$ ,  
24 for the Na-majorite garnet are fast compared to a wide variety of garnets such as  
25 pyrope, grossular, almandine, and majorite garnet. Although single crystal anisotropy

26 of Na-majorite is greater than pyrope, it is still low compared to the major mantle  
27 phases.

28

29 **Key words:** Elasticity, Equation of state, Sodium majorite, Mantle transition region

30

### 31 **1. Introduction**

32 In the mineralogical models of the deep Earth, garnet is a volumetrically  
33 dominant mineral phase in the upper mantle and transition zone (Ringwood, 1962). In  
34 a homogenous pyrolite model, modal abundance of garnet ranges from 20 vol % at  
35 upper mantle conditions to almost 40 vol % at transition zone conditions. The  
36 thermodynamic stability of garnet enhances upon complete dissolution of the  
37 pyroxene phase at transition zone conditions. In recent studies, it has been suggested  
38 that the Earth's mantle is far from homogenous and consists of a mixture of depleted  
39 harzburgite components and subducted oceanic crusts, or mid oceanic basalt (MORB)  
40 components (Xu et al., 2008; Stixrude and Lithgow-Bertelloni, 2012). The bulk  
41 chemistry of MORB is different from depleted harzburgite and has significantly  
42 greater proportions of incompatible elements including Na<sub>2</sub>O (~1.5-1.8 wt%) (Irifune  
43 and Ringwood, 1993). The alkali component is likely to be incorporated into  
44 pyroxene (Angel et al., 1988) and also into garnet at higher pressures (Irifune and  
45 Ringwood, 1993). Experimental studies on realistic mantle bulk compositions have  
46 shown Na-majorite to be stable at conditions relevant to the lower part of the upper  
47 mantle and transition zone (Gasparik, 1990; Bindi et al., 2011; Bobrov et al., 2008a,b,  
48 2009; Dymshits et al., 2010, 2013). Sodium bearing natural garnets with (Na<sub>2</sub>O > 1 wt  
49 %) have also been reported (Sobolev and Lavrentev, 1971), indicating that sodium is  
50 likely to be an important component within the garnet structure at higher pressures.

51 Na-Majorite is an important host for the alkali elements and the partitioning of alkali  
52 elements might affect the relative thermodynamic stability of other alkali bearing  
53 aluminous phases such as new aluminous phases (NAL) and calcium ferrite structured  
54 (CF) phases (Dymshits et al., 2013). Owing to the importance of Na-majorite in the  
55 upper mantle and transition zone conditions, extensive studies have been conducted  
56 on phase relations (Gasparik, 1990; Bindi et al., 2011; Bobrov et al., 2008a,b, 2009;  
57 Dymshits et al., 2010, 2013), crystal structure determinations (Bindi et al., 2011) and  
58 high pressure compressibility and the equation of state (Hazen et al., 1994; Dymshits  
59 et al., 2014). It is known that the bulk sound velocity i.e., compressional wave  
60 velocity,  $V_p$ , and shear wave velocity,  $V_s$ , for mineral phases and aggregates are  
61 important to relate seismological observation with the chemical composition of the  
62 mantle. The elastic moduli for majorite garnet with  $\text{Na}_2\text{MgSi}_5\text{O}_{12}$  stoichiometry are  
63 important. A complete knowledge of these end-member properties is instrumental in  
64 developing a thermodynamic mixing-model that will allow predictions for the seismic  
65 velocity for complex majorite solid solutions. It will also help us to test whether these  
66 phases that host alkali elements play important role in the upper mantle and in the  
67 transition zone. Despite being important, the full elastic moduli tensor and the sound  
68 wave velocity of  $\text{Na}_2\text{MgSi}_5\text{O}_{12}$  remains unknown. In this study we used *first*  
69 *principles* method to predict crystal structure, equation of state, and full elastic moduli  
70 tensor at high pressures.

71

## 72 **2. Method**

### 73 ***2.1. Crystal structure models for sodium garnet***

74 In this study we examine Na-majorite garnet with  $\text{Na}_2\text{MgSi}_5\text{O}_{12}$  stoichiometry.

75 Earlier experimental studies on Na-majorite with  $[(\text{Na}_{1.88}\text{Mg}_{1.12})(\text{Mg}_{0.06}\text{Si}_{1.94})\text{Si}_3\text{O}_{12}]$

76 stoichiometry reported a cubic symmetry with  $Ia\bar{3}d$  spacegroup (Pacalo et al., 1992;  
77 Hazen et al., 1994). Recent high-pressure synthesis of Na-majorite with  $\text{Na}_2\text{MgSi}_5\text{O}_{12}$   
78 stoichiometry and subsequent crystal structure determination suggested a tetragonal  
79 symmetry with  $I4_1/acd$  spacegroup (# 142) (Bindi et al., 2011). In the tetragonal  
80 Na-majorite garnet, the Na and Mg cations disordered over the two do-decahedral  
81 sites with 16e and 8b Wyckoff symmetry (Bindi et al., 2011). In our study we use the  
82 tetragonal crystal structure as our starting model. Since our calculations are static, we  
83 let the sodium cations reside in the 16e and the Mg cations reside in the 8b sites. This  
84 preserves the tetragonal symmetry with  $I4_1/acd$  spacegroup (**Table 1**). The Si  
85 cations are hosted in the octahedral and tetrahedral sites. There are two  
86 crystallographically distinct tetrahedral sites (**Table 1, Figure 1**).

87

## 88 **2. 2. First principle simulations**

89 We performed first principles calculation based on the density functional  
90 theory (Kohn and Sham, 1965) (DFT) using Vienna Ab initio Simulation Package  
91 (VASP) (Kresse and Hafner, 1993; Kresse and Furthmüller, 1996a,b). Two widely  
92 used approximations to the exchange-correlation functional: the local density  
93 approximation (LDA) and generalized gradient approximation (GGA) were used  
94 (Ceperley and Adler, 1980; Perdew and Wang 1986; Perdew et al., 1991, 1996). In  
95 order to facilitate solving the DFT equations, further approximations are made  
96 motivated by the argument that the core electrons participate little in bonding and  
97 structural changes. We have employed the pseudopotential method (Heine, 1970).  
98 DFT calculations have been instrumental in accurate prediction of energetics, physical  
99 properties such as compressibility, equation of state, and elasticity of mineral phases  
100 that are relevant for the Earth and planetary interiors (Karki et al., 2001; Oganov et

101 al., 2002; Wentzcovitch and Stixrude, 2010; ) including alkali bearing mineral phases  
102 (Mookherjee and Steinle-Neumann, 2009a,b; Mookherjee, 2011; Walker, 2012) and  
103 garnet (Li et al., 2007; Kawai and Tsuchiya, 2012; Erba et al., 2014). The  
104 pseudopotentials were successfully tested in previous studies on mineral phases and  
105 melts (Mookherjee, 2011; Mookherjee and Bezacier, 2012; Mookherjee et al., 2012).

106 All computations are performed in a non-primitive unit-cell with 160 atoms.  
107 We use an energy cutoff of  $E_{cut} = 600$  eV and a K-point mesh sampling of  $2 \times 2 \times 2$   $k$ -  
108 point mesh. A series of convergence tests demonstrated that these computational  
109 parameters yield pressures and total energies that are converged to within 0.01 GPa  
110 and 10 meV/atom respectively. Similar computational parameters were successfully  
111 used in recent study on grossular garnet (Kawai and Tsuchiya, 2012).

112 We analyzed the bulk compression behavior using a third order Birch  
113 Murnaghan equation-of state (Birch, 1978). To compute the full elastic moduli tensor,  
114 we strained the lattice and let the internal degree of freedom of the crystal structure  
115 relax consistent with the symmetry: elastic moduli were obtained through the changes  
116 in stress tensor ( $\underline{\underline{\sigma}}$ ) with respect to applied strain ( $\underline{\underline{\epsilon}}$ ). We applied positive and  
117 negative strain ( $\underline{\underline{\epsilon}}$ ) of magnitude 1% in order to accurately determine stress ( $\underline{\underline{\sigma}}$ ) in  
118 the appropriate limit of zero strain. The strained lattice ( $\vec{a}'$ ) is related to the fully  
119 relaxed lattice ( $\vec{a}$ ) by  $\vec{a}' = (\underline{\underline{I}} + \underline{\underline{\epsilon}}) \times \vec{a}$ , where  $\underline{\underline{I}}$  is an identity matrix (Karki et al.,  
120 1997). The details of the method is outlined in previous studies (Karki et al., 2001).  
121 For the Na-majorite with tetragonal symmetry and  $I4_1/acd$  spacegroup, four strains  
122 ( $\underline{\underline{\epsilon}}$ ) were used to derive the six independent elastic moduli  $C_{11}(=C_{22})$ ,  $C_{12}$ ,  
123  $C_{13}(=C_{23})$ ,  $C_{33}$ ,  $C_{44}(=C_{55})$ , and  $C_{66}$  (Nye, 1985).

### 124 **3. Results**

### 125 **3. 1. Equation of state and Linear Compressibility**

126 The predicted pressure-volume results based on *first-principles* simulations  
127 can be explained by a Birch Murnaghan finite strain formulation (Birch, 1978),

$$128 \quad P = 3K_0 f_v (1 + 2f_v)^{\frac{5}{2}} \left( 1 + \frac{3}{2} (K'_0 - 4) f_v \right) \quad (1)$$

129 where,  $f_v$  is Eulerian finite volume strain-

$$130 \quad f_v = \frac{1}{2} \left( \left( \frac{V_0}{V} \right)^{\frac{2}{3}} - 1 \right) \quad (2)$$

131 LDA predictions yields equation of state parameters with  $V_0 = 1447.6 (\pm 0.1) \text{ \AA}^3$ ,  $K_0 =$   
132  $177.3 (\pm 0.3) \text{ GPa}$ , and  $K'_0 = 3.93 (\pm 0.02)$ . In contrast, predictions based on GGA  
133 yields equation of state parameters,  $V_0 = 1525.8 (\pm 0.2) \text{ \AA}^3$ ,  $K_0 = 160.2 (\pm 0.4) \text{ GPa}$ ,  
134 and  $K'_0 = 3.96 (\pm 0.02)$ . The zero-pressure volume predicted using LDA and GGA are  
135 1.9 % smaller and 5.4 % larger than the recent experimental results respectively  
136 (Dymshits et al., 2014) (**Table 2, Figure 1**). The zero-pressure bulk modulus  
137 predicted using LDA is 2% smaller than the experimental result. The discrepancy  
138 between the zero-pressure bulk modulus predicted using GGA and experimental study  
139 is significantly larger  $\sim 9.6$  % (**Table 2, Figure 1**). The zero-pressure bulk modulus  
140 ( $K_0$ ) and volume ( $V_0$ ) are anti-correlated as observed in several mineral phases  
141 (Karki et al., 2001; Oganov et al., 2002). In comparison to the LDA prediction, the  
142 equation of state studies on single crystal Na-majorite with  
143  $[(\text{Na}_{1.88}\text{Mg}_{1.12})(\text{Mg}_{0.06}\text{Si}_{1.94})\text{Si}_3\text{O}_{12}]$  stoichiometry reports slightly stiffer  $\sim 7.4$  %, with  
144 zero-pressure bulk modulus of  $191.5 (\pm 2.5) \text{ GPa}$  (Hazen et al., 1994). This  
145 discrepancy is likely related to the limited pressure ( $< 5 \text{ GPa}$ ) range explored in the  
146 previous experiments. However, in an earlier Brillouin scattering study with the same  
147 single crystal of Na-majorite with  $[(\text{Na}_{1.88}\text{Mg}_{1.12})(\text{Mg}_{0.06}\text{Si}_{1.94})\text{Si}_3\text{O}_{12}]$  stoichiometry, a

148 zero-pressure bulk modulus of 173.5 ( $\pm$  2.6) GPa (Pacalo et al., 1992) was reported.  
149 This is within  $\sim$ 2 % of the LDA prediction (**Table 2**).

150 The pressure dependence  $a$ - and  $c$ - lattice parameters (**Supplementary Table**  
151 **1**) could be explained by finite strain formulations (Meade and Jeanloz, 1990 *and*  
152 *references therein*), where the Eulerian finite lattice strain,  $f_\alpha$ , along the  $\alpha$  i.e.,  $a$ - and  
153  $c$ -axes-

$$154 \quad f_a = \frac{1}{2} \left( \left( \frac{a_0}{a} \right)^2 - 1 \right) \quad (3)$$

155 and

$$156 \quad f_c = \frac{1}{2} \left( \left( \frac{c_0}{c} \right)^2 - 1 \right) \quad (4)$$

157 and, the normalized pressure  $F_\alpha$ , along  $\alpha$  i.e.,  $a$ - and  $c$ - axes directions are given by,

$$158 \quad F_a = \frac{P(ac_0/a_0c)^{2/3}}{f_a(1+f_v)^{3/2}} \quad (5)$$

159 and

$$160 \quad F_c = \frac{P(a_0c/ac_0)^{2/3}}{f_c(1+f_v)^{3/2}} \quad (6)$$

161 where  $f_v$ , is the Eulerian finite volume strain, defined in eq (2).

162 The finite strain expansions for the linear parameters (Meade and Jeanloz, 1990)  
163 could be written as:

$$164 \quad F_\alpha = K_\alpha + m_\alpha f_\alpha \quad (7)$$

165 where,  $K_\alpha$  is the linear moduli,  $f_\alpha$  is the finite lattice strain,  $m_\alpha$  is related to elastic  
166 constants, and  $\alpha$  corresponds to the  $a$ - and  $c$ - axes directions. The finite strain  
167 formulation using LDA yields  $K_a = 476$  GPa and  $K_c = 602$  GPa, whereas GGA  
168 predicts softer linear moduli with  $K_a = 456$  GPa and  $K_c = 539$  GPa (**Figure 1, Table**

169 2). The linear moduli obtained from the finite strain fits to pressure dependence of  $a$ -  
170 and  $c$ - lattice parameters are within 15% of the linear moduli predicted from the  
171 experimental bulk modulus for a cubic Na-majorite i.e.,  $3K_0 = K_\alpha$  ( $\sim 520.5$  GPa,  $l$   
172 refers to linear) (Pacalo et al., 1992). The linear moduli predicted using LDA and  
173 GGA methods, using the finite strain fits of  $a$ - and  $c$ - lattice parameters are consistent  
174 with the zero pressure bulk modulus i.e.,  $K_0^{-1} = 2K_a^{-1} + K_c^{-1}$ .

175 Among the polyhedral units, the dodecahedral sites have the largest zero  
176 pressure volume followed by octahedral and tetrahedral sites with the relation  $V_0^{NaO_8} >$   
177  $V_0^{MgO_8} > V_0^{SiO_6} > V_0^{SiO_4}$ , whereas, the tetrahedral sites are the stiffest followed by the  
178 octahedral and the dodecahedral sites i.e.,  $K_0^{NaO_8} < K_0^{MgO_8} < K_0^{SiO_6} < K_0^{SiO_4}$ . The  
179 relationship holds for both the LDA and GGA predictions (**Table 2**).

180

### 181 3. 2. Elasticity

182 The crystal structure of Na-majorite garnet with  $Na_2MgSi_5O_{12}$  stoichiometry  
183 has tetragonal symmetry ( $I4_1/acd$  spacegroup) and, hence, it has six independent  
184 elastic moduli. We note that both LDA and GGA predictions show that the principal  
185 elastic moduli along the x- and y- axes,  $C_{11}$  is softer than the principal elastic moduli  
186 along the z-axes,  $C_{33}$ . The off-diagonal elastic moduli are nearly identical, i.e.,  
187  $C_{12} \sim C_{13}$ . The shear elastic moduli are also nearly identical, i.e.,  $C_{44} \sim C_{66}$ . At zero-  
188 pressure, the elastic moduli predicted using GGA and LDA bracket the experimental  
189 full elastic moduli tensor based on Brillouin scattering on a cubic Na-majorite (Pacalo  
190 et al., 1992) (**Figure 2**). The principal component, off-diagonal component, and the  
191 shear elastic component of the full elastic tensor moduli stiffen upon compression



192 **(Figure 3)** and the pressure dependence of elastic moduli can be described by finite  
193 strain formulation (in full 4<sup>th</sup> rank tensor notation)

$$194 \quad c_{ijkl} = (1 + 2f_V)^{7/2} [c_{ijkl0} + b_1 f_V + 0.5b_2 f_V^2] - P\Delta_{ijkl}, \quad (8)$$

195 where,  $f_V$  is the Eulerian finite volume strain as defined in eq (6), and (9)

$$196 \quad b_1 = 3K_0(c'_{ijkl0} + \Delta_{ijkl}) - 7c_{ijkl0}, \quad (10)$$

$$197 \quad b_2 = 9K_0^2 c''_{ijkl0} + 3K_0'(b_1 + 7c_{ijkl0}) - 16b_1 - 49c_{ijkl0}, \quad (11)$$

$$198 \quad \text{and } \Delta_{ijkl} = -\delta_{ij}\delta_{kl} - \delta_{ik}\delta_{jl} - \delta_{il}\delta_{jk} \quad (12)$$

199 where  $c'_{ijkl0}$  and  $c''_{ijkl0}$  are the first and second derivative of  $c_{ijkl}$ , with respect to  
200 pressure.  $\Delta_{ijkl} = 1$  for principal and off-diagonal elastic constants ( $c_{ijij}$  in full tensor  
201 and  $c_{ij}$  in Voigt notation, with  $i=1,2,3, i \neq j$ ), -1 for shear constants ( $c_{ijij}$  in full tensor  
202 notation with  $i=1,2,3, i \neq j$  and  $c_{ij}$  in Voigt notation with  $i=4,5,6, i=j$ ), and 0 otherwise  
203 (Karki et al. 2001). The isotropic bulk (K) and shear (G) moduli are determined using  
204 the relations

$$205 \quad K_{\text{Voigt}} = (1/9)[2c_{11} + c_{33} + 2(c_{12} + 2c_{13})], \quad (13)$$

$$206 \quad K_{\text{Reuss}} = [2s_{11} + s_{33} + 2(s_{12} + 2s_{13})]^{-1}, \quad (14)$$

$$207 \quad G_{\text{Voigt}} = (1/15)[2c_{11} + c_{33} - (c_{12} + 2c_{13}) + 3(2c_{44} + c_{66})], \quad (15)$$

$$208 \quad \text{and } G_{\text{Reuss}} = 15[4(2s_{11} + s_{33} - (s_{12} + 2s_{13})) + 3(2s_{44} + s_{66})]^{-1} \quad (16)$$

209  $s_{ijkl}(=c_{ijkl}^{-1})$  are the elastic compliances (inverse of the elastic constant tensor) and  
210 Voigt notation for the compliance tensor is used. Hill averages (**Table 3**) are  
211 estimated as average of Voigt and Reuss bounds.

212 Based on the full elastic moduli results and the resulting compliance tensors,  
213 we have determined the linear compressibility,  $k$ , which is related to the elastic  
214 compliance tensor by the relation,

215  $k = s_{ij}l_i l_j$  (17)

216 where  $s_{ij}$  is the compliance matrix in voigt notation,  $l_i$  and  $l_j$  are the directions of  
217 unit-vector (Nye 1985). For tetragonal symmetry, the linear moduli could be  
218 expressed as-

219  $K_a^{-1} = k_{100} = (s_{11} + s_{12} + s_{13})$  (18)

220 and

221  $K_c^{-1} = k_{001} = (s_{13} + s_{23} + s_{33})$  (19)

222 The linear moduli determined using the compliance matrix show that  $K_c$  is  
223 stiffer than  $K_a$  at low pressures, consistent with the finite strain fit to the pressure  
224 dependence of  $a$ - and  $c$ - lattice parameters (i.e., results using equation 3 to 7). The  
225 linear moduli stiffens upon compression and the relation  $K_c > K_a$  is also held at  
226 higher pressures (**Figure 4**). This is similar to the relationship amongst the principal  
227 components of the full elastic moduli, i.e.,  $C_{33} > C_{11}$ .

228 In our study we notice that the difference in the  $a$ - and  $c$ -lattice parameters is  
229 negligible, indicating that the deviation from cubic symmetry is rather small. We  
230 define the tetragonal strain following previous work (Heinemann et al., 1997) as  
231 difference between the  $a$ - and  $c$ -lattice parameter in the tetragonal cell with a  
232 hypothetical  $a$ -lattice parameter of a cubic cell with same volume, i.e.,

233  $\epsilon_{11} = \frac{a_{tet} - a_{cubic}}{a_{cubic}}$  (20)

234 and

235  $\epsilon_{33} = \frac{c_{tet} - a_{cubic}}{a_{cubic}}$  (21)

236 where,

$$237 \quad a_{cubic} = \frac{2a_{tet} + c_{tet}}{3} \quad (22)$$

238       Based on the single-crystal X-ray diffraction study, at zero-pressures the  
239 tetragonal strain  $\epsilon_{11} > \epsilon_{33}$ . LDA and GGA methods predict similar results at zero  
240 pressures, however the magnitude of the  $\epsilon_{11}$  and  $\epsilon_{33}$  strain is smaller than the  
241 experimental values (**Figure 4**). At higher pressures, the relation between the  
242 tetragonal strains reverses with  $\epsilon_{33} > \epsilon_{11}$  (**Figure 4**). This is likely a manifestation of  
243 the relation  $K_c > K_a$  that, makes  $c$ - lattice parameter slightly larger than the  $a$ -lattice  
244 parameter at higher pressures.

245       The bulk compressional wave ( $V_p$ ) and the shear wave ( $V_s$ ) velocities  
246 obtained from the aggregate bulk and shear modulus (i.e., equation 16 to 19) increase  
247 upon compression (**Figure 5**). We computed the single crystal azimuthal anisotropy  
248 for P- ( $AV_p$ ) and S- waves ( $AV_s$ ) using the formulation for maximum polarization  
249 anisotropy (Mainprice, 1990)-

$$250 \quad AV_p(\%) = \frac{V_{Pmax} - V_{Pmin}}{(V_{Pmax} + V_{Pmin})} \times 200 \quad (23)$$

251 and

$$252 \quad AV_s(\%) = \frac{V_{Smax} - V_{Smin}}{(V_{Smax} + V_{Smin})} \times 200 \quad (24)$$

253 Based on the LDA predictions, the  $AV_p$  anisotropy of Na-majorite is  $\sim 6.5\%$  at low  
254 pressures and reduces to 5 % at transition zone conditions. The  $AV_p$  anisotropy is 5.1  
255 % at low pressures and increases slightly to 5.6 % at transition zone conditions  
256 (**Figure 5**).

257

#### 258 **4. Discussion**

259 Our *first principles* simulations indicate that the compressional wave  
260 velocity,  $V_p$ , and shear wave velocity,  $V_s$ , for Na-majorite are in good agreement with  
261 the ambient condition Brillouin scattering results (Pacalo et al., 1992). In comparison  
262 with the other garnets relevant for the upper mantle and the transition zone regions,  
263 such as, pyrope ( $\text{Mg}_3\text{Al}_2\text{Si}_3\text{O}_{12}$ ), grossular ( $\text{Ca}_3\text{Al}_2\text{Si}_3\text{O}_{12}$ ), almandine ( $\text{Fe}_3\text{Al}_2\text{Si}_3\text{O}_{12}$ ),  
264 and majorite ( $\text{Mg}_4\text{Si}_4\text{O}_{12}$ ), Na-majorite is relatively faster (**Figure 5**). The variation in  
265 the velocity of various garnet could be rationalized in terms of the specific site  
266 chemistry. For example, andradrite is iron-rich which leads to enhanced density and  
267 lower sound wave velocity, whereas in the  $\text{Na}_2\text{MgSi}_5\text{O}_{12}$  garnet, the faster  $V_p$  and  $V_s$   
268 velocities are likely to be related with stiffer bulk and shear modulus which in turn is  
269 due to the substitution of the large sodium cations into the dodecahedral site in  
270 addition to the silicon cations in the octahedral sites. A positive correlation between  
271 the large sodium cations in the dodecahedral sites and stiffer bulk and shear moduli  
272 might seem counter intuitive since among all the polyhedral units, the zero-pressure  
273 bulk modulus of the  $\text{NaO}_8$  polyhedral unit,  $K_0^{\text{NaO}_8}$  is the smallest (**Table 2**). However,  
274 the bulk and shear moduli of garnet crystal structures are controlled by the corner  
275 linked octahedral and tetrahedral units. There is a strong influence of chemistry i.e.,  
276 cation size and charge in these polyhedral units on the bulk and shear modulus of  
277 garnet structure. Unlike pyrope and grossular, the octahedral units in Na-majorite are  
278 occupied by silicon cation ( $\text{Si}^{4+}$ ) which has a smaller size and higher charge compared  
279 to the aluminum cation ( $\text{Al}^{3+}$ ). This leads to smaller unit-cell volume of Na-majorite.  
280 In addition, the  $\text{NaO}_8$  dodecahedral units have large proportion of shared edges and  
281 hence act as braces controlling the amount of rotation between the corner linked  
282 tetrahedral ( $\text{SiO}_4$ ) and the octahedral ( $\text{SiO}_6$ ) units. Since Na is a large cation, the  
283 rotational freedom within the corner linked octahedral and tetrahedral units are

284 minimized. This is also true for grossular that also has large Ca cations residing in the  
285 dodecahedral sites. In comparison, pyrope, with the smaller Mg cations, allows for a  
286 greater degree of freedom for the corner sharing octahedral and tetrahedral units,  
287 resulting in softer bulk and shear modulus (Pacalo et al., 1992).

288 In our static calculations, we have ordered the Na and Mg cations in the 16c  
289 and 8b Wyckoff symmetry. This results in a continuous edge sharing chain of NaO<sub>8</sub>  
290 units parallel to the 4<sub>1</sub> screw axes i.e., parallel to [001] direction. And hence the  
291 degree of rotational freedom for corner sharing octahedral and tetrahedral units along  
292 the 4<sub>1</sub> screw axes (|| [001] direction) is lower than the [100] direction. In contrast,  
293 NaO<sub>8</sub> units do not have continuous edge sharing along [100] and [010] directions  
294 (Figure S1). This is likely explanation for the relation  $C_{33} > C_{11}$  and  $K_c > K_a$  in our  
295 study.

296

## 297 **5. Implications**

298 Na-majorite garnet with Na<sub>2</sub>MgSi<sub>5</sub>O<sub>12</sub> stoichiometry is likely to be an  
299 important component in the deeply subducted oceanic crustal materials (Dymshits et  
300 al., 2013). The velocity of this phase is likely to be faster compared to the majorite  
301 garnet with MgSiO<sub>3</sub> stoichiometry. Na-majorite garnet with the NaAlSi<sub>2</sub>O<sub>6</sub>  
302 stoichiometry, i.e., jadeitic pyroxene, has also been proposed to have higher bulk and  
303 shear modulus (Duffy and Anderson, 1989). However, such end member majorite  
304 garnet composition (NaAlSi<sub>2</sub>O<sub>6</sub>) has not been examined or observed in experimental  
305 studies, and the physical properties have only been estimated (Bass and Anderson,  
306 1984). The effect of temperature on the sound velocities for the Na<sub>2</sub>MgSi<sub>5</sub>O<sub>12</sub>  
307 majorite garnet is unknown and needs to be investigated. However, recent thermal  
308 equation-of-state results indicate that the temperature derivative of the bulk modulus

309  $(\partial K_0 / \partial T)$  for  $\text{Na}_2\text{MgSi}_5\text{O}_{12}$  majorite garnet ( $-0.0023 \text{ GPa K}^{-1}$ ) (Dymshits et al.,  
310 2014) is similar to the  $\text{MgSiO}_3$  majorite garnet ( $-0.0022 \text{ GPa K}^{-1}$ ) (Nishihara et al.,  
311 2005). If the temperature derivative of the shear modulus for  $\text{Na}_2\text{MgSi}_5\text{O}_{12}$  is similar  
312 to  $\text{MgSiO}_3$  majorite garnet (Irifune et al., 2008), then it is likely that Na- majorite  
313 might still remain faster compared to majorite in the transition zone region. However,  
314 reduction of velocity is required to explain the seismologically observed velocity  
315 profiles (Dziewonski and Anderson, 1981) in the mantle and transition zone region.  
316 The velocity reduction could occur through an admixture of Na-majorite and other  
317 alkali bearing aluminous phases such as NAL and CF phases with slower velocity  
318 (Mookherjee et al., 2012).

319 Anisotropy in velocity based on the single crystal elastic constants indicate  
320 that  $\text{Na}_2\text{MgSi}_5\text{O}_{12}$  has slightly larger  $AV_p$  compared to pyrope (Lu et al., 2013) but  
321 lower than grossular (Kawai and Tsuchiya, 2012). Since the Na-majorite phase is not  
322 very anisotropic, velocity reduction through anisotropy could be ruled out. Important  
323 mantle phases such as olivine, wadsleyite, pyroxene, and calcium perovskite are  
324 likely to dominate the seismic anisotropy signatures in the upper mantle and transition  
325 zone region (Stixrude, 2007). Other possible mechanisms that might account for  
326 reductions of seismic velocity might include incorporation of protons in the garnet  
327 structure as reported in experimental studies on pyrope-rich garnet at mantle  
328 conditions (Lu and Keppeler, 1997; Mookherjee and Karato, 2010). Proton  
329 substitutions might lead to further reduction of velocities, as has been observed for the  
330 grossular garnet (O'Neill et al., 1993). However, further investigation on the effect of  
331 chemistry (e.g., sodium, proton, iron) and temperature on the elasticity of majorite  
332 garnet is required at conditions relevant to the lower part of the upper mantle and

333 transition zone region to explain the geophysical observations and to understand the  
334 composition and heterogeneity in the Earth's mantle.

335

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343

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502

## Figure Caption

503 **Figure 1. (a)** Crystal structure of  $\text{Na}_2\text{MgSi}_5\text{O}_{12}$  viewed down the  $c$ -axis. The eight  
504 fold coordinated Na cations (green symbols) occupy the 16e Wyckoff sites and  
505 the Mg cations (yellow symbols) occupy the 8b Wyckoff sites. Also shown are  
506 the corner-sharing octahedral units, i.e., 16c Wyckoff sites and tetrahedral units  
507 16e and 8a Wyckoff sites (**Table 1**). Both the octahedral and tetrahedral units  
508 are fully occupied by the silicon cations represented by blue symbols. **(b)** Plot of  
509 pressure vs. volume, the filled red symbols (LDA) and blue symbols (GGA) are  
510 from this study, grey filled symbols are from experimental study (Dymshits et  
511 al., 2014). The red and blue line represents finite strain fits to LDA and GGA  
512 results. The dashed line represents the finite strain fit to the experimental results  
513 (Dymshits et al., 2014), the black filled circles represent single crystal X-ray  
514 diffraction results (Hazen et al., 1994). The inset shows the plot of Eulerian  
515 linear finite strain vs. normalized pressure along  $a$ - and  $c$ - axes. Red and blue  
516 symbols represent LDA and GGA results respectively. The error bars are  
517 calculated based on an assumed numerical precision in the pressure of  $\pm 0.1$   
518 GPa, **(c)** Discrepancy between the equation of state parameters from  
519 experimental studies and the results from *first principles*- Red and blue bars  
520 represents the discrepancy between the experimental results with the prediction  
521 based on LDA and GGA methods respectively. The zero-pressure volume ( $V_0$ )  
522 and bulk modulus ( $K_0$ ) from LDA and GGA are compared with recent  
523 synchrotron based powder X-ray diffractions (Dymshits et al., 2014). The zero-  
524 pressure lattice parameters, i.e.,  $a_0$  and  $c_0$  from LDA and GGA are compared  
525 with the single-crystal X-ray diffraction results (Bindi et al., 2011). The linear  
526 moduli along the  $a$ - and  $c$ - axes, i.e.,  $K_a$  and  $K_c$  are compared with the linear

527 moduli determined from the bulk modulus of the cubic Na-majorite (Pacalo et  
528 al., 1992).

529 **Figure 2.** The zero pressure full elastic tensor moduli ( $C_{ij}$ ) for  $\text{Na}_2\text{MgSi}_5\text{O}_{12}$ - a  
530 comparison between from Brillouin scattering study (Pacalo et al., 1992) and the  
531 results from *first principles* are shown. The red, blue, and grey symbols  
532 represent LDA, GGA, and the experimental results (Pacalo et al., 1992)  
533 respectively. At the bottom of the figure, the red and the blue bars indicate the  
534 discrepancy between the experimental results (Pacalo et al., 1992) and the  
535 prediction based on LDA and GGA methods, respectively.

536 **Figure 3.** Pressure dependence of the full elastic tensor moduli ( $C_{ij}$ ) for  $\text{Na}_2\text{MgSi}_5\text{O}_{12}$   
537 - (a) principal components, (b) off-diagonal components, (c) shear components,  
538 and (d) bulk and shear moduli. Note- Red and blue symbols represents the  
539 predictions based on LDA and GGA methods respectively. The solid and dashed  
540 lines are finite strain fits to LDA and GGA predictions respectively. The grey  
541 symbols represents Brillouin scattering results at ambient conditions (Pacalo et  
542 al., 1992). The light green shaded region depicts the mantle transition zone  
543 region where Na-majorite is likely to be stable.

544 **Figure 4.** Plot of linear moduli,  $K_a$  and  $K_c$ , derived from the full elastic constant  
545 tensor moduli  $C_{ij}$ , as a function of pressure. The results are consistent with the  
546 linear moduli at zero pressure derived from the finite strain fits to the pressure  
547 dependence of *a*- and *c*-lattice parameters (**Figure 1, Table 2**). The red and blue  
548 symbols refer to predictions based on LDA and GGA methods. The lines refer  
549 to linear fits. The inset shows tetragonal strain,  $\epsilon_{11}$  and  $\epsilon_{33}$  for the single crystal  
550 X-ray diffraction results (grey bars) (Bindi et al., 2011), LDA (red bars), and

551 GGA (blue bars) predictions. The symbols for  $\epsilon_{11}$  are in darker shades whereas  
552 the symbols for  $\epsilon_{33}$  are lighter in shades. At zero pressure,  $a_0 > c_0$  and  $\epsilon_{11} > \epsilon_{33}$   
553 for the experiments, LDA, and GGA predictions. However, at higher pressure  
554 both LDA and GGA predicts a stiffer  $K_c$  and hence at high-pressures the  
555 magnitude of the strains are reversed with  $\epsilon_{33} > \epsilon_{11}$ .

556 **Figure 5a.** Plot of  $V_p$  and  $V_s$  as a function of pressure for garnet with varying  
557 compositions- ‘py’ refers to pyrope (Sinogeikin and Bass, 2000); ‘gr’ refers to  
558 grossular rich garnet (Jiang et al., 2004); ‘and’ refers to andradite rich garnet  
559 (Jiang et al., 2004); ‘maj’ refers to majorite (Irifune et al., 2008); ‘na-maj’ refers  
560 to sodium bearing majorite (Pacalo et al., 1992), and the blue and red symbols  
561 refer to the LDA and GGA results on Na-majorite with  $\text{Na}_2\text{MgSi}_5\text{O}_{12}$   
562 stoichiometry from this study. **(b)** Plot of seismic anisotropy based on single  
563 crystal elastic constant tensor for Na-majorite:  $AV_p$  (circles) and  $AV_s$  (rhombs)  
564 as a function of pressure. The red and blue symbols refer to LDA and GGA  
565 results respectively. The light green shaded region depicts the mantle transition  
566 zone region where Na-majorite is likely to be stable. Also, indicated is the  
567 thermodynamic stability range for garnets with varying chemistry.



1 **Table 1.** Fully relaxed crystal structure with Wyckoff positions, site symmetry, and  
 2 fractional coordinates of atoms of Na<sub>2</sub>MgSi<sub>5</sub>O<sub>12</sub> from LDA, for a unit-cell volume of  
 3 1480 Å<sup>3</sup> (*a*= 11.409, and *c*= 11.368 Å).

Symmetry	Site Symmetry	Wyckoff Symbol	Elements	<i>fractional coordinates</i>		
				<i>x</i>	<i>y</i>	<i>z</i>
Na <sub>2</sub> MgSi <sub>5</sub> O <sub>12</sub>						
I4 <sub>1</sub> / <i>acd</i>	2	16e	Na	0.1263	0.0000	0.2500
4/ <i>mmm</i>	222	8b	Mg	0.0000	0.2500	0.1250
142	-1	16c	Si <sup>VI</sup>	0.0000	0.0000	0.0000
	2	16e	Si <sup>IV</sup>	0.3740	0.0000	0.2500
	-4	8a	Si <sup>IV</sup>	0.0000	0.2500	0.3750
	1	32g	O	0.0360	0.0442	0.6463
	1	32g	O	0.6454	0.0375	0.0430
	1	32g	O	0.0406	0.6504	0.0294

4

1 **Table 2.** Equation of state parameters, linear compressibilities, and compressibilities  
 2 of the polyhedral units.

	$V_0$ $\text{\AA}^3$	$K_0$ GPa	$K'_0$	Reference
	1447.6(0.1)	177.3(0.3)	3.93(0.02)	LDA
	1525.8(0.2)	160.2(0.4)	3.96(0.02)	GGA
	1476.0(0.1)	181.0(0.9)	4.40(1.20)	SXRD <sup>a</sup>
	1484.9(0.3)	173.5(2.6)	*	SCBS <sup>b</sup>
	1485.5(0.3)	191.5(2.5)	4.00( <i>fixed</i> )	SCXRD <sup>c</sup>
	$a_0$ $\text{\AA}$	$K_a$ GPa		Reference
	11.326(0.0002)	476.4(2.9)		LDA
	11.529(0.0001)	455.6(1.9)		GGA
	11.409(0.001)	520.5(7.8)		SCBS <sup>b</sup>
	$c_0$ $\text{\AA}$	$K_c$ GPa		Reference
	11.284(0.0005)	601.9(2.8)		LDA
	11.479(0.0006)	539.3(2.3)		GGA
	11.409(0.001)	520.5(7.8)		SCBS <sup>b</sup>
<i>Compressibility of polyhedral units</i>				
	$V_0$ $\text{\AA}^3$	$K_0$ GPa	$K'_0$	Reference
SiO <sub>4</sub> (16e)	2.193(0.000)	314.9(2.1)	4.27(0.11)	LDA
	2.238(0.002)	287.1(6.7)	5.16(0.31)	GGA
SiO <sub>4</sub> (8a)	2.156(0.000)	318.3(3.1)	4.86(0.17)	LDA
	2.279(0.001)	281.8(3.0)	4.64(0.13)	GGA
SiO <sub>6</sub>	7.535(0.000)	247.8(0.6)	5.27(0.04)	LDA
	7.919(0.003)	212.9(2.0)	5.37(0.10)	GGA
MgO <sub>8</sub>	19.513(0.004)	126.0(0.5)	3.45(0.02)	LDA
	20.791(0.022)	114.7(1.7)	3.45(0.06)	GGA
NaO <sub>8</sub>	21.232(0.005)	123.1(0.5)	3.21(0.02)	LDA
	22.609(0.012)	111.5(0.8)	3.29(0.03)	GGA

3 <sup>a</sup>synchrotron X-ray diffraction (Dymshits et al., 2014); <sup>b</sup>single-crystal full elastic  
 4 moduli tensor using Brillouin scattering (Pacalo et al., 1992) \*this study was  
 5 conducted at ambient conditions and hence no pressure derivatives were reported;  
 6 <sup>c</sup>single-crystal X-ray diffraction (Hazen et al., 1994); LDA- local density  
 7 approximation, *this study*; GGA- generalized gradient approximation, *this study*

**Table 3.** Elasticity of garnet as a function of pressure and finite strain fit with the pressure derivatives ( $dC_{ij}/dP$ ) shown in brackets.

V ( $\text{\AA}^3$ )	$\rho$ ( $\text{gm/cm}^3$ )	P	$C_{11}$	$C_{33}$	$C_{12}$	$C_{13}$	$C_{44}$	$C_{66}$	$K_{\text{VRH}}$	$G_{\text{VRH}}$	$V_P$	$V_S$
(GPa)												
(km/sec)												
<b>Na<sub>2</sub>MgSi<sub>5</sub>O<sub>12</sub> (<math>I4_1/acd</math>) (LDA)</b>												
1480	3.61	-3.8	296.2	344.4	87.7	89.7	106.0	107.6	163.0	108.4	9.2	5.5
1440	3.72	0.9	323.9	368.0	102.6	101.4	109.2	110.3	180.5	112.9	9.4	5.5
1400	3.82	6.3	354.2	396.9	119.2	117.0	114.9	115.9	201.1	119.1	9.7	5.6
1360	3.93	12.5	386.5	428.8	139.0	135.0	120.8	122.1	224.2	125.4	10.0	5.6
1320	4.05	19.6	421.3	462.1	161.6	155.9	125.9	128.6	250.0	131.3	10.2	5.7
1280	4.18	27.8	459.1	497.9	188.8	181.7	133.0	134.6	279.9	137.5	10.5	5.7
1240	4.31	37.2	498.3	535.2	217.2	209.8	138.6	138.4	311.6	142.6	10.8	5.7
1200	4.46	48.1	542.3	578.6	250.1	242.6	146.3	149.3	348.1	150.1	11.1	5.8
<i>finite strain fit</i>												
1448	3.70	0.0	318.5 (5.8)	364.1 (5.5)	99.6 (3.2)	98.6 (2.9)	109.6 (0.9)	111.0 (1.0)	177.0 (3.9)	112.1 (1.2)	9.4	5.5
<b>Na<sub>2</sub>MgSi<sub>5</sub>O<sub>12</sub> (<math>I4_1/acd</math>) (GGA)</b>												
1560	3.43	-3.4	254.8	309.1	71.8	79.4	95.0	95.0	141.5	96.0	8.9	5.3
1520	3.52	0.6	294.7	340.0	93.5	96.0	101.9	103.8	166.3	104.3	9.3	5.4
1480	3.61	5.2	317.4	363.0	102.8	106.0	107.7	110.4	180.5	110.6	9.5	5.5
1440	3.72	10.4	345.4	386.8	118.2	115.9	111.7	113.6	197.3	115.7	9.7	5.6
1400	3.82	16.3	378.1	415.3	137.6	132.7	119.2	121.3	219.6	123.0	10.0	5.7
1360	3.93	23.1	419.8	456.0	164.9	158.3	132.1	130.5	250.9	133.2	10.4	5.8
1320	4.05	30.9	450.9	491.9	186.9	179.8	131.9	132.4	276.1	135.5	10.6	5.8
1280	4.18	39.9	485.0	521.6	210.7	204.1	139.5	134.6	303.2	140.8	10.8	5.8
<i>finite strain fit</i>												
1526	3.51	0	282.2 (7.1)	331.4 (6.4)	84.3 (3.6)	90.1 (3.1)	100.9 (1.3)	100.8 (1.6)	159.4 (3.9)	103.5 (1.3)	9.2	5.4

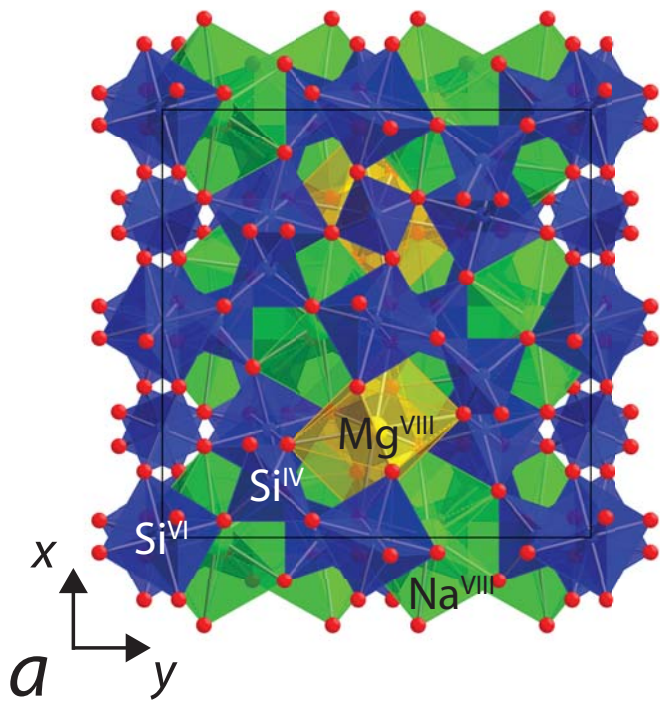
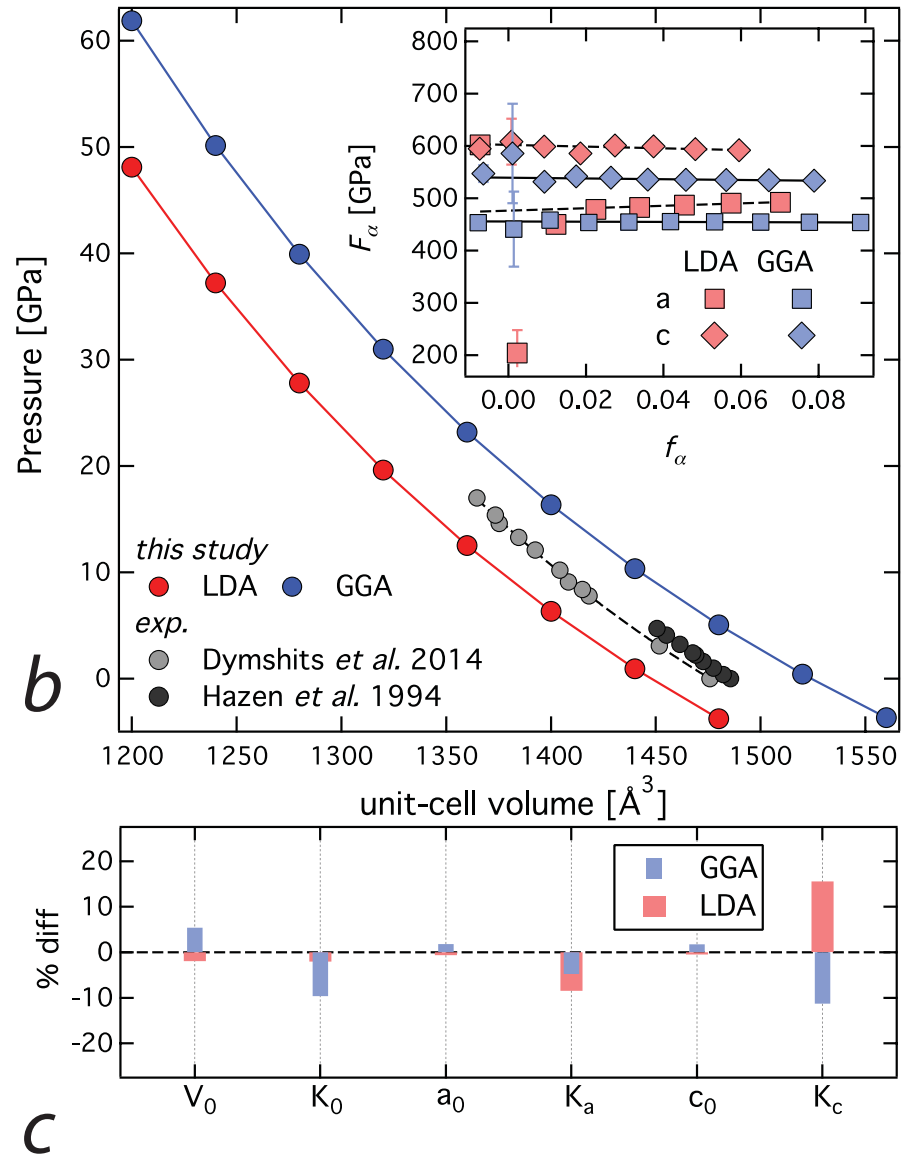


Figure 1



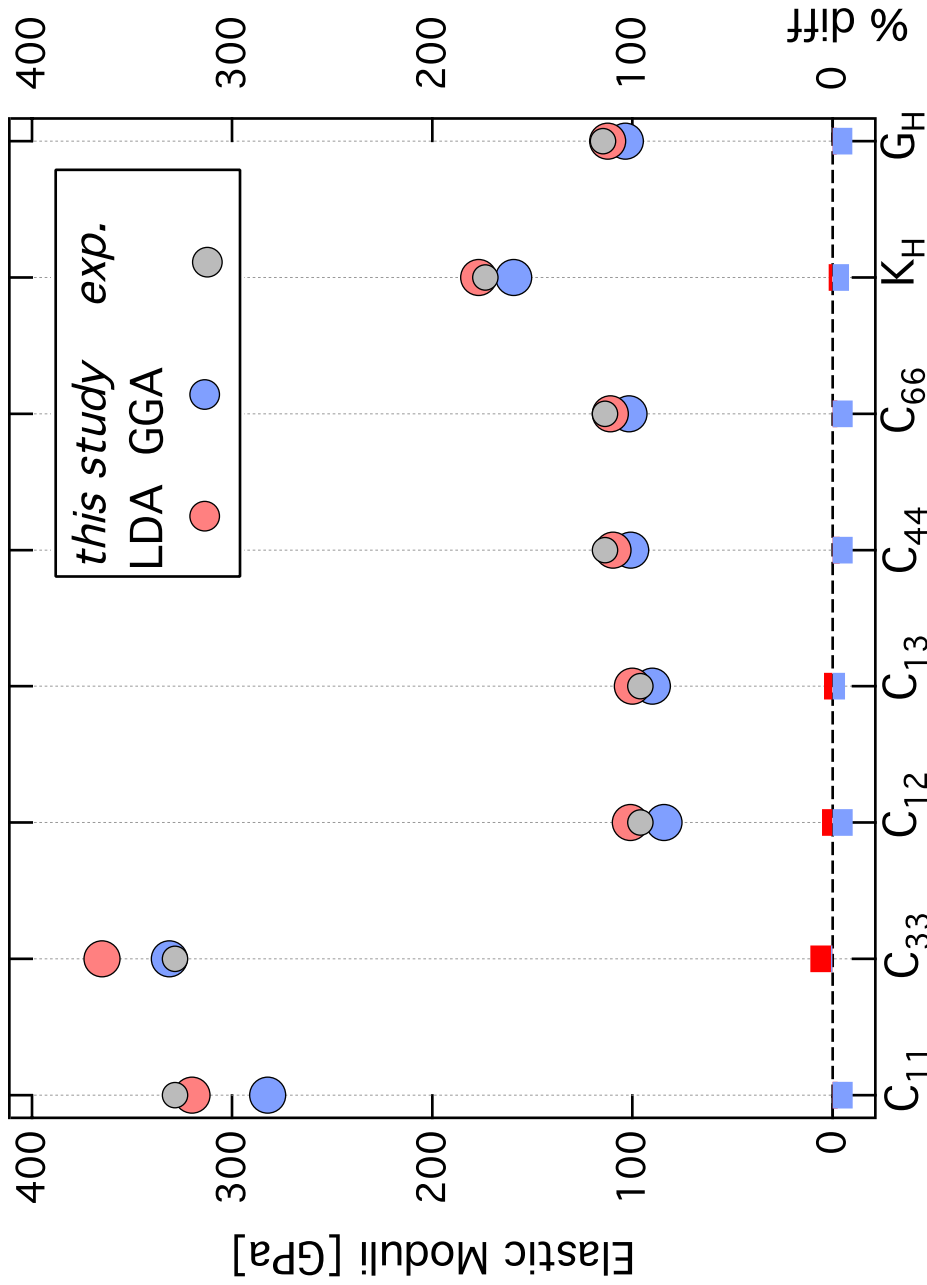
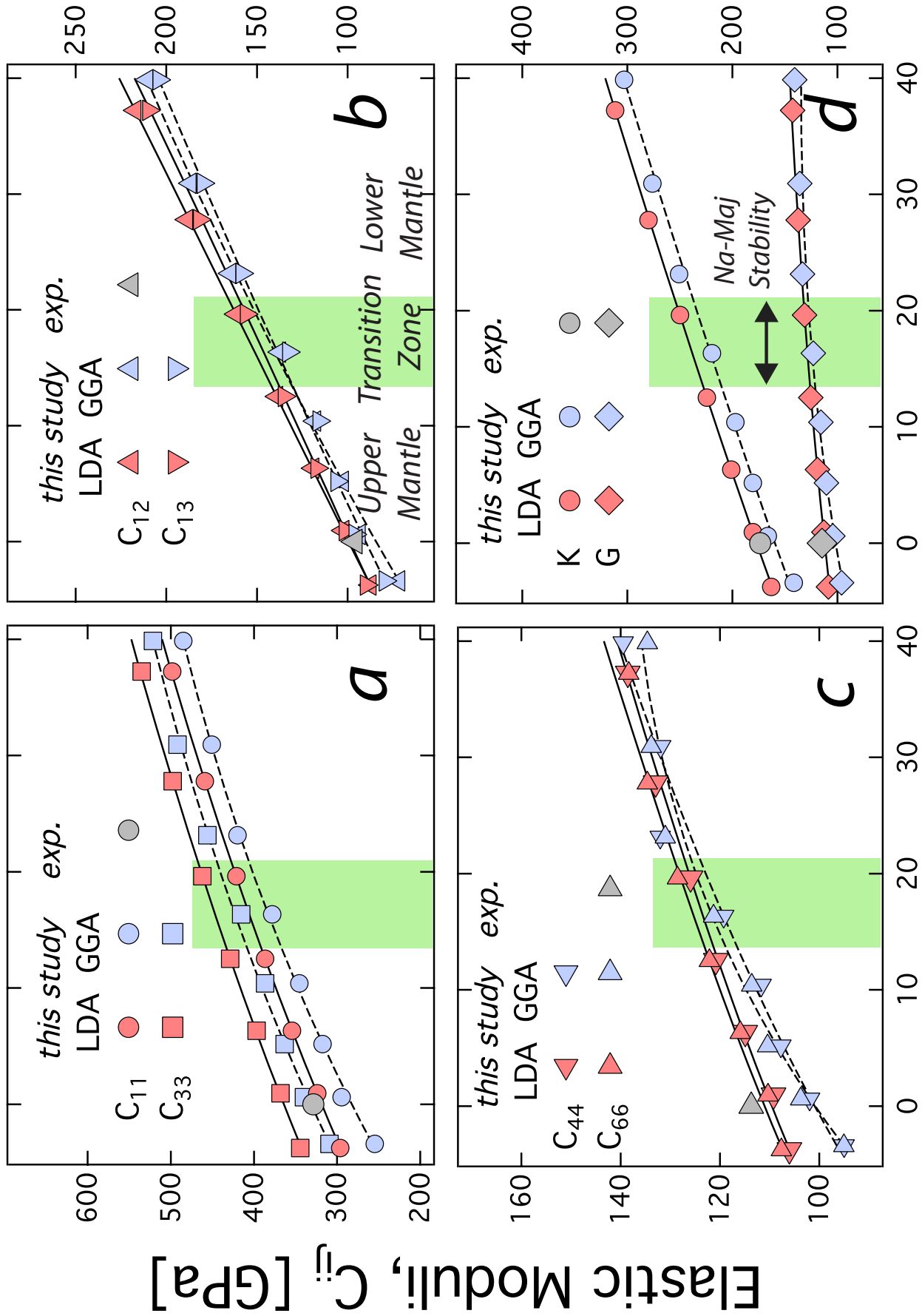


Figure 2

# Elastic Moduli, $C_{ij}$ [GPa]



Pressure [GPa]

Figure 3

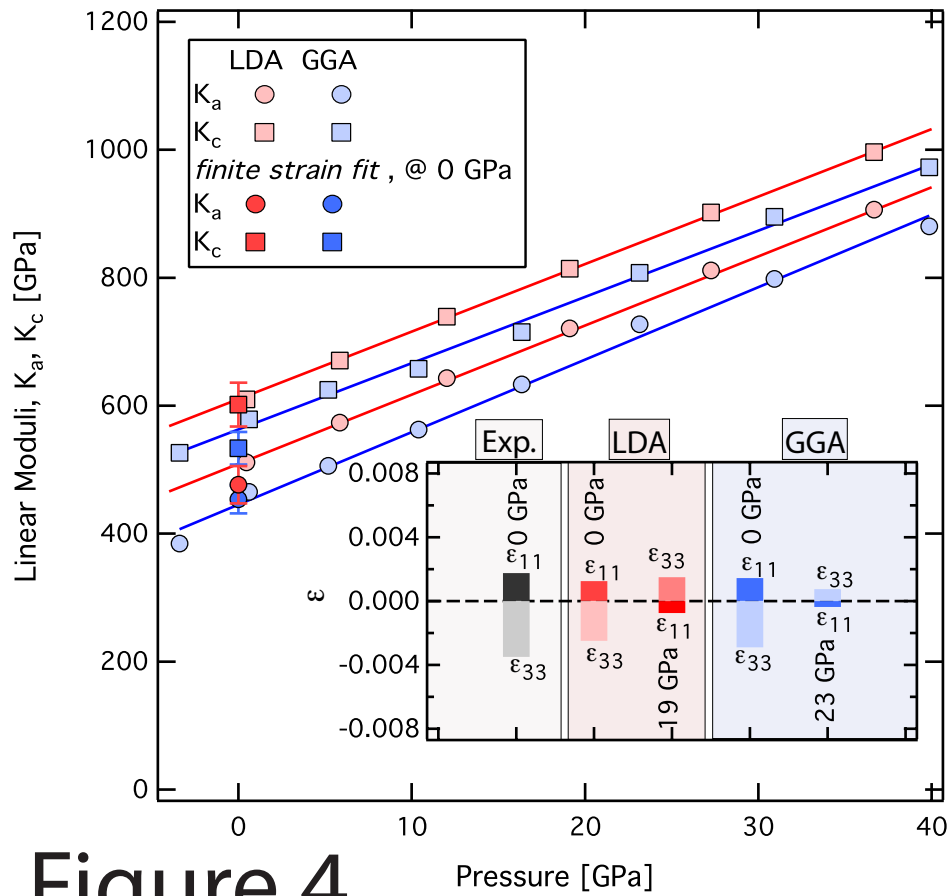


Figure 4

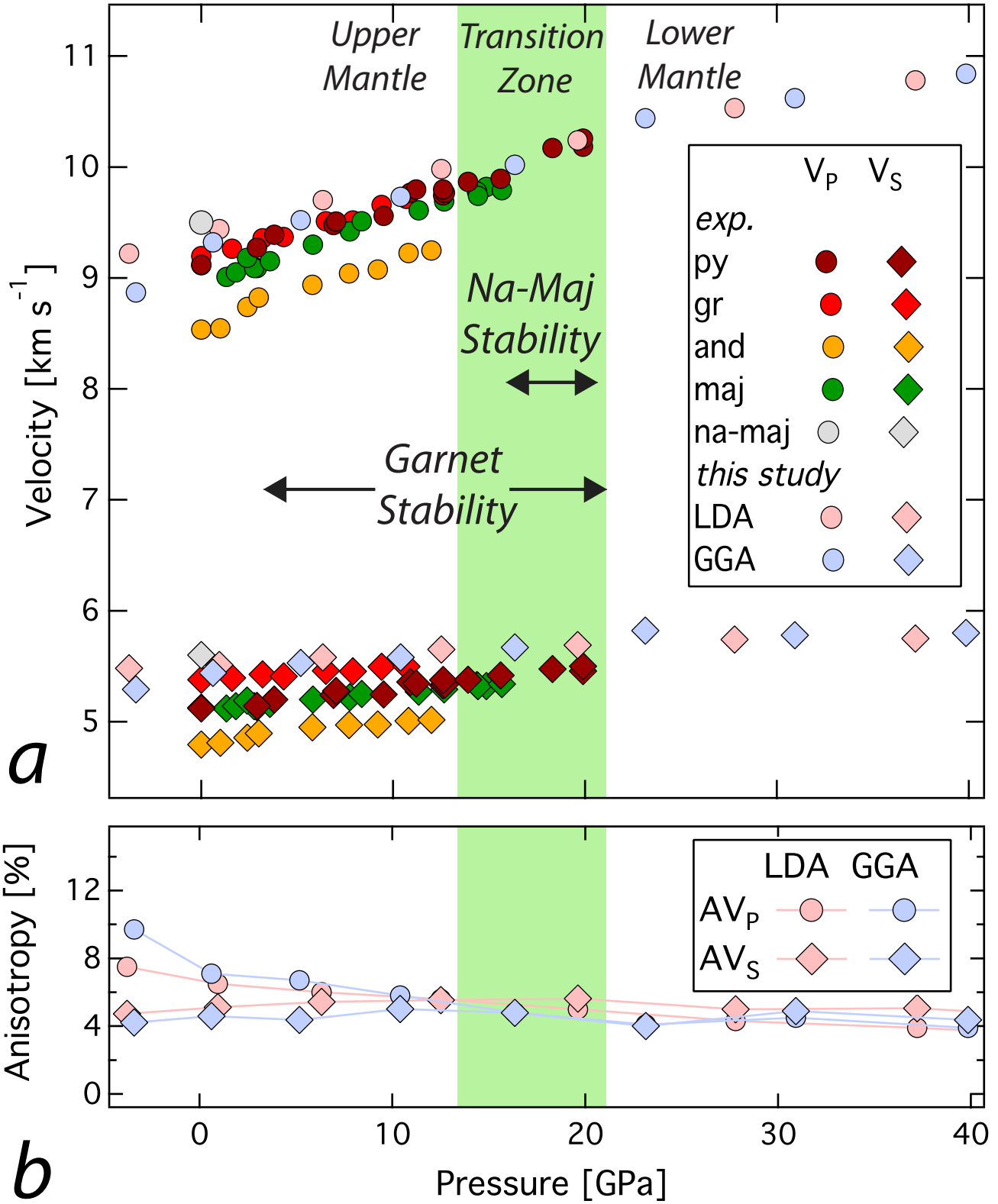


Figure 5