

1 **Revision 3**

2 **Z-contrast imaging and *ab initio* study on “d” superstructure in sedimentary**
3 **dolomite**

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23 **ABSTRACT**

24 Nano-precipitates with tripled periodicity along the *c*-axis are observed in a Ca-rich dolomite
25 sample from Proterozoic carbonate rocks with “molar tooth” structure. This observation is
26 consistent with previous description of *d* reflections. High-angle annular dark-field STEM
27 imaging (or *Z*-contrast imaging) that avoids dynamic diffraction as seen in electron diffraction
28 and high-resolution TEM imaging modes, confirms that *d* reflections correspond to nanoscale
29 precipitates aligned parallel to (001) of the host dolomite. The lamellae precipitates have a cation
30 ordering sequence of Ca-Ca-Mg-Ca-Ca-Mg along the *c* direction resulting in a chemical
31 composition of $\text{Ca}_{0.67}\text{Mg}_{0.33}\text{CO}_3$. This superstructure is attributed to the extra or *d* reflections,
32 thus is referred to as the *d* superstructure in this study. The structure can be simply described as
33 interstratified calcite/dolomite. The crystal structure of the *d* superstructure calculated from
34 density functional theory (DFT) has a space group of *P31c* and has *a* and *c* unit cell parameters
35 of 4.879 Å and 16.260 Å, respectively, values between those of dolomite and calcite. The
36 detailed structural characteristics and parameters obtained from *ab initio* calculations are also
37 reported in this paper. The method of combining *Z*-contrast imaging and *ab initio* calculation can
38 be used for solving structures of other nano-precipitates and nano-phases.

39

40 INTRODUCTION

41 The dolomite($R\bar{3}c$) structure has alternating Ca^{2+} and Mg^{2+} cation layers along the c -axis with
42 the triangular CO_3^{2-} anion layers lying between two cation layers. Since the size of Mg^{2+} ions is
43 smaller than that of Ca^{2+} , the CO_3^{2-} layers are closer to Mg^{2+} layers. The lack of c glides in the
44 dolomite structure due to the cation ordering causes the occurrence of extra reflections (b
45 reflections, $h\bar{h}0l$, $l = \text{odd}$) in dolomite diffraction patterns compared to that of calcite (a
46 reflections only) (Reeder, 1992). Two additional reflections (c and d reflections) have been
47 observed in natural sedimentary dolomite. Previously reported c reflections were interpreted as
48 being a result of cation ordering within dolomite basal planes (Van Tendeloo et al., 1985). Our
49 recent STEM work confirmed that the “ c ”-reflections could result from multiple diffraction
50 between the host dolomite and twinned Mg-calcite nano-lamellae under TEM imaging and
51 diffraction modes (Shen et al., 2013). The d reflection was first observed in Devonian dolomite
52 samples by Wenk and Zenger (1983) and have been also found in ankerite samples (Rekesten,
53 1990). The d reflections occur as satellites around a and b reflections with diffraction vector $\sim 1/3$
54 $(000l)^*$ and are usually streaking along c^* direction (Wenk and Zenger, 1983; Wenk and Zhang,
55 1985; Van Tendeloo et al., 1985).

56

57 Dynamical diffraction in transmission electron microscopy (TEM) mode has been a major
58 problem for structure determination as shown in previous work of analyzing “ c ” superstructures
59 (Shen et al., 2013). The scanning transmission electron microscopy (STEM) method uses the
60 high-angle annular dark-field (HAADF) detector to give the most highly localized 1s Bloch state
61 imaging, which eliminates most of the obvious effects of dynamical diffraction (Pennycook,

62 2002). With the advantage of a spherical aberration corrector, the resolution of HAADF STEM
63 or Z-contrast images is only limited by the size of the 1s Bloch state that is $\sim 0.6\text{-}0.8 \text{ \AA}$
64 (Pennycook et al., 2000). The intensity of Z-contrast images is dependent on the atomic number
65 of atoms through the $\sim Z^2$ dependence of the Rutherford scattering cross-section, which thus
66 provides chemical information for the material (Kirkland, 1998; Pennycook et al., 2000). A study
67 of microstructures in natural dolomite samples using Z-contrast images may help find answers to
68 previous observations of superstructures and explore new microstructures in dolomite.

69

70 In addition to experimental studies in mineralogy, the application of *ab initio* calculations of
71 crystal structure, phase stability, and physical properties of minerals at given pressure and
72 temperature has increased in the past few years (Ogonov et al., 2006; Barnard and Xu, 2008;
73 Chatterjee and Saha-Dasgupta, 2010; Stackhouse et al., 2010). Density functional theory (DFT)
74 uses the functional of electron density to solve the Schrödinger equation for a many electron
75 system to acquire the minimum energy of the system (Scholl and Steckel, 2009). This method
76 can calculate the enthalpy of a system at 0K and the corresponding structure. Since STEM work
77 only provides topological information for the crystal structure, the use of the DFT method is
78 needed in this study to calculate the detailed crystal structure of the *d* phase and confirm our
79 model for the *d* superstructure by comparing the energetic stability of our model to another
80 model proposed in literature.

81

82 The “molar tooth” structure refers to vertical and horizontal ribbons and blobs of fine-grained
83 calcite in a dolomite host (Bauerman, 1885). This structure has only been recognized in

84 Mesoproterozoic and Neoproterozoic marine carbonate rocks with a few exceptions (Frank and
85 Lyons, 1998). The origin and the temporally limited occurrence of molar tooth carbonate have
86 bewildered geologists for over a century (James et al., 1998; Pratt, 1998; Meng and Ge, 2002;
87 Marshall and Anglin, 2004; Pollock et al., 2006; Long, 2007; Kuang et al., 2012). However, no
88 previous work on the nano-scale mineralogy of carbonates with the “molar tooth structure” has
89 been conducted. In this present study, STEM imaging and DFT calculations were combined to
90 provide a complete crystallographic description of the *d* superstructure.

91

92 **SAMPLES**

93 The “molar tooth” carbonate samples were collected near Hungry Horse Dam, Montana, from
94 outcrops of the Helena Formation of the Mesoproterozoic Belt Group (Frank and Lyons, 1998).
95 The sample HHL-00H was chosen for STEM analysis. In the hand specimen of HHL-00H
96 (Figure 1), the thin sinuous vertical and horizontal ribbons with width of ~0.2-1.5cm intersect
97 each other. The “molar tooth” is composed of clean and fine-grained calcite crystals with similar
98 size (~10 μm). The “molar tooth” host has three major phases: dolomite, calcite, and quartz
99 (Figure 2). K-feldspar, illite, chlorite, rutile, and apatite are also present in the “molar tooth” host
100 rock. The XRD powder analysis of the “molar tooth” host rock shows that the dolomite is cation-
101 ordered with the presence of sharp (105) and (009) peaks. The dolomite has a composition of
102 $\sim\text{Ca}_{1.02}\text{Mg}_{0.98}(\text{CO}_3)_2$ based on the d_{104} value (2.890Å) and the relationship between d_{104} values
103 and MgCO_3 content in ordered dolomite (Goldsmith and Graf, 1958; Zhang et al., 2010).

104

105 **METHODS**

106 Specimens for STEM measurements were prepared by ion milling. Ion milling was performed
107 with a Fischione 1010 ion mill operated at an accelerating voltage of 4 kV and an incident ion-
108 beam angle of 10° , followed by gentler milling at an accelerating voltage of 2.6 kV and an
109 incident angle of 10° in order to reduce surface amorphous material. The ion-milled samples
110 were lightly carbon coated. The microstructures in the dolomite crystals were examined by using
111 a spherical aberration-corrected FEG- STEM (Titan 80-200) operating at 200 kV at the
112 University of Wisconsin-Madison. This instrument can image single atoms with ~ 0.1 nm or
113 better spatial resolution in STEM mode. Probe current was set at 24.5 pA. Camera length for the
114 image acquisition was set at 160 mm. Collection angle of the HAADF detector for acquiring the
115 Z-contrast images ranges from 54 to 270 milliradians (mrad).

116

117 The DFT calculations were performed by using the Vienna *ab initio* simulation package (VASP)
118 (Kresse et al., 1996). The general gradient approximation (GGA) with the Perdew, Burke, and
119 Ernzerhof (PBE) parameters was employed (Perdew et al., 1996). The projector-augmented wave
120 (PAW) method with an energy cutoff of 600 eV was used. A conventional hexagonal supercell
121 of calcite derived structures including 30 atoms or 6 chemical formula units was used. We tested
122 k-point convergence and a mesh of $3 \times 3 \times 1$ was found to be sufficient for the system. Two
123 starting structures for the *d* phase: one with calcite's unit cell parameters from experiments
124 (Graf, 1969) and another with dolomite's (Beran and Zemann, 1977). All the initial structures
125 were optimized using the static energy minimization scheme, where both the shape and volume
126 of the cell were allowed to relax. The structure with minimum energy calculated from the
127 previous step was further calculated by *ab initio* molecular dynamics simulations at 10K to better

128 explore the local minimum. The powder and electronic diffraction patterns of calculated
129 structures were generated by CrystalDiffract[®] and SingleCrystal[™] respectively.

130

131 **HIGH-RESOLUTION (S)TEM OBSERVATIONS**

132 In TEM images, the modulated microstructures with strong strain contrast are prevalent through
133 the “molar tooth” host dolomite. Calcite inclusions that are a common phenomenon in
134 sedimentary dolomites can be easily recognized because of being free of modulations (Figure 3).
135 Diffuse streaks along c^* occur in the electron diffraction pattern of the host dolomite (Figure 4).
136 Some maxima of the streaks are about one third of d_{006}^* (Fig. 4, and also see diffraction patterns
137 of Wenk and Zenger, 1983; Wenk and Zhang, 1985). According to the Fast Fourier Transform
138 (FFT) patterns of different areas in the dolomite images, the streaks in the diffraction pattern
139 come from the precipitates with linear features that are parallel to (001) (Figure 5). However, the
140 precipitates themselves are arranged in such a way that they are roughly parallel to $(1\bar{1}4)$. The
141 observations above match the features of d reflections that were first described by Wenk and
142 Zenger (1983). It was proposed that the streaks parallel to c^* could be from stacking disorder
143 (Van Tendeloo et al., 1985).

144

145 The dark areas of the precipitates in the bright field (BF) image under STEM mode become
146 bright areas in the HAADF image (Figure 6), which suggests higher Ca contents in the
147 precipitates than the host dolomite. The occurrence of d reflections in the FFT pattern of the
148 HAADF image excludes the possibility that they are caused by multiple diffraction. In the
149 HAADF of high magnification or Z-contrast image (Figure 7), the alternating bright and dark

150 layers along c axis represent the alternating Ca and Mg layers in the normal dolomite structure.
151 However, the d dolomite precipitate has the cation sequence of Ca-Ca-Mg-Ca-Ca-Mg- (Figure
152 7). The FFT pattern of this domain shows that the d reflections are attributed to this
153 superstructure. This observation is different from the previous model for the d superstructure in
154 which every third Mg layer is replaced by a Ca layer producing a sequence of Ca-Mg-Ca-Ca-Ca-
155 Mg- (Wenk and Zhang, 1985). But both sequences produce the same chemical stoichiometry of
156 $\text{Ca}_{0.67}\text{Mg}_{0.33}\text{CO}_3$. However, the repetition along the c -axis is doubled in the previous model with
157 respect to the observed Ca-Mg ordering in the Z -contrast image (Figure 7).

158

159 Mg-bearing calcite precipitates are also observed in the dolomite (Figure 8). Along the $(\bar{1}02)$
160 trace, the calcite exsolution region has six consecutive Ca layers as opposed to the dolomite host
161 that has alternating Ca and Mg layers. In the line profile of the calcite region, some Ca columns
162 have slightly lower intensities than the pure Ca columns in the dolomite host, which suggests the
163 existence of a small amount of Mg in this calcite precipitate (Figure 8). This calcite exsolution is
164 similar to Mg-calcite precipitates in Ca-rich dolomite (Shen et al., 2013).

165

166 **DFT CALCULATIONS**

167 The structural parameters for the optimized d superstructure from DFT calculations are listed in
168 Table 1. Tables 2 and 3 compare the calculated lattice parameters of dolomite and calcite
169 structures with the reported experimental data. The calculated equilibrium volumes for both
170 dolomite and calcite are slightly smaller than the reported data, the underestimation of the c
171 parameter being the major contribution, but are still within the range of previous theoretical

172 calculation data (see Table 2 and 3). The calculated values are for the structures at 0K. The
173 reported experimental values were measured at ambient environment. Temperature could be a
174 factor for the small discrepancy between calculated and measured the unit cell volumes. A small
175 discrepancy between calculated and experimental values of lattice parameters is not uncommon
176 for DFT calculations and may result from the use of an approximate exchange-correlation
177 potential (Hossain et al., 2011). In spite of this discrepancy, the trend found in experimental data
178 is maintained in our DFT calculations. The calculated d superstructure has an a parameter close
179 to that of dolomite but has a c parameter closer to stoichiometric calcite. This finding is
180 consistent with the observation from the diffraction patterns of the d superstructure that the
181 difference between a parameters of dolomite and the d superstructure is smaller than that
182 between c parameters, even though the DFT calculations are unconstrained bulk structure
183 calculations. This trend is reasonable because superstructure precipitates share the (001) interface
184 with the host dolomite. Smaller differences in the a dimensions between the host dolomite and d
185 superstructure would cause less strain at the interface.

186

187 The C-O bonds in CO_3^{2-} groups are rigid and the C-O distances in both experimental and
188 calculation data are constant no matter what the actual composition of the carbonate mineral is. It
189 is interesting to note that the C-O distances in the d superstructure are divided into two categories
190 (Table 1): 1.294Å when a CO_3^{2-} layer is between Ca^{2+} and Mg^{2+} layers along the c axis; and
191 1.297Å when a CO_3^{2-} layer is in between two Ca^{2+} layers. DFT calculations predict slightly
192 shorter Ca-O and Mg-O bonds in dolomite and calcite than experimental data (Table 2 and 3).
193 The Mg-O distance in the d superstructure increases from 2.069Å in calculated dolomite to 2.085
194 Å. The Ca-O distance in the d superstructure differs depending on the oxygen positions; the Ca-

195 O distances (2.374 Å) are larger when the oxygen ions are from CO₃²⁻ group sitting in between
196 Ca²⁺ and Mg²⁺ layers than those (2.355 Å) from CO₃²⁻ group between two Ca²⁺ layers. This is
197 because that the CO₃²⁻ layers are closer to Mg²⁺ layers than Ca²⁺ layers due to the smaller Mg²⁺
198 radius. The inversion center is missing in the *d* superstructure while the *c* glide is retained. The
199 existence of (100) and (200) reflections in the diffraction pattern proves that it is not a
200 rhombohedral unit cell, but a primitive hexagonal (Figure 9C). Therefore, the space group is
201 determined to be *P31c* (No. 159). The atom coordinates and symmetry equivalent positions of
202 the *d* superstructure are reported in Table 1.

203

204 The superstructure with cation sequence of Ca-Mg-Ca-Ca-Ca-Mg-Ca that was proposed to
205 explain the *d* reflections was referred to as δ structure (Wenk and Zhang, 1985; and Wenk et al.,
206 1991) (see Figure 9A for details). The calculated δ structure has slightly larger unit cell
207 parameters ($a = 4.883\text{\AA}$ and $c = 16.281\text{\AA}$) than the *d* superstructure indicated here. The enthalpy
208 of a unit cell of δ structure is slightly higher than that of the *d* superstructure by 0.54 kJ/mol per
209 MCO₃ unit. The previously proposed δ structure is unstable with respect to the current *d*
210 superstructure. The proposed δ structure model was based on the assumption that overlapped
211 diffraction patterns from host dolomite and *d* superstructure were from the precipitates only
212 (Wenk and Zhang, 1985). Careful examination of their diffraction pattern (Fig. 1A of Wenk and
213 Zhang, 1985) and our FFT pattern (Figure 9D), show that the position of 003 is not half way
214 between 002 and 004 of the *d* superstructure, and the position of 009 is not midway between 008
215 and 0010 of the *d* superstructure. The *d* superstructure does not have reflections with odd *l* due to
216 its *c*-glide. A powder X-ray diffraction pattern with d_{104} value of 2.930 Å for the *d* superstructure
217 is also calculated (Fig. 10).

218 The observed intermediate phases are analogous to those locally ordered domains in
219 mixed-layer clay minerals, such as interstratified chlorite/serpentine, chlorite/biotite, and
220 chlorite/pyrophyllite minerals (Banfield and Bailey, 1996; Xu and Veblen, 1996; Xu et al., 1996;
221 Wang and Xu, 2006). The observed intermediate phase between calcite and dolomite may be
222 described as interstratified calcite/dolomite, instead of δ -dolomite or *d*-dolomite.

223

224 **IMPLICATIONS**

225 The calculated energies of calcite ($\text{Ca}(\text{CO}_3)$), dolomite ($\text{CaMg}(\text{CO}_3)_2$), and *d* superstructure
226 ($\text{Ca}_2\text{Mg}(\text{CO}_3)_3$) are 3620.64, 7132.37 and 10740.22 kJ/mol, respectively. The energy for the *d*
227 superstructure ($\text{Ca}_2\text{Mg}(\text{CO}_3)_3$) is higher (~ 4.2 kJ/mol per MCO_3 unit) than the sum of the
228 calculated energies from dolomite ($\text{CaMg}(\text{CO}_3)_2$) and calcite (CaCO_3) end members. The *d*
229 superstructures serve as a metastable phase with respect to the stoichiometric dolomite and
230 calcite. The *d* superstructures have a maximum of 4 repeats of Ca-Ca-Mg- along the *c* axis ($<$
231 35\AA) in our sample and are only stable within dolomite host in the form of nano-precipitates. It
232 is very difficult for extra Ca ions in the dolomite structure to diffuse through carbonate layers
233 and to congregate to form lamellae parallel to the basal plane at low temperature (Wenk et al.,
234 1991; Shen et al., 2013). This explains all the reported occurrences of nano-precipitates of *d*-
235 reflections in natural samples. Wenk et al. (1991) summarized the various proposed
236 superstructures of Ca-Mg carbonates. By using Z-contrast imaging, we can image the cation
237 ordering directly and propose a more accurate structure model. By using the DFT method, we
238 can calculate the detailed structures and explore the energetics of these metastable nano-phases.

239 The methods may be applied to understand many other nano-phases, where there are challenges
240 or artifacts by using other methods.

241

242

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332

Figure 1. (A) The outcrop shows the brittle deformation inside the carbonate layers with “molar tooth” structure. (B) Sinuous dark blue ribbons of fine crystalline calcite (“molar tooth”) exist in the dolomite host that is weathered into buff color. (C) Weathered surface of HHL-00H specimen. (D) “Molar tooth” becomes light color in fresh cut surface of HHL-00H specimen.

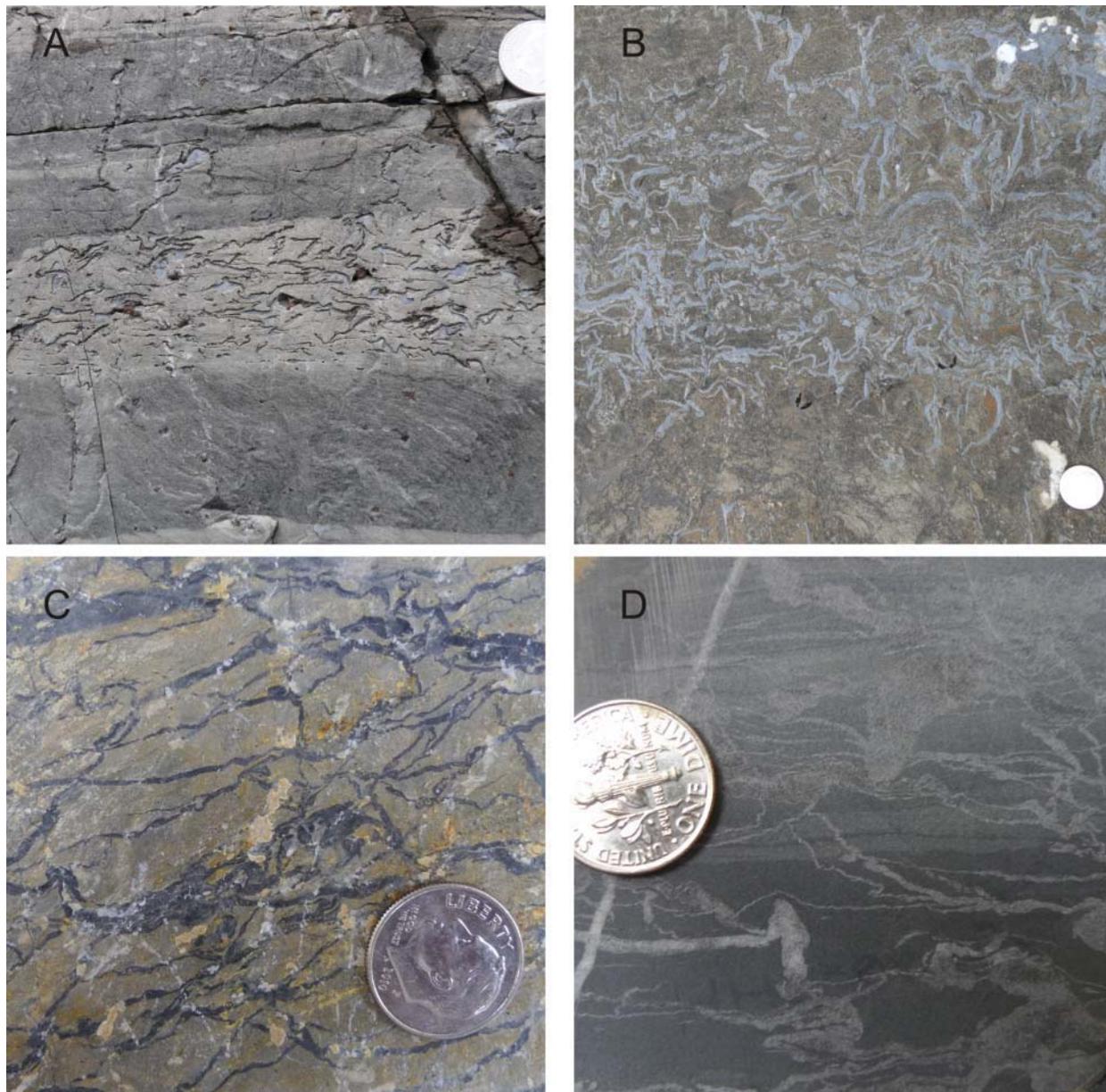


Figure 2. Powder XRD pattern of “molar tooth” host and “molar tooth” in sample HHL-00H.

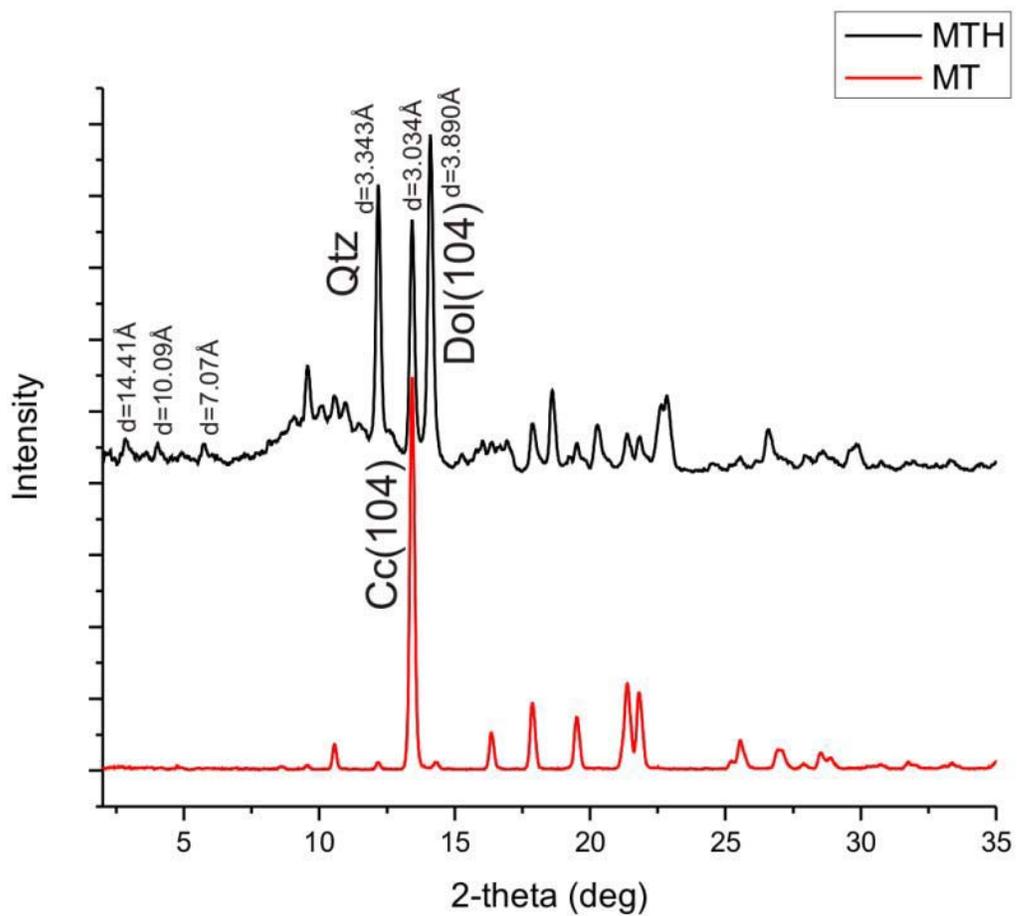


Figure 3. TEM image shows that calcite inclusion that is free of modulations or strain contrast in Ca-rich dolomite host.

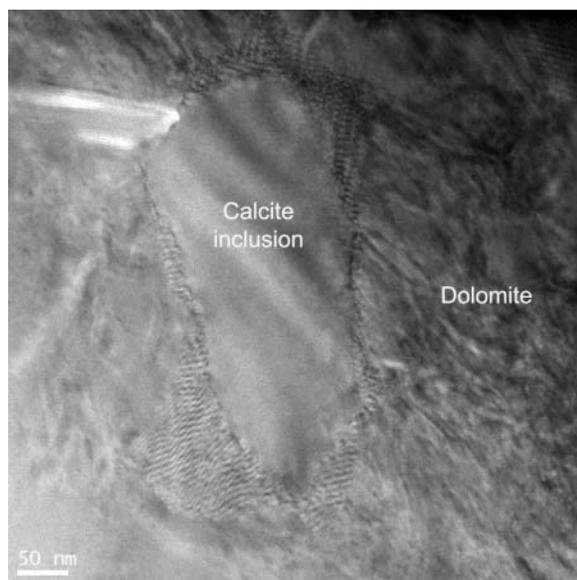


Figure 4. The diffuse streaks along c^* occur in the diffraction pattern of Ca-rich dolomite from Helena Formation, Montana. Some diffraction maxima occur at around $1/3$ of d_{006}^* .

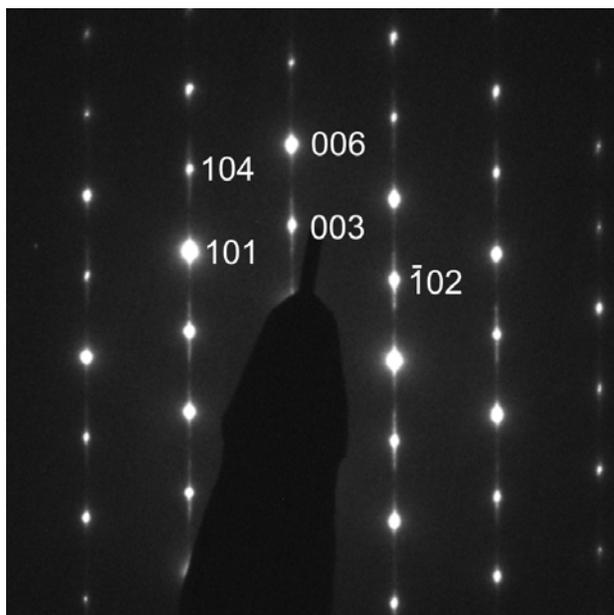


Figure 5. FFT patterns from TEM image show that the streaking and “*d*” reflections are from the precipitates (compare the FFT patterns from outlined areas of a and b).

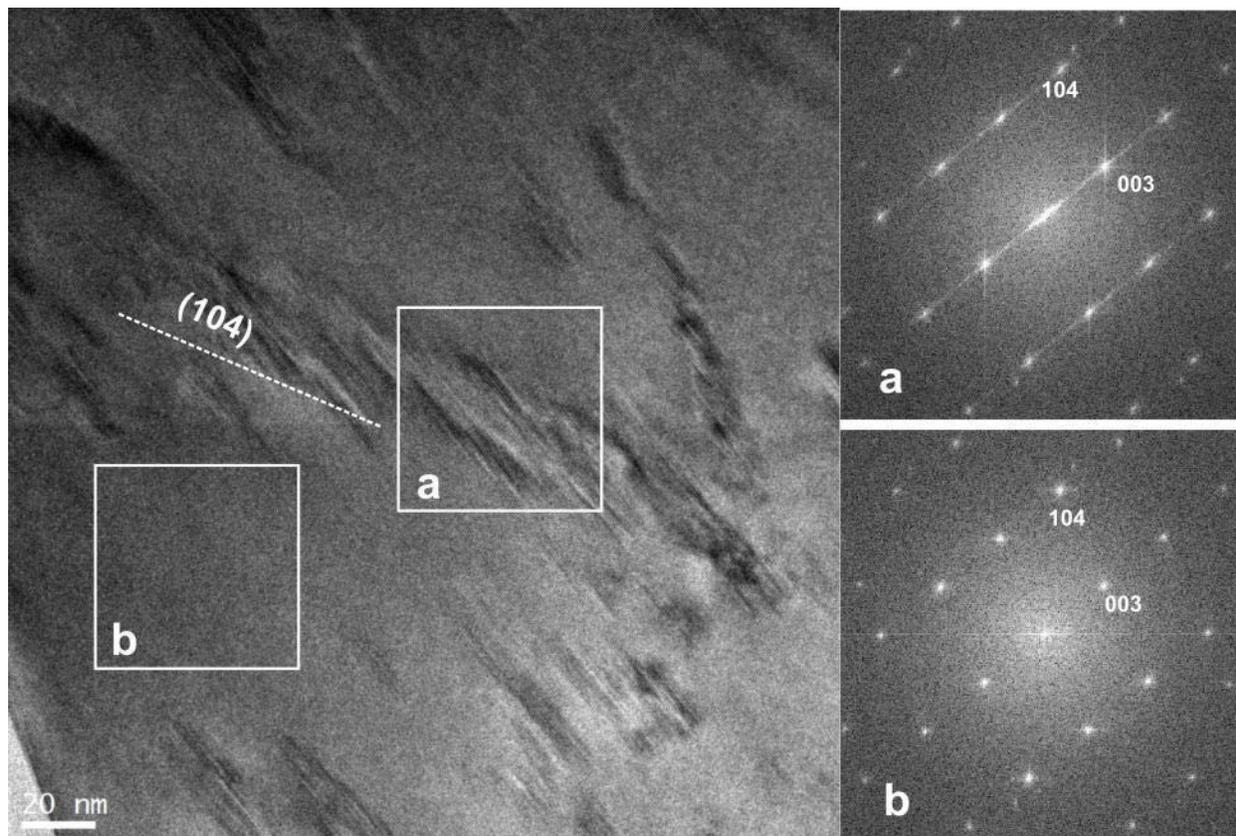


Figure 6. The dark areas of *d* superstructure in the bright-field image (left) under STEM mode become bright in HAADF image (right), which means higher Ca content in the precipitates of *d* superstructure than the host dolomite.

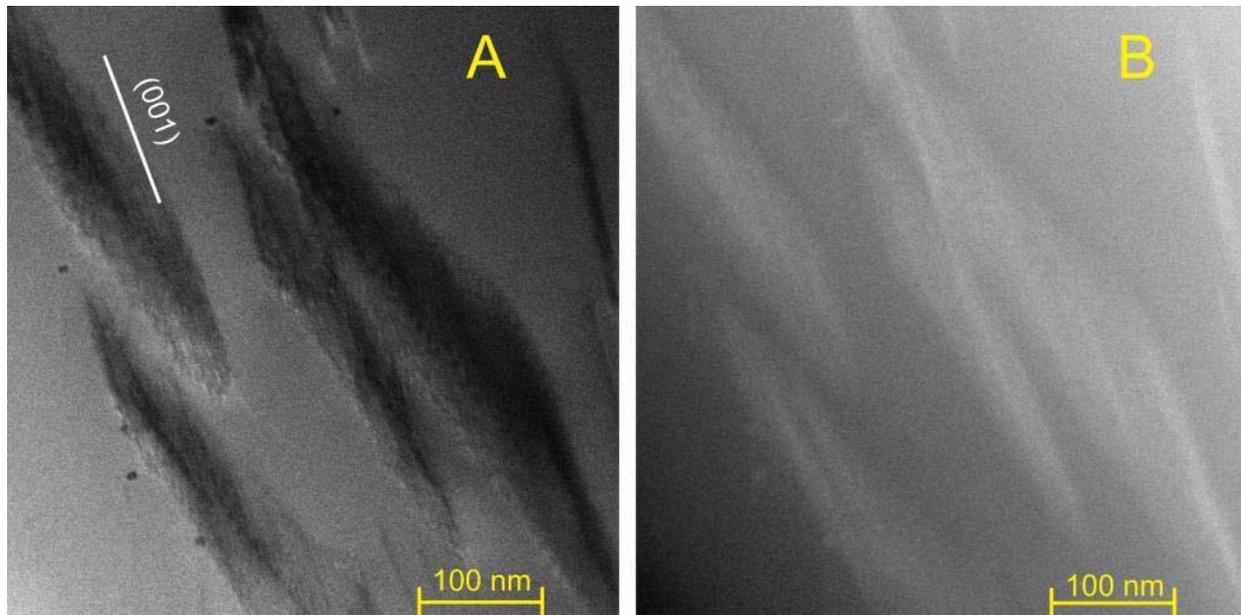


Figure 7. High magnification bright-field (A) and dark-filed (B) STEM images of *d* domains. Z-contrast (dark-filed) image shows that the Ca-rich precipitates have a cation sequence (bottom-right) of Ca-Ca-Mg-Ca-Ca-Mg- along *c* axis as opposed to dolomite cation sequence as shown in the middle-right corner (B). The occurrence of streaking and splitting along *c** (i.e., *d* reflections) in the FFT patterns (bottom-left corner). 003 and 009 spots are from host dolomite, not from the *d* superstructure, because the position of 003 half way between 002 and 004 of the *d* superstructure, and the position of 009 is half way between 008 and 0010 of the *d* superstructure. Noise-filtered Z-contrast image (C) shows enhanced cation sequences in the *d* superstructure and dolomite host. Line profiles from line 1 (dolomite host) and 2 (*d* superstructure) also illustrate the ordering of Ca and Mg atoms.

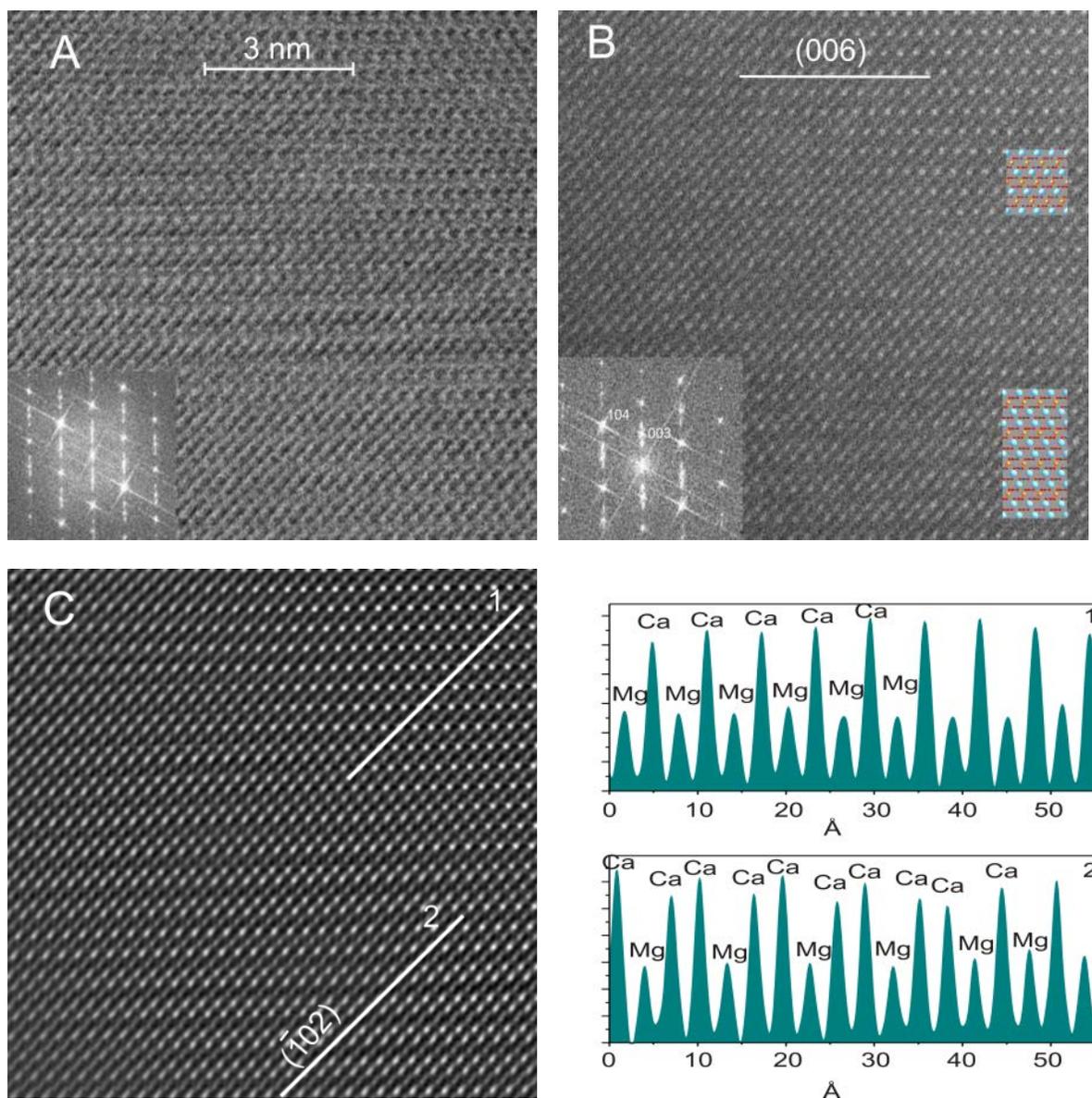


Figure 8. Z-contrast image shows calcite exsolution lamellae // (001) in dolomite host. The line profile 1 has six consecutive Ca columns along $(\bar{1}02)$ trace. The line profile 2 of the dolomite region shows the normal dolomite cation sequence of alternating Ca and Mg columns. The line profile 3 shows one repeat of d superstructure. The lines with arrows show the boundaries between calcite and dolomite and between dolomite and d superstructure. The atomic models for dolomite/calcite and dolomite/ d phase interfaces are shown at the bottom. Carbonate group are not shown proportionally in order to highlight the cations.

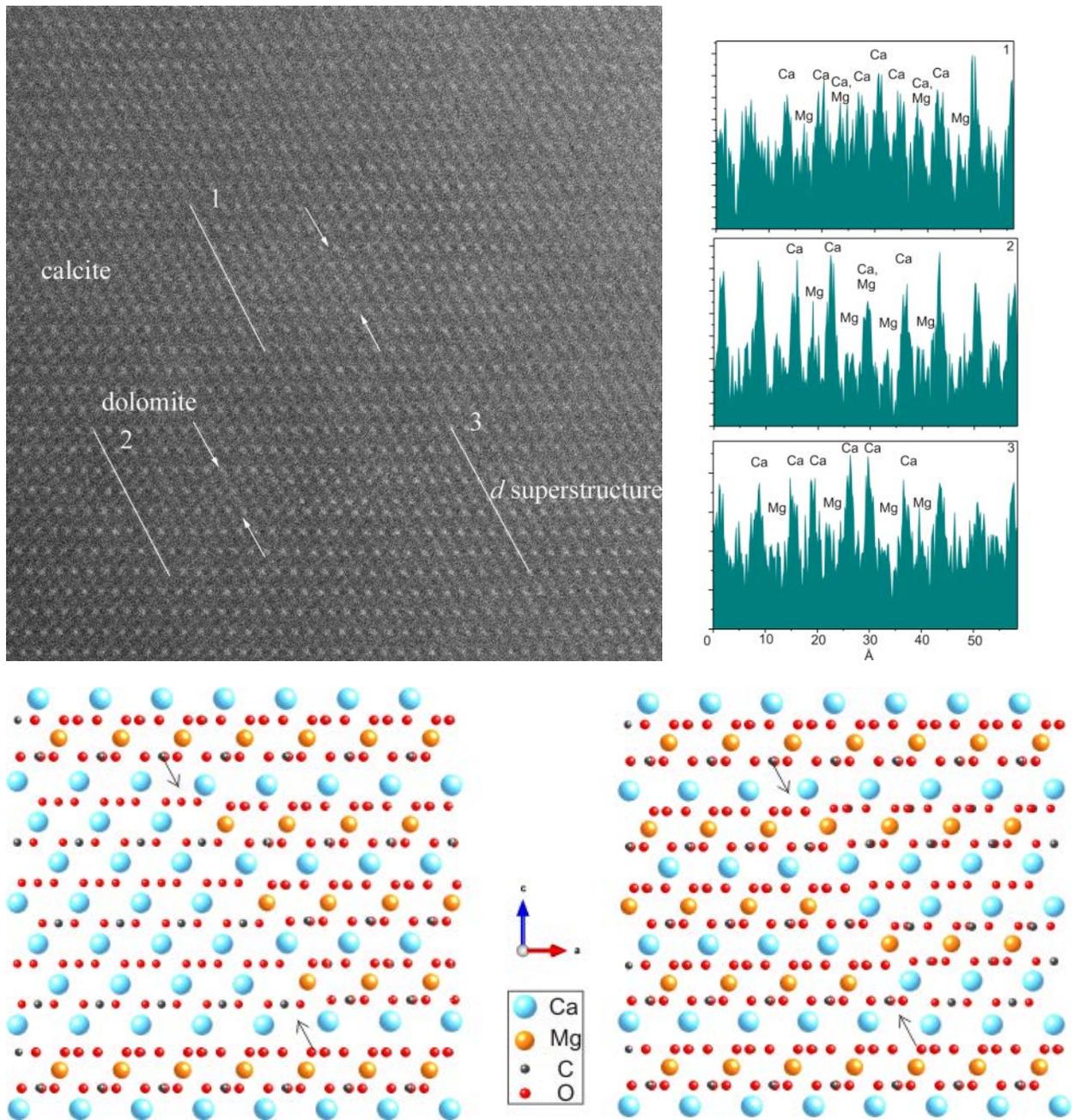


Figure 9. Previously proposed structure model for “ δ dolomite” (A), and calculated model for d superstructure (B), and corresponding calculated electron diffraction pattern (C), compared to the FFT pattern (D) from an area dominated by the d superstructure. Very weak 003 and 009 reflections also occur in FFT pattern due to contributions from the dolomite host. However, the position of 003 is not half way between 002 and 004 of the d superstructure, and the position of 009 is not half way between 008 and 0010 of the d superstructure. The d superstructure does not have ($h0l$) reflections with odd l due to its c -glide.

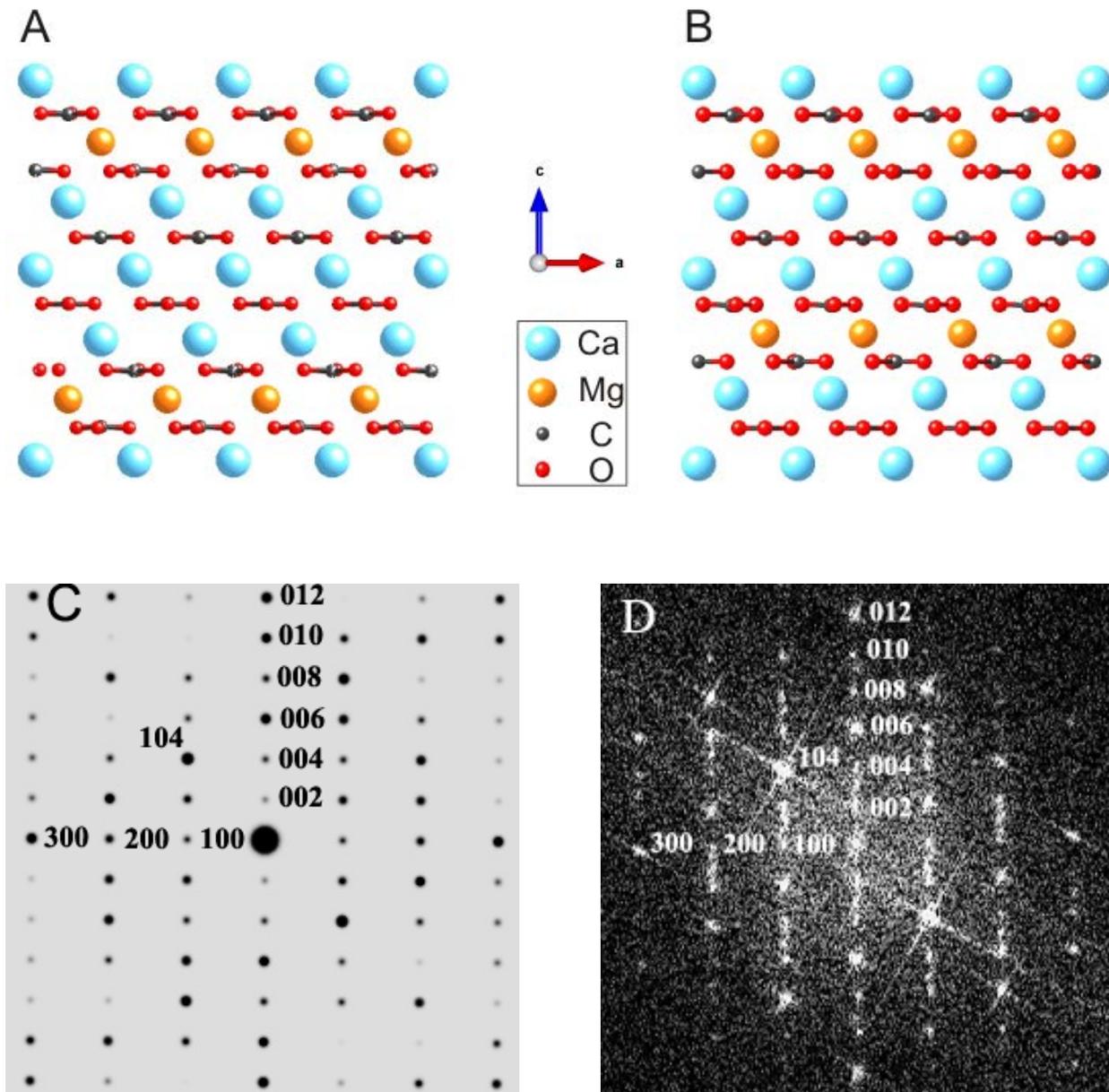


Figure 10: Calculated powder diffraction patterns of the *d* superstructure and stoichiometric dolomite (copper K-alpha radiation). The peak of (105) in dolomite does not appear in the *d* superstructure phase. However, the peak of (106) in the *d* superstructure phase does not appear in dolomite.

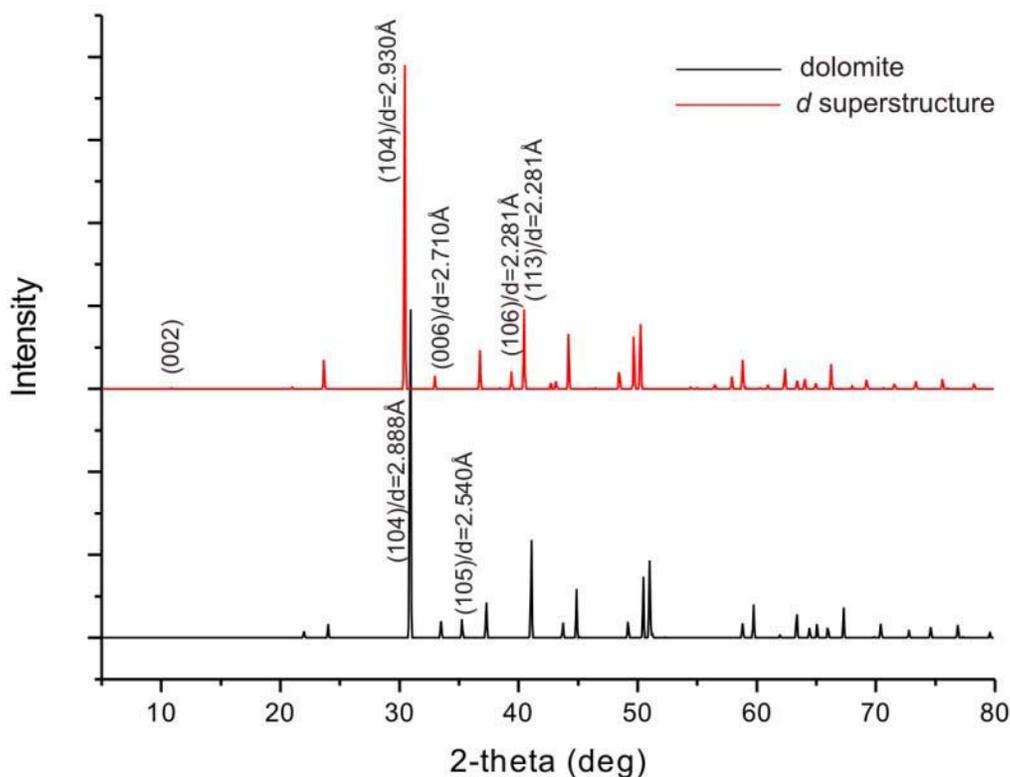


Table 1. Fractional coordinates of atoms, lattice parameters, and bond distances for the *d* superstructure calculated using DFT method.

Atom	x	y	z	Atom	x	y	z
Ca1	0.00000	0.00000	0.00000	Ca2	0.33333	0.66667	0.18457
Mg	0.66667	0.33333	0.34229	C1	0.66667	0.33333	0.09229
C2	0.00000	0.00000	0.26849	C3	0.33333	0.66667	0.41608
O1	0.66667	0.06803	0.09229	O2	0.28361	0.04018	0.26739
O3	0.04972	0.62648	0.41719				
<i>a</i>: 4.879Å <i>c</i>: 16.260Å <i>V</i>: 335.226Å³ Space group: <i>P31c</i> (no. 159)							
C-O bond (Å) : 1.294, 1.297 Ca-O bond (Å) 2.355, 2.374 Mg-O bond (Å) 2.085							

Table 2. Calculated structural parameters of dolomite as compared to experimental and previously calculated data.

Lattice parameters	This work	Experimental	Theoretical
a (Å)	4.810	4.808 ^a , 4.812 ^b	4.787 ^c , 4.877 ^d , 4.858 ^e
c (Å)	15.704	16.010 ^a , 16.020 ^b	15.55 ^c , 16.285 ^d , 16.109 ^e
V (Å ³)	314.611	320.504 ^a , 321.251 ^b	308.623 ^c , 335.409 ^d , 329.248 ^e
C-O bond (Å)	1.294	1.233 ^a , 1.286 ^b	1.286 ^c , 1.299 ^d
Ca-O bond (Å)	2.358	2.405 ^a , 2.382 ^b	2.328 ^c , 2.401 ^d
Mg-O bond (Å)	2.069	2.114 ^a , 2.087 ^b	2.071 ^c , 2.314 ^d
O-Mg-O bond angles (°)	89.03, 90.97, 180	89.17, 90.83, 180 ^a	89.335, 90.645, 180 ^c 88.546, 91.454, 180 ^d

^a Graf ; ^b Beran and Zemann; ^c Hossain et al., LDA functional was used; ^d Hossain et al., GGA functional was used; ^e Bakri and Zaoui, fitted to Birch-Murnaghan equation of state.

Table 3. Calculated structural parameters of calcite as compared to experimental and previously calculated data.

Lattice parameters	This work	Experimental	Theoretical
a (Å)	4.995	4.990 ^a	5.061 ^b , 4.981 ^c
c (Å)	16.685	17.062 ^a	17.097 ^b , 15.902 ^c
V (Å ³)	360.435	367.916 ^a	379.279 ^b , 341.676 ^c
C-O bond (Å)	1.295	1.286 ^a	
Ca-O bond (Å)	2.336	2.357 ^a	

^a Graf; ^b Ayoub, Zaoui, and Berghout; ^c Aydinol et al.