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Revision 2

Chromium solubility in perovskite at high pressure: The structure of $(Mg_{1-x}Cr_x)(Si_{1-x}Cr_x)O_3$ (with x = 0.07) synthesized at 23 GPa and 1600 °C

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ABSTRACT

The crystal structure and chemical composition of a crystal of $(Mg_{1-x}Cr_x)(Si_{1-x}Cr_x)O_3$ 17 perovskite (with x = 0.07) synthesized in the model system Mg₃Cr₂Si₃O₁₂-Mg₄Si₄O₁₂ at 23 18 GPa and 1600 °C have been investigated. The compound was found to be orthorhombic, 19 space group *Pbnm*, with lattice parameters a = 4.8213(5), b = 4.9368(6), c = 6.9132(8) Å, V =20 164.55(3) Å³. The structure was refined to R = 0.046 using 473 independent reflections. 21 Chromium was found to substitute for both Mg at the dodecahedral X site (with a mean bond 22 distance of 2.187 Å) and Si at the octahedral Y site (mean: 1.814 Å), according to the reaction 23 $Mg^{2+} + Si^{4+} = 2Cr^{3+}$. Such substitutions cause a shortening of the $\langle X-O \rangle$ and a lengthening of 24 the $\langle Y-O \rangle$ distances with respect to the values typically observed for pure MgSiO₃ 25 perovskite. Although no high Cr-contents are considered in the pyrolite model, Cr-bearing 26 perovskite may be an important host for chromium in the lower mantle. The successful 27 synthesis of perovskite with high-Cr contents and its structural characterization are of key 28 importance because the study of its thermodynamic constants combined with the data on 29 phase relations in the lower-mantle systems can provide new constraints on thermobarometry 30 of perovskite-bearing assemblages. 31

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33 KEYWORDS: perovskite, chromium, lower mantle, crystal structure, microprobe analysis,
34 synthesis

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INTRODUCTION

Although it is well accepted that (Mg,Fe)SiO₃ perovskite is one of the major phases in the 38 Earth's lower mantle, there is still a debate about the trace-elements incorporation in its 39 structure. As pointed out by Andrault (2003), three mechanisms can be invoked for the 40 incorporation of a trivalent cation in (Mg,Fe)SiO₃ perovskite: (*i*) incorporation of X^{3+} cations 41 in the octahedral Y site with formation of oxygen vacancies; (ii) coupled-substitution of X^{3+} 42 cations at both X and Y sites; and (*iii*) coupled-substitution of X^{3+} cations and Fe³⁺, after 43 oxidation of some Fe²⁺ into Fe³⁺. Obviously, all these mechanisms can vary with pressure, 44 temperature, X^{3+} cations content, and/or oxygen fugacity (fO₂). Among the X^{3+} elements, 45 chromium is considered to be very minor in the primitive mantle (~0.3-0.4 wt% Cr₂O₃; 46 Ringwood 1979; Allègre et al. 1995; O'Neill and Palme 1998) and the largest amount of Cr 47 reported so far in synthetic MgSiO₃ perovskite does not exceed 0.006-0.008 atoms per 48 formula unit (Irifune 1994; Eeckhout et al. 2007). Thus, as Cr is present in the lower mantle, 49 it is important to investigate the crystal chemistry of high-pressure Cr-bearing phases under 50 such conditions. Moreover, there is also considerable discussion about the valence state of 51 chromium in lower mantle (Cr^{2+} vs. Cr^{3+}) related to its role played to influence properties 52 such as electrical conductivity (Dobson and Brodholt 2000), ionic diffusion (Mackwell et al. 53 2005), and lower mantle oxygen fugacity (McCammon et al. 2004). 54

To verify the possible incorporation of large amounts of Cr into perovskite and the response of the crystal structure, we present a structural study by means of single-crystal Xray diffraction of an MgSiO₃-perovskite synthesized at high pressures exhibiting the highest Cr content ever reported.

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Experimental

61 Synthesis

Experiments at P = 23 GPa and T = 1600 °C were performed using a 2000-t Kawai-type 62 multianvil apparatus installed at the Ehime University (Matsuyama, Japan). The samples were 63 compressed by eight cubic tungsten carbide anvils with 2.5-mm truncation edge lengths. 64 Pyrophyllite gaskets were placed between the anvils to make the compression easier. Heating 65 of the sample was performed by a tubular LaCrO₃ heater. The samples were loaded into 66 platinum capsules isolated from the heater by a MgO insulator. Approximate sample volumes 67 after experiments were 1.0 mm³. As the starting materials, mixtures of MgO, SiO₂ and Cr₂O₃ 68 in stoichiometric proportions were prepared for the composition knorringite (Knr) - majorite 69 (Maj) Knr₇₀Maj₃₀ (mol %). Temperature was controlled by a W₉₇Re₃-W₇₅Re₂₅ thermocouple. 70 Pressure was calibrated at room temperature using semiconductor-metal transitions of Bi, ZnS 71

and GaAs (Irifune et al. 2004). The effect of temperature on pressure was further corrected 72 using the α - β and β - γ phase transitions of olivine (Katsura and Ito 1989; Yamada et al. 2004). 73 MgSiO₃ perovskite (Pv) prevailed in run products being accompanied by MgCr₂O₄ with

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calcium titanate structure (Ct) and stishovite (St) (Fig. 1). 75

Data collection and crystal-structure refinement 76

A small crystal $(30 \times 32 \times 40 \text{ }\mu\text{m})$, hand-picked under a reflected light microscope from 77 the run product 2413-70 (Fig. 1), was preliminarily examined with a Bruker-Enraf MACH3 78 single-crystal diffractometer using graphite-monochromatized MoK α radiation. The data 79 collection was then done with an Oxford Diffraction Xcalibur 3 diffractometer (X-ray MoK α 80 radiation, $\lambda = 0.71073$ Å) fitted with a Sapphire 2 CCD detector (see Table 1 for details). 81 Intensity integration and standard Lorentz-polarization corrections were done with the 82 CrysAlis RED (Oxford Diffraction 2006) software package. The program ABSPACK of the 83 CrysAlis RED package (Oxford Diffraction 2006) was used for the absorption correction. The 84 $R_{\rm int}$ (Laue group *mmm*) decreased from 0.081 to 0.048 after the absorption correction. 85

Reflection conditions (h0l: h + l = 2n; 0kl: k = 2n; h00: h = 2n; 0k0: k = 2n; 00l: l = 2n) are 86 consistent with the space group Pbnm (Pnma as standard), reported for orthorhombic 87 perovskites. The full-matrix least-squares program SHELXL-97 (Sheldrick 2008), working on 88 F^2 , was used for the refinement of the structure, which was carried out starting from the 89 atomic coordinates reported by Kojitani et al. (2007) for Al-bearing MgSiO₃ perovskite. Site-90 scattering values were refined using scattering curves for neutral species (Ibers and Hamilton 91 1974) as follows: Mg vs. Cr for the X site, Si vs. Cr for the Y site, and O vs. for the anion 92 sites. The O sites were found fully occupied, and the occupancy factors were then fixed to 93 1.00. The refined mean electron numbers at the X and Y cation sites were found to be 94 12.83(1) and 14.69(1), respectively, thus indicating the same amount of Cr substituting for 95 both Mg (at the X site) and Si (at the Y site). Therefore, in order to reduce the number of free 96 variables, only one parameter was refined to constrain ${}^{X}Cr = {}^{Y}Cr$. Successive cycles were run 97 introducing anisotropic temperature factors for all the atoms leading to R1 = 0.036 for 191 98 observed reflections $[F_0 > 4\sigma(F_0)]$ and $R_1 = 0.046$ for all 473 independent reflections. 99 Fractional atomic coordinates and atomic displacement parameters are shown in Table 2. 100 Table 3¹ lists the observed and calculated structure factors. Bond distances are reported in 101 Table 4 together with those of other synthetic MgSiO₃ perovskites. 102

¹ For a copy of Table 3, document item, contact the Business Office of the Mineralogical Society of America.

103 Chemical composition

A preliminary chemical analysis using energy dispersive spectrometry, performed on the 104 same crystal fragment used for the structural study as well as on other fragments from the 105 same run product, did not indicate the presence of elements (Z > 9) other than Cr, Mg and Si. 106 The chemical composition was then determined using wavelength dispersive analysis (WDS) 107 by means of a Jeol JXA-8600 electron microprobe. We used 40 s as counting time. The 108 matrix correction was performed with the Bence and Albee (1968) program as modified by 109 Albee and Ray (1970). The standards employed were forsterite (Mg, Si) and synthetic Cr₂O₃ 110 (Cr). The crystal used for the X-ray study was found to be homogeneous within the analytical 111 uncertainty. The average chemical composition (five analyses on different spots) is (wt %), 112 SiO_2 53.17(24); Cr_2O_3 10.35(14); MgO 35.66(16); total 99.18(23); corresponding, on the 113 basis of 3 oxygen atoms, to [Mg_{0.93(5)}Cr_{0.07(3)}][Si_{0.93(3)}Cr_{0.07(3)}]O₃. 114

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RESULTS AND DISCUSSION

The unit-cell parameters are strongly influenced by the entry of Cr into the structure. We 117 observed an increase of all the values with the most affected being the *a* parameter. It 118 increases by 0.9% with respect to pure MgSiO₃ whereas the increase of the b and c119 parameters is close to 0.1–0.2%. This feature is quite surprising as Kojitani et al. (2007) 120 observed a contraction, although small, of the *a* parameter with increasing the Al content in 121 perovskite-type MgSiO₃ by means of X-ray powder diffraction data. Al-bearing 122 stoichiometric perovskites should indeed behave in a similar way since Al substitutes for both 123 Mg and Si, that is it replaces a larger cation in the dodecahedral X site and a smaller cation in 124 the octahedral Y site. It should be pointed out, however, that O'Neill and Jeanloz (1994) and 125 Irifune et al. (1996) had described a linear increase of the *a* parameter with increase of the Al 126 content in stoichiometric perovskites according to the linear equations: $a(\text{\AA}) = 0.016(8)X_{\text{Al}} +$ 127 4.779(1) and a (Å) = 0.019(7) X_{Al} + 4.777(1), respectively. In general, it seems that Cr has a 128 larger effect on the perovskite structure than does Al. The Cr-for-Si substitution inducing the 129 enlargement of the Y site seems stronger than the Cr-for-Mg substitution inducing the 130 shrinkage of the X site, thus implying an overall increase of the unit-cell volume. Such a 131 feature can also be pointed out by the analysis of the degree of polyhedral tilting (Φ) , 132 calculated as $\Phi = 1 - \cos^2 \theta_m \cos \theta_z$ (Thomas 1996), as a function of the minor substituent for Si 133 in the perovskite structure. In Figure 2 the ratio of the two polyhedral volumes (V_X/V_Y) is 134 plotted against the degree of polyhedral tilting Φ . For an untilted perovskite ($\Phi = 0$), $V_X/V_Y =$ 135 5. On the basis of the analysis of the geometry of several synthetic perovskites, Thomas 136

(1996) predicted the tendency given as dotted line in Figure 2. Interestingly, pure MgSiO₃ 137 (Dobson and Jacobsen 2004), Al-bearing perovskites (Kojitani et al. 2007) and the Cr-bearing 138 crystal studied here, plot exactly on the trend. Although irreducible representations and 139 symmetry-adapted modes would provide a clear and unambiguous way to separate the effects 140 of distortion and tilting of octahedra in perovskites (Wang and Angel 2011), the data for the 141 Cr-bearing perovskite are markedly different with respect to those calculated for pure MgSiO₃ 142 thus indicating that the polyhedral tilting seems the dominant mechanism to accommodate 143 different sized X and/or Y cations. 144

The assignment of Cr to both the X and Y sites according to the reaction $Mg^{2+} + Si^{4+} =$ 145 $2Cr^{3+}$ is required both to account for the electron density at those sites and to justify the 146 decrease and the increase of the X-O and Y-O distances, respectively, relative to pure 147 $MgSiO_3$ (Table 4). The geometry of the X-polyhedron in the crystal studied here shows 148 specific characteristics never previously reported for the MgSiO₃-perovskite structure. The 149 mean $\langle X-O \rangle$ value of 2.187 Å is by far the shortest among those observed in MgSiO₃ 150 perovskites. In particular, the decrease of the X-O distances is not isotropic; the shortest X-151 O1^{*ii*} distance (i.e., 1.933 Å) decreases the most while there is a slight lengthening of the X– 152 O2^{iii,iv} distances at 2.064 Å. Moreover, the four long X–O2 distances become almost equal, 153 and do not show the usual difference ($\Delta = X - O2^{\nu i, \nu i i} - X - O2^{i, \nu}$) observed between them (e.g., 154 $\Delta = 0.143$ in pure MgSiO₃; Dobson and Jacobsen 2004). Similar, but reversed geometrical 155 changes are caused by the entry of Cr in the Y site. The Cr-for-Si substitution induces a 156 general increase of the mean octahedral distance, which pass from 1.793 Å in pure MgSiO₃ 157 (Dobson and Jacobsen 2004) to 1.814 Å in the present crystal. Such a value exceeds those 158 observed for ^{VI}Si in synthetic Na₂MgSi₅O₁₂ garnet (1.793 Å; Bindi et al. 2011) and that in 159 stishovite (1.757 Å; Hill et al. 1983), whereas it is similar to those found in pyroxenes 160 synthesized in the Na($Mg_{0.5}Si_{0.5}$)Si₂O₆-NaAlSi₂O₆ system (1.807–1.813 Å; Yang et al. 2009). 161 This is not surprising as the octahedral site of these pyroxenes are occupied by minor amounts 162 of Mg (or Al) substituting for silicon. The Cr-for-Si substitution also induces a distortion of 163 the octahedral site quantifiable with an increase of the octahedral angle variance σ^2 (Robinson 164 et al. 1971) from 1.56 in pure MgSiO₃ (Dobson and Jacobsen 2004) to 7.48 in the studied 165 crystal. A similar value ($\sigma^2 = 7.63$) can be calculated for (Mg_{1-x}Al_x)(Si_{1-x}Al_x)O₃ with x =166 0.050 (Kojitani et al. 2007). Taking into account that Cr^{3+} is not expected to be Jahn-Teller 167 active, it appears that the substitution of a trivalent cation for Si in the perovskite structure 168 increases the distortion of the octahedra. 169

Since Mg-rich perovskite is the most abundant phase in the pyrolitic lower mantle and its structure seems rather flexible to various crystal-chemical environments, this mineral may

accommodate Cr and other minor elements of the pyrolitic composition. Cr-bearing 172 perovskites were described in the lower mantle associations in diamonds from São Luiz, 173 Brazil (Harte et al. 1999). Most of them are Mg-rich, with Cr₂O₃ up to 0.5 wt % and Al₂O₃ up 174 to 2.7 wt %. The highest Cr₂O₃ content registered in these perovskites is 1.19 wt % (10.0 wt 175 % Al₂O₃), which is \sim 3 times higher than the estimated bulk Cr concentration in the mantle. In 176 this relation, there is no doubt that the perovskite phase would be capable of storing all of the 177 Cr content expected for the pyrolitic-type composition (i.e., 0.4 wt% of Cr₂O₃). It is important 178 to note that, as for Fe^{3+} , the entry of Cr^{3+} in perovskite is not related to the usual consideration 179 of oxygen fugacity (Lauterbach et al. 2000). On the usual basis of $Al-Fe^{3+}$ (Cr^{3+}) substitution 180 relationships in silicate structures, the high content of Cr^{3+} and Fe^{3+} in some natural 181 perovskites may be linked to the high Al content (McCammon 1997), which increases with 182 pressure. 183

Our results on the synthesis of Al-free perovskite demonstrate that the extremely high 184 Cr₂O₃ content results from incorporation of Cr in both the X and Y sites. The dissolution of 185 Cr^{3+} in MgSiO₃ perovskite may have a considerable effect on physical properties in the lower 186 mantle. Since the reported bulk modulus of Al-bearing MgSiO₃ perovskite (Zhang and 187 Weidner 1999) differs significantly from that of pure MgSiO₃ perovskite, estimation of the 188 lower mantle composition based on the elastic properties of perovskite should account for a 189 number of major-element substitutions (including Cr) in it. The successful synthesis and 190 structural refinement of an MgSiO₃ perovskite containing high amounts of Cr is important for 191 the study of impact of this element on its thermodynamic constants and phase relations in the 192 lower mantle. 193

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284	FIGURE CAPTIONS
285	FIGURE 1. SEM-BSE image of idiomorphic Cr-bearing perovskite (Pv) crystals associated with
286	MgCr ₂ O ₄ with a calcium titanate structure (Ct) and stishovite (St) in the run 2413-70
287	$[P = 23 \text{ GPa}, T = 1600 ^{\circ}\text{C}]$. CamScan electronic microscope MV2300.
288	FIGURE 2. Variation of V_X/V_Y as a function of the degree of tilt Φ (see text for explanation).
289	The dotted line refers to the predicted trend reported by Thomas (1996). Numbers
290	as follows: 1 = pure MgSiO ₃ (Dobson and Jacobsen 2004); 2 = $(Mg_{1-x}Al_x)(Si_{1-x$
291	_x Al _x)O ₃ with $x = 0.025$ (Kojitani et al. 2007); $3 = (Mg_{1-x}Al_x)(Si_{1-x}Al_x)O_3$ with $x =$
292	0.050 (Kojitani et al. 2007); 4 = Cr-bearing perovskite (present study).

TABLE 1. Data and experimental details for the selected crystal

Crystal data	
Formula Crystal size (mm) Form Colour Crystal system Space group a (Å) b (Å) c (Å) V (Å ³) Z	$\begin{array}{l} (Mg_{0.928}Cr_{0.072})(Si_{0.928}Cr_{0.072})O_{3}\\ 0.030\times0.032\times0.040\\ \text{block}\\ \text{transparent}\\ \text{orthorhombic}\\ Pbnm~(\#62-Pnma~\text{as standard})\\ 4.8213(5)\\ 4.9368(6)\\ 6.9132(8)\\ 164.55(3)\\ 4 \end{array}$
Data collection	
Instrument Radiation type Temperature (K) Detector to sample distance (cm) Number of frames Measuring time (s) Maximum covered 2θ (°) Absorption correction Collected reflections Unique reflections Unique reflections Reflections with $F_0 > 4 \sigma$ (F_0) R_{int} R_{σ} Range of <i>h</i> , <i>k</i> , <i>l</i>	Oxford Diffraction Excalibur 3 Mo $K\alpha$ ($\lambda = 0.71073$) 298(3) 5 2122 135 75.80 multi-scan (<i>ABSPACK</i> ; Oxford Diffraction, 2006) 3496 473 191 0.0480 0.1185 -8 $\leq h \leq 8, -8 \leq k \leq 8, -11 \leq l \leq 11$
Refinement	
Refinement Final <i>R</i> 1 [$F_o > 4 \sigma (F_o)$] Final <i>R</i> 1 (all data) W <i>R</i> 1[$F_o > 4 \sigma (F_o)$] W <i>R</i> 1(all data) GoF Number of least squares parameters $\Delta \rho_{max}$ (e Å ⁻³) $\Delta \rho_{min}$ (e Å ⁻³)	Full-matrix least squares on <i>F</i> ² 0.0364 0.0455 0.0905 0.0912 0.968 29 0.63 -0.87

TABLE 2. Atoms, Wyckoff positions, atom coordinates and atomic displacement parameters $(Å^2)$ for the selected crystal.

atom	Wyckoff	x	у	Ζ	U_{11}	<i>U</i> ₂₂	U ₃₃	U_{12}	U_{13}	U ₂₃	$U_{\rm iso}*/U_{\rm eq}$
X [Mg _{0.931(8)} Cr _{0.069}]	4 <i>c</i>	0.9879(3)	0.0726(2)	1⁄4	0.0193(6)	0.0171(5)	0.0192(5)	0.0001(7)	0	0	0.0185(3)
Y [Si _{0.931(8)} Cr _{0.069}]	4a	0	1/2	0	0.0205(5)	0.0201(4)	0.0215(4)	0.0004(6)	0.0001(6)	-0.0002(3)	0.0207(2)
O1 [O]	4 <i>c</i>	0.1025(7)	0.4478(6)	1/4	0.023(1)	0.022(2)	0.028(1)	0.002(1)	0	0	0.0243(7)
O2 [O]	8 <i>d</i>	0.6853(5)	0.3124(4)	0.0500(3)	0.025(1)	0.026(1)	0.0256(9)	-0.0001(8)	0.0003(9)	-0.0003(8)	0.0255(5)

TABLE 4. Geometric details of different perovskites.

	MgSiO ₃ *	$(Mg_{0.975}Al_{0.025})(Si_{0.975}Al_{0.025})O_3$	$(Mg_{0.95}Al_{0.05})(Si_{0.95}Al_{0.05})O_3$	$(Mg_{0.928}Cr_{0.072})(Si_{0.928}Cr_{0.072})O_3$
	(Dobson and Jacobsen 2004)	(Kojitani et al. 2007)	(Kojitani et al. 2007)	(this study)
a (Å)	4.7784(2)	4.7783(1)	4.7767(1)	4.8213(5)
b (Å)	4.9303(1)	4.9327(2)	4.9342(2)	4.9368(6)
c (Å)	6.8990(2)	6.9051(2)	6.9130(2)	6.9132(8)
$V(\text{\AA}^3)$	162.53(1)	162.75(1)	162.93(1)	164.55(3)
Mg-polyhedron				
Mg-O1 ^{<i>ii</i>}	2.017	2.008(4)	2.060(7)	1.933(3)
Mg-O2 ^{<i>iii,iv</i>}	2.055	2.048(3)	1.982(5)	2.064(2)
Mg-O1 ^{<i>ii</i>}	2.098	2.096(4)	2.086(6)	2.069(4)
$Mg-O2^{i,v}$	2.282	2.283(3)	2.253(5)	2.333(3)
Mg-O2 ^{vi,vii}	2.425	2.428(3)	2.496(5)	2.351(2)
Average	2.205	2.203	2.201	2.187
Si-octahedron				
Si-O1 ^{<i>i</i>,<i>xii</i>}	1.800	1.808(1)	1.822(2)	1.816(1)
$\text{Si-O2}^{x,xi}$	1.795	1.803(2)	1.807(4)	1.816(2)
$\text{Si-O2}^{\text{viii,ix}}$	1.783	1.779(2)	1.785(4)	1.811(2)
average	1.793	1.797	1.805	1.814
$O1^{xii}$ -Si- $O2^{xi}$	91.5	91.2(2)	93.3(2)	94.2(1)
$O1^{xii}$ -Si- $O2^i$	91.3	88.2(1)	86.9(3)	85.8(1)
$O2^{viii}$ -Si- $O2^{ix}$	90.6	89.5(1)	89.2(1)	89.19(4)
Si ⁱ -O1-Si ^{xiii}	146.8	145.5(2)	143.2(3)	144.3(2)
Si ⁱ -O2-Si ^{vii}	147.2	146.8(2)	145.8(2)	144.1(1)

Note: * Bond distances were calculated using lattice parameters given by Kojitani et al. (2007) for MgSiO₃ and with the atomic coordinates reported by Dobson and Jacobsen (2004). Symmetry codes are: *i*: *x*, *y*, *z*; *ii*: *x*+1, *y*, *z*; *iii*: *-x*+3/2, *y*-1/2, *z*; *iv*: *-x*+3/2, *y*-1/2, *z*; *v*: *x*, *y*, *-z*+1/2; *v*: *x*, *y*, *-z*+1/2; *v*: *x*, *y*, *-z*+1/2; *v*: *x*, *y*, *-z*+1/2; *v*: *x*+1/2, *-y*+1/2, *z*; *vii*: *x*+1/2, *-y*+1/2, *-z*; *viii*: *-x*+1, *-y*+1, *-z*; *ix*: *x*-1, *y*, *z*; *x*: *x*+1/2, *y*+1/2, *z*; *xii*: *x*-1/2, *-y*+1/2, *-z*; *xiii*: *-x*, *-y*+1, *-z*; *xiii*: *-x*, *-y*+1, *z*+1/2.

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