1	Revision 1: The determination of hydrogen positions in superhydrous phase B
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11	Abstract
12	In order to investigate the nature of hydrogen incorporation into the structure of
13	superhydrous phase B, neutron powder diffraction experiments have been performed. A
14	structural model based on Pnn2 symmetry has been used for the analysis of the data,
15	which is consistent with earlier spectroscopic studies. Application of Fourier synthesis
16	with subsequent analyses of difference nuclear density maps and Rietveld fits reveal two
17	distinct positions for deuterium, at 4c (0.194, 0.052, 0.596) and at 4c (0.186, 0.119,
18	0.388). This unambiguously <i>shows</i> that deuterium lies within large channels, which are
19	formed between the edge-shared octahedra and vertex-linked tetrahedra along the b-axis
20	of the structure. These results contrast with recent polarized single crystal infrared
21	spectroscopy studies where the position of one of two hydrogens was estimated to lie
22	close to the octahedral edge of an MgO ₆ octahedron, thereby leaving the large structural
23	channel empty.

24 Keywords: hydrous magnesium silicates, neutron diffraction, superhydrous phase B,

25 crystal structure, hydrous minerals, determination of hydrogen positions.

26 Introduction

27 Water exists in the Earth's mantle in a number of different forms. It may exist as a 28 free fluid phase; it may be dissolved in silicate melts or incorporated into nominally 29 anhydrous minerals, such as the polymorphs of Mg₂SiO₄ (forsterite, wadsleyite and 30 ringwoodite). It may also be incorporated into nominally hydrous high-pressure silicate 31 phases, such as dense hydrous magnesium silicate (DHMS) phases, which have to date 32 mainly been characterized in the in MgO–SiO₂–H₂O ternary system. The DHMS phases, 33 referred to as A, E, B and superhydrous phase B (SHyB), might be stable at the pressure 34 and temperature conditions compatible with cold portions of subducting slabs and they 35 may form a network of phases with overlapping stability fields that transport water into 36 the lower mantle. Although, the existence of DHMS in nature is still under debate, SHyB 37 *might* be stable within ultramafic compositions coexisting with ringwoodite at conditions 38 compatible with subducting slabs in the lower part of the transition zone (Angel et al. 39 2001). The formation of hydrous minerals such as SHyB may influence the mechanical 40 and rheological properties of slabs, and the ultimate dehydration might potentially be 41 related to the processes causing deep-focus earthquakes. Any attempt to understand the 42 stability or transport properties of such hydrous phases, however, *would benefit* from an 43 understanding of the crystal structure, including an appreciation of the hydrogen sites.

44 SHyB has an ideal chemical composition of $Mg_{10}Si_3O_{14}(OH)_4$, with a Mg/Si ratio 45 of 3.2 and 5.8 wt% of stoichiometrically incorporated water and lies along the brucite– 46 anhydrous phase B join of the MgO–SiO₂–Mg(OH)₂ ternary system. SHyB was first identified and characterized by Gasparik (1993). The current picture of the hydroxyl substitution mechanisms in $Mg_{10}Si_3O_{14}(OH)_4$ is based on results of X–ray diffraction studies (Pacalo and Parise 1992; Kudoh et al. 1994; Koch-Müller et al. 2005), nuclear magnetic resonance (NMR) studies (Xue et al. 2008), and infrared (Cynn et al. 1996; Koch-Müller et al. 2005) and Raman spectroscopy studies (Cynn et al. 1996; Frost and Fei 1998; Liu et al. 2002; Koch-Müller et al. 2005).

53 The first model of the crystal structure of SHyB was established from X-ray 54 diffraction studies by Pacalo and Parise (1992) who determined its orthorhombic Pnnm 55 symmetry with Z=2. However, in this study the positions of the hydrogen atoms were 56 determined only approximately from difference Fourier maps (Pacalo and Parise 1992). 57 Hydrogen was placed into a general position 8h at (-0.0769, 0.3026, 0.1085) with 58 multiplicity 8, which corresponds to a site within the channels along the *a*-axis (*b*-axis in 59 our crystallographic setting) (Pacalo and Parise 1992). Kudoh et al. (1994) and Koch-Müller et al. (2005) performed single crystal X-ray diffraction studies on 60 61 $Mg_{10}Si_3O_{14}(OH)_4$ and proposed two non-centrosymmetric space groups, $P2_1nm$ and $Pnn2_2$, 62 respectively. Although, the hydrogen sites were not determined in the former paper, the 63 $P2_1nm$ symmetry implies the existence of two distinct crystallographic sites for the 64 hydrogen atoms. Koch-Müller et al. (2005) proposed approximate coordinates for two 65 distinct positions for hydrogen atoms, at 4c with (0.3, 0.6, 0.1) and at 4c (0.27, 0.18, 66 (0.45). The first position was inferred from electron density Fourier difference maps 67 obtained by means of X-ray diffraction studies, *consistent with the polarized IR spectra*, 68 whereas, the second position came from polarized single-crystal infrared spectroscopic 69 studies *only*. The infrared and Raman spectroscopic studies (Cynn et al. 1996; Frost and

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70	Fei 1998; Liu et al. 2002; Koch-Müller et al. 2005) do not corroborate the
71	centrosymmetric <i>Pnnm</i> space group for SHyB suggested by Pacalo and Parise (1992).
72	These spectra show two intense peaks in the infrared O-H stretching region that is clear
73	evidence for the existence of two distinct crystallographic positions for hydrogen atoms
74	(Cynn et al. 1996; Frost and Fei 1998; Liu et al. 2002; Koch-Müller et al. 2005).
75	Furthermore, comprehensive 2D ¹ H and ²⁹ Si NMR investigations have also shown the
76	existence of dissimilar hydrogen (H1-H2) pairs in SHyB and found only one distinct
77	tetrahedral Si site (Xue et al. 2008). These findings are consistent with the Pnn2 space
78	group (Xue et al. 2008), which was proposed by Koch-Müller et al (2005). Moreover,
79	Xue et al. (2008) have also excluded Pnnm of Pacalo and Parise (1992) and $P2_1mn$ of
80	Kudoh et al. (1994). Xue et al. (2008) questioned the H positions and OH dipole
81	directions found <i>as proposed by</i> Koch-Müller et al. (2005). In fact, while the H1 position
82	at $(0.3, 0.6, 0.1)$ is located inside a channel along the <i>b</i> -axis, the H2 position at $(0.27, 0.18)$
83	0.45) is close to the octahedral edge O3–O9 of the $Mg3O_6$ octahedron (Xue et al. 2008).
84	The hydrogen bond of H2 belongs in this case to the same Mg3O ₆ octahedron and hence
85	the large structural channel remains unfilled. Furthermore, Koch-Müller et al. (2005)
86	presented sufficient evidences to claim the existence of a high-temperature polymorph
87	of Ni-doped SHyB. On the contrary, the existence of such polymorph was questioned
88	by Xue et al. (2008) based on analysis of undoped sample. Such discrepancies may
89	suggest the existence of a phase transition between the "high-temperature polymorph
90	of SHyB" and the "low-temperature polymorph of SHyB", which is induced by
91	changes in the Ni-content (Ni-SHyB and SHyB are substantially different compounds
92	and this might lead to different crystallographic symmetries).

Thus, the available information from the literature concerning the predominant locations of hydrogen atoms in the structure of the SHyB is inconsistent. Therefore, a detailed investigation that could shed light on fine details of water incorporation in the

96 structure of the SHyB is required for construction of realistic mineral physical models. 97 Neutron diffraction is the most rigorous method for determination of deuterium site 98 occupancies in crystalline phases because neutron scattering cross sections are not a 99 function of atomic mass as for X-ray diffraction and are relatively large for deuterium. In 100 this study the deuterium sites in the SHyB structure have been precisely determined by 101 means of neutron powder diffraction.

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Experimental details

103 A sample of the SHyB of about 20 mg was synthesised from Mg₂SiO₄ (forsterite) 104 + D₂O in a 5000 tonne large-volume press (Frost et al. 2004). An 18 mm edge length 105 Cr_2O_3 doped octahedral pressure assembly was used with tungsten carbide cubes that had 106 8 mm edge length corner truncations. $Mg_2SiO_4 + D_2O$ were placed inside a welded 107 platinum capsule of 3 mm diameter and approximately of 4 mm length. The capsule was 108 inserted into an MgO sleeve and placed into a cylindrical LaCrO₃ heater that was inserted 109 into the octahedral assembly. The temperature during the experiment was monitored 110 using a W3Re/W25Re thermocouple inserted within an alumina tube, with the junction in 111 contact with the top of the Pt capsule. Other details were similar to those reported in 112 previous studies (Frost et al. 2004). The synthesis was performed at 22 GPa, and a 113 temperature of 1100 C^o, with an annealed duration of 4 hours. The recovered sample was 114 milled into a powder and characterized by Raman spectroscopy and X-ray powder 115 diffraction.

116 Neutron powder diffraction measurements were performed at the D20 high-flux 117 instrument installed at the neutron reactor at the Institut Laue-Langevin in Grenoble, 118 France (Hansen et al. 2008). A vanadium sample holder of 4 mm in diameter was filled 119 with a sample of the SHyB and was loaded inside a cryostat sample chamber. The 120 measurements were carried out at ambient temperature and at 10 K. Reducing the 121 temperature to 10 K permitted the deuterium positions to be determined more precisely 122 since the atomic displacement parameters (a.d.p.) of atoms are strongly reduced at low temperatures. The wavelength of 2.415 Å was determined using a Na₂Ca₃Al₂F₁₄ 123 124 reference material; a second wavelength of 2.419 Å was also recognized and included in 125 the Rietveld fits. Despite the presence of the second wavelength, this experimental setup 126 gave the maximum of intensity at the sample position and allowed us to perform high-127 quality neutron diffraction experiments on a relatively small sample of approximately 20 128 mg. After switching on the oscillation of the sample, a microstrip position sensitive 129 detector was used for data collection in the range of 15–150 degrees of 2θ . The collected 130 diffraction patterns were analyzed by *full-profile Rietveld refinements and difference* 131 Fourier analyses using the WinPLOTR package (Rodriguez-Carvajal 1993). The resolution function of the diffractometer and the peak profile shape were modeled using a 132 133 pseudo-Voigt function, determined by fitting a $Na_2Ca_3Al_2F_{14}$ reference material. The 134 background of the diffraction pattern was fitted using a linear interpolation between 135 selected data points in non-overlapping regions. The scale factor, the lattice parameters, 136 zero shift and the Y profile shape parameter were varied during the fitting procedure. As 137 the crystal structure of SHyB is complex with 18 distinct sites for Mg, Si and O, the 138 atomic coordinates of these atoms were fixed at those values determined by single-crystal

synchrotron diffraction measurements (Koch-Müller et al. 2005), whereas, the fractional
coordinates and the occupancies of the deuterium atoms and a.d.p.'s parameters of all
atoms were varied.

142 **Results**

143 The initial crystal model and the crystallographic setting used for the refinement of 144 the SHyB structure were those proposed by Koch-Müller et al. (2005), i.e., with Pnn2 145 symmetry. Hydrogen positions proposed in this paper were not introduced into the 146 refinements. The Pnn2 symmetry is also consistent with NMR results of Xue et al. (2008). 147 In the sample secondary minor phases of stishovite and ringwoodite could be detected 148 which were included in the full-profile refinements. The fit residuals $R_p = 3.37\%$, R_{wp} =4.62% and χ^2 =8.97 (*Pnn2* space group) were achieved when deuterium positions 149 150 were not introduced into the structural model. Thereafter, we determined the initial 151 positions of the deuterium atoms from analyses of the Fourier nuclear density difference 152 maps (Fig. 1). These maps were calculated from the difference between observed neutron 153 structure factors and structure factors that were inferred from the 'deuterium-free' 154 structural model based on structural parameters found in synchrotron single-crystal 155 diffraction studies (Koch-Müller et al. 2005). Figure 1 demonstrates the difference 156 nuclear density at a level in the structure of $x \approx 0.19$ with two pronounced maxima around 157 $y\approx 0.05$, $z\approx 0.62$ and $y\approx 0.05$, $z\approx 0.4$. Therefore, it was possible to establish the initial 158 positions of the deuterium atoms, D1 and D2 as (0.19, 0.05, 0.62) and (0.19, 0.05, 0.4), 159 respectively. Rietveld refinement of the pattern by then varying the coordinates of the D1 160 and D2 atoms, a.d.p.'s and occupancies resulted in excellent convergence with residuals $R_p=2.27\%$, $R_{wp}=3.13\%$ and $\chi^2=4.15$, i.e., the residuals are considerably improved with 161

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respect to the "deuterium-free" model. The detailed crystallographic data obtained from the Rietveld refinements of the structural models at 300 and 10 K are given in the supplementary information in the form of standard crystallographic information files (ciffiles). We show the observed and calculated neutron diffraction profiles for SHyB in Figure 2.

167 Variation the occupation factors of the deuterium sites slightly improved the quality 168 of the Rietveld refinement. The refined deuterium content was lower than expected from 169 the SHyB stoichiometry, Mg₁₀Si₃O₁₈D₄. This suggests that some hydrogen contamination 170 of the deuterated sample may have occurred during the sample synthesis. This fact is also 171 supported by Raman spectroscopy which shows pronounced peaks in the O-H stretching region at frequencies near 3400 cm⁻¹. However, accounting for the presence of hydrogen 172 173 in the structural model does not result in any further improvement to the residual and this 174 was therefore neglected.

175 **Discussion**

176 Figure 3 shows the crystal structure of the SHyB together with the hydrogen positions 177 determined from the neutron diffraction experiments. The structure is closely related to 178 anhydrous phase B and phase B (Finger et al. 1991, Pacalo and Parise 1992). Its can be 179 considered as an ordered alternation of two structural motives along the a-axis: the first 180 motive is composed of double octahedral-tetrahedral OT layers consisting of the edge-181 shared MgO_6 octahedra and vertex-linked SiO₄ tetrahedra, whereas the second motive is 182 built up of octahedral O layers containing edge-shared MgO_6 and SiO_6 octahedra. These 183 layers interchange as ...-O-OT-OT-O-OT-OT-O-... sequence. Large channels occur 184 in the structure along the *b*-axis inside the OT layers, formed between the edge-shared

octahedra and the vertex–linked tetrahedra. These channels are responsible for the high
water capacity of the structure and would very likely lead to fast water diffusion through
the SHyB structure.

188 In accordance with our structural model, the deuterium atoms are accommodated in 189 the SHyB structure inside these channels in the OT layers (Fig. 3). D atoms form 190 hydroxyl groups together with oxygens that coordinate only to magnesium. The bond 191 length of the O4–D1 hydroxyl group is 0.89(4) Å, whereas, a slightly longer bond of 192 1.08(4) Å was found for the O3–D2 hydroxyl group. The O4–D1…O1 and O3–D2…O1 193 hydrogen bond lengths are equal to 2.89(4) Å and 2.83(4) Å, respectively, i.e., both 194 bonds form weak hydrogen bonds and should yield typical stretching vibration frequencies in the range $3200-3750 \text{ cm}^{-1}$. This is in good agreement with numerous 195 196 spectroscopic studies (Cynn et al. 1996; Frost and Fei 1998; Liu et al. 2002; Koch-Müller 197 et al. 2005). The internal angles of the O4–D1···O1 and O3–D2···O1 bonds were found to 198 be of 166 and 162 degrees. The O4–D1 and O3–D2 hydroxyl groups are bonded to the 199 same acceptor oxygen atom O1, and hence they form together a O4–D1···O1···D2–O3 200 configuration. Such a configuration is similar to that reported for the phase B (Finger 201 et al. 1991) as well as with results of density functional calculations for SHyB (Poswal 202 et al. 2010). The hydrogen repulsion could also influence the geometry of the hydrogen 203 bonding in solids, and hence, a precise determination of the D1–D2 distance of 1.90(5) Å 204 is also a crucial point for characterization of the geometry of the hydrogen bonding 205 configuration.

Koch-Müller at al. (2005) proposed the H1 position (0.3, 0.6, 0.1), which is also inside the channel along the *b*-axis in the OT layer, and practically coincides with the D1

208	position determined in this study. The H2 position (0.27, 0.18, 0.45) spectroscopically
209	assigned by Koch-Müller at al. (2005), shown as a large black sphere in Figure 3, lies
210	close to the octahedral O3–O9 edge of the $Mg3O_6$ octahedron and forms an O3–H2–O9
211	angle of 138 degrees. The O3-H2 and O9-H2 distances were found to be 1.25 and 1.78
212	Å, respectively, whereas the O3–O9 distance (the length of the hypothetical O3–H2···O9
213	hydrogen bond), was 2.84 Å. The H1-H2 distance is almost equal to 1.8 Å, and hence,
214	the geometrical configuration parameters of the hypothetical O3-H2···O9 hydrogen bond
215	are very close to those determined in this study. This last point likely explains why the
216	H2 position was spectroscopically assigned at (0.27, 0.18, 0.45) (Koch-Müller et al.
217	2005). Xue et al. (2008) reported that Koch-Müller et al. (2005) have estimated the H2
218	position from the polarization direction of the band near 3347 cm ⁻¹ in single-crystal
219	infrared spectra by using a number of assumptions (H2 position is shown as large black
220	sphere in Fig. 3). According to Xue et al. (2008), it is very unlikely that the H2 site is
221	close to the O3-O9 octahedral edge with H2 bonded to O9 acceptor inside the same
222	octahedron Mg3O ₆ . From a crystal structure point of view, the cationic H2–Mg3 distance
223	of 1.58 Å, which follows from the model of Koch-Müller et al. (2005), seems to be
224	slightly shorter than typical H-Mg distances. It seems also quiet strange if hydrogen
225	atoms would in fact reside at the H2 positions, close to the O3-O9 edge, and would not
226	go into the large empty cavities inside the OT layers, whereas the centers of these empty
227	channels are only about 1.3 Å away from the H2 position. Note, that deuterium atoms
228	cannot reside simultaneously at both D2 and H2 positions as this would lead to too short
229	and nonphysical cationic D2-H2 distance of 1.31 Å. Thus, if we fix our D2 position at
230	the coordinates of the H2 position (0.27, 0.18, 0.45) from reference of Koch-Müller et al.

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231 2005, we cannot achieve a satisfactory Rietveld refinement of the pattern. On the other 232 hand, varying the coordinates of the D2 position we can achieve good Rietveld 233 refinement where the resulting coordinates are (0.19, 0.05, 0.4), i.e., identical to those 234 derived from the difference nuclear density Fourier maps. We should mention that we 235 have also performed a difference Fourier analysis with subsequent Rietveld refinements 236 based on both *Pnnm* (Pacalo and Parise 1992) and *P2*₁*nm* (Kudoh et al. 1994) symmetries. 237 Reasonable residuals were achieved only for deuterium located inside the above cavities 238 in the OT layers. However, our preferred structural model is based on the *Pnn2* symmetry 239 in agreement with spectroscopic investigations (Koch-Müller et al. 2005; Xue et al. 2008). 240 Therefore, our results, supported also by the recent NMR study by Xue et al. (Xue et al. 241 2008), indicate two distinct positions of hydrogen accommodated in channels along the 242 *b*-axis within the OT layers of SHyB. 243 Further, we have to discuss a possibility of incorrect assignment of the H2 position 244 that was proposed by Koch-Müller et al. (2005). This could happen because of

246 their OH2 band with E parallel to the a-axis and the most intense band with E parallel

confusing of the a- and c-axes. According to the last paper, the lowest intensity for

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247 to c-axis, were observed in the (010) plane. This served as a basis for the assignment of

248 the H2 position. In fact, hypothetical O3–H2 hydroxyl group is oriented nearly parallel

250 parallel to the c-axis and to the minimum absorbance intensity with E perpendicular to

to the c-axis (figure 3) which corresponds to the maximum absorbance intensity with E

251 the c-axis (see figure 10 b in Koch-Müller et al. (2005)). Now, lets us to assume that

252 Koch-Müller et al. (2005) have changed the directions of the a- and c-axes in the (010)

253 oriented plate, i.e. the lowest intensity for their OH2 band with E parallel to the c-axis

254 and the strongest intensity for their OH2 with E parallel to the a-axis have to be 255 observed in the (010) plane. This means that such hydroxyl group must be oriented 256 nearly parallel to the a-axis, as in the case of our O3-D2 group (figure 3). This 257 corresponds to the maximum absorbance intensity with E parallel to the a-axis and to 258 the minimum absorbance intensity with E perpendicular to the a-axis (see figure 10 b 259 of Koch-Müller et al. (2005)). Hence, if mixing up of a- and c-axes would be true, our 260 assignment of D2 position would agree perfectly with the polarization behavior from 261 reference by Koch-Müller et al. (2005).

262 In summary, we have performed the first direct determination of deuterium locations 263 in the SHyB structure by means of neutron diffraction. This information is important for 264 understanding the mechanisms of water incorporation and hydrogen transportation 265 through mantle silicates and, therefore, for the construction of models of geological 266 processes in the deep mantle. Knowledge of the hydrogen sites is required, for instance, 267 for local structure studies, aimed at the identification of possible hydrogen pathways 268 responsible for macroscopic water transportation though the SHyB structure. The 269 findings of our work, in particular, the orientation, strength and H–H distance of the O-H 270 bond will assist in the understanding of the infrared hydroxyl stretching bands, which are 271 commonly used to quantify the water content in mantle minerals.

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- 318 D, and superhydrous B: New structural constraints from one- and two-319 dimensional ²⁹Si and ¹H NMR. American Mineralogist, 93, 1099-1111.
- 320 List of figure captions

- 321 Figure 1. Difference nuclear density at a level of $x \approx 0.19$ in Y-Z cross-section at 300 K as
- 322 an example, two pronounced maxima around $y \approx 0.05$, $z \approx 0.6$ and $y \approx 0.05$, $z \approx 0.4$ are clear;
- 323 the difference nuclear density is measured in $fm/Å^3$ units.
- 324 Figure 2. Results of the Rietveld refinements of SHyB diffraction patterns at 300 and 10
- 325 K. A small peak at ~ $2\theta=70^{\circ}$ was also present in a reference sample pattern collected
- 326 under the same conditions and apparently arises from the sample environment. The broad
- 327 feature in background at ~ 50–75 degrees of 2θ may be a signature from amorphous
- 328 silicate melt in the sample.
- 329 Figure 3. A polyhedral representation of the SHyB crystal structure along the *b*-axis and
- 330 stacking within OT layers with hydrogens along *a*-axis.

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