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REVISION 1

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New morphological, chemical, and structural data of woolly erionite-Na from

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Durkee, Oregon, USA

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5 GEORGIA CAMETTI¹, ALESSANDRO PACELLA¹, FRANCESCO MURA², MARCO ROSSI^{2,3},

6

AND PAOLO BALLIRANO^{1,*}

7

8 ¹Dipartimento di Scienze della Terra, Sapienza Università di Roma, Piazzale Aldo Moro 5, 00185
9 Roma, Italy.

10 ²Centro di ricerca per le Nanotecnologie applicate all'Ingegneria della Sapienza (CNIS), Sapienza
11 Università di Roma, Piazzale Aldo Moro 5, 00185 Roma, Italy.

12 ³Dipartimento di Scienze di Base Applicate all'Ingegneria (SBAI), Sapienza Università di Roma,
13 Piazzale Aldo Moro 5, 00185 Roma, Italy.

15 *E-mail address:* paolo.ballirano@uniroma1.it

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ABSTRACT

18 A detailed morphological, crystal-chemical, and structural characterization of erionite from the
19 type locality of Durkee, Oregon, has been carried out by combining field emission scanning
20 electron microscopy (FESEM) and laboratory parallel-beam transmission X-ray powder diffraction
21 (XRPD). According to the crystal-chemical formula $(\text{Na}_{5.38}\text{K}_{1.99}\text{Mg}_{0.24})[\text{Al}_{7.66}\text{Si}_{28.34}\text{O}_{72.09}] \cdot$
22 $29.83\text{H}_2\text{O}$, the sample has been classified as erionite-Na. The Rietveld refinement has indicated that
23 the extraframework cations are located at three Ca1, Ca2, and Ca3 sites, the first one containing all
24 available Mg. Moreover, it has been proved the absence of the additional K2 site found in both

25 dehydrated erionite and erionite-K. Notably, despite many works have reported the presence of a
26 variable Fe and Ca content in erionite samples from Durkee, our results revealed the absence of
27 both these elements. This is relevant information because it is well known from amphibole asbestos
28 that Fe²⁺ has been claimed to be one of the causes of carcinogenesis by participating in Fenton
29 chemistry and producing free radicals.

30

31 **Keywords:** Erionite-Na, Durkee Oregon, woolly morphology, crystal-chemistry, crystal structure,
32 Rietveld method, Scanning Electron Microscopy, laboratory parallel-beam transmission X-ray
33 powder diffraction.

34 **Introduction**

35 Erionite is a fibrous zeolite belonging to the so-called ABC-6 family (Gottardi and Galli 1985). It
36 usually occurs in volcanic ash altered by weathering processes. From the structural point of view, it
37 is hexagonal, space group $P6_3/mmc$. The erionite framework, topological code [ERI] (Meier and
38 Olson 1992), consists of (Si,Al)O₄ tetrahedra linked together to form single-six rings (S6R) and
39 double six-rings (D6R). S6R originate erionite cages by bridging adjacent columns at the level of
40 the cancrinite (ϵ) cages. D6R connect columns of ϵ cages along the z direction. An average chemical
41 formula $K_2(Na,Ca_{0.5})_8[Al_{10}Si_{26}O_{72}] \cdot 30H_2O$ has been proposed for erionite by Coombs et al.
42 (1997). However, due to a significant chemical variability, three different species, erionite-K,
43 erionite-Ca, and erionite-Na have been identified, depending on the most abundant extra-framework
44 (EF) cation.

45 In the past few years fibrous erionite has been thoroughly investigated because of its linking with
46 malignant mesothelioma (MM). In fact, *in vivo* studies unambiguously proved that erionite is
47 several times more tumorigenic than chrysotile and crocidolite asbestos. As a consequence of this,
48 the International Agency for Research on Cancer (IARC) listed erionite as a Group 1 Human-
49 Carcinogen (IARC 1987, 2011). Erionite toxicity has been partly ascribed to the presence of ion-
50 exchanged and/or surface deposited iron participating to Fenton chemistry and generating reactive
51 oxygen species that induce DNA damage (Eborn and Aust 1995). In fact, despite erionite is a
52 nominally Fe-free phase, Fe₂O₃ contents up to 3 wt% have been reported (Eberly 1964; Dogan et al.
53 2006). Besides, Dogan et al. (2006) reported that erionite from Rome (Oregon, USA) has a Fe₂O₃
54 content even higher than the Cappadocia erionite, the latter being responsible for the malignant
55 mesothelioma epidemic. Ballirano et al. (2009) carried out a combined spectroscopic and crystal-
56 chemical investigation of erionite-K from Rome hypothesizing that iron resides on the zeolite
57 surface under the form of oxide-like nanoparticles with dimensions spanning from 1 to 9 nm.
58 Notably, it was proposed that even upon inhalation of Fe-free fibers, they could imbibe Fe via ion
59 exchange because protein injury can lead to metal ion release (Carr and Frei 1999). In any case,

60 whether Fe is carried on from erionite or is captured in the lungs, it was highlighted that the Fenton
61 chemistry occurs on the surface of the fibers and that the coordination state of Fe²⁺ and Fe³⁺ is
62 critical for inducing cytotoxicity and genotoxic damage (Fach et al. 2003; Ruda and Dutta 2005).

63 Several recent researches aimed at the investigation of fibers toxicity have been dedicated to the
64 study of their surface reactivity, mainly by XPS analysis (Fantauzzi et al. 2010; 2012). The success
65 of those investigations is strongly dependent on a fine crystal-chemical and structural
66 characterization of the fibers. Coherently, a relevant effort has been devoted to improve the
67 reliability of the chemical data extracted from micro-analytical investigations carried out through
68 Transmission Electron Microscopy (TEM) (Dogan 2011), Scanning Electron Microscopy (SEM)
69 (Dogan and Dogan, 2008), and Electron MicroProbe Analysis (EMPA) (Lowers et al. 2010) to
70 properly constraint XPS experiments. This task is inherently very difficult for fibers and especially
71 for zeolites because of, a part of the well-known problems related to the small dimensions, their
72 instability under the electron beam that generally led to fairly reliable analytical results. As far as
73 the structural characterization of fibers is concerned to, a significant number of papers fully
74 exploiting the capabilities of laboratory parallel-beam transmission X-ray powder diffraction have
75 been recently published (Ballirano et al. 2008; Pacella et al. 2008; Andreozzi et al. 2009; Ballirano
76 et al. 2009; Ballirano and Cametti 2012). Such instrumental set up coupled with the preparation of
77 the sample as a capillary mount allows an almost complete removal of preferred orientation
78 significantly improving the reliability of the structural data even at non-ambient conditions
79 (Ballirano 2011a,b,c).

80 The present study is dedicated to the detailed crystal-chemical characterization of a sample of
81 woolly erionite from the type locality near Durkee, Oregon (USA), by combining data from X-ray
82 powder diffraction (XRPD) and SEM analysis. The reported results are expected to provide a sound
83 background for incoming experiments aimed at understanding the relationships occurring among
84 morphology, crystal chemical and structural features, and biological activity of erionite.

85

86 **EXPERIMENTAL METHODS**

87 **Scanning Electron Microscopy (SEM)**

88 The analyzed erionite sample was provided by the Smithsonian Institution, Washington (NMNH
89 R4066-1). At Durkee, Oregon, USA, erionite occurs in thin seams in gray, rhyolitic welded ash-
90 flow tuffs (Sheppard 1996). The hand specimen consists of ribbons and bundles aggregating to form
91 a hank having a white color and pearly luster as reported by Eakle (1898). SEM images were
92 collected at the Centro di ricerca per le Nanotecnologie applicate all'Ingegneria della Sapienza
93 (CNIS) using a field emission SEM (FESEM) Zeiss Auriga 405 with an accelerating voltage in the
94 range 1.3-6 kV at a working distance in the range 2.8-3.5 mm. A preliminary set of chemical
95 analyses was collected on bundles of erionite fibers. Moreover, two samples were prepared for a
96 micro-chemical characterization by dispersing a small aliquot of fibers in distilled water in a beaker
97 and subsequently pipetting and depositing a few droplets of the dispersion onto two aluminum stubs
98 with double adhesive carbon tape. One of the stubs was coated with carbon, the other with gold, to
99 check the possible occurrence of analytical differences arising from the coating. Analyses were
100 collected for each stub positively verifying that the analytical results are independent from the
101 coating type. The micro-chemical characterization was performed at the Dipartimento di Scienze
102 della Terra, Sapienza Università di Roma using a FEI Quanta 400 SEM equipped with an EDX
103 Genesis EDS system. Operating conditions were: 15 kV accelerating voltage, 11 mm working
104 distance, 0° tilt angle. Chemical data were collected at thirty analytical points. The final crystal
105 chemical formula was calculated, after renormalization of the chemical analyses hypothesizing a
106 water content of 18.5 wt% (corresponding to *ca.* 30 atoms per formula unit, *apfu*), on the basis of 36
107 (Si+Al+Fe³⁺) *apfu*.

108 In order to evaluate the quality of SEM-EDX microchemical data we performed a preliminary,
109 thorough comparison with EMPA, using a fairly large prismatic erionite-Ca crystal as a standard.
110 The sample, from Faedo, Colli Euganei, Italy, is a crystal of the same batch investigated by EMPA
111 by Passaglia and Tagliavini (1995). Compositions were determined using a Cameca SX50 electron

112 microprobe with the following conditions: 10 s counting time (peak), 5 s counting time
113 (background), beam diameter 20 μm , excitation voltage 15 kV, specimen current 20 nA,
114 wavelength-dispersive spectrometry (WDS). The following standards were used: wollastonite (Si
115 and Ca), rutile (Ti), corundum (Al), magnetite (Fe), metallic Mn (Mn), barite (Ba), celestine (Sr),
116 periclase (Mg), orthoclase (K), and jadeite (Na). The raw data were corrected on-line for drift, dead
117 time, and background; matrix correction has been performed with a standard ZAF program. Results
118 of the chemical analyses are compared in Table 1. As can be seen, no relevant difference were
119 observed among the three data sets, the SEM-EDX being characterized by a very marginal increase
120 of the data dispersion indicating the full reliability of the technique. At any case, we unsuccessfully
121 attempted to prepare a sample of erionite from Durkee for EMPA analysis by embedding in epoxy
122 small hanks of fibers. Unfortunately, the polishing procedure unavoidably produced the separation
123 of the fibrils resulting in irregular, unpolished, surfaces. As a consequence, the resulting EMPA
124 cannot be considered as reliable.

125

126 **X-ray Powder diffraction (XRPD)**

127 Bundles of fibers were selected under a binocular microscope on the basis of the absence of any
128 recognizable impurity phase. They were subsequently disaggregated using a knife and carefully
129 hand-ground, under ethanol, in an agate mortar. The powder was loaded in a 0.7 mm diameter
130 borosilicate capillary that was aligned onto a standard goniometer head. XRPD data were collected,
131 up to a $(\sin \theta/\lambda)_{\text{max}}$ of 0.619 \AA^{-1} , using a parallel-beam Bruker AXS D8 Advance Diffractometer,
132 operating in θ - θ geometry, equipped with Göbel mirrors on the incident beam, Soller slits on both
133 incident and (radial) diffracted beams, and a PSD VANTEC-1 detector. Experimental details of the
134 XRPD data collection are reported in Table 2. A careful scrutiny of the diffraction pattern confirmed
135 the absence of any detectable crystalline impurity, within the limits of the diffractometer sensitivity,
136 which is estimated to be of ca. 0.1 wt%. The structural refinement was carried out by the Rietveld
137 method using TOPAS v.4.2 (Bruker AXS 2009) running in launch mode. Peak shape was modeled

138 through FPA (Fundamental Parameters Approach), imposing the following full axial parameters:
139 divergence slit: 0.3° , source length 12 mm, sample length and receiving slit length 15.3(2) mm. The
140 starting structural model, taken from Alberti et al. (1997), consists, a part of the framework atoms,
141 of four EF cationic sites, K1, Ca1, Ca2, and Ca3; and six water molecules sites (OW7, OW8, OW9,
142 OW10, OW11, and OW12). Both the occupancy of all EF cationic and water molecules sites and
143 the displacement parameters of all sites were refined. However, because of the occurrence of
144 correlations, displacement parameters of the sites were constrained, according to the structural data
145 of Ballirano et al. (2009), as follow: $B_{T1} = B_{T2}$; $B_{O1} = B_{O2} = B_{O3} = B_{O4} = B_{O5} = B_{O6}$; $B_{Ca1} = B_{Ca2} =$
146 B_{Ca3} ; $B_{Ow8} = B_{Ow9} = B_{Ow10} = B_{Ow11} = B_{Ow12} = 2 * B_{Ow7}$. Absorption was modeled following the Sabine
147 model for cylindrical samples (Sabine et al. 1998) and the background was fitted by a Chebychev
148 polynomial of the first kind. The occurrence of preferred orientation was tested by means of
149 spherical harmonics (four refinable parameters up to the 6th order). The choice of the number of
150 terms to be used has been performed following the procedure described by Ballirano (2003). As
151 expected for a capillary mount, only a marginal improvement of the fit was observed as a result of
152 the nearly absence of texture. A first refinement was carried out considering an isotropic shape of
153 the crystallites. The refinement smoothly converged to the following agreement indices: $R_{wp} =$
154 4.92% , $R_p = 3.44\%$, $R_B = 0.89\%$ (Young 1993). However, a careful scrutiny of the Rietveld plots
155 (Figure 1a) indicated an imperfect fitting of the shape of a few peaks due to the strongly anisotropic
156 shape of the crystallites. Therefore, a second refinement was performed using the recently proposed
157 ellipsoid-model of Katerinopoulou et al. (2012) describing the diffraction-vector dependent
158 broadening of diffraction maxima. In the hexagonal symmetry, the shape ellipsoid parameters b_{ij} are
159 constrained as $b_{11} = b_{22} = 2b_{12}$; $b_{13} = b_{23} = 0$. The orientation of the ellipsoid is such that the
160 principal radii $r_a \perp c$ and $r_c \parallel c$. A significant improvement of the fit was obtained as compared to
161 the use of an isotropic shape of crystallites as indicated by the final agreement indices: $R_{wp} = 3.36\%$,
162 $R_p = 2.45\%$, $R_B = 0.47\%$. Final Rietveld plots are shown in Figure 1b.

163

164 **RESULTS AND DISCUSSION**

165 **Scanning electron microscopy (SEM)**

166 From a morphological point of view, the sample consists of sub-parallel or bent bundles of fibers
167 with woolly aspect. The diameter of the bundles ranges from 10 to 20 μm (Figure 2a). Each fiber is
168 composed of very thin and curled fibrils as outlined in Figure 2b. The diameter of the fibers ranges
169 from 0.3 to 2 μm , whereas that of a single fibril may be as small as 15 nm (Figure 2c). It is worth
170 noting that as long fibers are tensile, elastic, and flexible, they are potentially extremely pathogenic
171 upon inhalation because they cannot be phagocytized. The woolly aspect of the fibers is typical of
172 the locality of Durkee, Baker County, as firstly reported by Eakle (1898). In addition, Gude and
173 Sheppard (1981) described a woolly erionite sample from the Reese River deposit, Nevada,
174 identical in appearance to that from Durkee. Notably, the morphology of both of these erionite is
175 peculiar as it differs from that of erionite samples from other localities, which have been described
176 as prismatic or acicular in habit.

177 The EDX spectra acquired on the bundles of fibers revealed, coherently with reference data, the
178 occurrence of Ca, Mg, Na and, K as EF cations. In addition, a small amount of Fe content was
179 detected (Figure 3). Moreover, preliminary semi-quantitative analyses pointed out that both CaO
180 and Fe_2O_3 content widely differs depending on the analytical point, indicating a significant
181 inhomogeneity of the sample. In particular, Fe_2O_3 and CaO content ranged from zero up to 4 wt%,
182 and from zero up to 6 wt%, respectively. It must be noted that, as highlighted by FESEM images
183 (Figure 2), it was very hard to discriminate and analyze a single fiber. For this reason a new set of
184 analyses was carried out on isolated bundles obtained by dispersion in distilled water. In this case,
185 the collected spectra clearly indicated the absence of both CaO and Fe_2O_3 (Figure 4). On this basis,
186 the presence of these elements in the erionite fibers was ruled out and attributed to impurity phases.
187 As confirmation of this, FESEM images revealed the presence of small particles of variable size
188 (from 1 to 10 μm) and irregular shape resting at the fiber surface (Figure 5). Their chemical
189 composition was qualitatively evaluated and the corresponding spectra showed that both Ca and Fe

190 are associated to these particles (Figure 6).

191 The reliability of the chemical analyses of the fibers was evaluated using both the balance error
192 formula E% (Passaglia 1970) and the Mg-content test (Dogan and Dogan 2008). Using the above-
193 mentioned experimental cautions, approximately 70% of the analyses passed both tests. The
194 average chemical composition is reported in Table 3. The MgO content did not exceed 0.54 wt%,
195 whereas Na resulted the most abundant EF cation with a Na₂O content ranging from 4.78 to 6.44
196 wt%. The final average crystal-chemical formula, calculated from 16 analytical point passing both
197 the E% and the Mg-content tests, of (Na_{5.38}K_{1.99}Mg_{0.24})[Al_{7.66}Si_{28.34}O_{72.09}] • 29.83H₂O allows a
198 sample classification as erionite-Na. The calculated R = Si/(Al+Si) ratio of 0.787 is identical to that
199 found by Passaglia et al. (1998) for the erionite sample no. 25 from Durkee. The abundance of Na
200 and the absence of Ca result in a very high M/(M+D) ratio of 0.97 (M = Na+K; D = Ca+Mg+Mn).
201 A similar value of M/(M+D) = 0.95, was previously found in only one of the large set of 25 erionite
202 samples analyzed by Passaglia et al. (1998) (sample no. 8 from Phillip Island, Australia). In fact, its
203 chemical composition, in term of EF cations, is very close to that of the erionite investigated in this
204 work the exception being a higher K₂O content. Besides, sample no. 25 from Durkee has a chemical
205 composition significantly different with respect to that of the present sample, and as a result, it can
206 be classified as erionite-Ca. Moreover, chemical data obtained in the present work significantly
207 differ with respect to those previously obtained on woolly erionite samples by various authors
208 (Eakle 1898; Staples and Gard 1959; Gude and Sheppard 1981). In particular, in the case of
209 erionite-Na from Durkee the Si/(Al+Fe³⁺) ratio is significantly larger (reference samples do not
210 exceed 3.50 instead of 3.69 of the present sample) as well as the M/(M+D) ratio (less than 0.80 in
211 reference samples: Sheppard and Gude 1969).

212

213 **Structural refinement**

214 **Micro-structural parameters and framework**

215 Cell parameters $a = 13.2356(5)$ Å and $c = 15.0652(6)$ Å and cell volume $V = 2285.6(2)$ Å³ are

216 similar to those reported in Ballirano et al. (2009) for erionite-K from Rome, Oregon, as expected
217 from the similar R value. A comparison with reference data of cell parameters of erionite samples
218 from Durkee is reported in Table 4. A good agreement is observed with the data of Passaglia et al.
219 (1998) for a sample characterized by an identical R value but a different EF cationic content. The
220 cell parameters of Sheppard and Gude (1969) are systematically larger than those of both the
221 present work and Passaglia et al. (1998). However, the c/a ratios are identical within 1σ .

222 From the refined anisotropic crystal shape parameters b_{11} and b_{33} , the principal ellipsoid radii $r_a =$
223 $15.5(2)$ nm and $r_c = 79(5)$ nm were computed using the relations $r_a = 1/(a \cos 30^\circ b_{11}^{0.5})$ and $r_c =$
224 $1/(c b_{33}^{0.5})$, leading to a r_c/r_a ratio of 5.1(9). Moreover a ϵ_0 micro-strain (lattice strain) of 0.200(7)
225 was obtained (Ballirano and Sadun 2009). For comparison purposes, the refinement with unique
226 average crystallite size gave spherical crystallites with radius $r = 30(1)$ nm and a ϵ_0 micro-strain of
227 0.207(12).

228 The r_a value is comparable with the diameter of a single fibril observed by FESEM. On the
229 contrary, the coherency along the fiber axis is significantly shorter than the full length of the fibril.
230 This is clearly an effect of the bending of the fibrils that reduces the coherency along the c -axis.

231 Fractional coordinates, isotropic displacement parameters, and site scattering (*s.s.*) are reported in
232 Table 5, relevant bond distances and angles in Table 6, and bond valence analysis in Table 7.

233 The mean bond distances $\langle T1-O \rangle = 1.631$ Å and $\langle T2-O \rangle = 1.646$ Å clearly indicate a disordered
234 Si/Al distribution ($\langle T1-O \rangle - \langle T2-O \rangle = -0.015$ Å) and the preferential partition of Al at the T2 site.
235 The R ratio has been calculated by using both the Jones' determinative curves (Jones, 1968) and the
236 regression equation of Passaglia et al. (1998). The population of the two T1 ($Al_{4.21}Si_{19.79}$) and T2
237 ($Al_{3.26}Si_{8.74}$) sites, determined using the Jones' determinative curve, provides an R value of 0.793.
238 Besides, an R value of 0.783 has been obtained using the regression equation of Passaglia et al.
239 (1998). Both values are reasonably close with that of 0.787 obtained from chemical data. On the
240 contrary, the corresponding mean bond distances $\langle T1-O \rangle = 1.630$ Å and $\langle T2-O \rangle = 1.705$ Å
241 obtained from the isotropic crystallite shape refinement seem to indicate a very strong preference of

242 Al for T2 ($\text{Al}_{7.80}\text{Si}_{4.20}$). However, the mean $\langle\text{T-O}\rangle$ bond distances result in an R value of 0.671,
243 which is unsupported by both chemical and unit cell volume data. This result can be rationalized
244 considering the deleterious effects of an imperfect peak-shape modeling on the accuracy of the
245 structural parameters determination. Therefore, any effort has to be paid to adequately model the
246 anisotropic broadening of diffraction maxima to extract reliable structural information from fibrous
247 samples.

248

249 **Extraframework cations**

250 Chemical data indicate a total site scattering of 99.9 e^- for the EF cations. This value is in
251 reasonable agreement with the *s.s.* of 98(4) e^- obtained from the Rietveld refinement (Table 5).
252 Similarly, a total of 31.2(13) water molecules *pfu* has been obtained from the refinement, in good
253 agreement with reference data (Coombs et al. 1997; Ballirano et al. 2009). Notably, the *s.s.* value of
254 100(5) e^- and 42(5) water molecules *pfu*, obtained in the isotropic crystallite shape refinement,
255 confirm the fundamental role played by an accurate fitting of the peak shape for extracting reliable
256 structural parameters.

257 The K atom lies at the K1 site located at the center of the cancrinite cage. The fully occupied site
258 is 12-fold coordinated by six O2 atoms and six O3 atoms at 2.957(10) Å and 3.348(9) Å
259 respectively (Table 6). For this site, bond valence analysis, performed using the parameters of
260 Breese and O'Keeffe (1991), results in 0.86 valence units (*v.u.*) (Table 7). This site does not provide
261 a favorable coordination for Na as indicated by bond valence analysis (0.36 *v.u.*). Therefore,
262 reported values of less than 2 K *apfu* indicate a partly vacant K1 site or, more plausibly, an
263 inaccurate chemical analysis. Attempts to detect the occurrence of electron density at the K2 site,
264 located at the center of the boat-shaped 8-member rings (8MR) forming the walls of the erionite
265 cage, failed. As reported in Ballirano et al. (2009), this site corresponds to that found by Schlenker
266 et al. (1977) in dehydrated erionite and to the site labeled as Ca4 by Gualtieri et al. (1998) in two
267 erionite samples. It has been shown that the K2 site provides a very favorable coordination for K

268 atoms at room temperature. However, in our case the available K is entirely located within the
269 cancrinite cages and, as a consequence of this, the K2 site is empty. The remaining EF cationic sites,
270 Ca1, Ca2 and Ca3, are located at $(1/3, 2/3, z)$, along the three-fold axis. These sites were refined
271 using the Mg scattering factor for the Ca1 site and the Na scattering factor for the Ca2 and Ca3
272 sites, following the same approach adopted by Ballirano et al. (2009) for erionite-K from Rome.
273 Atomic fractional coordinates and displacement parameters are reported in Table 4. Values are in
274 good agreement with those reported for erionite-K from Rome in Ballirano et al. (2009), and those
275 reported for erionite-Ca from Tunguska (Russia) by Alberti et al. (1997). Differences in both *s.s.*
276 and population of the EF cationic sites are clearly related to the different chemical composition. The
277 Ca1 site has a very low *s.s.* of 5.4(14) e^- and the associated high standard deviation of bond and
278 contact distances render difficult a meaningful discussion about its coordination and site
279 assignment. However, it has been identified a fairly regular six-fold coordination (Ca1-OW8 x3 at
280 1.88(9) Å and Ca1-OW10 x3 at 2.68(12) Å) well fitting the occurrence of Mg (Table 6). This
281 assignment is confirmed by a bond-valence sum of 2.03 *v.u.* The value of 0.24 *apfu* obtained from
282 chemical analysis for Mg corresponds to a *s.s.* value of 2.9 e^- . It is worth noting that, this value is in
283 reasonable agreement with the results obtained from the structural refinement, under the hypothesis
284 that all Mg content is located at Ca1, as in the case of erionite-K (Ballirano et al. 2009). However,
285 the coordination of Mg is different in erionite-Na and -K as both Ca1 and Ca2 sites in erionite-K
286 are slightly displaced off-axis, and the OW9 water molecule site is significantly displaced in
287 erionite-Na as compared to both erionite-K and -Ca (Alberti et al., 1997). Therefore, Na is located
288 at both Ca2 and Ca3 sites. The Ca2 site has the largest *s.s.* value (35.5(10) e^-) among the three EF
289 cationic sites located within the erionite cage. It is characterized by a nine-fold coordination (Ca2-
290 OW8 x3 at 2.54(4) Å, Ca2-OW12 x3 at 2.68(3) Å, and Ca2-O5 x3 at 3.376(19) Å) leading to a
291 bond-valence sum of 0.72 *v.u.* The Ca3 site, not occurring in erionite-K, is at bonding distance with
292 OW7, OW10, and OW11 sites. However, according to the bond-valence analysis and taking into
293 account the short non occurring contact OW7-OW11 (1.73(6) Å), a nine-fold coordination may be

294 proposed, consisting of Ca3-OW10 x 3 at 2.47(3) Å, Ca3-OW11 x3 at 2.64(7), and Ca3-OW10 x3
295 at 3.10(4) Å, leading to a bond valence sum of 0.90 *v.u.*

296 Concerning the water molecule positions, the most relevant difference with respect to Rome and
297 Tunguska erionite samples is related to the displacement of OW9 (from x ca. 0.44, $y = 2x$; $z =$ ca.
298 0.92 to x ca. 0.50, $y = 2x$; $z =$ ca. 0.95).

299 Water molecules site scattering of erionite-K and -Na are similar, a part for an inversion between
300 neighboring OW8 and OW9 sites.

301 On the contrary, differences are very relevant with respect to erionite-Ca (Alberti et al. 1997).
302 Besides, in the case of the present erionite-Na sample, OW9 is linked exclusively to framework
303 oxygen atoms and other H₂O molecules differently from OW8, OW11, and OW12 of erionite-K
304 (Ballirano et al. 2009).

305

306 **Environmental and health relevance**

307 Despite many works have reported the presence of a variable Fe and Ca content in erionite
308 samples from Durkee, our results revealed the absence of both these elements. In particular, the
309 absence of Fe, confirming the findings of Ballirano et al. (2009) for erionite-K from Rome, it is of
310 relevant interest because it is well known from amphibole asbestos that Fe has been claimed to be
311 one of the causes of carcinogenesis by participating in Fenton chemistry and producing free radicals
312 (e.g., Kane et al. 1996; Fubini and Otero Aréan 1999; Kamp and Weitzman 1999; Robledo and
313 Mossman 1999). Notably, it was proposed that, in addition Fe presence, erionite ion exchanging
314 could play a primary role to induce toxicity (Eborn and Aust 1995). In fact, cations may be readily
315 exchanged whenever a zeolite is in contact with human body fluids. It is worth noting that Ballirano
316 and Cametti (2012) have shown that Mg in Ca1 is the most mobile cation during heating of erionite-
317 K, despite its large ionic potential. Notably, results of *in vitro* experiments on amphibole asbestos
318 revealed that the release of Mg²⁺ can have different effects on lipid packing, membrane
319 permeability and can counteract Ca²⁺ uptake by blocking the calcium uniporter (Bergamini et al.

320 2007; Pacella et al. 2012). Further studies are in progress on the erionite sample here studied,
321 concerning the exchange properties of erionite in contact with a physiological solution and the
322 reactivity (detection of oxy-radicals and monitoring of lipid peroxidation). The combined approach
323 consisting in the full characterization of the fibrous erionite coupled with the cation exchange
324 properties and reactivity studies is very promising to shed new light on the role of the ion exchange
325 in the interaction between fibers and organic environment.

326 Besides, the possible role of erionite as a carrier within the human body for nanoscale iron-bearing
327 materials needs to be carefully investigated.

328

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336

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459

460 **Figure Captions**

461

462 **FIGURE 1.** Above: experimental (dots) and total calculated (continuous); middle: calculated
463 erionite; bottom: difference plot of the refinement of erionite-Na from Durkee, Oregon: **a)**
464 refinement with an isotropic crystallites shape; **b)** refinement with an anisotropic crystallites shape.
465 Vertical bars refer to the position of the Bragg reflections. Onset: shape of the ellipsoid shape and
466 corresponding r_a and r_c principal radii.

467

468 **FIGURE 2.** FESEM images of woolly erionite-Na. Bundles of fibers (**a**) and fibrils (**b,c**).

469

470 **FIGURE 3.** Preliminary SEM-EDX spectrum of bundles of erionite-Na. The S and Au peaks arise
471 from, respectively, the stub adhesive and the coating.

472

473 **FIGURE 4.** SEM-EDX spectrum of dispersed fibers of erionite-Na.

474

475 **FIGURE 5.** FESEM images of small particles associated with erionite-Na fibers.

476

477 **FIGURE 6.** SEM-EDX spectrum of small particles occurring with erionite-Na fibers.

478

479 **FIGURE 7.** Ca1 site coordination (CrystalMaker 8.7; CrystalMaker Software 2012).

480

481 **FIGURE 8.** Ca2 site coordination (CrystalMaker 8.7; CrystalMaker Software 2012).

482

483 **FIGURE 9.** Ca3 site coordination (CrystalMaker 8.7; CrystalMaker Software 2012).

484

TABLE 1. Chemical data of erionite-Ca from Faedo, Colli Euganei, Italy as obtained from SEM-EDX and EMPA. Crystal chemical formula calculated on the basis of $36(\text{Si}+\text{Al}+\text{Fe}^{3+})$ *apfu*. EMPA of Passaglia and Tagliavini (1995), labeled as P&T1995, is reported for comparison. Minimum and maximum values are reported in italics.

R = Si/(Si + Al); M = Na + K; D = Ca + Mg + Sr + Ba;

E% = $100 * [(\text{Al} + \text{Fe})_{\text{ob}} - \text{Al}_{\text{th}}] / \text{Al}_{\text{th}}$ where $\text{Al}_{\text{th}} = \text{Na} + \text{K} + 2 * (\text{Ca} + \text{Mg} + \text{Sr} + \text{Ba})$.

	P&T1995	Present work*	
		EMPA	SEM-EDX
morphology		prismatic	
SiO ₂	54.87	54.38 (53.58-54.95)	55.46 (54.25-56.01)
Al ₂ O ₃	16.77	16.45 (16.17-16.68)	16.22 (16.17-16.43)
TiO ₂	-	0.07 (0.00-0.09)	-
SrO	0.12	0.02 (0.00-0.27)	-
BaO	0.02	0.08 (0.00-0.24)	-
CaO	6.72	6.95 (6.62-7.47)	6.38 (5.90-7.24)
MnO	-	0.06 (0.00-0.26)	-
MgO	0.05	0.11 (0.05-0.19)	0.12 (0.00-0.22)
Na ₂ O	0.32	0.12 (0.04-0.30)	0.12 (0.00-0.30)
K ₂ O	2.62	3.22 (2.89-3.51)	3.17 (2.60-3.90)
H ₂ O	18.50	18.50	18.50
total	100	100	100
Unit-cell atomic content			
Si	26.45	26.53 (26.43-26.58)	26.76 (26.71-26.77)
Al	9.54	9.47 (9.34-9.62)	9.24 (9.15-9.32)
Ti	-	0.02 (0.00-0.09)	-
Sr	0.01	0.01 (0.00-0.08)	-
Ba	-	0.02 (0.00-0.04)	-
Ca	3.47	3.64 (3.45-3.96)	3.30 (3.05-3.83)
Mn	-	0.03 (0.00-0.11)	-
Mg	0.04	0.08 (0.04-0.14)	0.09 (0.00-0.16)
Na	0.30	0.11 (0.04-0.29)	0.11 (0.00-0.28)
K	1.62	2.00 (1.80-2.21)	1.95 (1.59-2.53)
O	71.73	72.14	71.80 (71.53-72.44)
H ₂ O	29.81	30.17	29.84
R	0.735	0.737	0.743 (0.741-0.746)
M/(M+D)	0.539	0.562	0.609
Na/M	0.155	0.053	0.054
E%	5.8	-1.4	4.4

*Mean values renormalized hypothesizing a water content of 18.5 wt.%.

TABLE 2. Experimental details of XRPD data collection.

Instrument	Bruker AXS D8 Advance
Radiation	CuK α
Primary and secondary radius (mm)	250
Detector	PSD VÅNTEC-1
Sample mount	Rotating capillary (60 r/min)
Incident beam optics	60 mm multilayer (Göbel) X-ray mirror
Detector window (°)	6
Divergence slits (°)	0.3
Soller slits	Primary beam (2.3°); diffracted beam (radial)
Angular range (°2 θ)	4-145
($\sin \theta/\lambda$) _{max} (Å ⁻¹)	0.619
Step size (°2 θ)	0.022
Counting time (s)	10

TABLE 3. Results of quantitative SEM-EDX microanalyses of erionite from Durkee, Oregon. Chemical analyses of reference samples are reported for comparison.

	1	2	3	present work*
morphology	woolly	woolly	acicular/ prismatic	woolly
SiO ₂	57.16	57.40	58.35	58.74 (58.17-59.36)
Al ₂ O ₃	16.08	15.60	13.32	13.46 (13.19-13.83)
Fe ₂ O ₃	-	-	0.16	-
BaO	-	-	0.68	-
CaO	3.50	2.92	4.11	-
MgO	0.66	1.11	0.83	0.33 (0.14-0.54)
Na ₂ O	2.47	1.45	0.36	5.75 (4.78-6.44)
K ₂ O	3.51	3.40	2.72	3.22 (2.69-3.67)
TiO ₂	-	-	-	-
P ₂ O ₅	-	-	-	-
MnO	-	-	-	-
H ₂ O	17.30	17.58	19.47	18.50
total	100.68	99.46	100.30	100.00
Unit-cell atomic content				
Si	27.04	27.27	28.31	28.34 (28.24-28.44)
Al	8.96	8.73	7.61	7.66 (7.56-7.76)
Fe ³⁺	-	-	0.06	-
Fe ²⁺	-	-	-	-
Ba	-	-	0.13	-
Ca	1.77	1.49	2.14	-
Mg	0.47	0.79	0.60	0.24 (0.10-0.38)
Na	2.27	1.34	0.34	5.38 (4.43-6.05)
K	2.12	2.06	1.68	1.98 (1.66-2.29)
O	71.95	71.60	72.16	72.09 (71.69-72.39)
H ₂ O	27.29	27.85	31.57	29.77
R	0.751	0.757	0.787	0.787 (0.784-0.790)
M/(M+D)	0.662	0.599	0.425	0.969
Na/M	0.517	0.333	0.168	0.731
E%	1.2	10.0	-0.9	-2.3
species	-Na	-K	-Ca	-Na

*Mean values, from 16 analytical points renormalized hypothesizing a water content of 18.5 wt.%.

1. Chemical analysis by Eakle (1898).

2. Chemical analysis by Staples and Gard (1959).

3. Chemical analysis of sample no. 25, by Passaglia et al. (1998).

TABLE 4. Comparison of cell parameters of erionite samples from Durkee.

	1	2	present work
morphology	woolly	acicular/ prismatic	woolly
a (Å)	13.254(6)	13.233(1)	13.2357(5)
c (Å)	15.100(10)	15.055(4)	15.0652(6)
Vol. (Å ³)	2297(3)	2283.1(6)	2285.6(2)
c/a	1.1393(12)	1.1377(4)	1.1382(1)
<i>species</i>	-Na	-Ca	-Na

1. Sheppard and Gude (1969) recalculated from the original d -spacings of Staples and Gard (1959).

2. Passaglia et al. (1998): sample no. 25.

TABLE 5. Fractional coordinates, isotropic displacement parameters, site multiplicity and occupancy, and site scattering (*s.s.*) of erionite-Na. Si, Al site partition from Jones' (1968) determinative curves.

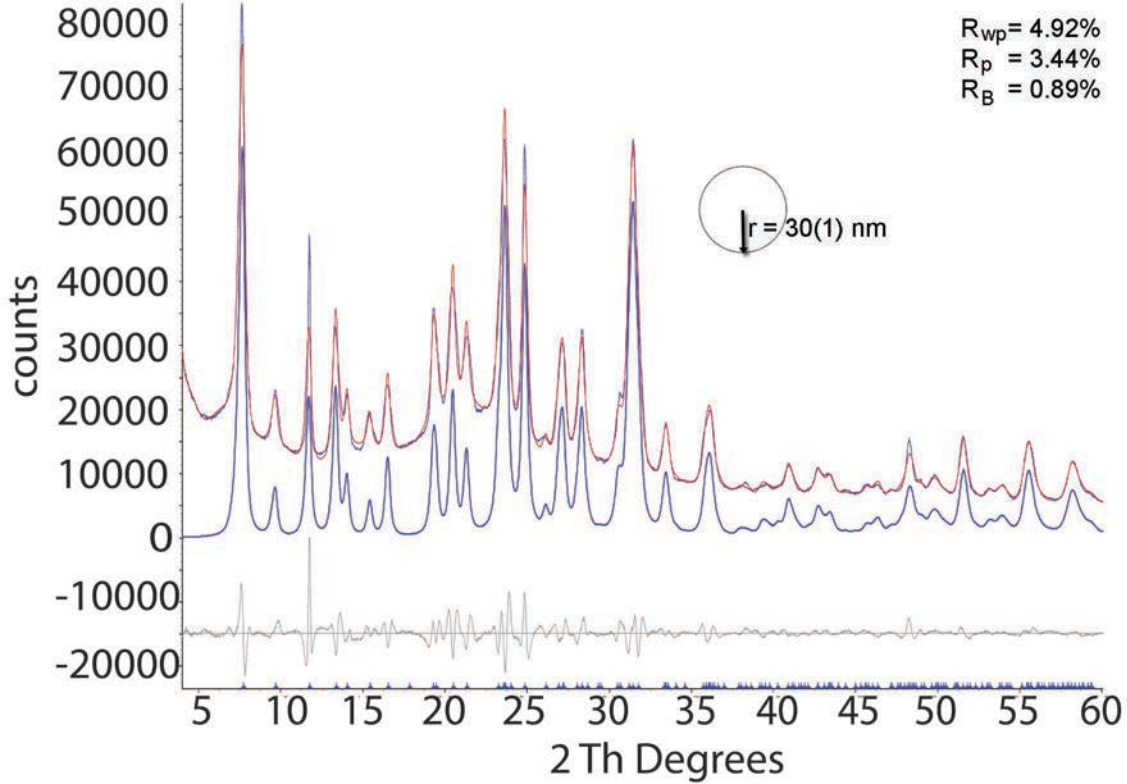
<i>Site</i>	<i>Site mult.</i>	<i>x</i>	<i>y</i>	<i>z</i>	$B_{iso}(\text{\AA}^2)$	Site occ.	<i>s.s.</i> (e^-)	Proposed site partition	<i>s.s.</i> (e^-) from site partition
Framework									
T1	24	0.2331(3)	0.0001(4)	0.10454(16)	0.88(5)	1		Si _{19.79} Al _{4.21}	
T2	12	0.0958(4)	0.4246(5)	1/4	0.88(5)	1		Si _{8.74} Al _{3.26}	
O1	24	0.3483(5)	0.0239(5)	0.6604(5)	1.34(13)	1			
O2	12	0.1005(4)	2 <i>x</i>	0.1269(7)	1.34(13)	1			
O3	12	0.1270(3)	2 <i>x</i>	0.6402(8)	1.39(19)	1			
O4	12	0.2634(6)	0	0	1.34(13)	1			
O5	6	0.2307(6)	2 <i>x</i>	1/4	1.39(19)	1			
O6	6	0.4616(7)	2 <i>x</i>	1/4	1.39(19)	1			
EF cations									
Ca1	4	1/3	2/3	0.924(16)	12.3(13)	0.112(29)	5.4(14)	Mg _{0.24}	2.9
Ca2	4	1/3	2/3	0.0893(15)	12.3(13)	0.806(23)	35.5(10)	Na _{3.48}	38.3
Ca3	4	1/3	2/3	0.7068(28)	12.3(13)	0.44(3)	19.4(14)	Na _{1.90}	20.9
K1	2	0	0	1/4	2.63(24)	1	38	K _{1.99}	37.8
						$\Sigma_{cat.}$	98(4)		99.9
Water molecules									
OW7	6	0.2375(14)	2 <i>x</i>	3/4	5.0(3)	0.641(22)	30.8(11)		
OW8	12	0.2565(18)	2 <i>x</i>	0.9674(28)	9.9(6)	0.404(17)	38.8(16)		
OW9	12	0.498(5)	2 <i>x</i>	0.950(4)	9.9(6)	0.239(18)	22.9(17)		
OW10	12	0.4365(13)	2 <i>x</i>	0.6601(16)	9.9(6)	0.565(15)	54.2(14)		
OW11	12	0.2707(26)	2 <i>x</i>	0.647(4)	9.9(6)	0.363(24)	34.8(23)		
OW12	12	0.4365(13)	2 <i>x</i>	0.050(17)	9.9(6)	0.704(23)	67.6(22)		
						Σ_{water}	249(10)		

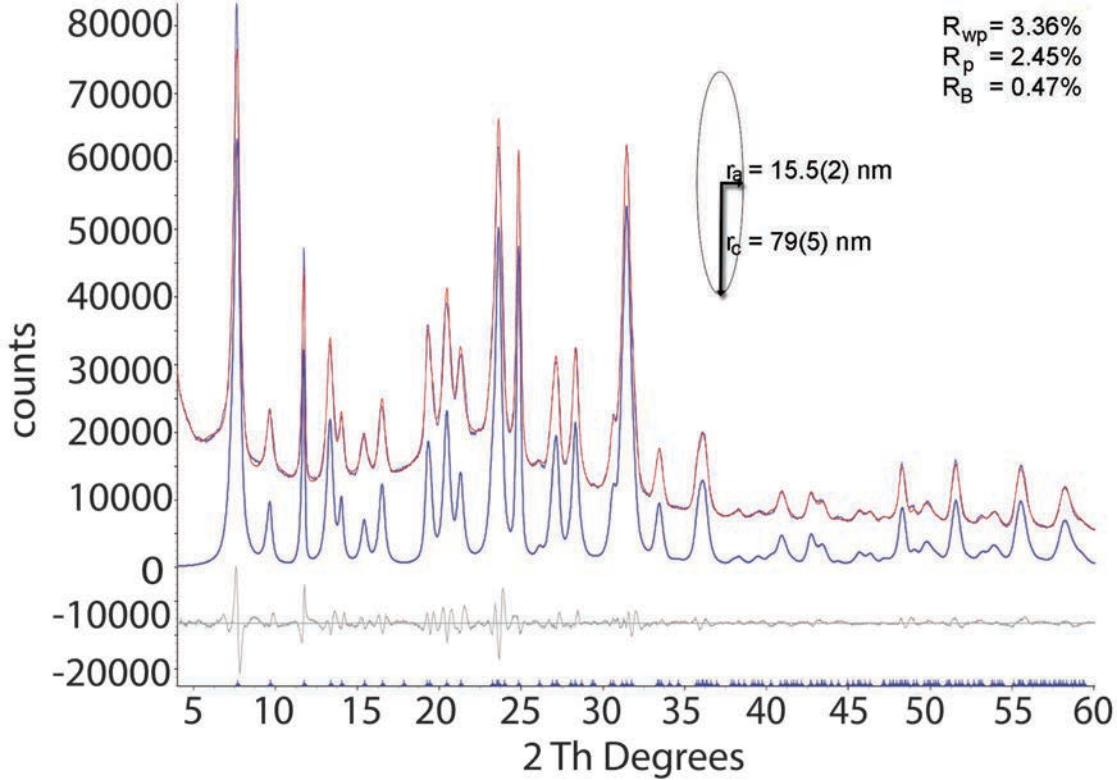
TABLE 6. Relevant bond distances (Å) and angles (°) of erionite-Na from Durkee, Oregon. In italic never occurring short distances.

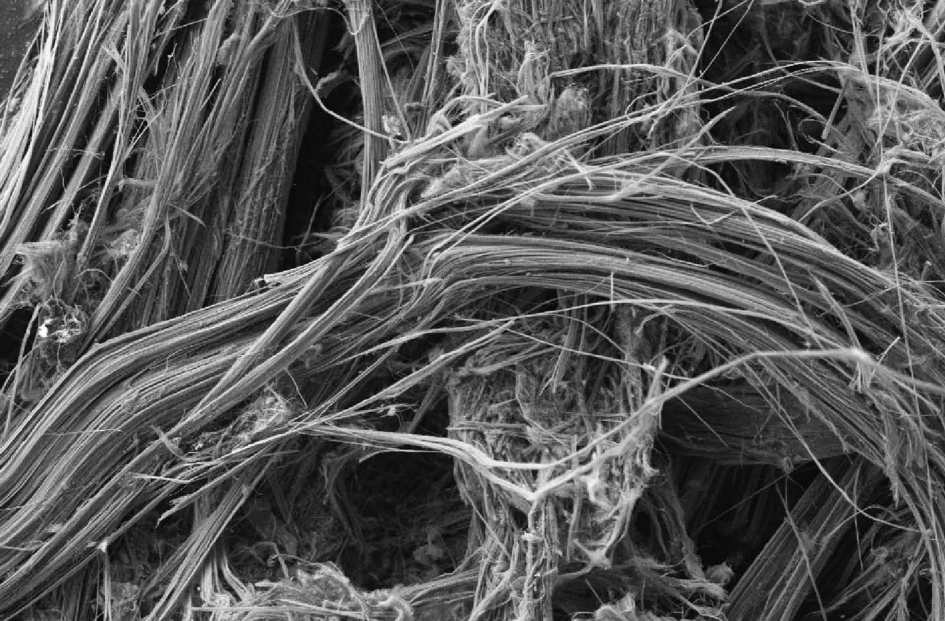
T1-O2	1.622(5)	O1-T1-O2	109.7(5)
T1-O4	1.625(3)	O1-T1-O3	105.5(4)
T1-O1	1.629(9)	O1-T1-O4	107.0(5)
T1-O3	1.649(7)	O2-T1-O3	109.2(5)
<T1-O>	1.631	O2-T1-O4	111.5(4)
		O3-T1-O4	113.6(5)
T2-O5	1.598(7)	O1-T2-O1	107.9(5)
T2-O6	1.647(8)	O1-T2-O5 x2	113.4(5)
T2-O1 x2	1.670(7)	O1-T2-O6 x2	104.6(5)
<T2-O>	1.646	O5-T2-O6	112.3(7)
T1-O1-T2	140.3(6)	T1-O4-T1	151.4(6)
T2-O2-T1	144.2(7)	T2-O5-T2	149.4(11)
T1-O3-T1	138.2(8)	T2-O6-T2	164.8(11)
K1-O2 x6	2.957(10)		
K1-O3 x6	3.348(9)		
Ca1-OW11x3	<i>1.79(16)</i>	Ca1-Ca3	<i>1.97(24)</i>
Ca1-OW8 x3	1.88(9)	Ca1-Ca2	<i>2.49(24)</i>
Ca1-OW12 x3	2.66(12)	OW8-OW11	<i>1.75(7)</i>
Ca1-OW10 x3	<i>2.68(12)</i>		
Ca2-OW8 x3	2.54(4)		
Ca2-OW12	2.68(3)		
Ca2-O5 x3	3.376(19)		
Ca3-OW11x3	<i>1.70(6)</i>	Ca3-Ca3	<i>1.30(8)</i>
Ca3-OW7 x3	2.29(3)	OW7-OW11	<i>1.73(6)</i>
Ca3-OW10 x3	2.47(3)		
Ca3-OW11 x3	2.64(7)		
Ca3-OW10 x3	3.10(4)		
Ca3-OW8 x3	3.16(5)		
		OW9-OW9	<i>1.51(13)</i>
		OW9-OW12	<i>1.63(10)</i>
		OW9-OW12	<i>1.66(10)</i>

TABLE 7. Bond valence analysis (*v.u.*) following Breese and O'Keeffe (1991).

	K	Na	Mg
K1-O2 x6	0.107 x6		
K1-O3 x6	0.037 x6		
Σ	0.864		
Ca1-OW8 x3			0.607 x3
Ca1-OW10 x3			0.069 x3
Σ			2.028
Ca2-OW8 x3		0.134 x3	
Ca2-OW12 x3		0.092 x3	
Ca2-O5 x3		0.014 x3	
Σ		0.720	
Ca3-OW10 x3		0.164 x3	
Ca3-OW11 x3		0.105 x3	
Ca3-OW10 x3		0.030 x3	
Σ		0.897	







10 μ m



EHT = 6.00 kV

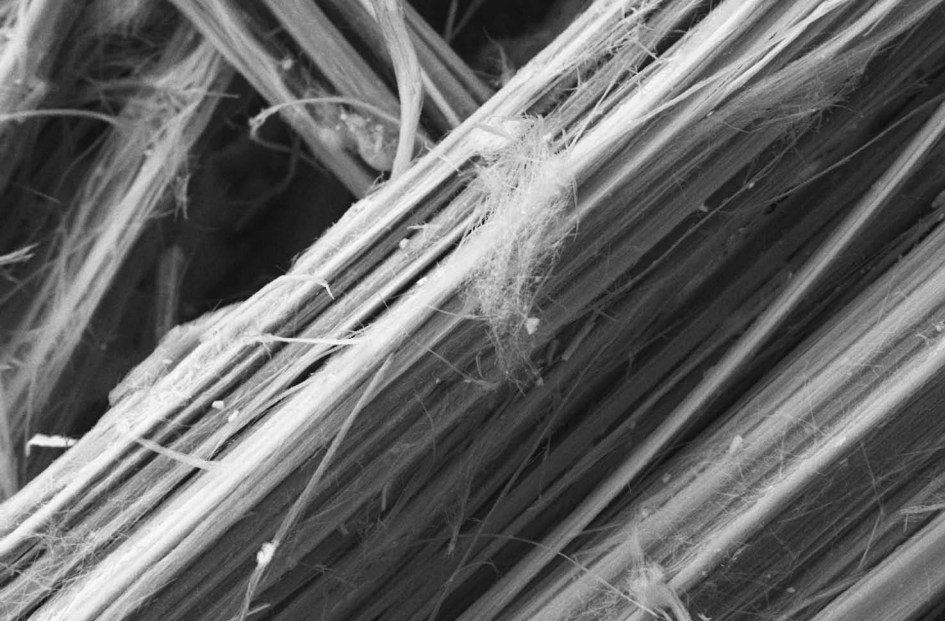
WD = 3.5 mm

Signal A = SE2

Mag = 1.00 K X

Date :24 Feb 2012

Sample ID = Ballirano



2 μm



EHT = 6.00 kV

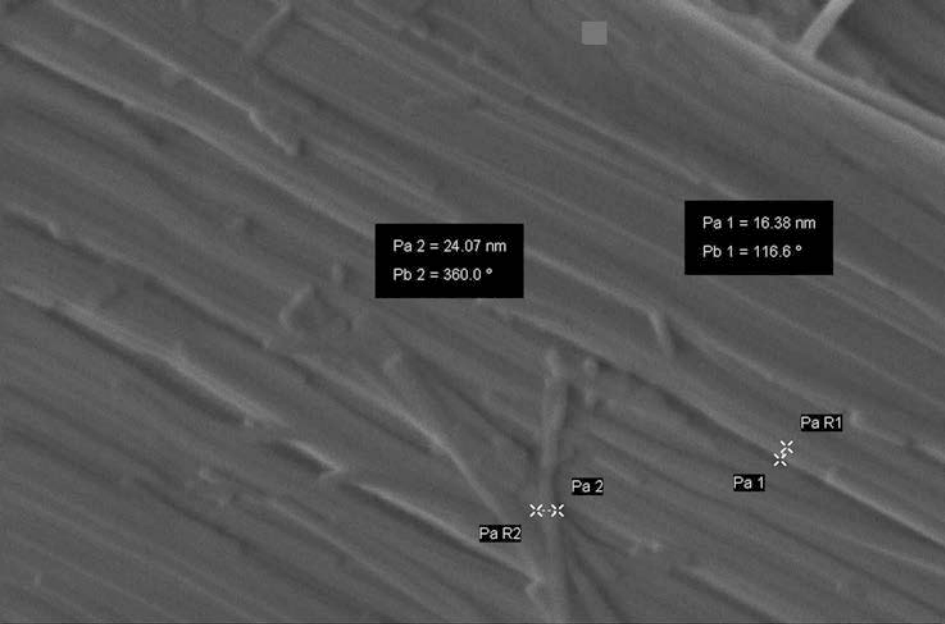
WD = 3.5 mm

Signal A = SE2

Mag = 8.00 K X

Date :24 Feb 2012

Sample ID = Ballirano



Pa 2 = 24.07 nm
Pb 2 = 360.0 °

Pa 1 = 16.38 nm
Pb 1 = 116.6 °

Pa R1

Pa 2

Pa 1

Pa R2

100 nm



EHT = 1.30 kV

WD = 3.5 mm

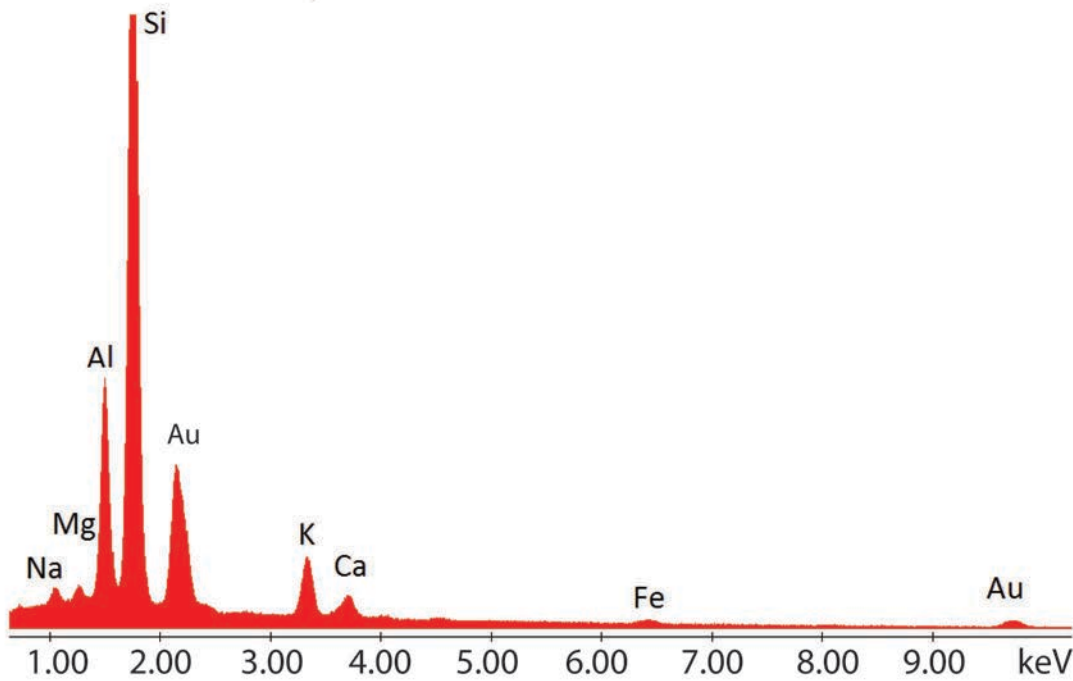
Signal A = InLens

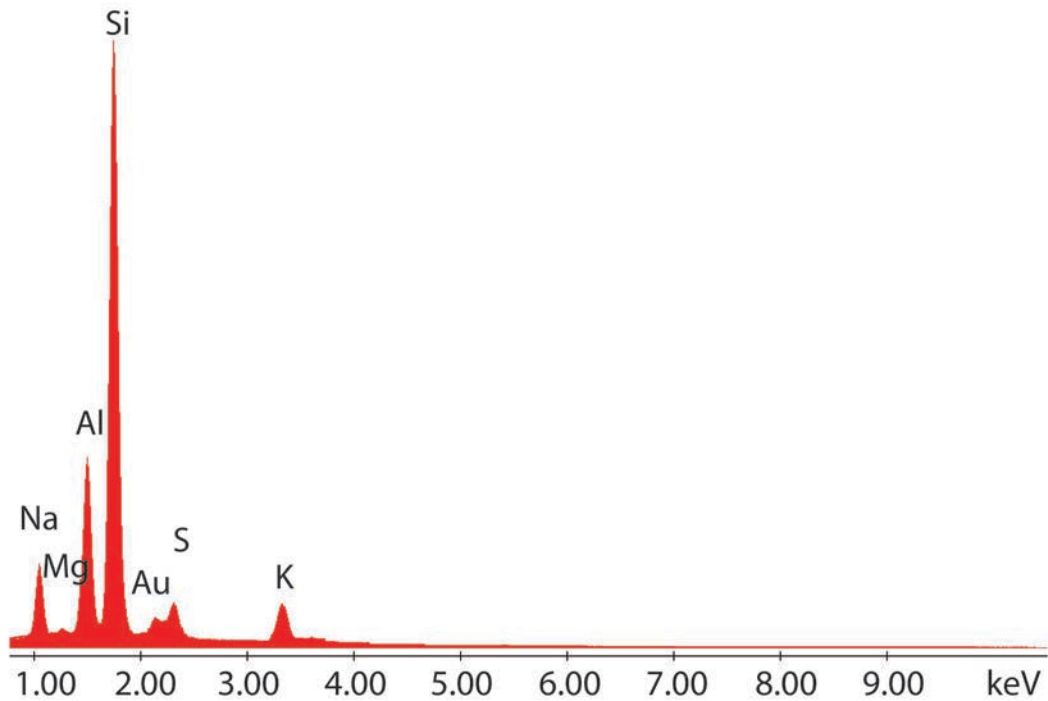
Mag = 280.00 K X

Date :24 Feb 2012

Sample ID = Ballirano









1 μ m


EHT = 5.00 kV

WD = 2.8 mm

Signal A = SE2

Mag = 15.00 K X

Date :24 Feb 2012

Sample ID = Ballirano

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applicata all'Ingegneria delle Impianti

