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2	Kangite, (Sc,Ti,Al,Zr,Mg,Ca,□) <sub>2</sub> O <sub>3</sub> , a new ultrarefractory scandia mineral
3	from the Allende meteorite: Synchrotron micro-Laue diffraction and
4	electron backscatter diffraction
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13	ABSTRACT
14	Kangite (IMA 2011-092), (Sc,Ti,Al,Zr,Mg,Ca,□) <sub>2</sub> O <sub>3</sub> , is a new scandia
15	mineral, occurring as micrometer-sized crystals with REE-rich perovskite and
16	spinel in a davisite-dominant ultrarefractory inclusion from the Allende CV3
17	carbonaceous chondrite. The phase was characterized by SEM, EBSD,
18	synchrotron micro-Laue diffraction, micro-Raman and EPMA. The mean
19	chemical composition of the type kangite is (wt%) $TiO_2$ 36.6, $Sc_2O_3$ 26.4, $ZrO_2$
20	11.3, Al <sub>2</sub> O <sub>3</sub> 7.0, Y <sub>2</sub> O <sub>3</sub> 5.4, CaO 3.9, MgO 3.14, Dy <sub>2</sub> O <sub>3</sub> 1.8, SiO <sub>2</sub> 1.7, V <sub>2</sub> O <sub>3</sub> 1.31,
21	Er <sub>2</sub> O <sub>3</sub> 0.92, FeO 0.8, Gd <sub>2</sub> O <sub>3</sub> 0.60, Ho <sub>2</sub> O <sub>3</sub> 0.40, Tb <sub>2</sub> O <sub>3</sub> 0.18, Cr <sub>2</sub> O <sub>3</sub> 0.09, ThO <sub>2</sub>
22	0.04, O -0.3, sum 101.28 with a corrected empirical formula calculated on the
23	basis of 3 O atoms of
24	$[(Sc_{0.54}Al_{0.16}Y_{0.07}V_{0.03}Gd_{0.01}Dy_{0.01}Er_{0.01})^{3+}{}_{\Sigma 0.83}(Ti_{0.66}Zr_{0.13})^{4+}{}_{\Sigma 0.79}(Mg_{0.11}Ca_{0.06}Y_{0.01}Y_{0.01}Y_{$
25	$\text{Fe}_{0.02}$ ) <sup>2+</sup> <sub><math>\Sigma 0.19</math></sub> $\square_{0.19}$ ] <sub><math>\Sigma 2.00</math></sub> O <sub>3</sub> . Synchrotron micro-Laue diffraction (i.e., an energy
26	scan by a high flux X-ray monochromatic beam) on one type domain at
27	submicrometer resolution revealed that kangite has a cation deficient $Ia\overline{3}$
28	bixbyite-type cubic structure. The cell parameters are $a = 9.842(1)$ Å, $V =$
29	953.3(1) Å <sup>3</sup> , Z = 16, which leads to a calculated density of 3.879 g/cm <sup>3</sup> . Kangite
30	is a new ultrarefractory mineral, likely originating through low temperature

31	oxidation of a Sc-, Ti <sup>3+</sup> -enriched high temperature condensate oxide dating to the
32	birth of the solar system.
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34	<b>Keywords</b> : Kangite, (Sc,Ti,Al,Zr,Mg,Ca, $\Box$ ) <sub>2</sub> O <sub>3</sub> , new ultrarefractory mineral, scandia,
35	Allende meteorite, CV3 carbonaceous chondrite, synchrotron micro-Laue diffraction, EBSD.
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38	INTRODUCTION
39	During a nanomineralogy investigation of a previously described inclusion, ACM-3,
40	in the Allende meteorite (Ma and Rossman 2009b), a new ultrarefractory Sc-, Ti-rich mineral
41	(Sc,Ti,Al,Zr,Mg,Ca,□) <sub>2</sub> O <sub>3</sub> , named "kangite," was discovered. Electron probe microanalysis
42	(EPMA), high-resolution scanning electron microscope (SEM), electron backscatter
43	diffraction (EBSD), synchrotron micro-Laue diffraction with subsequent energy scans, and
44	microRaman analyses were used to determine its composition, physical properties, and
45	structure and to characterize associated phases. Synthetic Sc <sub>2</sub> O <sub>3</sub> is well-known but synthetic
46	$(Sc,Ti,Al,Zr,Mg,Ca,\Box)_2O_3$ is unknown. Kangite is a new mineral species and a phase new to
47	meteoritics. In this paper, we describe the first occurrence of kangite in nature, as a new
48	ultrarefractory oxide, among the oldest solid materials in the solar system.
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50	MINERAL NAME AND TYPE MATERIAL
51	The mineral and the mineral name (kangite) have been approved by the Commission
52	on New Minerals, Nomenclature and Classification (CNMNC) of the International
53	Mineralogical Association (IMA 2011-092). The name is derived from "kang", the Chinese
54	word for the element scandium. Holotype material has been deposited under catalog USNM
55	7555 in the Smithsonian Institution's National Museum of Natural History, Washington DC,
56	USA. Section USNM 7555 also contains type davisite (IMA 2008-030; Ma and Rossman
57	2009b).
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59	OCCURRENCE
60	The Allende meteorite fell in and near Pueblito de Allende, Chihuahua, Mexico on
61	February 8, 1969 (Clarke et al. 1971). This CV3 carbonaceous chondrite is probably the most
62	highly studied meteorite and the results of these studies have greatly influenced current

63 thinking about processes, timing, and chemistry in the primitive solar nebula and in small 64 planetary bodies. The mineral kangite was found within one irregular ultrarefractory 65 inclusion, ACM-3, in one polished section (USNM 7555), prepared from a  $\sim$ 1 cm diameter Allende fragment (Caltech Meteorite Collection No. Allende12A). ACM-3 is about 130 µm 66 67 in diameter in the section plane and is surrounded by a matrix of mostly fine-grained olivine 68 and troilite. Kangite occurs interstitially, along with perovskite and spinel, within an 69 aggregate of the type davisite crystals (Figs. 1-2). All four kangite grains are in contact with 70 davisite and two are also in contact with perovskite. One was observed in contact with 71 spinel.

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## APPEARANCE, PHYSICAL, AND OPTICAL PROPERTIES

74 Kangite occurs as four irregular to subhedral grains, 1 to 4 um in size, which 75 constitute the type material (Fig. 2). In the optically thick section (150 µm in thickness), 76 kangite is opaque. Color, streak, luster, hardness, tenacity, cleavage, fracture, density and 77 refractive index were not determined because of the small grain size. Given the large extent 78 of solid solution in the natural material and its cation deficient stoichiometry, it is unlikely 79 that these properties are close to those of synthetic Sc<sub>2</sub>O<sub>3</sub>. The density, calculated from its crystal structure and the empirical formula, is  $3.879 \text{ g/cm}^3$ , which is similar to those of Sc<sub>2</sub>O<sub>3</sub> 80 (3.86) and panguite (3.75), but lower than those of TiO<sub>2</sub> (4.59) and Ti<sub>2</sub>O<sub>3</sub> (4.23). Kangite is 81 82 not cathodoluminescent under the electron beam in an SEM and we observed no crystal 83 forms or twinning. Visible light fluorescence can be excited by light of 514.5 nm; we did not 84 check for fluorescence using alternative wavelengths.

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# **CHEMICAL COMPOSITION**

87 Chemical analyses of kangite (on the two leftmost grains in Fig. 2) and associated 88 minerals were carried out using a JEOL 8200 electron microprobe (WDS: 15 kV; 5 nA and 89 25 nA; beam in focused mode) interfaced with the Probe for EPMA program from Probe 90 Software, Inc. Possible interferences on peak position and background position were checked 91 and corrected for all measured elements based on WDS scans and using the Probe for EPMA; 92 on-peak interferences of TiK $\alpha$  by Sc, VK $\alpha$  by Ti and Y, CrK $\alpha$  by V, ErL $\alpha$  by Tb, HoL $\alpha$  by 93 Gd, LaL $\alpha$  by Nd, CeL $\alpha$  by Ti and V, and NdL $\alpha$  by Ce were corrected. Standards for the 94 analysis were TiO<sub>2</sub> (Ti $K\alpha$ , OK $\alpha$ ), ScPO<sub>4</sub> (ScK $\alpha$ ), zircon (ZrL $\alpha$ ), anorthite (Ca $K\alpha$ , AlK $\alpha$ , 95 SiKa), YPO<sub>4</sub> (YLa), forsterite (MgKa), fayalite (FeKa), DyPO<sub>4</sub> (DyLa), ErPO<sub>4</sub> (ErLa), 96  $GdPO_4$  ( $GdL\alpha$ ),  $TbPO_4$  ( $TbL\alpha$ ),  $HoPO_4$  ( $HoL\alpha$ ),  $ThO_2$  ( $ThM\alpha$ ),  $V_2O_3$  ( $VK\alpha$ ),  $Cr_2O_3$  ( $CrK\alpha$ ),

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LaPO<sub>4</sub> (LaL $\alpha$ ), CePO<sub>4</sub> (CeL $\alpha$ ), NdPO<sub>4</sub> (NdL $\alpha$ ), SmPO<sub>4</sub> (SmL $\alpha$ ), and YbPO<sub>4</sub> (YbL $\alpha$ ). Quantitative elemental microanalyses were processed with the CITZAF correction procedure (Armstrong 1995) and analytical results are given in Table 1. No other element with an atomic number exceeding 4 was detected by WDS scans at 15 kV, 25 nA and 300 nA. In particular, none of the light rare earth elements (LREE) nor Yb was detected in kangite or in associated perovskite and davisite; detection limits for these elements at 99% confidence were 0.03 wt% for Sm and 0.04 wt% for La, Ce, Nd, and Yb. On-peak WDS X-ray mapping of Sc, Zr, Y, and Dy in ACM-3 (Fig. 3) was carried out at 15 kV and 100 nA, counting for 500 ms per pixel. The 250 x 320 pixel maps were collected using a 0.5  $\mu$ m step size and without a background correction. The empirical formula of type kangite as analyzed by EPMA, based on three oxygens,  $is \left[ (Sc_{0.53}Al_{0.19}Y_{0.07}V_{0.02}Gd_{0.01}Dy_{0.01}Er_{0.01})^{3+} \right]_{\Sigma 0.84} \left( Ti_{0.63}Zr_{0.13}Si_{0.04} \right)^{4+} \right]_{\Sigma 0.80}$  $(Mg_{0,11}Ca_{0,10}Fe_{0,02})^{2+}$   $\sum_{\Sigma_{0,23}} \Box_{0,13} \sum_{\Sigma_{2,00}} O_3$ , where the oxidation states of Ti, V, and Fe are assumed to be 4+, 3+, and 2+, respectively. Oxygen analyses by EPMA (at 15kV and 25 nA, using LDE1 crystal for  $OK\alpha$  measurement and synthetic TiO<sub>2</sub> as standard) in both kangite and perovskite support Ti<sup>4+</sup> (versus Ti<sup>3+</sup>) in kangite, as shown in Table 1 where leftover O in kangite (also perovskite) is roughly one standard deviation of the measured oxygen content when all Ti is assigned to 4+. If all Ti is assumed to be 3+, then leftover oxygen in kangite is 3.4 wt% (i.e., an order of magnitude higher than the analytical error). We cannot reject the possibility that minor Ti is trivalent as 8 % of the Ti in the kangite analysis in Table 1 would be  $Ti^{3+}$  if the measured oxygen content is used to calculate  $Ti^{3+}/Ti^{4+}$  (contamination from nearby davisite, which is Ti<sup>3+</sup>-enriched (Ma and Rossman 2009b), does not significantly affect this calculation). Nevertheless, EPMA analyses clearly indicate that Ti in kangite is dominantly 4+. We cannot be certain of the valence for Fe or V because the concentrations are too low. Some of the Fe in kangite could be ferric as alloys in Allende refractory inclusions were often oxidized to magnetite during low temperature metasomatism (Blum et al. 1989) and some of the V could be divalent as roughly half of the V in clinopyroxene and spinel from Ca-, Al-rich inclusions from Allende is  $V^{2+}$  (Simon et al. 2007; Pague et al. 2010). Changing the assigned valences for Fe and/or V has a negligible effect on the inferred crystallographic properties and no effect on the nomenclature of kangite or our conclusions regarding origin. The apparent Si content in kangite is likely derived from the surrounding davisite, because of the small grain size. If all Si and associated cations in the form of surrounding davisite are removed, kangite shows a formula of

 $\left[\left(Sc_{0.54}Al_{0.16}Y_{0.07}V_{0.03}Gd_{0.01}Dy_{0.01}Er_{0.01}\right)^{3+}\Sigma_{0.83}\left(Ti_{0.66}Zr_{0.13}\right)^{4+}\Sigma_{0.79}\left(Mg_{0.11}Ca_{0.06}Fe_{0.02}\right)^{2+}\Sigma_{0.19}\Box_{0.19}\right]$ 130  $]_{\Sigma 2.00}O_3$ . Kangite is much more Ti- and Sc-rich than previously reported zirconias and is 131 132 more Sc-rich than panguite (Fig. 4). Based on EPMA, kangite exhibits an ultrarefractory rare 133 earth element (REE) pattern in which most of the heavy REE (HREE) are greatly enriched 134 relative to the LREE and Yb is depleted relative to the other HREE (concentrations of La, Ce, 135 Nd, Sm, and Yb are all below EPMA detection limits, 0.03-0.04 wt%) (Fig. 5). The general formula for kangite is (Sc,Ti,Al,Zr,Mg,Ca,  $\Box$ )<sub>2</sub>O<sub>3</sub>, or (Sc,Ti,Al,Zr,Mg,Ca)<sub>18</sub>O<sub>3</sub>, where Sc<sup>3+</sup> is 136 137 the dominant trivalent component and the trivalent cations are dominant in the cation sites 138 without indications for cation ordering. The IMA-approved dominant-valency rule (Hatert 139 and Burke 2008) is applied here to designate this oxide as a new Sc mineral. The ideal end-140 member formula is Sc<sub>2</sub>O<sub>3</sub>. 141 Kangite occurs within an aggregate of 2-12  $\mu$ m diameter crystals of davisite, whose 142  $Sc_2O_3$  content ranges from as high as 17.66 wt% in portions of the inclusion near the spray of kangite and perovskite crystals, to as low as 8.05 wt% near the inclusion-matrix boundary 143 144 and in the two lower lobes of the inclusion (the lobe containing the "spinel" label in Fig. 1 145 and the smaller lobe above and to the left of it; see also the Sc X-ray map in Fig. 3). Davisite in the central region surrounding kangite and perovskite has a mean composition of 146  $Ca_{0.98}(Sc_{0.52}Ti^{3+}_{0.017}Mg_{0.14}Ti^{4+}_{0.09}Zr_{0.04}V^{3+}_{0.02}Y_{0.02}Al_{0.01}Fe^{2+}_{0.01})(Si_{1.00}Al_{1.00})O_6$ , where 147 partitioning of Ti<sup>3+</sup> and Ti<sup>4+</sup> is based on a stoichiometric formula unit containing 4.00 cations 148 149 and 6 oxygens. High Sc davisite is HREE enriched (LREE are below detection limits) but 150 concentrations are a factor of several to an order of magnitude lower than in perovskite and 151 kangite. Low Sc davisite is also lower in Zr and HREE than high Sc crystals, showing an average formula of  $Ca_{0.98}(Sc_{0.30}Ti^{3+}_{0.019}Mg_{0.31}Ti^{4+}_{0.08})$ 152  $Al_{0.07}Zr_{0.02}V_{0.01}^{3+}Fe_{0.01}^{2+}(Si_{1.20}Al_{0.80})O_6$ . If kangite and nearby davisite equilibrated with 153 154 each other (although this may not have been the case, as discussed below), the compositions 155 (Table 1; Fig. 3) imply that Zr, V, Y and the REE are strongly partitioned into kangite 156 relative to davisite and Ca and, to a lesser extent, Al in davisite relative to kangite; the Ti 157 oxide panguite displays similar partitioning relationships with associated davisite (Ma et al. 2012). Note that about two-thirds of the Ti in davisite is inferred to be trivalent while 158 coexisting kangite contains little if any Ti<sup>3+</sup>. One possible interpretation of this observation is 159 that kangite strongly preferred Ti<sup>4+</sup> and discriminated against Ti<sup>3+</sup> relative to davisite. 160 However, we argue below that kangite is the low temperature oxidation product of a high 161

temperature Ti<sup>3+</sup>-rich phase. Relative concentrations of cations were not significantly

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affected but the additional oxygen was accommodated through the transformation of Ti<sup>3+</sup> to 163 Ti<sup>4+</sup>. 164 165 Perovskite in the kangite type example (Figs. 1-2) is Y-rich (Table 1) but Zr-poor  $(\sim 0.1 \text{ wt }\%)$ , with a formula of  $(Ca_{0.82}Y_{0.08}Dy_{0.03}Gd_{0.02}Sc_{0.02}Fe_{0.01}Er_{0.01}V_{0.01})(Ti_{0.94}Al_{0.02})O_3$ . 166 167 As in kangite, the perovskite is HREE enriched with concentrations of LREE below EPMA 168 detection limits (Table 1; Fig. 5). Perovskite with similar REE abundances described by 169 Weber and Bischoff (1994) is relatively Zr-rich (ZrO<sub>2</sub> 4 wt%). Associated spinel (Fig. 2) is 170 Fe-rich (FeO 11.22 wt%) with 0.42 wt% TiO<sub>2</sub>, 0.38 wt% V<sub>2</sub>O<sub>3</sub>, and 0.30 wt% Cr<sub>2</sub>O<sub>3</sub> (Table 171 1), showing a formula of  $(Mg_{0.79}Fe_{0.24})(Al_{1.95}Ti_{0.01}V_{0.01}Cr_{0.01})O_4$ . We note that stoichiometry considerations require  $Fe^{2+} > Fe^{3+}$  for this phase (i.e., the spinel is hercynitic). The valence 172 state of Ti in the spinel, which can be either  $Ti^{3+}$  or  $Ti^{4+}$ -enriched, at least in type B1 173 174 inclusions (Paque et al. 2010), is unknown. Overall, V-Ti-Cr concentrations are higher than 175 for most reported spinels from meteoritic inclusions in carbonaceous chondrites (e.g., 176 Connolly and Burnett 1999; Connolly et al. 2003) but they are not diagnostic because 177 similarly enriched compositions, although rare, are occasionally encountered in wide variety 178 of inclusion types including fine-grained (Krot et al. 2004), compact type A (Srinivasan et al. 179 2000), fluffy type A (Allen et al. 1978), and type B1 inclusions (Wark and Lovering 1982; El 180 Goresy et al. 1985; Armstrong et al. 1987). 181 182 CRYSTALLOGRAPHY 183 184 EBSD AND SYNCHROTRON MICRO-LAUE DIFFRACTION 185 EBSD analyses at a submicrometer scale were performed on the vibration-polished 186 section using methods described in Ma and Rossman (2008a, 2009a). An HKL (now Oxford) 187 EBSD system on a Zeiss 1550VP scanning electron microscope was used for these 188 measurements and operated at 20 kV and 6 nA in a focused beam configuration with a 70° 189 tilted stage and variable pressure (25 Pa) mode, which allows the study of uncoated 190 specimens. EBSD patterns of the kangite grains were indexed against structures of synthetic 191 Sc<sub>2</sub>O<sub>3</sub> (Schleid and Meyer 1989), ScTiO<sub>3</sub> (Shafi et al. 2012), Sc<sub>2</sub>TiO<sub>5</sub> (Kolitsch & Tillmanns 192 2003), Ti<sub>2</sub>O<sub>3</sub> (Vincent et al. 1980), and perovskite CaTiO<sub>3</sub> (Liu & Liebermann 1993), and the minerals panguite  $(Ti^{4+}, Sc, Al, Mg, Zr, Ca)_{1,8}O_3$  (IMA 2010-057; Ma et al. 2012), and 193 194 tazheranite (Zr,Ti,Ca)O<sub>1.75</sub> (cubic zirconia; Rastsvetaeva et al. 1998). Among these, the 195 patterns can be indexed using a bixbyite-type structure (Sc<sub>2</sub>O<sub>3</sub>, ScTiO<sub>3</sub>), a bixbyite-type

196 related structure (panguite), and a fluorite-type structure (tazheranite). Since the bixbyite 197 structure can be derived from a fluorite-structured oxide through a systematic removal of 198 oxygens, it follows that the EBSD data suggest a fluorite- or defective fluorite-structure. 199 Synchrotron micro-Laue diffraction on one kangite domain was carried out at the 200 34ID-E undulator beamline of the Advanced Photon Source in the Argonne National Laboratory using a 250×400 nm<sup>2</sup> polychromatic X-ray beam and an array of Perkin-Elmer 201 202 amorphous Si area detectors, using methods described in detail by Ma et al. (2012) in our 203 study of the newly discovered Sc-bearing Ti-oxide panguite. An energy scan (energy range 204 16 to 27 keV, 1 eV steps) of the primary beam yielded a set of 16 reflections, which were 205 used for a reverse Monte Carlo optimization (Endeavour 1.4; Puetz et al. 1999) with the 206 Sc<sub>2</sub>O<sub>3</sub> C-type (bixbyite-type) structure as the initial configuration. The rMC optimization 207 converged to  $R_{\text{Bragg}} = 0.079$  for the cubic bixbyite-type structure (see Deposited Item). 208 Subsequently we used SHELXL (Sheldrick 2008) for refinement. Isotropic atomic 209 displacement  $(U_{iso})$  values, site occupancy factors (SOFs) of Sc1, Sc2, and x,y,z coordinates 210 for the cation and oxygen sites are given in Table 2. Refinement of the  $U_{iso}$  values, SOFs and fractional coordinates was conducted in sequential steps. 211

Kangite shows a cation-deficient  $Ia\overline{3}$  bixbyite-type structure with unit cell dimension 212 a = 9.842(1) Å, V = 953.3(1) Å<sup>3</sup>, Z = 16. The EBSD pattern (Fig. 6) is an excellent match to 213 214 the computed EBSD pattern obtained using the cell parameter of the cubic structure from the 215 synchrotron micro-diffraction analysis with resulting mean angular deviations as low as 0.35. 216 X-ray powder-diffraction data (Table 3, in Å for CuKa1, in Bragg-Brentano geometry) were 217 calculated using PowderCell (Kraus and Nolze 1996) version 2.4. Cation-cation distances in 218 kangite are around 2.6% larger than in synthetic Sc<sub>2</sub>O<sub>3</sub> (Schleid and Meyer 1989). Isotropic 219 thermal displacement factors of the 8a and the 24d cations are similar, 0.017 and 0.015 in 220 synthetic Sc<sub>2</sub>O<sub>3</sub> (Schleid and Meyer 1989) versus 0.012 and 0.009 in kangite (Table 2), 221 respectively. However, thermal displacement of oxygen is significantly larger in kangite than 222 in synthetic  $Sc_2O_3$ : 0.16 versus 0.019 Å<sup>-2</sup>. We relate this observation to another important difference between the structures of kangite and synthetic pure Sc<sub>2</sub>O<sub>3</sub>. In both kangite and 223 224 pure  $Sc_2O_3$ , cations residing at site 8a have six next neighbor anions, although cation-anion 225 distances are 5% shorter in kangite than in pure  $Sc_2O_3$ . In synthetic  $Sc_2O_3$  the cation on site 226 24d has six-fold coordination with Sc-O distances ranging from 2.08 to 2.17 Å, as expected 227 for bixbyite-type phases (Wyckoff 1960). In kangite, however, two apices of the 24d centered 228 polyhedron are clearly beyond Sc-O bond distances, leaving Sc with an apparent four-fold

229 coordination. The four shortest cation-O distances range from 1.8 to 2.2 Å with the other two 230 at 3.0 and 3.5 Å. Obviously, bond orbital distributions still require a 6 - 8 fold coordination of 231 the major cations (Sc, Ti, Zr, etc.). These unusual observations can be explained by the high 232 density of cation vacancies in kangite and the fact that site occupancies for the 8a and 24d 233 sites are quite different. The 24d site has only 60% occupancy whereas the 8a site has 80% 234 occupancy. It is therefore likely that cations in the 8a site attract O closer in unit cells for 235 which the 24d site is empty than in those for which 24d is occupied. In the averaged structure 236 sampled by the micro-Laue measurements, this results in an overall shift of the O-atom 237 position away from the 24d site leaving the latter with an apparent incomplete coordination. 238 This interpretation is supported by the unusually large thermal displacement factor of O, 239 which is almost a factor of ten larger than in pure  $Sc_2O_3$  (see above). In other words, this 240 large thermal displacement factor convolutes the static disorder of O atomic positions in 241 kangite. Such disorder can, in principle, be modeled by partial occupancies of multiple 48e 242 sites for O. However, such further refinement would require more unique reflections than are 243 available in our data set.

244 The structural relationships between kangite and panguite are very interesting because 245 of the similar compositions. Both phases are Ti-, Sc-rich oxides, whose compositions are well expressed using a pseudobrookite-type formula  $M_3O_5$ , but they reveal a bixbyite-type ( $M_2O_3$ ) 246 247 or related structure. Kangite, which contains more Sc and significantly higher HREE 248 contents, has a bixbyite-type structure, whereas panguite shows a reduction in symmetry from bixbyite-type  $Ia\overline{3}$  to Pbca. In bixbyite and kangite, the cations assume two Wykhoff sites, 8a 249 250 and 24d, whereas the anion resides on the general position 48e. The cubic bixbyite structure is related to the structure of panguite by a group-subgroup chain  $Ia\overline{3} \rightarrow Ibca \rightarrow Pbca$  where 251 252 all atoms occupy the general site 8c (Aroyo et al. 2006). The reduced crystal symmetry of 253 panguite is the result of lattice relaxation due to the large number of vacancies and large 254 concentration of cations whose ionic radii are smaller than expected for a bixbyite-type structure. 255 This has been discussed in detail in Ma et al. (2012). Kangite has similarly large cation deficiency 256 but a larger concentration of REE elements, all of which establish end-member oxides in the 257 bixbyite structure. It is this higher concentration of large cations that stabilizes the bixbyite 258 structure for kangite. Cooling rates during the formation process may also play a role. However, 259 this can only be assessed by experimental studies on synthetic equivalents of kangite and 260 panguite. 261

### 262 RAMAN SPECTROSCOPY

263 Raman spectroscopic microanalysis was carried out using a Renishaw M1000 micro-264 Raman spectrometer system on kangite domains of the sample in polished section, using the 265 methods described in Ma and Rossman (2008a, 2009a). Approximately 5 mW of 514.5 nm 266 laser illumination (at the sample) focused with a 100× objective lens provided satisfactory 267 spectra. The spot size was about 2 µm. The spectrum of kangite is characterized by intense 268 fluorescence features, similar to those of nearby perovskite grains, as shown in Figure 7. The dominant Raman feature of synthetic Sc<sub>2</sub>O<sub>3</sub> at 419 cm<sup>-1</sup> (Ubaldini and Carnasciali 2008) is 269 270 nearly invisible. The intense fluorescence features are due to significant amounts of 271 fluorescent *REE* elements in both kangite and perovskite (Table 1). Because features in the 272 Raman spectra arise from intense optical fluorescence due to the REE elements, there are no 273 identifiable Raman shifts and, hence, the spectra provide no information about the crystal 274 structure of the host phase. 275 276 DISCUSSION Kangite is a new Sc-, Ti-rich oxide,  $(Sc,Ti,Al,Zr,Mg,Ca,\Box)_2O_3$ , with a *Ia* $\overline{3}$  bixbvite-277 278 type structure, and a new member of the bixbyite group, which already contains bixbyite 279  $(Mn_2O_3)$ , avicennite  $(Tl_2O_3)$  and yttriaite-(Y)  $(Y_2O_3)$ . It is the first mineral with a bixbyite-280 type structure to be found in a meteorite. Kangite also joins a variety of Zr-, Y-, and Sc-rich 281 oxides and silicates, including allendeite, tazheranite, zirconolite, panguite, thortveitite, 282 lakargiite, eringaite, and davisite, that have been reported in refractory inclusions from 283 carbonaceous chondrites (Allen et al. 1980; Hinton et al. 1988; El Goresv et al. 2002; Ma and 284 Rossman 2008b, 2009b; Ma et al. 2009, 2011; Ma 2011; Ma 2012; Ma et al. 2012). Each of 285 these phases is a potential sensor of environment and kangite has the added advantage of 286 being heavy rare earth enriched and, therefore, potentially sensitive to condensation and 287 evaporative processes. The juxtaposition of kangite, which has little if any  $Ti^{3+}$ , within an aggregate of 288 289 davisite crystals whose Ti is largely trivalent, presents a challenge for understanding the 290 origin of kangite. There are three basic possibilities: (1) kangite formed or equilibrated in an 291 oxidizing environment separate from the reducing environment necessary to produce davisite and was later transported to the davisite (or vice versa); (2) davisite and kangite both form 292 under reducing conditions but kangite discriminates against Ti<sup>+3</sup> in much the same way that 293

294 perovskite, which is also present in this inclusion, does; (3) kangite originally formed as a

295 Ti<sup>3+</sup>-enriched phase that was incorporated into the davisite aggregate but these original grains 296 were later oxidized at low temperatures (too low to affect the davisite) to form kangite. 297 Phase equilibria place basic constraints on the stability of a mineral and the conditions 298 under which it may have formed. For kangite, the most relevant binary system under 299 oxidizing conditions is Sc<sub>2</sub>O<sub>3</sub>-TiO<sub>2</sub> but the phase relations of Sc<sub>2</sub>O<sub>3</sub>-TiO<sub>x</sub>-bearing systems in 300 general and of bixbyite-structured oxides within these systems in particular are poorly 301 constrained. The high temperature phase diagram of Magunov and Magunov (2002) for the 302 system  $Sc_2O_3$ -TiO<sub>2</sub> shows a solubility of only a few mole % TiO<sub>2</sub> in the terminal  $Sc_2O_3$  solid 303 solution (their data constrain it to < 10 mole%) with greater amounts of TiO<sub>2</sub> generating 304  $Sc_2TiO_5$  up to the melting point of that compound at 1347°C. Although they did not note it, 305 the compound  $Sc_4Ti_3O_{12}$  (e.g., Lyashenko et al. 2004) also appears to be stable in this system at high temperatures. There are also some constraints on phase stabilities for compositions 306 307 within the quaternary Sc<sub>2</sub>O<sub>3</sub>- TiO<sub>2</sub>-Y<sub>2</sub>O<sub>3</sub>-ZrO<sub>2</sub> (e.g., Tao and Irvine 2002) but the 308 experimentally examined compositions are far richer in Y+Zr ( $\geq 65$  mole % of the cations for 309 Tao and Irvine's study) than in kangite ( $\sim 10\%$ ; Fig. 4). Kangite has not been synthesized to 310 our knowledge but many trivalent end-member oxides are known to assume a bixbyite 311 structure, including  $Sc_2O_3$ ,  $Y_2O_3$ , trivalent oxides of the rare earths, and  $In_2O_3$ . Available data 312 for Sc<sub>2</sub>O<sub>3</sub>-TiO<sub>2</sub> imply that a bixbyite structured scandian oxide as TiO<sub>2</sub>-rich as kangite (molar 313  $Ti/Sc \sim 1.2$ ) is not a stable high temperature phase under oxidizing conditions relative to a 314  $Sc_2TiO_5$ - or  $Sc_4Ti_3O_{12}$ -bearing phase assemblage. This argues against scenario (1) given at 315 the beginning of this section and for a low temperature origin. If kangite is a high temperature 316 phase produced under oxidizing conditions, then one or more of the other cations present in 317 significant concentrations (i.e., Al, Zr, Mg, and/or Ca) must have stabilized the bixbyite 318 structure. It is important to consider the possible role of  $Ti^{3+}$  in the origin of kangite, both from 319

the perspective of whether or not kangite can accommodate  $Ti^{3+}$  (e.g., scenario 2) and what 320 the nature and stability of a  $Ti^{3+}$ -enriched precursor would have been (scenario 3). The 321 322 average ionic radius of trivalent cations in kangite is 0.70 Å and, if Si is removed from the 323 formula in the form of davisite (i.e., all Si in the kangite analysis is due to contamination of 324 the activation volumes for EPMA by the surrounding davisite), then the average ionic radius is 0.67 Å. Moreover, trivalent cations present in significant concentrations (Al, Sc, Y) have 325 ionic radii ranging from 0.53 to 0.89 Å. The ionic radius of  $Ti^{3+}$  is 0.67 Å, near the average, 326 327 which implies that there are no large size-related energetic penalties associated with the substitution of Ti<sup>3+</sup> into kangite. Thus, scenario (2) is unlikely. Kangite discriminated against 328

329  $Ti^{3+}$  because there was little or none to be had at the time it formed. Kangite formed (or equilibrated) in an oxidizing environment separate from the one that produced the davisite. 330 If kangite is the oxidized form of a  $Ti^{3+}$ -enriched precursor (scenario 3), then phase 331 332 relations in the system  $Sc_2O_3$ -Ti<sub>2</sub>O<sub>3</sub> are pertinent to the origin of the precursor but these are 333 even less well constrained than for Sc<sub>2</sub>O<sub>3</sub>-TiO<sub>2</sub>. Ti<sub>2</sub>O<sub>3</sub>, which takes on a corundum-type 334 structure (R3c), can be synthesized with up to 4 mole % Sc<sub>2</sub>O<sub>3</sub> (Chandrashekar et al. 1974) and the double oxide  $ScTi^{+3}O_3$ , which has a bixbyite-type structure, is readily synthesized at 335 336 high temperatures under reducing conditions (Reid and Sienko 1967). The solubility of Ti<sub>2</sub>O<sub>3</sub> 337 in a terminal  $Sc_2O_3$  solution is unknown but, given that bixbyite-type structures tend to be 338 mostly sensitive to ionic radius for a potential substituting cation, it seems likely that the 339 solubility is significantly greater than that of Sc in tistarite ( $Ti_2O_3$ ). Even these limited 340 observations place an important constraint on the origin of kangite. A kangite precursor 341 could, in principle have been an  $M_2O_3$  oxide but, given the low solubility of Sc in Ti<sub>2</sub>O<sub>3</sub>, it 342 was not tistarite. 343 Although there is a paucity of phase equilibrium data for kangite composition oxides and Ti<sup>3+</sup>-enriched analogs, there is a substantial literature concerning the solubility 344 345 relationships and energetics of solid solutions of bixbyite-type structured oxides. Systems 346 involving In<sub>2</sub>O<sub>3</sub> are the most heavily studied because of their use in energy efficient windows, 347 flat screen panel displays and an assortment of photovoltaics but there are enough data on

enough systems (e.g., Busker et al. 1999; Freeman et al. 2000; Saha et al. 2002; Mason et al.

2003; Levy et al. 2007; Hoel et al. 2010) to allow for the expression of some general

350 principles regarding the substitution of cations into a bixbyite structured end-member  $M_2O_3$ 

351 oxide such that the  $Ia\overline{3}$  space group is retained: (1) for an aliovalent substitution (i.e., a 3+

cation substituting for a 3+ cation), cations with ionic radii similar to those of the host are

energetically favored; (2) substitution of a single type of divalent or tetravalent cation for a

354 3+ host is compensated by oxygen interstitials or cation vacancies (for kangite this would 355 likely be via cation vacancies); (3) co-doping of 4+ and 2+ cations substantially enhances the 356 solubility of both cations relative to the solubility of either by itself; and (4) the solubility of 357 co-doped 4+ and 2+ cations is maximized when the average ionic radius of the two cations 358 approaches that of the host.

Consider a reduced precursor to kangite (scenario 3) in light of the above four principles. We begin with the premise that Si is a contaminant from davisite, although it is possible that Si is soluble in small amounts in kangite. As shown above, removing the Si by

subtracting davisite and renormalizing to a 3 oxygen basis leads to the kangite formula: 362  $\left[\left(\mathrm{Sc}_{0.54}\mathrm{Al}_{0.16}\mathrm{Y}_{0.07}\mathrm{V}_{0.03}\mathrm{Gd}_{0.01}\mathrm{Dy}_{0.01}\mathrm{Er}_{0.01}\right)^{3+}_{\Sigma_{0.83}}\left(\mathrm{Ti}_{0.66}\mathrm{Zr}_{0.13}\right)^{4+}_{\Sigma_{0.79}}\left(\mathrm{Mg}_{0.11}\mathrm{Ca}_{0.06}\mathrm{Fe}_{0.02}\right)^{2+}_{\Sigma_{0.19}}\mathrm{D}_{0.19}\right]$ 363 364  $\int_{\Sigma_2 00} O_3$ . This formula has slightly lower Al and Ca (and no Si) than the measured kangite 365 composition as given in Table 1 because these cations are in much greater abundance in davisite than in kangite. There are 0.79 quadrivalent cations in this formula unit but only 0.19 366 367 divalent cations, which means that there are far more 4+ cations than can be accommodated through co-doping of 2+ and 4+ cations. This is equivalent to saying that there are far more 368 Ti<sup>4+</sup> cations than can be accounted for through co-doping. As an end-member model, we 369 370 assume that all 4+ cations are co-doped with 2+ cations in the kangite precursor and that all 371 remaining Ti in the kangite precursor was trivalent. This yields a formula unit of  $[(Ti_{0.61}Sc_{0.53}Al_{0.19}Y_{0.06}V_{0.02}Gd_{0.01}Dy_{0.01}Er_{0.01})^{3+}z_{1.44}(Ti_{0.05}Zr_{0.14})^{4+}z_{0.19}(Mg_{0.11}Ca_{0.06}Fe_{0.02})^{2+}z_{0.19})$ 372  $]_{\Sigma 2.00}O_3$  for the kangite precursor. Note that this phase has an even higher Ti<sup>+3</sup>/Ti<sup>+4</sup> ratio (12) 373 374 than does davisite (2). Of course, intermediate formulae can be constructed in which some fraction of the Ti that is not co-doped with a divalent cation is accommodated as  $Ti^{+4}$  in the 375 precursor through cation defects or interstitial oxygen.  $Ti^{4+}$  (0.605 Å) has a significantly lower 376 ionic radius than does Sc (0.73 Å) or  $Ti^{3+}$  (0.67 Å) but the average ionic radius for a Ca: $Ti^{4+}$ 377 378 co-doping couple is higher (0.80 Å) than the average ionic radius for the trivalent cations 379 though still smaller than that of Y (0.89 Å), which is present in kangite in similar 380 concentrations. Perhaps Al, which has a much lower ionic radius (0.53 Å) than those of Sc or  $Ti^{3+}$ , partially stabilizes the structure for Ca: $Ti^{4+}$  co-doping by reducing the mean ionic radius 381 for the mineral. Connecting the proposed precursor composition to specific redox conditions 382 would require additional experimentation but we can say that Ti<sup>3+</sup> enriched precursors to 383 384 kangite are consistent with reducing conditions and that they may have formed in an 385 environment similar to or directly related to the one that produced the host davisite. The structure of the kangite precursor is uncertain but the fact that cations of  $Ti^{3+} \sim Sc^{3+}$  in the 386 387 davisite-corrected formula unit given above suggests the possibility that this may have been 388 an ScTiO<sub>3</sub> bixbyite- type phase with considerable solid solution towards various other 389 components. Alternatively, the inferred Ti:Sc ratio is fortuitous and the kangite precursor was 390 a terminal Sc<sub>2</sub>O<sub>3</sub> solid solution. Both phases have a bixbyite-type structure. The kangite 391 precursor, which may well have been a high temperature condensate, was incorporated into the host davisite aggregate but, at a presumably later time, the 0.6 cations of  $Ti^{3+}$  on a 3 392 oxygen basis, that was originally present, was oxidized to form 0.6 cations of  $Ti^{4+}$  and the 393 phase transformed into kangite. Shafi et al. (2012) showed that  $Ti^{3+}$  in  $ScTi^{3+}O_3$  is readily 394

395 oxidized at low temperatures to form a metastable intermediate with a stoichiometry of 396  $Sc_2Ti^{4+}2O_7$ , which has a defect fluorite structure. A similar process may have occurred with the kangite precursor but this phase, having only half the Ti<sup>3+</sup> present in the end-member 397 398 ScTiO<sub>3</sub> could only be oxidized to an overall  $M_3O_5$  (i.e., pseudobrookite-like) stoichiometry 399 and it therefore assumed or retained the bixbyite structure, as observed for kangite. The 400 oxidation process must have occurred at a sufficiently low temperature so that the kangite, so 401 produced, did not break down to form other, more stable oxides and davisite was unaffected. 402 We can place some constraints on the conditions for this process through a consideration of 403 CAIs in Allende. Based on alloy-oxide equilibria, the alloys in Allende CAIs were oxidized 404 at temperatures in the vicinity of  $\sim$ 500°C under conditions about a log unit more oxidizing 405 than the iron-wüstite buffer (e.g., Blum et al. 1989). During this event, clinopyroxene was 406 unaffected but iron diffused into the centers of large, 40-60 µm diameter, spinel grains from 407 the core region of an Allende type B inclusion, hundreds of um from the surface of the CAI 408 (Paque et al. 2007). Spinel in the kangite bearing inclusion described here is very iron rich 409 (11.2 wt % FeO) and it is likely that this Fe was introduced during the Allende metasomatic 410 event. Given the close proximity of kangite crystals ( $\leq 20 \,\mu m$  from the surface of the host 411 inclusion), the small grain size (2-12 µm) of the davisite (hence many grain boundaries that 412 could act as fast pathways for diffusion from the external medium outside the inclusion to the 413 surfaces of the kangite precursors), the small grain size of the kangite (1-4  $\mu$ m grain sizes but 414 not more than  $\sim 2 \,\mu m$  thick; see Fig. 2), and the highly ferroan compositions of nearby and 415 kangite-contacting spinel in the same inclusion, it is virtually certain that the kangite 416 precursor would have been exposed to these oxidizing metasomatic conditions. If so, the low concentration of Fe in kangite suggests that the solubility of Fe<sup>2+</sup> was very low compared to 417 418 spinel exposed to virtually identical conditions (cf., Table 1). This is consistent with the absence of exchangeable Ti<sup>4+</sup> from other phases during the alteration process that would have 419 been necessary to charge balance the substitution of  $Fe^{2+}$  into kangite. 420

421 We conclude that kangite most likely formed as the low-temperature oxidation 422 product of a Sc-,  $Ti^{3+}$ -enriched oxide that formed under reducing conditions, presumably as a 423 condensate and probably with a bixbyite-type structure. Note that this high temperature 424 precursor, were it to be found, would be a new mineral. It would not be kangite because 425 cations of  $Ti^{3+} > Sc$  and it would not be tistarite because it would have a different structure. 426 This reduced kangite precursor was incorporated into an aggregate of davisite crystals along 427 with small amounts of perovskite and spinel and it was later oxidized to form kangite. It is 428 possible that the oxidation event occurred prior to incorporation into the davisite aggregate 429 (i.e., the grains arrived as kangite crystals) but the most straightforward scenario is one in 430 which the same metasomatic event that introduced iron into spinels, including the ones shown 431 in Figures 1-2, and oxidized metal in Allende CAIs, also caused in situ oxidation of the 432 kangite precursor. If so, the composition of the kangite, except for oxygen and possibly Fe, 433 approximates that of the original precursor and, because there were  $\leq 0.6$  cations of Ti<sup>3+</sup> per 3 434 oxygens in the original crystals, oxidation yielded a roughly  $M_3O_5$  stoichiometry and a 435 bixbyite-type structure was retained.

436 Although we have emphasized the origin of kangite in this discussion, it is also important to place the proposed Ti<sup>3+</sup>-enriched precursor into a broader cosmochemical 437 context. Both kangite and perovskite in the inclusion described in this study are highly 438 439 enriched in HREE (Fig. 5) and this provides a key to understanding their origin. In general, 440 ultrarefractory inclusions are characterized by extraordinarily high HREE/LREE ratios and 441 negative Yb anomalies as shown in Figure 5. Only a handful of meteoritic inclusions with 442 ultrarefractory REE patterns have been described (e.g., Davis 1984; 1991; Simon et al. 1996; 443 El Goresy et al. 2002; Uchiyama et al. 2008) but Boynton (1975) proposed their existence 444 before any had been discovered based on thermodynamic modeling of high temperature 445 condensation. Moreover, the absence of isotopic mass fractionation in Ti and Mg in at least 446 one of these inclusions suggests that the ultrarefractory grains are condensates rather than 447 evaporative residues (Simon et al. 1996). Some meteoritic inclusions (described as Group II) 448 appear to have derived their REE budget via condensation from a reservoir that was depleted 449 in an ultrarefractory component. Group II inclusions are fairly common among CAIs and 450 especially common among fine-grained inclusions (Davis 1991; Hiyagon et al. 2011). 451 Ultrarefractory phases, including perovskite and the kangite precursor from this study, are 452 representatives of solid extracts from high temperature fractional condensation processes. 453 Although the occurrence of ultrarefractory material as original condensate grains is rare, its 454 effect is often observed through REE patterns of lower temperature condensates that are 455 missing an ultrarefractory component. 456 457 **ACKNOWLEDGEMENTS** 

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- 626 **Table 1**. Mean electron microprobe analytical results for type kangite, associated perovskite,
- 627 spinel, and host davisite.

			davisite next to kangite and	davisite near the	
Constituent	kangite	perovskite	perovskite	edge	spinel
wt%	n=7 <sup>a</sup>	n=6	n=6	n=2	n=6
TiO <sub>2</sub>	$36.6(4)^{b}$	51.2(3)	8.9(6)	9.2(7)	0.42(2)
Sc <sub>2</sub> O <sub>3</sub>	26.4(5)	0.8(2)	15(1)	9(1)	0.05(1)
ZrO <sub>2</sub>	11.3(3)	0.13(5)	2.1(5)	1.1(3)	b.d.
Al <sub>2</sub> O <sub>3</sub>	7.0(2)	0.84(9)	22(2)	19.1(5)	65.75(7)
$Y_2O_3$	5.4(1)	6.3(1)	0.8(1)	0.68(3)	b.d.
CaO	3.9(2)	31.3(2)	23.2(4)	23.70(2)	0.15(2)
MgO	3.14(6)	b.d. <sup>c</sup>	2(1)	5.4(5)	21.2(1)
SiO <sub>2</sub>	1.7(4)	b.d.	25(2)	31.1(7)	b.d.
V <sub>2</sub> O <sub>3</sub>	1.31(4)	0.64(4)	0.53(2)	0.52(4)	0.38(1)
FeO	0.8(6)	0.61(2)	0.40(7)	0.43(7)	11.22(7)
Cr <sub>2</sub> O <sub>3</sub>	0.09(2)	b.d.	b.d.	0.03(1)	0.30(1)
Dy <sub>2</sub> O <sub>3</sub>	1.8(1)	3.1(1)	0.30(7)	0.25(4)	n.a. <sup>d</sup>
Er <sub>2</sub> O <sub>3</sub>	0.92(5)	1.00(5)	0.10(5)	b.d.	n.a.
Gd <sub>2</sub> O <sub>3</sub>	0.60(4)	2.52(7)	0.18(9)	0.14(2)	n.a.
Ho <sub>2</sub> O <sub>3</sub>	0.40(2)	0.48(7)	b.d.	b.d.	n.a.
Tb <sub>2</sub> O <sub>3</sub>	0.18(7)	0.40(6)	b.d.	b.d.	n.a.
ThO <sub>2</sub>	0.04(2)	0.40(6)	b.d.	b.d.	n.a.
O <sup>e</sup>	-0.3(2)	-0.3(2)			
Total	101.28	99.42	100.51	100.65	99.47

629 <sup>a</sup>Number of analyses.

<sup>b</sup>Numbers in parentheses represent one standard deviation of the mean for the n analyses
 conducted on the phase.

<sup>632</sup> <sup>c</sup>Below detection. Detection limits at 99% confidence are Mg 0.09 wt%, Si 0.08 %, Cr 0.07

633 %, Zr 0.05%, Y 0.06%, Si 0.02%, Er 0.10%, Ho 0.09%, Tb 0.08%, Th 0.03%.

634 <sup>d</sup>Not analyzed.

<sup>e</sup>Difference between oxygen measured by EPMA and value obtained from elemental

636 concentrations of cations as determined by EPMA in which a single valence (listed in the left

hand column) has been assumed for each of the oxides. The quoted uncertainty reflects only

638 the analytical error on the oxygen analysis.

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	x/a	y/b	z/c	SOF	$U_{\rm iso}$
Sc1	0.25	0.25	0.25	0.767(2)	0.012(3)
Sc2	0.9784(2)	0	0.25	0.590(1)	0.0093(3)
0	0.3207(6)	0.227(5)	-0.44(5)	1	0.16(9)

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650 <b>Table 3</b> . Calculated X-ray powder diffraction data for kangite.
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Ì	h	k	l	<i>d</i> [A]	I <sub>rel</sub>	
	2	1	1	4.0190	16	
	2	2	2	2.8419	100	
	3	2	1	2.6311	1	
4	4	0	0	2.4612	10	
	4	1	1	2.3204	5	
	4	2	0	2.2013	1	
	3	3	2	2.0989	15	
	4	2	2	2.0095	2	
	4	1	3	1.9307	75	
	5	2	1	1.7974	3	
	4	4	0	1.7403	51	
	4	3	3	1.6883	3	
	4	4	2	1.6408	1	
	6	1	1	1.5970	4	
	6	0	2	1.5566	1	
	5	4	1	1.5191	29	
(	6	2	2	1.4841	23	
(	6	1	3	1.4515	4	
	4	4	4	1.4210	2	
4	5	4	3	1.3922	2	
(	6	0	4	1.3652	1	
	7	2	1	1.3397	4	
(	6	4	2	1.3155	1	
(	6	1	5	1.2503	1	
8	8	0	0	1.2306	4	
	8	1	1	1.2118	2	
8	8	2	0	1.1938	1	
(	6	5	3	1.1767	1	
8	8	2	2	1.1602	1	
8	8	3	1	1.1444	1	
(	6	6	2	1.1293	6	
	8	0	4	1.1007	2	
	8	3	3	1.0869	1	
	8	2	4	1.0739	1	
	6	5	5	1.0613	2	
	6	6	4	1.0492	1	
	7	4	5	1.0374	1	
	7	6	3	1.0151	2	
	8	4	4	1.0045	5	
	9	4	1	0.9942	1	
	8	5	3	0.9942	2	
	8	6	0	0.9842	2	
	7	7	2	0.9745	1	
	8	2	6	0.9651	4	
	9	3	4	0.9559	1	
10		2	2	0.9471	4	
	, 7	6	5	0.9384	1	
	8	5	5	0.9218	1	
(	5	5	5	0.7210	1	

# **Deposited Item**.

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2 6 8	92916.11	56643.00	1.77	0.575	0.97
4 4 1	) 52273.38	41034.36	0.97	0.489	0.86
0 6 1	) 27150.91	32576.42	0.93	0.436	0.84
0 0 1	55806.35	56775.83	0.17	0.576	0.98
4 4 8	168947.70	171334.06	0.14	1.000	1.00
					139, GooF 0.875 (S



- **Figure 1**. BSE image of ultrarefractory inclusion ACM-3 in section USNM 7555 from the
- Allende meteorite, where the type davisite (Ma and Rossman 2009b) and kangite (this study)
- 726 occur. Scale bar is  $10 \,\mu m$ .

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735 Figure 2. Enlarged BSE images of ACM-3 with contrasts optimized to show kangite microcrystals (type material) with perovskite and spinel in host davisite. The cross mark 736 737 shown in (a) indicates where the EBSD pattern (shown in Fig. 6) and the synchrotron micro-Laue diffraction data were collected. Scale bar is  $1 \, \mu m$ . 738

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Figure 3. Sc, Zr, Y, and Dy X-ray maps of ultrarefractory inclusion ACM-3. Scale bar is 50 μm.

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752 753 Figure 4. Compositions in mole % of meteoritic Sc-, Zr-enriched oxides in terms of the ternary  $Sc_2O_3$  - TiO<sub>2</sub>\* (where all Ti is computed as TiO<sub>2</sub>) - ZrO<sub>2</sub>. These three oxides account 754 for 80-97% of cations in the zirconias and 70-74% of the cations in kangite and panguite, the 755 balance being mostly Al, Ca, Mg, and Y. Data are taken from Noonan et al. (1977), Lovering 756 et al. (1979), Hinton et al. (1988), Weber and Bischoff (1994), Ma et al. (2012), and this 757 758 study.

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Figure 5. Abundance of REE and Y relative to CI chondrites (Anders and Grevesse 1989) in 766 kangite and perovskite from ACM-3. Error bars for analyses from this study are standard 767 768 deviations on multiple analyses where larger than the symbols. Downward pointing arrows 769 indicate maximum concentrations set at the detection limit for our EPMA analyses. Also 770 shown are the REE patterns for an ultrarefractory cluster of oxides in Ornans inclusion 1105-771 42 (Davis 1991) and inclusion HIB-11 from Murchison, which consists of perovskite and 772 spinel in a davisite-rich inclusion (Simon et al. 1996). Error bars for literature analyses are 773 not shown.

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Figure 6. (a) EBSD pattern of the kangite crystal marked with a cross in Figure 2a; (b) the same pattern indexed with the cubic  $Ia\overline{3}$  structure obtained by synchrotron micro-diffraction. 787





791 Figure 7. Raman spectra of kangite and perovskite in ACM-3, and synthetic Sc<sub>2</sub>O<sub>3</sub>. The 792 peaks for kangite and perovskite are dominated by optical fluorescence features due to high 793 concentrations of REEs in the crystals so that no Raman shifts can be distinguished.