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3	The structure of (Ca,Co)CoSi ₂ O ₆ pyroxenes and the Ca- M^{2+} substitution in
4	$(Ca, M^{2+})M^{2+}Si_2O_6$ pyroxenes $(M^{2+} = Co, Fe, Mg)$
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7	Luciana Mantovani ¹ , Mario Tribaudino ¹ , Francesco Mezzadri ² , Gianluca Calestani ² ,
8	Geoffrey Bromiley ³
9 10 11 12	¹ Dipartimento di Fisica e Scienze della Terra "Macedonio Melloni", Parco Area delle Scienze 157/A, 43124 Parma, Italy
13 14	² Dipartimento di Chimica, Parco Area delle Scienze 17/A, 43124 Parma, Italy
15 16 17 18 19	3 School of GeoSciences, University of Edinburgh, Grant Institute, West Mains Road, Edinburgh EH9 3JW, UK
20 21	Abstract
22	The crystal structure of three $C2/c$ clinopyroxenes with composition (Ca _{0.8} Co _{0.2})CoSi ₂ O ₆ ,
23	$(Ca_{0.6}Co_{0.4})CoSi_2O_6$ and $(Ca_{0.4}Co_{0.6})CoSi_2O_6$ was refined down to $R_{4\sigma}$ = 2.6% by single crystal X-
24	ray diffraction. The crystals were synthesized at $P = 3$ GPa by cooling from 1500 to 1200°C in a
25	piston cylinder apparatus. At the end of the refinement cycles electron density residuals (up to 2.1 e
26) were observed close to the M2 site and related to the site splitting of Ca and Co in the M2
27	polyhedron in the two subsites M2 and M2'. Split refinement significantly improved the agreement
28	factor and decreased the uncertainty in the atomic coordinates. Similar features were found in (Ca,
29	Mg)MgSi ₂ O ₆ and (Ca, Fe)FeSi ₂ O ₆ intermediate pyroxenes.
30	The average structural changes related to the cation substitution at the M2 site in (Ca, Co)CoSi ₂ O ₆ ,
31	(Ca, Mg)MgSi ₂ O ₆ and (Ca, Fe)FeSi ₂ O ₆ pyroxenes are similar: the T tetrahedron becomes more
32	regular, the difference between M1-O bond lengths increases, and the M2-O3 bond lengths with the
33	furthermost O3 oxygens become longer. The changes in the M2-O bond distances are, however, not
34	linear, and they are higher for more increased substitution. The largest structural deformation occurs
35	on the (010) plane, with higher deformation at about 60° from the c axis for any composition. The
36	orientation of the deformation ellipsoid is most related to a shift in tetrahedral chains. The scalar
37	deformation for the cation substitution, ε_s , is linearly related to the cation radius of the average M2

site (IR^{M2}), i.e. the deformation is higher as the cation size decreases, following the equation: ε_s =-38 0.0072(12)IR^{M2}+0.0082(13), R²=0.75. Increasing deformation with cation substitution is supported 39 40 as the major limiting factor for solid solution. 41 The displacement parameters for unsplit M2, O2 and O3 atoms increase up to the intermediate 42 composition, indicating a local configuration for the M2 polyhedron centered by Ca and Co. 43 However no significant change in U_{eq} of the O3 atom is observed up to 20% substitution of the 44 smaller cation in the M2 site. Comparison with Raman spectral data suggests that local chain 45 structural configurations occur only for the substitution of the smaller cation in the M2 site higher 46 than 20%, and that the substitution mechanism is different for C2/c clinopyroxenes with lower and 47 higher Ca content. 48

Keywords: CaCoSi₂O₆; single crystal X-ray diffraction; high pressure single crystal synthesis;
pyroxene.

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Introduction

Pyroxenes are chain silicates and are among the major constituents of the lower crust and the upper mantle. Their structural and crystal chemical behavior provides a simple model to test relations between structure and composition within a silicate family. Moreover it was recently found that pyroxene is a new class of ferroics (Jodlauk et al. 2007), which has now opened new interest in understanding the structural details of pyroxenes (Streltsov and Khomskii 2008, Redhammer et al. 2008, 2009, 2011).

59 The general formula of pyroxenes is $M2M1Si_2O_6$. In the holotypic structure of the 60 pyroxene diopside (CaMgSi₂O₆) Ca occupies an 8-fold coordinated M2 polyhedron, and Mg a 61 slightly distorted M1 octahedron. The Mg for Ca substitution in the M2 site of diopside is found in 62 natural pyroxenes, but, due to the different cation radius of Ca and Mg, it is limited by a wide and 63 asymmetric miscibility gap (Gasparik 1990). Only at high pressure and temperature, over a narrow 64 hypersolvus field, extended solid solution is possible (Newton et al. 1979). The iron equivalent of 65 diopside is hedenbergite, CaFeSi₂O₆. Again, replacement of Fe for Ca is possible at high pressure, 66 but limited by a miscibility gap, albeit more narrow (Lindsley and Munoz 1969, Ohashi et al. 1975). 67 The solid solutions between Ca, Mg and Fe represent almost completely the composition of the 68 most abundant "quadrilateral" pyroxenes.

The detailed structure changes involved in the above substitutions were investigated in several papers on the (Ca, Mg)MgSi₂O₆ and (Ca, Fe)FeSi₂O₆ intermediate pyroxenes (hereafter respectively Ca-Mg and Ca-Fe pyroxenes) (Ohashi and Finger 1976, Bruno et al. 1982, Tribaudino 72 et al. 1989, Tribaudino et al. 2002, Tribaudino and Nestola 2002, Nestola and Tribaudino 2003, 73 Tribaudino et al. 2005a for Ca-Mg pyroxenes, Burnham 1966, Ohashi et al. 1975, Nestola et al. 74 2007 for Ca-Fe pyroxenes). It was found that the structure of the intermediate pyroxenes is different 75 from a linear combination of the end members as: 1) changes in structure with composition are not 76 linear; 2) a phase transition occurs between Ca-rich C2/c to Ca poor $P2_1/c$ pyroxenes; 3) 77 displacement parameters of several atoms (M2, O2 and O3) are significantly larger in intermediate 78 solid solutions, due to positional disorder, whereas in other atoms they are not (Ohashi et al. 1975, 79 Tribaudino et al. 1989, Tribaudino and Nestola 2002, Nestola and Tribaudino 2003).

80 In spite of the wealth of experimental data, a model relating the different cation substitution 81 in the M2 site, the structural deformation and the extension of the solid solution was not provided.

Here the results of single crystal refinements on three (Ca, Co)CoSi₂O₆ clinopyroxene solid solutions (hereafter Ca-Co) representing the Ca-Co analogues of the Ca-Fe and Ca-Mg ones will be reported. The structural changes in Ca-Co pyroxenes will be examined and compared with those occurring in Ca-Mg and Ca-Fe pyroxenes. Local and average structures will be discussed aiming to a comprehensive model of solid solutions in pyroxenes. In broader sense this paper aims to model the structural features that limit the solid solution in quadrilateral monoclinic pyroxenes, which are the basis for two pyroxene thermometry (Lindsley 1983)

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Experimental

92 Synthesis

93 Three samples of pyroxene with composition (Ca_{0.8}Co_{0.2})CoSi₂O₆, (Ca_{0.6}Co_{0.4})CoSi₂O₆ and 94 (Ca_{0.4}Co_{0.6})CoSi₂O₆ (hereafter Ca_{0.8}Co_{0.2}, Ca_{0.6}Co_{0.4} and Ca_{0.4}Co_{0.6} respectively) were synthesized using as starting material a stoichiometric mixture of Co₃O₄, CaCO₃ and SiO₂. The mixtures were 95 previously annealed in an electric furnace for 12 hours at $T = 1000^{\circ}C$ and room pressure. 96 Decarbonation and reduction of the Co^{3+} in Co_3O_4 to Co^{2+} were obtained during the synthesis. An 97 98 assemblage of cristobalite, Co-pyroxene, Co-olivine and Co-akermanite was obtained, with 99 cristobalite and Co-akermanite as metastable phases. This assemblage, finely ground, was put in a 5 100 mm inner diameter, 10 mm long, Pt capsule with 20 wt% distilled water to aid crystal growth, and 101 welded shut. The capsules were placed into a 0.5 inch pyrex-talc piston-cylinder assembly 102 containing an internal graphite furnace. The experiments were run at P = 3 GPa; the sample was first heated at 1500°C for 30 minutes and then cooled from 1500°C to 1200°C for Ca_{0.8}Co_{0.2} and 103 104 Ca_{0.6}Co_{0.4} and to 1350°C for Ca_{0.4}Co_{0.6}, with subsequent annealing for 6 hours. The higher temperature of annealing of sample Ca_{0.4}Co_{0.6} was chosen in order to avoid exsolution. The 105

temperature was measured with a Pt-Pt10%Rh thermocouple (for more details see Bromiley et al. 2004). An assemblage of Co-pyroxene crystals sized up to 600 µm was obtained, among which those examined for single crystal XRD were selected by sharp optical extinction. No glass was found in any sample after the synthesis. XRD powder diffraction and SEM-EDS characterization confirmed that pyroxene is the only phase in all samples, with stoichiometric compositions.

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X-ray single crystal refinement

113 Single crystal X-ray diffraction was performed using a Smart-Apex CCD device, measuring 114 full reflection sphere up to $2\theta_{max} = 64^{\circ}$. The intensity data were corrected for absorption using the 115 SADABS program (Sheldrick 1996). The C2/c space group was found for all the samples by the 116 analysis of reflection extinctions and intensities: the intensities of h+k odd reflections, which would 117 indicate switch to the P lattice were always lower than 3σ . Weighted structural anisotropic 118 refinements were performed using the SHELX-97 program (Sheldrick 1997), within the WinGX 119 suite (Farrugia 1999) and the coordinates of CaCoSi₂O₆ by Ghose et al. (1987) as starting model. 120 Full occupancy of Co and Si in the M1 and T sites was confirmed by site refinement. In the M2 site 121 mixed occupancy of Ca and Co was present; therefore the site occupancy of the M2 for each 122 composition was determined from the EDS analysis, ascribing all Ca in the M2 site, and filling the 123 remaining with Co.

124 After the first cycles of full matrix refinements significant positive and negative peaks in 125 the Fourier difference map were found, located along the diad axis at ≈ 0.7 Å from the M2 site. In 126 the Ca_{0.8}Co_{0.2} and Ca_{0.6}Co_{0.4} samples the positive peaks of 2.1 and 1.9 electrons, respectively, are 127 located at 0.21 y/b. This position matches that of the residual found in C2/c clinopyroxenes when 128 Fe²⁺ and/or Mg exchange for Ca (Bruno et al. 1982; Rossi et al. 1987; Tribaudino et al. 1989, Boffa 129 Ballaran et al. 1998). Similar residuals were found by Tabira et al. (1992) in a study on C2/c130 clinopyroxenes in the system $CaMgSi_2O_6$ (Di)– $CaCoSi_2O_6$ (CaCoPx), and related to a small content 131 of cobalt in the M2 site, likely in excess of the CaMgSi₂O₆–CaCoSi₂O₆ stoichiometry. In contrast, 132 in $Ca_{0.4}Co_{0.6}$ at the same position there is a negative peak of -1.9 electrons, whereas the positive 133 peak of 1.2 electrons is located at 0.33 y/b. Electron density maps near the M2 site show in our 134 samples an asymmetric distribution along the diad axis (Fig. 1); the asymmetry is inverted between 135 $Ca_{0.6}Co_{0.4}$ and $Ca_{0.4}Co_{0.6}$.

The interpretation provided by previous works (Rossi et al., 1987; Tribaudino and Nestola, 2002) on the observed residuals can also apply to ours: Ca and the smaller cation in the M2 cavity split in two different positions along the diad axis, , respectively the M2 and M2' subsites; the observed residuals occur when split electron densities are refined as a single atom. If the cation smaller than Ca is present in minor amounts the asymmetric electron distribution is most evident; if Ca and the smaller cation are present at similar amounts, so that the electron distribution of the two cations is similar, a large apparent displacement parameter is found, although, as shown in Fig. 1, some asymmetry in the Fourier map is still present. Coherently, the size of the displacement parameters of the M2 site is almost doubled between CaCoSi₂O₆ and Ca_{0.6}Co_{0.4} (Fig. 2) but it is similar between Ca_{0.6}Co_{0.4} and Ca_{0.4}Co_{0.6} (Table 2).

146 We performed then a split refinement of the M2 site, with the new M2 site occupied by Ca 147 and the M2' subsite by Co. As indicated by Rossi et al. (1987), Ca was located in a position at y/b 148 0.30, close to the center of the M2 polyhedron. This is close to the position of the M2 site after the 149 unsplit refinements in $Ca_{0.8}Co_{0.2}$ and $Ca_{0.6}Co_{0.4}$, and also to the positive maximum at y/b 0.33 for 150 $Ca_{0.4}Co_{0.6}$. The Co in M2' was located at the position of the maximum at y/b 0.21 in $Ca_{0.8}Co_{0.2}$ and 151 Ca_{0.6}Co_{0.4}, and at the position of the M2 site in Ca_{0.4}Co_{0.6}. This was done because the higher 152 electron contribution of Co in Ca_{0.4}Co_{0.6}, indicates that the refined M2 position in Ca_{0.4}Co_{0.6} mostly 153 represents that of Co. After few refinement cycles, allowing Ca and Co to vary independently their 154 coordinate along the diad axis, two distinct atomic positions at y/b 0.30 and 0.26 were found, 155 occupied respectively by Ca and Co. In agreement with Rossi et al. (1987), we will refer to the Ca 156 site as M2 and to the Co one as M2'. The refinement was done with Ca anisotropic and Co isotropic 157 in $Ca_{0.8}Co_{0.2}$, and with Ca and Co both anisotropic in $Ca_{0.6}Co_{0.4}$ and $Ca_{0.4}Co_{0.6}$. A significant 158 decrease in the agreement factor and in the uncertainty of other atomic positions was observed; 159 moreover residuals became featureless. Although this procedure gave better phasing and therefore 160 more reliable positions of the other atoms, the M2 and M2' peaks after the split refinements are at a 161 distance between 0.3 and 0.4 Å, lower than the resolution of the present refinement data (0.67 Å), 162 and the reliability of the M2 and M2' split positions is lower than that of the unsplit M2.

Tables from 1 to 4 report refinement parameters, structural coordinates, atomic distances and equivalent displacement factors. For the M2-M2' sites the M2 position and M2-O bond distances before the split refinement were also reported, whereas for other atoms only those obtained after the split refinement model. CIF files relevant to the best refinements in the M2 split and unsplit models for each composition are deposited online.

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170 Average structure

171 **M2 polyhedron**. The M2 is a distorted 8-fold coordinated polyhedron (Fig. 3; atomic 172 crystallographic nomenclature as in Burnham et al. 1967). On one side the M2 cation is linked to 173 four O3 oxygens, namely those bridging the tetrahedra within the silicate chains. On the other side

Results

174 two O1 oxygens are shared with the M1 polyhedron, while in between there are the O2 atoms. In 175 C2/c pyroxenes the M2 cation is constrained by symmetry to lie on the diad axis, but can move 176 along the axis without breaking the symmetry; the M2-O2 bonds are almost normal to the diad axis. 177 The Co for Ca substitution into the M2 site increases the average distance with the O3 chain 178 bridging oxygen and decreases that with O1 and O2 (Fig. 4), changing the average coordination of 179 M2 towards a 4+2 one like it was found in augitic pyroxenes (Domeneghetti et al. 2005). The 180 polyhedral volume decreases also. The distances with O3 and O1 change for the shift of the average 181 M2 position along the diad axis, towards the O1 oxygen. The shift of the average M2 site is by as 182 much as 0.268 Å between CaCoSi₂O₆ and Ca_{0.4}Co_{0.6} towards the O1 oxygen, and the bond distances 183 change likewise. The M2 shift occurs while the size of the polyhedron along the diad axis does not 184 decrease: the distance between the glide plane at 0.5 y/b, which corresponds to the average position 185 of the O3 oxygen, and the position of the O1 along the b axis (y/b of the O1, being the two O1 186 oxygen symmetry equivalent, and fixed to the same y/b coordinate by the diad axis in C2/c187 symmetry) shows only the negligible decrease of 0.007 Å between $CaCoSi_2O_6$ and $Ca_{0.4}Co_{0.6}$.

188 The average bond lengths represent an average situation: from the results of the refinements 189 Co and Ca are split in the two subsites M2 and M2' into the M2 polyhedron, each with its own 190 bond distances, but both lying on the diad axis. As shown in Fig. 5, M2-O and M2'-O distances 191 change little with composition, with the exception of the M2-O2 and M2'-O2 distances that 192 decrease both by the same extent. The M2 configuration is always eight fold, but in the M2' the O3 193 distances are out of coordination, and a strongly asymmetric fourfold like configuration is found 194 (Rossi et al. 1987). The positions and the y/b coordinate of the M2 and M2' site, at approximately 195 y/b 0.30 and 0.26, correspond to those of the M2 site in diopside and clinoenstatite for the Ca-Mg 196 (Cameron et al. 1973, Ohashi and Finger 1976), or in hedenbergite and clinoferrosilite for the Ca-197 Mg pyroxenes (respectively Nestola et al. 2007 and Burnham 1966), and can be here interpreted as 198 local configurations for the Ca (M2) and Co (M2') atoms.

199 Changes in the position of the M2 cation along the diad axis do not account for the decrease 200 in the M2-O2 distance, being the bond almost normal to the axis. An approach of the O2 atom 201 towards the M2 site takes place (Table 4), and is related to a tetrahedral tilting, as it has been 202 observed in diopside-enstatite pyroxenes (Bruno et al. 1982). The decrease in the M2-O2 distance accounts for the significant decrease in the M2 polyhedral volume, by 1.3 Å³ (5 %) between 203 204 CaCoSi₂O₆ and Ca_{0.4}Co_{0.6}: a linear change between M2-O2 and volume is observed in Fig. 6. To 205 note, volume changes less in the first part of the series, but more with higher Co content, and the 206 same trend, albeit shifted by the difference in end members occurs in Ca-Mg and Ca-Fe pyroxenes. 207 Also the changes in the M2-O distances are very similar in the three series, dictated solely by the changes in the average ionic radius and position, apart from the shorter M2-O3 distance, where significant differences exist already in the Ca rich end members and are increased by substitution of the smaller cation (Fig. 7).

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M1 polyhedron. In the Ca-Co series the M1 site is always occupied by Co. As expected, the volume and the average bond lengths change little within the series, although in $Ca_{0.4}Co_{0.6}$ the M1-O average bond length and M1 polyhedral volume are significantly higher. Individual bond lengths and angles show instead a continuous change: the M1-O1A2 distance decreases with Ca, down to the value of the shorter M1-O2 distance, whereas the M1-O1A1, which is the longest M1-O distance, shows a further increase (Fig. 3).

218 The changes in M1-O bond distances are interpreted as a secondary effect of the shift of the 219 cation in the M2 site towards the O1A1, B1 atoms. As shown in Fig. 8, when Ca is substituted by a 220 smaller cation, the difference between the longer and shorter M1-O bonds increases at the same rate 221 in Ca-Co, Ca-Mg and Ca-Fe pyroxenes. The shift of the M2 average position along the diad axis 222 affects the distance of the M2 to the three M1 cations whose polyhedron shares an edge with the 223 M2 polyhedron (Figs. 3 and 4). One of the above M1 cations faces M2 along the diad axis, at a 224 distance of about 3.5 Å, the two others, M1(1) in Burnham et al. (1967) nomenclature, are 225 symmetry equivalent at a distance of about 3.2 Å with M2 (Fig. 3). As a consequence of the 226 approach of the average M2 position with decreasing Ca, the O1A1-O2D2 and O1B1-O2C2 edges 227 shared with the M2 polyhedron by the symmetry equivalent M1(1) decrease their length, and the 228 O1 atoms shifts towards the O2. The shift of the O1A1 and O1B1 towards O2D2 and O2C2 opens 229 the edge O1A1-O1B1, shared between M2 and the M1 polyhedron along the diad axis, from 2.79 in 230 $CaCoSi_2O_6$ to 2.84 Å in $Ca_{0.4}Co_{0.6}$ (Fig. 3), causing lower electrostatic shielding between M2 and 231 M1 along the diad axis, when the two cations approach with decreasing Ca. As a consequence the 232 M1 cation moves off M2 along the diad axis, increasing the bond distance with the O1 atoms. The 233 O1A1-Co-O1B1 angle maintains its value of 81° throughout the series, significantly deviated from 234 the expected 90° of a regular octahedron due to the combined shift of Co in M1 away from the M2 235 and the increase of the O1A1-O1B1 shared edge.

Furthermore, angular distortion changes little up to $Ca_{0.6}Co_{0.4}$, but $Ca_{0.4}Co_{0.6}$ is significantly more distorted. The changes in the O-M1-O angles with composition are continuous and small, apart from the two O1-M1-O2 couples that change by 5°, crossing the value of 90° at $Ca_{0.6}Co_{0.4}$ (Table 4).

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T polyhedron. As the M1 octahedron, the tetrahedral site, which is always filled with Si, is also affected by cation substitution in the M2 polyhedron. The polyhedral volume does not change, but the average bond distances decrease between $CaCoSi_2O_6$ and $Ca_{0.4}Co_{0.6}$. This is due to a decrease in bond lengths of Si with the oxygen shared by two tetrahedra, i.e the T-O3 bridging, while the distances with other oxygen (T-O non bridging) increase. Also the tetrahedral angle with the two bridging oxygen becomes closer to the 109.4° ideal value with a strong increase in the O3-M1-O3 angle (from 104.3 to 106.9) and a decreased tetrahedral distortion (Fig. 3).

The decrease in the difference between bridging and non bridging oxygen is related to the cation substitution in the M2 site (Fig. 9). The increased average M2-O3 bond distance decreases the nominal charge on the O3 atom, whereas the electrostatic contribution of the M2 cation to O2 and O1A increases. The lower contribution to the bridging O3 from the M2 site is compensated by a decrease in the distance of the bridging oxygen with tetrahedral Si. Also the O3-O3 distance, which is a shared edge with the M2 polyhedron, increases, and as a consequence the polyhedral angle opens.

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Cell parameters and structure. The substitution of a smaller cation like Co for the larger Ca decreases, as expected, the size of the unit cell. The decrease is comparable to that of the sum of the volumes of the 4 eight coordinated M2 polyhedra in the unit cell (-5.4 vs -5.1 Å³) for the cell parameters between CaCoSi₂O₆ and Ca_{0.4}Co_{0.6}. The cell volume decreases mostly as a consequence of contraction along a* (d₁₀₀), whereas the b axis increases its size, and the c axis is unchanged. The present data do not show a deviation from the linear behavior in volume, but the axial changes are not linear (Fig. 10).

263 Along the b axis the increase is by about 0.3 Å. As discussed in Tribaudino et al. (2005b), 264 the length of the b axis in $C^{2/c}$ pyroxenes is determined by the sum of the contributions along the 265 M2 polyhedron, and the M1 octahedral and T tetrahedral chains (Fig. 11). The contribution of the 266 M2 polyhedron is determined by the distance along the diad axis between the O1 atoms and the 267 glide plane at 0.5 y/b, i.e the average y/b of the O3 oxygen (Fig. 11). It changes little along the 268 series: as discussed above the cation substitution in the M2 site occurs with little change in the 269 positions of the surrounding oxygen, apart from the O2 atoms, and, as it will be discussed later, a 270 shift in the O3 oxygen. The change in the M2-O2 distance and the O3 shift do not affect the size of 271 the b axis, being almost normal to the diad axis. Likewise the tetrahedral chain contribution is very 272 small. The increase along b is due to an increase in the size of the M1 polyhedron, in turn due to the 273 increase in the M2-O1A1 distances (Figs. 3 and 4).

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274 The size of the c axis is related to the tetrahedral chain size (Cameron et al. 1973). It may 275 change by a kinking of the O3-O3-O3 tetrahedral angle, or by a variation in the O3C1-O3C2 276 distance, which is oriented almost along the c axis. In C2/c Co-pyroxenes the c axis changes little 277 along the series, like previously found in Ca-Mg pyroxenes (Tribaudino 2000): a balance occurs 278 between the small chain kinking along the series, which would decrease, and the O3C1-O3C2 279 distances, which would increase the c axis.

280 The parameters that change the most along the series are thea axis and the β angle, both 281 lying on the (010) plane. The deformation on the (010) plane mostly accounts for the deformation 282 due to chemical substitution, as shown by an analysis of the strain tensor.

283 The changes in cell parameters with composition give rise to a macroscopic compositional 284 strain whose tensor is calculated from the cell parameters of the different compositions; it is 285 geometrically represented by a strain ellipsoid, whose axes are the diagonal components of the 286 strain tensor (Ohashi and Burnham 1973). In monoclinic pyroxenes the strain ellipsoid is 287 constrained by symmetry to have two axes lying onto the (010) plane and the third coincident with 288 the **b** axis. The relevant variables are the size of the axes and the orientation onto the (010) plane 289 respect to the crystallographic **a** and **c** axes. The usefulness of the strain analysis is to relate atomic 290 changes to the macroscopic cell shape deformation, and to compare such effect within isostructural 291 series.

292 The strain for the substitution of a smaller cation for Ca in the M2 site was therefore 293 calculated comparing cell parameters of different composition, in our Co-pyroxenes and the related 294 Ca-Mg and Ca-Fe pyroxenes. An Eulerian strain formalism, with the orientation of the reference 295 Cartesian with crystallographic axes as x / a^* , y / b and z / c, was used. The compliance tensor 296 sums were calculated from the unit-cell parameter variation with pressure using the WinStrain 297 program, a modified version of the Strain program by Ohashi (1972), kindly provided by R.J. Angel 298 (University of Padova). The results are reported in Table 5.

299 As shown in Table 5 the two strain axes showing higher deformation lie both on the (010) 300 plane: one shows a compression at 60° from the c axis, the other a smaller expansion normal to that 301 direction. Instead the strain along the b axis is the smallest. The orientation of the strain ellipsoid is 302 almost the same in the three series, indicating a similar mechanism of deformation.

303 The highest deformation direction (ε_1) is almost coincident to the projection onto the (010) 304 of the M2-O3C1,D1 bonds, i.e. the distance O3C1-O3D1 (Fig. 12). This distances decreases strongly along the series, from 3.404 to 3.219 Å, i.e by about 6%. The expansion normal to the ε_1 , 305 306 along the ε_3 axis, occurs very close to the O3C2-O3D2 distance, along the projection of the longest 307 M2-O3 bonds onto (010): again the changes along this distance, with increases from 4.369 to 4.483

Å account for the deformation. This pattern of deformation is present in Ca-Co, Ca-Fe and Ca-Mg pyroxenes, but the degree of deformation varies: the deformation increases with increasing substitution in Ca-Co and Ca-Mg pyroxenes, much less in Ca-Fe pyroxenes. It looks that the same substitution has a more profound effect with larger ionic radius difference.

The structural mechanism involved in this deformation is that of a shift of the tetrahedral chains (Fig. 12). Tetrahedral chain shift was described in detail in several papers (Benna et al. 1988, Tribaudino 1996, Tribaudino and Nestola 2002), to describe the change in position of the O3 bridging atoms on facing tetrahedral chains. Such shift in C2/c clinopyroxenes enables to decrease the average coordination for the M2 site with increasing substitution of a smaller cation for a larger one.

- 318
- 319 Local structure

320 The displacement parameters of this work pyroxenes increase their size from $Ca_{0.8}Co_{0.2}$ to 321 $Ca_{0.6}Co_{0.4}$, whereas in $Ca_{0.4}Co_{0.6}$ they are similar to $Ca_{0.6}Co_{0.4}$ (Tab. 2, Figs 2 and 13). A plot of the 322 displacement parameters for the M2 site (unsplit) in the Ca-Co, Ca-Mg and Ca-Fe series shows that, 323 in spite of the different experimental setup and collection methods, the displacement parameters 324 follow a well defined trend: an increase up to the intermediate composition, and a symmetric 325 decrease towards end members, so that in intermediate compositions the vibration ellipsoid is much 326 larger than in end members. The C2/c to $P2_1/c$ transition at ~ 0.60 and 0.30 atoms p.f.u. in Ca-Mg 327 and Ca-Fe pyroxenes (Ohashi et al 1975, Tribaudino 2000) does not affect this trend.

The U_{eq} of all atoms increase from the end members towards the intermediate solid solutions, but the increase is higher in the M2, O2 and O3 atoms. Also the orientation of the displacement parameters changes significantly with composition, best seen on the M2 site, which in intermediate samples has a displacement ellipsoid elongated with the diad axis, whereas in end members it has the major axis normal to the diad axis (Fig. 2).

The results in Fig. 13 are simply interpreted as an effect of positional disorder overlapping to thermal vibration. The positional disorder is higher in some atoms, like M2, O2 and O3. The coexistence of local configurations centered by Ca and Co in the M2 site, affecting also the O2 and O3 atoms is the most obvious interpretation: disorder disappears in end members, where only one configuration is present.

The position of split O2 and O3 atoms was previously refined in C2/c and $P2_1/c$, Ca-Mg pyroxenes, collected with high resolution, and full configuration of Ca and Mg centered M2 polyhedra could be obtained (Tribaudino et al. 1989, Tribaudino and Nestola 2002). It was observed that the split M2 polyhedral configurations are quite similar to those in end member structures. However, the behavior of the M2 and O2 atoms is different from that of the O3 atom: the former increase continuously from the Ca-richer end member, whereas the O3 has a non-linear increase, smaller at first, and higher then (Fig. 13).

- 345
- 346

Discussion

347 The above results detail the structural changes resulting from the substitution of a smaller 348 cation for Ca in the M2 site of C2/c pyroxenes. The solid solution involves M2 site splitting, a shift 349 in the average M2 position off the O3 atoms along the diad axis, and a shift in the tetrahedral chain, 350 normal to the diad axis (Fig. 14). The chain shift and the related change of interchain O3-O3 351 distances accommodate most of the compositional macroscopic strain. During this process the M2 352 polyhedron does not change its size along the b axis, but normal to this axis the O2 atom 353 approaches the M2 cation. The approach of the O2 atom is obtained by a tilt of the tetrahedral 354 chain, which does not affect the cell deformation. This is found when Ca is substituted by either Co, 355 Mg or Fe (Figs. 3, 6, 7, 8, 9, 14), for any degree of cation substitution; changes in M2-O bond 356 lengths, in the deformation of the M2 polyhedron, in chain shift, and, in general, in the 357 compositional strain are higher as Ca content decreases. The substitution of a smaller cation in M2 358 (like Mg) also promotes higher deformation than that of a larger one (like Fe): higher chain 359 deformation is obtained with less substitution in Ca-Mg, in respect to Ca-Co and Ca-Fe pyroxenes 360 (Ohashi et al. 1975, Tribaudino 2000).

The proposed mechanism of solid solution appears limited to a critical composition when the C2/c phase is no more stable at room conditions, and the transition to $P2_1/c$ occurs. In Ca-Co pyroxenes the transition occurs between Ca_{0.4}Co_{0.6}CoSi₂O₆ and Ca_{0.3}Co_{0.7}CoSi₂O₆ (Mantovani et al. 2012), and in Ca-Mg ones at Ca_{0.6}Mg_{0.4}MgSi₂O₆ (Tribaudino 2000), i.e. at a composition where the chain shift approaches 0.9 Angstrom, which is near a limiting value (Fig. 14).

366 In Ca-Fe the transition was observed at $Ca_{0.3}Fe_{0.7}FeSi_2O_6$, by the appearance of symmetry 367 violating reflections (Ohashi et al. 1975). Although at this composition the shift is lower than in Ca-368 Mg and Ca-Co pyroxenes, the same non-linear mechanism applies.

The above data indicate that substitution in the M2 site strongly affects the tetrahedral chain shift, in a non linear way, accounting for the compositional strain. In C2/c pyroxenes the substitution may occur maintaining the equivalence in facing chains by a shift of the tetrahedral chains. As the substitution of a smaller cation for Ca goes on, the compositional deformation increases, and at a given composition, the equivalence of facing chains is lost, and a transition to the $P2_1/c$ symmetry occurs. The M2-O bond distances, chain shift, O3-O3-O3 angle and compositional strain show that small degrees of substitution of a smaller cation for Ca has a smaller effect on the tetrahedral chain than when the substitution occurs to a greater extent.

An analytical representation of the increased deformation of the unit cell with the decreased size of the average cation radius in the M2 site is shown in Fig. 15. The axial compositional strain was calculated for several intervals in Ca-Fe, Ca-Co and Ca-Mg pyroxenes (deposited, partially reported in Table 5). To obtain a single scalar value a unit scalar strain ε_s was also calculated from the three ε_1 , ε_2 and ε_3 axes of the strain ellipsoid as :

383
$$\varepsilon_{\rm s} = \sqrt{\varepsilon_1^2 + \varepsilon_2^2 + \varepsilon_3^2}$$

It is observed that the unit strain ε_s is linearly related to the average M2 ionic radius (IR^{M2}, Fig 15); it appears that some difference exists between Ca-Co and Ca-Mg pyroxenes, the latter showing higher strain, but an overall general trend can be all the same fitted as:

387

$$\varepsilon_{s}$$
=-0.0072(12)IR^{M2}+0.0082(13), R²=0.75.

388 To note, the strain is very close to zero at the ionic radius of Ca (1.12Å). The deformation 389 needed to accommodate the same amount of the smaller cation in the M2 polyhedron increases 390 therefore with substitution, up to a point where solid solution is no more possible: a higher 391 deformation increases the energetic demand for a further substitution, thus increasing the free 392 energy of the solid solutions. This explains the presence of a miscibility gap in quadrilatedal 393 pyroxenes. Further investigation is in progress to constrain the presence of a miscibility gap also in 394 Ca, Co pyroxenes.

395 At the level of the local structure the same is apparent also from the displacement 396 parameters of the O3 atoms: the U_{eq} of the O3 atoms do not increase significantly their size up to 397 20% substitution of the smaller cation (Fig. 13), whereas M2 and O2 do. The O3 atoms are tightly 398 bonded to the tetrahedral chain, from which they receive most of their formal charge, whereas the 399 displacement of the O2 atoms is most sensitive to local M2-O2 distances. The suggestion is that 400 small substitution does not affect local chain configurations. An independent confirmation comes from the analysis of the Raman peaks in Ca-Mg pyroxenes. The strong peak at 670 cm⁻¹, which by 401 402 quantum-mechanical simulation was ascribed mostly to the tetrahedral chain bending (Prencipe et 403 al. 2012), does not increase its linewidth from diopside up to $(Ca_{0.8}Mg_{0.2})MgSi_2O_6$, whereas the linewidth of peak at 340 cm⁻¹, which was instead related to bending and stretching of the M2-O 404 405 bonds, increases significantly (Tribaudino et al. 2012).

406 With higher substitution also the tetrahedral chains are deformed differently for the 407 different substituting cations in the M2 polyhedron, and local chain configurations occur. As the

408	tetrahedral chain is more and more affected by cation substitution, higher strain occurs, eventually
409	hindering further solid solution.
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562 **Table 1.** Cell and structure refinement parameters.

Samples	Ca _{0.8} Co _{0.2}	Ca _{0.6} Co _{0.4}	Ca _{0.4} Co _{0.6}
a (Å)	9.794(2)	9.7754(4)	9.747(6)
b (Å)	8.953(2)	8.9686(4)	8.982(5)
<i>c</i> (Å)	5.243(1)	5.2421(4)	5.249(3)
eta(°)	105.66(3)	106.302(4)	107.46(8)
$V(Å^3)$	442.64	440.11	438.37
Space group	C2/c	C2/c	C2/c
Z	4	4	4
Radiation	ΜοΚα	$MoK\alpha$	ΜοΚα
No. of indep. refl. $I > 4\sigma$	684	671	590
$2\theta_{\max}(^{\circ})$	64	64	64
R ₄₀ (F) (%)*	2.6	2.6	2.6
No. parameters	49	52	52
Goof	0.800	0.807	1.107

563 * split model

565

Table 2. Fractional atomic coordinates and isotropic displacement parameters. x and z for the M2,
 M2* and M2' special position are 0 and 0.25, for the M1 0 and 0.75 (both in e Wyckoff position)

	1 1	Ca _{0.8} Co _{0.2}	Ca _{0.6} Co _{0.4}	0.75 (both in e Wyck Ca _{0.4} Co _{0.6}
M2	у	0.2965(1)	0.2857(1)	0.2690(1)
1012	y U _{eq}	0.0134(2)	0.0222(3)	0.0233(2)
M2*	y y	0.3016(2)	0.3003(4)	0.2966(5)
1012	y U _{eq}	0.0072(3)	0.0093(5)	0.0118(8)
M2'	y y	0.2721(6)	0.2650(4)	0.2580(2)
1112	y U _{eq}	0.0199(16)	0.0173(8)	0.0144(5)
M1	-	0.9072(1)	0.9063(1)	0.9052(1)
1011	y U _{eq}	0.0056(2)	0.0068(2)	0.9032(1) 0.0098(2)
Т	U _{eq} X	0.0030(2) 0.2881(1)	0.0008(2) 0.2903(1)	0.2940(1)
1		0.0924(1)	0.2903(1) 0.0914(1)	0.2940(1) 0.0900(1)
	У			
	Z	0.2322(2)	0.2370(2)	0.2476(2)
01	U_{eq}	0.0047(2)	0.0062(2)	0.0098(2) 0.1202(2)
01	Х	0.1177(2)	0.1185(3)	0.1203(2)
	У	0.0884(2)	0.0886(3)	0.0888(2)
	Z	0.1473(4)	0.1478(5)	0.1503(4)
~ •	U_{eq}	0.0064(4)	0.0079(4)	0.0108(4)
O2	Х	0.3617(2)	0.3646(3)	0.3700(3)
	У	0.2487(2)	0.2478(3)	0.2459(3)
	Z	0.3239(4)	0.3329(6)	0.3475(5)
	U_{eq}	0.0096(4)	0.0144(5)	0.0185(5)
03	Х	0.3511(2)	0.3515(2)	0.3529(2)
	У	0.0198(2)	0.0221(3)	0.0232(3)
	Z	0.9948(4)	0.9972(5)	1.0064(5)
	U_{eq}	0.0074(4)	0.0101(5)	0.0200(5)

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570	Table	e 3 (to be depo	o sited) Anisotr	opic displacement parameters.
571	Ca _{0.8}		$Co_{0.4}$ $Ca_{0.4}$	
572	M2	Ca _{0.6}	C00.4 Ca0.4	00.6
		0.0124(4)	0.0114(2)	0.0125(2)
573	U_{11}	0.0124(4)	0.0114(2)	0.0135(3)
574	U ₂₂	0.0177(4)	0.0450(6)	0.0410(5)
575	U33	0.0084(3)	0.0086(4)	0.0133(3)
576	U ₁₂	0	0	0
577	U ₁₃	-0.0001(2)	0.0002(3)	0.0008(3)
578	U ₂₃	0	0	0
579	M2*			
580	U11	0.0103(5)	0.0108(7)	0.0136(11)
581	U ₂₂	0.0040(5)	0.0080(11)	0.0086(22)
582	$U_{33}^{}$	0.0057(4)	0.0079(7)	0.0108(10)
583	U ₁₂	0	0	0
584	U_{13}	-0.0012(3)	0.0005(5)	-0.0001(8)
585	U_{23}	0	0	0
586	M1	v	Ū.	Ŭ
587	U_{11}	0.0065(2)	0.0073(2)	0.0106(3)
588			0.0070(2)	0.0083(3)
	U_{22}	0.0054(2)		
589	U_{33}	0.0054(2)	0.0064(2)	0.0105(2)
590	U_{12}	0	0	0
591	U ₁₃	0.0011(2)	0.0021(2)	0.0036(2)
592	U_{23}	0	0	0
593	Т			
594	U_{11}	0.0052(3)	0.0062(3)	0.0102(4)
595	U ₂₂	0.0044(3)	0.0061 (3)	0.0071(3)
596	U33	0.0051(3)	0.0073(3)	0.0128(3)
597	U ₁₂	-0.0005(2)	-0.0011(2)	-0.0013(2)
598	U13	0.0013(2)	0.0032(2)	0.0047(2)
599	U ₂₃	-0.0004(2)	-0.0008(2)	-0.0011(2)
600	02			
601	Ū11	0.0049(7)	0.0070(7)	0.0111(8)
602	U_{22}	0.0075(7)	0.0098(7)	0.0097(8)
603	U_{33}	0.0077(7)	0.0075(7)	0.0109(8)
604	U_{12}	0.0005(5)	0.0009(5)	0.0009(6)
605	$U_{12} U_{13}$	0.0014(6)	0.0009(5)	0.0036(6)
606				
	U_{23}	0.0005(5)	0.0002(5)	0.0001(6)
607	01	0.0100(7)	0.01(7(0))	0.0015(0)
608	U ₁₁	0.0120(7)	0.0167(8)	0.0215(9)
609	U_{22}	0.0070(7)	0.0096(8)	0.0112 (8)
610	U ₃₃	0.0114(8)	0.0206(9)	0.0290(10)
611	U_{12}	-0.0030(6)	-0.0058(6)	-0.0074(7)
612	U ₁₃	0.0043(6)	0.0112(7)	0.0167(8)
613	U ₂₃	-0.0016(6)	-0.0048(6)	-0.0080(7)
614	03			
615	U_{11}	0.0071(7)	0.0063(7)	0.0102(8)
616	U ₂₂	0.0098(7)	0.0170(7)	0.0290(11)
617	$U_{33}^{}$	0.0065(7)	0.0073(7)	0.0201(9)
618	U ₁₂	0.0001(5)	0.0001(6)	0.0011 (7)
619	U_{13}^{12}	0.0025(6)	0.0030(6)	0.0035(7)
620	U_{23}	-0.0021(5)	-0.0033(6)	-0.0117(8)
621	- 23			
622				
623				
023				

624 **Table 4:** atomic distances and angles.

Table 4: atomic d		angles.		
	CaCoSi ₂ O ₆	$Ca_{0.8}Co_{1.2}$	$Ca_{0.6}Co_{1.4}$	$Ca_{0.4}Co_{1.6}$
M2-O1A1,B1 ×2	2.378(1)	2.329(2)	2.259(2)	2.154(2)
M2-O2C2,D2 ×2	2.332(1)	2.315(2)	2.236(3)	2.117(3)
M2-O3C1,D1 ×2	2.454(1)	2.619(2)	2.697(3)	2.793(3)
M2-O3C2,D2 ×2	2.744(1)	2.736(2)	2.796(3)	2.917(3)
<m2 o="" –=""></m2>	2.502(1)	2.500(2)	2.497(3)	2.495/3)
M2*-O1A1,B1 ×2		2.366(2)	2.364(3)	2.347(4)
M2*-O2C2,D2 ×2		2.323(2)	2.257(2)	2.147(3)
M2*-O3C1,D1 ×2		2.585(2)	2.598(3)	2.594(4)
M2*-O3C2,D2 ×2		2.710(2)	2.716(3)	2.767(4)
<m2 -="" o=""></m2>		2.496	2.484	2.463
M2'-O1A1,B1 ×2		2.159(4)	2.118(3)	2.082(2)
M2'-O2C2,D2 ×2		2.286(2)	2.219(2)	2.113(3)
M2'-O3C1,D1 ×2		2.790(5)	2.848(3)	2.874(3)
M2'-O3C2,D2 ×2		2.874(4)	2.913(3)	2.983(3)
<m2' -="" o=""></m2'>		2.527(4)	2.524(3)	2.513(3)
V_{M2} (Å ³)	25.85(3)	25.73(4)	25.32(4)	24.51(10)
M2 along b axis**	3.686	3.685	3.690	3.693
M2-M1	3.505(1)	3.485(1)	3.403(1)	3.268(1)
M2-M1(1)	3.208(1)	3.194(1)	3.136(1)	3.056(1)
M1-O1A2,B2 ×2	2.095(1)	2.091(2)	2.082(3)	2.077(2)
M1-01A1,B1 ×2	2.095(1) 2.136(1)	2.071(2)	2.082(3) 2.157(2)	2.077(2)
M1-O1A1,B1 ×2 M1-O2C1,D1 ×2	2.071(1)	2.069(2)	2.070(3)	2.074(2)
<m1 -="" o=""></m1>	2.071(1) 2.101(1)	2.009(2) 2.100(2)	2.103(3)	2.074(2) 2.109(2)
V_{M1} (Å ³)	12.28(1)	12.28(2)	12.32(2)	12.42(4)
OAV***	12.28(1)	12.28(2)	12.32(2)	12.42(4)
01-M1-O2 x2	92.8(1)	92.2(1)	92.8(1)	92.9(1)
O1-M1-O2 x2	92.6(1) 92.6(1)	92.2(1) 92.3(1)	90.8(1)	88.3(1)
O1-M1-O2 x2	88.9(1)	89.2(1)	90.9(1)	93.8(1)
01-M1-01 x2	92.6(1)		90.9(1) 92.6(1)	
01-M1-01 x2 01-M1-01 x2	92.0(1) 85.8(1)	92.7(1) 85.7(1)	92.0(1) 85.5(1)	92.4(1) 85.3(1)
O2C1-M1-O2D1	93.3(1)	93.3(1)	93.2(1)	92.8(1)
O1A1-M1-O1B1 O1A1-O1B1	81.7(1) 2.793(1)	81.6(1)	81.5(1)	81.5(1)
		2.797(2)	2.815(2)	2.840(3)
O1B1-O2C2	3.012(1)	2.999(2)	2.956(2)	2.891(3)
M1 along b axis**	3.038	3.041	3.056	3.080
M1-M1	3.100(1)	3.104(1)	3.114(1)	3.128(1)
T - O1 T - O2	1.605(1)	1.606(2)	1.612(3)	1.615(3)
T - O2	1.588(1)	1.589(2)	1.595(3)	1.598(3)
T -O3A1	1.666(1)	1.663(2)	1.658(3)	1.652(3)
T-O3A2	1.684(1)	1.681(2)	1.672(3)	1.655(3)
$\langle T - O \rangle$	2.181(1)	2.180(2)	2.179(3)	2.173(3)
$V_T (Å^3)$	2.23(1)	2.23(1)	2.22(1)	2.21(1)
TAV***	25.3	23.3	18.7	15.7
tilting angle****	2.8	2.8	3.4	4.5
O3A1-T-O3A2	104.3(1)	104.7(1)	105.5(1)	106.9(1)
03-03-03	165.05(11)	164.5(2)	162.8(2)	161.9(2)
O3A1-O3A2	2.644(1)	2.646(1)	2.651(1)	2.657(2)
T along b axis**	2.226	2.227	2.222	2.209

625 * split model; ** M2 contribution: $(0.5-y/b_{01})$ *b; M1 contribution: $(y/b_{02}-y/b_{01})$ *b; T contribution $(y/b_{02}-0.5)$ *b *** OAV and TAV are angle variance for tetrahedra and octahedra (Robinson et al. 1971); **** 627 tilting angle: angle between (100) and the basal face O2-O3-O3 of a tetrahedron. Data for CaCoSi₂O₆ were 628 taken from Ghose et al. (1987)

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2	Table 5. Unit strain and axial orientation.							
	Compositional range	Unit strain (x 10 ⁻⁴ /°C)		⁴ /°C)	angle with c (°)			axial expansion
		ϵ_1	ϵ_2	E ₃	ϵ_1	ϵ_2	E ₃	ellipsoid
	$Di_{100} - Di_{80}En_{20}$	-2.5(1)	- 0.7(1)	0.1(1)	62	90	152	1:0.28:-0.03
	$Di_{80}En_{20}\ -Di_{66}En_{40}$	-5.6(1)	- 0.5(1)	3.1(1)	57	90	147	1:0.08:-0.55
	CaCoSi ₂ O ₆ - Ca _{0.8} Co _{0.2}	-1.6(1)	0.2(1)	0.4(1)	62	90	152	1:-0.12:-0.25
	$Ca_{0.8}Co_{0.2} - Ca_{0.6}Co_{0.4}$	-4.2(1)	0.8(1)	1.6(1)	57	90	147	1:-0.19:-0.38
	$Ca_{0.6}Co_{0.4}\ -Ca_{0.4}Co_{0.6}$	-7.2(1)	0.8(1)	3.4(1)	59	90	149	1:0.11:-0.47
	$Hd_{100} \ - Hd_{70}Fs_{30}$	-2.8(1)	0.9(1)	-0.1(1)	56	90	146	1:-0.32:0.04
	$Hd_{70}Fs_{30}\ -Hd_{40}Fs_{60}$	-5.2(1)	1.1(1)	1.4(1)	58	90	148	1:-0.21:-0.27

633 Notes: an angle of 90° corresponds to the orientation along the **b** axis. The orientation of the other axes on (010) is 634 measured with the c axis, from c towards a. The unit cell used for the calculation are taken from: (Ca, Mg)MgSi₂O₆ 635 pyroxenes: Tribaudino et al (1989), (Ca, Co)CoSi₂O₆ pyroxenes: this work; (Ca, Fe)FeSi₂O₆ pyroxenes: Ohashi et al 636 (1975).

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640 Captions of the figures.641

Figure 1: electron density map at the M2 site (x across, y down); the cation occupancy of the site is reported. CaM2 and CoM2 site positions are always located respectively at the M2* and M2' positions in Table 2 of $Ca_{0.4}Co_{0.6}$, to enlighten the changes in the electron density. The structure factors for the calculation in the "Ca only in the M2 site" map were taken from the refinement of the hedenbergite (Nestola et al. 2007).

- Figure 2: atomic displacement parameters of the unsplit M2 site and surrounding oxygen at the M2polyhedron.
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Figure 3: main changes with composition in the M1, M2 and T (occupied by Si) polyhedra onto the (100) plane. In between it is shown a sketch showing the changes discussed in the paper. The b and c axes are shifted from their position. Only one chain of the two tetrahedral chains linked to M2 is shown, with the fourth O1 oxygen pointing downwards, not drawn for clarity.

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- Figure 4: M2-O (unsplit model), M1-O and T-O bond lengths with composition. The 0 % corresponds to pure $Co_2Si_2O_6$.
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- Figure 5: M2-O and M2'-O bond lengths with composition in the split model.
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665 Figure 7: M2-O bond distances (unsplit model) vs ionic radius of the 8-coordinated M2 site, in 666 (Ca_xM_{1-x})MSi₂O₆ pyroxenes with M= Mg, Co and Fe. Δ M2 = M2-O3C2 - (M2-O3C1 + M2-O2 + 667 M2-O1)/3. Ionic radii from Shannon (1976)

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Figure 8: difference between the longest and shortest distance in the M2 site, divided by the samedifference in the end meber with Ca only in the M2 site, vs Ca content in the M2 site.

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Figure 9: difference between average bridging and non bridging T-O distances vs Ca content in theM2 site and average M2-O3 distance (unsplit model).

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Figure 10: relative changes in cell parameters and volume in $(Ca, Co)CoSi_2O_6$ pyroxenes. The 0 % corresponds to pure $Co_2Si_2O_6$.

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Figure 11: contributions of the different polyhedra along the b axis.

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Figure 12: C2/c clinopyroxene structure on the (010) plane: O3-O3 distances discussed in the text, the amount of deformation (exaggerated, but with relative deformation correct) and the directions of compression and expansion are shown, as well as the projection of the chain shift vector.

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Figure 13: U_{eq} vs composition (unsplit model).

- Figure 14: chain shift vs composition. Chain shift is calculated as $2a\cos\beta(x_{O3}-0.5)+2cz_{O3}$.
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688 Figure 15: Unit strain vs average ionic radius of the M2 cation. The unit strain was calculated in an

689 interval of 0.01 Ca atoms p.f.u. The average M2 ionic radius for the M2 site is the average ionic

690 radius of the composition in between those over which the strain was calculated.



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