**REVISION 1** 

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3	An X-ray diffraction study of the pressure-induced hydration in cordierite at 4-5 GPa
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6	
7	Abstract

8 The elastic and structure behavior of natural cordierite compressed in aqueous medium up to 6 9 GPa was studied by means of in situ synchrotron powder diffraction with a diamond anvil cell. 10 In the range between 1-4 GPa the elastic behavior is regular and slightly anisotropic, with linear 11 compressibilities  $\beta_a:\beta_b:\beta_c=4:4:5$ , the most rigid *a-b* plane coinciding with the orientation of 6-12 membered rings. A distinct decrease of compressibility in the range of 4-5 GPa indicates a 13 pressure-induced hydration (PIH), which is confirmed by the structure refinements. The addition 14 of about 60% of the initial water content into the cordierite channels proceeds through positional 15 disordering of the H<sub>2</sub>O sites inside the channel cavity and a stepwise filling of the H<sub>2</sub>O position 16 inside the 6-membered rings, leading to the phase transition at about 4.7 GPa. The appearance of 17 H<sub>2</sub>O molecules inside 6-membered rings prevents their contraction and even causes their slight 18 enlargement along a direction, apparently related to the orientation of H-bonds. This results in 19 anisotropic deformation of the unit cell and the increase of a parameter in the HP phase at 4.9 20 GPa, as well as the decrease of linear compressibility along a upon the further compression up to 21 6 GPa ( $\beta_a:\beta_b:\beta_c=5:9:10$ ).

Key-words: cordierite, high pressure, compressibility, pressure-induced hydration, crystal
 structure.

24 Introduction

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25 Cordierite, a ring framework aluminosilicate  $(Mg,Fe)_2[Al_4Si_5O_{18}]*n(H_2O,CO_2)$ , has 26 attracted much interest due to its widespread formation in moderate- to high-grade metamorphic

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27 rocks, as well as due to its channel structure accessible for small molecules (predominantly  $H_2O$ 28 and CO<sub>2</sub>) (Shrever and Yoder 1964; Gibbs 1966; Cohen et al. 1977; Goldman et al. 1977; 29 Hochella 1979; Armbruster 1985; Poon et al. 1990; Le Breton and Schreyer 1993; Lepezin et al. 30 1999; Toohill et al. 1999; Kolesov and Geiger 2000; Bul'bak and Shvedenkov 2005). Despite its 31 petrological importance, only few works were devoted to the high-pressure behavior of cordierite 32 (Mirwald 1982; Mirwald et al. 1984; Koepke and Schulz 1986). In the low pressure region the 33 volume compressibility of cordierite ( $K_0$ =90 GPa) is close to that of feldspars, whereas in the 34 range of 1-3 GPa it approaches the order of that of beryl ( $K_0$ =143 GPa) (Mirwald et al. 1984; 35 Hazen et al. 1986). This indicates a dominant role of the ring elements in compression behavior 36 of cordierite structure. In the cordierite structure, the 6-membered (Si,Al)O<sub>4</sub>-tetrahedral rings 37 (6mR) form channels by stacking along the *c*-direction. The rings are included in a frame of 38 edge-sharing (Mg,Fe)O<sub>6</sub> octahedra and (Si,Al)O<sub>4</sub>-tetrahedra, with the octahedra located between 39 the ring planes (Gibbs 1966; Cohen et al. 1977).

The compression behavior of "empty" (waterless) cordierite was found to depend on the 40 41 type of pressure-transmitting medium. In a non-penetrating (methanol-ethanol) medium, the 42 compression is approximately regular up to 3 GPa, whereas in penetrating water medium it 43 sharply decreases at P>2 GPa due to the incorporation of excess H<sub>2</sub>O molecules (Koepke and 44 Schulz 1986). At that, a significant shrinking of 6mR tetrahedra was observed, which was 45 supposed to introduce a phase transition at higher pressure. Indeed, our recent Raman 46 spectroscopic study has revealed a reversible, low-kinetics phase transition in natural cordierite 47 upon the compression in aqueous medium up to about 4.7 GPa (Likhacheva et al. 2012). This 48 transition is evinced by the abrupt shifts of all the framework and O-H stretching modes. The 49 shift magnitudes of different framework modes indicate the predominance of distortion over 50 contraction of the framework polyhedra upon this transition. However, the influence of 51 penetrating water medium onto the observed transition remains unclear. Since there is no 52 structural data on the HP-behavior of cordierite at P>2.5 GPa, the present study aims at a

detailed description of the elastic and structural behavior of cordierite in the pressure interval between 2 and 6 GPa, and, in particular, the elucidation of the penetrating role of water in the observed phase transition.

56 **Experimental** 

57 A sample of cordierite (Z-cordierite) used in this study is from the Altai amphibolite 58 complex (Russia), with the chemical formula  $Na_{0.07}(Mg_{1.57}Fe_{0.36}Mn_{0.07})[Al_{3.96}Fe_{0.06}Si_{4.98}O_{18}]$ 59 \*0.45H<sub>2</sub>O (the average from 7 microprobe analyses). The same sample was used in the HP 60 Raman study of Likhacheva et al. (2012). It presents homogenous, inclusion-free, transparent 61 and uniformly colored grains. The  $H_2O$  content was determined by gas chromatographic analysis 62 with the accuracy of 0.08 wt%. The cell parameters at ambient conditions determined by Rietveld refinement are a = 17.104(2), b = 9.746(1), c = 9.333(1) Å, V = 1555.7(2) Å<sup>3</sup>, space 63 64 group Cccm. The powdered sample was placed into a 400×100 µm gasket hole of a diamond 65 anvil cell (DAC) and compressed up to 6 GPa. The DAC is based on a modified Mao-Bell design (Fursenko et al. 1984) and employs two diamonds with 1 mm diameter culets. Two HP 66 67 experiments were carried out using both penetrating (H<sub>2</sub>O+30% methanol:ethanol 4:1) and nonpenetrating (methanol:ethanol 4:1) medium. The pressure values were measured before and after 68 69 the diffraction experiment from the R1 ruby fluorescence line (Mao et al. 1986) excited by a 514 nm line of Ar laser. The powder diffraction experiments were performed at the 4<sup>th</sup> beamline of 70 71 the VEPP-3 storage ring of Synchrotron center SSTRC of the Institute of nuclear physics, 72 Novosibirsk (Ancharov et al. 2001), at a constant wavelength of 0.3685 Å. An MAR345 imaging 73 plate detector (pixel dimension 100 µm) was used. The sample to detector distance was 368 mm, and the focused X-ray beam was  $100 \times 100 \ \mu\text{m}^2$  in size. The program FIT2D (Hammersley et al. 74 75 1996) was used for integrating the two-dimensional images up to  $2\Theta_{max} = 25^{\circ}$ . The lattice 76 parameters of cordierite at different pressures were refined by whole pattern fitting using the 77 Rietveld method, with the GSAS package (Larson and Von Dreele 2000). The diffraction

profiles collected at ambient pressure (hereafter 0 GPa), 4.4 and 4.9 GPa (in aqueous medium)

79 were used for the structure refinements.

Rietveld refinement of the crystal structure was performed using data in the 20 range 3-80 81  $25^{\circ}$ . The instrumental background was approximated by a Chebyshev polynomial with 28 82 coefficients. The Bragg peak profiles were refined by a pseudo-Voight function. The scattering 83 factors for silicon, aluminum and sodium atoms were used to model the framework (Si,Al) and 84 extra-framework species (Na), respectively. To model the  $H_2O$  molecules, we used the scattering 85 factor for fluorine atom, providing a better approximation to the scattering capability of  $H_2O$ 86 molecule in comparison with oxygen atom, as it was shown earlier (Servotkin et al. 2003). The 87 Si-O and Al-O distances were restrained to 1.61(2) Å and 1.75(2) Å, respectively; the restraints 88 were removed only in several final cycles of the refinement. The  $U_{iso}$  values assigned to 89 (Mg,Si,Al), O, Na and F atoms were taken from the data of Sokol et al. (2010) and Koepke and 90 Shulz (1986). Due to a narrow 2 $\theta$  range used, the  $U_{iso}$  values for all atoms were not refined. The 91 crystal structure of natural (Mg,Fe)-cordierite (Hochella et al. 1979) was used as a starting model 92 for the refinement of the LP structure at 0 GPa. The HP structures were derived from the refined 93 LP structure with addition of a new  $H_2O$  split site close to the original position Ow1 in the 94 channel, as well a new H<sub>2</sub>O site inside 6mR. The final refined structure parameters were the 95 atomic coordinates (for all the atoms) and the occupancy factors for the  $H_2O$  sites.

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#### **Results and discussion**

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### Variation of the lattice parameters at high water pressure

During the compression in aqueous medium up to 6 GPa, no change in the profile shape of Z-cordierite is observed, which indicates the preservation of the original space group *Cccm* at high pressure. In the pressure range between 0-4 GPa the compression is slightly anisotropic (Fig. 1a), in agreement with Brillouin measurements of Toohill et al. (1999). The mean axial compressibilities estimated from linear regressions ( $\beta_a = \beta_b = 2.4(1) \cdot 10^{-3}$ ,  $\beta_c = 3.1(1) \cdot 10^{-3}$  GPa<sup>-1</sup>) are similar to the previous data of Mirwald et al. (1984) and Koepke and Schulz (1986) obtained

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104 for non-penetrating medium. The less compressible plane *a-b* is parallel to the orientation of 105 rigid 6mR. Within the pressure range of 0-4 GPa, the two sets of the volume data obtained using 106 both penetrating and non-penetrating medium closely coincide (Fig. 1b). Therefore, the P-V data 107 for the compression in aqueous medium at 0-4 GPa and those obtained for the compression in 108 non-penetrating medium at 0-5.5 GPa were used as a whole to estimate the elastic parameters. 109 Fitting the volume data with Murnaghan EoS (Murnaghan 1937, Angel 2001) gives:  $V_0 =$ 1569(2) Å<sup>3</sup>,  $K_0 = 115(3)$  GPa, K' = 4 (fixed) (Fig. 1a). The  $K_0$  value is slightly lower than  $K_0 =$ 110 111 129(1) GPa obtained from Brillouin measurements by Toohill et al. (1999), and is intermediate 112 between the  $K_0$  values of 91(5) and 143(5) GPa found by Mirwald et al. (1984) for the pressure 113 ranges of 0-1 and 1-3 GPa.

114 In the pressure range between 4 and 4.4 GPa, the volume compressibility of Z-cordierite 115 in aqueous medium is lowered to almost zero (Fig. 1b, Table 1) due to a substantial drop of the 116 compressibility along a and b directions. This may be regarded as an indirect evidence for a 117 pressure-induced hydration (PIH) of the framework channels. A similar decrease of 118 compressibility due to PIH was observed by Koepke and Schulz (1986) in "empty" cordierite at 119 2.3 GPa. The comparison with the volume data obtained in non-penetrating medium at P > 3.5120 GPa (Fig. 1b) argues for the observed compression anomaly to be due to the penetrating function 121 of water.

122 Upon the pressure rise up to 4.9 GPa, an anisotropic volume expansion of  $\approx 0.1$  % results 123 from the predominant increase of the a parameter and less enlargement along c, whereas the b124 parameter does not change. Upon the further compression at P > 5 GPa the a and b 125 compressibilities change into opposite directions, whereas the compressibility along c and the bulk modulus retain the initial value:  $\beta_a = 1.6(1) \cdot 10^{-3}$ ,  $\beta_b = 2.7(1) \cdot 10^{-3}$ ,  $\beta_c = 3.0(1) \cdot 10^{-3}$  GPa<sup>-1</sup>,  $K_0 = 1.6(1) \cdot 10^{-3}$ ,  $K_0 = 1.6(1) \cdot 10^{-3$ 126 115(2) GPa (fitting with Murnaghan EoS, K' = 4 (fixed)). This change of the compression 127 128 mechanism is apparently related with the transition to a new HP phase associated with PIH in the 129 pressure range between 4.4 and 4.9 GPa. The existence of phase transition is in line with the

130 previous spectroscopic data showing the abrupt shifts of the Raman bands in Z-cordierite in the

131 same pressure interval (Likhacheva et al. 2012).

Based on the observed deviation from regular compression at P > 4 GPa, we can suppose that the transition to the H*P* phase is preceded by the formation of intermediate PIH state of cordierite structure in the pressure range between 4 and 4.5 GPa.

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#### The crystal structure of Z-cordierite at 4.4 and 4.9 GPa

136 Refinement details for the LP and HP structures of Z-cordierite are reported in Table 2, 137 and the refined atomic parameters at 0, 4.4 and 4.9 GPa are given in Tables 3 and 4. The 138 Rietveld plots and the LP and HP structures are drawn in Figures 2 and 3, correspondingly. In 139 the LP structure at 0 GPa all the distances and angles in the framework polyhedra are in 140 reasonable agreement with the previous single-crystal data (Hochella et al. 1979; Yakubovich et 141 al. 2004). The refined occupancy of Na site in the center of 6mR and that of  $H_2O$  (Ow1) site 142 located in the large cavity of the channel (Fig. 3a, Table 3) agree with the data of chemical 143 analysis.

144 Based on the compression trends described above, we consider the HP structures refined 145 at 4.9 and 4.4 GPa to represent the HP phase and its precursor, correspondingly. As compared to 146 the LP structure, the occupancy of the position at 0,0,0 in the center of 6mR gradually increases 147 in the precursor and HP phase (Table 3). We should emphasize that the increase of the 0,0,0148 occupancy in both structures was obtained in several repeated refinements. Provided the Na 149 occupancy was fixed at the initial value of 0.08, the observed increase of the 0.00, occupancy is 150 obviously related with the penetration of extra  $H_2O$  molecules from the pressure-transmitting 151 medium. The  $H_2O$  position inside the 6mR was labeled as Ow3 (Figs. 3b,c, Tables 3,4). Due to 152 the appearance of  $H_2O$  inside 6mR, we had to introduce into the structure model an additional 153 H<sub>2</sub>O site (Ow2) slightly shifted from the initial position Ow1 in the center of channel cavity 154 (Figs. 3b,c), in order to provide a reasonable distance to the site Ow3 inside 6mR (Table 4). This 155 has led to redistribution of the occupancy factors between the Ow1 and Ow2 sites inside the

channel cavity (Table 3), the total (Ow1+Ow2) occupancy being increased by about 20% of the initial value in the precursor phase at 4.4 GPa (Table 3). Appreciable lowering of R-factor confirmed the addition of the position Ow2 to be reasonable. It should also be noted that the positional disordering of H<sub>2</sub>O inside the cage due to the appearance of Ow2 agrees with the observed broadening of the intense Raman band of O-H stretching vibrations (Likhacheva et al. 2012).

Taking into account the contribution of new H<sub>2</sub>O sites Ow2 and Ow3, the total amount of water in Z-cordierite increases from 0.45 (0 GPa) to 0.69 and 0.72 molecules (per 18 oxygens) in the precursor and HP phase, correspondingly (Table 2). At that, the increase of the water content in the HP phase proceeds only at the expense of additional filling of the Ow3 site inside 6mR (Table 3), which can therefore be regarded as a driving force for this transition.

167 The appearance of  $H_2O$  molecules inside 6mR prevents the contraction of these rings 168 along a direction, which is seen from the preservation of the initial O2-O2 distance in 6mR in 169 both the HP structures (Fig. 3a, Table 4). It seems reasonable to suppose the H-vectors of  $H_2O$ 170 molecule inside 6mR to be oriented towards O2 atoms, because the Ow3-O2 distance of  $\approx 2.98$ 171 Å (Table 4) is the only favorable for the accommodation of H-bond. This supposition is in line 172 with the observed anisotropic deformation of the unit cell and the increase of a parameter upon 173 the transition at  $\approx 4.7$  GPa, as well as the decrease of linear compressibility along a during the 174 further compression up to 6 GPa.

The (Mg,Fe)-O<sub>6</sub> octahedra, presenting the "inner" part of the cordierite framework, contract almost uniformly. The Si(2)O<sub>4</sub>- and AlO<sub>4</sub>-tetrahedra, joining octahedral polyhedra and 6mR, undergo an anisotropic deformation without contraction: the mean (Si,Al)-O distances in these tetrahedra remain at the initial level in both the HP structures. The Si(2)O<sub>4</sub>-tetrahedron expands along *a* and slightly contracts along the *c* direction, whereas the AlO<sub>4</sub>-tetrahedron contracts mainly along the *c* axis. On the whole, both the precursor and HP phases are characterized by similar PIH level and associated deformation of the framework. The structure of

the HP phase at 4.9 GPa differs from its precursor by a minor increase of  $H_2O$  content and slight concerted rotation and twisting of all the framework polyhedra. This smooth transition is apparently driven by exceeding some critical amount of extra  $H_2O$  entering the Ow3 site inside 6mR, as it is mentioned above.

186 The location of H<sub>2</sub>O molecules in a "tight" position inside the 6mR, commonly occupied 187 only by cations in natural cordierites (Cohen et al. 1977; Armbruster 1985; Yakubovich et al. 188 2004), is quite unusual. Another example of such an "exotic" siting is provided by the high-189 temperature structural study of natural H<sub>2</sub>O-bearing cordierite (Hochella et al. 1979). It is 190 noteworthy that a minor amount of H<sub>2</sub>O inside the 6mR was located only in the sample cooled 191 down to room temperature after partial dehydration at  $775^{\circ}$ C. Therefore, the H<sub>2</sub>O molecules 192 appeared inside the 6mR as a result of their re-distribution throughout the framework channel 193 "quenched" by a relatively quick cooling. It seems probable that in both the HP and HT 194 experiments the appearance of H<sub>2</sub>O inside the 6mR reflects a kinetically hindered character of 195 water diffusion in cordierite channels, which is known to stop in the temperature range between 196  $200^{\circ}$  and  $300^{\circ}$ C (Bul'bak et al. 2005). It is interesting that the compression in aqueous medium 197 not only provides the formation of such a "non-equilibrium" distribution of H<sub>2</sub>O molecules 198 throughout the cordierite channel, but even makes the new HP phase quasi-stable in a certain 199 pressure range, according to a regular compression at P > 5 GPa.

200 To summarize, the presented structural data reveal the formation of PIH state and 201 associated phase transition in natural H<sub>2</sub>O-bearing cordierite in the pressure range between 4.4 202 and 5 GPa. This is about 2.5 GPa higher than the pressure needed for PIH in "empty" cordierite 203 (Koepke and Schulz 1986). The enlargement and lowering of the lattice compressibility along a 204 axis is related to the penetration of  $H_2O$  molecules preventing the contraction of the channel-205 forming 6mR. This is an unusual case when a nearly isotropic compressibility of cordierite 206 within the *a-b* plane is violated under the influence of  $H_2O$  molecules residing in 6mR. This HP 207 behavior differs from the deformation mechanism observed at high temperature, where the

- 208 expansion along a and b axes proceeds at the expense of the enlarged Mg-octahedra, the 6mR
- 209 remaining undeformed (Hochella et al. 1979).
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331 332	Captions for figures
333 334	Fig.1. Pressure dependence of the normalized lattice parameters (a) and volume (b) of Z-
335	cordierite compressed in aqueous medium. The volume curve obtained using an aqueous
336	medium (filled squares) is compared with our data (empty squares) and the data of Mirwald et al.
337	(1984) (dotted line) for the compression in non-penetrating alcohol medium in DAC. The
338	Murnaghan EoS fit is shown by solid line.
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340	Fig.2. Rietveld refinements of Z-cordierite at 0 GPa (a) and 4.9 GPa (b). The observed and
341	calculated profiles are indicated by (+) and by a continuous line, respectively. The tick marks
342	indicate the positions of allowed Bragg peaks in the respective symmetry. A difference curve is
343	plotted at the bottom. The demonstrated diffraction profile is limited to the region of $2\theta$ =2.5-14°
344	because this part is the most representative: it contains all the intense reflections of Z-cordierite
345	and is free from the peaks of ice 7.
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347	Fig.3. Crystal structure of Z-cordierite at 0 GPa (a), 4.4 GPa (b) and 4.9 GPa (c).
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Table 1. Lattice parameters of	of Z-cordierite (	space group	Cccm)
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at compression up to 6 GPa.						
<b>P</b> (GPa)	a (Å)	<b>b</b> (Å)	c (Å)	$V(\text{\AA}^3)$		
Compress	Compression in aqueous medium ( $H_2O+30\%$ methanol:ethanol 4:1)					
0	17.104(2)	9.746(1)	9.333(1)	1555.7(2)		
1.4	17.047(3)	9.711(2)	9.293(2)	1538.4(3)		
2.0	17.021(2)	9.692(1)	9.274(1)	1529.9(2)		
2.6	16.987(3)	9.683(1)	9.252(1)	1521.8(3)		
3.0	16.976(3)	9.670(2)	9.246(1)	1517.7(3)		
3.5	16.969(4)	9.660(2)	9.231(2)	1513.2(5)		
3.85	16.944(3)	9.659(1)	9.221(1)	1509.2(3)		
4.4	16.944(2)	9.656(1)	9.216(1)	1507.8(2)		
4.9	16.959(1)	9.653(1)	9.217(1)	1508.9(2)		
5.6	16.941(3)	9.633(1)	9.200(1)	1501.3(3)		
6.15	16.924(3)	9.621(1)	9.182(1)	1495.0(3)		
Compression in methanol:ethanol 4:1						
1.3	17.051(4)	9.703(2)	9.307(2)	1539.8(4)		
2.05	17.017(4)	9.692(3)	9.280(2)	1530.5(4)		
3.5	16.966(1)	9.639(7)	9.230(6)	1510(2)		
4.5	16.914(7)	9.625(4)	9.203(4)	1498(1)		
5.2	16.867(8)	9.625(5)	9.200(4)	1494(1)		

at nigh pressures.				
Pressure (GPa)	0	4.4	4.9	
<i>a</i> (Å)	17.104(2)	16.944(2)	16.959(1)	
<i>b</i> (Å)	9.746(1)	9.656(1)	9.653(1)	
<i>c</i> (Å)	9.333(1)	9.216(1)	9.217(1)	
$V(Å^3)$	1555.7(2)	1507.8(2)	1508.9(2)	
Space group		Cccm		
Radiation		$\lambda = 0.3685 \text{ Å}$		
2Θ range (°)	3-25			
Number of observations		2443		
Number of variables	60	65	68	
Number of reflections	650	650	650	
R <sub>p</sub>	0.004	0.004	0.006	
R <sub>wR</sub>	0.005	0.005	0.009	
$R_{\rm F}^{2}$	0.10	0.15*	0.13*	
$\chi^2$	0.23	0.21	0.34	
Calculated amount of $H_2O$ mol./f.u. (18 oxygens)	0.45(1)	0.69(2)	0.72(1)	

 

 Table 2. Basic crystallographic and experimental data for Z-cordierite

 at high pressures

 

\* The  $R_{\rm F}^2$  values are higher than that for 0 GPa because several regions are excluded from the refinement due to the presence of ice 7 

Atom	P GPa		v/h	$\frac{10, 4.4 \text{ and } 4.}{7/c}$	9 OI a.	U.
Ma	1,01 <i>u</i>	0.163(1)	y/v	0.2500000(0)	1 000(0)	0.008
wig	4.4	0.103(1)	0.5000000(0)	0.2500000(0)	1.000(0)	0.008
	4.4	0.1623(2)	0.5000000(0)	0.2500000(0)	1.000(0)	0.008
<b>Si</b> (1)	4.9	0.104(1)	0.079(1)	0.0000000000	1.000(0)	0.008
51(1)	0	0.193(1)	0.079(1)	0.0000000(0)	1.000(0)	0.008
	4.4	0.1935(2)	0.0790(3)	0.0000000(0)	1.000(0)	0.008
Si(2)	ч.) 0	0.000000000	0.5000000(0)	0.2500000(0)	1.000(0)	0.008
51(2)	44	0.0000000(0)	0.5000000(0)	0.2500000(0)	1.000(0)	0.008
	49	0.0000000(0)	0.5000000(0)	0.2500000(0)	1.000(0)	0.008
Si(3)	0	-0.137(1)	0.236(1)	0.0000000(0)	1.000(0)	0.008
- (-)	4.4	-0.1378(2)	0.2372(3)	0.0000000(0)	1.000(0)	0.008
	4.9	-0.139(1)	0.233(1)	0.0000000(0)	1.000(0)	0.008
Si(4)	0	0.049(1)	0.309(1)	0.0000000(0)	1.000(0)	0.008
	4.4	0.050(1)	0.313(1)	0.0000000(0)	1.000(0)	0.008
	4.9	0.050(1)	0.318(1)	0.0000000(0)	1.000(0)	0.008
Al	0	0.2500000(0)	0.2500000(0)	0.251(3)	1.000(0)	0.008
	4.4	0.2500000(0)	0.2500000(0)	0.248(1)	1.000(0)	0.008
	4.9	0.2500000(0)	0.2500000(0)	0.258(2)	1.000(0)	0.008
0(1)	0	0.247(1)	0.102(2)	0.144(2)	1.000(0)	0.01
	4.4	0.2468(2)	0.0984(4)	0.1435(3)	1.000(0)	0.01
	4.9	0.247(1)	0.103(2)	0.146(1)	1.000(0)	0.01
O(2)	0	-0.168(1)	0.081(3)	0.0000000(0)	1.000(0)	0.01
	4.4	-0.171(1)	0.0831(3)	0.000000(0)	1.000(0)	0.01
	4.9	-0.170(1)	0.079(2)	0.000000(0)	1.000(0)	0.01
O(3)	0	0.123(1)	0.187(2)	0.000000(0)	1.000(0)	0.01
	4.4	0.1232(3)	0.190(1)	0.000000(0)	1.000(0)	0.01
	4.9	0.122(1)	0.191(2)	0.000000(0)	1.000(0)	0.01
O(4)	0	0.064(1)	0.416(1)	0.154(1)	1.000(0)	0.01
	4.4	0.0647(3)	0.413(1)	0.153(1)	1.000(0)	0.01
	4.9	0.066(1)	0.410(1)	0.159(1)	1.000(0)	0.01
O(5)	0	-0.044(1)	0.249(2)	0.000000(0)	1.000(0)	0.01
	4.4	-0.0436(2)	0.247(2)	0.0000000(0)	1.000(0)	0.01
	4.9	-0.044(1)	0.249(2)	0.0000000(0)	1.000(0)	0.01
O(6)	0	-0.172(1)	0.313(1)	0.143(1)	1.000(0)	0.01
	4.4	-0.1706(3)	0.3152(3)	0.1424(3)	1.000(0)	0.01
	4.9	-0.172(1)	0.317(1)	0.140(1)	1.000(0)	0.01
Na	0	0.0000000(0)	0.0000000(0)	0.0000000(0)	0.08(1)	0.025
	4.4	0.0000000(0)	0.0000000000000000000000000000000000000	0.0000000(0)	0.08	0.025
01	4.9	0.0000000(0)	0.0000000(0)	0.0000000000000000000000000000000000000	0.08	0.025
UWI	4.4			0.2300000(0)	0.43(1)	0.1
	4.4			0.2300000(0)	0.13(2)	0.1
0	4.9	0.00000000000000000000000000000000000	0.00000000000000000000000000000000000	0.2300000(0)	0.12(1)	0.1
OW2	4.4	0.01(2)	0.02(3)	0.32(1)	0.110(5)	0.1
<u>Our?</u>	4.9	-0.01(2)	-0.03(2)	0.010(0)	0.113(4) 0.12(2)	0.1
0W3	4.4		0.0000000(0)	0.0000000(0)	0.12(2) 0.14(1)	0.1
	4.9	0.0000000000000000000000000000000000000	0.0000000000000000000000000000000000000	0.0000000000000000000000000000000000000	0.14(1)	0.1

423 Table 3. Atomic coordinates for Z-cordierite at 0, 4.4 and 4.9 GPa.

Table 4. Selected structural parameters for Z-cordierite at

			1.
com	pression in	aqueous	medium.

•	0 GPa	4.4 GPa	4.9 GPa
6-ring			
O(2) – O(2) IIa	5.96(3)	5.99(1)	5.95(2)
O(3) – O(3) IIa	5.56(3)	5.56(1)	5.53(2)
O(5) – O(5) IIb	5.07(3)	4.99(1)	5.03(2)
mean diameter	5.53	5.51	5.50
		•	Į.
Si(1) - O(1) = x2	1.65(1)	1.614(2)	1.65(1)
O(2)	1.61(3)	1.617(3)	1.60(2)
O(3)	1.59(2)	1.598(3)	1.60(2)
mean	1.63	1.61	1.62
Si(3) – O(2)	1.60(3)	1.588(3)	1.58(2)
O(5)	1.60(3)	1.599(3)	1.63(2)
O(6) x2	1.64(1)	1.605(2)	1.62(1)
mean	1.62	1.60	1.61
Si(4) – O(3)	1.73(2)	1.71(1)	1.72(2)
O(4) x2	1.79(1)	1.73(1)	1.74(1)
O(5)	1.70(3)	1.71(2)	1.72(2)
mean	1.75	1.72	1.73
(Mg,Fe) - O(1) x2	2.08(2)	2.056(3)	2.05(2)
O(4) x2	2.10(2)	2.060(4)	2.06(2)
O(6) x2	2.08(2)	2.060(2)	2.04(1)
mean	2.08	2.06	2.05
O(1) – O(1) IIb	5.07(2)	4.98(1)	5.01(2)
Al – O(1) x2	1.76(2)	1.751(3)	1.76(2)
O(6) x2	1.77(2)	1.792(3)	1.75(2)
mean	1.765	1.77	1.755
O(1) - O(6) IIc	4.92(3)	4.89(1)	4.82(2)
O(1) – O(1) IIb	2.89(3)	2.93(1)	2.84(2)
Si(2) – O(4)	1.63(1)	1.64(1)	1.65(1)
O(4) – O(4) llc	2.43(2)	2.44(1)	2.41(2)
O(4) – O(4) IIa	3.95(2)	3.94(1)	4.07(2)
Na/Ow3 – O(2) x2	2.98(2)	3.00(1)	2.98(2)
- O(3) x2	2.78(2)	2.78(1)	2.77(2)
– O(5) x2	2.54(2)	2.50(2)	2.52(1)
– Ow1	2.3333(2)	2.3040(2)	2.3043(2)
- Ow2		2.9(1)	2.95(5)



Fig.1a



Fig.1b

## Fig.2a



## Fig.2b



## Fig. 3a

a



# Fig. 3b b Ow2 Ow3 Ow1

## Fig. 3c

С

