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3	Characterization of ferric arsenate-sulfate compounds: Implications for
4	arsenic control in refractory gold processing residues
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### ABSTRACT

34 A combination of techniques, including powder X-ray diffraction (XRD), electron 35 microprobe analysis (EPMA), transmission electron microscopy (TEM) and X-ray 36 absorption spectroscopy (XAFS), is used to characterize the common ferric-arsenate-37 sulfate compounds which could result from the pressure oxidation of refractory gold ores 38 at elevated temperatures. Three general types of precipitate are identified; namely, 39 arsenate-bearing basic ferric sulfate (FeSO<sub>4</sub>(OH) and designated as BFS), ferric arsenate-40 sulfate (an extensive solid solution Fe(AsO<sub>4</sub>)<sub>0.2-0.7</sub>(SO<sub>4</sub>)<sub>0.7-0.2</sub>(OH)<sub>0.7-0.2</sub> and designated as 41 FAS) and hydrated ferric orthoarsenate (FeAsO<sub>4</sub>·0.75H<sub>2</sub>O). The crystal structure of FAS 42 is solved by precession electron-diffraction experiments. The structures of BFS and FAS are constructed from octahedral Fe<sup>3+</sup> chains which are cross-linked by sulfate and 43 44 arsenate tetrahedra. Extensive substitution of arsenate for sulfate occurs in both types of 45 compound with charge neutrality being maintained of variations in the (OH) content. The 46 XAFS spectra indicate that the local structures of both BFS and FAS are made of corner-47 linked single chains of FeO<sub>6</sub> octahedra where the chains are linked by AsO<sub>4</sub> or SO<sub>4</sub> 48 tetrahedra forming alternating layers of FeO<sub>6</sub> octahedra and AsO<sub>4</sub> or SO<sub>4</sub> tetrahedra. 49 Preliminary Toxicity Characteristics Leaching Procedure (TCLP) testing of the 50 precipitates indicates that FAS with a molar ratio As/(As+S) ratio of  $\leq 0.5$  could be an 51 acceptable material for disposal in a tailings impoundment whereas more As-rich FAS 52 and BFS may require further treatment. The results for the laboratory-prepared 53 precipitates are compared with those obtained on three residues from the processing of 54 refractory gold ores. The major As-carrier in one of the residues is FAS, whereas As-55 bearing goethite and hematite are the dominant As-carriers in the other two residues. 56 Thus, the mineralogical characteristics of the residues dictate the appropriate arsenic 57 management and disposal options in the processing of refractory gold ores.

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59 *Keywords*: arsenic, arsenate, sulfate, ferric arsenate sulfate, refractory gold, arsenic
60 control, hydrometallurgy, autoclave residue

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## **INTRODUCTION**

64 In refractory gold ores, gold occurs as submicroscopic particles or in solid solution in 65 arsenopyrite and pyrite at ppb to ppm levels; these occurrences are often referred to as 66 "invisible gold". Pyrite in refractory gold-sulfide deposits is often rich in As, and As 67 contents as high as 20 wt % have been reported (Abraitis et al. 2004; Paktunc et al. 2006). 68 Therefore, like the less abundant arsenopyrite, pyrite can be an important primary source 69 of As in refractory gold processing circuits. Gold extraction from refractory gold ores 70 requires a pre-treatment process such as roasting, pressure oxidation or bacterial 71 oxidation to destroy the host sulfide so that the gold is liberated and made available for 72 cyanide leaching. Roasting and pressure oxidation are the most commonly used pre-73 treatment technologies. In addition to its technical advantages, such as improved 74 liberation of gold, pressure oxidation is preferred over roasting because of environmental 75 concerns associated with the generation of gaseous emissions with SO<sub>2</sub> and As<sub>2</sub>O<sub>3</sub> in the roasting process (Fleming 2010). Pressure oxidation is performed in large autoclaves 76 77 commonly operating in acidic media at relatively high temperatures (i.e., 190-230  $^{\circ}$ C) to 78 achieve complete oxidization of the sulfide minerals within a few hours. With the 79 destruction of arsenopyrite and pyrite, significant amounts of As can be incorporated in 80 the residues along with sulfate and ferric iron.

Swash and Monhemius (1994) were among the first to examine the Fe<sup>+3</sup>-AsO<sub>4</sub>-SO<sub>4</sub> 81 82 system at elevated temperatures under hydrothermal conditions. They synthesized a series 83 of compounds including scorodite (FeAsO<sub>4</sub>·2H<sub>2</sub>O), basic ferric sulfate (Fe(SO<sub>4</sub>)(OH)) 84 and two unknown ferric arsenate and ferric arsenate-sulfate compounds, designated as 85 Type-1 and Type-2 (Table 1). Scorodite formed at temperatures below 175 °C whereas 86 the other compounds formed at higher temperatures which were dependent on the starting 87 solution composition. Despite the compositional differences, the X-ray diffraction (XRD) 88 pattern of Type-1 is similar to the ferric orthoarsenate hydrate (FeAsO<sub>4</sub>·0.75H<sub>2</sub>O) 89 reported by Jakeman et al. (1991). Type-2 has a composition defined as 90  $Fe_4(AsO_4)_3(SO_4)_v(OH)_x$  (where x+2y=3). Similarly, Dutrizac and Jambor (2007) 91 performed synthesis experiments in the same system to determine the compositions of the 92 precipitates that may be analogous to those generated in industrial autoclaves. In addition 93 to scorodite, Fe(SO<sub>4</sub>)(OH) and two unknown compounds labelled as Phase 3 and Phase 4

94 formed. Among these, Phase 4 is a ferric arsenate with a composition of 95  $Fe_{0.96}(AsO_4)_{0.99}(SO_4)_{0.01}(H_2O)_{0.54}$ . Like Type-1, Phase 4 is similar to the ferric 96 orthoarsenate hydrate of Jakeman et al. (1991). Phase 3 has a wider compositional range 97 between Fe<sub>0.8-1.1</sub>(AsO<sub>4</sub>)<sub>0.25</sub>(SO<sub>4</sub>)<sub>0.75</sub>(OH)<sub>0.75</sub> and Fe<sub>0.8-1.1</sub>(AsO<sub>4</sub>)<sub>0.54</sub>(SO<sub>4</sub>)<sub>0.46</sub>(OH)<sub>0.46</sub>. Based 98 on our experimental work conducted at CANMET and earlier studies conducted at 99 Imperial College (Ugarte and Monhemius 1992; Swash and Monhemius 1994), it is 100 anticipated that As and S released during the oxidation of arsenopyrite and pyrite under 101 industrial autoclave conditions would likely precipitate predominantly as a ferric 102 arsenate-sulfate phase (FAS). Despite their likelihood as the most abundant arsenic 103 compounds in autoclave residues, the characteristics and crystal structures of Phase 3 and 104 Type-2 remained unknown.

The objectives of this study are to characterize the common arsenical compounds which would result from the pressure oxidation of refractory gold ores and to assess their solubilities. In addition, three residue samples from industrial gold processing operations and two samples of bukovskýite ( $Fe_2(AsO_4)(SO_4)(OH) \cdot 9H_2O$  (Majzlan et al. 2012a)) were characterized to provide a basis for comparison. Mineralogical characteristics and the stability of the As species in the residues will dictate the development of appropriate arsenic management and disposal options at mine sites.

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### **METHODOLOGY**

114 In order to determine the As-bearing compounds that are typically present in pressure 115 oxidation residues, a series of experiments was performed in the ferric iron, arsenate and sulfate system using solutions containing reagent grade Fe(SO<sub>4</sub>)<sub>1.5</sub>, As<sub>2</sub>O<sub>5</sub>, H<sub>2</sub>SO<sub>4</sub> and 116 117 Fe(NO<sub>3</sub>)<sub>3</sub>. Synthesis work was performed in Parr 2-L autoclaves under hydrothermal 118 conditions at temperatures greater than ~100 °C with retention times varying from 1 to 5 119 hours. These experiments produced various ferric arsenate-sulfates, scorodite, ferric 120 orthoarsenate, ferric sulfate and jarosite; some of compounds were previously described 121 as Phase 3 and Phase 4 by Dutrizac and Jambor (2007). In addition, pressure oxidation 122 residue samples from three commercial gold operations, designated as POX-2, POX-9, 123 POX-14, were obtained to provide a basis for comparison with the synthesized samples.

124 The samples were characterized by a combination of techniques including powder X-125 diffraction (XRD), electron microprobe analysis (EPMA), transmission electron 126 microscopy (TEM) and X-ray absorption spectroscopy (XAFS).

127 Powder XRD analyses were performed using a RIGAKU D/MAX 2500 rotating-128 anode powder diffractometer with CuKa radiation (1.5406 Å) at 50 kV, 260 mA, a step-129 scan of  $0.02^{\circ}$ , and a scan rate of  $1^{\circ}$  per minute in 20 from 5 to 70°. The samples were 130 finely ground in a mortar and pestle, and packed either on Teflon covered aluminum 131 sample holders or on zero-background plates for the XRD analyses. Powder XRD 132 patterns were also collected at the bending-magnet beamline PDIFF at the synchrotron 133 light source ANKA (Angströmquelle Karlsruhe, Germany). X-rays of wavelength of 134 0.82694(1) Å were selected by a double crystal Si(111) monochromator. The wavelength 135 and the zero angle of the diffractometer were determined with silicon powder (NIST 136 standard reference material 640). The sample was loaded into a 1.0 mm glass capillary 137 which was rotated about its axis during data collection. The intensity of the incoming 138 beam was monitored during data collection by an ion chamber and the measured 139 intensities of the diffracted beam were corrected for the decay and fluctuations of the 140 primary beam. The XRD patterns were collected at room temperature, over an angular range of 4 to  $40^{\circ}$  in 20, with a step of 0.005° and counting time of 1 s per point. The 141 142 diffraction data were treated with a full-profile Rietveld refinement with the GSAS 143 (General Structure Analysis System) program of Larson and von Dreele (1994).

Electron microprobe analyses were performed by wavelength-dispersive spectrometry (WDS) on carbon-coated polished sections using JEOL JXA 8900 (CANMET) and JEOL 8230 (Earth Sciences, University of Ottawa) microanalyzers, operated at 15 kV using a probe current of 10 to 30 nA, counting times of 10 seconds and a ZAF correction. The following X-ray lines and standards were used: Fe*Ka* (marcasite), S*Ka* (marcasite) and As*La* (synthetic GaAs).

Transmission electron microscopy (TEM) examination of the samples, including structural determinations, was conducted on a Philips CM120 microscope with a LaB<sub>6</sub> cathode operating at 120 kV. The microscope is equipped with a DigiStar (NanoMegas) precession device for application of the precession electron diffraction (PED) and with a CCD Camera Olympus SIS Veleta with a 14-bit dynamical range. Precession electron 155 diffraction was used to suppress the dynamical diffraction effects to as large an extent as possible. Small, homogeneous crystals were selected for the diffraction experiment. The 156 157 crystals were not oriented along a special zone axis prior to the data collection but were 158 intentionally left in a random orientation. During the data collection the crystal was tilted 159 in steps of 1° around the main tilt axis of the sample holder, and at each tilt a diffraction 160 pattern was recorded. This experimental method dubbed "electron diffraction 161 tomography" and pioneered by Kolb and coworkers (Mugnaioli et al., 2009) allows a 162 collection of 3-dimensional electron diffraction data set and subsequent ab initio solution 163 of crystal structures from electron diffraction data. The data were processed using the 164 computer programs PETS (Palatinus, 2011) and Jana2006 (Petříček et al., 2006), to yield the unit cell parameters and integrated reflection intensities. The crystal structures were 165 166 solved *ab initio* using the computer program Superflip (Palatinus and Chapuis, 2007), 167 and refined with Jana2006 (Petříček et al., 2006).

168 XAFS spectroscopy measurements were made at the Pacific Northwest Consortium's 169 (PNC-CAT) bending magnet beam line of the Advanced Photon Source. Samples were 170 packed in Teflon sample holders after mixing with boron nitride to achieve a sample 171 thickness of about one absorption length. Each sample was scanned 5 to 8 times at room 172 temperature in both the transmission and fluorescence modes. Data reduction and 173 analysis were accomplished by ATHENA and ARTEMIS (Ravel and Newville 2005) 174 with data analysis considering the theoretical phase and amplitude functions generated in 175 FEFF6 (Zabinsky et al. 1993).

176 The samples were subjected to the Toxicity Characteristics Leaching Procedure 177 (TCLP) test (USEPA 1992). Due to the homogenous and fine-grained nature of the 178 precipitates, only 5 g samples of the precipitates were contacted for 20 hours at room 179 temperature by 100 mL of deionised water buffered at pH 5. Prior to the tests, the 180 samples were water-washed to remove any possible remaining soluble secondary salts 181 and dried at 100°C. The residue materials following the TCLP tests were filtered through 182 0.1 µm filters and the solutions were analysed for As, Fe and S by ICP-AES (Varian, 183 Vista RL) and ICP-MS (Thermo Scientific X-series 2) techniques.

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## **RESULTS AND DISCUSSION**

## 187 Long-range structures and chemical compositions

188 The chemical composition of the precipitates formed from solutions with initial Fe/As molar ratios >1.6 is highly variable from 29.0 to 33.3 wt % Fe, 0.1 to 43.2 wt % AsO<sub>4</sub> 189 190 and 15.6 to 56.6 wt % SO<sub>4</sub>. In terms of the Fe, AsO<sub>4</sub> and SO<sub>4</sub> molar contents, the 191 variation is essentially continuous as illustrated in Figure 1. The composition of the 192 samples projects from the point with a Fe:SO<sub>4</sub> ratio of 1 to  $Fe(AsO_4)_{0.5}(SO_4)_{0.5}$ , but 193 deviates towards Fe<sub>1.5</sub>AsO<sub>4</sub> as the AsO<sub>4</sub> content increases and the SO<sub>4</sub> content decreases. This trend requires that OH initially decreases from 1 to 0.5 in unison with the decrease 194 in SO<sub>4</sub>, and this substitution can be written as  $SO_4^{2-} + OH^- = AsO_4^{3-}$ . In this case, the 195 change in the formula would be from Fe(SO<sub>4</sub>)(OH) to Fe(AsO<sub>4</sub>)<sub>0.5</sub>(SO<sub>4</sub>)<sub>0.5</sub>(OH)<sub>0.5</sub>. For the 196 197 compositions in which AsO<sub>4</sub> dominates over SO<sub>4</sub>, the formulae change between 198  $Fe(AsO_4)_{0.5}(SO_4)_{0.5}(OH)_{0.5}$  and  $Fe_{1.5}AsO_4(OH)_{1.5}$ . This substitution can be formally written as  $SO_4^{2-} = Fe^{3+} + AsO_4^{3-} + 2OH^-$ . Based on the powder XRD patterns (Fig. 2), 199 the precipitates were grouped as Type-2, Phase 3 and Phase 5. In the following text, we 200 201 describe the structures of all the identified phases and the variations thereof with the 202 observed AsO<sub>4</sub>-SO<sub>4</sub> substitutions. We begin with the sulfur-rich phases and proceed 203 toward the most arsenic-rich one, as shown in Figure 1.

Jarosite. Jarosite, nominally  $KFe_3(SO_4)_2(OH)_6$ , has a dense heteropolyhedral structure with octahedral kagomé sheets, sulfate tetrahedra, and monovalent cations (K<sup>+</sup>, but several other ions including  $H_3O^+$ ) in twelve-fold coordination (Fig. 3a). Because of its relatively dense atomic arrangement, the AsO<sub>4</sub>-SO<sub>4</sub> substitution is limited and the maximum was reported to be 17 % AsO<sub>4</sub>/(AsO<sub>4</sub>+SO<sub>4</sub>) (Paktunc and Dutrizac 2003). These authors investigated the incorporation of As<sup>5+</sup> in the jarosite structure in detail and their analyses and conclusions will not be repeated here.

Basic ferric sulfate (BFS). Another nominally arsenic-free phase with limited AsO<sub>4</sub>-SO<sub>4</sub> substitution is Fe(SO<sub>4</sub>)(OH), hereafter referred to as basic ferric sulfate (BFS). The structure of this phase was solved by Johansson (1962) and was shown to be in the orthorhombic space group *Pnma*. Ventruti et al. (2005) investigated the order-disorder nature of the structure of BFS and showed that the orthorhombic phase of Johansson (1962) is one of two maximum-degree-of-order (MDO) phases. The other MDO

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217  $Fe(SO_4)(OH)$  phase is monoclinic, space group  $P2_1/c$ . Small changes in the stacking 218 order between the two MDO phases results in small modifications in the powder XRD 219 patterns. These changes, especially visible in the intensities of selected peaks, can be qualitatively traced in our XRD data. Structures of both MDO Fe(SO<sub>4</sub>)(OH) phases are 220 heteropolyhedral frameworks. In contrast to jarosite, the  $Fe^{3+}$  octahedra in  $Fe(SO_4)(OH)$ 221 222 are arranged in chains and cross-linked by the tetrahedral units (Fig. 3b). A peculiar 223 change in lattice parameters upon substitution of  $AsO_4$  for  $SO_4$  can be observed (Fig. 4). 224 With the addition of a small amount of AsO<sub>4</sub>, the unit-cell volume initially decreases 225 despite the fact that the AsO<sub>4</sub> tetrahedron is significantly larger than the SO<sub>4</sub> tetrahedron. 226 Only upon further insertion of AsO<sub>4</sub> into the structure does the unit-cell volume begin to 227 expand. A closer inspection of the lattice parameters (in the orthorhombic phase setting 228 *Pnma*) reveals the non-linear change of a and near linear expansion of b and c (Fig. 4b). 229 This observation is interesting because the octahedral chains extend in the direction of the 230 a axis. Tetrahedra are bridging FeO octahedra along the chains and the shared oxygen 231 atoms are perfectly aligned with the oxygen atoms linking the octahedra; therefore, 232 substitution of smaller sulfate by the larger arsenate (e.g. S-O=1.47 Å vs. As-O=1.68 Å) 233 would translate to the largest increases along the *a* axis. Three adjacent octahedral chains 234 are cross-linked or anchored by the tetrahedra; therefore, expansions along the b and c 235 axes would be restricted. It seems that with a small amount of AsO<sub>4</sub>, the chains are 236 allowed to relax and actually contract. The expansion of the unit cell (i.e., the octahedral 237 chains) begins only with higher amounts of AsO<sub>4</sub>, at 2 % AsO<sub>4</sub>/(AsO<sub>4</sub>+SO<sub>4</sub>). It appears that BFS is able to accommodate up to about 10 wt % AsO4, which corresponds to the 238  $X_{As}$  mole fraction of 0.12. Interestingly, this is similar to the limit of AsO<sub>4</sub>-SO<sub>4</sub> 239 240 substitution in jarosite (Paktunc and Dutrizac 2003).

Our powder XRD data suggest that the BFS samples contain a minor amount of hydronium jarosite. In addition, the BFS samples with the highest As content contain Phase 3, which is chemically and structurally related to BFS and is described in detail below.

Ferric arsenate sulfate (FAS). Phase 3 which spans most of the mixed SO<sub>4</sub>-AsO<sub>4</sub> compositions (Fig. 1) is a ferric arsenate-sulfate (FAS). Structure of this phase was not known previously, but the powder XRD data bear a remarkable similarity to the 248 monoclinic Fe(SO<sub>4</sub>)(OH). Despite this similarity, the structure solution of FAS turned out
249 to be a difficult problem.

250 Early attempts to index the synchrotron powder XRD data failed because of the small 251 amount of impurities in the samples. TEM examinations indicated that FAS is the major 252 phase and occurs as isometric or blocky crystals, often twinned (Fig. 5). Energy-253 dispersive X-ray microanalyses indicate that the crystals have a variable S:As ratio in the 254 range of 0.7-1. The electron diffraction tomography data were collected on a small crystal 255 with dimensions 130x180 nanometres. In total, 118 diffraction patterns with exposure 256 time 2 s were collected. The data analysis revealed that the FAS is triclinic (pseudo-257 orthorhombic), with approximate lattice parameters a=5.42 Å, b=5.44 Å, c=13.58 Å, and  $\alpha$ =89.21°,  $\beta$ =89.68°,  $\gamma$ =87.89°, space group *I*-1. Just as BFS, the structure of FAS is 258 constructed from octahedral Fe<sup>3+</sup> chains which are cross-linked by sulfate and arsenate 259 tetrahedra (Fig. 6). The atomic positions of the structure as refined against the electron 260 261 diffraction data are given in Table 2. It should be emphasized that this structure is only 262 approximate, as the refinement does not take into account the dynamical scattering 263 effects. However, the experience with the refinement from precession electron diffraction 264 data shows that the errors in the refined atomic positions often range around 0.1-0.2 Å, 265 and very rarely exceed 0.5 Å (Gorelik et al. 2011).

The lattice parameters refined from the X-ray powder diffraction data are a=5.1663, b=5.1892, c=12.917 Å and  $\alpha=89.000^{\circ}$ ,  $\beta=89.645^{\circ}$ ,  $\gamma=87.868^{\circ}$ . These refined parameters agree well with those determined from electron diffraction data supporting the triclinic symmetry with the exception of an overall scaling factor of ~1.05, resulting from imperfect calibration of the electron diffraction patterns.

The FAS phase precipitates from solutions with initial Fe/As molar concentrations  $\geq 1.5$  at temperatures between 180 and 220 °C. It is the most abundant precipitate with  $X_{As}$ mole fractions varying from 0.26 to 0.63. The lower end of this range seems to correspond to higher synthesis temperatures.

Profile fitting of the in-house and synchrotron powder XRD data obtained from the samples with variable  $X_{As}$  mole fractions shows an intriguing trend (Fig. 4, Table 3). As was the situation for the BFS, the unit-cell volume does not change linearly with increasing amounts of AsO<sub>4</sub> in the structure. Within the scatter of the data, the unit-cell volume increases linearly up to  $X_{As} \sim 0.55$ , but then levels off for the most As-rich compositions. These changes are most likely related to the capability of the octahedra within the chains to rotate and adjust for the attachment of the sulfate or arsenate tetrahedra.

283 Type-2 synthesized by Swash and Monhemius (1994) with the  $X_{As}$  mole fraction of 284 0.75 has an XRD pattern similar to that of FAS, and this indicates that they are essentially 285 the same compound (Fig. 2). Electron microprobe analyses of the material synthesized by 286 Swash and Monhemius (1994) revealed that Type-2 is not homogeneous. The grains 287 showing hedge-hog type morphology display zoning with respect to As and S (Fig. 7a). 288 Some of the grains have margins with increased As and decreased S contents. In addition, 289 the X-ray maps show sulfur-deficient rosettes within the grains indicating a ferric 290 arsenate phase exsolved from the compound (Fig. 7b). Electron microprobe analyses 291 indicate that the exsolved phase has  $X_{As}$  mole fractions varying from 0.77 to 0.96 with an 292 average value of 0.90±0.05; however, it is possible that this exsolved phase is nearly 293 sulfate-free. Furthermore, because of the small size, it is likely that some of the S counts 294 in the microanalyses originate from the host phase. The  $X_{As}$  mole fraction of the main 295 phase is 0.69±0.05. The structural formula of the dominant phase is determined to be 296 Fe<sub>1.1-1.3</sub>(As<sub>0.6-0.8</sub>S<sub>0.2-0.4</sub>O<sub>4</sub>)(OH)<sub>0.2-0.4</sub>. It appears that this composition represents an end-297 member of FAS with the highest  $X_{As}$  fraction in the structure. The structural formula of 298 the exsolved phase is  $Fe_{1,1-1,5}(As_{0,8-1,0}S_{0,0-0,2}O_4)(OH)_{0,0-0,2}$ . In summary, the composition of 299 FAS can be defined as  $Fe(AsO_4)_{0,2-0,7}(SO_4)_{0,7-0,2}(OH)_{0,7-0,2}$  where  $AsO_4+SO_4=1$ .

Scorodite. Scorodite, nominally  $FeAsO_4 \cdot 2H_2O_5$ , is another heteropolyhedral network among the studied phases. The structure consists of monomers of  $FeO_4(H_2O)_6$  polyhedra which attach via the oxygen ligands to the four neighboring  $AsO_4$  tetrahedra (Fig.3c). Little  $SO_4$  can be incorporated into the scorodite structure and from the limited data acquired herein, it seems that the unit cell contracts monotonically with increasing  $SO_4$ substitution (Table 3).

**Ferric orthoarsenate hydrate.** In our synthesis experiments,  $FeAsO_4 \cdot 0.75H_2O$  is the phase with the least amount of SO<sub>4</sub>. The structure of this phase was reported by Jakeman et al. (1991). The structure is built by kinked tetramers of edge-sharing of  $Fe(O,H_2O)_6$ octahedra (Fig.3d). The tetramers are interlinked by the AsO<sub>4</sub> tetrahedra in a tightly knitted heteropolyhedral framework which probably does not leave much room for the substitution of the smaller  $SO_4$  group. This phase forms when the Fe/As molar ratio of the initial solution is low (i.e., near 1) and at synthesis temperatures of 180°C and greater. At lower temperatures, between 150 and 175 °C, the precipitate is scorodite.

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## 315 Short-range structures

The  $k^3$ -weighted As K-edge spectra of BFS and FAS samples are broadly similar 316 (Fig. 8a). The  $k^3$ -weighted Fe K-edge XAFS spectra also display similarities (Fig. 8b). 317 318 The changes in the spectra point to the progressive variation of the  $X_{As}$  mole fractions in 319 the precipitates. This indicates that the local structures around the As and Fe atoms in 320 BFS and FAS are similar, and this confirms the continuity in the long-range structures 321 discussed earlier. The gradual changes in the oscillation features of the Fe K-edge XAFS 322 spectra reflect the differences in the number of As and S atoms surrounding the central Fe 323 atoms. For instance, the gradual development of humps on the low k sides of the oscillation features at ~4, 6 and 8 Å<sup>-1</sup> in samples with  $X_{As}$  mole fractions  $\leq 0.35$  suggests 324 the influence of S. The changes in the As K-edge XAFS spectra are not as apparent 325 326 suggesting that the local As environment is not significantly affected by the changes in the  $X_{As}$  mole fraction. Instead, it appears that Fe is the closest neighbour surrounding the 327 328 AsO<sub>4</sub> tetrahedra in both compounds. Considering these similarities and the gradual 329 changes in the spectral features, the XAFS fitting was intended to test the similarity of the 330 compounds to the Fe(SO<sub>4</sub>)(OH) structure in terms of local structures around the As and 331 Fe atoms. In the Fe(SO<sub>4</sub>)(OH) structure, a central Fe atom is surrounded by four S atoms 332 at 3.31 Å and two Fe atoms at 3.67 Å, whereas a central S atom, which is assumed to be 333 substituted by As in this case, is surrounded by four Fe atoms at 3.31 Å and four S atoms at 4.09 Å. 334

Local structural parameters determined by As and Fe XAFS fitting for the BFS and FAS compounds are given in Tables 4 and 5, and the fits are shown in Figure 8. By limiting the number of variables through constraining the coordination numbers from the crystal structure, it was possible to include up to 4 coordination shells in fitting. This allowed an independent assessment of the interatomic distances around As and Fe atoms and the changes due to arsenate-for-sulfate substitution. The As-O radial distances are

341 relatively uniform at 1.68±0.01 and 1.69±0.01 Å. Beyond the oxygen shell, spectral 342 features at 3.36±0.02 (3.34-3.39), 3.73±0.02 (3.70-3.76) and 4.03±0.03 (3.99-4.07) Å 343 correspond to Fe, S and As atoms, respectively. As expected, the variations in the As-Fe, 344 As-S and As-As distances do not appear to be influenced by the changes in the  $X_{As}$  mole 345 fraction in the structures. The As-S radial distances are shorter than those of the As-As 346 due to smaller size of the sulfate tetrahedra. Fit quality was improved with the inclusion 347 of the multiple scattering paths, As-O1-O2, As-O1-As-O2 and As-O1-As-O1 derived 348 from FEFF calculations (Ankudinov et al. 1998) based on the crystal structure of 349 scorodite from Kitahama et al. (1975).

350 Fitting of the Fe K-edge XAFS spectrum of the BFS samples resulted in Fe-O 351 distances of 1.99±0.01 - 2.00±0.01 Å in octahedral coordination and Fe-S distances of  $3.34\pm0.03 - 3.38\pm0.03$  Å (Table 5). In contrast, the FAS samples have slightly shorter 352 353 Fe-O distances at 1.96±0.01 and 1.97±0.01 Å. Because of the similarity of the Fe-As and 354 Fe-S interatomic distances, simulations of the second shell were performed with As for 355 the FAS samples which resulted in Fe-As distances of  $3.31\pm0.02 - 3.34\pm0.02$  Å. These are comparable to the As-Fe distances of 3.34±0.02 - 3.39±0.03 Å from the As-XAFS 356 357 and the nominal FAS distances varying from 3.25 to 3.38 Å (3.33±0.05 Å). Constraining 358 the coordination numbers to their structural values made it possible to include two Fe 359 shells in the simulations. The first Fe shell with a coordination number of 2 is at 360  $3.69\pm0.03 - 3.71\pm0.05$  Å for the BFS and  $3.62\pm0.10 - 3.68\pm0.04$  Å for the FAS samples, 361 representing corner-linked FeO<sub>6</sub> octahedra along the octahedral chains. The second Fe 362 shell with a coordination number of 8 is at  $5.10\pm0.03 - 5.12\pm0.02$  Å for the BFS and  $5.14\pm0.04 - 5.18\pm0.04$  Å for the FAS samples representing corner-linked FeO<sub>6</sub> dimers in 363 364 the four adjacent octahedral chains. These Fe-Fe radial distances are comparable to the 365 Fe-Fe distances of 3.67 and 5.14 Å of the Fe(SO<sub>4</sub>)(OH) structure (Johansson 1962), and 366 3.76 and 5.17 Å of the FAS structure (this study). The increased Fe-Fe2 distances of the 367 FAS samples indicate that the arsenate-for-sulfate substitution causes increases in distance between the octahedral chains which reflect the increases along a and b. 368

In summary, the XAFS-derived interatomic distances are comparable to their counterparts from the crystal structures of FAS and BFS. The average S-O distances of 1.457 Å in Fe(SO<sub>4</sub>)(OH) and 1.473 Å in sulfate minerals (Hawthorne et al. 2000) are shorter by about 0.22 Å from the measured As-O distance of 1.68 Å. It is possible that the observed differences between the As-As and As-S distances, ranging from 0.29 $\pm$ 0.06 to 0.32 $\pm$ 0.07 Å, are reflecting this difference. In addition, substitution of sulfate ions by the larger arsenate ions would cause the Fe-Fe distance to become greater, lengthening the iron chains and widening the distance between the octahedral chains. This would explain the gradual shifts to lower 20 or wider *d*-spacings and the enlargement of the unit-cell volume with increased arsenate substitution.

379

## 380 Leachability

381 TCLP testing was used to make a preliminary assessment of the environmental 382 stability of the BFS and FAS precipitates. Materials that produce effluents with As 383 concentrations exceeding the Canadian metal-mine effluents limit of 0.5 mg/L 384 (Government of Canada 2002) could be considered unacceptable for safe disposal in the 385 environment, and even materials which pass this short-term test often require testing for longer times at various pH values and under a range of oxidation/reduction conditions. 386 387 TCLP testing of the BFS with an  $X_{As}$  mole fraction of 0.08 resulted in an As concentration of 16.2 mg/L in the leachate. The FAS samples produced As concentrations 388 389 that varied between 0.06 and 0.88 mg/L. The upper end of this range is represented by the 390 sample with the highest  $X_{As}$  molar fraction (i.e., 0.62). Swash and Monhemius (1994) 391 reported that Type-2 with an  $X_{As}$  mole fraction of 0.75 produced a similar As 392 concentration (i.e., 0.8 mg/L) in the TCLP leachate at pH 5. Some of the FAS samples 393 contain trace to minor amounts of BFS, as evidenced by the presence of overlapping peaks at 20 values of  $\sim 26.5^{\circ}$  and 27.5° which appear as asymmetric and broadened peaks. 394 395 The BFS samples containing minor amounts of FAS have leachate As concentrations 396 between 0.46 and 2.26 mg/L. Arsenic concentrations in the leachates of the FAS samples 397 increase exponentially with a decrease in the leachate Fe concentration (Fig. 9), although 398 all the As concentrations are less than 1 mg/L. This is with the exception of the samples 399 containing minor amounts of BFS. In contrast, the As and Fe concentrations in the 400 leachates of BFS and those containing both compounds display a positive correlation. 401 Leachate S concentrations are variable from 55.6 to 252.7 mg/L in the samples containing 402 BFS and less than 1 to 39.6 mg/L in the FAS samples, forming a trend of increasing Fe

and S concentrations from both types of precipitates (Fig. 9). The highest releases of As,
Fe and S are from BFS. Leachate Fe concentrations of 13 to 259 mg/L released from BFS
and the FAS samples containing BFS are unusually high for solutions at the TCLP test
value of pH 4.5.

407 The FAS samples with  $X_{As}$  mole fractions of 0.40, 0.42 and 0.52 gave the lowest As 408 concentrations in the leachates (i.e., 0.063-0.069 mg/L) and also yielded molar Fe/As 409 ratios that are greater than 20. These samples contain trace amounts of BFS. For 410 precipitates with molar Fe/As+S ratios between 0.002 and 0.572, the releases of Fe, As 411 and S from both the FAS and BFS precipitates are non-stoichiometric. The highest ratio 412 is from a BFS sample indicating that dissolution of BFS releases about twice as much As 413 and S as Fe. With the exception of the samples containing minor to trace BFS, the molar 414 Fe/As+S ratios of the TCLP extracts of the FAS samples are less than 0.008 indicating 415 that very little iron is released in comparison to the combined As and S releases from the 416 dissolution of FAS. In consideration of the crystal structure of BFS, the non-417 stoichiometric dissolution behaviour can be explained by the precipitation of a ferric 418 oxyhydroxide phase or by the preferential release of SO<sub>4</sub> leaving behind chains of FeO<sub>6</sub> 419 octahedra.

420

## 421 Implications for the disposal of gold processing residues

422 Fine grained residue samples from three autoclaves treating refractory gold ores were 423 obtained. In terms of bulk chemical compositions, the sample designated as POX-2 424 analyzed 22.7% Fe, 5.42% As and 3.90% SO<sub>4</sub>. The POX-9 sample analyzed 12.9% Fe, 425 2.39% As and 1.40% SO<sub>4</sub>; the analysis of the POX-14 sample was 20.0% Fe, 4.01% As 426 and 3.50% SO<sub>4</sub>. Electron microprobe analyses of the heterogeneous oxidation residues 427 indicate that the arsenic carrier in the POX-2 sample is a FAS compound (Fig. 10). The 428 POX-9 sample has jarosite, hematite and goethite as the main As carriers. The jarosite 429 has significant amount of Al and contains between 0.5 and 8.6 wt % As<sub>2</sub>O<sub>5</sub>. Hematite has 430 3.6 to 6.1 wt % As<sub>2</sub>O<sub>5</sub> whereas goethite contains much higher As concentrations, in the 431 7.9 to 23.5 wt % range. Goethite, with As<sub>2</sub>O<sub>5</sub> concentrations in the 0.1 to 14.9 wt % 432 range, is the dominant As carrier in the POX-14 sample. The compositional similarity of 433 the POX-2 residues to the As-rich members of Phase 3 or Type-2 is apparent on the Fe434 AsO<sub>4</sub>-SO<sub>4</sub> plot (Fig. 10). Spherical ferric arsenate-sulfate grains from the Red Lake 435 autoclave (McCredie et al. 1998) have a similar composition. The POX-14 data cluster 436 near the Fe<sup>3+</sup> corner along the Fe-AsO<sub>4</sub> join in agreement with the compositions of As-437 adsorbed goethite and hematite. The arsenic carriers in the POX-9 residue plot with those 438 of POX-14 and are close to the Fe<sup>3+</sup> corner and near the mid point along the Fe-SO<sub>4</sub> join 439 close to the theoretical jarosite composition. In terms of contained As, these residues are 440 similar to the synthetic arsenical jarosites reported by Paktunc and Dutrizac (2003).

441 In agreement with their chemical compositions, analysis of the As-XAFS spectra of 442 the POX-2 and POX-14 residue sample (Fig. 11) yielded local structural parameters that 443 are similar to those of FAS and goethite, respectively (Table 6). POX-9 has an optimized 444 iron coordination number of 2 but the As-Fe radial distance is rather long at 3.40±0.10 Å in comparison to the 3.24-3.32 Å distances for goethite (Foster et al. 1998; Sherman and 445 446 Randall 2003; Paktunc et al. 2004; 2008), 3.25-3.26 Å for jarosite (Paktunc and Dutrizac 447 2003), and 3.24-3.35 Å for hematite (Sherman and Randall 2003). However, considering 448 the large uncertainty associated with the measured As-Fe distance for POX-9 and the 449 number of As carriers in the sample, the calculated "average" As-Fe distance is not 450 unreasonable.

451 Minerals in the Fe-AsO<sub>4</sub>-SO<sub>4</sub> system are also shown on the Fe-AsO<sub>4</sub>-SO<sub>4</sub> plot (Fig. 452 10). The minerals are essentially restricted to near Fe/anion ratios of 1 and 1.5. 453 Kaatialaite with a Fe/As molar ratio of 1/3 is the exception but this is a very rare mineral 454 found only in extremely acidic environments. Other than scorodite, jarosite, hematite and 455 goethite, the compounds that form during pressure oxidation do not have natural analogs. 456 There are three ferric arsenate-sulfate minerals, bukovskýite 457  $(Fe(AsO_4)_{0.5}(SO_4)_{0.5}(OH)_{0.5} \cdot 4.5H_2O)$ , sarmientite  $(Fe(AsO_4)_{0.5}(SO_4)_{0.5}(OH)_{0.5} \cdot 2.5H_2O)$ 458 and zykaite ( $Fe_4(AsO_4)_3(SO_4)(OH) \cdot 15H_2O$ ). The exsolved ferric arsenate phase present in 459 the FAS sample with the highest As content is similar to ferrisymplesite 460 (Fe<sub>3</sub>(AsO<sub>4</sub>)<sub>2</sub>(OH)<sub>3</sub>·5H<sub>2</sub>O in terms of the Fe/As ratio. Ferrisymplesite is an amorphous 461 ferric arsenate supposedly resulting from the oxidation of symplesite, but little is known 462 about this compound and its existence as a valid mineral species may be questionable. 463 Bukovskýite and sarmientite could be considered as hydrated natural analogs of FAS 464 with equal numbers of AsO<sub>4</sub> and SO<sub>4</sub> molecules in the structure. Along with zykaite and 465 kaatialaite, these are uncommon arsenate minerals, occurring as alteration products of 466 scorodite and arsenopyrite in several mine waste sites. The As-XAFS spectra of two 467 bukovskýite specimens from Kutná Hora (Czech Republic) and the Ketza River mine in 468 Yukon (Canada) display similarities to those of FAS (Fig. 11). Local structural 469 parameters of bukovskýite determined by As-XAFS are four O at 1.69±0.01 Å, four Fe at 470 3.33±0.03 Å and two As at 3.97±0.07 Å (Table 6). Fitting of the Fe-XAFS spectra (Fig. 11b) resulted in six O at 1.99±0.01 Å, two As at 3.35±0.02 Å and one Fe at 3.61±0.03 Å 471 472 and one Fe at 5.09±0.05 Å (Table 7). These independently determined local structures are 473 reasonable in consideration of the crystal structure of bukovskýite (Majzlan et al. 2012a) 474 and they are broadly similar to the local structures of FAS, indicating that bukovskýite 475 and FAS have similar local structures around As and Fe in the sense that arsenate 476 tetrahedra link octahedral "chains" which are only dimers in bukovskýite.

477 Most of the FAS samples have TCLP As concentrations that are less than the 478 Canadian metal-mine effluent limit of 0.5 mg/L (Government of Canada 2002) 479 suggesting that FAS could meet the current regulatory guideline for mine effluents at pH 480 5. The precipitates having  $X_{As}$  mole fractions between 0.4 and 0.5 could even meet 481 stricter mine effluent limit of 0.2 mg/L imposed by the Provincial Government of Québec 482 for new operations (Gouvernement du Québec 2005). In contrast, the BFS with an  $X_{As}$ mole fraction of 0.08 yields 16.2 mg/L As in the TCLP leachate. This is only a 483 484 preliminary assessment, however, as these short-term tests may not represent the 485 solubility or near equilibrium conditions. Type-2 which has the highest As content and an 486 exsolved ferric arsenate compound, has As leachate values of less <0.5 to 7.3 mg/L, 487 determined at pH values of 3, 5, 7 and 10 by Swash and Monhemius (1994). This 488 suggests that As control by FAS precipitation would be acceptable only over the limited 489 pH range from 3 to 5. Nevertheless, these preliminary assessment results for FAS are 490 promising in that they are comparable to the solubility levels of scorodite (Krause and 491 Ettel 1989; Paktunc and Bruggeman 2010; Majzlan et al. 2012b).

Limited leaching tests indicate that the basic orthoarsenate hydrate (i.e., Phase 4 of Dutrizac and Jambor 2007 and the Type-1 compounds of Swash and Monhemius 1994) would not be suitable for disposal. Similarly, BFS would be an unwanted compound because of its high solubility which would result in the release of much of the contained 496 As. In addition to environmental concerns, the formation of Fe(SO<sub>4</sub>)(OH) can cause 497 problems during subsequent cyanidation and impact on the economics of the process by 498 causing excessive lime consumption during cyanidation and potentially exposing the 499 plant workers to hazardous HCN gas as pointed out by Fleming (2010). According to 500 Fleming et al. (2008), the autoclave temperatures should be around 190°C for the 501 optimum recovery of gold from the refractory Kittila ore. Higher temperatures, around 502  $210-230^{\circ}$ C, promote the formation of Fe(SO<sub>4</sub>)(OH). Accordingly, As control practices at 503 mine sites could aim to optimize the conditions for the formation of FAS at the lower end 504 of the temperature range (185-225°C) during pressure oxidation, without compromising 505 gold recoveries.

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632	
633	FIGURE CAPTIONS
634	FIGURE 1. Compositions of Phase 3, Phase 4, Phase 5 and Type-2 in terms of molar
635	$\mathrm{Fe}^{3+}$ , AsO <sub>4</sub> and SO <sub>4</sub> contents. Note that only the upper half of the plot is shown as Fe-
636	0.5AsO <sub>4</sub> -0.5SO <sub>4</sub> . Type-2 plots are based on electron microprobe analyses. Cores are
637	represented by blue stars and exsolution-like features by purple stars (Type-2 exs).
638	Synthetic jarosite data from Paktunc and Dutrizac (2003) and scorodite from Dutrizac
639	and Jambor (1970).
640	FIGURE 2. XRD patterns of Phase 3, Phase 5 and Type-2. Numbers next to the patterns
641	are $X_{As}$ mole fractions of the samples.
642	FIGURE 3. Polyhedral representation of the crystal structures of (a) jarosite, (b) BFS, (c)
643	scorodite, (d) FeAsO <sub>4</sub> ·0.75H <sub>2</sub> O. Fe: brown; As: purple; S: yellow; K: green.
644	FIGURE 4. (a) changes in unit-cell volume as a function of substitution of AsO <sub>4</sub> for SO <sub>4</sub>
645	in BFS and FAS, (b) changes in the <i>a</i> , <i>b</i> and <i>c</i> lattice parameters of BFS (normalized to
646	XAs0 parameters) as a function of the AsO <sub>4</sub> fraction.

FIGURE 5. TEM photomicrographs of the FAS precipitates. The acicular crystal marked

- 648 04 is ferric orthoarsenate hydrate which occurs in trace amounts in the sample. The other 649 labels indicate the spots where microanalysis of the FAS crystals was performed. FIGURE 6. Crystal structure of FAS as visualized projected along a (a) and along the 650 651 FeO<sub>6</sub> octahedral chains (b). A comparative visualization of the crystal structures of FAS 652 (c) and BFS (d), showing the rotation of the  $FeO_6$  octahedra and  $AsO_4/SO_4$  tetrahedra in 653 FAS. Fe: brown; As: purple; S: yellow. 654 FIGURE 7. (a) Backscattered electron (BSE) image and (b) X-ray maps of As and S 655 showing depletion of As along the outer margin of Type-2 grains and its enrichment in 656 exsolution-like small spotty areas or circular zones in the interior. Note that the changes 657 in S are inversely related to those of As. Scale bar of X-ray maps is 100 µm. **FIGURE 8**. (a)  $k^3$ -weighted As XAFS spectra and (b)  $k^3$ -weighted Fe XAFS spectra of 658
- BFS and FAS. Numbers next to the curves are As mole fractions:  $100 \times AsO_4/(AsO_4+SO_4)$
- with 0 to 12 being BFS and 34 to 75 being FAS. Experimental curves indicated by solid
- 661 lines and fitted spectra by circles.
- **FIGURE 9**. (a) As and Fe concentrations (mg/L) and (b) Fe and S concentrations (mg/L)
- 663 in the leachates resulting from TCLP testing.
- **FIGURE 10**. Fe<sup>3+</sup>-AsO<sub>4</sub>-SO<sub>4</sub> plot showing the compositions of the pressure-oxidation
- 665 (POX) residues from three gold mining operations. POX-RL represents ferric arsenate-
- sulfate grains from the Red Lake autoclave. Also shown for comparative purposes are the
- 667 representative mineral compositions and the synthesized compounds.
- **FIGURE 11**. (a)  $k^3$ -weighted As-XAFS spectra of the pressure oxidation residues from
- 669 the three gold mining operations and of bukovskýite. (b)  $k^3$ -weighted Fe-XAFS spectra of
- 670 bukovskýite. Experimental spectra shown by solid curves and fitted spectra by circles.
- 671 Bukovskýite specimens from Kutna Hora (bukKH) and the Ketza River mine (bukKe).
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# TABLE 1. Compositions of precipitates synthesized in the Fe-AsO<sub>4</sub>-SO<sub>4</sub> system

Compound	Formula	Temperature (°C)	Ref.	Structure*
Type-1	Fe <sub>2</sub> (HAsO <sub>4</sub> ) <sub>3</sub> ·zH <sub>2</sub> O (z<4)	125-225	1	
Ferric orthoarsenate hydrate	FeAsO <sub>4</sub> ·¾H <sub>2</sub> O	700-800	2	
Phase 4	$Fe_{0.96}(AsO_4)_{0.99}(SO_4)_{0.01}(H_2O)_{0.54}$	205-215	3	
Туре-2	Fe <sub>4</sub> (AsO <sub>4</sub> ) <sub>3</sub> (SO <sub>4</sub> ) <sub>y</sub> (OH) <sub>x</sub> (x+2y=3)	175-225	1	FAS
Phase 3	Fe <sub>0.8-1.1</sub> (AsO <sub>4</sub> ) <sub>0.25-0.54</sub> (SO <sub>4</sub> ) <sub>0.46-</sub> <sub>0.75</sub> (OH) <sub>0.46-0.75</sub>	185-225	3,4	FAS
Basic iron sulfate	Fe(SO <sub>4</sub> )(OH)	200	5	BFS
Basic iron sulfate	Fe(SO <sub>4</sub> )(OH)	>175	1	BFS
Phase 5	Fe(SO <sub>4</sub> )(OH)	205-215	3,4	BFS
Scorodite	FeAsO <sub>4</sub> ·2H <sub>2</sub> O	<175	1,2,6	
Jarosite	$KFe_3(SO_4)_2(OH)_6$	98	7	

676 1: Swash and Monhemius (1994); 2: Jakeman et al. (1991); 3: Dutrizac and Jambor (2007); 4: present study;
677 5: Johansson (1962); 6: Paktunc et al. (2008); 7: Paktunc and Dutrizac (2003); Structure\*: in reference to the abbreviations used in text.

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#### 682 TABLE 2. Lattice parameters, space group and atomic coordinates of ferric

arsenate sulfate (FAS), as refined from the electron diffraction data. 683

684

685 Space group I-1

Lattice parameters: a = 5.1639 Å, b = 5.1863 Å, c = 12.9457 Å,  $\alpha = 89.209^{\circ}$ ,  $\beta = 89.675^{\circ}$ ,  $\gamma = 87.892^{\circ}$ , V = 346.433 Å<sup>3</sup> 686 687

V = 346.433  A				
atom	Х	у	Z	U <sub>iso</sub>
As/S*	0.7474(14)	0.018(2)	0.1227(6)	0.0038(18)
Fe1	0.25	-0.25	0.25	0.038(4)
Fe2	0	0.5	0	0.024(3)
01	0.782(5)	-0.190(6)	0.0442(18)	0.026(6)
O2	0.746(5)	0.309(6)	0.083(2)	0.028(6)
O3	0.471(4)	0.014(6)	0.1906(17)	0.023(5)
O4	0.949(9)	-0.001(10)	0.208(3)	0.069(10)
OH	0.241(7)	-0.449(9)	0.123(3)	0.065(10)

\* occupancy 50 % As, 50 % S

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#### 693 TABLE 3. Lattice parameters of basic ferric sulfate (BFS) and ferric arsenatesulfate (FAS)

	а	b	С	α	β	Ŷ	V (Å <sup>3</sup> )
BFS							
XAs0	7.3434(4)	6.4133(4)	7.1330(4)				335.93(3
XAs1	7.3353(4)	6.4143(3)	7.1377(4)				335.84(3
XAs2	7.3357(4)	6.4145(4)	7.1359(4)				335.78(3
XAs4	7.3431(4)	6.4178(4)	7.1393(4)				336.45(3
XAs8	7.3565(6)	6.4215(5)	7.1407(6)				337.33(5
XAs12	7.3936(7)	6.4261(6)	7.1414(6)				339.30(5
Nominal(OR)	7.33	6.42	7.14				336.00
Nominal(MO)	7.33	7.14	7.39		119.7		295.27
FAS							
XAs26	5.1744(3)	5.1644(3)	12.9929(9)	89.641(5)	89.910(5)	87.852(4)	346.95(4
XAs30	5.1841(5)	5.1599(5)	13.0179(12)	89.518(6)	90.022(7)	87.939(5)	347.98(5
XAs32	5.1852(4)	5.1653(4)	13.0624(11)	89.437(4)	89.796(6)	87.837(4)	349.59(5
XAs33	5.1801(4)	5.1674(4)	13.0483(10)	89.503(5)	89.851(5)	87.873(4)	349.02(4
XAs34	5.1908(4)	5.1807(4)	13.0265(9)	89.599(4)	89.506(4)	87.692(4)	350.01(4
XAs35	5.1932(3)	5.1834(4)	12.9906(11)	89.530(6)	89.460(5)	87.633(4)	349.36(4
XAs36	5.1990(3)	5.1836(3)	12.9697(9)	89.581(4)	89.226(4)	87.590(3)	349.18(4
XAs37.1	5.2018(6)	5.1744(6)	13.091(2)	89.321(6)	90.15(1)	88.044(7)	352.14(7
XAs37.3	5.1835(3)	5.1739(3)	13.0515(8)	89.415(4)	89.818(4)	87.830(3)	349.76(3
XAs38	5.2082(6)	5.1721(6)	13.095(2)	89.409(6)	89.920(9)	87.999(7)	352.50(7
XAs40	5.1985(3)	5.1764(3)	13.1062(8)	89.427(4)	89.775(4)	87.870(3)	352.42(4
XAs42	5.1939(4)	5.1649(4)	13.1357(13)	89.267(5)	89.679(6)	87.771(5)	352.07(5
XAs44	5.2131(3)	5.2051(3)	13.0583(9)	89.556(4)	89.208(4)	87.544(3)	353.97(4
XAs45	5.2042(3)	5.1815(3)	13.0774(8)	89.600(4)	89.911(4)	87.996(3)	352.41(4
XAs46	5.2068(4)	5.1854(4)	13.0626(10)	89.615(5)	89.913(5)	88.023(5)	352.46(5
XAs46.6	5.2147(3)	5.1934(3)	13.0690(8)	89.621(4)	89.909(4)	88.129(4)	353.74(4
XAs47	5.2192(3)	5.1956(4)	13.1330(10)	89.366(4)	89.755(4)	87.861(4)	355.85(4
XAs48	5.2264(5)	5.1979(4)	13.1119(12)	89.550(6)	89.924(6)	88.040(5)	355.98(5
XAs51.3	5.2341(3)	5.2140(4)	13.0527(9)	89.564(5)	90.048(4)	88.340(4)	356.05(4
XAs51.6	5.2293(4)	5.1974(4)	13.1172(12)	89.487(5)	89.929(6)	88.122(5)	356.31(5
XAs51.8	5.2319(4)	5.2142(4)	13.0881(10)	89.731(6)	89.952(6)	88.186(4)	356.86(5
XAs52.0	5.2124(3)	5.1960(4)	13.1187(9)	89.464(5)	89.621(5)	87.835(5)	355.03(4
XAs52.1	5.2059(5)	5.1849(4)	13.1244(11)	89.381(5)	89.679(6)	87.760(5)	353.96(5
XAs52.2	5.2219(3)	5.1946(4)	13.1404(9)	89.408(4)	89.809(4)	87.930(4)	356.19(4
XAs52.3	5.2118(5)	5.1898(5)	13.1267(12)	89.386(6)	89.721(6)	87.788(5)	354.77(6
XAs59	5.2187(4)	5.2397(4)	13.0298(11)	89.578(6)	88.938(5)	87.888(5)	355.98(5
XAs62	5.2173(4)	5.2506(3)	13.0150(10)	89.558(5)	88.900(5)	88.168(5)	356.28(4
XAs63	5.2225(3)	5.20014(33)	13.1390(8)	89.435(4)	89.779(4)	87.854(4)	356.55(4
XAs75	5.2134(4)	5.2415(4)	13.0156(12)	89.519(6)	88.837(5)	87.781(5)	355.31(5
Scorodite	- ()			- \ - /	- (-)	- (-)	(-

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Nominal*	8.937(1)	10.278(2)	9.996(2)	918.2(3)
XAs99	8.9514(9)	10.3254(10)	10.0369(9)	927.7(1)
XAs93	8.9335(8)	10.2610(8)	10.0484(8)	921.1(1)
XAs92±3	8.9394(6)	10.2871(6)	10.0435(6)	923.61(9)
XAs84	8.9365(3)	10.2617(4)	10.0553(4)	922.11(6)

695 696 697 Nominal FeOHSO<sub>4</sub> lattice parameters (OR) from Johannson (1962) and (MO) from Ventruti et al. (2005);

Nominal scorodite is based on Hawthorne (1976); Numbers in sample labels next to XAs refer to 100×AsO<sub>4</sub>/(AsO<sub>4</sub>+SO<sub>4</sub>); *a*, *b* and *c* in Å;  $\alpha$ ,  $\beta$  and  $\gamma$  in degrees; numbers in bracket are uncertainty values.

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## 706 **TABLE 4**. Arsenic XAFS fitting results

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Sample		N	R	$\sigma^2$	E0	rf	Ni/Nv
XAs8	0	4*	1.68±0.01	0.0008	-0.9	0.033	1.7
	Fe	4*	3.34±0.02	0.0074			
	S	3.5*	3.72±0.02	0.0047			
	MS2-1	12*	3.08±0.12	0.0085			
XAs12	0	4*	1.68±0.01	0.0009	0.4	0.029	1.7
	Fe	4*	3.35±0.02	0.0087			
	S	3.5*	3.73±0.02	0.0046			
	MS2-1	12*	3.07±0.09	0.0044			
XAs34	0	4*	1.68±0.01	0.0020	0.5	0.017	1.5
	Fe	4*	3.34±0.02	0.0111			
	S	2.6*	3.72±0.03	0.0064			
	As	1.4*	4.01±0.05	0.0085			
	MS2-1	12*	3.14±0.08	0.0127			
XAs35	0	4*	1.69±0.01	0.0012	2.0	0.012	1.5
	Fe	4*	3.38±0.02	0.0108			
	S	2.6*	3.76±0.02	0.0028			
	As	1.4*	4.06±0.02	0.0043			
	MS2-1	12*	3.11±0.07	0.0040			
XAs37	0	4*	1.68±0.01	0.0010	-0.2	0.016	1.5
	Fe	4*	3.35±0.02	0.0091			
	S	2.5*	3.73±0.03	0.0038			
	As	1.5*	4.02±0.04	0.0064			
	MS2-1	12*	3.11±0.11	0.0079			
XAs47	0	4*	1.69±0.01	0.0014	1.4	0.009	1.5
	Fe	4*	3.39±0.03	0.0123			
	S	2*	3.76±0.03	0.0024			
	As	2*	4.07±0.04	0.0084			
	MS2-1	12*	3.12±0.09	0.0060			
XAs59	0	4*	1.68±0.01	0.0012	-0.5	0.015	1.5
	Fe	4*	3.35±0.02	0.0103			
	S	1.6*	3.70±0.03	0.0011			
	As	2.4*	4.02±0.06	0.0115			
	MS2-1	12*	3.15±0.18	0.0017			
XAs63	0	4*	1.68±0.01	0.0013	1.4	0.011	1.5
	Fe	4*	3.38±0.03	0.0045			
	S	1.5*	3.74±0.02	0.0015			
	As	2.5*	4.05±0.03	0.0089			
	MS2-1	12*	3.10±0.07	0.0045			
XAs75	0	4*	1.68±0.01	0.0043	-0.4	0.015	1.7
/ 10/0	Fe	+ 4*	3.38±0.03	0.0014	0.7	0.010	1.7
	As	<del>-</del> 3*	3.99±0.07	0.0155			
	AS MS2-1	3 12*	3.10±0.07	0.0135			

Fit performed in *k*-space (3-13 Å<sup>-1</sup>); amplitude reduction factor  $(S_0^{-2})$  is constrained to 0.9 following its optimization; *N*: coordination number; *R*: interatomic distance (Å);  $\sigma^2$ : Debye–Waller parameter (Å<sup>2</sup>); *E0*: energy offset (eV); *rf*: r-factor as goodness-of-fit; *Ni/NV*: number of independent points/number of variables; \* Fixed value; Multiple scattering paths, MS2-1, MS3-1 and MS3-2 refer to As–O1–O2, As–O1–As–O2 and As–O1–As–O1, respectively (Manceau et al. 2007); *N* and *R* values of MS3-1 and MS3-2 were fixed to their crystallographic values (i.e. 12 at 3.34 Å and 4 at 3.34 Å, respectively) and their Debye-Waller parameters were constrained to be identical with that of MS2-1; Numbers in sample labels next to XAs refer to 100×AsO<sub>4</sub>/(AsO<sub>4</sub>+SO<sub>4</sub>).

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718	<b>TABLE 5</b> . Iron XAFS fitting results

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Sample		N	R	$\sigma^2$	E0	rf
XAs0	0	6*	2.00±0.01	0.0054	03	0.017
	S	4*	3.38±0.02	0.0111		
	Fe1	2*	3.69±0.03	0.0122		
	Fe2	8*	5.12±0.02	0.0122 <sup>a</sup>		
XAs8	0	6*	1.99±0.01	0.0062	-2.3	0.025
	S	3.7*	3.38±0.03	0.0147		
	Fe1	2*	3.69±0.03	0.0132		
	Fe2	8*	5.12±0.02	0.0132 <sup>a</sup>		
XAs12	0	6*	1.99±0.01	0.0076	-3.6	0.039
	S	3.5*	3.34±0.03	0.0174		
	Fe1	2*	3.71±0.05	0.0163		
	Fe2	8*	5.10±0.03	0.0163 <sup>a</sup>		
XAs34	0	6*	1.97±0.01	0.0094	0.6	0.027
	As	1.4*	3.32±0.06	0.0145		
	Fe1	2*	3.68±0.04	0.0193		
	Fe2	8*	5.14±0.04	0.0193 <sup>a</sup>		
XAs35	0	6*	1.97±0.01	0.0093	0.8	0.026
	As	1.4*	3.32±0.04	0.0155		
	Fe1	2*	3.67±0.06	0.0200		
	Fe2	8*	5.14±0.04	0.0200 <sup>a</sup>		
XAs47	0	6*	1.96±0.01	0.0106	0.9	0.027
	As	2*	3.31±0.02	0.0117		
	Fe1	2*	3.62±0.06	0.0201		
	Fe2	8*	5.18±0.04	0.0201 <sup>a</sup>		
XAs62	0	6*	1.96±0.01	0.0111	0.8	0.031
	As	2.5*	3.32±0.02	0.0112		
	Fe1	2*	3.62±0.10	0.0241		
	Fe2	8*	5.17±0.06	0.0241 <sup>a</sup>		
XAs75	0	6*	1.97±0.01	0.0110	2.6	0.029
	As	3*	3.34±0.02	0.0140		
	Fe1	2*	3.65±0.08	0.0213		
	Fe2	8*	5.17±0.04	0.0213 <sup>a</sup>		

Fit performed in *k*-space (2.5-12 Å<sup>-1</sup>); amplitude reduction factor  $(S_0^2)$  is constrained to 0.9 following its optimization; *N*: coordination number; *R*: interatomic distance (Å);  $\sigma^2$ : Debye–Waller parameter (Å<sup>2</sup>); *E0*: energy offset (eV); *rf*: r-factor as goodness-of-fit; number of independent points/number of variables=2 for all; \* Fixed value optimized to crystallographic value; <sup>a</sup>: constrained to be identical with preceding Debye– Waller parameter. Numbers in sample labels next to XAs refer to 100×AsO<sub>4</sub>/(AsO<sub>4</sub>+SO<sub>4</sub>).

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## TABLE 6. Arsenic XAFS fitting of pressure oxidation residues and bukovskýite

		N	R	$\sigma^2$	E0	rf	Ni/Nv
POX-2	0	4*	1.68±0.01	0.0013	6.2	0.013	1.5
	Fe	4*	3.38±0.03	0.0129			
	S	2*	3.72±0.04	0.0055			
	As	2*	4.05±0.05	0.0097			
	MS2-1	12*	3.11±0.09	0.0081			
POX-9	0	4*	1.70±0.01	0.0019	7.8	0.032	1.6
	Fe	2*	3.43±0.04	0.0086			
	MS2-1	12*	3.13±0.07	0.0009			
POX-14	0	4*	1.69±0.01	0.0016	5.8	0.020	1.6
	Fe	2*	3.35±0.05	0.0133			
	MS2-1	12*	3.11±0.06	0.0010			
bukKH	0	4*	1.69±0.01	0.0014	5.8	0.011	1.7
	Fe	4*	3.36±0.02	0.0092			
	As	2*	3.95±0.06	0.0113			
	MS2-1	12*	3.10±0.07	0.0034			
bukKe	0	4*	1.69±0.01	0.0011	6.3	0.011	1.7
	Fe	4*	3.35±0.01	0.0077			
	As	2*	3.95±0.05	0.0113			
	MS2-1	12*	3.13±0.07	0.0045			

Fit performed in *k*-space  $(3-13 \text{ Å}^{-1})$ ; amplitude reduction factor  $(\text{S}_0^{-2})$  is constrained to 0.9 following its optimization; *N*: coordination number; *R*: interatomic distance (Å);  $\sigma^2$ : Debye–Waller parameter (Å<sup>2</sup>); *E0*: energy offset (eV); *rf*: r-factor as goodness-of-fit; Multiple scattering paths, MS2-1, MS3-1 and MS3-2 refer to As–O1–O2, As–O1–As–O2 and As–O1–As–O1, respectively (Manceau et al. 2007); \* Fixed value; *N* and *R* values of MS3-1 and MS3-2 were fixed to their crystallographic values (i.e. 12 at 3.34 Å and 4 at 3.34 Å, respectively) and their Debye-Waller parameters were constrained to be identical with MS2-1; bukKH: bukovskýite from Kutna Hora, Czech Republic; bukKe: bukovskýite from Ketza River Mine, Yukon, Canada; POX-2, -9, -14 pressure oxidation residues from gold processing operations.

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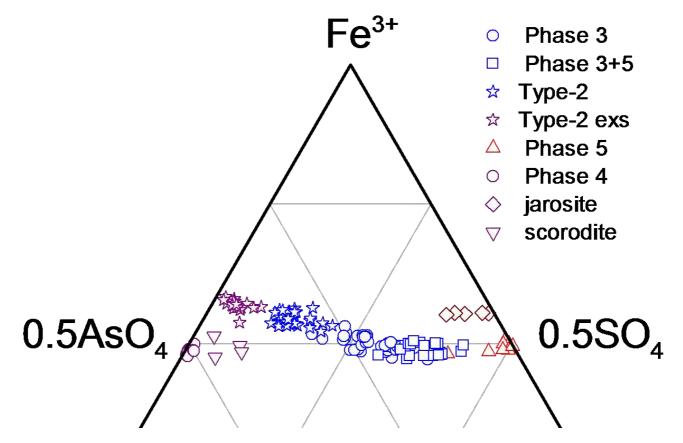
## 741 **TABLE 7**. Fe-XAFS fitting of bukovskýite

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		N	R	$\sigma^2$	rf	E0
bukKH	0	6*	1.99±0.01	0.0084	0.029	-3.5
	As	2*	3.35±0.01	0.0076		
	Fe1	1*	3.62±0.02	0.0064		
	Fe2	1*	5.08±0.04	0.0064 <sup>a</sup>		
bukKe	0	6*	1.99±0.01	0.0079	0.033	-4.4
	As	2*	3.34±0.01	0.0057		
	Fe1	1*	3.60±0.02	0.0058		
	Fe2	1*	5.09±0.04	0.0058 <sup>a</sup>		

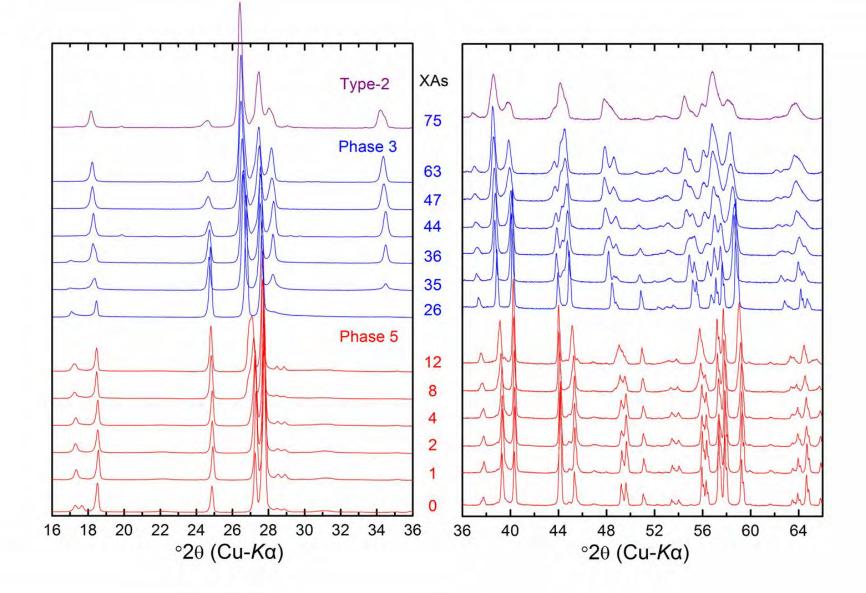
Fit performed in *k*-space (3-13 Å<sup>-1</sup>); amplitude reduction factor (S<sub>0</sub><sup>2</sup>) is constrained to 0.9 following its optimization; *N*: coordination number; *R*: interatomic distance (Å); o<sup>2</sup>: Debye–Waller parameter (Å<sup>2</sup>); *E0*: energy offset (eV); *rf*: r-factor as goodness-of-fit; number of independent points/number of variables=2;
\*:Fixed value optimized to crystallographic value; <sup>a</sup>: constrained to be identical with preceding Debye–Waller parameter. bukKH: bukovskýite from Kutna Hora, Czech Republic; bukKe: bukovskýite from Ketza River Mine, Yukon, Canada.

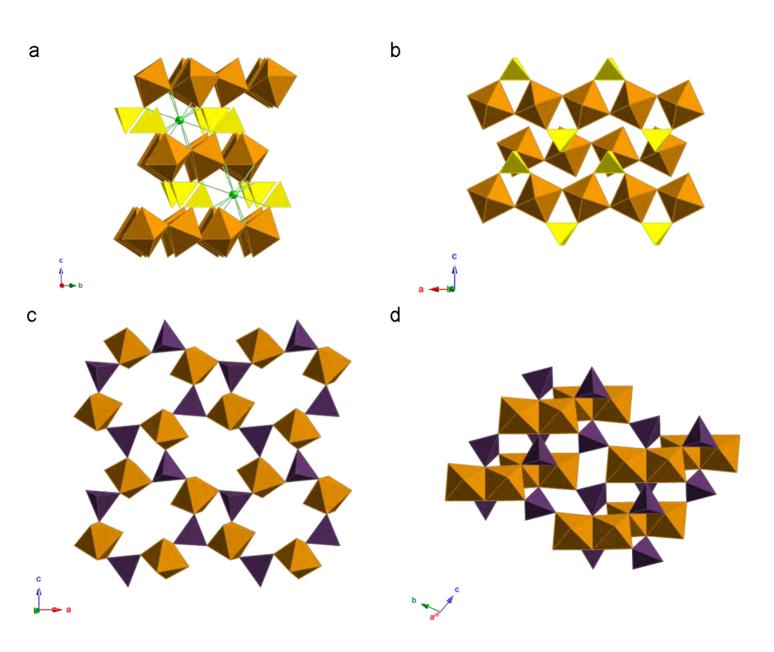
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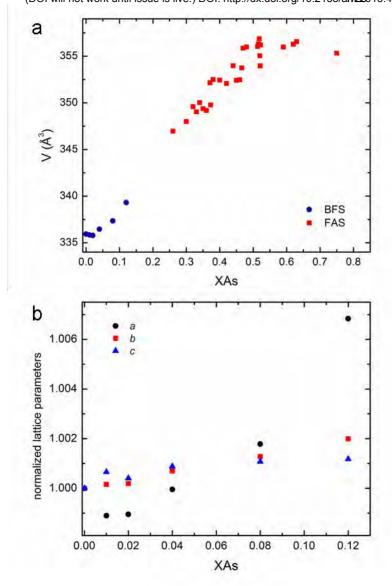
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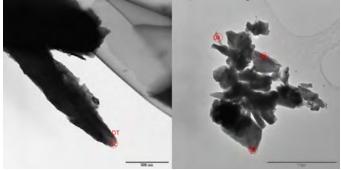




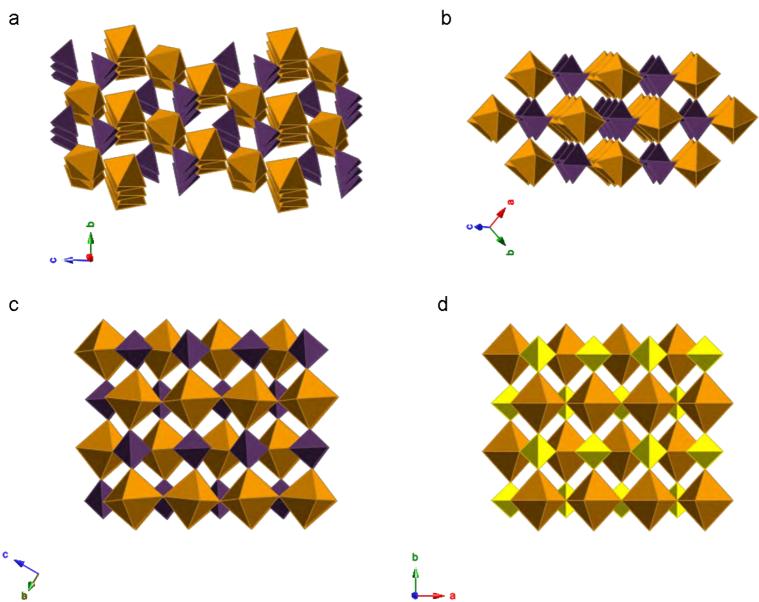
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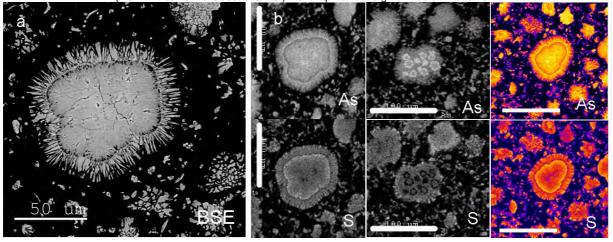


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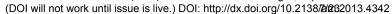
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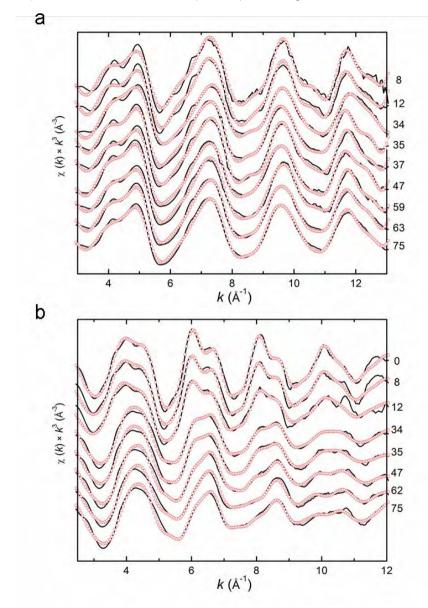


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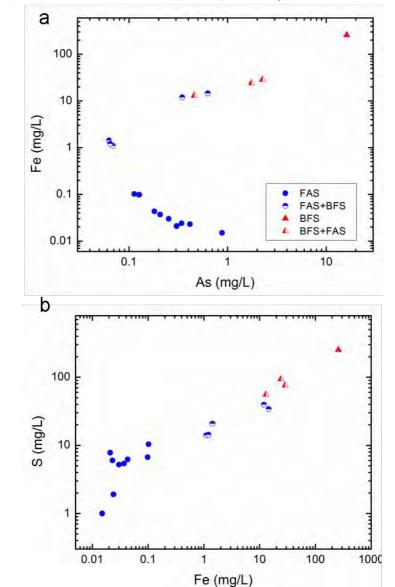
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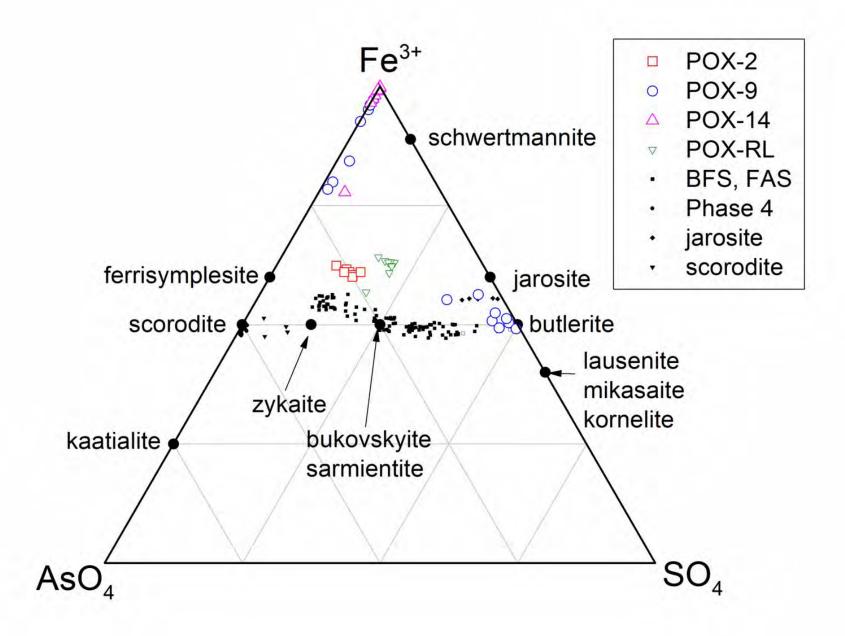
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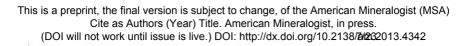
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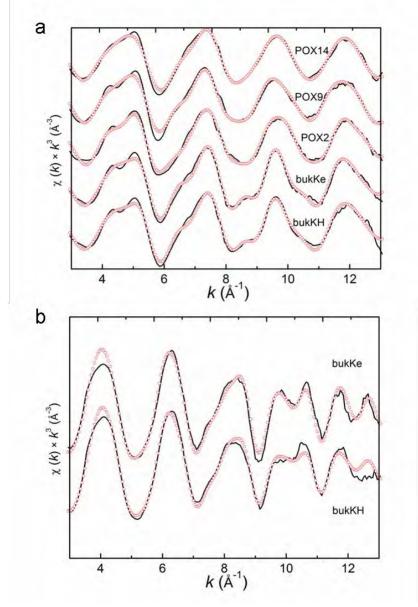


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