

1 Revision 1

2 **Low-temperature magnetism of alabandite: crucial**
3 **role of surface oxidation**

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5 J. Cuda^{a,*}, T. Kohout^{b,c,*}, J. Filip^a, J. Tuček^a, A. Kosterov^{d,e}, J. Haloda^f, R.
6 Skala^c, E. Santala^g, I. Medrik^a, R. Zboril^a

7
8 ^a*Regional Centre of Advanced Technologies and Materials, Departments of Physical Chemistry*
9 *and Experimental Physics, Faculty of Science, Palacky University Olomouc, 17. listopadu 12, 771 46 Olomouc,*
10 *Czech Republic*

11 ^b*Department of Physics, University of Helsinki, P.O. Box 64, 00014 Helsinki University, Finland*

12 ^c*Institute of Geology, Academy of Sciences of the Czech Republic v.v.i., Rozvojová 269, 165 00 Prague, Czech*
13 *Republic*

14 ^d*Earth Physics Department, Faculty of Physics, Saint-Petersburg University, 198504 Peterhoff, Saint-*
15 *Petersburg, Russia*

16 ^e*Resource Center "Geomodel", Saint-Petersburg University, 198504 Peterhoff, Saint-Petersburg, Russia*

17 ^f*Czech Geological Survey, Geologická 6, 152 00 Praha 5, Czech Republic*

18 ^g*Department of Chemistry, University of Helsinki, P.O. Box 55, 00014 Helsinki University, Finland*

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***Corresponding author(s):** e-mail:"

jan.cuda@upol.cz (Jan Cuda); e-mail:

22 kohout@gli.cas.cz (Tomas Kohout)

23

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25

26 **Abstract**

27 Manganese(II) monosulphide crystallizes into three different polymorphs (alpha-, beta-, and
28 gamma-MnS). Out of these, alpha-MnS, also known as mineral alabandite, is considered the
29 most stable and is widespread in terrestrial materials as well as in extraterrestrial objects such
30 as meteorites.

31 In this study, a low-temperature antiferromagnetic state of alpha-MnS was investigated using
32 macroscopic magnetic measurements as induced and remanent field-cooled (FC) and zero-
33 field-cooled (ZFC) magnetizations and magnetic hysteresis. Both natural alabandite and
34 synthetic samples show (i) Néel temperatures in a narrow temperature range around 153 K
35 and (ii) a rapid increase of the magnetization around 40 K. An anomalous magnetic behavior
36 taking place at about 40 K was previously ascribed to the magnetic transition from a high-
37 temperature antiferromagnetic to a low-temperature ferromagnetic state documented for non-
38 stoichiometric alpha-MnS slightly enriched in manganese. However, our detailed microscopic
39 observations and, in particular, oxidation experiments indicate that the anomalous magnetic
40 behavior around 40 K is caused by a presence of oxide layer of ferrimagnetic hausmannite
41 (Mn_3O_4) on the surface of alpha-MnS rather than being an intrinsic property of nearly
42 stoichiometric alpha-MnS.

43

44 **1. Introduction**

45 Alabandite is a manganese sulfide with theoretical composition MnS crystallizing in the cubic
46 lattice of galena type (PbS). It occurs as an accessory mineral at many localities worldwide,
47 mainly in epithermal base-metal sulfide veins, in low-temperature manganese deposits
48 [Doelter, 1926; Hewett and Rove, 1930; Anthony *et al.*, 2012] and also in marine sediments
49 [Lepland and Stevens, 1998]. Locally, it is an important ore mineral of Mn. Its name is

50 derived from its supposed discovery locality at Alabanda, Turkey. The type locality of
51 alabandite is Sacarimb, Romania [Anthony *et al.*, 2012].

52 Apart from terrestrial localities, alabandite is also relatively abundant in certain types of
53 meteorites, e.g., in E chondrites [Keil, 1968; Zhang *et al.*, 1995; Zhang and Sears, 1996;
54 Brearley and Jones, 1998] and related achondritic aubrites [Keil and Fredriksson, 1967;
55 Ryder and Murali, 1987; Lin *et al.*, 1989; Mittlefehldt *et al.*, 1998]. It was also reported in
56 some ureilites [Fioretti and Molin, 1998] and winonaites [Mason and Jarosewich, 1967].

57 Alabandite is paramagnetic at room temperatures and orders antiferromagnetically below its
58 Néel temperature $T_N \sim 148$ K [Heikens *et al.*, 1977]. A slightly higher T_N (~ 153 K) was later
59 published by Pearce *et al.* [2006]. A structural transition occurs at $T \sim 130$ K which is
60 interpreted as an abrupt inversion of the rhombohedral distortion of the f.c.c. lattice along [1 1
61 1] plane accompanied by a discontinuous change in the magnetic susceptibility observed in
62 alabandite single crystals [Heikens *et al.*, 1977].

63 Magnetic susceptibility and induced field-cooled and zero-field-cooled magnetization (in
64 10 mT of external magnetic field) of antiferromagnetic alabandite below its T_N are low,
65 typically in the range of 10^{-7} m³/kg and 3-4 mA m²/kg, respectively. The substitution of Mn²⁺
66 ions by Fe²⁺ has a pronounced effect on the Néel temperature which increases with increasing
67 iron content reaching ~ 185 K for the Fe_xMn_{1-x}S system with $x = 0.2$ [Petrakovski *et al.*,
68 2002]. Still more iron-rich alabandite samples ($x > 0.25$) exhibit ferrimagnetic behavior above
69 room temperature with Curie temperatures T_C between 730 K ($x = 0.27$) and 860 K ($x = 0.38$)
70 [Loseva *et al.*, 1998; Petrakovski *et al.*, 2002]. However, magnetization of this ferrimagnetic-
71 ordered alabandite is weak, close to that of paramagnetic MnS.

72 It was reported that iron-free MnS samples with a slight excess of Mn show antiferro-
73 ferromagnetic transition at $T \sim 40$ K [Petrakovski *et al.*, 2001]. This transition manifests itself
74 in a sharp, two orders of magnitude, increase of induced magnetization on cooling. Such a

75 sharp change in magnetic properties can significantly increase magnetic response of
76 alabandite at low temperatures and can potentially contribute to low-temperature magnetic
77 properties of extraterrestrial bodies [Kohout *et al.*, 2010]. Similar low-temperature magnetic
78 transition is observed in troilite (FeS) [Kohout *et al.*, 2007; Cuda *et al.*, 2011]. Moreover,
79 Gattacceca *et al.* [2011] recently reported that chromite with Curie temperature in the 40–
80 80 K range exist in certain meteorites and may significantly modify their low-temperature
81 magnetic properties. Therefore, verification, interpretation and quantification of this magnetic
82 phenomenon in alabandite samples and its comparison to low-temperature magnetic
83 properties of troilite and chromite are required and are the subject of this study.

84

85 **2. Materials and methods**

86 Basic characteristics of our samples are summarized in the Table 1. Natural polycrystalline
87 sample of alabandite (NA) originates from Broken Hill, N.S.W., Australia, sample Bm 1972,
88 294, kindly provided by Natural History Museum, London. Additionally, nearly
89 stoichiometric alpha-MnS was synthesized adopting two alternative procedures.

90 SA1 sample was prepared using a slightly modified solvothermal process of Biswas and
91 Chaudhuri [2007]. First, manganese acetate [(CH₃CO₂)Mn·4H₂O, purity >99.0%, Sigma
92 Aldrich] and thiourea [CH₄N₂S, purity >99.0%, Sigma Aldrich] was mixed in a molar ratio of
93 1:3 with a water solvent, loaded into an ace pressure tube (Sigma Aldrich) and placed into a
94 furnace at 190°C for 17 hours. Subsequently dried product was annealed in a helium
95 atmosphere at temperatures up to 450°C.

96 SA2 sample was synthesized by direct thermal fusion of sulfur (purity >98.0%, Sigma
97 Aldrich) and manganese (purity >99.0%, Sigma Aldrich) powders in stoichiometric molar
98 ratio. The precursors were annealed twice in a sealed quartz tube under reduced pressure for
99 12 hours at 700°C including sample homogenization between two subsequent runs.

100 Measurements of the macroscopic magnetic response such as induced and remanent field-
101 cooled (FC) and zero-field-cooled (ZFC) magnetizations and magnetic hysteresis
102 measurements were carried out at the Institute for Rock Magnetism, University of Minnesota
103 and at the Regional Centre of Advanced Technologies and Materials, Palacky University
104 Olomouc, using MPMS5S and MPMS XL-7 (both Quantum Design) SQUID magnetometers.
105 Details of FC and ZFC measurement procedure are provided in auxiliary material.

106 X-Ray Diffraction (XRD) patterns of all samples were recorded with a PANalytical X'Pert
107 PRO MPD diffractometer (iron-filtered $\text{CoK}\alpha$ radiation: $\lambda = 0.178901$ nm, 40 kV and 30 mA)
108 in the Bragg-Brentano geometry. Samples were placed on a zero-background Si slides and
109 scanned in a continuous mode (resolution of $0.017^\circ 2\theta$, scan speed of $0.0016^\circ 2\theta$
110 per second). Identification of crystalline phases and Rietveld refinements were obtained using
111 the software High Score Plus (PANalytical) in conjunction with the PDF-4+ and ICSD
112 databases (ICSD collection codes: MnS – 41331, Mn_3O_4 – 31094, Mn_2O_3 – 76087, S –
113 63082). Peak shapes were modeled using the pseudo-Voigt function, separately refining the
114 Caglioti parameters (u, v, w), unit cell parameters, and scale factor for each phase.

115 Bulk chemical composition of the SA2 sample was determined using a quantitative X-ray
116 wavelength dispersive spectral analysis on a MICROSPEC 3PC X-ray wavelength dispersive
117 system (WDS) on a CamScan 3200 scanning electron microscope (SEM) at the Czech
118 Geological Survey. The analyses were performed using an accelerating voltage of 20 kV,
119 25 nA beam current, 1 μm beam size and ZAF correction procedures. The counting times
120 were 30 s for all analyzed elements. A combination of natural and synthetic standards was
121 used for calibration.

122 SEM TESCAN VEGA 3XM at Institute of Geology, Academy of Sciences of the Czech
123 Republic has been used to document surface features of the natural alabandite (NA sample).
124 To avoid potential deterioration of the sample, the specimen has not been coated and the SEM

125 has been operated at low-vacuum mode. Energy dispersive x-ray (EDX) spectra of individual
126 phases observed on the surface of the studied specimen have been acquired with a Bruker
127 XFlash detector. Subsequently, part of the sample has been polished and analyzed with an
128 electron microprobe (EMPA) CAMECA SX-100 instrument at the Institute of Geology,
129 Academy of Sciences of the Czech Republic to determine the stoichiometry of the sample
130 interior.

131

132 **3. Results and discussion**

133 **3.1. Characterization and low-temperature magnetic properties of alpha-MnS**

134 XRD patterns of a natural alabandite (NA) and of two synthetic alpha-MnS (SA1 and SA2)
135 samples are shown in Figure 1a, 1b, and 1c. They perfectly correspond to the cubic structure
136 of alpha-MnS (PDF No. 01-088-2223). In the NA sample, some amount of elemental sulfur
137 (16% by Rietveld refinement) and hausmannite (see below) has been found in addition to the
138 alabandite main phase. The presence of sulfur on NA sample surface was also confirmed by
139 SEM/EDX (Figure S1 of the auxiliary material). In contrast, EMPA of the NA sample interior
140 did not reveal any presence of sulfur or hausmannite. SEM-WDS analysis of the SA2 sample
141 show similar result. Based on the information provided above, all three studied samples can
142 be considered as a representative of alabandite with only a minor presence of other phases
143 limited to the surface of the individual grains.

144 Temperature dependence of induced FC and ZFC magnetizations at 10 mT (Figure 2) as well
145 as remanent FC and ZFC magnetizations (imprinted by 2.5 T at 5 K, Figure 3) yield Néel
146 temperatures (T_N) for natural and synthetic samples in a narrow temperature range around
147 153 K as expected for alabandite [Pearce *et al.*, 2006]. Néel temperature manifests itself
148 through peaks on the induced magnetization curves, and through merging of the FC and ZFC
149 remanent magnetization curves at T_N .

150 In all alabandite samples, a feature at 40 ± 2 K is further observed, manifested through a rapid
151 increase of magnetic response with decreasing temperature (Figure 2 and 3). Enhanced
152 magnetic response is also seen in the hysteresis properties at 5 K, namely, an S-shaped
153 hysteresis loop and a tendency to saturation in high external magnetic field (Figure 4). Such
154 behavior is typical for the ferro/ferrimagnetic materials rather than for an antiferromagnetic
155 one.

156 Enhanced magnetic response below 40 ± 2 K is similar to that reported for iron-free alpha-
157 MnS samples slightly enriched in Mn with respect to a stoichiometric alpha-MnS and
158 interpreted as an antiferro- to ferromagnetic transition at 40 K upon cooling [*Petrakovski et*
159 *al.* 2001]. However, in our case both NA and SA2 samples are highly stoichiometric
160 examples of alabandite (Table 1). According to our measurements, the enhanced magnetic
161 response at around 40 K occurs in all studied samples and its amplitude does not correlate
162 with Mn/S ratio. Thus, a question arises whether such low-temperature behavior is limited to
163 alabandite samples slightly enriched in Mn as reported by *Petrakovski et al.* [2001], or it is a
164 general phenomenon occurring in stoichiometric or Mn-depleted alabandite samples as well.
165 Alternatively, a presence of small amounts of another phase on alabandite grain surfaces
166 identified above may be responsible for the observed low-temperature magnetic behavior and
167 will be evaluated in following section.

168

169 **3.2. Effects of surface oxidation on low-temperature magnetic properties of alpha-MnS**

170 In order to confirm or exclude the role of manganese oxides on the magnetic response of
171 alpha-MnS (sulfur should not significantly influence the low-temperature magnetic properties
172 of alabandite), we artificially oxidized the SA1 sample by hydrogen peroxide (H₂O₂). About
173 20 mg of SA1 material was briefly submerged in hydrogen peroxide and air-dried at room
174 temperature for 20 hours (sample labeled as SA1_OX). Subsequently, XRD and magnetic

175 measurements were carried out following the same procedures as for other samples (Figures
176 1d and 5).

177 Upon hydrogen peroxide treatment, surface of MnS was partially oxidized into manganese
178 oxide phases and thus the ratio of alabandite to manganese oxides in the SA1_OX sample
179 changed compared to original SA1 sample. Presence of manganese oxide Mn₃O₄, known also
180 as mineral hausmannite, was confirmed by XRD analysis in the SA1_OX sample (Figure 1d).
181 Hausmannite has Curie temperature $T_C \sim 41 \div 43$ K [*Dwight and Menyuk*, 1960; *Robie and*
182 *Hemingway*, 1985]. It is very close to the temperature where enhanced magnetic response of
183 alabandite is observed upon cooling, making hausmannite a meaningful candidate to explain
184 the observed low-temperature magnetic behavior of alabandite. Other phases identified in the
185 XRD pattern are bixbyite (alpha-Mn₂O₃, antiferromagnetic below 80 K [*Robie and*
186 *Hemingway*, 1985] or 90 K [*Mukherjee et al.*, 2006]) and sulfur (diamagnetic [*O'Handley*,
187 2000; *Blundell*, 2001]). These phases do not have any magnetic transitions around 40 K which
188 would explain observed magnetic behavior at this temperature.

189 Oxidized SA1_OX sample shows a significant (by a factor of eight) increase in its magnetic
190 response below 40 ± 2 K compared to the unoxidized alpha-MnS sample (insets in Figure 5).
191 Dependence of magnitude of the 40 K magnetic response on the amount of manganese
192 oxides, including hausmannite, suggests that the low-temperature behavior observed in
193 studied alabandite samples is not an intrinsic property of alabandite itself, but rather
194 hausmannite governs the low-temperature magnetic response below ~ 40 K. Similar example
195 was described for antiferromagnetic MnO nanoparticles with Mn₃O₄ surface layers [*Berkowitz*
196 *et al.*, 2008].

197 Assuming that ferromagnetic-like hysteresis loops measured at 5 K are produced entirely by
198 Mn₃O₄, it is possible to estimate its content in the studied samples. However, caution is
199 needed when interpreting the hysteresis data, since, unlike many other compounds of spinel

200 type, Mn_3O_4 does not reach magnetic saturation even in fields as high as 34 T [*Nielsen and*
201 *Roeland*, 1976]. Corresponding saturation magnetization value, determined from
202 extrapolation to zero magnetic field of the magnetization field dependence measured along
203 [010] direction, amounts to $47.2 \text{ Am}^2/\text{kg}$. Experiments performed in much lower maximum
204 fields, which are more appropriate to compare with ours, yielded respectively lower values of
205 the high-field magnetization extrapolated to zero field (M_E): $34.2 \text{ Am}^2/\text{kg}$ (1.4 Bohr
206 magnetons per formula unit) for the 1 T field [*Wickham and Croft*, 1958] and $38.1 \text{ Am}^2/\text{kg}$
207 (1.56 Bohr magnetons per formula unit) for the 14 T field [*Jacobs*, 1959].

208 For our experiments, carried out in the 7 T maximum field, we adopted an intermediate value
209 of $36 \text{ Am}^2/\text{kg}$. M_E of the SA1_OX sample (i.e., after oxidation) amounts to $7.3 \text{ Am}^2/\text{kg}$ at 5 K
210 (Table 2) corresponding to a presence of ~20 wt.% of hausmannite. This is in a reasonably
211 good accordance with the result of Rietveld refinement where the amount of hausmannite in
212 this oxidized sample appears to be 22.9(4) wt.%. The remaining difference may be due to a
213 small size of at least a fraction of hausmannite grains as suggested by broad diffraction peaks
214 of hausmannite in the XRD pattern (Figure 1d; see also *Siskova et al.*, 2012 and *Markova et*
215 *al.*, 2012). Indeed, nanosized Mn_3O_4 shows considerably reduced M_E values as determined
216 from hysteresis loops measured in maximum fields up to 7 T [*Winkler et al.*, 2004, *Vázquez-*
217 *Olmos et al.*, 2005].

218 We expect that hausmannite governs the magnetic response below 40 K in other studied
219 samples as well. Comparing, as above, the M_E values at 5 K (Table 2) with the Mn_3O_4 M_E
220 bulk value of $36 \text{ Am}^2/\text{kg}$, we can estimate the hausmannite content. Approximately 2.4 wt.%,
221 0.1 wt.%, and 5.3 wt.% of hausmannite are required to produce the low-temperature magnetic
222 response observed in SA1 (before oxidation), SA2, and natural alabandite (NA) samples,
223 respectively. These values are close to, or below, the detection limit of XRD measurements

224 and thus not observed in the XRD pattern of these samples except for NA sample (6.9(4)
225 wt.% of Mn_3O_4 as calculated using Rietveld refinement).

226 Furthermore, a profile of the temperature-dependent induced FC magnetization curve of
227 samples NA, SA1 and especially SA1_OX displays a tendency to follow the Curie-Weiss law
228 above 40 K. The hysteresis loops below 40 K show also shift along field (horizontal) axis
229 towards negative values (compare B_{c+} and B_{c-} in Table 2) which seems to be a manifestation
230 of exchange interaction between antiferromagnetic and ferro- or ferrimagnetic phases
231 [Nogues and Schuller, 1999]. Exchange bias or exchange anisotropy is present in bi-layers (or
232 multi-layers) of magnetic materials which in our case are represented by alabandite
233 (antiferromagnetic) and hausmannite (ferrimagnetic). On the other hand, we cannot exclude a
234 random canting of the particle surface spins caused by competing antiferromagnetic exchange
235 interactions at the surface of hausmannite particles producing exchange anisotropy between
236 their core and surface spins.

237 In the SA2 sample with an extremely low ($\sim 0.1\%$) hausmannite content, the magnetic signal
238 above 40 K does not show a tendency to follow the Curie-Weiss law, and the low-temperature
239 behavior is dominated by a stronger alabandite antiferromagnetic response than a
240 paramagnetic behavior of Mn_3O_4 present in the oxidized surface layer of alabandite grains.

241 Upon heating in the temperature range between T_C of hausmannite and T_N of alabandite, the
242 decreasing paramagnetic response of hausmannite overlaps with the slightly increasing
243 antiferromagnetic response of alabandite. A local minimum in the induced FC curve can be
244 observed in this temperature range and it is shifted to the higher temperatures with increasing
245 hausmannite content (insets in Figure 2). In this case, it appears that there is a direct
246 correlation between the amount of hausmannite present in the sample and temperature at
247 which the minimum occurs. The estimated temperatures are 44 ± 1 K, 89 ± 1 K and 146 ± 1 K
248 for SA2, SA1, and NA samples, respectively, and follow positive trend with increasing

249 hausmannite content. For sample SA1_OX, we do not observe the minimum in temperature
250 range from 40 to 155 K because the paramagnetic response of hausmannite (and perhaps also
251 of bixbyite) dominates over the antiferromagnetic response of alpha-MnS.
252 Last but not least, the temperature dependence of induced ZFC measurements of NA, SA1,
253 SA2 and SA1_OX samples exhibits a sharp peak below the Curie temperature ascribed to
254 hausmannite, and then on subsequent heating drops down to a very low values (Figures 2 and
255 5a). The observed peak can be interpreted as a Hopkinson peak observed just prior to
256 transition from magnetically ordered state to a paramagnetic one [Dunlop and Özdemir, 1997]
257 rather than as an effect accompanying a magnetic transition from ferromagnetic to a
258 antiferromagnetic state as suggested by *Petrakovski et al.* [2001].
259 The positive correlation between Mn enrichment and magnitude of the 40 K feature observed
260 in synthetic alabandite by *Petrakovski et al.* [2001] can be explained as the extra Mn added
261 did not enter alabandite structure and rather reacted with oxygen to produce manganese
262 oxides including hausmannite. Thus, higher addition of Mn resulted in higher production of
263 hausmannite causing higher amplitude of the 40 K feature.

264

265 **3.3. Comparison to other low-temperature magnetic minerals**

266 Similar low-temperature magnetic transition, as described above in alabandite – hausmannite
267 system, is observed at ~70 K in other sulfur monosulfide – troilite FeS [*Kohout et al.*, 2007;
268 *Cuda et al.*, 2011]. *Gattacceca et al.* [2011] recently reported that chromite with Curie
269 temperature in 40-80 K range exists in certain meteorites and chromite contamination within
270 troilite samples may be responsible for the observed ~70 K feature in troilite. In this respect
271 alabandite with hausmanite contamination is analogue to troilite with proposed chromite
272 contamination. Thus, in following paragraph we briefly compare alabandite – hausmannite
273 system to the troilite and chromite.

274 At the first look both systems show very similar behavior with a sharp increase in both
275 induced and remanent magnetization and onset of ferromagnetic-like hysteresis below the
276 transition temperature. The difference between these two systems is in nature of the
277 contaminant. Hausmannite is of similar composition to alabandite (both manganese bearing
278 phases) and is localized at the surface of alabandite grains. Thus it can be easily overlooked in
279 EMPA analysis of polished grains. In contrast, chromite contamination is supposed to be
280 present within the interior of troilite grains and thus should be more easily observable.
281 Thorough analytical data (EMPA and SEM-BSE (BackScattered Electrons) observation of
282 polished grain sections, AAS (Atomic Absorption Spectroscopy), XRD and Mössbauer
283 spectroscopy of bulk troilite samples in *Cuda et al.* [2011]) reveal chromium content one to
284 three orders of magnitude lower than predicted for chromite amount explaining low-
285 temperature magnetic observations. Another difference can be observed on induced ZFC
286 magnetization curves. While in alabandite – hausmannite case the ZFC induced magnetization
287 stays well below FC and shows a pronounced Hopkinson peak just below T_C of hausmannite
288 (Figure 2), the ZFC curve of troilite do not show obvious presence of Hopkinson peak (Figure
289 3 in *Kohout et al.* [2007]). Thus, the nature of the contaminant or mechanism of the low-
290 temperature transition in troilite is likely to be different than proposed chromite (or similar to
291 alabandite – hausmannite case).

292

293 **4. Conclusions**

294 Based on our detailed investigation, the low-temperature phenomenon at ~40 K, previously
295 observed in some synthetic alabandite samples and ascribed to nonstoichiometry of the latter,
296 is not the intrinsic property of alabandite. It appears to be a result of ferri- to paramagnetic
297 transition of hausmannite (Mn_3O_4) present in oxidized surface layer on crystals/grains of
298 alabandite. Presence of hausmannite even in amounts below 1 wt.% can have a detectable

299 effect on magnetic response of alabandite, which is otherwise a purely antiferromagnetic
300 material below its Néel temperature of ~ 153 K without any other low-temperature magnetic
301 transitions. This conclusion rules out pristine alabandite to significantly contribute to
302 remanent or induced magnetism of minor Solar System bodies. In contrary, no prove of
303 similar contamination has been found in troilite showing similar transition ~ 70 K.

304

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435 **Figure captions**

436 **Figure 1** XRD patterns of the samples: (a) NA, (b) SA1, (c) SA2, and (d) SA1_OX. Result of
437 the Rietveld refinement of SA1_OX sample is: 24.0(2) wt.% of MnS, 23(1) wt.% of Mn₃O₄,
438 22.9(4) wt.% of Mn₂O₃, and 30(1) wt.% of sulfur. The respective PDF cards shown below
439 XRD patterns are: 1 - 01-088-2223 (MnS); 2 - 01-075-1560 (Mn₃O₄); 3 - 01-089-4836
440 (Mn₂O₃); 4 - 01-078-1888 (S).

441

442 **Figure 2** Induced-ZFC and FC magnetization curves in the external magnetic field of 10 mT
443 for samples: (a) NA, (b) SA1 and (c) SA2.

444

445 **Figure 3** ZFC-FC curves of remanent magnetization imprinted by a field of 2.5 T for
446 samples: (a) NA and (b) SA1.

447

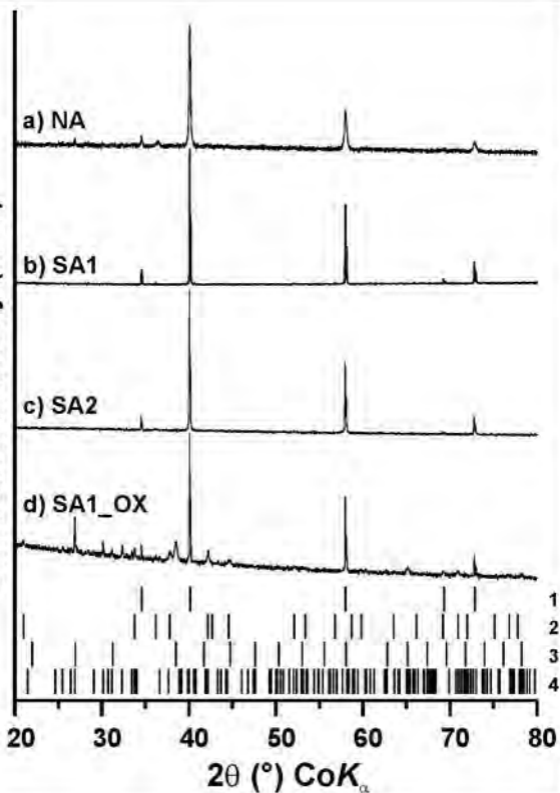
448 **Figure 4** Hysteresis loops of samples: (a) NA, (b) SA1 and (c) SA2, measured at 5 K (the
449 high-field slope has been subtracted).

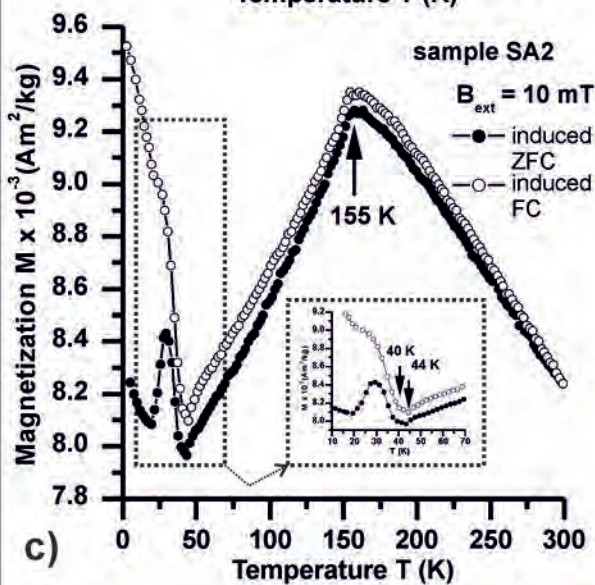
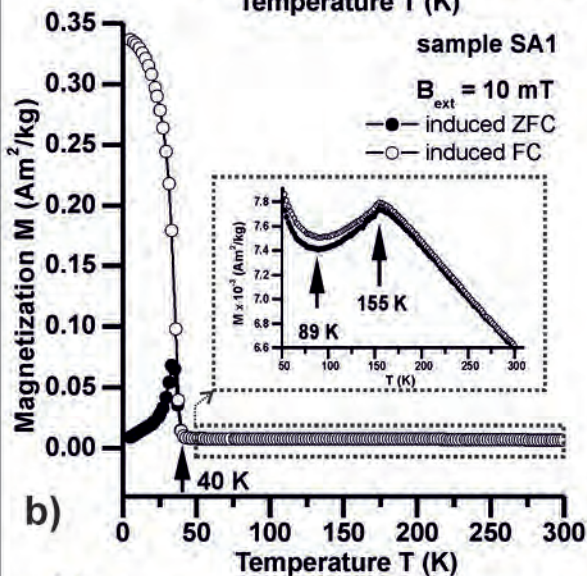
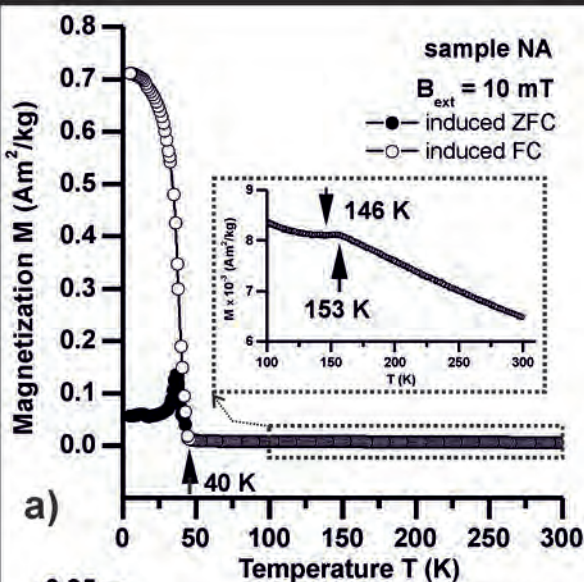
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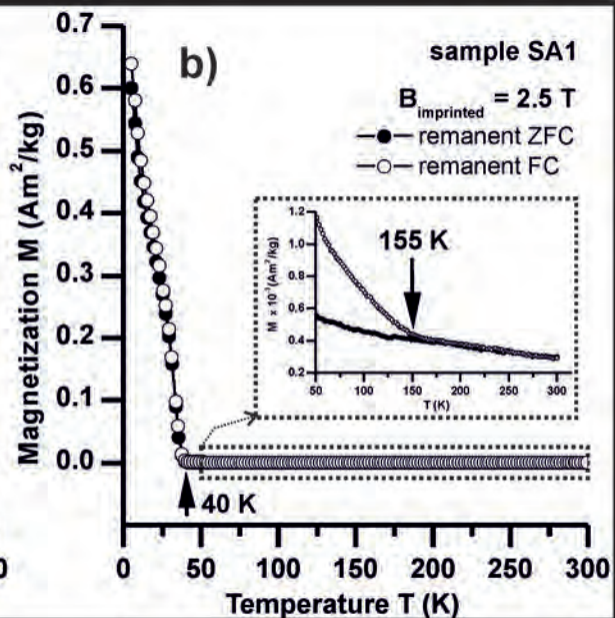
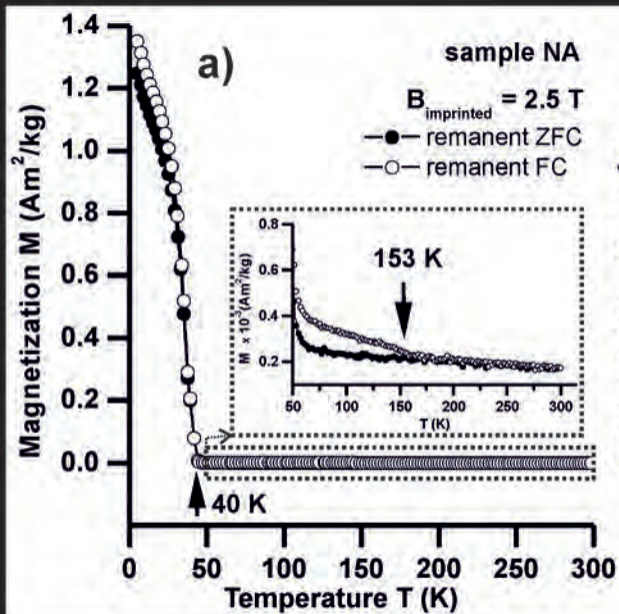
451 **Figure 5** (a) Induced-ZFC and FC magnetization curves in the external magnetic field of
452 10 mT for SA1_OX sample. Induced-FC curves of the SA1 sample before and after oxidation
453 are displayed for comparison in the inset. (b) ZFC-FC curves of remanent magnetization
454 imprinted by a field of 2.5 T for the SA1_OX sample. Remanent-FC curves of the SA1 before
455 and after oxidation are displayed for comparison in the inset. (d) Hysteresis loops of the SA1
456 sample before and after oxidation at 5 K after slope correction.

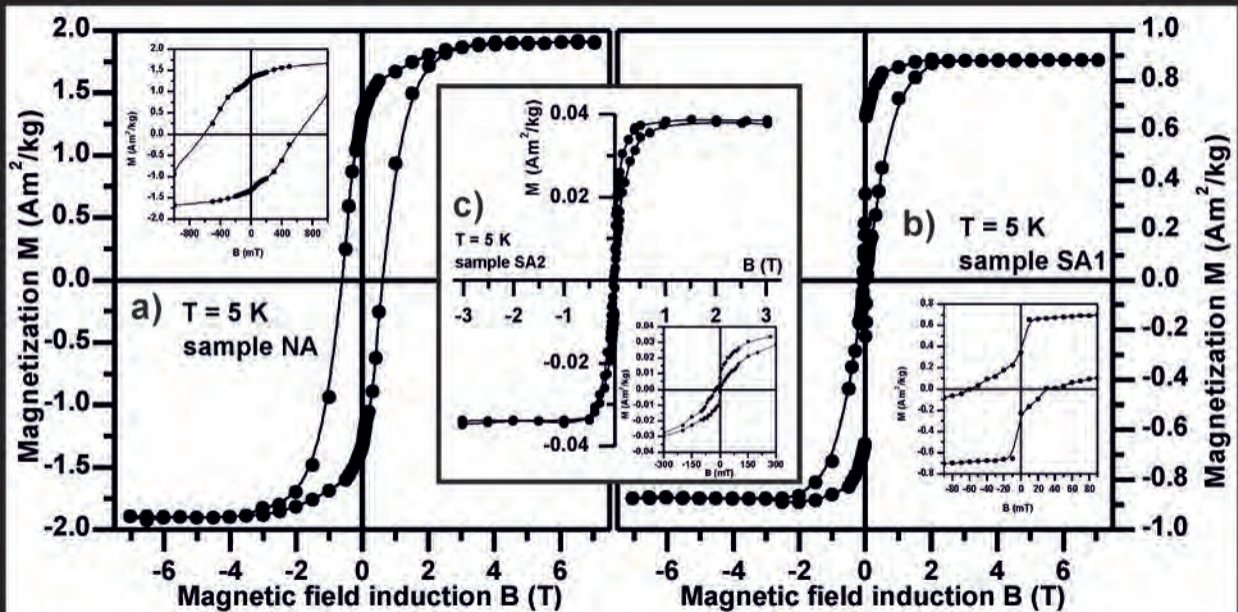
457

Diffraction intensity (a. u.)









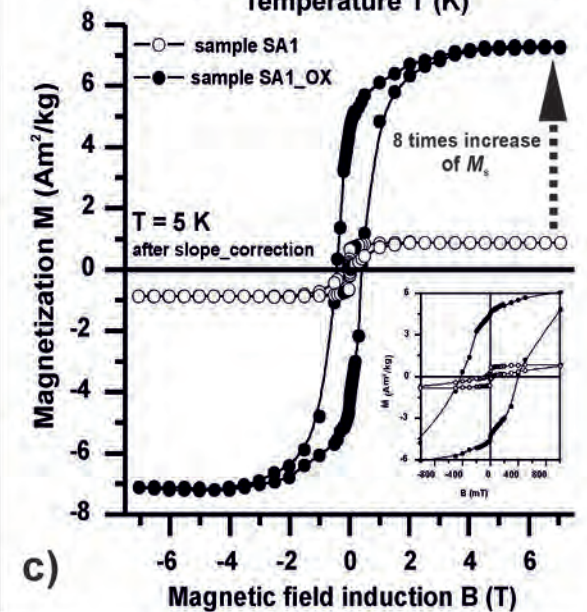
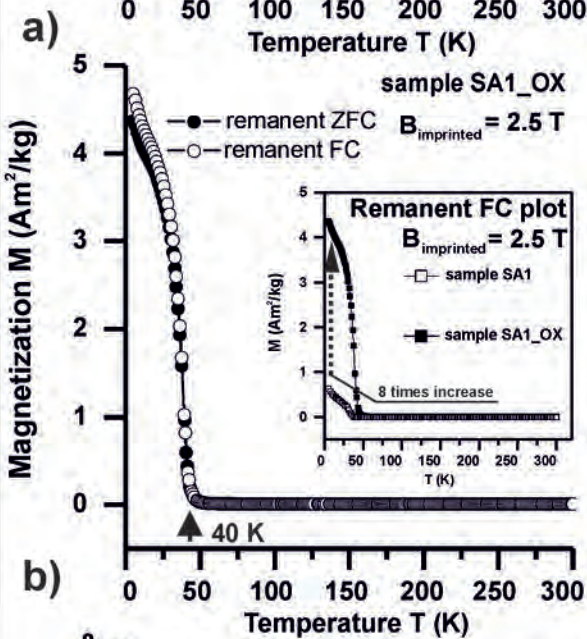
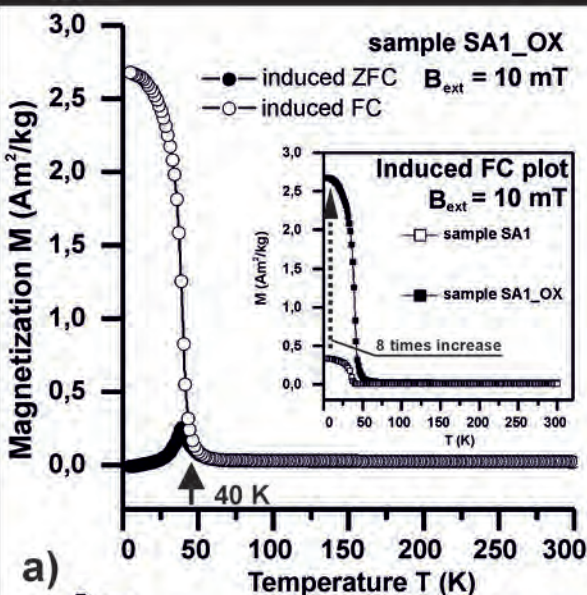


Table 1 Alabandite samples.

Sample	Description of sample preparation	Stoichiometry of alabandite	Stoichiometry by means of
NA	natural alabandite	$\text{Mn}_{1.007}\text{S}$	EMPA on polished specimen
SA1	solvothermal process	n.d.	
SA2	direct thermal synthesis from S and Mn powders	$\text{MnS}_{0.998}\text{S}$	SEM/ WDS on polished specimen
SA1_OX	oxidation product of sample SA1	n.d.	

n.d. stands for not determined values.

Table 2 Parameters of the hysteresis loops of (i) the NA, (ii) SA1, (iii) SA2 and (iv) SA1_OX at 5 K after the high-field slope has been subtracted.

Sample	T (K)	M_{E+} (Am ² /kg)	M_{E-} (Am ² /kg)	B_{C+} (mT)	B_{C-} (mT)	M_{R+} (Am ² /kg)	M_{R-} (Am ² /kg)
NA	5	1.910 ± 0.001	1.905 ± 0.001	424 ± 1	-608 ± 1	1.315 ± 0.001	-1.326 ± 0.001
SA1	5	0.882 ± 0.001	-0.874 ± 0.001	29 ± 1	-59 ± 1	0.344 ± 0.001	-0.225 ± 0.001
SA2	5	0.039 ± 0.001	-0.035 ± 0.001	4 ± 1	-24 ± 1	0.008 ± 0.001	-0.002 ± 0.001
SA1_OX	5	7.277 ± 0.001	7.235 ± 0.001	395 ± 1	-426 ± 1	4.564 ± 0.001	-4.376 ± 0.001

M_{E+} is the positive saturation magnetization, M_{E-} is the negative saturation magnetization, B_{C+} is the positive coercivity, B_{C-} is the negative coercivity, M_{R+} is the positive remanent magnetization and M_{R-} is the negative remanent magnetization.